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Electronic Structure of Superconducting KC₈ and Nonsuperconducting LiC₆ Graphite Intercalation Compounds: Evidence for a Graphene-Sheet-Driven Superconducting State

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We have performed photoemission studies of the electronic structure in LiC₆ and KC₈, a nonsuperconducting and a superconducting graphite intercalation compound, respectively. We have found that the charge transfer from the intercalant layers to graphene layers is larger in KC₈ than in LiC₆, opposite of what might be expected from their chemical composition. We have also measured the strength of the electron-phonon interaction on the graphene-derived Fermi surface to carbon derived phonons in both materials and found that it follows a universal trend where the coupling strength and superconductivity monotonically increase with the filling of graphene π^* states. This correlation suggests that both graphene-derived electrons and graphene-derived phonons are crucial for superconductivity in graphite intercalation compounds.

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In graphite intercalation compounds (GICs), the intercalation of various atomic or molecular species in between graphene layers in graphite leads to novel properties and a very rich physics, including superconductivity [1]. In graphite intercalated with alkaline metals, superconductivity has been known for decades [2], but after the recent discovery of relatively high T_c superconductivity in CaC₆ $(T_c = 11.5 \text{ K})$ [3,4], research in this field has been intensified. Even though the electron-phonon coupling (EPC) is most likely responsible for pairing in GICs [5–7], it is still not clear what electronic states, intercalant- or graphenederived ones, and what phonons are responsible for pairing [8–11]. Because of differences in structure and composition, no clear trends have been identified that could unambiguously resolve these issues. For example, KC₈ is a superconductor and LiC₆ is not. Further, in GICs intercalated with alkaline earths, T_c ranges from zero to 11.5 K, even though they share the same chemical formula MC₆, where M is an alkaline-earth atom. On the other hand, band structure calculations show that in graphite and GICs an interlayer state exists above the π^* band [12,13], prompting some researchers to propose that its partial filling and coupling to soft intercalant phonons induces superconductivity in GICs [8,14]. The experimental situation is still inconclusive, with strong advocates for intercalant [7] and graphene dominated superconductivity [5,15–17]. A recent angle-resolved photoemission spectroscopy (ARPES) study on CaC₆ [15] reported that EPC on a graphenederived Fermi surface (FS) to graphene phonons is strong enough to explain a T_c in the range of tens of Kelvin, indicating that graphene sheets provide crucial ingredients for superconductivity in GICs. However, to test this idea, it would be important to extend similar studies to GICs with different T_c .

In this Letter, we report ARPES studies of the electronic structure and the EPC in the nonsuperconducting LiC₆ and in the superconducting KC₈ ($T_c = 0.39$ K) and compare these materials with several other GICs. We find that the EPC on the graphene-derived π^* states to the graphenederived phonons increases with the filling of π^* states in a sequence from LiC₆ to KC₈ to CaC₆, following the same trend as T_c . The positive correlation between these quantities implies that superconductivity originates in graphene sheets while the main role of intercalants is to provide the charge for filling of the graphene π^* states.

The experiments were carried out on a Scienta SES-100 electron spectrometer operating in the angle-resolved mode at the beam line 12.0.1 of the Advanced Light Source. The spectra were recorded at the photon energy of 50 eV, with the combined instrumental energy resolution of 20-25 meV and the momentum resolution of $\pm 0.008 \text{ Å}^{-1}$ in geometry where the polarization of light was perpendicular to the probed momentum line. The LiC₆ and KC₈ samples were prepared by intercalating natural, single-crystal graphite flakes (Madagascan) using the vapor transport method as described in Ref. [1]. X-ray diffraction showed very high sample purity with no graphite or secondary stage phases. To avoid degradation, all samples were unsealed and glued to the sample holder with Ag-epoxy in an Ar filled glow box. Protected by the cured epoxy, they were then quickly transferred to the ARPES prep chamber, and cleaved at low temperature (15–20 K) under ultrahigh vacuum conditions $(2 \times 10^{-9} \text{ Pa})$. All data were collected at 15–20 K.

Figure 1 shows the ARPES spectra near the K point in the graphene Brillouin zone for LiC₆ and KC₈. The upper panels (a) and (b) show the contours of photoemission intensity as a function of binding energy for a momentum line going through the K point. The intensity from a narrow interval (± 10 meV) around the Fermi level, representing the FS, is shown in the lower panels (c) and (d). The dispersing states are the graphene-derived π and π^* bands, as marked in Figs. 1(a) and 1(b). In KC₈ the low energy band structure is essentially graphenelike, with π and π^* bands touching at the Dirac point [16] which is shifted below the Fermi level due to doping. In LiC₆, a sizable gap exists between the π and π^* band. The Dirac point is determined by extrapolating the linear part of the π^* dispersion at low binding energies to the K point. The arrows in Figs. 1(a) and 1(b) indicate the position of the Dirac point (E_D) , 0.825 eV and 1.35 eV for LiC₆ and KC₈, respectively. It is clear from Fig. 1 that the π^* band is filled more in KC₈ than in LiC₆ and that it forms a larger FS in the former material. The area enclosed by the FS (lower panels in Fig. 1) is a direct measure of doping of graphene π^* states, i.e., of the charge transferred into the graphene sheets. The FS is determined from peak positions of

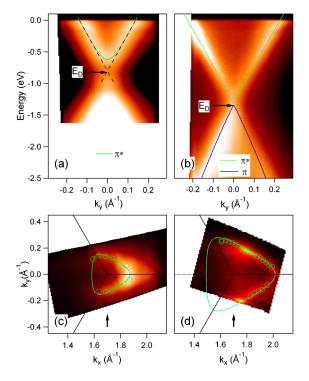


FIG. 1 (color online). Photoemission spectra from LiC₆ (a) and KC₈ (b) along the same momentum line in the graphene Brilluoin zone traversing the K point, as indicated by the arrows in panels (c) and (d). Lines represent the π and π^* bands. Arrows indicate the binding energy of the Dirac point. (c) and (d) Photoemission intensity from a narrow energy interval around the Fermi level ($\omega=\pm 10$ meV), representing the graphene-derived π^* FS, for LiC₆ and KC₈, respectively. Circles represent the MDC peak positions, while lines represent the tight-binding fits to the data, as described in the text.

momentum distribution curves (MDCs) at E_F (open circles) and compared to the 3rd nearest neighbor hopping tight-binding band structure (lines). In KC₈, the occupied FS area is 0.399 Å⁻², which corresponds to 0.11 electrons per graphene unit cell (GUC), or 44% of the nominal value of 0.25 for the complete charge transfer. In LiC₆, the occupied π^* (green circles and lines) FS area is 0.125 Å⁻², corresponding to the doping of only 0.0344 electrons per GUC. This is far below the nominal value of 0.33 electrons per GUC.

The incomplete charge transfer into the graphene π^* states would suggest that the remaining charge occupies the so-called interlayer band. The occupation of the interlayer state has been recently reported in CaC₆ [18], but the observed feature was very weak and broad. Even the graphene-derived π^* state was very broad and did not form an enclosed contour at the Fermi level, casting doubts on these results. Our experiments always show a relatively sharp π^* band that forms a well-defined FS. However, in $MC_6(M = \text{Li, Ca, Ba})$ materials, in addition to the π^* band, we always see a broad feature at slightly higher binding energy that follows the π^* band, dispersing upward from the K point [Fig. 1(a)]. Farther from the K point, the broad feature loses intensity and its dispersion cannot be precisely traced. In CaC₆, this feature is observable over a larger region of k space (Fig. 1 in [15]). It is possible that this feature is a remnant of an interlayer band, smeared out by a disorder within the intercalant layers and folded into the K point of the graphene Brillouin zone. However, our measurements do not show any evidence of the interlayer band in the region from which it should be folded to the K point—the Γ point. We note that in pristine graphite the interlayer hopping t_{\perp} splits both π and π^* bands into the bonding and antibonding counterparts due to the ABstacking of graphene sheets. It would be tempting to assign the broad feature as a bonding π^* state, due to similarities in initial dispersion. However, all first stage GICs have the AA stacking of graphene sheets and such an assignment would be incorrect. On the other hand, in LiC₆ and CaC₆ the π^* band might be split due to the AB stacking of the intercalant (if there is not too much disorder in the intercalant sites). If this was indeed the case, the correct charge transfer to the graphene layers would be 0.0616 and 0.349 electrons per GUC, for LiC₆ and for CaC₆, respectively.

Irrespective of these issues, our experimental observation that the doping of graphene sheets is larger in KC_8 than in LiC_6 is opposite of the expected nominal doping, but is in line with the existence of superconductivity in these materials: KC_8 is a superconductor and LiC_6 is not. In the following, we identify the reason for the correlation between superconductivity and doping of the graphene sheets. It is evident from Fig. 1 that in both LiC_6 and KC_8 an anomaly or a kink in dispersion of the π^* band occurs at approximately 0.165 eV below the E_F . This is a hallmark of the interaction of the electronic states with

phonons [19] that have been previously observed in CaC₆ and KC₈ [15,16] and attributed to a coupling to graphene in-plane high-frequency phonons. To quantify the electron-phonon coupling, we have used the standard MDC fitting procedure [20,21] which uses a tight-binding dispersion as the starting approximation for the bare band and gives the real (Re Σ) and imaginary (Im Σ) part of self-energy as fitting parameters. The bare band dispersion is then refined until the obtained Re Σ and Im Σ satisfy Kramers-Kronig transformations [22]. Panels (a) and (c) in Fig. 2 show the Im Σ while panels (b) and (d) show Re Σ for both materials for several different locations on the FS, as indicated in figure.

Re Σ in both materials shows a peak at around -0.165 eV, while Im Σ shows a decrease below that energy, indicating a coupling to the phonon mode. The only phonons with such high energy are graphene-derived inplane phonon modes. A small variation in the energy at which $Re\Sigma$ has a maximum at different points on the FS indicates a slight dispersion of the mode. We note that the sharp increase of Re Σ below 20 meV is an artifact of finite energy resolution of the experimental apparatus [23]. We have excluded the affected interval $|\omega| < 20$ meV from the considerations and any fine structure, related to a possible coupling to the intercalant modes, is out of our detection limits. However, the lack of broadening of the π^* states with increasing temperature over the range of 15 K < T <200 K suggests that the low energy modes play an insignificant role and that the EPC is dominated by the graphene-derived high-frequency modes.

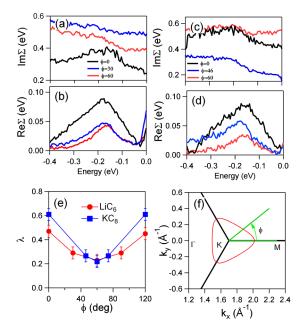


FIG. 2 (color online). $\text{Im}\Sigma(\omega)$ (a) and $\text{Re}\Sigma(\omega)$ (b) for LiC_6 and KC_8 [(c) and (d)] for several different points at the FS, as indicated in (e) and (f). (e) The electron-phonon coupling strength, λ , for LiC_6 and KC_8 , extracted from $\text{Re}\Sigma(\omega)$ as a function of the polar angle ϕ as defined in (f).

The coupling constant λ can be extracted directly from $\operatorname{Re}\Sigma$ as $(\lambda = -[\partial(\operatorname{Re}\Sigma)/\partial\omega]_0)$ by fitting the low energy part to a straight line. It shows some anisotropy [Fig. 2(e)], with the maximum along the KM direction and the minimum along the ΓK direction, similar, but significantly smaller than in CaC_6 [15] and what was recently reported for KC_8 [16]. The most important observation, however, is that the momentum-averaged $\langle \lambda_k \rangle$ is stronger in KC_8 than in LiC_6 . The coupling constant and its anisotropy both increase from LiC_6 to KC_8 to CaC_6 , exactly in the same sequence as the filling of the graphene π^* band and in the previously established sequence for T_c . Strengthening of EPC with the filling of the π^* band has also been observed in the epitaxial graphene [24]. This is not surprising because the density of states near the E_F increases with

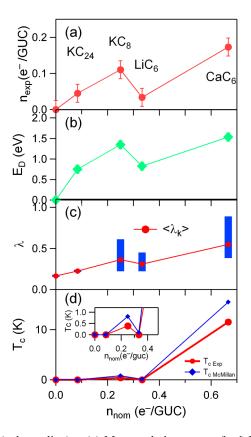


FIG. 3 (color online). (a) Measured charge transfer [electrons per graphene unit cell (GUC)] from the intercalant to the graphene π^* states. (b) Binding energy of Dirac point. (c) EPC coupling constant λ . The range of each vertical bar indicates the anisotropy with the maximum along the KM line and minimum along the $K\Gamma$ line. Red circles represent the momentum-averaged coupling constant $\langle \lambda_k \rangle$. (d) Superconducting transition temperature T_c in GICs. Red dots represent experimental values [2-4], and blue diamonds are calculated from the measured $\langle \lambda_k \rangle$, shown in (c), using McMillan's formula [26]. The inset zooms in on the low T_c values. All the quantities are shown as functions of nominal chemical composition for measured GICs. Zero corresponds to the pristine graphite. Data for CaC6 and pristine graphite are from Ref. [15] and for KC₂₄ are from Ref. [27].

the filling of the π^* band and a larger FS makes an EPC process more probable as the phase space available for the scattering events grows. In the pristine graphite, the FS is nearly a point and the EPC is strongly suppressed[15,25].

To better illustrate a positive correlation between T_c , λ , and doping of the graphene π^* band in different GICs, we plot these quantities in Fig. 3 as functions of nominal chemical composition for several different materials. Actual (measured) charge transfer (in electrons/GUC) is shown in Fig. 3(a). The increase in size of the FS is consistent with the energy of the Dirac point [panel (b)] as the chemical potential, μ , shifts from the pristine graphite to CaC₆. It is interesting to note that $\mu \propto \sqrt{n_{\rm exp}}$ still holds, regardless of the shape of FS and number of FS sheets in these five different materials. The coupling constant λ and T_c follow the same trend. This suggests that the graphene π^* states and their coupling to graphene in-plane phonons are crucial for superconductivity and that the only role that the intercalants seem to play is to provide the charge for filling of the π^* states. This is further reinforced by the calculated T_c using McMillan's formula [26]:

$$T_c = \frac{\Theta}{1.45} \exp\left(-\frac{1.04(\lambda+1)}{\lambda - (0.62\lambda+1)\mu^*}\right),$$
 (1)

where we use measured $\langle \lambda_k \rangle$ [panel(c)], Debye teperature $\Theta=1926$ K, and Coulomb pseudopotential $\mu^*\sim 0.14$. As shown in Fig. 3(d), the calculated T_c values are very close to the ones measured experimentally. The threshold-like behavior of T_c near $\langle \lambda_k \rangle=0.3$ places LiC₆ on one side and KC₈ on another side of a steep increase in T_c .

We note that superconductivity in LiC_3 and LiC_2 , materials in which more Li is pushed in under pressure, supports our picture where the EPC and superconductivity strengthen with the filling of graphene π^* states. The increase in T_c from 0.39 K for stoichiometric KC₈ to 0.55 K in material with excess K is also in line with this picture. A further test would be a systematic ARPES study on alkaline-earth GICs (Ca, Sr, Ba) where T_c decreases with the atomic mass of the alkaline-earth intercalant.

In conclusion, we have identified the universal trend in alkali and alkaline-earth GICs where superconductivity is tightly correlated with the doping of graphene-derived π^* states and with the coupling of these states to graphene phonons. This implies that the graphene sheets play the crucial role in superconductivity in GICs.

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