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Structural and adhesion properties of poly(ethyl 2-cyanoacrylate) post-cured at different temperatures and times

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ABSTRACT

The structure and properties of poly(ethyl 2-cyanoacrylate)s (pECNs) cured at 25°C during 24 hours, and post-cured at different temperatures (45-90°C) and times (2-6 hours) were studied. Irrespective of the post-cure conditions, pECN was not fully polymerized and two different structures of different molecular weights and topographies were obtained. The post-cure decreased the shear strength values in joints made with aluminum and ethyl cyanoacrylate due to the decrease in the percentage of higher molecular weight polymeric chains and the formation of new short polymeric chains by unzipping de-polymerization mechanism, both contributed to decrease the mechanical properties. On the other hand, the post-cure changed the morphology of pECN because of nano-spheres were produced by curing at 25°C which were not present after post-cure at 90°C.

ARTICLE HISTORY

Received 7 April 2018 Revised 15 July 2018 Accepted 6 August 2018

KEYWORDS

Poly(ethylcyanoacrylate); thermal post-cure; adhesion; unzipping mechanism; DSC; TGA

1. Introduction

The polymerization conditions of cyanoacrylates such as the initiator [1], the temperature [2–4], and the pH [5], among other, determine their structure and properties. For instance, Dossi et al. [5] have shown that during cyanoacrylate polymerization and depending on the pH, two main structures with different molecular weights can be formed. Thus, high molecular weight fractions were not observed under acidic polymerization conditions carried out at 0 or 65 °C. However, the polymerization of cyanoacrylates at 25 °C produced the formation of two main structures with different molecular weights, and the polymerization under neutral pH led to high molecular weight polymers only. On the other hand, Puglisi et al. [1] established differences in the size, the topography and the properties of the nano-capsules formed during the polymerization of ethyl cyanoacrylate by using different solvents as initiators. Thus, the use of aprotic and water-miscible solvents, such as acetone and acetonitrile, caused the formation of suspensions of polymer nano-capsules, while water-miscible solvents,

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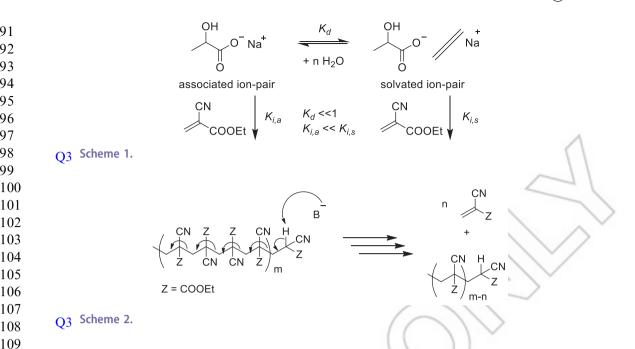
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such as alcohols, led to colloidal suspensions consisting of nano-spheres and nanocapsules. Furthermore, the solvent used as polymerization initiator determined the nanoparticle size and the polydispersity of poly(ethyl cyanoacrylate)s. These results have been explained by the co-existence of two different polymerization mechanisms, i.e., interfacial precipitation of pre-formed polymer and interfacial polymerization.

The influence of the temperature on the polymerization of ethyl cyanoacrylate has been studied elsewhere [2-4,6-8], but different conclusions have been obtained depending on the polymerization conditions. For example, the particle size of poly(ethyl 2-cyanoacrylate)s (pECNs) has been related to the polymerization temperature by Reddy et al. [2] whose have demonstrated that the higher was the polymerization temperature, the larger was the polymer particle size because of some agglomeration of particles was produced. Under vapour phase conditions, the influence of the polymerization temperature on the morphology of pECN has been also studied, and it has been shown the formation of nanofibers at 100 °C but spheres at $180 \,^{\circ}C$ [3]. On the contrary, Algaier et al. [4] have considered the use of ethyl cyanoacrylate in revealing fingerprints in thermostatic chamber (20-80 °C) and they have found that higher polymerization degree of ethyl cyanoacrylate at lower temperature by using sodium lactate as polymerization initiator was obtained. Algaier et al. [4] have related the ionic pair of the sodium lactate with the polymerization rate of the ethyl cyanoacrylate, concluding that they were more reactive than the associated ionic pairs, especially at low temperature (Scheme 1). Interestingly, these authors did not observe a significant variation of the molecular weight of the poly-cyanoacrylate by changing the polymerization temperature, likely due to high polymerization rate caused by the solvated ion pairs in the sodium lactate initiator.

70 On the other hand, Senchenya et al. [6] have studied the influence of the polymer-71 ization temperature on the adhesion properties of different cyanoacrylates. They have 72 demonstrated that the polymerization at high temperature decreased the adhesion 73 because of the thermal degradation of the cyanoacrylate was favoured. Thus, the com-74 plete degradation and the loss of adhesion were produced by curing joints made with 75 composites and ethyl cyanoacrylate at 150 °C during 8 hours. In the same direction, 76 Guthrie et al. [7] have studied the effect of the thermal treatment at 90 °C during dif-77 ferent times on the adhesion of steel-ethyl cyanoacrylate joints, and they have 78 observed an important decrease in adhesion and a slight reduction in the molecular 79 weight of the polymer by increasing the time of the thermal treatment. The negative 80 effects of the increase of the temperature on the poly-cyanoacrylate properties have 81 also been studied by Chorbadjiev et al. [8], ascribing them to the formation of free 82 monomer during the thermal decomposition of the polymer. In addition, Robello 83 et al. [9] have established that the presence of residual amounts of organic or 84 inorganic base (usually the initiator) at the end of the polymerization may favour the 85 degradation of the poly-cyanoacrylates. This base-induced depolymerization process, 86 also known as "unzipping" mechanism (Scheme 2), has been proposed for explaining 87 the low stability and the decreased adhesion of poly-cyanoacrylates after long storage 88 periods under aqueous conditions [7]. Ryan et al. [10] have also studied the 89 influence of the temperature on the polymerization-depolymerization reactions of 90 poly-cyanoacrylates, and they have proposed that the unzipping depolymerization

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process in polycyanocrylates was initiated by the abstraction of one terminal proton in the "parent" polymer chain producing monomeric units by a retro-Michael cascade reaction. The liberated free monomer was then able to initiate a new base-catalyzed polymerization generating new low molecular weight "daughter" polymer chains.

On the other hand, depending on the initiator nature, the polymerization of cyanoacrylates can take place thought three different mechanisms: radical, anionic, and zwitterionic. Pepper [11] has proposed a SINT (slow-initiated, non-terminated) process for zwitterionic polymerizations using tertiary amines as initiators. Furthermore, based on kinetic studies, Johnston and Pepper [12] supported the absence of intrinsic termination process in the polymerization of cyanoacrylates.

Because of the contradictory evidences found in the literature dealing with the 120 121 influence of the polymerization temperature on the properties of ethyl cyanoacrylate, 122 the effect of the temperature and the time on the structure and properties of the 123 pECNs are considered in this study. To the best of our knowledge, no systematic 124 studies have been performed so far on the influence of the post-cure temperature and 125 time on the structure and properties of the pECNs. Thus, in this study, ethyl cyano-126 acrylate was polymerized at 25 °C and later post-cured at different temperatures dur-127 ing different times and the changes in the structure, the topography and the adhesion 128 properties of the pECNs were determined. 129

2. Experimental

2.1. Materials

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Ethyl cyanoacrylate (ECN) was supplied by Adhbio S.L. (Elche, Spain). ECN contains hydroquinone and *p*-toluenesulphonic acid stabilizers, and its water content obtained by Karl-Fisher titration is 566 ± 40 ppm.

4 😉 G. ESTAN-CEREZO ET AL.

A 1.0% (v/v) aqueous (ultrapure water) solution of tryethylamine (Sigma-Aldrich, Steinheim, Germany) was used as polymerization initiator.

2.2. Polymerization of ethyl cyanoacrylate (ECN)

ECN was polymerized in an aluminum container by drop-wise addition of 1.0% (v/v) aqueous solution of tryethylamine over 1 ml of ECN under continuous stirring. The mixture was allowed to react under open air at 25 °C during one hour, and a white polymer was obtained. For removing the non-reacted initiator, the polymer was washed several times with ultrapure water until neutral pH in the washing liquid was obtained. The solid pECN was completely dried under open air at 25 °C during 24 hours. Then, the cured pECN was post-cured at 45-90 °C during 2-6 hours in Selecta 2000208 oven (Barcelona, Spain).

2.3. Experimental techniques

2.3.1. Attenuated total reflection infrared spectroscopy (ATR-IR)

ATR-IR spectra of the pECNs were obtained in a Tensor 27 FT-IR spectrometer (Bruker Optik GmbH, Erlinger, Germany) by using Germanium prism ATR accessory. In absorbance mode, 64 scans were recorded with resolution of 4 cm^{-1} in the wavenumber range of 4000 to 400 cm⁻¹.

2.3.2. Nuclear Magnetic Resonance (NMR)

¹H NMR spectra of the pECNs were obtained in a Bruker AC-400 spectrometer (Bruker, Rheinstetten, Germany), proved with 400 MHz magnet. 30-40 mg simple disolved in deuterated chloroform (CDCl₃) was placed in NMR flasks and the spectra were processed with Mestrec 5.0 program (Mestrelab Research S.L., Santiago de Compostela, Spain).

¹H DOSY experiment was performed in a Bruker Advance DRX500, QNP300MHz S1 with Z-gradient equipment (Bruker, Rheinstetten, Germany). Stimulated echo sequence using bipolar gradient pulses for diffusion and 1 spoil gradient, with a spectral width of 6172.84 Hz, were used. Stebpgp1s avance-version pulse program for acquisition data was used. D1 value of relaxation delay was 1 second, the delay for gradient recovery was 0.2 ms and the diffusion time (big DELTA) was 80.0 ms (16 gradient, Gradient Calib Contant 5.19 G/mm). The gas flow used at 25 °C was 535 l/h. Data were processed after the Fourier transformation of each FID with TopSpin3.2 to obtain the chemical shift as the x-axis along with logD as thee y-axis. Diffusion coefficient values (D) were corrected based on the D value of CDCl₃ as an internal standard.

2.3.3. Differential Scanning Calorimetry (DSC)

1772.5.5. Differential Sectioning Calorimetry (DSC)178The structural changes and the existence of post-polymerization of the pECNs were179analyzed in DSC Q100 instrument (TA Instruments, New Castle, DE, USA). 5–10 mg180of sample were placed in a hermetic aluminum pan and heated between –50 and150 °C under nitrogen atmosphere (flow rate: 50 ml/min) by using a heating rate of

10 °C/min. Then, the sample was cooled down from 150 to -50 °C by using a cooling 181 rate of 10 °C/min. The existence of post-polymerization in the cured and post-cured 182 pECN was evidenced from the first DSC heating run. Afterwards, a second heating 183 run from -50 to 300 °C was also performed by using a heating rate of 10 °C/min, the 184 glass transition temperature (T_{g}) and the decomposition temperature (T_{dec}) of the 185 pECNs were determined from the DSC thermograms. 186

2.3.4. Thermal gravimetric analysis (TGA) 188

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The thermal stability of the pECNs was analyzed in TGA Q500 instrument 189 190 (TA Instruments, New Castle, DE, USA). 5-10 mg of sample were placed in a platinum crucible under nitrogen atmosphere (flow rate: 100 ml/min) and they were 192 heated from room temperature to 350 °C by using a heating rate of 10 °C/min.

194 2.3.5. Molecular weight measurement

195 The molecular weights of the pECNs were determined in Waters gel permeation 196 chromatograph (Waters, Milford, MA, USA) equipped with refractive index detector. 197 Tetrahydrofurane (THF) was used as solvent, and the measurements were carried out 198 at 35 °C and by using a flow rate of 1 ml/min that was controlled by 1515 HPLC 199 pump (Waters, Milford, MA, USA). Polystyrene standards and Stryrabel columns 200(300 x 7.9 mm, 5 µm) (Waters, Milford, MA, USA) were employed. 201

2.3.6. Transmission Electronic Microscopy (TEM)

The topography of the pECN surfaces was analyzed by transmission electron microscopy in Jeol JEM-1400 Plus equipment (Jeol, Tokyo, Japan) provided with LaB₆ filament. A voltage of 120 V was used and TEM micrographs were recorded with Orius camera (Gatan, Pleasanton, USA).

2.3.7. Scanning Electron Microscopy (SEM)

The topography of the pECN surfaces was also analyzed by scanning electron microscopy in Jeol JSM 840 equipment (Jeol, Tokyo, Japan) equipped with a scintillator-photomultiplier of 4nm resolution. A voltage of 10 keV was employed and the samples were gold coated before analysis.

2.3.8. Single lap-shear test

215 The adhesion properties of ethyl cyanoacrylate (ECN) were obtained from single lap-216 shear tests of aluminum/ECN/aluminum joints. Aluminum 5754 test samples of 217 dimensions $150 \times 30 \times 1.5$ mm were used, and, before joint formation, they were 218 mildly scored with green Scotch Brite® scourer followed by washing with isopropyl 219 alcohol (Javer, Móstoles, Spain) and drying at room temperature during 15 minutes. 220 The joined area of the aluminium test samples was 30×30 mm, and an amount of 221 15 µl of ethyl cyanoacrylate was placed over one of the aluminium test sample; then, 222 the other aluminium test sample was immediately placed over and a pressure of 223 0.11 MPa was applied during 10 minutes. The joints were allowed to cure at $25\,^{\circ}\text{C}$ 224 during 24 hours. Afterwards, some of the adhesive joints were post-cured in oven at 225 45 or 90 °C during 2 hours, and then kept at room temperature during 72 hours.

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Single lap-shear tests were carried out in universal testing machine Instron 8516 (Instron, Buckhinghanshire, UK), a cross-head speed of 10 mm/min was used. Five replicates were carried out and averaged. The loci of failure of the joints were assessed by visual inspection.

3. Results and discussion

3.1. Influence of the post-cure temperature on the properties of pECN

234 Figure 1 shows, as typical example, the ATR-IR spectra of pECN cured at 25°C for 235 24 hours and post-cured at 90 °C for 2 hours. Typical bands of cyanoacrylates can be 236 distinguished in the ATR-IR spectra of Figure 1: C=C stretching at 3128 and 237 1625 cm^{-1} , methylene bands at 2987-2860, 1447, 720 y 856 cm^{-1} , C=O stretching at 238 1731 cm^{-1} , and C–O–C bands at 1090, 1150 and 1243 cm^{-1} . The post-cure does not 239 cause the formation of new absorption bands pointing to the de-polymerization of 240 pECN chains by unzipping mechanism. However, the intensities of the C=C bands at 241 3128 and 1625 cm^{-1} are reduced after post-cure, indicating higher degree of polymer-242 ization. Furthermore, after post-cure the intensities of the C-O-C bands at 1090, 243 1150 and 1243 cm⁻¹ increase and the ones of the methylene bands at 2987-2860, 720 244 y 856 cm^{-1} decrease, likely due to the existence of polymer with lower molecular 245 weight caused by de-polymerization of pECN by unzipping mechanism. Therefore, 246 the post-cure of pECN cured at 25 °C for 24 hours seems to create two different 247 structures of different molecular weights, one with higher degree of polymerization 248 and the other with lower degree of polymerization; this hypothesis has also been 249 proposed elsewhere [5]. 250

¹H NMR spectra of pECN cured at 25 °C for 24 hours and post-cured at 90 °C for 2 hours are given in Figure 2(a,b) respectively. The ¹H NMR spectra of pECN cured at 25 °C during 24 hours (Figure 2(a)) show the broads signals of the protons of the ethyl group at 1.32 and 4.32 ppm, and the signals of the protons of the double C=C bond appear at 6.67 and 7.09 ppm. The post-cure of pECN at 90 °C for 2 hours reduced notably the intensities of the signals of the protons of the double C=C bond (Figure 2(b)), confirming the higher degree of polymerization evidenced by ATR-IR spectroscopy.

258 Differential scanning calorimetry (DSC) was used for assessing the structural 259 changes in the cured and post-cured pECNs. Figure 3(a) shows the DSC thermograms 260 corresponding to the first heating run of the cured pECN at 25 °C for 24 hours and 261 post-cured at 45 and 90 °C for two hours. All pECNs show an endothermic post-262polymerization peak indicating that, under the experimental conditions used in this 263 study, the polymerization of ECN is not complete at 25 °C not even after being post-264 cured at 90 °C for 6 hours. The post-polymerization peak of the cured pECN appears at 265 106 °C and the one for the post-cured pECN is produced at higher temperature (115 to 266 128 °C), the higher is the post-cure temperature, the higher is the temperature of the 267 post-polymerization peak (Table 1). On the other hand, the post-polymerization enthalpy 268 of pECN cured at 25 °C for 24 hours is 26 J/g and it is reduced noticeably (11 to 3 J/g) 269 after post-cure (Table 1). Therefore, the post-cure of pECN at 45°C and, more 270markedly, at 90 °C decreases the extent of post-polymerization of pECN, since the

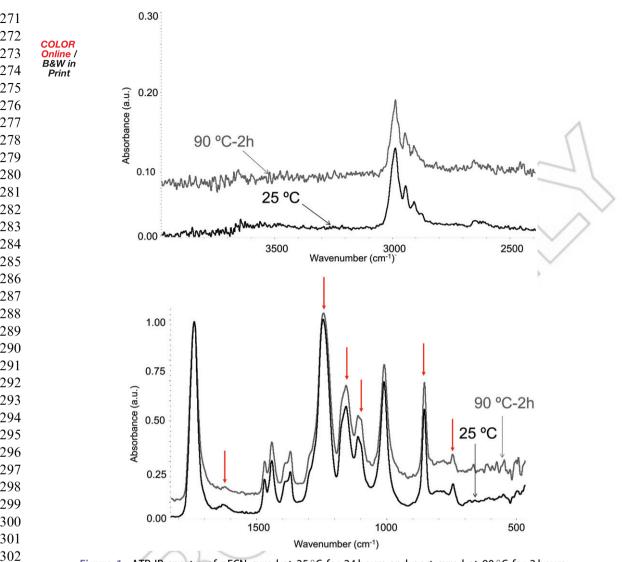


Figure 1. ATR-IR spectra of pECN cured at 25 °C for 24 hours and post-cured at 90 °C for 2 hours.

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thermal treatments usually favored the termination step by a SINT (slow-initiated, non-terminated) mechanism in zwitterionic polymerizations initiated by tertiary amines [11].

During the second DSC heating run, none of the DSC thermograms show an exothermic peak (Figure 3(b)), indicating the complete polymerization of pECN during the DSC experiment. Therefore, vitrification is produced during post-cure of pECN as the post-cure temperatures are below the glass transition temperature (T_g) of the polymer (above 130 °C). During the first DSC heating run carried out above 130 °C, the vitrification does not persist and polymerization of ECN is completed. The second DSC heating run of pECN shows the glass transition temperature as an endothermic inflexion in the thermogram and the decomposition of pECN at higher temperature as well. Table 1 shows that the T_g value of pECN is somewhat similar regardless the polymer is post-cured or not (130 °C for cured pECN at 25 °C and 134-136 °C for

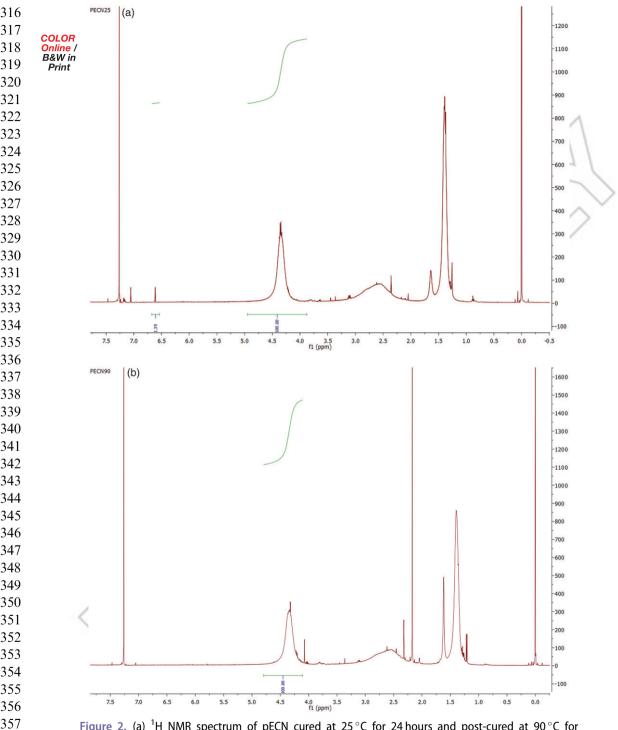


Figure 2. (a) ¹H NMR spectrum of pECN cured at 25 °C for 24 hours and post-cured at 90 °C for 2 hours. (b) ¹H NMR spectrum of pECN post-cured at 90 °C for 2 hours.

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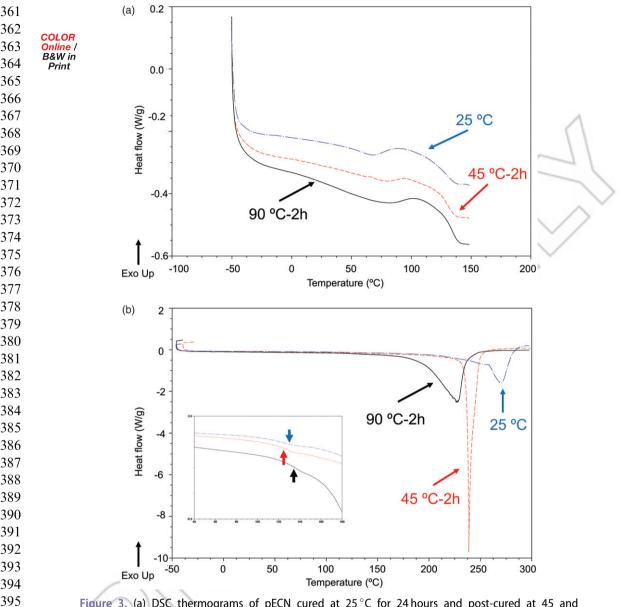


Figure 3. (a) DSC thermograms of pECN cured at 25 °C for 24 hours and post-cured at 45 and 90 °C for 2 hours. First heating run. (b) DSC thermograms of pECN cured at 25 °C for 24 hours and post-cured at 45 and 90 °C for 2 hours. Second heating run.

Table 1. Temperature (T_{post}) and enthalpy (ΔH_{post}) of post-polymerization, glass transition temperature (T_g), and decomposition temperature (T_{dec}) and enthalpy (ΔH_{dec}) of pECNs cured at 25 °C for 24 hours and post-cured at 45 and 90 °C for different times.

	1 st he	1 st heating run		2 nd heating run		
Post-cure condit	on T _{post} (°C)	ΔH_{post} (J/g)	T _g (°C)	T _m (°C)	ΔH_m (J/g)	
25 °C	106	26	130	226	775	
45 °C-2h	115	11	124	239	571	
90 °C-2h	124	6	134	264	504	
90 ° C-6h	128	3	136	268	500	

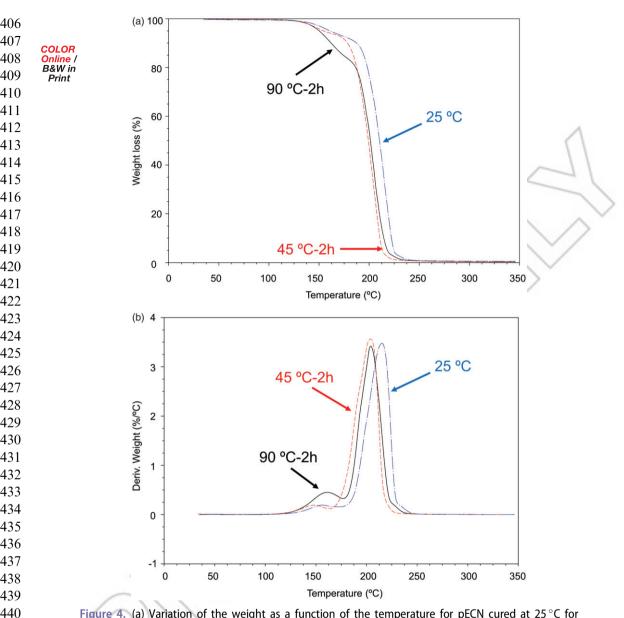


Figure 4. (a) Variation of the weight as a function of the temperature for pECN cured at 25 °C for 24 hours and after being post-cured at 45 and 90 °C for 2 hours. TGA experiments. (b) Variation of the derivative of the weight as a function of the temperature for pECN cured at 25 °C for 24 hours and after being post-cured at 45 and 90 °C for 2 hours. TGA experiments.

post-cured pECN at 90 °C); the T_g values obtained in this study agree well with the ones previously reported for pECN [13]. On the other hand, the decomposition temperature increases and the decomposition enthalpy decreases for the post-cured pECNs, in greater extent by increasing the post-cure temperature, confirming higher extent of polymerization of pECN caused by a SINT mechanism [11].

TGA was used to assess the thermal stability and the structural changes in the cured and post-cured pECN. Figure 4(a) shows, as typical example, the TGA curves of pECN

Table 2. Temperatures at which 5 ($T_{5\%}$) and 50 ($T_{50\%}$) wt% losses are produced, and their corresponding weight losses for the two thermal decompositions of pECN cured at 25 °C for 24 hours and post-cured at 45 and 90 °C for 2 and 6 hours.

Post-cure condition	T ₁ (°C)	Weight loss ₁ (%)	T ₂ (°C)	Weight loss ₂ (%)	T _{5%} (°C)	T _{50%} (°0
25 °C	156	10	215	85	160	210
45 °C-2h	148	6	204	90	153	199
90 °C-2h	161	18	205	77	150	201
90 °C-6h	163	21	208	72	149	204

Table 3. Diffusion coefficients (D) of pECN cured at 25 °C for 24 hours and pECN post-cured at 90 °C for 2 hours.

	D (10 ⁻	¹⁰ ms ⁻¹)	
Post-cure condition	TMS	pECN	D ^{pECN} /D ^{TMS}
25 °C	30.2	8.39	0.28
90 °C-2h	30.1	8.98	0.30

cured at 25 °C for 24 hours and after being post-cured at 90 °C for 2 hours. The postcure diminishes the thermal stability of pECN, more markedly by increasing the postcure temperature, as it starts to degrade at a lower temperature and the temperatures at which 5 and 50% mass of pECN decreased (Table 2). The main thermal decomposition is obtained at 215 °C for pECN cured at 25 °C for 24 hours and at 204-208 °C after post-curing at 45 or 90 °C for two hours. Furthermore, the post-cure of pECN increases the amount of mass loss at 148-163 °C, particularly for the post-cure at 90 °C, due to the formation of new short polymer chains (18 wt% in pECN post-cured at 90 °C and 10 wt% in pECN cured at 25 °C for 24 hours). These evidences point out again to the de-polymerization of pECN by unzipping mechanism produced during post-cure, and is also confirmed by the decrease of the temperatures at which 5 and 50 wt% are lost after post-cure of pECN (Table 2). On the other hand, the pECN shows two main thermal decompositions that are better evidenced in the derivative TGA curves (Figure 4(b)). The two thermal decompositions appear at 156 °C and 215 °C (for the pECN cured at 25 °C for 24 hours) and at 148-163 °C and 204-208 °C (for the pECN post-cured at 45 and 90 °C for two hours). These results clearly indicate the existence of two different main polymeric structures of different molecular weights in both polymers, in accordance with previously reported data [5]. The thermal decomposition of pECN at higher temperature points to the dominance of polymeric chains with high molecular weights whereas the one at lower temperature can be ascribed to shorter polymeric chains.

Diffusion-ordered NMR spectroscopy (DOSY) studies were also performed in pECN cured at 25 °C for 24 hours and after being post-cured at 90 °C. According to Table 3, the diffusion coefficient of the pECN post-cured at 90 °C is higher than the one of the pECN cured at 25 °C for 24 hours, this confirm the existence of shorter polymeric chains in the polymer post-cured at 90 °C, in agreement with the evidences shown by TGA experiments.

491 shown by TGA experiments.
492 The molecular weights of the pECN cured at 25 °C for 24 hours and after being post-cured at 90 °C for 2 hours were determined by GPC (Table 4). The molecular weight distribution of the pECNs is relatively narrow (polydispersity: 2.2 for pECN cured at 25 °C and 1.6 for pECN post-cured at 90 °C). After the post-cure of pECN at 90 °C, a decrease in Mw and Mz values of the polymer is observed indicating a

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Mn (Da)	Mw (Da)	Mz (Da)	Mw/Mn
10670	23270	39760	2.2
12700	20240	30700	1.6
	10670	10670 23270	10670 23270 39760

 Table 4.
 Molecular weights of pECN cured at 25°C for 24 hours and post-cured at 90°C for 2 hours.

reduction of the medium and larger polymeric chains. Furthermore, a slight increase in the Mn value is also noticed in pECN post-cured at 90 °C because of the formation of new shorter polymeric chains of pECN through an unzipping de-polymerization mechanism. These results confirm the evidences shown above by ATR-IR spectroscopy, DOSY and TGA studies. On the other hand, Guthrie et al. [7] have also found a slight reduction in the molecular weight of the polycyanoacrylate by increasing the time of the thermal treatment at 90 °C, although the reported values of the molecular weights of the pECNs are lower than the ones in this study likely due to the different initiators of the polymerization used in both studies.

3.2. Influence of the duration of the post-cure on the structure and morphology of pECNs

514 The duration of the post-cure may also affect the structure of the pECN. The struc-515 tural characterization of the pECNs post-cured at 90 °C for 2 and 6 hours was carried 516 out. Table 1 shows some parameters obtained from the DSC thermograms of the 517 pECNs post-cured at 90 °C for 2 and 6 hours. The post-polymerization temperature is 518 higher and the enthalpy of post-polymerization of pECN is slightly lower when the 519 duration of the post-cure treatment at 90 °C increases. On the other hand, the T_g 520 value of pECN slightly increases by extending the duration of the post-cure at 90 °C. 521 522 Therefore, the longer is the post-cure at 90 °C, the more complete is the polymeriza-523 tion, although a complete polymerization is not observed.

524 Figure 5(a) shows the TGA curves of pECN post-cured at 90 °C for 2 and 6 hours. 525 The thermal stability up to 170 °C of pECN post-cured at 90 °C for 2 and 6 hours is 526 similar but above this temperature, the thermal stability of the post-cured pECN dur-527 ing 6 hours is higher. In fact, according to Table 2, the temperature at which 50 wt% 528 is lost is higher when the post-cure treatment lasted up to 6 hours. On the other 529 hand, the post-cured pECN showed two main thermal decompositions clearly seen in 530 the derivative TGA curves (Figure 5(b)). These decompositions appear at $161 \,^{\circ}$ C and 531 205 °C in the post-cured pECN at 90 °C for 2 hours whereas they appear at 163 °C 532 and 208 °C in the post-cured pECN at 90 °C for 6 hours. These results confirm again 533 the existence of two different main polymeric structures of different molecular 534 weights in the post-cured pECN. On the other hand, the increase of the time of the 535 post-cure at 90 °C increases the weight loss and the maximum temperature of thermal 536 decomposition of the post-cured pECN in the TGA curve. Thus, the longer duration 537 of the post-cure at 90 °C, the higher is the percentage of the polymeric chains with 538 higher molecular weight. This result contradicts the so far accepted unzipping mech-539 anism in thermally treated poly-cyanoacrylates [7]. In our study, a longer post-cure at 540 90 °C might favour the termination step in the post-polymerization of pECN with

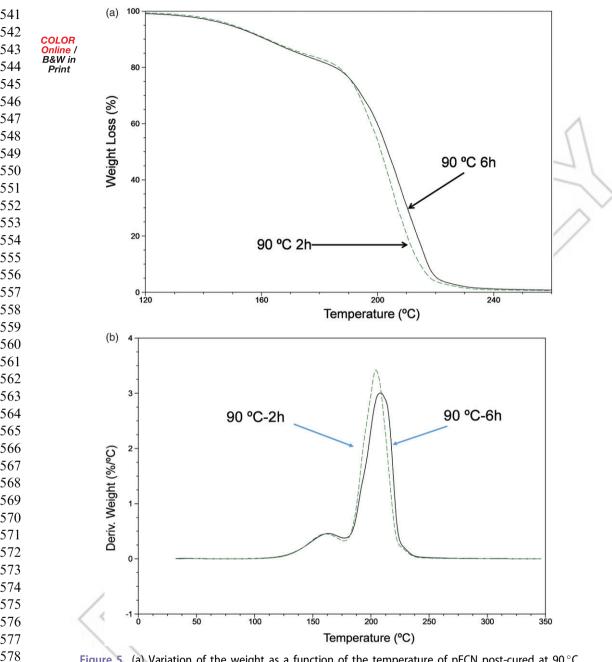


Figure 5. (a) Variation of the weight as a function of the temperature of pECN post-cured at 90 °C for 2 and 6 hours. TGA experiments. (b) Variation of the derivate weight as a function of the temperature of pECN post-cured at 90 °C for 2 and 6 hours. TGA experiments.

respect to the unzipping mechanism, being this result consistent with the small decrease in the post-polymerization enthalpy obtained in the first DSC heating run.

The post-cure treatment also affected the morphology of the pECN. Figure 6(a,b) shows the SEM micrographs at different magnifications of the pECN cured at 25 °C for 24 hours and post-cured at 90 °C for 6 hours. The non-post-cured pECN shows a

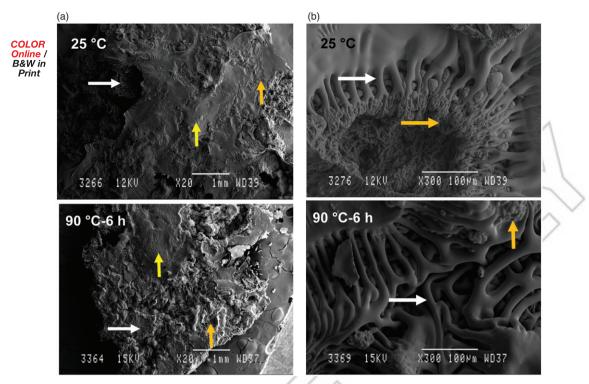


Figure 6. (a) SEM micrographs of pECN cured at 25 °C for 24 hours (up) and post-cured at 90 °C for 6 hours (down). White arrows: porosity. Orange arrows: wrinkle structures. Yellow arrows: compact crust. X 20. (b) SEM micrographs of pECN cured at 25 °C for 24 hours (up) and post-cured at 90 °C for 6 hours (down). White arrows: smooth areas. Orange arrows: rough areas. X 300.

heterogeneous structure, some porosity and wrinkle structures can be distinguished over a compact crust. The pECN post-cured at 90 °C shows a change of the topography with respect to that of pECN cured at 25 °C as the presence of compact crust is less important and the number of wrinkled structures increases (Figure 6(a)). On the other hand, Figure 6(b) shows the existence of two different structure domains in the pECN, one relatively smooth and another quite rough that may correspond to the two polymeric structures with different molecular weights evidenced by the TGA experiments. Figure 6(b) shows that the relatively smooth structure domains are dominant over the rough ones.

Figure 7 shows the TEM micrographs of the pECN cured at 25 °C for 24 hours and post-cured at 90 °C for 6 hours. The non-post-cured pECN shows abundant wrinkled structures and about 5 nm size nano-spheres (shown as black dots) which are disseminated into the continuous polymeric matrix. The existence of the nano-spheres in pECNs has been previously shown [2]. On the contrary, in the postcured pECN at 90 °C for 6 hours, the nano-spheres and the wrinkles disappear and a homogeneous and relatively smooth surface is observed.

3.3. Influence of the post-cure temperature of pECN on the adhesion properties

ECN was cured at 25 °C for 24 hours and later was post-cured at 45 and 90 °C for 2 hours. Table 5 shows that the shear strength value of the joint made with ECN

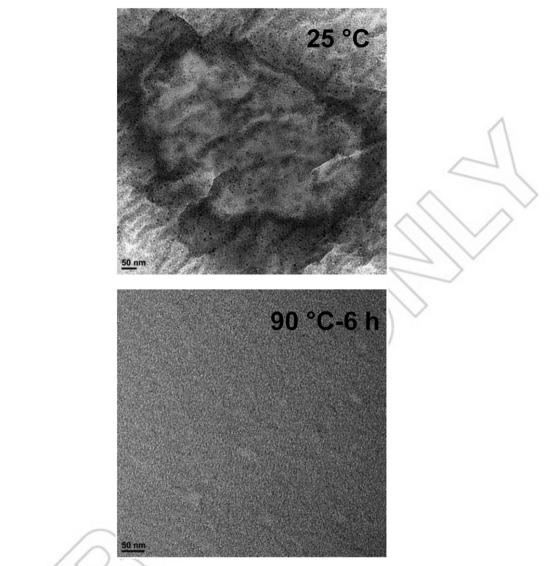


Figure 7. TEM micrographs of pECN cured at 25 °C for 24 hours (up) and post-cured at 90 °C for 6 hours (down).

cured at 25 °C is high and a mixed locus of failure (50% adhesion and 50% cohesive failure of the adhesive) is obtained, indicating that the cure of pECN was not complete. The post-cure of the adhesive joints decreases the shear strength more markedly by increasing the post-cure temperature, and always mixed loci of failure of adhesion and cohesive failure of the adhesive are observed; however, the increase of the post-cure temperature causes an increase of the percentage of cohesive failure in the adhesive which may explain the decrease in shear strength of the adhesive joints.

The decrease in the shear strength values of the post-cured joints made with ECN is in agreement with previously reported studies made with composite-ethyl cyanoacrylate [6] and steel-ethyl cyanoacrylate [7] joints. The decrease in the shear strength values of the joints made with ECN after being post-cured can be related to

Post-cure condition	Shear strength (MPa)	Locus of failure*
25 °C	3.21 ± 0.31	50% A+ 50% C
45 °C-2h	1.94 ± 0.07	40% A+ 60% C
90 °C-2h	1.40 ± 0.05	25% A + 75% C

Table 5. Single lap-shear strength values and loci of failure of aluminium 5754/ECN/aluminium 5754 joints.

changes in the physical and chemical structure of the cured pECN. The decrease in the shear strength value of the post-cured pECN can be ascribed to the degradation through an "unzipping" mechanism according to Scheme 2, i.e., the unzipping de-polymerization process initiated by the abstraction of one terminal proton in the "parent" polymer chain producing monomeric units by a retro-Michael cascade reaction [10]. The post-cure creates new low molecular weight polymeric chains that decrease the mechanical properties of pECN. As a consequence, the post-cure produces an increase in the T_g value, a decrease of the enthalpy and an increase in the temperature of post-polymerization. Furthermore, the post-cure changed the structure of the pECN and the molecular weight distribution of the two distinct polymeric domains.

4. Conclusions

The post-cure at 90 °C reduced the single lap-shear strength values of aluminum/ ECN/aluminum joints because of the termination step by SINT (slow-initiated, nonterminated) mechanism in zwitterionic polymerizations initiated by tertiary amines was favored. The increase in the post-cure temperature produced the de-polymerization of high molecular weight polymeric chains of pECN by an unzipping mechanism creating new low molecular weight polymeric chains that decreased the mechanical properties of pECN. The post-cure produced a slight increase in the T_g value, a decrease of the enthalpy and an increase in the temperature of post-polymerization. Furthermore, the post-cure changed the structure of the pECN and the molecular weight distribution of the two distinct polymeric domains. On the other hand, the topography of the pECN surface was affected by the post-cure favoring the creation of smooth structures and removing the wrinkles and the nano-spheres produced in pECN cured at 25°C for 24 hours. Finally, the influence of the temperature of the post-cure determined in greater extent the structure, thermal and topographical properties of pECN than the time of post-cure did.

Acknowledgments

The Authors thank Dr. E. Lorenzo from the Research Technical Services of Alicante University for carrying out the ¹H NMR experiments.

Disclosure statement

 O_2 No potential conflict of interest was reported by the authors.

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Funding

 This work was supported by the University of Alicante [VIGROB-001, VIGROB-173] and the O5 Spanish Ministerio de Economía y Competitividad [CTQ2015-66624-P].

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