

## Graphene-manganese oxide hybrid porous material and its application in carbon dioxide adsorption

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Graphene-Mn<sub>3</sub>O<sub>4</sub> (GMNO) hybrid porous material is prepared by a hydrothermal method and its performance in carbon dioxide adsorption is investigated. In the synthesis of the GMNO materials, MnO(OH)<sub>2</sub> colloid obtained by the hydrolysis of Mn<sup>2+</sup> in basic solution was using as the precursor of the Mn<sub>3</sub>O<sub>4</sub>. After a hydrothermal reaction of the mixture of graphene oxide (GO) and MnO(OH)<sub>2</sub>, GO was reduced into graphene and the MnO(OH)<sub>2</sub> was transformed into Mn<sub>3</sub>O<sub>4</sub> with enhanced crystallization. X-ray diffraction, thermal gravimetric analysis, transmission electron microscopy, infrared spectra and Raman spectroscopy were taken to characterize the hybrid material. The porosity and the carbon dioxide adsorption ability are measured by gas sorption analysis, in which the as-prepared GMNO hybrid materials exhibit a specific surface area ranging from 140 to 680 m<sup>2</sup> g<sup>-1</sup> and a maximum carbon dioxide capacity of about 11 wt%.

**graphene, manganese oxide, porous material, carbon dioxide adsorption**

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The greenhouse gas, carbon dioxide, is the leading contributor of the global warming and climate change [1]. In the purpose of sustainability, scientists are devoting to find ways to control the carbon dioxide emission and separate/adsorb the greenhouse gas. For the latter one, there are four main approaches: cryogenic distillation, membrane purification, absorption with liquids [2], and adsorption using solids [3,4]. Using solid adsorbent is widely considered, such as zeolites [5,6], activated carbons [7,8], metal oxides [9,10], and supported amines [11].

Considering the acidic nature of carbon dioxide, metal oxides which offer various basic sites are investigated widely as adsorbent. Many kinds of metal oxides display well in the performance of carbon dioxide adsorption, such as calcium oxide [9] and magnesium oxide [10]. Besides the basic sites contained in the solid adsorbent, the surface area is also an important factor. Porous metal oxide materials are

synthesized and show high carbon dioxide uptake volume [12]. Meanwhile, metal oxide nanoparticles loading on or incorporating in the supporting material also show a significant increase in the surface area and gas adsorption ability [13].

In recent years, various kinds of graphene-inorganic hybrid materials are fabricated owing to the superior properties of graphene, i.e., the large theoretical specific surface area (about 2600 m<sup>2</sup> g<sup>-1</sup>) and superior electronic and thermal properties [14,15]. Inorganic nanoparticles, i.e., metal [16] and metal oxides [17] have been successfully incorporated between graphene sheets, such materials have been investigated in a wide variety of applications in the fields of catalysis [18], gas sorption [13], and electrode materials [19]. Meanwhile, nanoparticles incorporated between graphene sheets effectively prevent the aggregation between graphene sheets in the hybrid system of the graphene-nanoparticle composites [20,21], and the high porosity would increase their performance as an adsorbent and in

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other applications.

Manganese oxide ( $\text{MnO}_x$ ) is a kind of low-cost and environmentally benign metal oxides with a high electrochemical activity and an environmental compatibility [22]. Graphene-manganese oxide hybrid materials are widely studied as electrode material in supercapacitor and lithium ion battery [23–25]. The hybrid materials show an enhanced performance compared to each component. However, the fabrication of graphene-manganese oxide porous material with a high surface area and the investigation of its gas sorption have not been reported.

Herein, we present a hydrothermal method of fabricating porous graphene- $\text{Mn}_3\text{O}_4$  (GMNO) material. In a hydrothermal reaction of the aqueous dispersion of  $\text{MnO}(\text{OH})_2$  and GO, the synthesis of  $\text{Mn}_3\text{O}_4$  and the reduction of GO could proceed simultaneously. Owing to the driving force of chemisorption interaction, the  $\text{Mn}_3\text{O}_4$  nanoparticles and graphene obtained *in-situ* assembled into three-dimensional structure with a high porosity. The gas sorption abilities of the as-prepared GMNO samples are different depending on the content of the  $\text{Mn}_3\text{O}_4$  contained in the hybrid material. The largest surface area value of GMNO is about  $680 \text{ m}^2 \text{ g}^{-1}$  and the maximum carbon dioxide capacity could reach up to 11 wt%.

## 1 Experimental

### 1.1 Materials

Natural flake graphite with an average particle diameter of  $20 \mu\text{m}$  (99 wt% purity) was obtained from Yingshida graphite Co. Ltd., Qingdao, China. Sulfuric acid (98 wt%), hydrogen peroxide (30 wt%), sodium nitrate, manganese (II) acetate hydrate ( $\text{Mn}(\text{Ac})_2 \cdot 4\text{H}_2\text{O}$ ) and ethanol were purchased from Beijing chemical works, China. All these reagents were of reagent grade and used without further purification. Ultra-pure water ( $18.2 \text{ M}\Omega \text{ cm}$ ) was obtained by the Millipore-ELIX water purification system.

### 1.2 Preparation of $\text{MnO}(\text{OH})_2$ nanoparticles

Aqueous NaOH solution ( $0.1 \text{ mol L}^{-1}$ ) was added into aqueous  $\text{Mn}(\text{CH}_3\text{COO})_2$  solution ( $5 \text{ mg mL}^{-1}$ ) dropwisely until the pH value was about 12. The transparent solution turned into yellow brown colloid without the appearance of a precipitate.

### 1.3 Preparation of graphene- $\text{Mn}_3\text{O}_4$ hybrid porous materials

Aqueous GO dispersion was prepared by chemical exfoliation of the natural flake graphite by a modified Hummers' method [26,27]. Aqueous GO dispersion was sonicated for 30 min prior to use. The colloidal dispersion of  $\text{MnO}(\text{OH})_2$  was added into the GO dispersion ( $1 \text{ mg mL}^{-1}$ , 20 mL), and

the mixture was diluted to 30 mL with ultra-pure water. After stirred for 6 h, the mixture was transferred into a Teflon-lined stainless-steel autoclave (capacity: 50 mL), and then heated to  $180^\circ\text{C}$  and maintained for 12 h. After cooled down to room temperature, a cylinder-shaped black solid was obtained. The black product was collected by filtration and washed with ethanol thrice, and dried at  $60^\circ\text{C}$  for 12 h. The products are named as GMNO-1, GMNO-2, GMNO-3, GMNO-4, and GMNO-5 with different weight ratio of  $\text{Mn}(\text{Ac})_2 \cdot 4\text{H}_2\text{O}$  to GO (8:1, 4:1, 1:1, 1:4, and 1:8), respectively.

$\text{Mn}_3\text{O}_4$  and hydrothermal reduced graphene (HTG) were prepared as control samples by hydrothermal reaction of  $\text{MnO}(\text{OH})_2$  or GO with the same condition as mentioned above.

### 1.4 Instrumental characterization

Nitrogen sorption and carbon dioxide sorption isotherms were obtained with a Micromeritics TriStarII3020 Surface Area and Porosity Analyze. The samples were degassed overnight at  $150^\circ\text{C}$ . Nitrogen sorption analysis was measured at 77 K, and the obtained nitrogen adsorption-desorption isotherms were evaluated to give the pore parameters, including Brunauer-Emmett-Teller (BET) specific surface area, pore size, and pore volume. The carbon dioxide adsorption isotherms of the samples were collected at 273 K.

Transmission electron microscopy (TEM) observations were carried out using a Tecnai G<sup>2</sup>20 S-TWIN microscope (FEI, USA) at an accelerating voltage of 200 kV. GMNO samples were dispersed in ethanol and the dispersions were dropped on a copper grid and dried in vacuum at  $60^\circ\text{C}$  overnight. Scanning electron microscopy (SEM) observations were carried out using a Hitachi S-4800 microscope (Hitachi Ltd., Japan) at an accelerating voltage of 6.0 kV and equipped with a Horiba energy dispersive X-ray spectrometer (EDXS). The accelerating voltage is 20.0 kV when taking SEM mapping images. X-ray diffraction (XRD) patterns of the samples were measured from  $5^\circ$  to  $90^\circ$  by a Philips X'Pert PRO X-ray diffraction instrument. Thermal gravimetric analysis (TGA) was performed on a Pyris Diamond thermogravimetric/differential thermal analyzer (Perkin-Elmer Instruments Co. Ltd, USA) by heating the samples from 30 to  $800^\circ\text{C}$  at a speed of  $5^\circ\text{C min}^{-1}$  in the atmosphere of air.

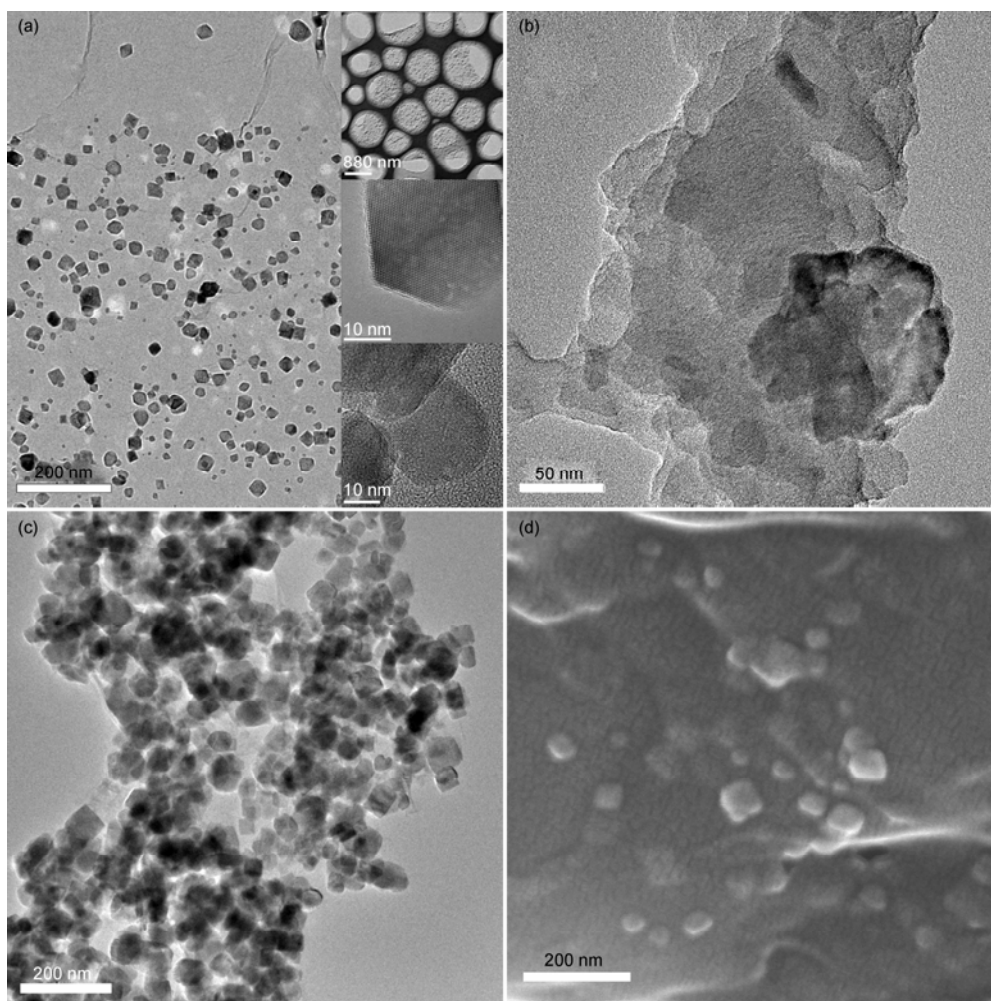
Infrared (IR) spectra were recorded in potassium bromide pellets using a Spectrum One Fourier transform infrared (FTIR) spectrometer (Perkin-Elmer Instruments Co. Ltd, USA). Raman spectra were recorded with a Renishaw inVia Raman spectrometer (Renishaw plc, UK). All samples were tested in powder form on silicon wafer without using any solvent. The laser excitation was provided by a regular model laser operating at 514 nm.

## 2 Results and discussion

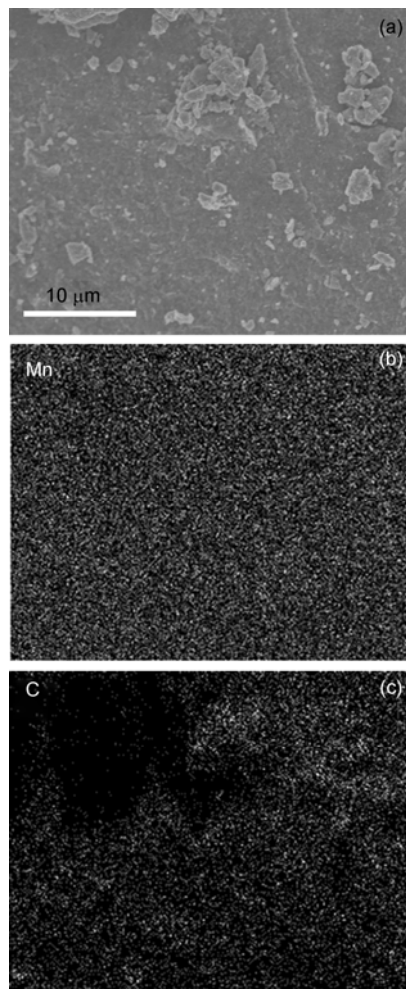
GO was prepared by the same method as our previous reports [28–30], in which the single-layered structure of GO sheet was conformed by atomic force microscope and the C/O atomic ratio of GO was measured as 2.2 by X-ray photoelectron spectroscopy [31–33]. GO is frequently used as a precursor of graphene in the fabrication of graphene-based porous materials [13,15]. As reported in our previous studies, most of the oxygen-containing groups on GO sheets can be removed in a solvo-/hydro-thermal reaction [32]. In our preparation of GMNO materials, aqueous NaOH solution was added into aqueous  $\text{Mn}(\text{CH}_3\text{COO})_2$  to prepare  $\text{MnO}(\text{OH})_2$ . The as-prepared  $\text{MnO}(\text{OH})_2$  colloid was mixed with GO dispersion and the mixture was loaded into a Teflon-lined stainless-steel autoclave. After the hydrothermal reaction, GO was reduced into graphene and  $\text{MnO}(\text{OH})_2$  transformed into  $\text{Mn}_3\text{O}_4$ . Owing to the chemisorption interaction between the two species, graphene sheets and  $\text{Mn}_3\text{O}_4$  nanoparticles assembled together. Graphene sheets could also be considered as the substrate for

the growth of  $\text{Mn}_3\text{O}_4$  nanoparticles, therefore, the  $\text{Mn}_3\text{O}_4$  nanoparticles do not tend to aggregate together and show better dispersibility in GMNO samples than in  $\text{Mn}_3\text{O}_4$  sample. In the TEM and SEM images (Figure 1(a) and (d)) of the GMNO-3 sample, cubic nanoparticles with a size of about 20 nm are well dispersed on the graphene sheets. However, the particles in  $\text{Mn}_3\text{O}_4$  sample (Figure 1(c)) and the graphene sheets in HTG sample (Figure 1(b)) show strong aggregation. Meanwhile, owing to the space between graphene sheets are limited, the size growth of  $\text{Mn}_3\text{O}_4$  nanoparticles becomes more difficult with the existence of graphene, therefore, and the particle sizes of  $\text{Mn}_3\text{O}_4$  nanoparticles are smaller in GMNO-3 than in  $\text{Mn}_3\text{O}_4$  sample (50 nm). The carbon and manganese elements mapping images of GMNO-3 show that the different elements are dispersed uniformly in the hybrid material (Figure 2).

Figure 3 shows the XRD patterns of the hybrid materials,  $\text{MnO}(\text{OH})_2$  and  $\text{Mn}_3\text{O}_4$ .  $\text{MnO}(\text{OH})_2$  showed low-crystalline structure, and was transformed into  $\text{Mn}_3\text{O}_4$  with increased crystalline after hydrothermal process. The XRD peaks of



**Figure 1** TEM images of GMNO-3 (a), HTG (b), and  $\text{Mn}_3\text{O}_4$  (c), and SEM image of GMNO-3 (d).

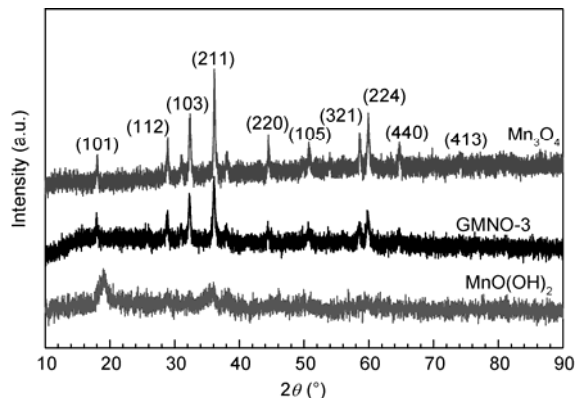


**Figure 2** SEM image (a), Mn (b) and C (c) element mapping images of GMNO-3 (the white color in the element mapping images represent the dispersibility of the corresponding elements).

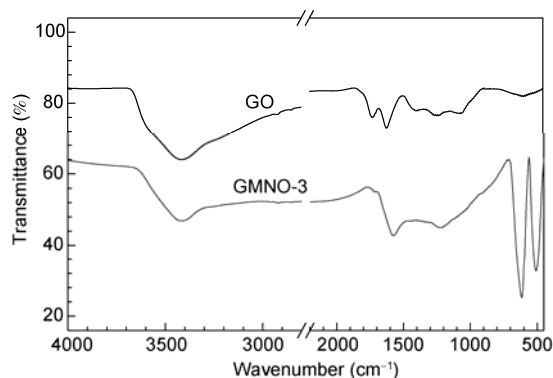
$\text{Mn}_3\text{O}_4$  and GMNO-3 are sharper, and the manganese oxide in  $\text{Mn}_3\text{O}_4$  and GMNO are hausmannite. The crystal structure of the  $\text{Mn}_3\text{O}_4$  growing on the graphene sheets is also revealed in the Figure 1(a). GMNO samples show a series of characteristic peaks at  $17.9^\circ$ ,  $28.9^\circ$ ,  $32.3^\circ$ ,  $36.1^\circ$ , and  $59.8^\circ$ , which can be assigned to the (101), (112), (103), (211), and (224) plane of the  $\text{Mn}_3\text{O}_4$  (JCPDS No. 24-0734) [34].

Figure 4 shows the IR spectra for GO and GMNO-3. The absorption peak at  $1730\text{ cm}^{-1}$  assigned to the carbonyl groups, which could be observed in the IR spectrum of GO but almost disappeared in the spectrum of GMNO-3. It is an evidence of the removal of the oxygen-containing groups on GO. GMNO-3 sample also shows two prominent absorption peaks at  $506$  and  $610\text{ cm}^{-1}$  that can be assigned to the stretch vibration of the manganese and oxygen bond.

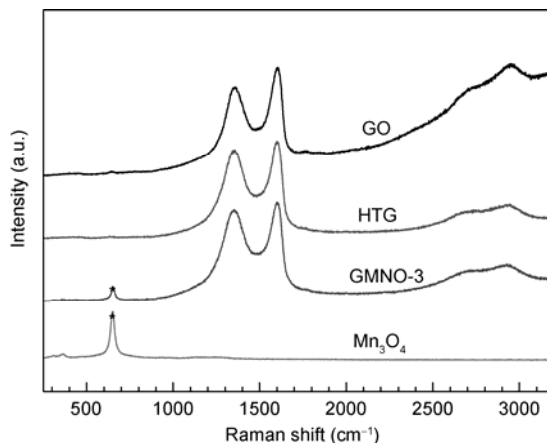
Figure 5 shows the Raman spectra of GO, HTG, GMNO-3, and  $\text{Mn}_3\text{O}_4$ . D- (ca.  $1350\text{ cm}^{-1}$ ) and G- (ca.  $1590\text{ cm}^{-1}$ ) bands are characteristic signal of carbon materials [31,35]. Compared with GO, HTG and GMNO show a slight increase in the D/G intensity ratio, which might be



**Figure 3** XRD patterns of  $\text{MnO}(\text{OH})_2$ ,  $\text{Mn}_3\text{O}_4$  and GMNO-3.



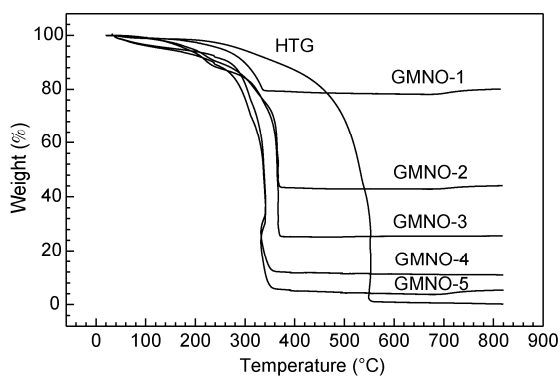
**Figure 4** IR spectra of GO and GMNO-3 samples.



**Figure 5** Raman spectra of GO, HTG, GMNO-3 and  $\text{Mn}_3\text{O}_4$ .

because the “cutting effect” of hydrothermal treatment on GO [36]. Meanwhile, another peak at about  $653\text{ cm}^{-1}$  is observed in the Raman spectrum of GMNO-3, which could be attributed to the  $\text{Mn}_3\text{O}_4$  contained in the hybrid material.

Figure 6 shows the TGA results of GMNO samples. The skeleton of the graphene in the GMNO decomposed at a lower temperature ( $300\text{--}400^\circ\text{C}$ ) compared to that of the HTG (about  $550^\circ\text{C}$ ), which might be caused by the reaction of skeleton carbon atoms and the manganese oxide

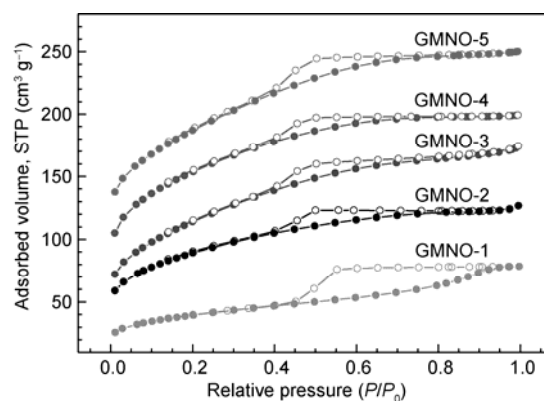


**Figure 6** TGA results of GMNO samples.

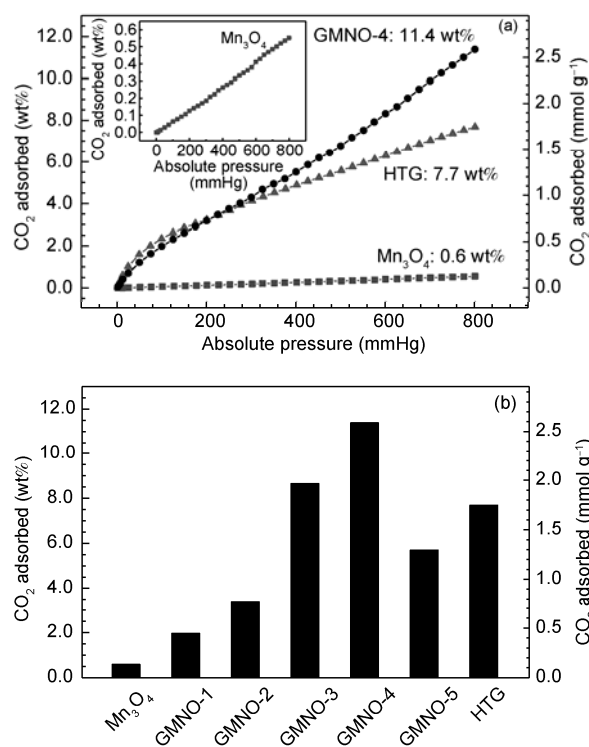
nanoparticles. Meanwhile, the content of the  $\text{Mn}_3\text{O}_4$  in the GMNO materials increase with the weight ratio of the manganese precursor to GO. The weight percentages of the  $\text{Mn}_3\text{O}_4$  are 78%, 42.8%, 25.6%, 11.0%, and 3.8% from GMNO-1 to GMNO-5, respectively.

GMNO samples show a small hysteresis loop at the relative pressure about 0.4–1.0 (Figure 7), which is similar to the nitrogen sorption isotherms of the graphene-based inorganic porous material we reported before [13]. Owing to the different contents of the  $\text{Mn}_3\text{O}_4$ , the surface area values of the GMNO samples are varying from about 140 to 685  $\text{m}^2 \text{g}^{-1}$ . The specific surface area of the GMNO samples decrease with the increase in the content of the  $\text{Mn}_3\text{O}_4$  nanoparticles (Figure 7 and Table 1), which might be because the density of the  $\text{Mn}_3\text{O}_4$  nanoparticles are larger than carbon materials.

The carbon dioxide capacities are list in Table 1, and Figure 8 shows the comparison of the HTG,  $\text{Mn}_3\text{O}_4$  and GMNO samples.  $\text{Mn}_3\text{O}_4$  shows the lowest carbon dioxide adsorption capacity, and the value of HTG is about 7.7 wt% which is comparable to the carbon material previously reported [1,3,4]. The carbon dioxide adsorption capacities of the GMNO samples show a maximum value of about 11.4 wt% at GMNO-4, which is larger than HTG and  $\text{Mn}_3\text{O}_4$ . As observed in Figure 8(a), the carbon dioxide adsorption isotherms of HTG and  $\text{Mn}_3\text{O}_4$  are slightly different, which might be owing to their different adsorption mechanisms. There are two main factors affect the carbon dioxide adsorption capacity for the GMNO samples, the surface area and the basic sites. HTG exhibits a similar carbon dioxide adsorption isotherm to other carbon materials and its



**Figure 7** Nitrogen sorption isotherms of GMNO samples.



**Figure 8** Carbon dioxide adsorption isotherms (a), and capacity comparison (b) of HTG,  $\text{Mn}_3\text{O}_4$  and GMNO samples. 1 mmHg=1.33 $\times 10^2$  Pa.

carbon dioxide adsorption capacity is mainly depended on its surface area [1,4]. Whereas, the specific surface area of the  $\text{Mn}_3\text{O}_4$  sample is low and the carbon dioxide adsorption

**Table 1** Porosity parameters measured by nitrogen sorption and  $\text{CO}_2$  adsorption capacities of the samples

	Weight ratio ( $\text{Mn}(\text{Ac})_2 \cdot 4\text{H}_2\text{O} : \text{GO}$ )	$S_{\text{BET}}$ ( $\text{m}^2 \text{g}^{-1}$ )	Pore size (nm)	Pore volume ( $\text{cm}^3 \text{g}^{-1}$ )	$\text{CO}_2$ uptake (wt%)
GMNO-1	8:1	140	4.9	0.12	2.0
GMNO-2	4:1	283	4.3	0.19	3.4
GMNO-3	1:1	410	4.3	0.26	8.7
GMNO-4	1:4	541	4.3	0.31	11.4
GMNO-5	1:8	685	4.3	0.48	5.7
HTG	—	647	—	—	7.7
$\text{Mn}_3\text{O}_4$	—	18	—	—	0.6

capacity is mainly contributed by the basic sites contained in the sample, the carbon dioxide uptake amount linearly increases with the pressure (inset in Figure 8(a)). Therefore, GMNO-4 shows the largest carbon dioxide uptake owing to the two factors.

### 3 Conclusion

Graphene-Mn<sub>3</sub>O<sub>4</sub> materials with high porosity were prepared through the hydrothermal reaction, in which the reduction of GO and the formation of Mn<sub>3</sub>O<sub>4</sub> are processing simultaneously. The specific surface area values of the as-prepared GMNO samples are different depending on the content of the Mn<sub>3</sub>O<sub>4</sub>, and the GMNO samples have a carbon dioxide adsorption amount up to 11.4 wt%, which is comparable with other carbon materials.

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