

University of Nebraska - Lincoln DigitalCommons@University of Nebraska - Lincoln

Faculty Publications from the Department of
Electrical and Computer Engineering

Electrical & Computer Engineering, Department of

1-3-2018

Electron effective mass in Sn-doped monoclinic single crystal b-gallium oxide determined by mid-infrared optical Hall effect

Sean Knight

University of Nebraska-Lincoln, sean.knight@engr.unl.edu

Alyssa Mock

University of Nebraska-Lincoln

Rafal Korlacki

University of Nebraska-Lincoln

Vanya Darakchieva


Linköping University

Bo Monemar

Linköping University & Tokyo University of Agriculture and Technology

See next page for additional authors

Follow this and additional works at: <https://digitalcommons.unl.edu/electricalengineeringfacpub>

 Part of the [Computer Engineering Commons](#), and the [Electrical and Computer Engineering Commons](#)

Knight, Sean; Mock, Alyssa; Korlacki, Rafal; Darakchieva, Vanya; Monemar, Bo; Kumagai, Yoshinao; Goto, Ken; Higashiwaki, Masataka; and Schubert, Mathias, "Electron effective mass in Sn-doped monoclinic single crystal b-gallium oxide determined by mid-infrared optical Hall effect" (2018). *Faculty Publications from the Department of Electrical and Computer Engineering*. 452.
<https://digitalcommons.unl.edu/electricalengineeringfacpub/452>

This Article is brought to you for free and open access by the Electrical & Computer Engineering, Department of at DigitalCommons@University of Nebraska - Lincoln. It has been accepted for inclusion in Faculty Publications from the Department of Electrical and Computer Engineering by an authorized administrator of DigitalCommons@University of Nebraska - Lincoln.

Authors

Sean Knight, Alyssa Mock, Rafal Korlacki, Vanya Darakchieva, Bo Monemar, Yoshinao Kumagai, Ken Goto, Masataka Higashiwaki, and Mathias Schubert

Electron effective mass in Sn-doped monoclinic single crystal β -gallium oxide determined by mid-infrared optical Hall effect

Sean Knight,^{1,(a)} Alyssa Mock,¹ Rafał Korlacki,¹ Vanya Darakchieva,² Bo Monemar,^{2,3} Yoshinao Kumagai,^{3,4} Ken Goto,^{4,5} Masataka Higashiwaki,⁶ and Mathias Schubert^{1,2,7}

¹Department of Electrical and Computer Engineering, University of Nebraska-Lincoln, Lincoln, Nebraska 68588, USA

²Terahertz Materials Analysis Center, Department of Physics, Chemistry and Biology (IFM), Linköping University, SE 58183 Linköping, Sweden

³Institute of Global Innovation Research, Tokyo University of Agriculture and Technology, Koganei, Tokyo 184-8588, Japan

⁴Department of Applied Chemistry, Tokyo University of Agriculture and Technology, Koganei, Tokyo 184-8588, Japan

⁵Tamura Corporation, Sayama, Saitama 350-1328, Japan

⁶National Institute of Information and Communications Technology, Koganei, Tokyo 184-8795, Japan

⁷Leibniz Institute for Polymer Research, 01069 Dresden, Germany

(Received 30 October 2017; accepted 21 December 2017; published online 3 January 2018)

The isotropic average conduction band minimum electron effective mass in Sn-doped monoclinic single crystal β -Ga₂O₃ is experimentally determined by the mid-infrared optical Hall effect to be $(0.284 \pm 0.013)m_0$ combining investigations on (010) and $(\bar{2}01)$ surface cuts. This result falls within the broad range of values predicted by theoretical calculations for undoped β -Ga₂O₃. The result is also comparable to recent density functional calculations using the Gaussian-attenuation-Perdew-Burke-Ernzerhof hybrid density functional, which predict an average effective mass of $0.267m_0$. Within our uncertainty limits, we detect no anisotropy for the electron effective mass, which is consistent with most previous theoretical calculations. We discuss upper limits for possible anisotropy of the electron effective mass parameter from our experimental uncertainty limits, and we compare our findings with recent theoretical results. *Published by AIP Publishing.*

<https://doi.org/10.1063/1.5011192>

Single crystal gallium (III) oxide is a desirable material for optical and electronic applications due to its unique physical properties such as its transparent conducting nature and wide bandgap.¹ As a transparent conductor, Ga₂O₃ is useful for various types of transparent electrodes, for example, in flat panel displays,² smart windows,^{3,4} photovoltaic cells,³ and gas sensors.⁵ Due to its wide bandgap, Ga₂O₃ has a larger breakdown voltage than SiC and GaN, which makes it an excellent candidate for power devices.^{6–10} Among the five phases, the monoclinic β phase is the most stable and is expected to possess highly anisotropic properties which may prove useful for various applications.^{11,12} Tuning of the free charge carrier concentration to enhance the electrical conductivity has been achieved by Sn doping, for example, which is a well established technique.⁶

Precise knowledge of the free charge carrier properties is imperative for electronic and optoelectronic device design and operation. Experimentally determined results for effective mass, free charge carrier concentration, and mobility parameters are currently scarce for β -Ga₂O₃. Numerous theoretical investigations have yielded a wide range of values for the electron effective mass: from $0.12m_0$ to $0.39m_0$, where m_0 is the free electron mass.^{10,13–19} Most calculations predict only minimal anisotropy. However, recent Gaussian-attenuation-Perdew-Burke-Ernzerhof (Gau-PBE) hybrid density functional calculations predict slightly higher anisotropy.¹⁵ Using a combination of optical transmission and electrical Hall effect measurements,

the authors in Ref. 20 estimate a range of values for electron effective mass parameter along the **b** and **c** crystal directions to be $m_b^* = 0.5m_0$ to $1.0m_0$ and $m_c^* = 1.0m_0$ to $2.0m_0$, respectively. Electrical Hall effect measurements on β -Ga₂O₃ allow access to free charge carrier concentration and mobility,^{20–23} but this technique alone cannot resolve the effective mass parameter.

The optical Hall effect is a physical phenomenon exploited in our measurement technique, which employs generalized spectroscopic ellipsometry in combination with external magnetic fields to obtain the free charge carrier properties of semiconducting materials without electrical contacts.^{24–28} This technique measures the change in the polarization of light after the interaction with a sample due to a Lorentz force acting on the free charge carriers. In contrast with the electrical Hall effect, the optical Hall effect is capable of obtaining the effective mass, carrier concentration, mobility, and charge carrier type parameters simultaneously.

In this work, we experimentally determine the electron effective mass in Sn-doped monoclinic single crystal β -Ga₂O₃ by mid-infrared optical Hall effect (MIR-OHE) measurements. We compare our results to values reported in previous theoretical and experimental work, and we discuss the anisotropy of the effective electron mass parameter. Here, we find no discernible anisotropy and assume an isotropic average parameter. We discuss the amount of finite anisotropy that may remain hidden within our present experimental error bars for the effective mass to be potentially discovered by subsequent experiments.

^{a)}Electronic mail: sean.knight@engr.unl.edu

Two surface cuts, (010) and $(\bar{2}01)$, of Sn-doped single crystal β -Ga₂O₃ are investigated in this work. The crystals were grown using the edge-defined film-fed growth method by Tamura Corp. (Japan).^{29–31} The dimensions are (0.65 × 10 × 10) mm for the (010) surface and (0.65 × 10 × 15) mm for the $(\bar{2}01)$ surface. The optical response of β -Ga₂O₃ is governed by the monoclinic Cartesian dielectric function tensor.³² Here, the Cartesian direction \mathbf{x} is contained within the sample surface plane and is oriented along the propagation direction of light incident on the sample. The \mathbf{z} direction is oriented into the sample surface. The crystal directions in β -Ga₂O₃ are denoted \mathbf{a} , \mathbf{b} , and \mathbf{c} , where the monoclinic angle $\beta = 103.7^\circ$ lies between \mathbf{a} and \mathbf{c} .³³ We choose to align \mathbf{a} and \mathbf{b} along \mathbf{x} and $-\mathbf{z}$, respectively, such that \mathbf{c} lies within the \mathbf{x} - \mathbf{y} plane. For practicality, we introduce the direction \mathbf{c}^* parallel to \mathbf{y} , so that \mathbf{a} , \mathbf{b} , and \mathbf{c}^* form a pseudo-orthorhombic system. We define azimuth angle ϕ as a rotation about the \mathbf{z} axis for a given crystal axis orientation.³² For the (010) surface, $\phi = 0^\circ$ corresponds to \mathbf{a} aligned along \mathbf{x} . For the $(\bar{2}01)$ surface, $\phi = 0^\circ$ corresponds to \mathbf{b} aligned along \mathbf{y} .

Generalized spectroscopic ellipsometry is the measurement technique employed here to determine the free charge carrier properties of β -Ga₂O₃. Ellipsometric data are obtained using the Mueller matrix formalism.^{34,35} WVASE (J.A. Woollam Co., Inc.) is used to acquire and analyze the data. The MIR-OHE data are measured using a home-built Fourier transform infrared ellipsometer in the spectral range of 550 cm⁻¹–1500 cm⁻¹ with a resolution of 2 cm⁻¹. The home-built ellipsometer is capable of attaining the upper-left 3 × 3 block of the complete 4 × 4 Mueller matrix.²⁴ The MIR-OHE data are obtained at +6 T, 0 T, and -6 T, with the magnetic field parallel to the incoming infrared beam. Each surface cut is measured at one in-plane azimuth orientation. These measurements are performed at an angle of incidence of $\Phi_a = 45^\circ$ and at a temperature of $T = 300$ K. Additional measurements at zero field were performed at multiple in-plane orientations and included into the data analysis. Note that the anisotropy of the effective mass parameter is determined by the anisotropy in the plasma frequency as discussed further below. The anisotropy of the plasma frequency parameter is determined at zero field and multiple azimuth orientations. Hence, OHE data were only measured at one azimuth orientation for each sample.

In addition to the MIR-OHE measurements, zero magnetic field Mueller matrix data are measured at multiple azimuth orientations for each surface cut. The data are obtained using a commercially available MIR ellipsometer (IR-VASE, J.A. Woollam Co., Inc.) and the aforementioned home-built ellipsometer in the spectral range of 150 cm⁻¹–1500 cm⁻¹ with a resolution of 2 cm⁻¹. The zero magnetic field data are not shown here but are included in Ref. 32. These measurements are performed at $\Phi_a = 50^\circ$, 60° , and 70° and at room temperature.

Ellipsometry is an indirect measurement technique which requires a physical parameterized model be fit to experimental data to determine the desired parameters.³⁶ The model approach used here is very similar to that of Ref. 32. The two phase optical model consists of ambient air and β -Ga₂O₃ joined at a planar interface. The dielectric function tensor of β -Ga₂O₃ at long wavelengths consists of contributions from

optical phonon modes and free charge carriers. These contributions are modeled using the eigendielectric displacement vector summation approach described in Refs. 32 and 37. In this approach, contributions from individual dielectric resonances, in this case phonon modes and free charge carriers, are added to a high frequency dielectric constant tensor ϵ_∞ . The anharmonically broadened Lorentz oscillator model is used to represent phonon resonance contributions.³⁷ No substantial Drude contribution was detected in the off-diagonal components of the monoclinic dielectric function tensor.³² Thus, we employ an orthorhombic Drude model where three independent Drude contributions are added to the dielectric function response along axes \mathbf{a} (ϵ_{xx}), \mathbf{b} (ϵ_{zz}), and \mathbf{c}^* (ϵ_{yy}).

The magnetic field dependent free charge carrier contribution to the dielectric function tensor $\epsilon^{\text{FC}}(\omega)$ is described using the classical Drude formalism including the change induced by the Lorentz force^{24,28}

$$\epsilon^{\text{FC}}(\omega) = \frac{\omega_p^2}{-\omega^2 \mathbf{I} - i\omega\boldsymbol{\gamma} + i\omega \begin{pmatrix} 0 & -b_z & b_y \\ b_z & 0 & -b_x \\ -b_y & b_x & 0 \end{pmatrix}} \boldsymbol{\omega}_c \quad (1)$$

Here, \mathbf{I} is the identity matrix, and $\langle b_x, b_y, b_z \rangle$ are the scalar components of magnetic field vector \mathbf{B} , where each component is the projection along \mathbf{x} , \mathbf{y} , and \mathbf{z} , respectively. At zero magnetic field, the classical Drude model parameters include the screened plasma frequency tensor $\omega_p = \sqrt{Nq^2/\epsilon_0\epsilon_\infty m^*}$ and the plasma broadening tensor $\boldsymbol{\gamma} = q/\mu m^*$. These parameters depend on the free charge carrier properties which include effective mass m^* , free charge carrier volume density N , and mobility μ , where m^* and μ are the diagonal second rank tensors. In the isotropic average approximation of a given tensor, its values are replaced by an isotropic scalar and the corresponding unit matrix. The parameter ϵ_0 is the vacuum dielectric permittivity, and q is the elementary electric charge. At non-zero magnetic field, the cyclotron frequency tensor is $\boldsymbol{\omega}_c = q|\mathbf{B}|/m^*$.

Figures 1 and 2 show experimental and best-match model calculated MIR-OHE difference spectra for the (010) and $(\bar{2}01)$ surface cut of β -Ga₂O₃, respectively. The MIR-OHE signals are the strongest in the vicinity of the zero magnetic field reflectance minima for samples with sufficient free charge carrier contributions.³⁸ For the samples investigated here, the reflectance minimum and MIR-OHE signal appear at the edge of the reststrahlen band at around 900 cm⁻¹. Due to the coupling of longitudinal optical phonon modes and free charge carriers, the so called longitudinal phonon plasmon (LPP) modes are now experimentally observed. Since the spectral locations of the reflectance minima are governed by the LPP modes, the strongest MIR-OHE signatures occur in the vicinity of the highest frequency LPP modes, which are indicated by the vertical dotted lines in Figs. 1 and 2.³² The unique shape of the signal is governed by changes in the dielectric function tensor due to phonon modes near this spectral range. We note that until this report, MIR-OHE difference data between positive and negative magnetic fields were only seen in the off-block-diagonal Mueller matrix elements (i.e., M_{13} , M_{23} , M_{31} , and M_{32}).³⁸

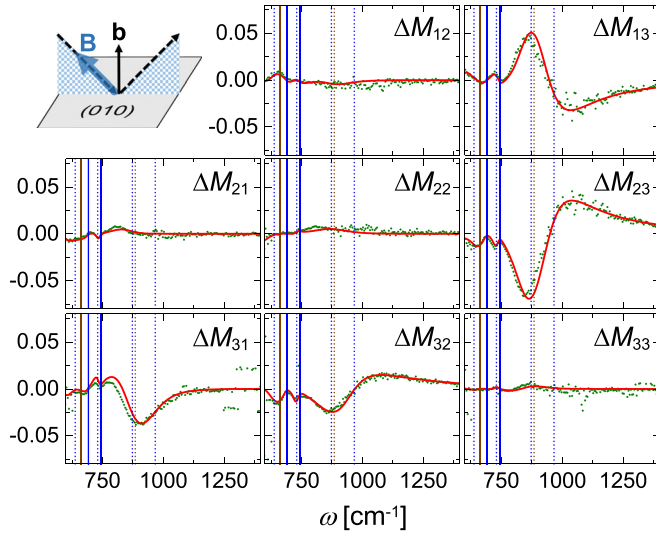


FIG. 1. MIR-OHE experimental (green dots) and best-match model calculated (solid red lines) Mueller matrix difference spectra ($\Delta M_{ij} = M_{ij}(+6\text{ T}) - M_{ij}(-6\text{ T})$) for the (010) cut $\beta\text{-Ga}_2\text{O}_3$ sample at azimuth angle $\phi = 112.5^\circ$. All measurements are performed at temperature $T = 300\text{ K}$ and at an angle of incidence of $\Phi_a = 45^\circ$. The magnetic field \mathbf{B} is parallel to the incoming infrared beam. Taken from Ref. 32, vertical lines signify the wave numbers of LPP (dotted lines) and transverse optical phonon modes (solid lines) polarized in the \mathbf{a} - \mathbf{c} plane (blue) and along the \mathbf{b} axis (brown).

However, for the (010) surface (Fig. 1), a small difference is seen in the on-block-diagonal elements (i.e., M_{12} , M_{21} , M_{22} , and M_{33}). This is due to the dielectric function tensor for the (010) surface at azimuth angle $\phi = 112.5^\circ$ possessing non-zero off-diagonal elements at $|\mathbf{B}| = 0$. In contrast, the (201) surface at azimuth angle $\phi = 181.7^\circ$ possesses negligible off-diagonal tensor components at $|\mathbf{B}| = 0$ since the \mathbf{a} - \mathbf{c} plane lies within the plane of incidence.

Assuming separate sets of isotropic free charge carrier properties for the (010) and $(\bar{2}01)$ surfaces, the model parameters are fit to the MIR-OHE difference data and zero magnetic field data simultaneously. The final best-match model fit is presented in Figs. 1 and 2, and the resulting parameters are shown in Table I. The zero magnetic field data alone

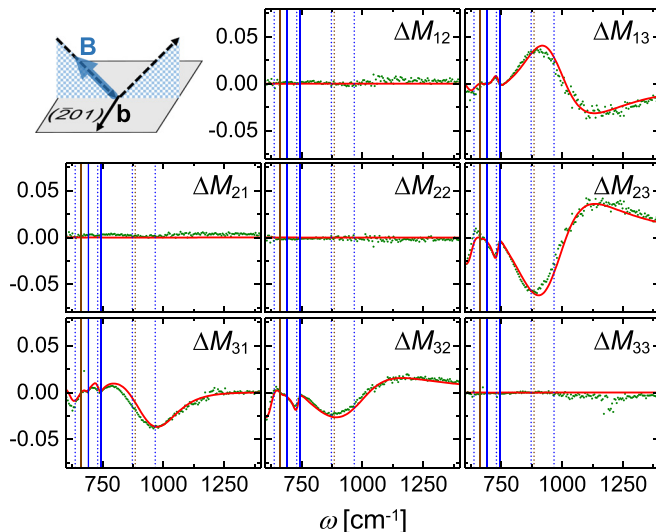


FIG. 2. Same as Fig. 1 for the $(\bar{2}01)$ cut $\beta\text{-Ga}_2\text{O}_3$ sample at azimuth angle $\phi = 181.7^\circ$.

TABLE I. Results for isotropic average free charge carrier properties in $\beta\text{-Ga}_2\text{O}_3$. The theoretical Gau-PBE average effective mass parameter is calculated by taking the harmonic mean of the predicted values along the \mathbf{a} , \mathbf{b} , and \mathbf{c} crystal axes, which can be found in Table II. Error bars shown correspond to the 90% confidence interval within the best-match model data analysis.

Method	Parameter	Value
MIR-OHE for the (010) surface ^a	m^*	$(0.284 \pm 0.013)m_0$
	N	$(4.2 \pm 0.1) \times 10^{18}\text{ cm}^{-3}$
	μ	$(44 \pm 2)\text{ cm}^2/\text{V s}$
MIR-OHE for the $(\bar{2}01)$ surface ^a	m^*	$(0.283 \pm 0.011)m_0$
	N	$(5.9 \pm 0.1) \times 10^{18}\text{ cm}^{-3}$
	μ	$(43 \pm 1)\text{ cm}^2/\text{V s}$
Gau-PBE ^b	m_{avg}^*	$0.267m_0$

^aThis work.

^bTheory, Ref. 15.

would allow one to determine the isotropic plasma frequency ω_p and broadening γ , which are functions of m^* , N , and μ . The addition of the MIR-OHE data in the analysis allows m^* , N , and μ to be accurately resolved. In order to improve the best match between model calculated and experimental data, the model for the (010) and $(\bar{2}01)$ surfaces must be assigned independent sets of free charge carrier properties to account for a potentially different Sn dopant distribution and activation. However, each surface shares the same phonon mode parameters and ϵ_∞ . Since the phonon mode parameters and ϵ_∞ in Ref. 32 were derived assuming identical Drude parameters for the (010) and $(\bar{2}01)$ surfaces, these quantities were also included in the best-match model fit to properly determine the free charge carrier properties. The analysis confirms the expected n-type conductivity for each surface cut. The azimuth angles $\phi = 112.5^\circ$ and $\phi = 181.7^\circ$ for the (010) and $(\bar{2}01)$ surface, respectively, are determined by applying the zero magnetic field model to the zero-field MIR-OHE measurement.

The electron effective mass parameters experimentally determined in this work are $m^* = (0.284 \pm 0.013)m_0$ for the (010) surface and $m^* = (0.283 \pm 0.011)m_0$ for the $(\bar{2}01)$ surface. These fall within the broad range of values reported for various density functional theory calculations: $(0.12$ to $0.13)m_0$,¹³ $(0.22$ to $0.30)m_0$,¹⁵ $(0.23$ to $0.24)m_0$,¹⁴ $(0.26$ to $0.27)m_0$,¹⁰ $(0.27$ to $0.28)m_0$,^{16,17} $(0.34)m_0$,¹⁸ and $(0.39)m_0$.¹⁹ The best-match model parameter results for the isotropically averaged mobility parameter μ for the two surfaces compare well with values determined previously by electrical Hall effect measurements for samples with similar free electron densities.²¹ Our best-match model parameter results for the electron density N are in excellent agreement with the nominal Sn density of $1.7 \times 10^{18}\text{ cm}^{-3}$ provided by the crystal manufacturer. The electron density obtained by electrical Hall effect measurements is approximately equal to the doped Sn density.

Ueda *et al.* estimated electron effective mass parameters $m_b^* = 0.5m_0$ to $1.0m_0$ and $m_c^* = 1.0m_0$ to $2.0m_0$ from optical transmission and electrical Hall effect measurements.²⁰ There is a rather large discrepancy between the effective mass parameters reported in this work and by Ueda *et al.* A critical discussion of the results by Ueda *et al.* was given by

Parisini and Fornari³⁹ suggesting revision of data analysis in Ref. 20.

The anisotropy of m^* may be defined by considering the ratios (m_a^*/m_b^*) and $(m_b^*/m_{c^*}^*)$. These quantities are comparable to the squares of the ratios of the plasma frequencies determined by the orthogonal Drude model approximation, via $(\omega_{p,b}/\omega_{p,a})^2 = (m_a^*/m_b^*)$ and $(\omega_{p,c^*}/\omega_{p,b})^2 = (m_b^*/m_{c^*}^*)$. The parameters $\omega_{p,a}$, $\omega_{p,b}$, and ω_{p,c^*} are the plasma frequencies corresponding to the **a**, **b**, and **c**^{*} directions, respectively. Information about the plasma frequencies can be gathered without the use of magnetic fields. Generalized ellipsometry measurements at zero field and at multiple sample azimuth orientations, for both the (010) and (201) surfaces, were taken and subsequently analyzed simultaneously with the MIR-OHE data. This approach provided sufficient sensitivity to determine the anisotropy of the free charge carrier parameters. The resulting effective mass parameters are shown in Table II. The ratios are $(m_a^*/m_b^*) = (1.02_{-0.28}^{+0.38})$ and $(m_b^*/m_{c^*}^*) = (0.99_{-0.27}^{+0.37})$ for the (010) surface, where the upper and lower scripted numbers refer to the upper and lower uncertainty limits of the effective mass parameter ratio. The upper/lower limits come from taking the ratios within the maximum/minimum parameter deviations in the numerator and denominator using the mass parameters and error bars as shown in Table II. For the (201) surface, the ratios are $(m_a^*/m_b^*) = (1.07_{-0.25}^{+0.33})$ and $(m_b^*/m_{c^*}^*) = (0.89_{-0.22}^{+0.29})$. Our findings suggest a small deviation from isotropy, however, which is well within the uncertainty limits for both surfaces investigated. Nonetheless, the possibility of a small anisotropy would be consistent with recent theoretical investigations.^{10,13,14,17}

Yamaguchi calculated the electron effective mass ratios of $(m_a^*/m_b^*) = 0.96$ and $(m_b^*/m_{c^*}^*) = 1.07$ at the Γ point, where m_a^* , m_b^* , and $m_{c^*}^*$ are the diagonal Cartesian effective mass tensor components.¹⁴ Furthmüller and Bechstedt predict ratios of $(m_a^*/m_b^*) = 1.01$ and $(m_b^*/m_{c^*}^*) = 1.03$.¹⁰ He *et al.* find ratios of $(m_a^*/m_b^*) = 0.95$ and $(m_b^*/m_{c^*}^*) = 1.05$.¹³ Recent

TABLE II. Results for anisotropic free charge carrier properties in β -Ga₂O₃. Error bars shown correspond to the 90% confidence interval within the best-match model data analysis.

Method	Parameter	Value
MIR-OHE for the (010) surface ^a	m_a^*	$(0.288 \pm 0.044)m_0$
	m_b^*	$(0.283 \pm 0.046)m_0$
	$m_{c^*}^*$	$(0.286 \pm 0.044)m_0$
	N	$(4.1 \pm 0.3) \times 10^{18} \text{ cm}^{-3}$
	μ_a	$(45 \pm 4) \text{ cm}^2/\text{V s}$
	μ_b	$(42 \pm 4) \text{ cm}^2/\text{V s}$
	μ_{c^*}	$(42 \pm 3) \text{ cm}^2/\text{V s}$
MIR-OHE for the (201) surface ^a	m_a^*	$(0.295 \pm 0.039)m_0$
	m_b^*	$(0.276 \pm 0.037)m_0$
	$m_{c^*}^*$	$(0.311 \pm 0.044)m_0$
	N	$(6.0 \pm 0.5) \times 10^{18} \text{ cm}^{-3}$
	μ_a	$(44 \pm 3) \text{ cm}^2/\text{V s}$
	μ_b	$(44 \pm 3) \text{ cm}^2/\text{V s}$
	μ_{c^*}	$(41 \pm 3) \text{ cm}^2/\text{V s}$
Gau-PBE ^b	m_a^*	$0.224m_0$
	m_b^*	$0.301m_0$
	$m_{c^*}^*$	$0.291m_0$

^aThis work.

^bTheory, Ref. 15.

density functional calculations using the Gau-PBE approach predict ratios of $(m_a^*/m_b^*) = 0.74$ and $(m_b^*/m_{c^*}^*) = 1.03$ for undoped β -Ga₂O₃.¹⁵ We note that the theoretical results reported so far are inconsistent; however, all theoretical predicted ratios could fall within our experimental error bars and no conclusive statement about a finite anisotropy of the effective electron mass parameter can be made at this point. Within our uncertainty limits, the mobility parameter is found to be essentially isotropic. This is consistent with previous theoretical investigations for intrinsic mobility⁴⁰ and experimental Hall effect measurements using the bar method⁴¹ and using the Van der Pauw method.^{22,42} A nearly isotropic mobility is also reported for electron channel mobility in silicon-doped Ga₂O₃ metal-oxide-semiconductor field-effect transistors (MOSFETs).⁴³

Electron effective mass values for monoclinic oxides similar to β -Ga₂O₃ have been calculated by density functional methods. Monoclinic θ -Al₂O₃, an isomorph of β -Ga₂O₃, is predicted to possess similar effective mass values and anisotropy. The values are calculated to be $0.41m_0$, $0.41m_0$, and $0.37m_0$ along the [100], [010], and [001] crystallographic directions, respectively.⁴⁴ The effective mass is virtually isotropic regardless of the structural complexity of the low symmetry monoclinic structure. In contrast, monoclinic transition metal oxides ZrO₂ and HfO₂ are predicted to have much higher effective masses and stronger anisotropy.⁴⁵ The predicted values for ZrO₂ are $3.1m_0$, $3.2m_0$, and $3.7m_0$ along the x' , y' , and z' , respectively, where z' is analogous to c^* . The predicted values for HfO₂ are $8.2m_0$, $1.6m_0$, and $1.0m_0$ along the same respective directions.

This work was supported by the Swedish Research Council (VR) under Grant Nos. 2013-5580 and 2016-00889, the Swedish Governmental Agency for Innovation Systems (VINNOVA) under the VINNMER international qualification program, Grant No. 2011-03486, the Swedish Government Strategic Research Area in Materials Science on Functional Materials at Linköping University, Faculty Grant SFO Mat LiU No. 2009 00971, and the Swedish Foundation for Strategic Research (SSF), under Grant Nos. FL12-0181 and RIF14-055. The authors further acknowledge financial support by the University of Nebraska-Lincoln, the J. A. Woollam Co., Inc., the J. A. Woollam Foundation, and the National Science Foundation (Award Nos. MRSEC DMR 1420645, CMMI 1337856, and EAR 1521428).

¹S. Stepanov, V. Nikolaev, V. Bougrov, and A. Romanov, *Rev. Adv. Mater. Sci.* **44**, 63 (2016).

²U. Betz, M. K. Olsson, J. Marthy, M. Escola, and F. Atamny, *Surf. Coat. Technol.* **200**, 5751 (2006).

³C. G. Granqvist, *Handbook of Inorganic Electrochromic Materials* (Elsevier, 1995).

⁴D. Gogova, A. Iossifova, T. Ivanova, Z. Dimitrova, and K. Gesheva, *J. Cryst. Growth* **198–199**, 1230 (1999).

⁵F. Réti, M. Fleischer, H. Meixner, and J. Giber, *Sens. Actuators, B* **19**, 573 (1994).

⁶M. Higashiwaki, K. Sasaki, H. Murakami, Y. Kumagai, A. Koukitu, A. Kuramata, T. Masui, and S. Yamakoshi, *Semicond. Sci. Technol.* **31**, 034001 (2016).

⁷J. F. Wager, *Science* **300**, 1245 (2003).

⁸C. Sturm, R. Schmidt-Grund, C. Kranert, J. Furthmüller, F. Bechstedt, and M. Grundmann, *Phys. Rev. B* **94**, 035148 (2016).

- ⁹C. Sturm, J. Furthmüller, F. Bechstedt, R. Schmidt-Grund, and M. Grundmann, *APL Mater.* **3**, 106106 (2015).
- ¹⁰J. Furthmüller and F. Bechstedt, *Phys. Rev. B* **93**, 115204 (2016).
- ¹¹R. Roy, V. G. Hill, and E. F. Osborn, *J. Am. Chem. Soc.* **74**, 719 (1952).
- ¹²H. H. Tippins, *Phys. Rev.* **140**, A316 (1965).
- ¹³H. He, M. A. Blanco, and R. Pandey, *Appl. Phys. Lett.* **88**, 261904 (2006).
- ¹⁴K. Yamaguchi, *Solid State Commun.* **131**, 739 (2004).
- ¹⁵A. Mock, R. Korlacki, C. Briley, V. Darakchieva, B. Monemar, Y. Kumagai, K. Goto, M. Higashiwaki, and M. Schubert, “Band-to-band transitions, selection rules, effective mass and exciton binding energy parameters in monoclinic β -Ga₂O₃,” *Phys. Rev. B* (submitted).
- ¹⁶J. B. Varley, J. R. Weber, A. Janotti, and C. G. Van de Walle, *Appl. Phys. Lett.* **97**, 142106 (2010).
- ¹⁷H. Peelaers and C. G. V. de Walle, *Phys. Status Solidi B* **252**, 828 (2015).
- ¹⁸H. He, R. Orlando, M. A. Blanco, R. Pandey, E. Amzallag, I. Baraille, and M. Rérat, *Phys. Rev. B* **74**, 195123 (2006).
- ¹⁹M.-G. Ju, X. Wang, W. Liang, Y. Zhao, and C. Li, *J. Mater. Chem. A* **2**, 17005 (2014).
- ²⁰N. Ueda, H. Hosono, R. Waseda, and H. Kawazoe, *Appl. Phys. Lett.* **71**, 933 (1997).
- ²¹N. Suzuki, S. Ohira, M. Tanaka, T. Sugawara, K. Nakajima, and T. Shishido, *Phys. Status Solidi C* **4**, 2310 (2007).
- ²²K. Irmscher, Z. Galazka, M. Pietsch, R. Uecker, and R. Fornari, *J. Appl. Phys.* **110**, 063720 (2011).
- ²³E. G. Villora, K. Shimamura, Y. Yoshikawa, T. Ujiie, and K. Aoki, *Appl. Phys. Lett.* **92**, 202120 (2008).
- ²⁴P. Kühne, C. M. Herzinger, M. Schubert, J. A. Woollam, and T. Hofmann, *Rev. Sci. Instrum.* **85**, 071301 (2014).
- ²⁵M. Schubert, T. Hofmann, and C. M. Herzinger, *J. Opt. Soc. Am. A* **20**, 347 (2003).
- ²⁶T. Hofmann, T. Chavdarov, V. Darakchieva, H. Lu, W. J. Schaff, and M. Schubert, *Phys. Status Solidi C* **3**, 1854 (2006).
- ²⁷T. Hofmann, V. Darakchieva, B. Monemar, H. Lu, W. Schaff, and M. Schubert, *J. Electron. Mater.* **37**, 611 (2008).
- ²⁸M. Schubert, P. Kühne, V. Darakchieva, and T. Hofmann, *J. Opt. Soc. Am. A* **33**, 1553 (2016).
- ²⁹H. Aida, K. Nishiguchi, H. Takeda, N. Aota, K. Sunakawa, and Y. Yaguchi, *Jpn. J. Appl. Phys., Part 1* **47**, 8506 (2008).
- ³⁰K. Sasaki, A. Kuramata, T. Masui, E. G. Villora, K. Shimamura, and S. Yamakoshi, *Appl. Phys. Express* **5**, 035502 (2012).
- ³¹K. Shimamura and E. Villora, *Acta Phys. Pol., A* **124**, 265 (2013).
- ³²M. Schubert, R. Korlacki, S. Knight, T. Hofmann, S. Schöche, V. Darakchieva, E. Janzén, B. Monemar, D. Gogova, Q.-T. Thieu, R. Togashi, H. Murakami, Y. Kumagai, K. Goto, A. Kuramata, S. Yamakoshi, and M. Higashiwaki, *Phys. Rev. B* **93**, 125209 (2016).
- ³³S. Geller, *J. Chem. Phys.* **33**, 676 (1960).
- ³⁴H. Fujiwara, *Spectroscopic Ellipsometry: Principles and Applications* (John Wiley & Sons, 2007).
- ³⁵M. Schubert, *Phys. Rev. B* **53**, 4265 (1996).
- ³⁶R. M. Azzam and N. M. Bashara, *Ellipsometry and Polarized Light* (North-Holland Publ. Co., Amsterdam, 1984).
- ³⁷A. Mock, R. Korlacki, S. Knight, and M. Schubert, *Phys. Rev. B* **95**, 165202 (2017).
- ³⁸S. Schöche, P. Kühne, T. Hofmann, M. Schubert, D. Nilsson, A. Kakanakova-Georgieva, E. Janzén, and V. Darakchieva, *Appl. Phys. Lett.* **103**, 212107 (2013).
- ³⁹A. Parisini and R. Fornari, *Semicond. Sci. Technol.* **31**, 035023 (2016).
- ⁴⁰Y. Kang, K. Krishnaswamy, H. Peelaers, and C. G. Van de Walle, *J. Phys.: Condens. Matter* **29**, 234001 (2017).
- ⁴¹E. G. Villora, K. Shimamura, Y. Yoshikawa, K. Aoki, and N. Ichinose, *J. Cryst. Growth* **270**, 420 (2004).
- ⁴²C. Golz, “Ga₂O₃ conductivity anisotropy analysed by van-der-Pauw measurements,” in 2nd International Workshop on Gallium Oxide and Related Materials (2017).
- ⁴³M. H. Wong, K. Sasaki, A. Kuramata, S. Yamakoshi, and M. Higashiwaki, *Jpn. J. Appl. Phys., Part 1* **55**, 1202B9 (2016).
- ⁴⁴J. E. Medvedeva, E. N. Teasley, and M. D. Hoffman, *Phys. Rev. B* **76**, 155107 (2007).
- ⁴⁵T. Perevalov, A. Shaposhnikov, K. Nasyrov, D. Gritsenko, V. Gritsenko, and V. Tapilin, *Defects in High-k Gate Dielectric Stacks*, NATO Science Series II: Mathematics, Physics and Chemistry Vol. 220 (Springer, Dordrecht, 2006).