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Enhanced Photocatalytic Activity of Titania by Co-Doping with Mo and W

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Received: 22 October 2018; Accepted: 19 November 2018; Published: 6 December 2018



Abstract: Various W and Mo co-doped titanium dioxide (TiO₂) materials were obtained through the EISA (Evaporation-Induced Self-Assembly) method and then tested as photocatalysts in the degradation of 4-chlorophenol. The synthesized materials were characterized by thermogravimetric analysis (TGA), Fourier transform infrared (FTIR) spectroscopy, X-ray diffraction (XRD), Raman spectroscopy (RS), N₂ physisorption, UV-vis diffuse reflectance spectroscopy (DRS), X-ray photoelectron spectroscopy (XPS), and transmission electron microscopy (TEM). The results showed that the W-Mo-TiO₂ catalysts have a high surface area of about 191 m²/g, and the presence of an anatase crystalline phase. The co-doped materials exhibited smaller crystallite sizes than those with one dopant, since the crystallinity is inhibited by the presence of both species. In addition, tungsten and molybdenum dopants are distributed and are incorporated into the anatase structure of TiO₂, due to changes in red parameters and lattice expansion. Under our experimental conditions, the co-doped TiO₂ catalyst presented 46% more 4-chlorophenol degradation than Degussa P25. The incorporation of two dopant cations in titania improved its photocatalytic performance, which was attributed to a cooperative effect by decreasing the recombination of photogenerated charges, high radiation absorption capacity, high surface areas, and low crystallinity. When TiO2 is co-doped with the same amount of both cations (1 wt.%), the highest degradation and mineralization (97% and 74%, respectively) is achieved. Quinones were the main intermediates in the 4-chlorophenol oxidation by W-Mo-TiO₂ and 1,2,4-benzenetriol was incompletely degraded.

Keywords: W-Mo dopants; titanium dioxide; nanoparticles; photocatalytic activity

1. Introduction

Among the advanced oxidation processes (AOPs), heterogeneous photocatalysis is considered as an efficient method for the degradation of organic pollutants in water and air [1,2]. Numerous semiconductors have been investigated as photocatalysts. Among the semiconductors used, titanium dioxide (TiO_2) is the most promising and widely studied material for photocatalytic applications due to its chemical stability, high efficiency, photostability, high oxidizing power, abundance, nontoxicity, and low cost [3,4]. The main quality of TiO_2 is attributed to oxidative power of hydroxyl radicals generated when the electrons are photoexcited by UV light absorption [5]. However, the recombination of the charge pair (holes and electrons) should not be ignored because it decreases the photoactivity.

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In order to avoid this and improve the photocatalytic activity, the photogenerated charges must be trapped on TiO₂ surface, thus reducing recombination [6]. In this context, several techniques such as doping with metals and non-metals [7], dye sensitization [8], deposition with noble metals [9], and coupled semiconductor [10] have been assessed. Among these studies, doping has shown positive effects on titania because it gives unique electronic and structural properties that translate into better activity. Doping with non-metallic ions usually introduces energy levels above the valence band of the semiconductor for photon absorption in the visible-light region [11]. Alternatively, the use of dopant metals promotes charge transfer and separation of photogenerated charges [12]. In addition, concentration and distribution of dopant ions in TiO₂ are factors that must be considered for a good photocatalytic performance [13]. The incorporation of two types of cations into TiO2 lattice and its photocatalytic performance has been reported in several studies [14,15]. Estrellan et al. [14] reported that when the TiO2 is co-doped with iron and niobium, the photocatalytic efficiency is improved due to synergistic actions between the doping species, which favor the e-/h+ generation and reduce the recombination rate. Shi et al. [15] reported enhanced photoactivity of titania with iron and cerium by co-doping, which is due to cooperative effects of both dopants, by broadening the absorption spectrum and retarding the recombination of the photogenerated charges.

The synthesis of the catalysts was carried out by Evaporation-Induced Self-Assembly (EISA) method, which allows obtaining mesoporous structures with high surface areas [16]. On the other hand, 4-chlorophenol oxidation was chosen as the reaction to evaluate the synthesized materials, because it is considered as a model molecule for photocatalytic evaluations [17]. However, it is important to mention that the removal of chlorophenols can be carried out by other techniques, such as hydrodechlorination, which has been the subject of many investigations [18,19]. 4-chlorophenol is a pollutant commonly found in the effluents from industries related to insecticides, dyes, plastics, herbicides, detergents, wood preservatives, and petroleum reforming [20]. In addition, it is classified by the U.S. Environmental Protection Agency (USEPA) as a very toxic pollutant in these effluents, since it causes damage to human health and aquatic environments, so its effective removal is of great interest [21].

In this work, samples of TiO_2 co-doped with transition metals tungsten (W) and molybdenum (Mo) were synthesized by the EISA method. In addition, these materials have not been reported previously. The resulted solids were characterized and their photocatalytic activity evaluated in the degradation and mineralization of 4-chlorophenol. This was conducted with the main objective of demonstrating a synergistic effect in terms of improved photoactivity by co-doping compared to W-doped and Mo-doped TiO_2 catalysts. Results were compared to those obtained with commercial Degussa P25 TiO_2 .

2. Results and Discussion

2.1. Photocatalysts Characterization

Thermal analysis (TGA) of the synthesized titania without dopants is shown in Figure 1. A weight loss (9%) from room temperature to 200 °C can be observed, which is assigned to desorption of water and residual organic solvents [22]. Subsequently, a significant weight loss (34%) between 200 °C and 300 °C is attributed to the elimination of P123 organic surfactant. Finally, the dehydroxylation process is observed above 300 °C. The thermogravimetric graph of the TiO₂ sample co-doped with 1 wt.% tungsten and 1 wt.% molybdenum is shown in Figure 1. For comparison purposes, it can be observed that the incorporation of dopant cations into titania reduces the percentage of total weight loss, which resulted in a better thermal stability, as previously observed by Hussain et al. [23]. The applied heat treatment (400 °C as maximum temperature) ensured the complete elimination of the organic surfactant in all synthesized samples. On the other hand, infrared analysis (FTIR) before and after heat treatment were carried out to corroborate the P123 elimination (see Figure 2). Before thermal treatment, the characteristic signals of the surfactant by IR at 1090, 1640, 2850, and 3250 cm⁻¹ are attributed

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to the C-H, H-O-H, O-H, and C-C bonds vibrations, respectively [24]. After heat treatment, these characteristic bands disappear, which confirms the complete elimination of the organic compound incorporated during the synthesis.

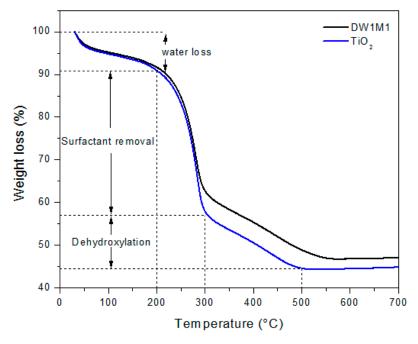


Figure 1. Thermogravimetric analysis of pure TiO₂ and TiO₂ co-doped with 1 wt.% tungsten and molybdenum (DW1M1).

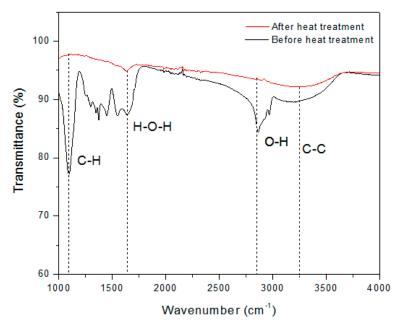


Figure 2. Infrared spectra of pure TiO₂ sample before and after thermal treatment.

Figure 3 shows the X-ray diffraction (XRD) patterns of all synthesized samples. The materials exhibited reflections at $20 \approx 25.4^{\circ}$, 37.8° , 48.1° , 54.2° , 55.2° , and 62.6° , which are representative of the crystalline planes (101), (004), (200), (105), (211), and (204) of the anatase TiO_2 phase, respectively [25]. The average crystallite sizes according to the Scherrer equation, as well as the lattice distortion of the samples are shown in Table 1. The crystal size of TiO_2 (8.6 nm) and DM2 (8.6 nm) is higher compared to the co-doped titania samples, whose values are in the range between 6.6 nm and 8.3 nm. This suggests that the inclusion of tungsten and molybdenum into the TiO_2 matrix suppresses crystalline

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growth. No relationship was observed due to the increase in weight percentage of a second cation by co-doping with respect to the average crystallite size. The lattice parameters and the unit cell volume of the materials are shown in Table 1. As can be seen, the incorporation of W and Mo in the titania by co-doping further increases the c-axis in comparison with the mono-doping, which indicates a greater expansion of the crystal lattice. Since the ion radii of Mo^{6+} (0.062 nm) and W^{6+} (0.060 nm) are very similar to that of Ti^{4+} (0.0605 nm), they can replace titanium within the anatase structure [26]. These changes in the lattice parameters are associated with lattice distortion due to the inclusion of dopants in the TiO_2 matrix.

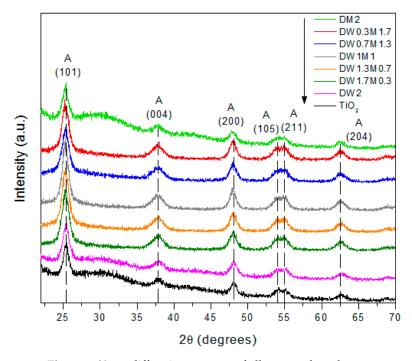


Figure 3. X-ray diffraction patterns of all prepared catalysts.

Table 1. Crystal size, lattice distortion and lattice parameters of the synthesized materials.

C 1 .	Average Crystallite	Lattice	Lattice Parameters		
Sample	Size (nm)	Distortion (ϵ)	a (nm)	c (nm)	V (nm ³)
TiO2	8.6	0.0183	0.377	0.931	0.132
DM2	8.6	0.0183	0.378	0.933	0.133
DW0.3M1.7	7.7	0.0207	0.378	0.982	0.140
DW0.7M1.3	6.6	0.0238	0.378	0.943	0.135
DW1M1	8.3	0.0191	0.379	0.937	0.134
DW1.3M0.7	7.3	0.0216	0.378	0.947	0.136
DW1.7M0.3	7.1	0.0224	0.379	0.959	0.138
DW2	8.1	0.0195	0.379	0.933	0.134

Figure 4 shows Raman spectra (RS) of TiO_2 , W- TiO_2 , Mo- TiO_2 , and W-Mo- TiO_2 samples. As observed in Figure 4a, some peaks centered at 144, 197, 397, 517, and 640 cm⁻¹ are assigned to the $E_{g(1)}$, $E_{g(2)}$, $B_{1g(1)}$, ($A_{1g} + B_{1g(2)}$), and $E_{g(3)}$ vibration modes, respectively, corresponding to anatase crystalline phase [27]. This suggests that, after doping with W and/or Mo cations, the anatase phase is maintained on the TiO_2 surface, which is in accordance with the XRD results. The position and intensity of the most intense Raman peak at 144 cm⁻¹ change when titania is mono-doped and co-doped with these kinds of cations (see Figure 4b), which is attributed to changes in lattice parameters and unit cell volume by XRD analysis (see Table 1) [28]. This shift Raman is related to the increase in the c lattice parameter and the lattice expansion, which is due to the incorporation of W and/or Mo dopant ions

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into anatase, generating structural distortion [29]. In addition, the absence of new Raman bands in the spectra implies that there are no additional phases, which means that the dopants have been incorporated and they are well distributed in the crystal structure of TiO₂ [30].

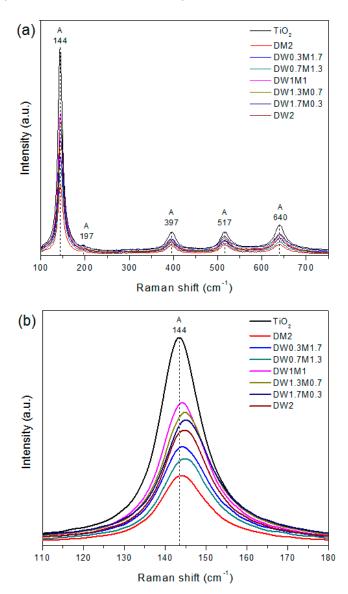


Figure 4. (a) Raman spectra and (b) Raman peak at 144 cm⁻¹ of all synthesized materials.

The nitrogen adsorption-desorption isotherms and pore size distributions of all prepared materials are shown in Figures 5 and 6, respectively. The samples exhibited type IV isotherm, which is characteristic of mesoporous materials [31] according to the IUPAC (International Union of Pure and Applied Chemistry) classification. Type H2 hysteresis loop with monomodal pore size distribution is attributed to porous structures formed by interconnected networks. Table 2 presents the specific surface area, average pore diameter, and pore volume of all samples. The specific surface area of the co-doped samples is higher than the doped ones (DW2 and DM2), which is probably due to the incorporation of W and Mo in the crystal structure as well as the decrease in crystallinity. The maximum specific surface area of 191 m²/g is obtained with sample DW0.7M1.3. No relationship was observed between the surface area and the concentration of a second dopant cation, however, when W or Mo is incorporated in a small amount (0.33 wt.%) by co-doping (DW1.7M0.3 and DW0.3M1.7 samples), the specific surface area increases by approximately 15% with respect to mono-doped TiO₂. High surface areas may offer more adsorption sites for photocatalytic reactions, which could result in

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better photoactivity [32]. Additionally, the average pore diameter of the co-doped catalysts remained constant (5.6 nm) and with a value between that of the mono-doped catalysts (4.6 nm and 6.1 nm), with the exception of sample DW1M1 that exhibited the highest average pore diameter of 6.6 nm and the maximum pore volume of $0.391 \, \text{cm}^3/\text{g}$.

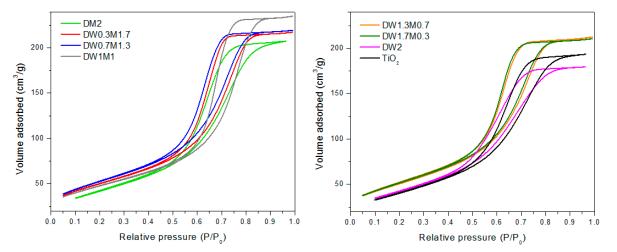


Figure 5. N_2 adsorption-desorption isotherms of undoped, doped, and co-doped TiO_2 with different weight percentages of W and Mo ions.

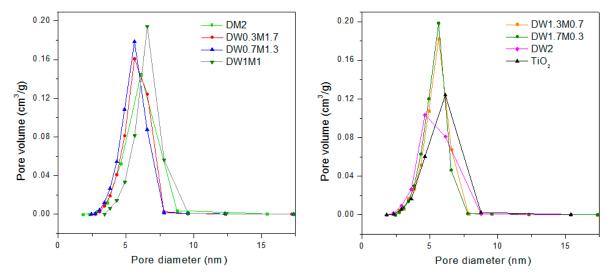


Figure 6. Pore size distributions of undoped, doped, and co-doped TiO₂ with different weight percentages of W and Mo ions.

Table 2. Textural properties of TiO₂, W-TiO₂, Mo-TiO₂, and W-Mo-TiO₂ catalysts.

Catalyst	Specific Surface Area (m ² /g)	Average Pore Diameter (nm)	Pore Volume (cm ³ /g)
TiO ₂	144	6.1	0.328
DM2	161	6.1	0.354
DW0.3M1.7	181	5.6	0.362
DW0.7M1.3	191	5.6	0.365
DW1M1	172	6.6	0.391
DW1.3M0.7	185	5.6	0.352
DW1.7M0.3	188	5.6	0.352
DW2	160	4.6	0.307

The catalysts were analyzed by UV-vis diffuse reflectance spectroscopy (DRS) to estimate their band gap energy. Figure 7a shows the UV-vis absorption spectra for all materials. As can be seen,

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the concentration of molybdenum cations in the titania gradually shifts the absorption edge towards the long wavelength region. Figure 7b shows the Tauc plots from the Kubelka-Munk function [33]. The band gap and its corresponding wavelength are presented in Table 3 for all synthesized samples. TiO₂ doped with 2 wt.% tungsten did not show a significant reduction in band gap energy (3.08 eV). However, titania doped with 2 wt.% molybdenum showed a slight shift towards visible radiation absorption at 452 nm and a band gap energy of 2.74 eV. The incorporation and increase of tungsten by co-doping does not favor the band gap reduction, since the "d" orbitals of W are located into the conduction band of TiO₂, making it difficult to generate energy levels under this band [34]. On the other hand, the increase of molybdenum by co-doping favors the band gap reduction due to charge transfer transitions between the "d" orbitals of molybdenum and TiO₂ located under the conduction band. The incorporation of this type of dopant cations (W and Mo) does not exhibit cooperative effects by co-doping towards an effective reduction of the band gap in anatase, so that its absorption of energy for electronic excitation is between the limits of mono-doped materials (DW2 and DM2 samples) at 403–452 nm.

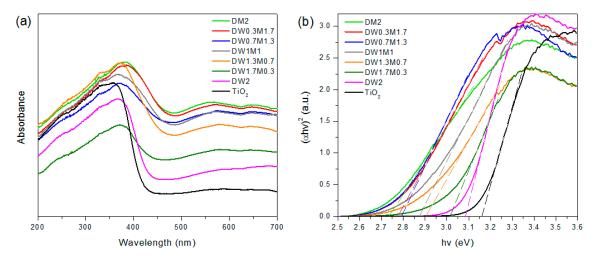


Figure 7. (a) UV-vis absorbance spectra and (b) band gap energies (Tauc plots) of synthesized catalysts.

Sample	Band Gap (eV)	Wavelength (nm)
TiO ₂	3.16	392
DM2	2.74	452
DW0.3M1.7	2.78	446
DW0.7M1.3	2.80	443
DW1M1	2.87	432
DW1.3M0.7	2.90	428
DW1.7M0.3	3.01	412
DW2	3.08	403

Table 3. Band gap energy and wavelength of synthesized materials.

X-ray photoelectron spectroscopy (XPS) was used to determine the chemical state of the elements on the surface of the synthesized materials. Figure 8 shows the XPS spectra of Ti 2p in samples TiO₂, DM2, DW2, and DW1M1. Two peaks located with binding energies at 458.1 eV and 463.8 eV are assigned to Ti $2p_{3/2}$ and Ti $2p_{1/2}$ states, respectively, and these correspond to Ti⁴⁺ into TiO₂ lattice. A slight shift towards higher binding energies at 0.3 eV after doping and co-doping with W and Mo cations is evidence that these dopant metals are part of the anatase crystalline structure [35]. The O 1s XPS spectra for the sample co-doped with 1 wt.% W and 1 wt.% Mo is shown in Figure 9a. The peak at 529.8 eV is attributed to the Ti-O bond in TiO₂, whereas that at 532.2 eV is assigned to surface hydroxyl groups. In photocatalysis, these hydroxyl groups play an important role, since they react with the holes generated during photoexcitation to produce hydroxyl radicals, which degrade organic

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compounds [36]. Figure 9b presents the XPS spectrum in the Mo 3d region for sample DW1M1. In this figure, two main peaks, attributed to Mo $3d_{3/2}$ and Mo $3d_{5/2}$, show contributions of Mo⁶⁺ at 233.2 eV and 234.7 eV, and Mo⁵⁺ at 232.8 eV and 234.3 eV. The percentage of these two species in the catalyst surface is 77.5% and 22.5% for Mo⁶⁺ and Mo⁵⁺, respectively. The XPS spectrum shown in Figure 9c corresponds to the W 4f region. As it is observed, the surface of sample DW1M1 is composed of 91.7% W⁶⁺ and 8.3% W⁵⁺ at 36.0 eV and 34.3 eV, respectively. These signals confirm the presence of both W and Mo dopants on the surface layers of TiO₂, which are involved in the photocatalytic reactions.

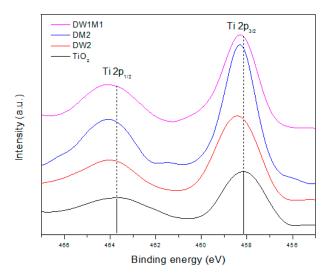


Figure 8. XPS spectra of Ti 2p for undoped, doped, and co-doped TiO₂ samples with W and Mo.

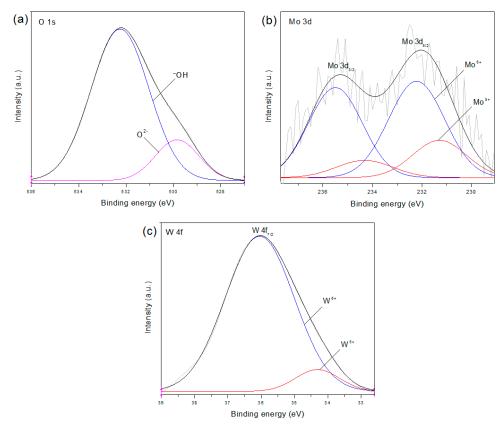


Figure 9. XPS spectra of **(a)** O 1s, **(b)** Mo 3d, and **(c)** W 4f for co-doped TiO₂ sample with W and Mo cations.

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Figure 10 shows the transmission electron micrographs, the high resolution transmission micrographs, as well as the electron diffraction patterns of mono-doped samples DM2 and DW2, and co-doped sample DW1M1. The TEM images in Figure 10a,d,g for mono-doped and co-doped titania with Mo and W cations show individual nanoparticles from five to 12 nm in size. The HRTEM micrographs for samples DM2 and DW2 in Figure 10b,e, exhibit interlayer spacing (0.35 nm) that corresponds to the (101) plane of anatase crystalline structure. In the HRTEM image for sample co-doped with 1 wt.% of both cations (Figure 10h); the (101) and (004) planes of anatase phase can be observed [37]. Furthermore, the SAED patterns shown in Figure 10c,f,i indicate a sequence of rings, which are characteristic of the crystalline planes (101), (004), (200), and (105). All these results are consistent with those obtained by the XRD analysis.

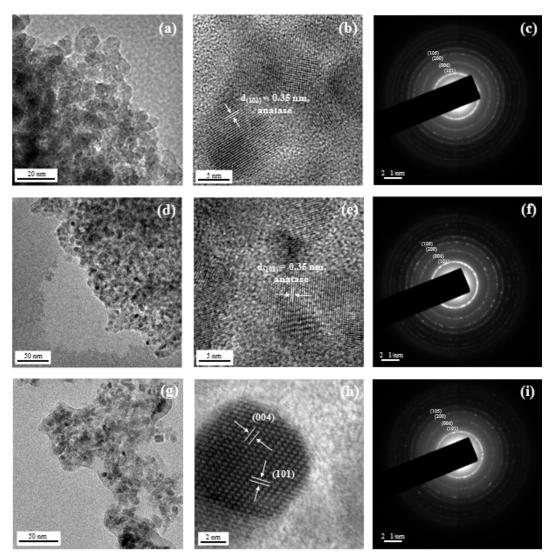


Figure 10. TEM images, HRTEM images and SAED patterns of samples (**a**–**c**) DM2, (**d**–**f**) DW2, and (**g**–**i**) DW1M1.

In order to visualize the dopant atoms in the anatase crystalline structure, Figure 11 shows STEM-HAADF micrographs with corrected aberration of synthesized catalysts DW1, DM1, and DW1M1. It can be observed in Figure 11a (W-mono-doped TiO₂) that the tungsten atoms are well distributed and they are part of well-defined atomic columns within the anatase phase. However, when titania is mono-doped with molybdenum atoms (Figure 11b), these can have two locations: (i) As being part of the crystalline structure with a good distribution (yellow circles), and (ii) forming some dispersed agglomerates that are not part of the atomic structure (orange circles). On the other hand,

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when TiO_2 is co-doped with W and Mo, the dispersion of both atoms inside the crystalline lattice is favored (atomic agglomerations disappear) (see Figure 11c).

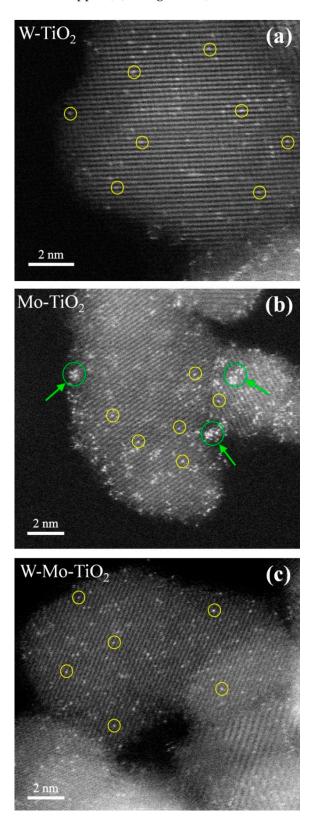


Figure 11. Aberration-corrected STEM-HAADF images of TiO_2 mono-doped with (a) W or (b) Mo, and (c) co-doped with both cations.

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2.2. Photocatalytic Tests

All synthesized samples of pure titania, mono-doped, and co-doped with molybdenum and tungsten cations, as well as the commercial catalyst Degussa P25 were evaluated in the photocatalytic degradation of 4-chlorophenol (4CP). Figure 12 shows the photodegradation profiles with 100 min of reaction as total time. It can be seen that pure synthesized TiO₂ and Degussa P25 exhibit the lowest percentages of 4CP degradation. With respect to the co-doped TiO₂ samples, these show better degradation than the mono-doped materials (DM2 and DW2), which might be because of synergistic effects by the two doping species, thus improving their photoactivity. In addition, the co-doped samples exhibited higher surface areas and smaller crystallite sizes than the mono-doped catalysts, which increase the sites density for substrates adsorption during photocatalytic activity [38]. The degradation is gradually improved when the amount of tungsten increases from 0.3 wt.% to 1 wt.%, achieving more than 97% for the DW1M1 sample. Concentrations above 1 wt.% W in co-doped TiO₂ do not favor photocatalytic activity, which results in less degradation. While the samples co-doped with high molybdenum concentration (DW0.3M1.7 and DW0.7M1.3) exhibited higher radiation absorption, the significant improvement in photoactivity is very likely due to suitable concentration of both dopants. The co-doped TiO₂ with W:Mo = 1:1 ratio exhibits a synergistic effect between the species towards the best photocatalytic activity, since it presents a good radiation absorption capacity (see Figure 7) that favors the generation of e^-/h^+ pairs, as well as the ability to reduce the recombination processes of the photogenerated charges [39]. The process to inhibit the recombination (Equation (2)) after the e^-/h^+ pair generation (Equation (1)) is shown in Equations. (3) and (4). The electrons can be trapped by the most stable states of W and Mo, the W⁶⁺ and Mo⁶⁺ species, respectively, reaching a local charge compensation that leads to a longer lifetime of the generated holes [40]. In this way, the holes can be trapped by the adsorbed H₂O or by the -OH groups on the TiO₂ surface (Equations (5) and (6)) (see Figure 9a), allowing the generation of more •OH radicals, which are responsible for the oxidation of organic compounds (Equation (7)). On the other hand, the redox potentials of the doping species involved should not be left aside. In this case, the potentials of the $\mathrm{Mo^{6+}/Mo^{5+}}$ and $\mathrm{W^{6+}/W^{5+}}$ pairs are +0.4 V and -0.03 V, respectively [41,42], and they are more positive than the potential of the TiO_2 conduction band (-0.51 V) [43], so their electronic capture is favored when both are present in titania. The prepared catalysts showed no 4CP removal due to adsorption, and likewise, the photolysis effect was negligible.

$$W-Mo-TiO_2 + hv \rightarrow e^- + h^+$$
 (1)

$$e^- + h^+ \to TiO_2 \tag{2}$$

$$W^{6+} + e^- \to W^{5+}$$
 (3)

$$Mo^{6+} + e^- \rightarrow Mo^{5+}$$
 (4)

$$H_2O + h^+ \rightarrow \bullet OH + H^+$$
 (5)

$$-OH + h^{+} \rightarrow \bullet OH \tag{6}$$

$$\bullet OH + 4CP \rightarrow CO_2 + H_2O + Cl^-$$
(7)

The photocatalytic degradation profiles of 4CP were adjusted to a pseudo-first-order kinetics [44]. Apparent kinetic constants as well as half-life for all catalysts are summarized in Table 4. It can be seen that the maximum degradation is achieved by using the titania co-doped at 1 wt.% of both cations with a 4CP half-life of approximately 20 min, where about 75% of the initial concentration of 4CP is degraded in 40 min.

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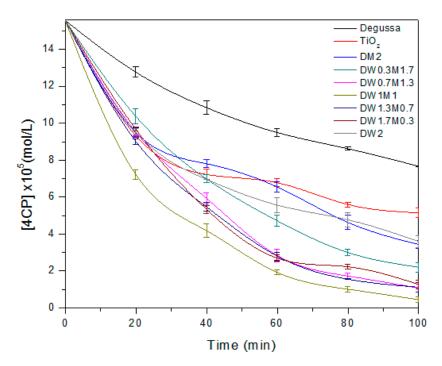


Figure 12. 4-Chlorophenol photodegradation profiles over TiO_2 , Mo- TiO_2 , W- TiO_2 , W-Mo- TiO_2 , and Degussa P25 catalysts. $[4CP]_0 = 15.56 \times 10^{-5}$ M, $C_{cat} = 2 \times 10^{-4}$ kg/L, pH₀ = 2, T = 298 K.

Table 4. Removal percentages, apparent kinetic constant, half-time and regression coefficient for 4CP degradation.

Catalyst	4CP Degradation (%)	$k \times 10^2 (min^{-1})$	t _{1/2} (min)	r ²
DM2	76	1.31	52.8	0.97
DW0.3M1.7	86	1.98	34.9	0.99
DW0.7M1.3	93	2.75	25.2	0.99
DW1M1	97	3.49	19.9	0.99
DW1.3M0.7	93	2.73	25.4	0.99
DW1.7M0.3	92	2.51	27.6	0.99
DW2	75	1.26	54.9	0.95
TiO ₂	67	1.04	66.8	0.90
Degussa P25	51	0.67	102.9	0.97
_				

The percentage of mineralization, which is estimated by the total organic carbon content (TOC), is an important parameter of the degree of deep or complete oxidation of 4-chlorophenol to carbon dioxide and water (Equation (8)) [45].

$$C_6H_5ClO + W-Mo-TiO_2 + hv \rightarrow Intermediates \rightarrow CO_2 + H_2O + Cl^-$$
 (8)

Figure 13 shows the 4CP mineralization percentages using the TiO_2 catalysts mono and co-doped with W and/or Mo. W-Mo-co-doped TiO_2 with a tungsten weight ratio of W/(W + Mo) = 0.5 showed the highest percentage of mineralization achieving more than 74%. On the other hand, the mono-doped samples (DM2 and DW2) exhibited average mineralization percentages of 50% and 53%, respectively.

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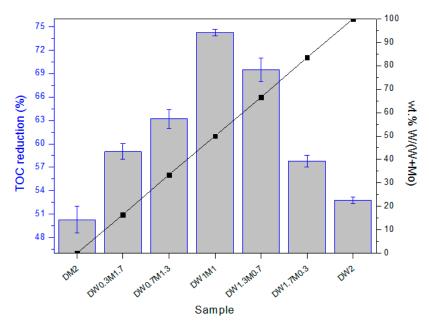


Figure 13. TOC reduction percentages and amount of tungsten dopant on mono-doped and co-doped TiO_2 . [4CP] $_0$ = 15.56 \times 10⁻⁵ M, C_{cat} = 2 \times 10⁻⁴ kg/L, pH $_0$ = 2, T = 298 K.

Table 5 shows the photocatalytic performance of the synthesized sample DW1M1 in comparison with other reported co-doped TiO_2 materials. It can be seen that, although the reaction conditions are a little different, the synthesized material of W-Mo- TiO_2 exhibits a higher degradation/mineralization percentage of 4-chlorophenol, as well as enhanced kinetic parameters.

Table 5. Photocatalytic performance of W-Mo-co-doped TiO₂ compared to other reported co-doped TiO₂ materials.

Photo-Catalyst	$[4CP]_0 \times 10^5 \text{ (mol/L)}$	Radiation	4CP Degradation (%)	4CP Mineralization (%)	$\begin{array}{c} k \times 10^2 \\ \text{(min}^{-1}) \end{array}$	Ref.
W-Mo-TiO ₂	15.56	UV	97 (100 min)	74 (100 min)	3.49	-
B-N-TiO ₂	5.00	UV	98 (120 min)	=	3.41	[46]
N-F-TiO ₂	7.78	VL	72 (300 min)	-	-	[47]
C-W-TiO ₂	50.00	VL	-	57 (300 min)	-	[48]

The intermediate compounds formed during the photocatalytic degradation of 4-chlorophenol were determined by HPLC. Figure 14 shows the concentration profiles of the main intermediaries when the catalyst DW1M1 is used. The name and structure of these identified intermediaries are summarized in Table 6. The aromatic ring of 4-chlorophenol can be attacked by hydroxyl radicals in several positions, and for this reason some mechanistic pathways are possible towards the formation of these species (see Figure 15). Furthermore, the generation of hydroxylated compounds is clear evidence of the involvement of hydroxyl radicals in the degradation of 4CP [49].

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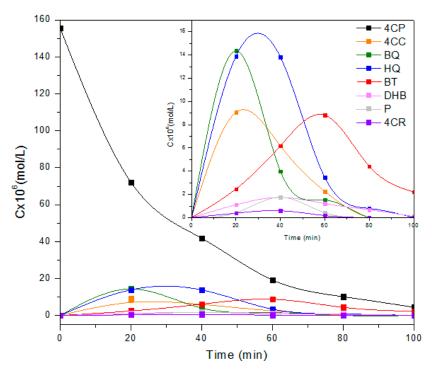


Figure 14. Concentration profiles of intermediates formed during 4CP photodegradation over W-Mo-TiO₂. [4CP] $_0$ = 15.56 \times 10⁻⁵ M, C_{cat} = 2 \times 10⁻⁴ kg/L, pH $_0$ = 2, T = 298 K.

Table 6. Intermediate compounds formed in the photocatalytic degradation of 4-chlorophenol over TiO_2 co-doped with W and Mo.

Intermediate Compound ID	Name	Molecular Structure
HQ	Hydroquinone 1,4-benzenediol 1,4-dihydroxybenzene	но
BQ	Benzoquinone 1,4-benzoquinone p-benzoquinone Quinone	
4CC	4-chlorocatechol 4-chloro-1,2-benzenediol	СІОН
BT	1,2,4-benzenetriol Hydroxyhydroquinone	но ОН
DHB	1,2-dihydroxybenzene 1,2-benzenediol Catechol Pyrocatechol	ОН
P	Phenol Hydroxybenzene	OH
4CR	4-chlororesorcinol 1,3-dihydroxy-4-chlorobenzene	HOOOH

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Figure 15. Proposed photocatalytic degradation pathway of 4-chlorophenol by W-Mo-TiO₂.

According to Figure 15, the oxidation process of 4CP is favored towards the quinone route (routes I and II) with maximum concentrations in the first minutes of reaction. Hydroquinone is generated by hydroxylation and dechlorination of 4CP in para position of the aromatic ring, which can be oxidized to benzoquinone. Additionally, benzoquinone can be formed from the hydroxyphenyl radical via route I [50].

4CP hydroxylation in ortho position generates 4-chlorocatechol (route III). The *para*-dechlorination of 4CC forms the 1,2-dihydroxyphenyl radical [51], which, when combined with a proton and an electron, produces 1,2-dihydroxybenzene. The peak with the highest concentration of 1,2,4-benzenetriol is found at 60 min, because its formation is after hydroxylation of hydroquinone (route II) or 4-chlorocatechol (route III). The oxidation process of 4CP can be initiated by the attack of the C-Cl bond (route IV) to form the hydroxyphenyl radical [52], which subsequently combined with a proton generates phenol. On the other hand, DHB (1,2-dihydroxybenzene) can also be formed by *ortho*-hydroxylation of phenol. However, this oxidation route is not favored due to low concentrations of phenol and 1,2-dihydroxybenzene observed during photocatalytic tests (see Figure 14). Similarly, *meta*-hydroxylation of 4CP generates very little 4-chlororesorcinol concentration (route V). Finally, the aromatic ring of these intermediate compounds is opened by subsequent oxidations to form aliphatic species (organic acids), which are later mineralized to carbon dioxide and water according to Equation (8) [53].

3. Materials and Methods

3.1. Chemicals

Titanium (IV) butoxide [Ti(OC₄H₉)₄, 97%], ammonium metatungstate [(NH₄)₆H₂W₁₂O₄₀·xH₂O] and ammonium heptamolybdate [(NH₄)₆Mo₇O₂₄·4H₂O] served as metal precursors Ti, W, and Mo, respectively. Pluronic P123 [EO₂₀PO₇₀EO₂₀] was used as a surfactant. Ethanol (C₂H₆O, 99.5%) and

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nitric acid (HNO₃, 70%) were employed as solvents and catalysts, respectively. 4-chlorophenol (ClC₆H₄OH, 99%) and deionized water were used during the photocatalytic tests. All reagents were of analytical grade and supplied by Sigma-Aldrich (St. Louis, MO, USA). Methanol and acetonitrile (HPLC grade) were used to prepare the mobile phases during the chromatographic analysis.

3.2. Preparation of Undoped, Doped and Co-Doped TiO₂

Pure TiO_2 was prepared by mixing ethyl alcohol and $Ti(OC_4H_9)_4$ under stirring for 15 min. This mixture was added to the P123 surfactant with constant stirring for 30 min and finally nitric acid was dropwise incorporated to the reaction media. The homogenous mixture was kept under vigorous stirring for 3 h. The molar ratio of reagents $Ti(OBu)_4$: C_2H_6O :P123: HNO_3 was maintained constant at 1:18.71:0.018:3.55. To prepare co-doped TiO_2 , tungsten and/or molybdenum precursors were added before incorporating the HNO_3 . The resulting solution was dried in a rotary evaporator until a solid was formed. This was subsequently calcined at 300 °C for 1 h (surfactant removal) and then at 400 °C for 4 h with a heating rate of 1 °C/min. Various samples with different weight percentages of dopant were synthetized. However, the total amount of dopant (s) was kept constant at 2 wt.% with respect to titania. The catalysts were labeled as DWAMB. W stands for tungsten and M for molybdenum. In addition, A and B are the weight percentages in samples. The DM2 and DW2 catalysts contain 2 wt.% molybdenum and 2 wt.% tungsten, respectively. Commercial TiO_2 (Degussa P25) was used for comparison.

3.3. Characterization of Photocatalysts

The temperature for the removal of the surfactant during heat treatment was determined by thermogravimetry (TGA 4000, Pyris, Perkin Elmer, Waltham, MA, USA) with N_2 gas flow at 20 mL/min in a range of 30 to 700 °C (5 °C/min). Before and after heat treatment, Infrared spectroscopy (FT-IR, Perkin Elmer Spectrum Two, Waltham, MA, USA) was used. The FT-IR spectra were acquired in the range of 4000–1000 cm⁻¹ and using a resolution of 1 cm⁻¹.

All synthesized samples were analyzed with an X-ray diffractometer (Bruker Advance D8, Billerica, MA, USA) to determine the crystalline properties. Cu K α radiation at 1.5406 Å was used as the X-ray source. The voltage and current applied were 30 kV and 25 mA, respectively. The diffraction patterns were determined over 20 range of 20° – 70° with a resolution of 0.02° 20/16 s. The average crystallite size (D) was estimated by the Scherrer equation D = $0.9\lambda/\beta$ cos0, the lattice distortion (d) by d = $\beta/4$ tg0, and the lattice parameters according to the Bragg equation 2dhkl sin0 = λ , where λ is the applied wavelength, 0 is the Bragg angle, and β is the FWHM value. Raman spectroscopy was also used to determine the structural properties. For this purpose, a Renishaw micro-Raman spectrometer, provided with a laser of 514 nm, was used.

Average pore diameters and specific surface areas of the synthesized materials were obtained through nitrogen adsorption–desorption isotherms on a Quantachrome (Boynton Beach, FL, USA) Autosorb-1 at 77 K by BJH and BET methods, respectively. Before measurements, the materials were degassed out for 2 h at 200 °C.

The band gaps of all samples were determined through UV-vis Diffuse Reflectance Spectroscopy (DRS) using a Perkin Elmer spectrophotometer (Lambda 35, Waltham, MA, USA) equipped with an integration sphere (Labsphere rsape-20). The Kubelka-Munk function $\alpha = (1-R)^2/2R$ was used to transform the reflectance spectra into absorption spectra, which were then used to estimate the band gap energies by constructing Tauc plots of hv vs. $(\alpha hv)^2$ and extrapolating the linear part to $(\alpha hv)^2$ equal to zero.

The chemical states of the surface elements present in the materials were examined by X-ray photoelectron spectroscopy (JEOL JPS-9200 spectrometer, Akishima, TYO, Japan) with an Al K α X-ray source. The energy of C 1s at 284.6 eV was used as a reference for charge correction during the estimation of the binding energies.

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High resolution-transmission electron microscopy (HR-TEM, JEOL-2100 at 200 kV, Akishima, TYO, Japan) and selected area electron diffraction (SAED) were used to determine the particle size and crystalline structure. High-angle annular dark-field scanning microscopy with a spherical aberration corrector (ac-HAADF-STEM, JEOL 2200FS+CS, Akishima, TYO, Japan) was used to visualize the doping atoms in the crystalline structures.

3.4. Photocatalytic activity and analysis

The photoactivity of all catalysts was evaluated in the degradation of 4-chlorophenol. The constant reaction conditions used were 2×10^{-4} kg/L of catalyst loading, 20×10^{-6} kg/L of 4CP and temperature of 298 K. A UV lamp was used as a radiation source with a wavelength and intensity of 254 nm and 4500 $\mu W/cm^2$, respectively. The 4-chlorophenol solution was adjusted to an initial pH of 2 and stirred at 1000 rpm to keep the catalyst in suspension. Aliquots were withdrawn from the reactor every 20 min and subsequently centrifuged to remove the catalyst. The solutions were then analyzed on a UV-vis spectrophotometer (Perkin Elmer Lambda 25, Waltham, MA, USA) to determine the amount of 4-chlorophenol at 280 nm, according to its absorbance peak. The total organic carbon concentration in the solution was measured by using a Shimadzu TOC-L analyzer to evaluate the 4CP mineralization. The identification of intermediate compounds during 4CP degradation was carried out by HPLC (Varian 230, isocratic mode, Santa Clara, CA, USA). Ascentis Express C18 column (2.7 μm , 3 cm \times 4.6 mm, Sigma-Aldrich) with a mobile phase methanol/water (20/80 v/v, 1.0 mL min $^{-1}$) and Eclipse XDB-C18 column (5 μm , 15 cm \times 4.6 mm, Agilent, Santa Clara, CA, USA) with a mobile phase acetonitrile/water (10/90 v/v, 0.6 mL/min) were used at 25 °C.

4. Conclusions

TiO₂ nanoparticles were prepared by mono and co-doping with molybdenum and tungsten cations by the Evaporation-Induced Self-Assembly (EISA) method. All synthesized materials presented anatase crystalline phase and larger specific surface areas than the commercial photocatalyst Degussa P25. The presence of a second cation by co-doping increased the surface area and decreased the crystallinity, as well as changes in lattice parameters due to structural distortion, since the doping species were successfully incorporated into TiO2 lattice. The dopant cations exhibited oxidation states with valence 5+ and 6+ in the titania. Low tungsten concentrations by co-doping favored the reduction of the anatase band gap. Raman analysis showed that no additional phases by co-doping were created, and both dopants are distributed inside the anatase. All synthesized materials exhibited photocatalytic activity in the oxidation of 4-chlorophenol and all of them outperformed commercial TiO2 P25. Synergistic effect with an appropriate concentration of both dopant cations improved photocatalytic activity by reducing the recombination of photogenerated charges and by increasing the absorption of radiation to promote their generation. The experiments performed at ambient temperature and during 100 min showed that titania, co-doped with 1 wt.% W and 1 wt.%, Mo exhibited the best photoactivity among the catalysts studied, with 97% and 74% of degradation and mineralization of 4-chlorophenol, respectively. The oxidation route of 4-chlorophenol was favored towards quinones as main intermediates via hydroxyl radicals. Remaining TOC (26%) is attributed to 1,2,4-benzenetriol as secondary intermediate and aliphatic compounds.

Author Contributions: J.E.-V. and R.N.-R. designed the experiments and supervised the project; O.A.-G. performed the experiments and analyzed the data; D.A.S.-C. contributed with characterization of materials; O.A.-G. wrote the manuscript; All authors contributed to a review of the manuscript before submission.

Funding: This research was funded by PROMEP-Mexico (financial support through project 103.5/13/S257) and CONACYT-Mexico (project 269093). Scholarship 378292 by CONACYT.

Acknowledgments: Authors are grateful to CCIQS from UAEM-UNAM and CIMAV-Mexico for the granted support. The technical support of Gustavo López Téllez, Alfredo Rafael Vílchis Néstor, Uvaldo Hernández, Citlalit Martínez and Carlos Elías Ornelas Gutiérrez is also acknowledged.

Conflicts of Interest: The authors declare no conflict of interest.

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