

Revision_1**Order of $^{[6]}\text{Ti}^{4+}$ in a Ti-rich calcium amphibole from Kaersut, Greenland: a combined X-ray and neutron diffraction study**

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Order of $^{16}\text{Ti}^{4+}$ in a Ti-rich calcium amphibole from Kaersut, Greenland: a combined X-ray and neutron diffraction study

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Abstract

In order to characterize the role of Ti in the crystal structure of calcium amphiboles with high or even dominant oxo-component, the crystal structure of a Ti-rich calcium amphibole from a gabbro at Kaersut, Greenland, has been refined with single-crystal $\text{MoK}\alpha$ X-ray intensity data to an $R_I(F)$ index of ~ 0.025 , and with single-crystal Laue neutron intensity data to an $R_I(F)$ index of ~ 0.053 . The crystal used for X-ray structure refinement was characterized by electron- and ion-microprobe analysis. The site populations of the C-group cations Mg, Fe and Ti were calculated from the refined site-scattering values for the $M(1)$, $M(2)$ and $M(3)$ sites derived by both X-ray and neutron diffraction. Ti is distributed among all the three 6-fold coordinated M sites, with a strong preference for the $M(1)$ and $M(3)$ sites, where its main role is maintaining electroneutrality at the deprotonated O(3) site. The pattern of distortion of the $M(1)$, $M(2)$ and $M(3)$ octahedra differs from that in F-free deprotonated or partly deprotonated amphiboles, where Ti^{4+} does not occur at the $M(3)$ site. The neutron structure refinement provides also a clear picture of the environment of the proton, anisotropic-displacement behaviour and potential hydrogen-bonding arrangements. A trifurcated hydrogen-bonding configuration has been identified, with two O(6) and one O(7) oxygen atoms as *acceptors* of weak hydrogen-bonds.

Keywords: Ti-rich amphibole, kaersutite, single-crystal X-ray diffraction, single-crystal Laue neutron diffraction, crystal chemistry.

72 Introduction

73 The amphiboles are probably the mineral supergroup with the highest chemical variability
74 and the highest ability to crystallize under almost all the conditions relevant to petrogenesis. The
75 general amphibole formula is written (Hawthorne and Oberti 2007) as



77 where: $A = Na^+, K^+, \square, Ca^{2+}, Li^+$;

78 $B = Na^+, Li^+, Ca^{2+}, Mn^{2+}, Fe^{2+}, Mg^{2+}$;

79 $C = Mg^{2+}, Fe^{2+}, Mn^{2+}, Al^{3+}, Fe^{3+}, Mn^{3+}, Ti^{4+}, Li^+$;

80 $T = Si^{4+}, Al^{3+}, Ti^{4+}$;

81 $W = (OH)^-, F^-, Cl^-, O^{2-}$.

82 With regard to the present work, C defines the cations occupying the octahedrally coordinated $M(1)$,
83 $M(2)$ and $M(3)$ sites and T defines the cations occupying the tetrahedrally coordinated $T(1)$ and $T(2)$
84 sites. The amphibole structure (with site nomenclature) is shown in Figure 1. Recent studies have
85 shown that the oxo component, $^wO^{2-}$ or $^{O(3)}O^{2-}$ to identify the relevant site, is particularly important in
86 affecting both the geometry and the crystal-chemical behaviour of the amphibole, to the extent that
87 oxo-dominant compositions are now classified as a separate group by the current nomenclature
88 scheme (Hawthorne et al. 2012).

89 The ability of the amphibole structure to incorporate cations in different coordinations and at
90 different structural sites, particularly where related to coupled substitutions, makes this mineral
91 supergroup the ideal case study to test crystal-chemical mechanisms controlling the incorporation of
92 Ti^{4+} in silicates.

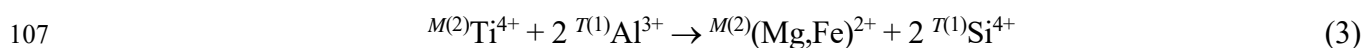
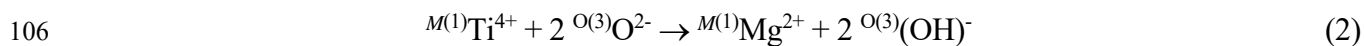
94 Previous work

95 Leake (1968) compiled a large number of amphibole analyses from the literature and classified
96 them as to quality. Saxena and Ekström (1970) examined 639 analyses, all first-class and some second-
97 class analyses as classified by Leake (1968), by principal component analysis. The strongest
98 correlation between chemical variables in this dataset is between Ti^{4+} and $(OH)^-$. They proposed that
99 Ti^{4+} enters calcium and sodium-calcium amphiboles *via* the substitution

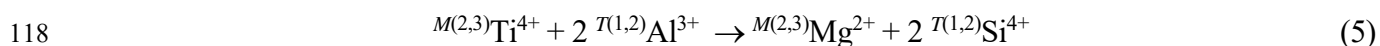


101 and found no correlation between the oxo component and the Fe^{3+} content of the amphibole.

102 Kitamura et al. (1975) and Pechar et al. (1989) showed by neutron diffraction that Ti is a C cation and
 103 is very strongly ordered at $M(1)$ in two kaersutites. Oberti et al. (1992) showed by X-ray structure
 104 refinement that $^{[6]}\text{Ti}^{4+}$ is incorporated into the amphibole richterite structure *via* three distinct
 105 substitutions:



109 **In richterites from lamproites**, the partitioning between the C and T cation-groups depends on the
 110 pressure of crystallization (Konzett 1997; Konzett et al. 1997) and the F content of the amphibole
 111 (Della Ventura et al. 1991, 1993). In an examination of a series of synthetic Ti-rich Fe-free kaersutites
 112 by X-ray diffraction, Tiepolo et al. (1999) showed that $^{[6]}\text{Ti}$ is disordered over $M(1)$, $M(2)$ and $M(3)$
 113 with a strong preference for the $M(1)$ site, and proposed two separate mechanisms for incorporation of
 114 $^{\text{C}}\text{Ti}$ into the amphibole structure: The first mechanism couples $^{[6]}\text{Ti}$ to the occurrence of O^{2-} at O(3) and
 115 leads to complete order of $^{[6]}\text{Ti}$ at $M(1)$ (*i.e.*, substitution (2) above); the local association of $M(1)\text{Ti}^{4+}$
 116 with $2 \text{O}^{(3)}\text{O}^{2-}$ was confirmed spectroscopically by Della Ventura et al. (2007). The second substitution
 117 couples $^{[6]}\text{Ti}$ to the occurrence of Al at the T sites (*i.e.*, substitution (5) below):



119 augmenting the local association of other small high-valence cations (*e.g.*, Al, Fe^{3+} , Cr^{3+}) at $M(2)$ and
 120 $M(3)$ with Al at $T(1)$ and $T(2)$ in accord with the valence-sum rule **(Brown 1981, 2002; Hawthorne**
 121 **1997; Hawthorne et al. 1998)**. Della Ventura et al. (1996) showed that in substitution (4), there is no
 122 short-range order involving $^{[4]}\text{Ti}$ and Si.

123 Therefore, the incorporation and site ordering of Ti in amphiboles is a function of both amphibole
 124 composition and conditions of crystallization. Understanding the crystal-chemical constraints on $^{\text{C}}\text{Ti}$
 125 partitioning in amphiboles is of great interest as the common occurrence of the oxo component in
 126 amphiboles is becoming more apparent due to advances in the structural and analytical techniques and
 127 advancing crystal-chemical knowledge. The present work was designed to examine rigorously the
 128 ordering of Ti in a calcium amphibole **le** for which comparative crystal-chemistry, based on X-ray
 129 diffraction data, indicates that $^{\text{C}}\text{Ti}$ is distributed over all the octahedrally coordinated sites.

130

131 **Sample provenance**

132 One sample was selected from the crystal-chemical database of the Pavia unit of the CNR-IGG
133 institution, which contains the results of complete (EMP+SIMS) chemical analysis, X-ray crystal-
134 structure refinement and in some cases Mössbauer spectroscopy. This sample was chosen because of
135 its composition (*i.e.*, high Ti content) and the availability of large (centimetric) crystals suitable for
136 neutron diffraction. The sample comes from a gabbro at Kaersut (Greenland), the type locality of
137 kaersutite. The sample code at the Muséum National d'Histoire Naturelle in Paris is MHN109.991, and
138 the crystal code in the CNR-IGG database is 632. Previous work based on comparison with X-ray
139 refinement of synthetic high-Ti Fe-free kaersutite (Tiepolo et al. 1999) had suggested that also this
140 sample had a significant amount of Ti at the $M(3)$ site, and that it cannot be considered as an oxo-
141 amphibole (namely, kaersutite) because its (OH+F) content, as measured by SIMS, is higher than 1.0
142 atoms per formula unit (*apfu*).

143

144

145 Experimental

146 - X-ray diffraction

147 Some fragments of a centimetric crystal ($\sim 0.8 \text{ cm}^3$) of the amphibole sample from Kaersut,
148 here labelled as “K(1)”, were used for this multi-methodological study.

149 Unit-cell dimensions of crystal K(1) were calculated from least-squares refinement of the d
150 values obtained from 50 rows of the reciprocal lattice by measuring the centroid of gravity of each
151 reflection in the range $-30 < \theta < +30^\circ$. Intensity data were collected for the monoclinic equivalent
152 pairs (hkl and $h-kl$) in the range $2 < \theta < 65^\circ$. Intensities were then corrected for absorption, Lorentz
153 and polarization effects, averaged and reduced to structure factors. Unit-cell dimensions and
154 miscellaneous information pertaining to intensity-data collection and refinement are given in Table
155 1. Only reflections with $I_o > 3\sigma_I$ in the range $2 < \theta < 30^\circ$ were considered as “observed” during
156 unweighted full-matrix least-squares refinement on F , done with an extensively modified version of
157 the program ORFLS (Busing and Levy 1962) which is able to deal with complex solid-solutions
158 (Cannillo et al. 1983). Fully-ionized scattering factors of the appropriate chemical species were
159 used for non-tetrahedrally coordinated cation sites, whereas appropriate combinations of neutral vs.
160 ionized scattering-factors were used for the T sites and the O atoms (see Oberti et al. 1992). The
161 electron density in the A cavity was modelled using three different subsites, of which only $A(m)$ and
162 $A(2)$ are likely to be occupied by A-group cations. Site populations were derived and optimized by

163 combining refined site-scattering values with refined mean bond lengths, and using all the
164 information derived from statistical treatment of the amphibole data base. More details on the
165 procedure are given in the following sections. Refined atom coordinates and displacement parameters
166 are given in Table 2; selected interatomic distances and angles are given in Table 3, and refined site-
167 scattering values (Hawthorne et al. 1995) are listed in Table 4.

168

169 - Neutron diffraction

170 A crystal of K(1) ($1.5 \times 1.2 \times 0.75 \text{ mm}^3$), optically free of defects and twinning, was selected
171 under a transmitted-light polarizing microscope for the neutron-diffraction experiment. Neutron
172 Laue data were collected at room temperature on the Laue diffractometer KOALA on the OPAL
173 reactor at the Australian Nuclear Science and Technology Organisation. KOALA, which is
174 essentially a clone of the Laue diffractometer VIVALDI at the Institut Laue-Langevin, Grenoble
175 (McIntyre et al. 2006; Edwards 2011; Gatta et al. 2012), uses the Laue diffraction technique on an
176 unmonochromated thermal-neutron beam with a large solid-angle (8 sterad) cylindrical image-plate
177 detector to increase the detected diffracted intensity by one to two orders of magnitude compared to
178 a conventional monochromatic experiment. The crystal was mounted with the high-symmetry axis
179 well away from the single instrument-rotation axis, in order to avoid bias in the final refined
180 anisotropic-displacement parameters due to the blind region in reciprocal space around the rotation
181 axis. Twenty-nine Laue diffraction patterns, each accumulated over 60 min on average and with two
182 different crystal orientations to ensure the best coverage of reciprocal space, were collected at 15°
183 intervals of rotation of the K(1) crystal. Two crystal orientations were used to ensure the best
184 coverage of reciprocal space and avoid bias in the final refined anisotropic-displacement parameters
185 due to the blind region in reciprocal space around the single rotation axis in one orientation. Data
186 extends to a minimum d spacing of $\sim 0.5 \text{ \AA}$ (with $\lambda_{\min} = 0.8 \text{ \AA}$ and $\lambda_{\max} = 5.2 \text{ \AA}$).

187 The Laue patterns were indexed on the basis of the unit-cell parameters previously measured
188 by single-crystal X-ray diffraction (Table 1) and compatible with the reflection conditions of the
189 space group $C2/m$, using the program LAUEGEN of the Daresbury Laboratory Laue Suite
190 (Campbell 1995; Campbell et al. 1998), and the reflections were corrected for background and
191 integrated using the program INTEGRATE+, which uses a two-dimensional version of the
192 minimum $\sigma(I)/I$ algorithm (Wilkinson et al. 1988). Corrections for absorption were deemed
193 unnecessary due to the small crystal volume and low fraction of highly absorbing elements.

194 Reflections were normalised to the incident wavelength, using a curve derived by comparing
195 equivalent reflections and multiple observations, and corrected for the different angles of incidence
196 to the detector surface via the local program LAUE4 (Piltz 2012), and $|F_0| > 4\sigma(F_0)$. Diffraction
197 peaks were indexed on the basis of the unit-cell parameters previously measured by single-crystal
198 X-ray diffraction (Table 1) and compatible with the reflection conditions of the space group $C2/m$.

199 Anisotropic crystal-structure refinement based on the neutron intensity data was done in the
200 space group $C2/m$ using the SHELX-97 software (Sheldrick 2008), starting from a structure model
201 without any H atom included. The neutron scattering lengths of Na, Ca, Fe, Ti, Al, Si, O and H
202 were taken from Sears (1986). Secondary isotropic extinction was corrected by the formalism of
203 Larson (1967), as implemented in the SHELXL-97 package. The structure refinement was done
204 with the following strategy: the $A(m)$ site was modelled with Na alone; the $M(1)$, $M(2)$, and $M(3)$
205 sites were modelled as occupied by Fe and Ti (and their fractions were mutually refined) and $M(4)$
206 by Ca and Na (fractions mutually refined); the $T(1)$ and $T(2)$ sites were modelled with Al and Si
207 (and their fractions mutually refined). Convergence was reached with one intense negative residual
208 peak (at $x \sim 0.208$, $y \sim 0$, $z \sim 0.768$) in the final difference-Fourier map of the nuclear density.
209 Further cycles of refinement were done with H assigned to these coordinates, and its site occupancy
210 was refined. Refinement of all variable parameters converged with all the principal mean-square
211 atomic displacement parameters positive, including those for the H site. The variance-covariance
212 matrix showed no significant correlation among the refined parameters. Site coordinates and
213 displacement parameters are listed in Table 2; selected interatomic distances and angles are given in
214 Table 3, and refined site-scattering values and aggregate neutron-scattering lengths are listed in
215 Table 4.

216

217 - **Microprobe analysis**

218 Subsequent to the collection of the X-ray intensity data, the same fragment of the crystal K(1)
219 was mounted in epoxy, polished, and analysed by electron- and ion-microprobe techniques (for H
220 and Li; Li was found to be at the wt ppm level) following the procedures described by Oberti et al.
221 (1992) and Ottolini et al. (1993, 2001). Chemical compositions (Table 5) are the mean of at least 10
222 determinations on the same crystal; because no significant compositional zoning was observed, they
223 should be representative of the bulk composition of the sample. Formulae were calculated on the
224 basis of $24(\text{O,OH,F}) = 24 \text{ apfu}$. The total number of cations was adjusted so as to obtain the best fit

225 between the refined group-site-scattering values and those calculated from EMP analysis. Given that
226 OH and F were measured by SIMS, this procedure implies a unique solution for the calculation of Fe³⁺
227 within the uncertainty of the SIMS analysis. The initial cation assignment was done as discussed by
228 Hawthorne and Oberti (2007). This showed C-group cations (*i.e.*, Mg, Fe²⁺, Mn²⁺, Ni, Al, Fe³⁺, Cr³⁺,
229 Ti⁴⁺) in excess of 5.0 *apfu*, as is commonly the case in calcium amphiboles. What specific C-group
230 cations should be assigned to the B-group? Usually, one assigns Mn²⁺ and Fe²⁺ in that order, using the
231 site preference $M^{(4)}\text{Mn}^{2+} > M^{(4)}\text{Fe}^{2+}$ observed in monoclinic Fe-Mg-Mn amphiboles as the rationalization
232 for this procedure. In the present case, we know the total scattering from both the B-group [$\equiv M(4)$ site]
233 and the C-group [$\equiv M(1)$, $M(2)$ and $M(3)$ sites] cations, and can check the assignment on this basis. In
234 conclusion, all Mn²⁺ was assigned to $M(4)$ together with 0.73 *apfu* of Fe²⁺. The resultant unit-formula
235 assignment is shown in Table 5.

236

237 **Results: Site assignment of cations**

238 - **T-group cations**

239 Initially, the Al contents at the $T(1)$ and $T(2)$ sites were allowed to vary during refinement of
240 the neutron data. However, the resultant site-populations were far too Al-rich for (1) the chemical
241 formula of the crystal; (2) the observed $\langle T(1)\text{-O} \rangle$ and $\langle T(2)\text{-O} \rangle$ distances, and (3) the grand $\langle T\text{-O} \rangle$
242 distance. We suspected that too much scattering (all the cation sites) was being varied in the neutron
243 refinement, leading to poor scaling, and hence we refined the structure with the site populations of the
244 $T(1)$ and (2) sites fixed at their values obtained from the X-ray structure refinement. Indeed, the $\langle T(1)\text{-O} \rangle$
245 and $\langle T(2)\text{-O} \rangle$ distances derived from the neutron refinement are almost identical to those obtained
246 from the X-ray refinement. According to the systematic work reported in Oberti et al. (2007), in
247 amphiboles the majority of ^TAl occurs at the $T(1)$ site and only minor ^TAl occurs at the $T(2)$ site.
248 Moreover, the ^{T(1)}Al content can be obtained from the regression equation of Oberti et al. (2007) [^{T(1)}Al
249 = ($\langle T(1)\text{-O} \rangle$ - 1.6193) · 34.2199], which gives 1.84 *apfu*. Because the size of the $T(2)$ tetrahedron
250 depends on many compositional factors (including the Fe content), the amount of ^{T(2)}Al cannot be
251 estimated quantitatively by a similar equation, and should be calculated by difference from the ^TAl
252 values obtained from EMP analysis: 1.97 - 1.84 = 0.13 *apfu*. Thus, $T(1) = 2.16 \text{ Si} + 1.84 \text{ Al}$ and $T(2) =$
253 $3.87 \text{ Si} + 0.13 \text{ Al apfu}$.

254

255 - **C-group cations**

256 The assignment of site populations to the $M(1)$, $M(2)$ and $M(3)$ sites is the major goal of this
 257 work. Inspection of Table 5 shows that there are five cations to be assigned to these sites: Mg^{2+} ,
 258 Fe^{2+} , Fe^{3+} , Al^{3+} and Ti^{4+} . We have assigned all Mn^{2+} and some Fe^{2+} to $M(4)$, as indicated by the
 259 chemical formula and the refined site-scattering values at $M(4)$. With a combination of just two
 260 different sets of diffraction data, we cannot uniquely determine site populations for five scattering
 261 species over three sites (Hawthorne 1983), and hence we must reduce this number. In terms of
 262 scattering, we may treat Fe^{2+} and Fe^{3+} as Fe, and assign CAl to $M(2)$ in accord with the known site-
 263 preference for this species in amphiboles containing significant Fe (Hawthorne and Oberti 2007;
 264 Oberti et al. 2007). This leaves three scattering species to be refined over three sites, $M(1,2,3)$.

265 We may write the refined site-scattering values from the X-ray and neutron-scattering
 266 experiments as follows:

267
 268
 269
$$12 \cdot Mg_a^{M(1)} + 26 \cdot Fe_a^{M(1)} + 22 \cdot Ti_a^{M(1)} = XRAY_S^{M(1)}$$

270
$$12 \cdot Mg_a^{M(2)} + 26 \cdot Fe_a^{M(2)} + 22 \cdot Ti_a^{M(2)} = XRAY_S^{M(2)} - 13 \cdot Al_a^{M(2)}$$

271
$$12 \cdot Mg_a^{M(3)} + 26 \cdot Fe_a^{M(3)} + 22 \cdot Ti_a^{M(3)} = XRAY_S^{M(3)}$$

272
 273
$$Mg_B \cdot Mg_a^{M(1)} + Fe_B \cdot Fe_a^{M(1)} + Ti_B \cdot Ti_a^{M(1)} = NEUTRON_S^{M(1)}$$

274
$$Mg_B \cdot Mg_a^{M(2)} + Fe_B \cdot Fe_a^{M(2)} + Ti_B \cdot Ti_a^{M(2)} = NEUTRON_S^{M(2)} - Al_B \cdot Al_a^{M(2)}$$

275
$$Mg_B \cdot Mg_a^{M(3)} + Fe_B \cdot Fe_a^{M(3)} + Ti_B \cdot Ti_a^{M(3)} = NEUTRON_S^{M(3)}$$

276
 277 where $XRAY_S^{M(1)}$ is the refined site-scattering value for the $M(1)$ site from the refinement of the X-
 278 ray data, $NEUTRON_S^{M(1)}$ is the refined site-scattering value for the $M(1)$ site from the refinement of the
 279 neutron data, Mg_B is the neutron scattering length for Mg, $Mg_a^{M(1)}$ is the amount (site population) of
 280 Mg assigned to the $M(1)$ site (etc. for the other sites and the other scattering species). Stoichiometry
 281 requires the following equations (*i.e.*, there are no vacancies at the $M(1,2,3)$ sites):

282
 283
$$Mg_a^{M(1)} + Fe_a^{M(1)} + Ti_a^{M(1)} = 2$$

284
$$Mg_a^{M(2)} + Fe_a^{M(2)} + Ti_a^{M(2)} = 2 - Al_a^{M(2)}$$

285
$$Mg_a^{M(3)} + Fe_a^{M(3)} + Ti_a^{M(3)} = 1$$

286

287 There are nine site-population variables $Mg_a^{M(1)}$ etc. and nine equations, and we may solve for the
288 site populations $X_a^{M(n)}$ where $X = Mg, Fe$ and Ti , and $n = 1-3$. The resultant values are given in
289 Table 6. They confirm that Ti^{4+} occurs in all the three octahedrally coordinated sites in this sample
290 (as suggested by Tiepolo et al. 1999). The unit-cell contents derived from the chemical analysis
291 (Table 5) provide a check on the accuracy of these site populations by comparing the sums of the
292 different scattering species assigned from the diffraction experiments with the same scattering
293 species determined by electron-microprobe analysis, as is done in the two columns to the right in
294 Table 6. The closeness of the two sets of compositions strongly suggests that the derived site
295 populations are correct.

296 The values given in Table 6 are derived solely from the site-scattering values and do not
297 take into account the presence of both Fe^{2+} and Fe^{3+} . We may examine this issue in terms of the
298 geometrical parameters (mean bond-lengths and distortion) at the $M(1)$, $M(2)$ and $M(3)$ sites, in two
299 ways as we now describe:

300 1) Use optimised ideal mean bond-lengths for complete occupancy of the $M(1,2,3)$ sites by Mg
301 (2.078 \AA), Fe^{2+} (2.125 \AA), Fe^{3+} (2.025 \AA), Ti^{4+} (1.960 \AA) and Al (1.929 \AA) for (OH, F, Cl)
302 amphiboles based on the CNR-IGG database in Pavia (Oberti et al. 2007), adjusted for the effect of
303 F at O(3) on the $\langle M(1)-O \rangle$ and $\langle M(3)-O \rangle$ distances ($-0.13 F \text{ apfu}$), and taking into account the
304 observation that significant Fe^{3+} content at the $M(1)$ site causes a considerable increase in the $M(1)$
305 octahedron distortion parameter Δ (Brown and Shannon 1973) from the common value of 2-3. The
306 value of Δ for the $M(1)$ octahedron in crystal K(1) is 2.49, thus excluding the presence of Fe^{3+} at
307 $M(1)$. With no Fe^{3+} at $M(1)$, the calculated $\langle M(1)-O \rangle$ distance is 2.064 \AA , already shorter than the
308 refined value of 2.078 \AA . Thus Fe^{3+} is either ordered at the $M(2)$ site, which would lead to a
309 calculated $\langle M(2)-O \rangle$ distance of 2.049 \AA and a calculated $\langle M(3)-O \rangle$ distance of 2.069 \AA (to be
310 compared to the refined values of 2.074 \AA and 2.070 \AA , respectively), or is distributed between the
311 $M(2)$ and the $M(3)$ sites. This latter hypothesis would make the overall shrinking of the strip of
312 octahedra more homogeneous in crystal K(1), which is consistent with the presence of a significant
313 oxo component. This issue will be explored further in the last section of this paper.

314 2) The regression equations of Hawthorne and Oberti (2007, Table 7) may be used to calculate the
315 Fe^{3+} content of each $M(1,2,3)$ site using the occupancies given in Table 6. The equation for the $M(2)$
316 octahedron involves just the aggregate radius of the constituent $M(2)$ cations, whereas the equations
317 for the $M(1)$ and $M(3)$ octahedra take into account not only the aggregate radius of the constituent

318 $M(1)$ and $M(3)$ cations, but also the aggregate radius of the anion(s) at O(3) and the cations at $M(4)$,
319 the amount of ${}^{\text{T}}\text{Al}$ and the aggregate net charge at the A site. The results are shown below:

320

321 $M(1) \text{Fe}^{3+} = 0.00 \text{ apfu}$, $\langle M(1)\text{-O} \rangle_{\text{obs}} = 2.078$ and $\langle M(1)\text{-O} \rangle_{\text{calc}} = 2.077 \text{ \AA}$;

322 $M(2) \text{Fe}^{3+} = 0.00 \text{ apfu}$, $\langle M(2)\text{-O} \rangle_{\text{obs}} = 2.073$ and $\langle M(2)\text{-O} \rangle_{\text{calc}} = 2.073 \text{ \AA}$;

323 $M(3) \text{Fe}^{3+} = 0.10 \text{ apfu}$, $\langle M(3)\text{-O} \rangle_{\text{obs}} = 2.070$ and $\langle M(3)\text{-O} \rangle_{\text{calc}} = 2.070 \text{ \AA}$.

324

325 This gives a significantly lower Fe^{3+} content than that obtained by formula recalculation with the
326 constrain of H and F measured by SIMS (i.e., 0.33 apfu). The predicted $\langle M\text{-O} \rangle$ values have
327 standard errors of estimate of 0.0027, 0.0053 and 0.0058 \AA for $M(1)$, $M(2)$ and $M(3)$, respectively,
328 which at their limits would predict a further Fe^{3+} content of 0.023 Fe^{3+} . Anyway, these results
329 indicate that Fe^{3+} is strongly ordered at the $M(3)$ site and is in reasonable accord with
330 electroneutrality using the neutron-derived H content of 1.02(2) apfu.

331 The presence of a significant amount of Ti^{4+} at the $M(1)$ site is usually detected by a much
332 higher value of the anisotropic displacement parameters (*adp*) refined at this site than at the $M(2)$
333 and $M(3)$ sites, indicating a shift of the Ti^{4+} ions toward the edge of the octahedron connecting the
334 two O(3) sites (Tiepolo et al. 1999). In some cases, two split positions, $M(1)$ and $M(1')$, can even be
335 refined (e.g., Hawthorne et al. 2000). In this work, a higher value of the *adp* at $M(1)$ and $M(3)$ was
336 obtained from the X-ray data but not from the neutron data. The presence of Ti^{4+} at $M(1)$, $M(2)$ and
337 $M(3)$, as well as the presence of significant F, may be the reason for the unusual distortion pattern of
338 the ribbon of octahedra observed in crystal K(1), which is generally more shrunk than expected
339 based on the ideal $\langle \text{cat-O} \rangle$ distances used in the crystal-chemical analysis of amphiboles (cf. Oberti
340 et al. 2007 and the discussion above).

341

342 - B-group cations

343 Although two different subsites (i.e., $M(4)$ and $M(4')$; Table 2) can be refined from X-ray
344 data due to the oblong shape of the electron density, their distance is close to the resolution of the
345 data. Hence, the total scattering obtained is more accurate than that based on the refinement of a
346 unique position, but its partitioning cannot be considered accurate. However, inspection of the cat-O
347 distances and comparison with the shape of the electron density found in many other amphibole
348 compositions show that smaller cations (Mg , Fe^{2+} , Mn^{2+}) order at the $M(4')$ position (Oberti et al.

349 2007). Cations were assigned to the B-group in the usual fashion (e.g., Hawthorne et al. 2012).
350 Those C-group cations in excess of 5 *apfu* were assigned as B-group cations in the following
351 sequence: Mn²⁺>Fe²⁺>Mg, and all Ca was assigned to this site until the site is completely occupied
352 (Hawthorne et al. 2012). The resultant calculated X-ray/neutron site-scattering values (Table 4) are
353 in close agreement with the analogous refined values for both the X-ray and neutron refinements.

354

355 - **A-group cations**

356 The site populations at the *A(m)*, *A(2)* and *A* sites were refined unconstrained in both X-ray
357 and neutron refinements. Although the model used for neutron data includes only the *A(m)* subsite,
358 as the *A* and *A(2)* site populations were found not to be significant, the aggregate refined
359 scattering/neutron-scattering value in the *A* cavity is close to that calculated from the A-group
360 cations based on electron- and ion-microprobe analysis (Table 4). The higher value obtained by X-
361 rays structure refinement may be due to the relatively high values of the atomic displacement
362 parameters (Table 2).

363

364 - **W-group anions**

365 The site population of the H site was refined directly in the neutron structure refinement,
366 with convergence to the following value: 1.02(2) H + 0.98 □ *apfu*. This may be compared to the
367 value derived by SIMS of 0.87 H *apfu*. Ottolini and Hawthorne (2001) give an uncertainty of 5-10%
368 for the determination of H by SIMS for the analytical method used here, *i.e.*, 0.04-0.09 *apfu* for crystal
369 K(1). The upper limit of the uncertainty plus the standard deviation of the H value derived by neutron
370 scattering suggest that there is little significant difference between these two values.

371

372

373 **Discussion and conclusions**

374 - **Hydrogen bonding**

375 The neutron structure refinement provides a clear picture of the H site location, anisotropic
376 displacement regime and potential H-bonding configuration. The coordinates of the H site obtained
377 in this study are consistent with previous experimental findings based on single-crystal neutron
378 structure refinement reported by Pechar et al. (1989) for a kaersutite from Bohemia. The refined O-
379 H bond distance of this study is 0.951(2) Å (Table 3), whereas that reported by Pechar et al. (1989)

380 is 0.93(1) Å. A correction for the “riding motion effect” (following the protocol of Busing and Levy
381 1964) gives 0.982 Å (Table 3), showing that the O–H distance is significantly affected by the
382 motion of the proton. The magnitude and orientation of the displacement ellipsoid of the H site of
383 this study is shown in Fig. 2, along with the potential H-bonding network. The shape of the
384 ellipsoid (root-mean-square components: $R1:R2:R3 = 1.89:1.84:1$, with $R1$ and $R2$ dispersed on
385 (100), Fig. 2) is strongly influenced by the geometry of the weak interactions with the oxygens of
386 the two independent tetrahedra. **Inspection** of the difference-Fourier map of the nuclear density does
387 not show any evidence of splitting of the H site. Pechar et al. (1989) did not report any indication of
388 possible hydrogen bonding. However, our data lead us to consider that at least three potential weak
389 H-bonds may occur: with **O(6) x 2 and O(7)** as *acceptors* (i.e., $H\cdots O(6) = 2.763(2)$ Å, $O(3)\cdots O(6)$
390 $= 3.316(1)$ Å and $O(3)–H\cdots O(6) = 117.86(1)^\circ$; $H\cdots O(7) = 2.694(3)$ Å, $O(3)\cdots O(7) = 3.270(2)$ Å and
391 $O(3)–H\cdots O(7) = 119.6(2)^\circ$, Table 3). A similar trifurcated configuration is observed in trioctahedral
392 phyllosilicates, in which the *donor* is an oxygen site shared by three adjacent octahedra (belonging
393 to the octahedral sheet) and the *acceptors* are oxygens of the superimposed 6-membered ring of
394 tetrahedra (belonging to the tetrahedral sheet) (e.g., Gatta et al. 2011, 2013, 2014). In all these
395 cases, the $O_{donor}–H\cdots O_{acceptor}$ angles range between 115–140°.

396

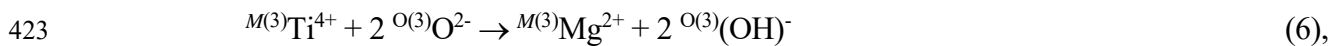
397 - The relation between $^{6l}Ti^{4+}$ and $^{O(3)}O^{2-}$

398 The Mg-Fe-Ti site populations of the C cations calculated from the refined X-ray and
399 neutron site-scattering values are in excellent agreement with the analysed chemical compositions
400 (Table 6). Ti^{4+} is preferentially ordered at the $M(1)$ site, in accord with the dominance of
401 substitution (2) (see **above**). **The** refined Ti^{4+} content of $M(1)$ is 0.38 *apfu* (Table 6); using the
402 neutron site population for hydrogen at the H site, the site population of O(3) is $1.02 (OH)^- + 0.20 F^-$
403 $+ 0.78 O^{2-}$. These values, $^{M(1)}Ti^{4+} = 0.38$ *apfu* and $^{O(3)}O^{2-} = 0.78$ *apfu*, are in almost exact accord
404 with substitution (2), suggesting that the incorporation of O^{2-} at O(3) is controlled by the occurrence
405 of Ti^{4+} at $M(1)$, and the bond-valence requirements of Ti^{4+} at $M(1)$ are met primarily by very short
406 bonds to O^{2-} at the O(3) site.

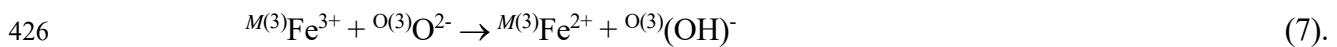
407 However, there is significant Ti^{4+} at the $M(2)$ and $M(3)$ sites (Table 6). Incorporation of Ti^{4+}
408 at the $M(2)$ site is a result of substitution (3) and is unrelated to dehydrogenation in amphiboles. The
409 incorporation of Ti^{4+} at $M(3)$ is more problematic. Incorporation of Ti^{4+} at $M(3)$ could in principle
410 be related to dehydrogenation as the $M(3)$ site is also coordinated by two O(3) sites. However, in

411 amphibole K(1), all O²⁻ at O(3) is associated with Ti⁴⁺ at M(1), and Ti⁴⁺ at M(3) cannot be
 412 associated independently with O²⁻ at O(3) as all O²⁻ at O(3) is already associated with Ti⁴⁺ at M(1).
 413 This being the case, it is not clear how the bond-valence requirements of Ti⁴⁺ at M(3) can be
 414 satisfied. Possibly, ^{M(3)}Ti⁴⁺ could be incorporated into the amphibole structure by, for example, a
 415 substitution such as ^{M(3)}Ti⁴⁺ + 2 ^{M(2)}Mg²⁺ → ^{M(3)}Mg²⁺ + 2Al³⁺, that is locally spatially associated with
 416 substitution (2), such that ^{M(3)}Ti⁴⁺ bonds to ^{O(3)}O²⁻. An analogous mechanism could also account for the
 417 occurrence of Fe³⁺ at M(3).

418 The Ti⁴⁺ content at the M(1) site compensates for 0.76 *apfu* O²⁻ at O(3). This leaves a
 419 difference of 0.02 (0.78 – 0.76 from the neutron H-site occupancy) to 0.17 *apfu* (0.93 – 0.76 from
 420 the SIMS H content) to be compensated by additional substitutions. Given the chemical formula of
 421 the sample, two possibilities remain: (1) the additional positive charge is provided by half the Ti⁴⁺
 422 occurring at M(3), which implies substitution (6) below:

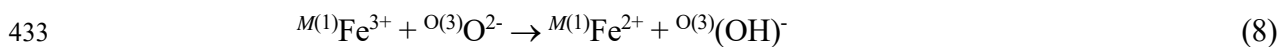


424 or (2) the additional positive charge is provided by up to 0.17 Fe³⁺ *apfu* occurring at M(3) according
 425 to substitution (7):



427 Note that substitution (7) is common in oxo-amphiboles where the Ti content is low or
 428 dehydrogenation has occurred after crystallization (*e.g.*, Oberti et al. 2016).

429 Above, we considered two approaches concerning the amount and distribution of Fe³⁺ at the
 430 M(1), M(2) and M(3) sites. Both exclude the presence of Fe³⁺ at the M(1) site and are compatible
 431 with the occurrence of Fe³⁺ at the M(3) site. In this regard, crystal K(1) is unusual as
 432 dehydrogenation of Fe-rich amphiboles generally proceeds according to substitution (8) below:



434 The second approach predicts 0.13 *apfu* Fe³⁺ at M(3), within the range 0.02-0.17 *apfu* required by
 435 electroneutrality, highlighting the unusual nature of this amphibole.

436
437

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Table 1. Summary of crystallographic, data collection and structure refinement information.

	K(1) X-ray	K(1) neutron*
<i>a</i> (Å)	9.882(4)	9.882(4)
<i>b</i> (Å)	18.084(7)	18.084(7)
<i>c</i> (Å)	5.312(1)	5.312(1)
β (°)	105.26(6)	105.26(6)
<i>V</i> (Å ³)	915.84	915.84
Space group	<i>C2/m</i>	<i>C2/m</i>
<i>Z</i>	2	2
Crystal size (mm)	0.75 x 0.43 x 0.22	2.0 x 1.5 x 0.9
Rad./mono.	MoK α	Thermal neutrons / Laue, (λ_{\min} =0.8, λ_{\max} =5.2 Å)
No. of reflections	16801	31909
Unique reflections	1254	3099
Unique observed reflections	1210	2261
<i>d</i> _{min} (Å)	0.7	0.5
<i>R</i> _{sym} %	3.70	6.13
No. of refined parameters	129	116
** <i>R</i> ₁ (<i>F</i>) %	1.78	5.25
<i>R</i> _{all} %	2.28	7.45
Maximum residual peaks	***0.47 e-/Å ³	+2.1/-2.8 fm/Å ³

*Cell dimensions taken from X-ray data; ***R*₁(*F*) % calculated on the basis of *F*_o > 3σ(*F*_o) for the X-ray data and *F*_o > 4σ(*F*_o) for the neutron data; ***close to the *M*(1) site, can be attributed to the ordering of Ti⁴⁺.

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Table 2. Atom coordinates and displacement parameters (\AA^2) for the amphibole of this work.

Site	<i>x</i>	<i>y</i>	<i>z</i>	U^{11*}	U^{22*}	U^{33*}	U^{12*}	U^{13*}	U^{23*}	$U^{\text{eq}*}$
<i>K(1)–X-ray</i>										
<i>T</i> (1)	0.28234(5)	0.08503(3)	0.30227(10)	0.0058(2)	0.0066(2)	0.0066(2)	–0.0009(2)	0.0007(2)	–0.0003(2)	0.0065(2)
<i>T</i> (2)	0.29067(5)	0.17235(3)	0.81009(9)	0.0056(2)	0.0075(2)	0.0060(2)	–0.0009(2)	0.0011(2)	0.0001(2)	0.0064(2)
<i>M</i> (1)	0	0.08646(4)	½	0.0098(3)	0.0176(3)	0.0081(3)	0	0.0040(2)	0	0.0115(2)
<i>M</i> (2)	0	0.17730(4)	0	0.0065(4)	0.0073(3)	0.0067(4)	0	0.0018(2)	0	0.0068(2)
<i>M</i> (3)	0	0	0	0.0102(5)	0.0073(5)	0.0086(5)	0	–0.0009(3)	0	0.0093(3)
<i>M</i> (4)	0	0.27936(3)	½	0.0115(3)	0.0078(3)	0.0113(3)	0	0.0068(2)	0	0.0095(2)
<i>M</i> (4')	0	0.2621(3)	½	0	0	0	0	0	0	0.0171(10)
<i>A</i>	0	½	0	0.076(7)	0.021(4)	0.076(7)	0	0.074(7)	0	0.048(4)
<i>A</i> (<i>m</i>)	0.0537(7)	½	0.1067(12)	0.053(3)	0.025(2)	0.047(3)	0	0.036(3)	0	0.037(2)
<i>A</i> (2)	0	0.4737(5)	0	0.051(5)	0.037(4)	0.079(7)	0	0.055(5)	0	0.049(4)
<i>O</i> (1)	0.10776(13)	0.08645(7)	0.2191(3)	0.0078(6)	0.0128(7)	0.0093(6)	–0.0012(4)	0.0017(5)	–0.0001(5)	0.0101(4)
<i>O</i> (2)	0.11914(13)	0.17219(7)	0.7283(3)	0.0073(6)	0.0114(5)	0.0098(6)	–0.0004(4)	0.0014(4)	0.0009(5)	0.0097(3)
<i>O</i> (3)	0.1082(2)	0	0.7150(4)	0.0092(8)	0.0156(8)	0.0136(9)	0	0.0021(7)	0	0.0130(5)
<i>O</i> (4)	0.3656(2)	0.24973(8)	0.7864(3)	0.0140(6)	0.0109(7)	0.0120(6)	–0.0027(5)	0.0037(5)	0.0008(5)	0.0123(4)
<i>O</i> (5)	0.34973(14)	0.13938(8)	0.1091(3)	0.0094(6)	0.0167(7)	0.0130(6)	–0.0005(5)	0.0012(5)	0.0051(5)	0.0134(4)
<i>O</i> (6)	0.34609(14)	0.11660(8)	0.6080(3)	0.0094(6)	0.0167(7)	0.0150(6)	0.0009(5)	0.0027(5)	–0.0051(5)	0.0138(4)
<i>O</i> (7)	0.3428(2)	0	0.2782(4)	0.0119(9)	0.0157(10)	0.0206(11)	0	0.0015(8)	0	0.0166(6)
<i>K(1)–Neutron</i>										
<i>T</i> (1)	0.28216(6)	0.08500(3)	0.30226(12)	0.0052(2)	0.0060(2)	0.0060(2)	–0.0006(1)	0.0006(2)	–0.0006(1)	0.0059(2)
<i>T</i> (2)	0.29056(5)	0.17238(3)	0.81021(11)	0.0059(2)	0.0065(2)	0.0060(2)	–0.0009(1)	0.0012(2)	0.0003(1)	0.0062(1)
<i>M</i> (1)	0	0.09012(3)	½	0.0074(3)	0.0023(2)	0.0059(3)	0	0.0016(2)	0	0.0052(2)
<i>M</i> (2)	0	0.17697(3)	0	0.0071(3)	0.0069(2)	0.0068(3)	0	0.0018(2)	0	0.0069(2)

<i>M</i> (3)	0	0	0	0.0070(3)	0.0062(3)	0.0056(4)	0	0.0016(3)	0	0.0063(2)
<i>M</i> (4)	0	0.27828(4)	½	0.0119(3)	0.0145(3)	0.0112(3)	0	0.0065(3)	0	0.0119(2)
<i>A</i> (<i>m</i>)	0.0369(7)	½	0.0762(13)	0.089(7)	0.118(5)	0.093(6)	0	0.077(5)	0	0.090(3)
O(1)	0.10791(4)	0.08643(2)	0.21876(9)	0.0074(2)	0.0119(2)	0.0095(2)	-0.0012(1)	0.0012(1)	-0.0001(1)	0.0098(1)
O(2)	0.11925(4)	0.17211(2)	0.72815(8)	0.0066(2)	0.0104(1)	0.0095(2)	-0.0002(1)	0.0008(1)	0.0012(1)	0.0091(1)
O(3)	0.10828(6)	0	0.71491(13)	0.0095(2)	0.0142(2)	0.0148(3)	0	0.0013(2)	0	0.0132(1)
O(4)	0.36545(5)	0.24984(2)	0.78717(9)	0.0136(2)	0.0093(2)	0.0121(2)	-0.0029(1)	0.0035(2)	0.0001(1)	0.0116(1)
O(5)	0.34970(4)	0.13944(2)	0.10881(9)	0.0099(2)	0.0157(2)	0.0118(2)	-0.0009(1)	0.0010(2)	0.0050(1)	0.0128(1)
O(6)	0.34592(4)	0.11669(2)	0.60790(9)	0.0106(2)	0.0151(2)	0.0140(2)	0.0005(1)	0.0027(2)	-0.0057(1)	0.0134(1)
O(7)	0.34272(7)	0	0.27894(14)	0.0126(3)	0.0139(2)	0.0210(3)	0	0.0020(2)	0	0.0163(1)
H	0.2080(3)	0	0.7672(7)	0.013(1)	0.044(2)	0.045(2)	0	0.004(1)	0	0.034(1)

* for *K*(1) – *X*-ray, the U^{ij} values are calculated from the refined β^i values. The atom- displacement parameter takes the form: $\exp [-2\pi^2 (h^2a^{*2}U^{11} + k^2b^{*2}U^{22} + l^2c^{*2}U^{33} + 2hka^*b^*U^{12} + 2hla^*c^*U^{13} + 2klb^*c^*U^{23})]$.

Table 3. Selected interatomic distances (Å) and angles (°) for the amphibole of this work.

	X-ray	Neutron		X-ray	Neutron
<i>T</i> (1)–O(1)	1.665(2)	1.661(1)	<i>T</i> (2)–O(2)	1.635(2)	1.633(1)
<i>T</i> (1)–O(5)	1.680(2)	1.683(1)	<i>T</i> (2)–O(4)	1.604(2)	1.604(1)
<i>T</i> (1)–O(6)	1.680(2)	1.680(1)	<i>T</i> (2)–O(5)	1.653(2)	1.651(1)
<i>T</i> (1)–O(7)	1.667(1)	1.666(1)	<i>T</i> (2)–O(6)	1.667(2)	1.667(1)
< <i>T</i> (1)–O>	1.673	1.673	< <i>T</i> (2)–O>	1.640	1.639
<i>M</i> (1)–O(1) x2	2.049(1)	2.052(1)	<i>M</i> (2)–O(1) x2	2.130(2)	2.125(1)
<i>M</i> (1)–O(2) x2	2.123(2)	2.075(1)	<i>M</i> (2)–O(2) x2	2.091(1)	2.092(1)
<i>M</i> (1)–O(3)	2.062(1)	2.112(1)	<i>M</i> (2)–O(4) x2	2.000(2)	2.002(1)
< <i>M</i> (1)–O>	2.078	2.080	< <i>M</i> (2)–O>	2.073	2.073
<i>M</i> (3)–O(1) x4	2.069(1)	2.068(1)	<i>A</i> –O(5) x4	3.058(2)	
<i>M</i> (3)–O(3) x2	2.071(2)	2.072(1)	<i>A</i> –O(6) x4	3.071(2)	
< <i>M</i> (3)–O>	2.070	2.070	<i>A</i> –O(7) x2	2.410(2)	
			< <i>A</i> –O>	2.846	
<i>A</i> (<i>m</i>)–O(5) x2	3.026(2)	3.027(4)			
<i>A</i> (<i>m</i>)–O(6) x2	2.633(2)	2.754(5)	<i>M</i> (4)–O(2) x2	2.421(2)	2.407(1)
<i>A</i> (<i>m</i>)–O(7)	2.485(2)	2.435(7)	<i>M</i> (4)–O(4) x2	2.328(2)	2.328(1)
<i>A</i> (<i>m</i>)–O(7)	2.516(2)	2.481(6)	<i>M</i> (4)–O(5) x2	2.656(2)	2.667(1)
< <i>A</i> (<i>m</i>)–O>	2.720	2.746	<i>M</i> (4)–O(6) x2	2.577(2)	2.591(1)
			< <i>M</i> (4)–O>	2.495	2.523
<i>A</i> (2)–O(5) x2	2.679(2)				
<i>A</i> (2)–O(6) x2	2.766(2)		<i>M</i> (4')–O(2) x2	2.180(1)	
<i>A</i> (2)–O(7) x2	2.456(2)		<i>M</i> (4')–O(4) x2	2.278(2)	
< <i>A</i> (2)–O>	2.634		<i>M</i> (4')–O(5) x2	2.840(2)	
			<i>M</i> (4')–O(6) x2	2.811(2)	
			< <i>M</i> (4')–O>	2.527	
O(3)–H		0.951(2)	O(3)–H \cdots O(6) x2		117.86(1)
O(3)–H*		0.982	H \cdots O(7)		2.694(3)
H \cdots O(6)		2.763(2)	O(3) \cdots O(7)		3.270(2)
O(3) \cdots O(6)		3.316(1)	O(3)–H \cdots O(7)		119.6(2)

*Bond distances corrected for riding motion effect following Busing and Levy (1964).

Table 4. Site-scattering values refined for the amphibole of this work.

Site	X-ray (<i>epfu</i>)		Neutron (barns <i>pfu</i>)
	Refined	EMPA	Refined
<i>M</i> (1)	34.08		9.144
<i>M</i> (2)	30.83		10.487
<i>M</i> (3)	17.62		5.120
<i>Sum C</i>	82.53	82.62	24.75
<i>M</i> (4)	36.45		
<i>M</i> (4')	4.14		
<i>Sum B</i>	40.59	40.52	9.53
<i>A</i> (<i>m</i>)	6.14		
<i>A</i> (2)	4.26		
<i>A</i>	2.55		
<i>Sum A</i>	12.96	12.64	3.69

Table 5. Chemical composition (wt.%) and unit formula (*apfu*) derived from EMP and SIMS analysis (H, F) of crystal K(1). *bdl* = below detection limit.

SiO ₂	39.99(47)
TiO ₂	6.07(11)
Al ₂ O ₃	12.02(24)
Cr ₂ O ₃	0.02(2)
*Fe ₂ O ₃	2.88
*FeO	6.77
MnO	0.13(2)
NiO	0.02(2)
MgO	13.50(22)
CaO	12.19(23)
Na ₂ O	2.50(15)
K ₂ O	0.93(4)
H ₂ O	0.86
F	0.42(5)
Cl	<i>bdl</i>
-O=F	0.18
Total	98.13
Si	6.031
Al	1.969
<i>sum T</i>	8.000
Al	0.168
Fe ³⁺	0.327
Ti	0.688
Cr	0.002
Ni	0.002
Mg	3.032
Fe ²⁺	0.781
<i>sum C</i>	5.000
Fe ²⁺	0.073
Mn ²⁺	0.017
Ca	1.910
<i>sum B</i>	2.000
Ca	0.060
Na	0.731
K	0.179
<i>sum A</i>	0.970
OH	0.865
F	0.200
O	0.935
<i>sum W</i>	2.000
<i>Note:</i> *FeO _{tot} = 9.36(31) wt%; OH and F were measured by SIMS, this procedure implies a unique solution for the calculation of Fe ²⁺ /Fe ³⁺ .	
Li was found to be at the wt ppm level.	

Table 6. Site populations (*apfu*) derived from X-ray and neutron site-scattering refinements for the amphibole of this work

	<i>M</i> (1)	<i>M</i> (2)	<i>M</i> (3)	Σ	EMPA
Mg	1.17	1.28	0.55	3.00	3.03
Fe	0.45	0.39	0.29	1.13	1.11
Ti	0.38	0.16	0.16	0.70	0.69
Al	-----	0.17	-----	0.17	0.17
Σ	2.00	2.00	1.00	5.00	5.00

Figure 1. Projection onto (100) of the structure of amphibole K(1). The site nomenclature is also shown. Structure model based on the neutron refinement of this study; displacement probability factor: 50%.

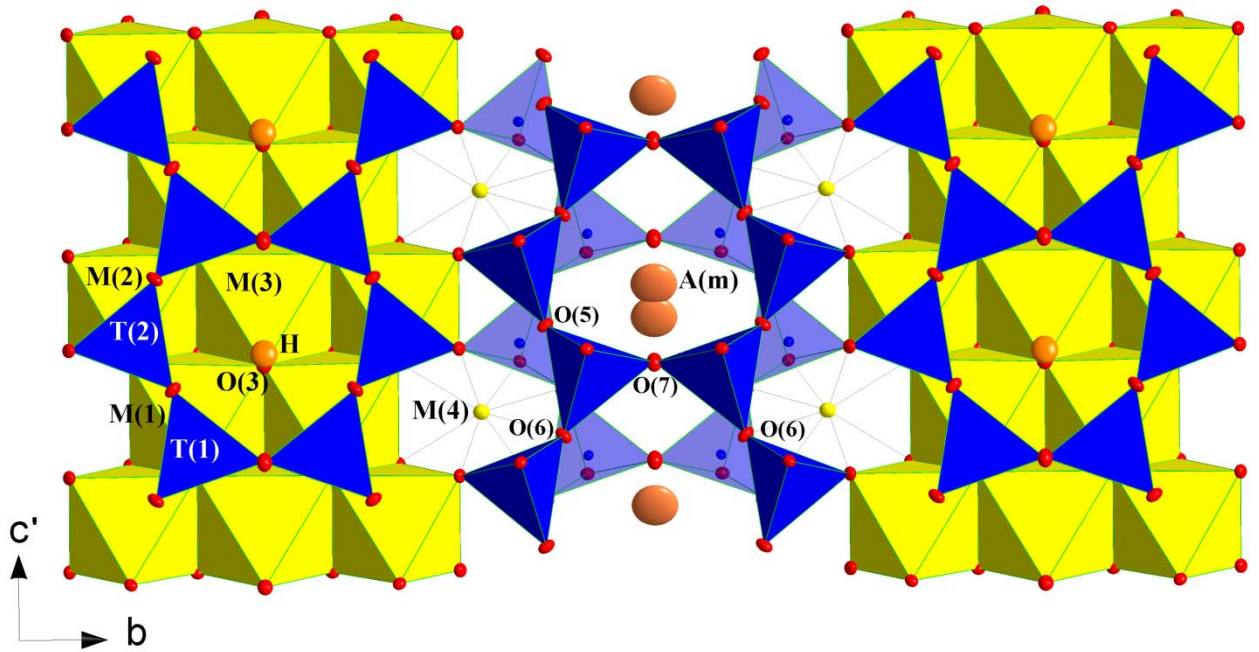


Figure 2. H-bonding configuration in the structure of K(1) amphibole deduced on the basis of the neutron structure refinement of this study. Displacement probability factor: 50%.

