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## Anomalous magnetic hysteresis in an amorphous Nd<sub>54</sub>Co<sub>36</sub>B<sub>10</sub> alloy

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The hysteresis and crystallization behavior of an amorphous  $Nd_{54}Co_{36}B_{10}$  alloy have been examined over a range of temperatures. Crystallization studies show a glass transition temperature around 210 °C, two crystallization peaks at around 220 and 360 °C, and two endothermic peaks at around 530 and 600 °C, respectively. ac susceptibility measurements show an ordering temperature of 38 K for the amorphous phase. Four additional magnetic phases have been observed in the crystallized samples with ordering temperatures 17, 45, 90, and 190 K, respectively. Magnetic measurements on melt-spun ribbons reveal a coercivity of 6 kOe at 4.2 K. The coercivity at 4.2 K of a sample heat treated at the first crystallization temperature is increased to 8 kOe but is reduced gradually upon annealing at higher temperatures and is only 2.5 kOe in a sample annealed at a temperature close to the second endothermic peak.

#### INTRODUCTION

Metallic glasses based on rare-earth elements show a magnetic behavior<sup>1,2</sup> which is quite different from that observed in transition-metal-metalloid glasses. The former exhibit a much larger anisotropy which leads to large coercivities at cryogenic temperatures. Although there are several models<sup>3–5</sup> discussing this anomalous magnetic hysteresis, there is still a confusion about the origin of the observed high coercive fields.

In this study we examine the thermal and magnetic properties of Nd<sub>54</sub>Co<sub>36</sub>B<sub>10</sub> in both the amorphous and crystallized state. ac susceptibility measurements are used to determine the ordering temperatures of the amorphous and crystalline phases that are present after crystallization. The hysteresis loops have been measured down to liquid helium temperature and their dependence on crystallization is examined. The observed high coercive fields are discussed in the light of various existing models.

#### EXPERIMENT

Samples of  $Nd_{54}Co_{36}B_{10}$  were prepared from high-purity elements by arc-melting the constituents several times in argon atmosphere. The as-cast samples were then rapidly quenched into ribbons using the melt-spinning technique. The heat treatment of the samples was done either by using the differential thermal analyzer or by sealing the samples wrapped in Ta foils under vacuum in quartz tubes and heating them inside a furnace equipped with a temperature controller.

The crystallization studies were made with differential scanning calorimetry using a Dupont 900 system with a heating rate of 30 K/min. The hysteresis loops and high field (80 kOe) magnetization measurements were made at cryogenic temperatures using a vibrating sample magnetometer equipped with a vari-temp Janis dewar. The magnetic susceptibility was measured with an ac technique in a rms field of 0.01 Oe and a frequency of 280 Hz.

#### **RESULTS**

Figure 1 shows the crystallization data on an asquenched  $Nd_{54}Co_{36}B_{10}$  sample using the differential scanning calorimetry (DSC) method. A glass transition temperature is observed at about 210 °C, two sharp exothermic peaks around 220 and 360 °C, and two endothermic peaks at about 530 and 600 °C, respectively. The exothermic peaks are characteristic of the crystallization of the amorphous phase.

The hysteresis loops and magnetization curves of an amorphous sample measured at cryogenic temperatures are shown in Fig. 2. A coercive force of about 6 kOe is observed at 4.2 K. The variation of  $H_c$  with temperature as deduced from Fig. 2 is shown in Fig. 3. The coercive force decreases rapidly with increasing temperature and is only about 250 Oe at 25 K which is well below the ordering temperature

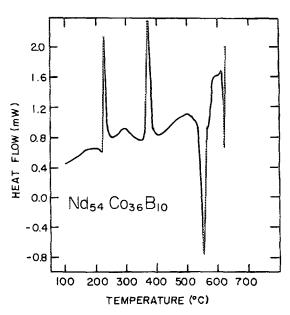


FIG. 1. Differential scanning calorimetry data on as-quenched  $Nd_{54}Co_{36}B_{10}$  ribbons.

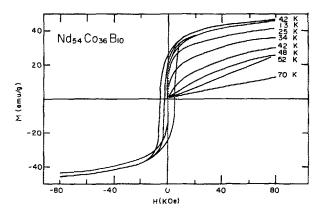


FIG. 2. Magnetization curves and hysteresis loops of amorphous  $Nd_{54}Co_{36}B_{10}$  ribbons.

estimated to be around 40 K. An attempt was made to study the effect of heat treatment on the magnetic properties of ribbons. Four melt-spun samples were heat treated in argon gas for 15 min at temperatures of 240, 370, 550, and 630 °C, respectively which are close to the peaks observed in the DSC studies. Figure 4 shows the variation of  $H_c$  measured at 4.2 K with the annealing temperature. The coercive force is increased to 8 kOe for a sample heat treated at the first crystallization temperature and then is decreased upon annealing at higher temperatures reaching only the value of 2.5 kOe in a sample heat treated at a temperature around the second endothermic peak.

ac susceptibility measurements have been performed on amorphous as well as heat-treated ribbons. The as-quenched sample exhibit a single sharp peak at 38 K (Fig. 5) which is close to that observed in the amorphous Nd<sub>60</sub>Co<sub>40</sub> ribbons. After the first crystallization, two new phases are formed with ordering temperatures 17 K and 45 K. At the second crystallization these two crystalline phases and the amorphous phase disappear and two other phases are formed with ordering temperatures around 90 and 190 K. Further an-

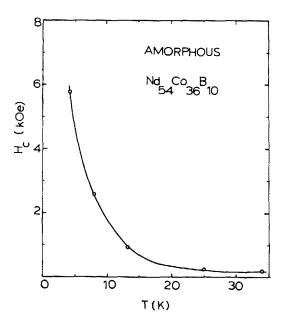


FIG. 3. Temperature dependence of coercivity in an amorphous sample.

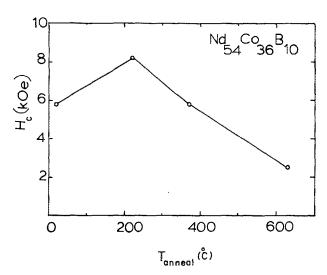


FIG. 4. Coercive force as function of annealing temperature.

nealing at 630 °C resulted in the appearance of a new phase with an ordering temperature about 115 K (Table I).

#### DISCUSSION

The high-field magnetization measurements on the melt-spun ribbons at different temperatures showed that a field of 80 kOe is not enough to saturate the sample indicating the presence of a large anisotropy. The law of approach to saturation<sup>6</sup> is used to estimate both the saturation magnetization  $M_s$  and the anisotropy constant K at 4.2 K. The values of 48 emu/g and  $1.2 \times 10^7$  erg/cm<sup>3</sup> have been obtained for  $M_s$  and K, respectively. The high-field susceptibility appears to be the same at different temperatures and this suggests that the anisotropy does not change much with temperature in the temperature range studied.

In order to determine the mechanism leading to the observed magnetic hysteresis in the amorphous samples an attempt is made to interpret the experimental data in the light of some of the existing models.<sup>3-5</sup> The single domain

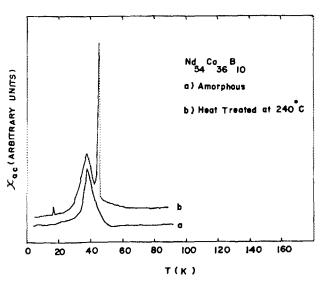


FIG. 5. ac susceptibility measurements on an amorphous sample and a crystallized sample heat treated at 240 °C for 15 min.

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TABLE I. Ordering temperatures of magnetic phases determined by ac susceptibility measurements on melt-spun and crystallized samples.

Heat Treatment	Ordering Temperature (K
As-quenched	38
15 min at 240 °C	17, 45
15 min at 370 °C	90, 190
15 min at 630 °C	115

particle theory<sup>3</sup> predicts a coercive force  $H_c$  for noninteracting single domain particles equal to  $K/M_s$ . After substituting the values determined from the law of approach to saturation a value of  $H_c = 36$  kOe is found which is much larger than the value obtained by extrapolating the data of Fig. 3 to T = 0 which is only  $\sim 12$  kOe. The random anisotropy model of Callen et al.4 took into account the interaction between the domains and showed that the exchange interaction can cause a decrease in the coercivity and an increase in the reduced remanence up to the value of 1. From the observed reduced remanence one can find the value of the exchange interaction and use it to predict  $H_c$ . A reduced remanent value of  $\sim 0.58$  was found for the amorphous ribbon at 4.2 K leading to a value of  $H_c = 27$  kOe which is still larger than the experimentally determined value. A domain-wall pinning model by Paul<sup>5</sup> has taken into account the difference between the magnetic properties of the magnetic matrix and defect and predicted the coercivity as a function of the dimensionless constants E and F where  $E = A_2K_2/A_1K_1$ , and  $F = A_2 M_2 / A_1 M_1$ . K is the anisotropy constant, A is the exchange constant, M is the magnetization and the subscripts 1 and 2 refer to the matrix and defect, respectively. The estimated value of coercivity at T = 0 K can be explained with Paul's model using values for E and F in the range of 0.25– 0.50 assuming a defect size equal to the domain wall thickness. These values are not very unreasonable for amorphous magnetic materials where one expects large fluctuations in anisotropy and exchange interactions.

The shape of the ac susceptibility peaks provides infor-

mation about the kind of magnetic phases present in the sample. The sharp ac peaks observed in the as-quenched samples may indicate sperimagnetic type of ordering. This is in contrast to the leveling off of ac observed below  $T_C$  for ferromagnetic type transitions. The phases with ordering temperatures of 17 and 115 K may be associated with Nd and NdCo<sub>2</sub>, respectively. The higher Curie temperature phase of 190 K has been also observed in Nd-Co alloys and has been associated with a Nd-Co oxide. The phase with the ordering temperature of 45 K cannot be identified with any of the existing phases in the Nd-Co system and it is probably a ternary Nd-Co-B phase. This phase may be highly anisotropic as indicated from the increase in coercivity after the first crystallization.

The influence of crystallization on hysteresis is presently studied more analytically. Lorentz microscopy studies will be made down to cryogenic temperatures to verify the domain wall pinning hypothesis and identify the defect responsible for the observed hysteresis.

#### **ACKNOWLEDGMENTS**

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