

Research Article

Synthesis of Metal Oxide Decorated Polycarboxyphenyl Polymer-Grafted Multiwalled Carbon Nanotube Composites by a Chemical Grafting Approach for Supercapacitor Application

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We present grafting of polycarboxyphenyl polymer on the surface of multiwalled carbon nanotube (MWCNT) via a free radical polymerization and subsequent anchoring of the metal oxide nanoparticles for the evaluation of their potential applicability to supercapacitor electrodes. Here, metal oxide nanoparticles, Fe_3O_4 and Sm_2O_3 , were created after the oxidation of metal precursors $\text{Sm}(\text{NO}_3)_3$ and FeCl_2 , respectively, and attached on the surface of polycarboxyphenyl-grafted MWCNT (P-CNT) in aqueous medium. This approach shows a potential for enhancing the dispersion of Fe_3O_4 and Sm_2O_3 nanoparticles on the wall of P-CNT. The structure and morphological characteristics of the purified MWCNT, P-CNT, and metal oxide-anchored polycarboxyphenyl-grafted MWCNT (MP-CNT) nanocomposites were studied by X-ray diffraction (XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM), and thermogravimetric analysis (TGA). The electrochemical performance of the purified MWCNT electrode, P-CNT electrode, and MP-CNT electrodes was tested by cyclic voltammetry (CV) and galvanostatic charge discharge in a 1.0 M H_2SO_4 aqueous electrolyte. The results showed that the specific capacitance of the purified MWCNT was 45.3 F/g at the scan rate of 5 mV/s and increased to 54.1 F/g after the modification with polycarboxyphenyl polymer. Further modification of P-CNT with Sm_2O_3 and Fe_3O_4 improved the specific capacitance of 65.84 F/g and 173.38 F/g, respectively, at the same scan rate.

1. Introduction

With increasing the awareness for pollution control, industry and research centers around the world are searching for the possible alternatives to the development of renewable energy production for the replacement of fossil fuel [1, 2]. In this mission, supercapacitors can be an alternative to fulfill the rising demand of transferable power systems for electronics and automotive applications [3]. The two basic mechanisms for energy storage in supercapacitors are formations of electrical double layer (EDL) capacitance and pseudocapacitance [4]. The first one is a non-Faradic process in which charges are stored electrostatically via reversible ion absorption at the electrode electrolyte interface, while in

the second type transformation of Faradic charge takes place at the electrode materials. Furthermore, supercapacitors store the energy in the electric field of the electrochemical double layer (Helmholtz Layer) at the electrode : electrolyte interface [5]. The positive and negative ionic charges within the electrolyte collect at the surface of the solid electrodes and balance for the electronic charge at the electrode surface. The concentration of the electrolyte and the size of the ions determine the thickness of the double layer in the range of 5–10 Å [4].

Demand of large supercapacitors with high voltage and improved energy and power density are under discussion for different applications [3]. Because of good thermal and chemical stability, high electrical conductivity, large surface

area, porous structures, and long cycling stability of carbon nanomaterials, they are generally used as symmetric electrodes for EDL capacitors [5–7]. Furthermore, the specific surface area and functional groups of carbon nanomaterials impact the electrode performance. The electrode surface area can increase by using porous electrodes with an extremely large internal effective surface [6, 7]. Intercalation of natural graphite and subsequent microwave irradiation can increase the surface area of graphite nanoplatelets [8]. Microwave irradiation of graphite oxide [9, 10] is also an effective way to increase the surface area of graphene sheets and resulting porous material shows high capacitance. Carbon nanomaterials have been also used to improve electrical, thermal, and mechanical properties of polymer nanocomposites [11–15]. Multiwalled CNTs (MWNTs) showed limited specific capacitance, typically ranging from 10 to 100 F/g [4–7]. Certain modification of MWCNT is essential to enhance the specific capacitance and energy density of MWCNT [16, 17]. Conducting polymers (e.g., polypyrrole, polyaniline (PANI), and polythiophene) [18–21] and metal oxides (e.g., RuO_2 , Fe_2O_3 , Bi_2O_3 , MnO_2 , and Mn_3O_4) [22–32] are accepted for pseudocapacitors with relatively high capacitance. During the charge-discharge process in conducting polymers and metal oxide, the generation of positively charged sites on the polymer chains and metal oxides creates structural deterioration with inferior cycling stability which, in turn, lowers the electrical conductivity of electrode [17]. To overcome the abovementioned problems, an appropriate stoichiometric combination of metal oxides or conducting polymers with carbonaceous materials have been used for the development of efficient supercapacitors [18, 21]. Many CNT/metal oxides composites have been reported as the electrochemical electrodes, which showed remarkable capacitance, much higher than those of pure CNT or individual metal oxides due to the prevention of the nanoparticles and CNT agglomeration by the anchored metal oxide nanoparticles on the wall of CNT [25, 28, 29]. Furthermore, the electron transmission across the metal oxides can also be improved by the well conductivity of CNT [25, 28].

Generally, an activated carbon electrode shows the rectangular shaped cyclic voltammogram (CV) as an ideal capacitor. RuO_2 and IrO_2 also have an almost rectangular shaped CV and exhibit good capacitor behavior [24]. However, the shape of the CV is not an outcome of pure double layer charging but of a sequence of redox reactions occurring in the metallic oxide. Zheng and Jow [22] observed a very high specific capacitance of up to 750 F/g for RuO_2 prepared at relatively low temperatures. Kim and Park [28] reported the effects of nanosized Fe_3O_4 on supercapacitive properties of MWCNT. In their study, Fe_3O_4 nanoparticles were deposited by chemical coprecipitation of Fe^{2+} and Fe^{3+} in the presence of MWNTs in alkaline solutions in which a considerably superior specific capacitance of Fe_3O_4 dispersed MWCNT compared to that of the pure MWNT electrode was observed with retaining 85.1% of its initial capacitance over 1000 cycles. Kotal et al. [18] showed the effects on amidation of carbon nanofiber and, consequently, the grafting of PANI to obtain high capacitance and good cycling stability.

The synergistic combination of outstanding conducting properties of carbonaceous materials and high pseudocapacitance of conducting polymers and metal oxide can amalgamate for the development of high-performance supercapacitors. Though supercapacitors of carbon-based materials with conducting polymers are in their growing state [18, 21], there are not any reports that show the electrochemical properties of metal oxide decorated-conjugated polymer-grafted-carbon nanomaterial composites. In this paper, a novel route to produce Fe_2O_3 and Sm_2O_3 particles-decorated polycarboxyphenyl-grafted MWCNT composite is developed at the first time by the grafting of polycarboxyphenyl group on the wall of MWCNT and subsequent oxidation of metal oxide precursor in an autoclave. Here, grafting of polycarboxyphenyl groups on MWCNT can not only develop long conjugated structure with MWCNT, but also reduce the aggregation of MWCNT. Furthermore, the anchoring of the metal oxide nanoparticles was assisted after grafting the polycarboxyphenyl group on the surface of MWCNT. The polycarboxyphenyl-grafted MWCNT leads to the development of a π -conjugated system and such a type of conducting network may facilitate the effective transfer of the Faradic charge through the MWCNT network. Then, the effect of the covalent functionalization as well as metal oxide decoration was compared for the electrochemical performance of the composites.

2. Experimental

2.1. Materials. Acetonitrile, azobisisobutyronitrile (AIBN), 4-aminobenzoic acid, sodium nitrite (NaNO_2), hydrochloric acid, tetrabutyl ammonium tetrafluoroborate (NBu_4BF_4), ammonium sulfate, potassium persulfate, samarium nitrate, ferrous chloride, sulfuric acid, and nitric acid were of analytical reagent grade (Sigma-Aldrich) and were used as received. Pristine MWNT (CM-95) was supplied by Hanwha Nanotech Co., Ltd. (Korea). To prepare solutions for the experiments, water was purified in a Milli-Q Plus water purification system (Millipore Co. Ltd., USA).

2.2. Synthesis of Polycarboxyphenyl-Grafted MWCNT (P-CNT). Purified MWCNT was obtained after the treatment of pristine MWCNT with the mixture of concentrated sulfuric acid and nitric acid (1:1) and subsequent washing several times with deionized water. 4-Carboxyphenyl diazonium salt was prepared by diazotization as reported in our previous study [33]. Typically, the grafting of polycarboxyphenyl group on the wall of MWCNT was performed by dispersing 0.2 g of purified MWCNT in 40 mL of acetonitrile with continuous stirring for 4 h, and carboxyphenyl diazonium salt (1.0 g) and AIBN (0.15 g) were added at room temperature. The reaction mixture was stirred for 12 h at 60°C. Next, the reaction mixture was vacuum-filtered, washed with deionized water sufficiently, and finally dried in a vacuum oven at 70°C overnight. Here, the grafting of polycarboxyphenyl group on the MWCNT was controlled by adjusting the amount of diazonium salt in the synthesis medium.

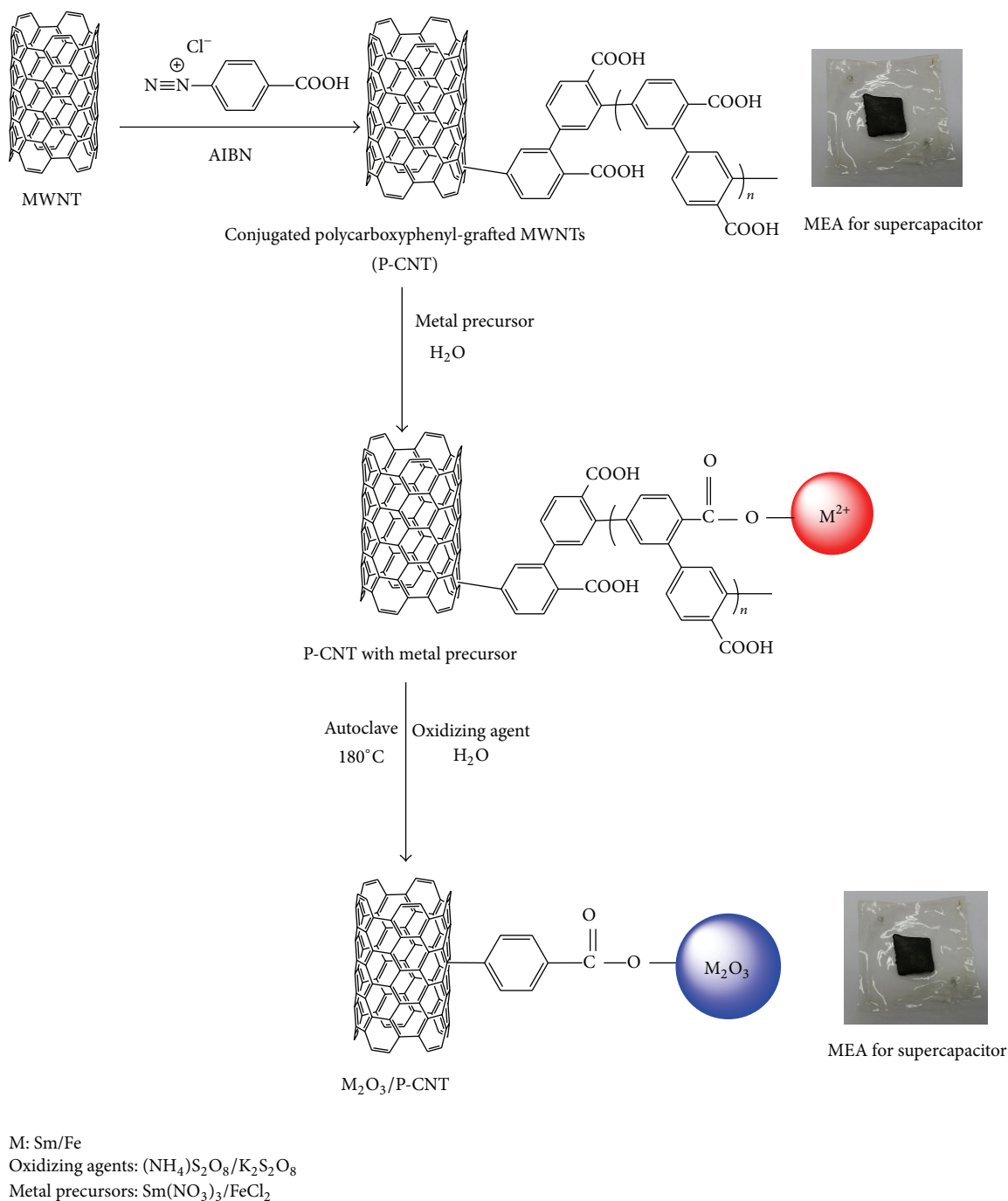


FIGURE 1: Schematic diagrams for the grafting of conjugated polycarboxyphenyl polymer on the wall of MWCNT and subsequent decoration with metal oxide nanoparticles.

2.3. *Synthesis of Metal Oxide-Anchored Polycarboxyphenyl-Grafted MWCNT (MP-CNT)*. Metal oxide nanoparticles were decorated onto the P-CNT in an autoclave using solution process. The P-CNT (0.2 g) was mixed with the metal oxide precursor (2 g), oxidizing agent (2 g), and an excess of deionized water. The oxidizing agents, $(\text{NH}_4)_2\text{S}_2\text{O}_8$ and $\text{K}_2\text{S}_2\text{O}_8$, were used for the oxidation of metal precursors, $\text{Sm}(\text{NO}_3)_3$ and FeCl_2 , respectively. The oxidation of metal precursors into metal oxides was performed in an autoclave

along with P-CNT at 180°C for 5 h, and the resulting reaction mixture was filtered, washed with deionized water, and then dried in a vacuum oven at 70°C for 24 h.

2.4. *Characterization of Purified MWCNT, P-CNT, and MP-CNT*. An X-ray diffractometer (Rigaku, Japan, with Cu KR ($\lambda = 1.540 \text{ \AA}$) radiation) was used to measure the XRD patterns of purified MWCNT, P-CNT, and MP-CNT. Thermogravimetric analysis of purified MWCNT, P-CNT,

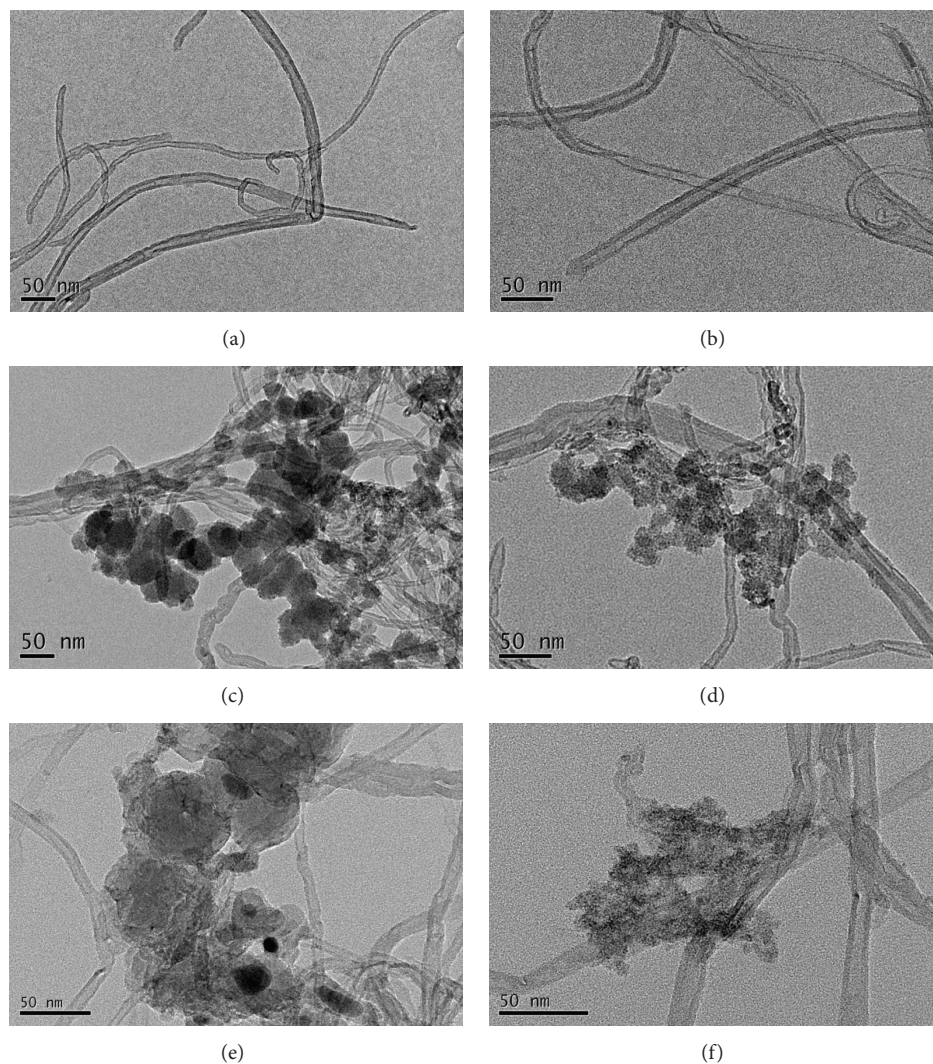


FIGURE 2: TEM images of (a) purified MWCNT, (b) P-CNT, (c) P-CNT with $\text{Sm}(\text{NO}_3)_3$, (d) Sm_2O_3 nanoparticle decorated P-CNT, (e) P-CNT with FeCl_2 , and (f) Fe_2O_3 particle decorated P-CNT.

and MP-CNT was performed using a sino TGA N-1000 with a heating rate of $10^\circ\text{C}/\text{min}$ under a nitrogen atmosphere. The morphological features of the electrode materials were observed with transmission electron microscopy (TEM, JEM-2100F, JEOL) and scanning electron microscopy (SEM, VEGA II SBH, Tescan). The sample for TEM measurement was prepared by ultrasonication of ~ 0.01 mg/mL of the electrode materials in deionized water followed by dipping the TEM grid on colloidal suspension. Powder sample without dispersion in any solvent was used for SEM observation.

2.5. Electrochemical Measurements. A conventional two-electrode configuration was used to study the electrochemical behavior of the prepared electrode materials (purified MWCNT, P-CNT, and MP-CNT) by cyclic voltammetry (CV) and galvanostatic charge-discharge (GCD) measurements. Working electrode (cathode) was prepared by mixing 90% MWCNT, P-CNT, and MP-CNT individually with 10% Nafion solution as a binder. A small amount of isopropyl

alcohol was added to obtain a more homogeneous dispersion. Then, the slurry was coated onto highly conductive hydrophobic carbon paper followed by a vacuum dry at 70°C for 24 h. The mass of the as-synthesized electrode materials coated on the electrode was about 1 mg. Activated carbon electrode was used as anode and aqueous H_2SO_4 solution (1M) was used as the electrolyte. The applied potential range for CV and GCD was kept in the limits from 0.0 to 1.00 V because the decomposition voltage of 1M aqueous H_2SO_4 electrolyte is ~ 1.4 V.

3. Results and Discussion

Our approach covers the grafting of 4-carboxyphenyl diazonium salt on the wall of purified MWCNT to give carboxyphenyl-grafted MWCNT. The reaction of diazonium salt with the carboxyphenyl-grafted MWCNT and AIBN by free radical polymerization ultimately generated polycarboxyphenyl-grafted MWCNT (P-CNT). The carboxyl

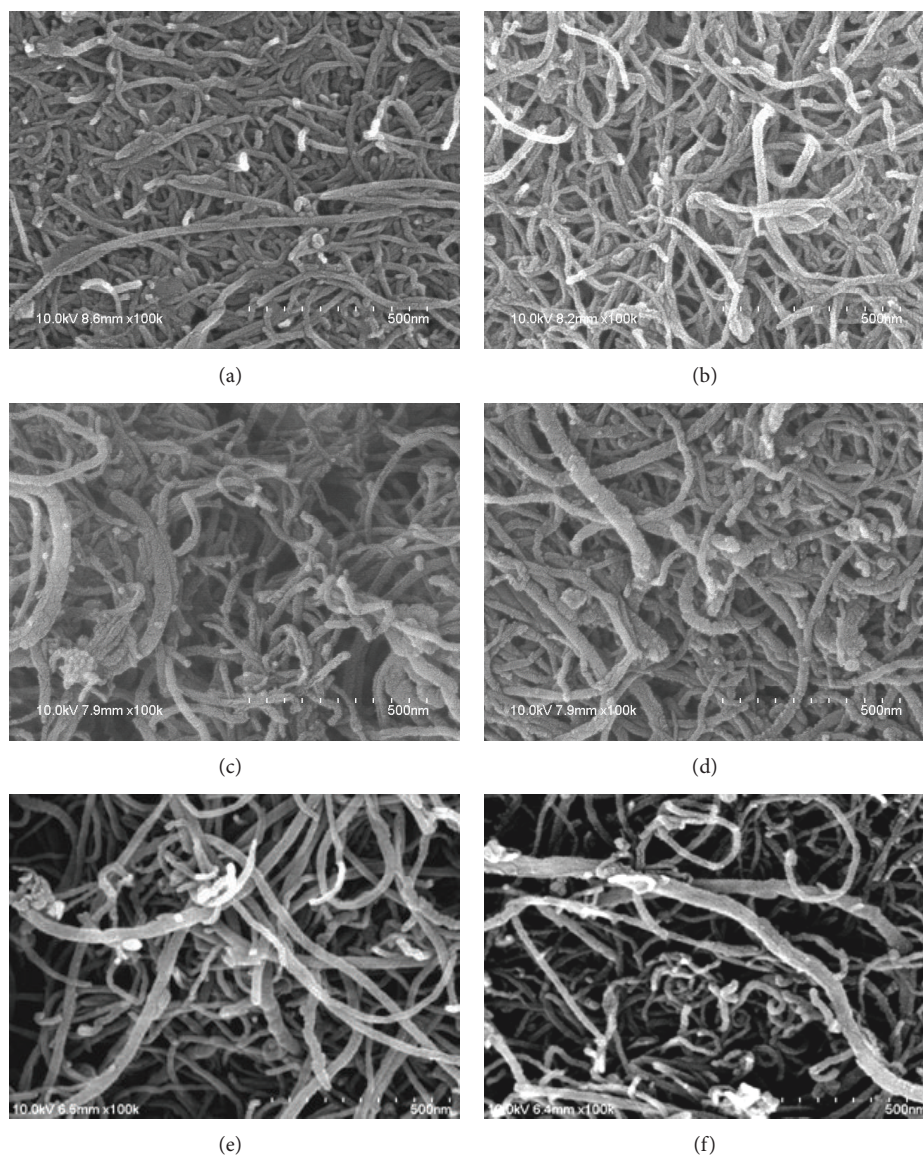


FIGURE 3: SEM images of (a) purified MWCNT, (b) P-CNT, (c) P-CNT with $\text{Sm}(\text{NO}_3)_3$, (d) Sm_2O_3 nanoparticle decorated P-CNT, (e) P-CNT with FeCl_2 , and (f) Fe_2O_3 particle decorated P-CNT.

group on the diazonium salt also assisted in the polymerization, leading to the development of a π -conjugated system in P-CNT. Then, the P-CNT was subsequently reacted with the excess of metal precursors separately to produce a complex derivative on its surface, followed by oxidation of the metal precursor in the presence of oxidizing agent to form a MP-CNT composite (Figure 1).

The morphologies of the surfaces, defects, and surface irregularities of purified MWCNT, P-CNT, and MP-CNT are the most important factors for evaluating the electrochemical properties. In this study, TEM and SEM were used to provide reliable and visual measurements about the surface morphologies of the electrode materials. Figure 2 shows the TEM images of the (a) purified MWCNT, (b) P-CNT, (c) P-CNT with $\text{Sm}(\text{NO}_3)_3$, (d) Sm_2O_3 /P-CNT, (e) P-CNT with FeCl_2 , and (f) Fe_2O_3 /P-CNT. The sp^2 hybridization state of carbon in MWCNT was not changed due to the grafting

of conjugated polycarboxyphenyl polymer. Obviously, the diameter of P-CNT was observed greater in Figure 2(b) than the purified MWCNT in Figure 2(a) as a proof of the grafting of polycarboxyphenyl polymer. In our previous study, we have already reported the mechanism of the formation of polycarboxyphenyl-grafted MWCNT with plausible structures by the grafting of diazonium compound for the development of enzyme-free biosensors [33]. The morphological feature of the P-CNT with metal precursor was changed after the treatment with oxidizing agent. Here, the fine distribution of Fe_2O_3 nanoparticles was observed compared to Sm_2O_3 nanoparticles on the surface of P-CNT.

Figure 3 displays the SEM images of (a) purified MWCNT, (b) P-CNT, (c) P-CNT with $\text{Sm}(\text{NO}_3)_3$, (d) Sm_2O_3 /P-CNT, (e) P-CNT with FeCl_2 , and (f) Fe_2O_3 /P-CNT. Similar to TEM images, grafting of polycarboxyphenyl polymer on MWCNT increased the diameter of MWCNT. Relatively,

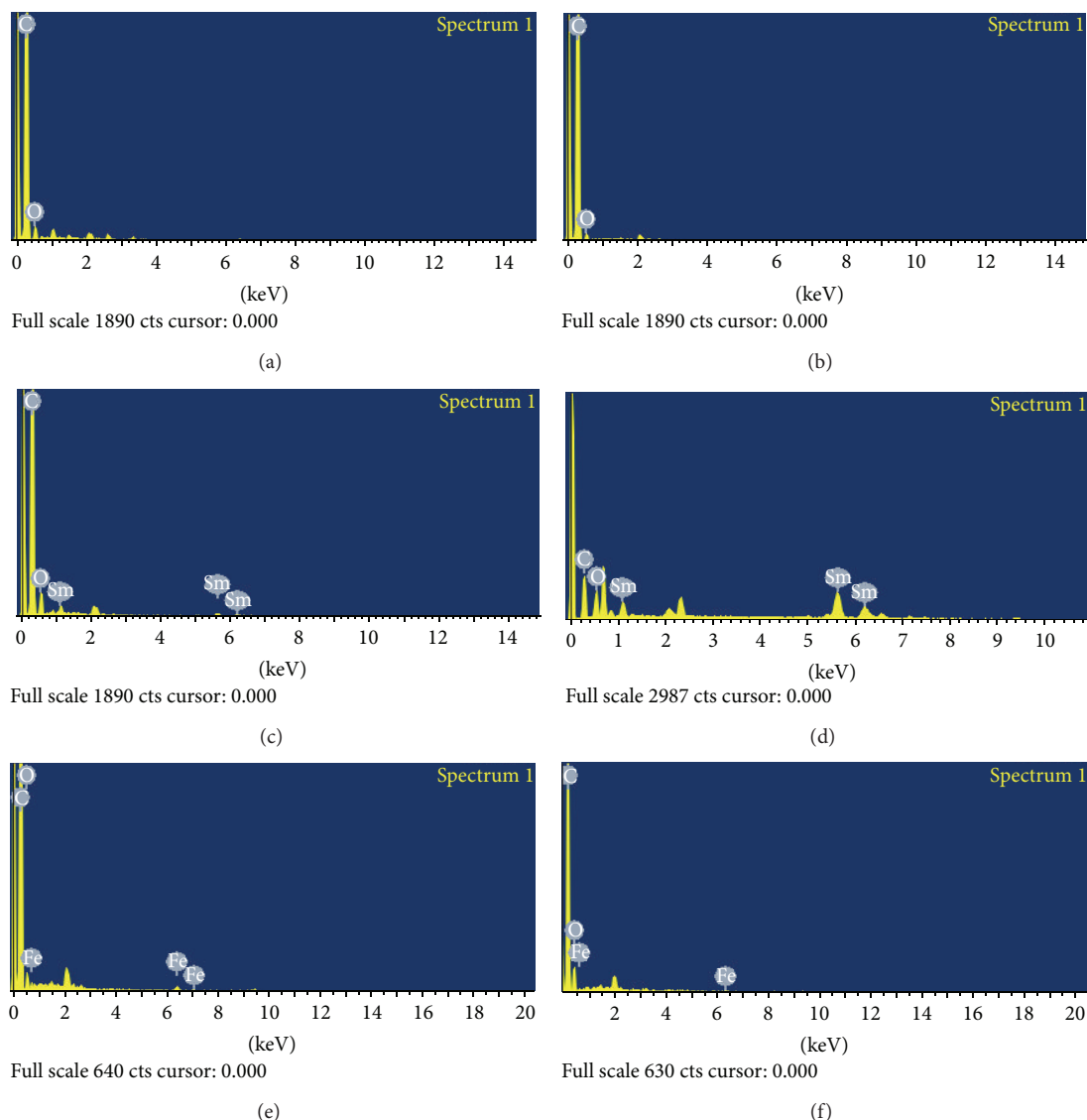


FIGURE 4: EDX spectra of (a) purified MWCNT, (b) P-CNT, (c) P-CNT with $\text{Sm}(\text{NO}_3)_3$, (d) Sm_2O_3 nanoparticle decorated P-CNT, (e) P-CNT with FeCl_2 , and (f) Fe_2O_3 nanoparticle decorated P-CNT.

dense decoration of metal oxide nanoparticles on P-CNT was observed in $\text{Sm}_2\text{O}_3/\text{P-CNT}$ compared to $\text{Fe}_2\text{O}_3/\text{P-CNT}$ composite. Energy dispersive X-ray (EDX) analysis in Figure 4 confirms that only carbon and oxygen are present in the analysis of (a) purified MWCNT and (b) P-CNT. After the reaction with metal precursors, Fe and Sm were also observed in Figures 4(c) and 4(e), respectively. The calculated atomic ratios of Sm to O in Figure 4(d) and Fe to O in Figure 4(f) are close to 2:3, which agree well with the stoichiometric ratio of Fe_2O_3 and Sm_2O_3 . Figure 5(a) presents the X-ray diffraction patterns of (A) purified MWCNT, (B) P-CNT, (C) P-CNT with $\text{Sm}(\text{NO}_3)_3$, and (D) $\text{Sm}_2\text{O}_3/\text{P-CNT}$. The diffraction peak of MWCNT at $2\theta = 26.1^\circ$ and 43.3° was observed corresponding to the 002 and 100, respectively. The $\text{Sm}_2\text{O}_3/\text{P-CNT}$ shows diffraction peaks at 222, 321, 024, 332, 422, 125, and 400 as the indication of the formation of Sm_2O_3 nanoparticle [34] on the surface

of MWCNT. Figure 5(b) shows X-ray diffraction patterns of (A) purified MWCNT, (B) P-CNT, (C) P-CNT with FeCl_2 , and (D) Fe_2O_3 nanoparticles decorated P-CNT. Except for the characteristic diffraction peak of MWCNT, we could not observe any additional peak in P-CNT with FeCl_2 , but P-CNT with $\text{Sm}(\text{NO}_3)_3$ showed sharp diffraction peak at 102. Probably, the difference in the interaction of different metal precursors ($\text{Sm}(\text{NO}_3)_3$ and FeCl_2) with P-CNT was responsible for showing such type of variation on the X-ray diffraction peaks. Furthermore, $\text{Fe}_2\text{O}_3/\text{P-CNT}$ composite shows the diffraction peaks of Fe_2O_3 at 220, 311, 222, and 400 [35]. Here, the broader size of the diffractive peaks of Fe_2O_3 indicates that the crystalline size of Fe_2O_3 particles is quite small.

Figure 6(a) shows TGA thermograms of (A) purified MWCNT, (B) P-CNT, (C) P-CNT with $\text{Sm}(\text{NO}_3)_3$, and (D) $\text{Sm}_2\text{O}_3/\text{P-CNT}$. Except for $\sim 3\%$ mass loss below 60°C ,

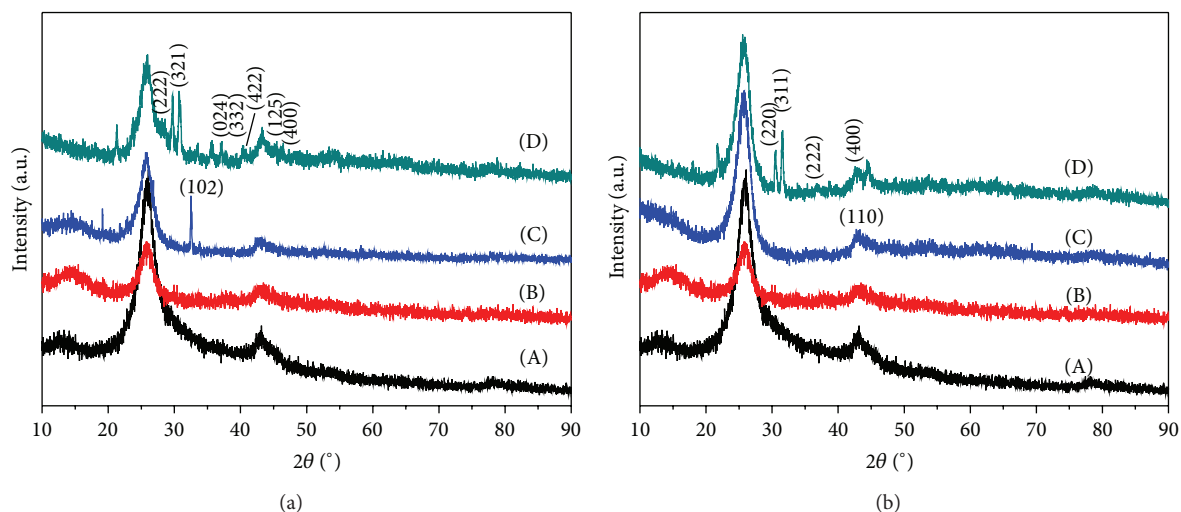


FIGURE 5: (a) The XRD patterns of (A) purified MWCNT, (B) P-CNT, (C) P-CNT with $\text{Sm}(\text{NO}_3)_3$, and (D) Sm_2O_3 nanoparticles decorated P-CNT. (b) The XRD patterns of (A) purified MWCNT, (B) P-CNT, (C) P-CNT with FeCl_2 , and (D) Fe_2O_3 nanoparticles decorated P-CNT.

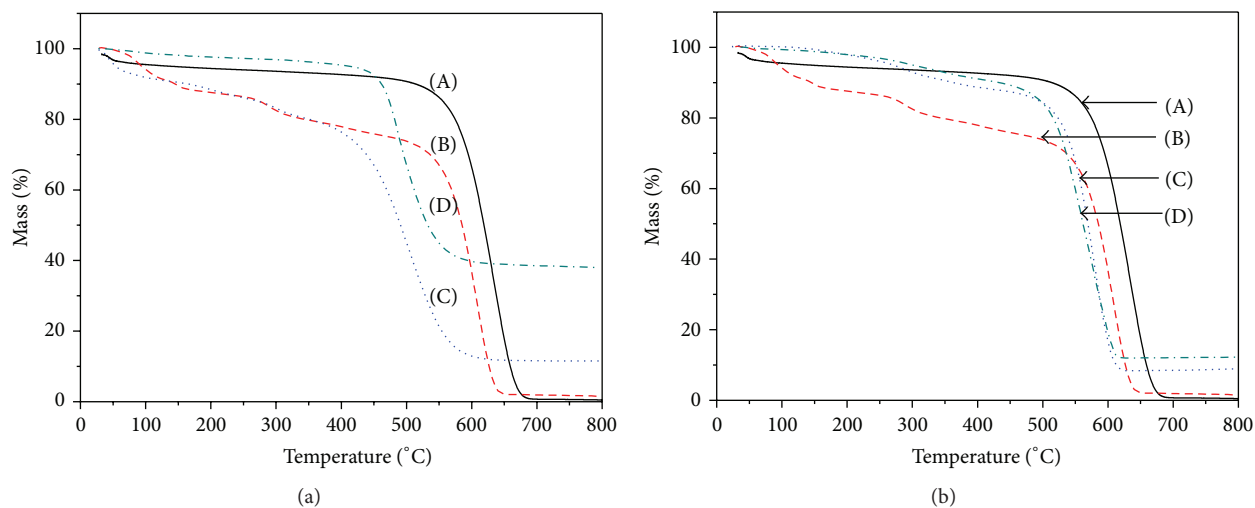


FIGURE 6: (a) TGA thermograms of (A) purified MWCNT, (B) P-CNT, (C) P-CNT with $\text{Sm}(\text{NO}_3)_3$, and (D) Sm_2O_3 nanoparticles decorated P-CNT. (b) TGA thermograms of (A) purified MWCNT, (B) P-CNT, (C) P-CNT with FeCl_2 , and (D) Fe_2O_3 nanoparticles decorated P-CNT.

purified MWCNT did not lose major mass before $\sim 500^\circ\text{C}$, whereas P-CNT lost 16.3% mass in the two steps between 110 and 400°C and underwent a major mass loss ($\sim 65\text{ wt}\%$) at $\sim 500^\circ\text{C}$. The mass loss of P-CNT below 400°C occurred due to the degradation of grafted polymer in MWCNT. Probably, the degradation of carboxyl group happened at the earlier stage and conjugated benzene rings at the later stage. Here, lower thermal stability of P-CNT than purified MWCNT was due to the grafting of long polymer chain on the wall of nanotube. Figure 6(b) presents TGA thermograms of (A) purified MWCNT, (B) P-CNT, (C) P-CNT with FeCl_2 , and (D) $\text{Fe}_2\text{O}_3/\text{P-CNT}$. When we compared the TGA thermograms of MWCNT with metal precursor and metal oxide nanoparticles in Figures 6(a) and 6(b), obviously thermal stability of the metal oxide particle decorated P-CNT was higher than P-CNT with metal precursor. Interestingly, the residue

after TGA of $\text{Sm}_2\text{O}_3/\text{P-CNT}$ and $\text{Fe}_2\text{O}_3/\text{P-CNT}$ was observed to be $\sim 40\%$ and $\sim 12\%$, respectively. The dense decoration of Sm_2O_3 nanoparticles than Fe_2O_3 on P-CNT was responsible for obtaining different nature of the TGA thermograms of $\text{Sm}_2\text{O}_3/\text{P-CNT}$ and $\text{Fe}_2\text{O}_3/\text{P-CNT}$ and that was also supported by the FESEM images in Figures 3(d) and 3(f).

The electrochemical performances of the purified MWCNT, P-CNT, $\text{Sm}_2\text{O}_3/\text{P-CNT}$, and $\text{Fe}_2\text{O}_3/\text{P-CNT}$ were analyzed using CV and GCD measurements. Figure 7(a) shows CV curves obtained in two-electrode cell for (A) purified MWCNT, (B) P-CNT, (C) $\text{Sm}_2\text{O}_3/\text{P-CNT}$, and (D) $\text{Fe}_2\text{O}_3/\text{P-CNT}$ electrodes at a voltage scan rate of 5 mV/s in $1.0\text{ M H}_2\text{SO}_4$ electrolyte in potential ranges of 0.0 to 1.0 V . From the above CV curves, the difference in the electrochemical surface activity of all electrode materials can easily be identified. It is noted that the area of the CV

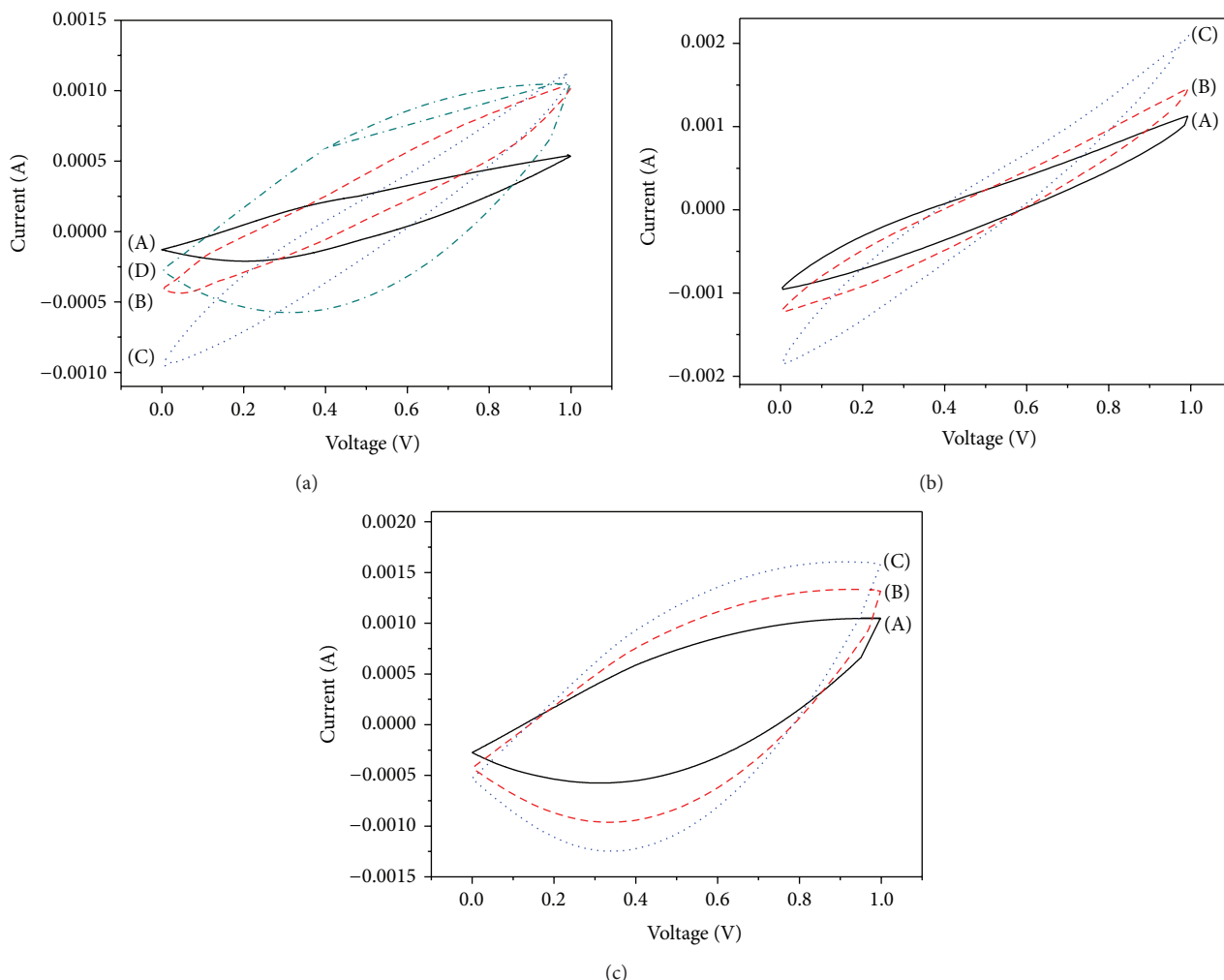


FIGURE 7: (a) Cyclic voltammograms of (A) purified MWNT, (B) P-CNT, (C) $\text{Sm}_2\text{O}_3/\text{P-CNT}$, and (D) $\text{Fe}_2\text{O}_3/\text{P-CNT}$ electrodes at a scan rate of 5 mV s^{-1} in $1.0 \text{ M H}_2\text{SO}_4$ electrolyte. (b) Cyclic voltammograms of $\text{Sm}_2\text{O}_3/\text{P-CNT}$ electrodes measured in $1.0 \text{ M H}_2\text{SO}_4$ electrolyte at the scan rates of (A) 5 mV s^{-1} , (B) 10 mV s^{-1} , and (C) 15 mV s^{-1} . (c) Cyclic voltammograms of $\text{Fe}_2\text{O}_3/\text{P-CNT}$ electrodes measured in $1.0 \text{ M H}_2\text{SO}_4$ electrolyte at the scan rates of (A) 5 mV s^{-1} , (B) 10 mV s^{-1} , and (C) 15 mV s^{-1} .

curves of P-CNT is larger than that of purified MWCNT, indicating a better capacitive response of P-CNT. The grafting of conjugated polycarboxyphenyl group on MWCNT was useful for enhancing the capacitance performance. Significant synergistic effect of polycarboxyphenyl polymer and metal oxide was observed in metal oxide decorated P-CNT. Interestingly, Fe_2O_3 decorated P-CNT showed the largest capacity among all of the electrode materials without changing the shape of the CV profiles. Furthermore, P-CNT exhibited a much lower area of the CV curve than that of Fe_2O_3 and Sm_2O_3 decorated P-CNT, indicating the better capacitance performance of the metal decorated P-CNT nanocomposite. Moreover, the intimate interaction or strong charge coupling of Fe_2O_3 with P-CNT arising from the chemical grafting provides a synergistic effect and results in enhanced accessibility of the electrolyte by every Fe_2O_3 nanoparticle, leading to improving capacitance of the $\text{Fe}_2\text{O}_3/\text{P-CNT}$ composites. On the contrary, dense

decoration of Sm_2O_3 on the P-CNT results in agglomeration of metal particles and also decreases the effective surface area of $\text{Sm}_2\text{O}_3/\text{P-CNT}$ nanocomposite. As a result, the $\text{Sm}_2\text{O}_3/\text{P-CNT}$ nanocomposite was the low specific capacitance. In this study, we could not observe any redox peaks in P-CNT and MP-CNT probably due to the sequence of redox reactions occurring in the conjugated polymer and metallic oxide decorated MWCNT nanocomposite [3, 25]. The area of the CV curves of $\text{Sm}_2\text{O}_3/\text{P-CNT}$ and $\text{Fe}_2\text{O}_3/\text{P-CNT}$ nanocomposites increased with increasing the scan rate from 5 mV/s to 15 mV/s as shown in Figures 7(b) and 7(c). Table 1 shows the specific capacitance of the purified MWCNT, P-CNT, $\text{Sm}_2\text{O}_3/\text{P-CNT}$, and $\text{Fe}_2\text{O}_3/\text{P-CNT}$ electrodes from cyclic voltammograms at a scan rate of 5 mV/s . A Fe_2O_3 decorated P-CNT composite electrode exhibited the highest specific capacitance (173.3 F g^{-1}), almost 2.8 times higher than that of purified MWCNT electrode and 2.2 times higher than that of P-CNT electrode, indicating a significant

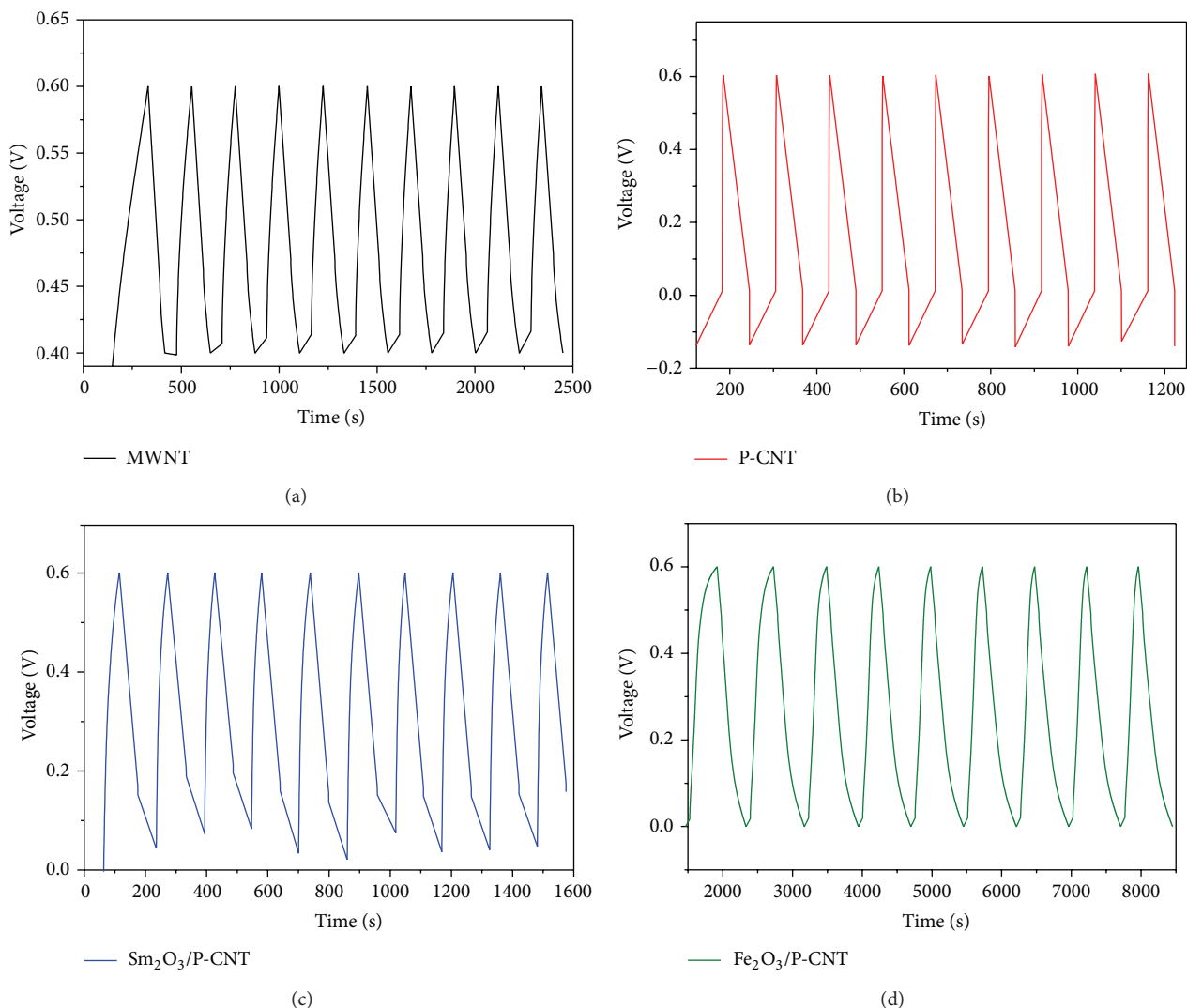


FIGURE 8: Galvanostatic charge discharge of (a) MWNT, (b) P-CNT (b), (c) $\text{Sm}_2\text{O}_3/\text{P-CNT}$, and (d) $\text{Fe}_2\text{O}_3/\text{MWNT}$ at an applied constant current of 0.3 mA in 1.0 M H_2SO_4 aqueous electrolyte.

TABLE 1: Specific capacitance (F/g) of the purified MWNT, P-CNT, $\text{Sm}_2\text{O}_3/\text{P-CNT}$, and $\text{Fe}_2\text{O}_3/\text{P-CNT}$ electrodes from cyclic voltammograms at the scan rate of 5 mV s^{-1} in 1.0 M H_2SO_4 electrolyte.

	Purified MWNT	P-CNT	$\text{Sm}_2\text{O}_3/\text{P-CNT}$	$\text{Fe}_2\text{O}_3/\text{P-CNT}$
Specific capacitance (F g^{-1})	45.36	54.1	65.84	173.38

synergistic effect of conjugated polymer and Fe_2O_3 . The specific capacitance of the $\text{Fe}_2\text{O}_3/\text{P-CNT}$ composites was superior to that of $\text{Sm}_2\text{O}_3/\text{P-CNT}$, probably due to the different nature of metal oxides and their different interaction properties with P-CNT.

Figure 8 shows the GCD plots of (a) purified MWCNT, (b) P-CNT, (c) $\text{Sm}_2\text{O}_3/\text{P-CNT}$, and (d) $\text{Fe}_2\text{O}_3/\text{P-CNT}$ electrodes at an applied constant current of 0.3 mA. It can be

seen that the charge-discharge curves of the purified MWNTs were very symmetrical and had a triangular shape as shown in Figure 8(a). This implies that the purified MWNT electrodes clearly had electrochemical double layer capacitive behavior [4, 16]. The P-CNT, $\text{Sm}_2\text{O}_3/\text{P-CNT}$, and $\text{Fe}_3\text{O}_4/\text{P-CNT}$ composite electrodes also showed reversible charge-discharge process of the electrode. Compared to the other electrode materials, $\text{Fe}_3\text{O}_4/\text{P-CNT}$ composite electrode showed the longest discharge time ascribed to the correct quantity of Fe_2O_3 contributing reversible redox transitions relating the exchange of protons and/or cations with the electrolyte.

4. Conclusions

The grafting of polycarboxyphenyl group on MWCNT and decoration with Fe_2O_3 and Sm_2O_3 nanoparticles separately presented herein offer a promising and facile approach to prepare a new class of electrode materials. The TEM, SEM,

EDX, TGA, and X-ray diffraction of electrode materials provide the evidences for the grafting of polycarboxyphenyl group on the wall of MWCNT and subsequent metal oxide nanoparticles decoration. Electrochemical test of the purified MWCNT, P-CNT, $\text{Sm}_2\text{O}_3/\text{P-CNT}$, and $\text{Fe}_2\text{O}_3/\text{P-CNT}$ electrodes from cyclic voltammograms at a scan rate of 5 mV/s shows specific capacitance of 45.3 F/g, 54.1 F/g, 65.84 F/g, and 173.3 F/g, respectively. The remarkably highest specific capacitance of $\text{Fe}_2\text{O}_3/\text{P-CNT}$ composite electrode compared to other electrodes is due to the grafting of the conjugated polycarboxyphenyl polymer on MWCNTs in the development of a π -conjugated structure and proper decoration of P-CNT by Fe_2O_3 nanoparticles for the charge transfer facilitation.

Conflict of Interests

The authors declare that there is no conflict of interests regarding the publication of this paper.

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References

- [1] P. G. Bruce, S. A. Freunberger, L. J. Hardwick, and J.-M. Tarascon, "Li-O₂ and Li-S batteries with high energy storage," *Nature Materials*, vol. 11, pp. 19–29, 2012.
- [2] B. Dunn, H. Kamath, and J. M. Tarascon, "Electrical energy storage for the grid: a battery of choices," *Science*, vol. 334, no. 6058, pp. 928–935, 2011.
- [3] L. L. Zhang and X. S. Zhao, "Carbon-based materials as supercapacitor electrodes," *Chemical Society Reviews*, vol. 38, no. 9, pp. 2520–2531, 2009.
- [4] R. Kötz and M. Carlen, "Principles and applications of electrochemical capacitors," *Electrochimica Acta*, vol. 45, no. 15-16, pp. 2483–2498, 2000.
- [5] J.-S. Ye, X. Liu, H. F. Cui, W.-D. Zhang, F.-S. Sheu, and T. M. Lim, "Electrochemical oxidation of multi-walled carbon nanotubes and its application to electrochemical double layer capacitors," *Electrochemistry Communications*, vol. 7, no. 3, pp. 249–255, 2005.
- [6] J. Chmiola, G. Yushin, Y. Gogotsi, C. Portet, P. Simon, and P. L. Taberna, "Anomalous increase in carbon at pore sizes less than 1 nanometer," *Science*, vol. 313, no. 5794, pp. 1760–1763, 2006.
- [7] Y. Zhai, Y. Dou, D. Zhao, P. F. Fulvio, R. T. Mayes, and S. Dai, "Carbon materials for chemical capacitive energy storage," *Advanced Materials*, vol. 23, no. 42, pp. 4828–4850, 2011.
- [8] Q.-T. Truong, P. Pokharel, G. S. Song, and D.-S. Lee, "Preparation and characterization of graphene nanoplatelets from natural graphite via intercalation and exfoliation with tetraalkylammoniumbromide," *Journal of Nanoscience and Nanotechnology*, vol. 12, no. 5, pp. 4305–4308, 2012.
- [9] P. Pokharel, Q.-T. Truong, and D. S. Lee, "Multi-step microwave reduction of graphite oxide and its use in the formation of electrically conductive graphene/epoxy composites," *Composites Part B: Engineering*, vol. 64, pp. 187–193, 2014.
- [10] Y. Zhu, S. Murali, M. D. Stoller, A. Velamakanni, R. D. Piner, and R. S. Ruoff, "Microwave assisted exfoliation and reduction of graphite oxide for ultracapacitors," *Carbon*, vol. 48, no. 7, pp. 2118–2122, 2010.
- [11] P. Pokharel and D. S. Lee, "Thermal and mechanical properties of reduced graphene oxide/polyurethane nanocomposite," *Journal of Nanoscience and Nanotechnology*, vol. 14, no. 8, pp. 5718–5721, 2014.
- [12] P. Pokharel, H. Bae, J.-G. Lim, K. Y. Lee, and S. Choi, "Effects of titanate treatment on morphology and mechanical properties of graphene nanoplatelets/high density polyethylene nanocomposites," *Journal of Applied Polymer Science*, vol. 132, no. 23, 2015.
- [13] P. Pokharel and D. S. Lee, "High performance polyurethane nanocomposite films prepared from a masterbatch of graphene oxide in polyether polyol," *Chemical Engineering Journal*, vol. 253, pp. 356–365, 2014.
- [14] P. Pokharel, S. Choi, and D. S. Lee, "The effect of hard segment length on the thermal and mechanical properties of polyurethane/graphene oxide nanocomposites," *Composites Part A: Applied Science and Manufacturing*, vol. 69, pp. 168–177, 2015.
- [15] P. Pokharel, S. H. Lee, and D. S. Lee, "Thermal, mechanical, and electrical properties of graphene nanoplatelet/graphene oxide/polyurethane hybrid nanocomposite," *Journal of Nanoscience and Nanotechnology*, vol. 15, no. 1, pp. 211–214, 2015.
- [16] A. G. Pandolfo and A. F. Hollenkamp, "Carbon properties and their role in supercapacitors," *Power Sources*, vol. 157, no. 1, pp. 11–27, 2006.
- [17] B. E. Conway, *Electrochemical Supercapacitors: Scientific Fundamentals and Technological Applications*, Kluwer-Plenum, New York, NY, USA, 1999.
- [18] M. Kotal, A. K. Thakur, and A. K. Bhowmick, "Polyaniline-carbon nanofiber composite by a chemical grafting approach and its supercapacitor application," *ACS Applied Materials and Interfaces*, vol. 5, no. 17, pp. 8374–8386, 2013.
- [19] Y. Huang, J. Taob, W. Menga et al., "Super-high rate stretchable polypyrrole-based supercapacitors with excellent cycling stability," *Nano Energy*, vol. 11, pp. 518–525, 2015.
- [20] Q.-F. Wu, K.-X. He, H.-Y. Mi, and X.-G. Zhang, "Electrochemical capacitance of polypyrrole nanowire prepared by using cetyltrimethylammonium bromide (CTAB) as soft template," *Materials Chemistry and Physics*, vol. 101, no. 2-3, pp. 367–371, 2007.
- [21] E. Hür, G. A. Varol, and A. Arslan, "The study of polythiophene, poly(3-methylthiophene) and poly(3,4-ethylenedioxythiophene) on pencil graphite electrode as an electrode active material for supercapacitor applications," *Synthetic Metals*, vol. 184, pp. 16–22, 2013.
- [22] J. P. Zheng and T. R. Jow, "High energy and high power density electrochemical capacitors," *Journal of Power Sources*, vol. 62, no. 2, pp. 155–159, 1996.
- [23] C.-C. Hu, K.-H. Chang, M.-C. Lin, and Y.-T. Wu, "Design and tailoring of the nanotubular arrayed architecture of hydrous

- RuO₂ for next generation supercapacitors,” *Nano Letters*, vol. 6, no. 12, pp. 2690–2695, 2006.
- [24] C. C. Hu, K. H. Chang, M. C. Lin, and Y. T. Wu, “Design and tailoring of the nanotubular arrayed architecture of hydrous RuO₂ for next generation supercapacitors,” *Nano Letters*, vol. 6, no. 12, pp. 2690–2695, 2006.
- [25] A. L. M. Reddy and S. Ramaprabhu, “Nanocrystalline metal oxides dispersed multiwalled carbon nanotubes as supercapacitor electrodes,” *The Journal of Physical Chemistry C*, vol. 111, no. 21, pp. 7727–7734, 2007.
- [26] J. Chen, K. Huang, and S. Liu, “Hydrothermal preparation of octadecahedron Fe₃O₄ thin film for use in an electrochemical supercapacitor,” *Electrochimica Acta*, vol. 55, no. 1, pp. 1–5, 2009.
- [27] D. Xuan, W. Chengyang, C. Mingming, J. Yang, and W. Jin, “Electrochemical performances of nanoparticle Fe₃O₄/activated carbon supercapacitor using KOH electrolyte solution,” *Journal of Physical Chemistry C*, vol. 113, no. 6, pp. 2643–2646, 2009.
- [28] Y.-H. Kim and S.-J. Park, “Roles of nanosized Fe₃O₄ on supercapacitive properties of carbon nanotubes,” *Current Applied Physics*, vol. 11, no. 3, pp. 462–466, 2011.
- [29] L. Fang, B. Zhang, W. Li, J. Zhang, K. Huang, and Q. Zhang, “Fabrication of highly dispersed ZnO nanoparticles embedded in graphene nanosheets for high performance supercapacitors,” *Electrochimica Acta*, vol. 148, pp. 164–169, 2014.
- [30] F.-L. Zheng, G.-R. Li, Y.-N. Ou, Z.-L. Wang, C.-Y. Su, and Y.-X. Tong, “Synthesis of hierarchical rippled Bi₂O₃ nanobelts for supercapacitor applications,” *Chemical Communications*, vol. 46, no. 27, pp. 5021–5023, 2010.
- [31] G. X. Wang, B. L. Zhang, Z. L. Yu, and M. Z. Qu, “Manganese oxide/MWNTs composite electrodes for supercapacitors,” *Solid State Ionics*, vol. 176, no. 11–12, pp. 1169–1174, 2005.
- [32] R. Liu, J. Duay, and S. B. Lee, “Heterogeneous nanostructured electrode materials for electrochemical energy storage,” *Chemical Communications*, vol. 47, no. 5, pp. 1384–1404, 2011.
- [33] D.-J. Chunga, S.-H. Ohb, S. Komathic, A. I. Gopalanc, K.-P. Lee, and S.-H. Choi, “One-step modification of various electrode surfaces using diazonium salt compounds and the application of this technology to electrochemical DNA (E-DNA) sensors,” *Electrochimica Acta*, vol. 76, pp. 394–403, 2012.
- [34] B. Renganathan, D. Sastikumar, R. Srinivasan, and A. R. Ganesan, “Nanocrystalline samarium oxide coated fiber optic gas sensor,” *Materials Science and Engineering B: Solid-State Materials for Advanced Technology*, vol. 186, no. 1, pp. 122–127, 2014.
- [35] L.-F. Chen, Z.-Y. Yu, X. Ma, Z.-Y. Li, and S.-H. Yu, “In situ hydrothermal growth of ferric oxides on carbon cloth for low-cost and scalable high-energy-density supercapacitors,” *Nano Energy*, vol. 9, pp. 345–354, 2014.



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