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# A critical review of high entropy alloys (HEAs) and related concepts

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# A CRITICAL REVIEW OF HIGH ENTROPY ALLOYS AND RELATED CONCEPTS

## *Beyond Nickel-Based Superalloys-II*

Cambridge, UK

20 July 2016

A CRITICAL REVIEW OF HIGH ENTROPY ALLOYS AND RELATED CONCEPTS  
DBM and O.N. Senkov, *Acta Mater.*, OVERVIEW, In Review.

NEW STRATEGIES AND TESTS TO ACCELERATE DISCOVERY AND DEVELOPMENT OF MULTI-  
PRINCIPAL ELEMENT STRUCTURAL ALLOYS DBM, B.S. Majumdar, K. Wertz and S.  
Gorsse, *Scripta Mater.*, Accepted.

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# OUTLINE



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## **INTRODUCTION**

*History, Definitions, Hypotheses, Visualizations*

## **THERMODYNAMICS**

*Enthalpy and Entropy of Solid Solution and Intermetallic Phases*

## **TAXONOMY**

*Elemental Constituents, Alloy Families*

## **MICROSTRUCTURES**

*Definitions, Observed and Calculated Phases, Assessment*

## **PROPERTIES**

*Functional, Mechanical, Assessment*

## **APPLICATIONS & DESIGN**

*Functional, Structural*

## **FUTURE WORK**

*Basic and Applied*

## **MAJOR ACCOMPLISHMENTS**

*Ideas, Achievements, Closing Remarks*



# MANUSCRIPT APPROACH



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## Each section is written in an independent style

- Extensively cross-referenced to other sections, separate summaries

## Only major results are shown here

- Emphasize *Mechanical Properties* and *Major Accomplishments*

## Critical review of main hypotheses recently published

- Scope is different but conclusions are the same



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**High-entropy alloys: a critical assessment of their founding principles and future prospects**

E. J. Pickering & N. G. Jones

**To cite this article:** E. J. Pickering & N. G. Jones (2016): High-entropy alloys: a critical assessment of their founding principles and future prospects, International Materials Reviews, DOI: [10.1080/09506608.2016.1180020](https://doi.org/10.1080/09506608.2016.1180020)

**To link to this article:** <http://dx.doi.org/10.1080/09506608.2016.1180020>





# EARLY HEA HYPOTHESES

## *It's Time to Learn and Move Ahead!*



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**If it's published often enough, it's accepted as fact, even if there are no data to support it!**

- ✓ *...more than expected 1-phase, solid solution microstructures with 'simple' crystal structures...*
- ✓ *...unusually strong and ductile...*
- ✓ *...sluggish diffusion...*
- ✓ *...violates Gibbs phase rule...*

**NONE OF THESE STATEMENTS ARE CORRECT**

**THERE ARE GREAT REASONS TO STUDY HEAs, BUT NOT THESE**

***The problem lies not with the new ideas, but in letting go of the old ones.***

**John Maynard Keynes**



# INTRODUCTION

## Definitions and Use of Terms



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**Intended or not, HEAs are associated with controlling configurational entropy to produce 1-phase solid solutions**

- This stirs unproductive controversy that distracts from the major objective of exploring vast composition space by placing limits on possibilities

**We apply a careful use of terms to avoid these implications**

- We use the term, “HEA” when configurational entropy or the intent to produce 1-phase, solid solution microstructures are important
- We use multi-principle element alloys (MPEAs) or complex concentrated alloys (CCAs) to evoke vast composition space with no restrictions concerning entropy or the phases present
- *This is a practical approach to clarify discussion, to focus efforts, and to avoid an unproductive controversy, and is not intended to detract from contributions of pioneers in the field*

**(CCA, MPEA) = HEA w/o (SS, 1-phase, entropy)**



# THERMODYNAMICS

## All Terms Can Be Important



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**Four primary thermodynamic terms must be considered in the competition between SS and IM phases**

- Each of these four terms can be significant

**The competition between phases is controlled by relatively small differences between these four larger values**

- Usually (*but not always!*),  $(H^{SS} - H^{IM}) > 0$  and  $(S^{SS} - S^{IM}) > 0$

**No single value or pair of values consistently dominate phase selection**

**These considerations make it difficult to accurately predict phases by focusing on a single thermodynamic term**

$$\Delta G^{SS-IM} = G^{SS} - G^{IM} = (H^{SS} - H^{IM}) - T(S^{SS} - S^{IM})$$



# TAXONOMY Summary

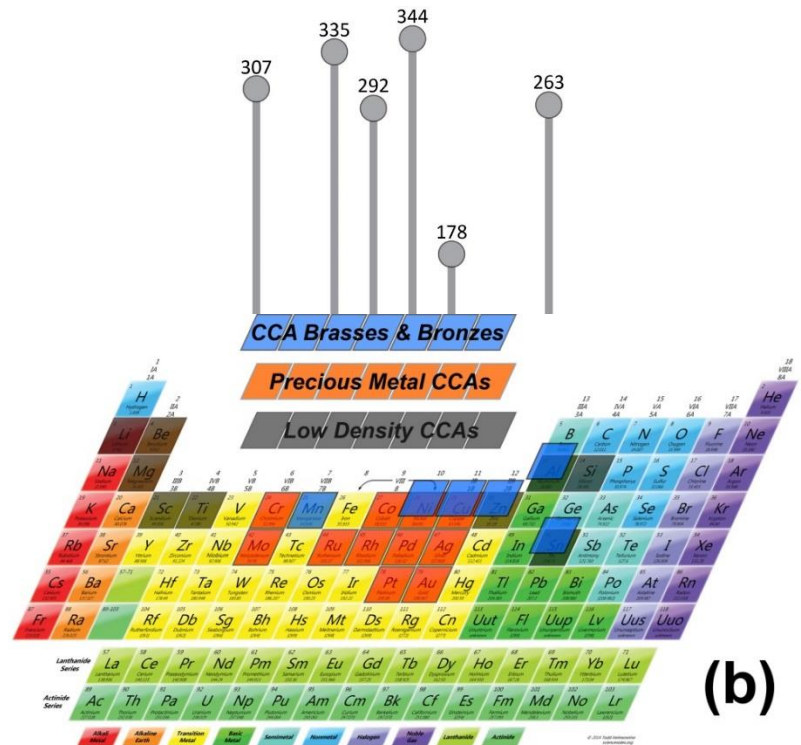
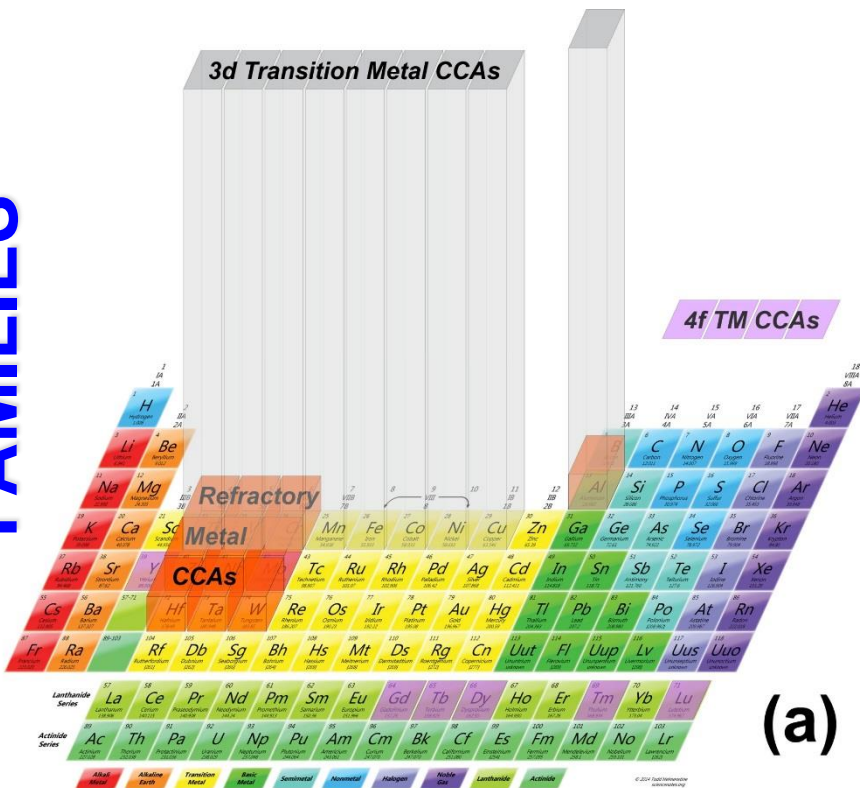


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A remarkable focus is found on elements used in MPEAs, and in groupings of elements

A broader use of element groupings is now underway

**MPEA ALLOY  
FAMILIES**







# MECHANICAL PROPERTIES

## 3d Transition Metal MPEAs



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**MPEAs are often claimed to be unusually strong and ductile, but available data don't support these claims**

**3d transition metal MPEAs have essentially the same properties as austenitic stainless steels and nickel alloys**

- Generally have RT  $\sigma_y$  below 300 MPa and  $\sigma_{uts}$  below 700 MPa
- Significant strengthening from work-hardening and grain refinement are not expected to be effective above about  $T_m/2$  (about 500C)

**Austenitic stainless steels and nickel alloys are 3d transition metal CCAs**

- All austenitic stainless steels and nickel alloys have Cr-Fe-Ni as principal elements and can also have significant additions of Co, Cu, Mn, Mo

**3d TM MPEA strengths do not compete with precipitation hardened stainless steels or nickel superalloys**

- $\sigma_y < 100$  MPa at about 1100 K, but superalloy sheet requires  $\sigma_y > 200$  MPa and blades require  $\sigma_y > 600$  MPa

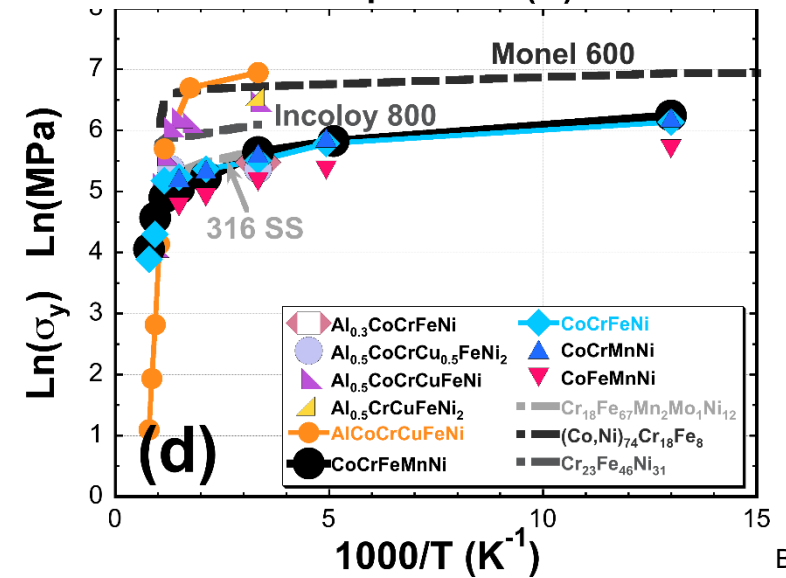
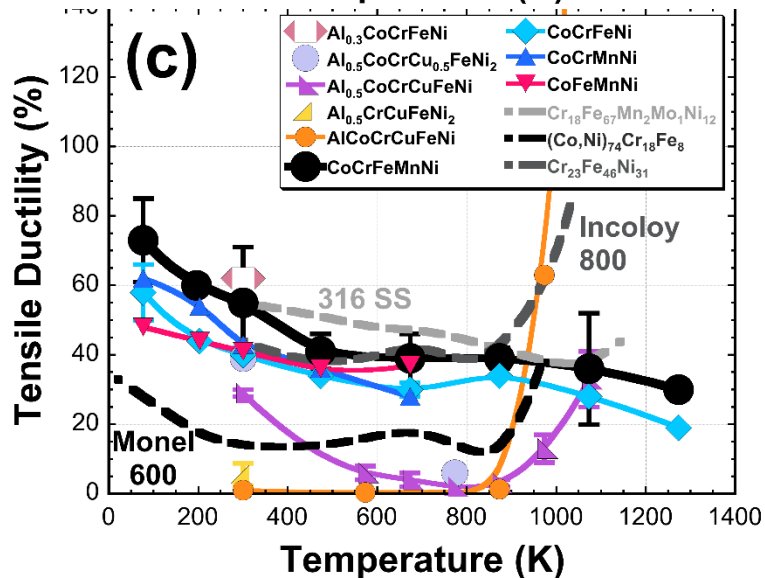
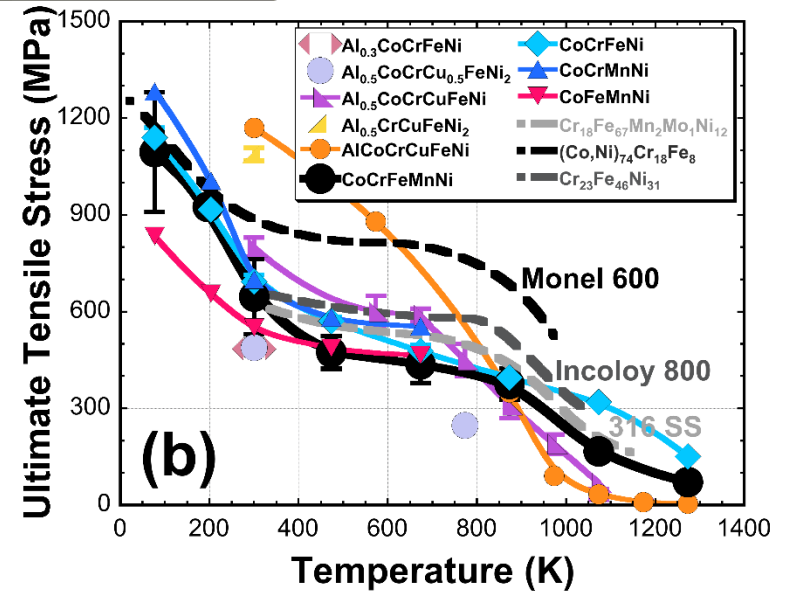
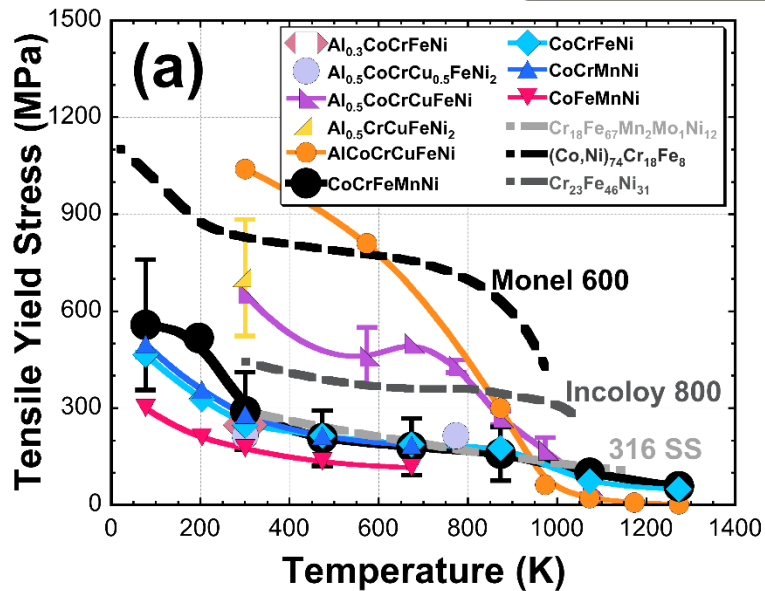


# MECHANICAL PROPERTIES

## 3d Transition Metal CCAs (Tension)



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# REFRACTORY METAL CCAs

## Structural Properties Summary



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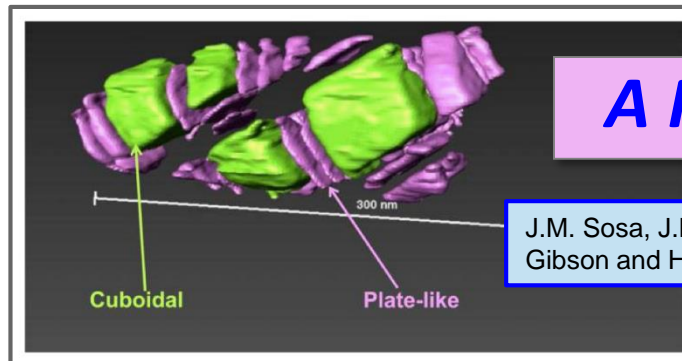
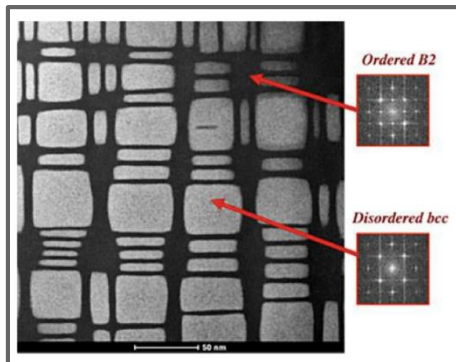
### Refractory metal CCAs were created to compete with nickel superalloys

- Refractory metal CCAs are still in relatively early stages of exploration (only 26 alloys in this assessment)
- A wide range in elemental properties (density, modulus, environmental resistance) suggest a wide range in alloy properties

**Densities range from 5.59 to 13.75 g cm<sup>-3</sup>**

**Most alloys have one or two BCC phases, one or more Laves phases, and one alloy reports a B2 phase**

- The full extent of the BCC solid solution phase field has not been explored



***A R-CCA Superalloy!***

J.M. Sosa, J.K. Jensen, D.E. Huber, G.B. Viswanathan, M.A. Gibson and H.L. Fraser; *Mat. Sci. Tech.*, **31**, 1250 (2015).





# REFRACTORY METAL CCAs

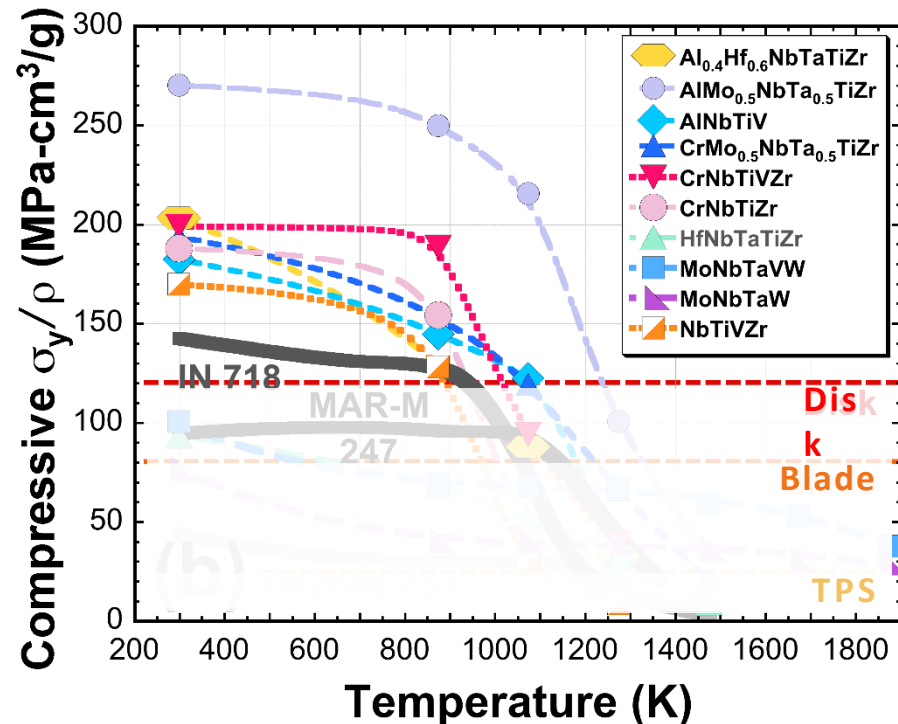
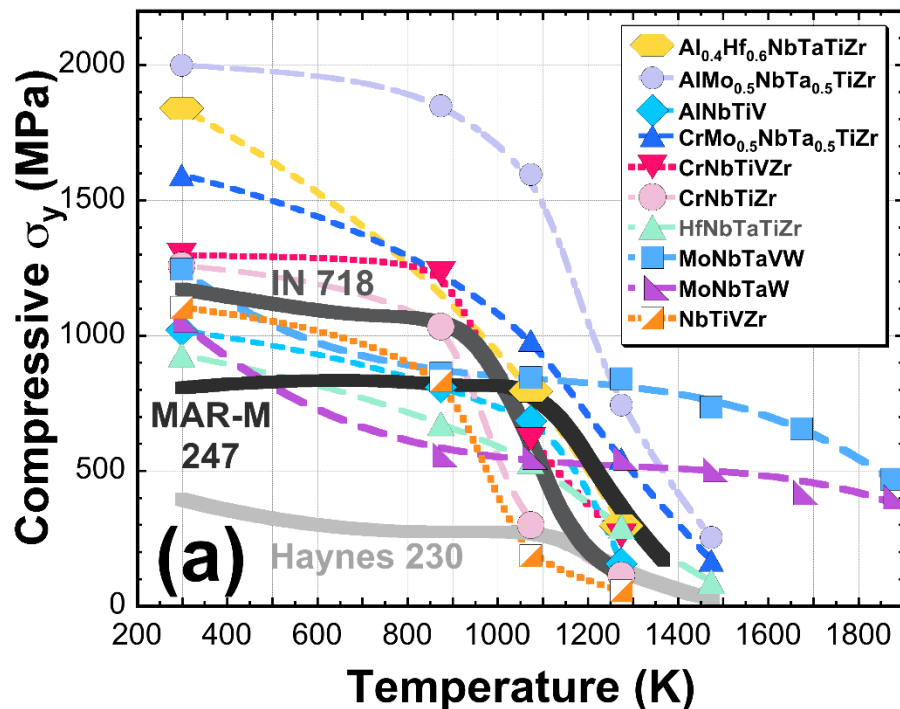
## Structural Properties Summary



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**Most mechanical properties are measured in compression with only a few tensile studies**

- Several alloys have yield and specific yield strengths that show potential to increase stresses and/or temperatures compared to superalloys
- The strongest alloys have  $\geq 10\%$  RT compressive ductility, offering the possibility of useful tensile ductility





# REFRACTORY METAL CCAs

## *Suggested Future Work*



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### **Continued exploration of principal element combinations**

- New alloy bases
- Control the type, volume fraction, size, morphology and distribution of second phases to design precipitation strengthened microstructures
- Extent of solid solution phases

### **Tensile properties of alloys with compressive ductility $\geq 10\%$**

### **Characterize oxidation behavior and explore alloying to improve environmental stability**

- Simple, hi-throughput oxidation screening tests are recommended

### **Effect of deformation processing on microstructure, properties**

- Cold-rolling has been demonstrated in one alloy



# A NEW STRATEGY TO RAPIDLY CHARACTERIZE ALLOYS



NEW STRATEGIES AND TESTS TO ACCELERATE DISCOVERY AND DEVELOPMENT OF MULTI-PRINCIPAL ELEMENT STRUCTURAL ALLOYS  
DBM, B.S. Majumdar, K. Wertz and S. Gorsse, *Scripta Mater.*, Accepted.

## Conventional Characterization

### 1<sup>st</sup> Tier

- Tensile strength, ductility

### 2<sup>nd</sup> Tier

- Fracture toughness
- Fatigue
- Creep

### Final Characterizations

- Environmental resistance
- '...ilities'

### Strengths

- Enables intuitive 'pre-selection' when a knowledge base already exists for a small number of candidate alloy systems.

### Weaknesses

- Starts with difficult-to-measure properties that depend on both composition and microstructure, making it hard to reject alloys quickly.

## New Characterization Strategy

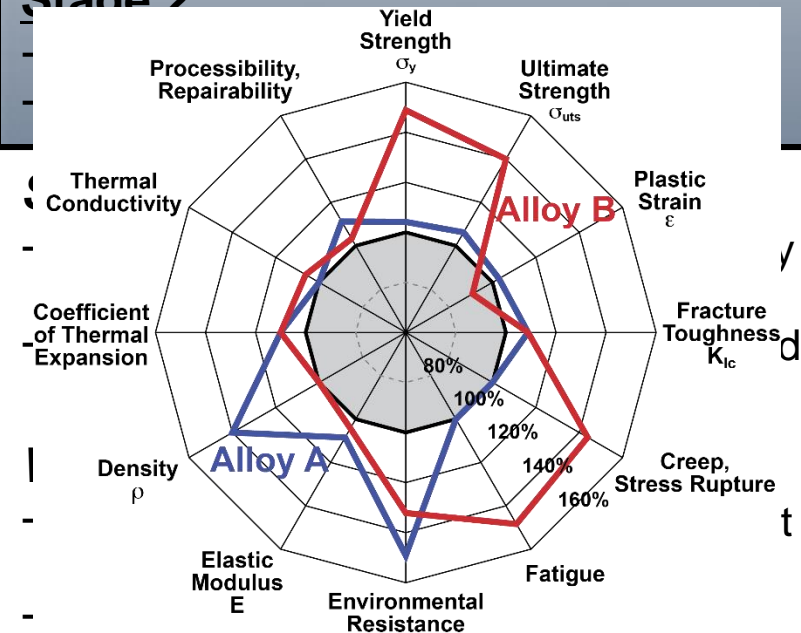
### Stage 0

- Hi throughput phase diagram calculations

### Stage 1

- Microstructure-independent properties
- Environmental resistance
- Modulus, density, thermal properties

### Stage 2





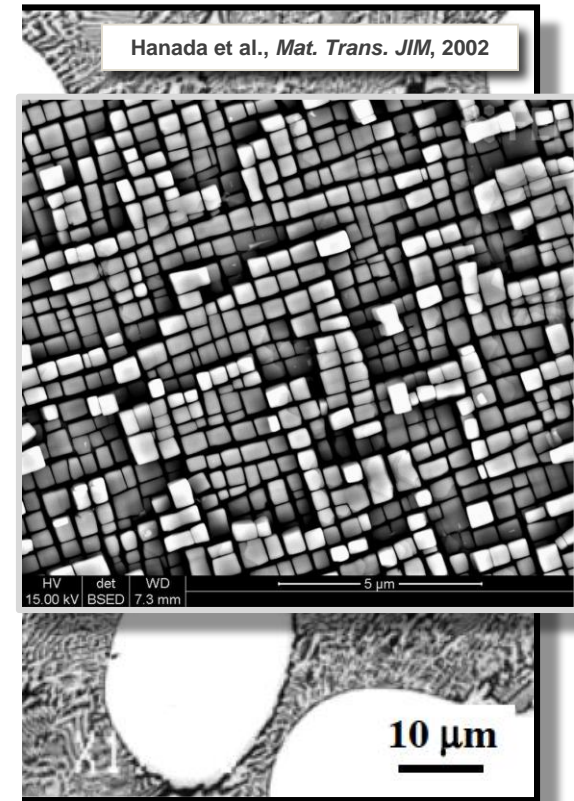
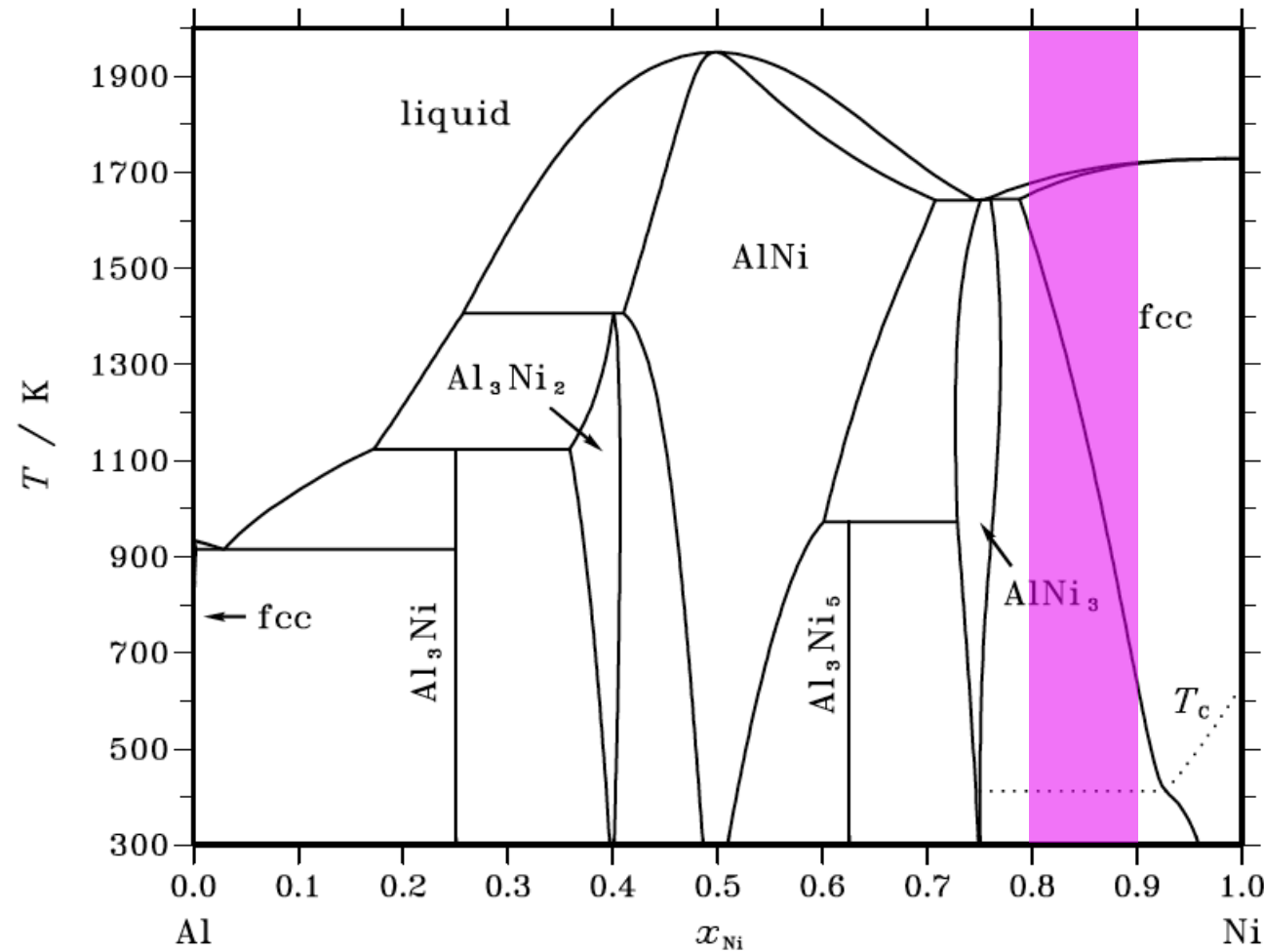
# PHASE DIAGRAM CALCULATIONS



## A Quick Head Start to Alloy Discovery

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Specific phase diagram features enable two-phase microstructures that may give a good balance of properties





# PHASE DIAGRAM CALCULATIONS



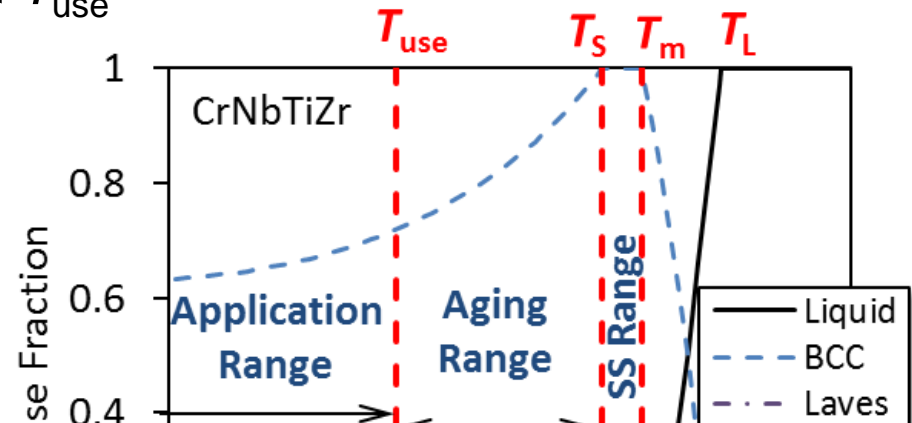
## A Quick Head Start to Alloy Discovery

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Specific phase diagram features enable two-phase microstructures that may give a good balance of properties

### Phase diagram criteria from CALculated PHase Diagram (CALPHAD) calculations

- $T_m > T_{use}$ , no phase transformation  $< T_{use}$
- At least 1 SS phase and no more than 1 IM phase at  $T_{use}$
- One phase must dissolve above  $T_{use}$
- fcc, bcc, hcp or ordered derivative crystal structures
- *These are selective criteria*



Calculation of 135,000 phase diagrams has identified ~100 candidate high temperature structural alloy systems

Senkov et al., Acta

CALPHAD: Computer Coupling of Phase Diagrams and Thermochemistry 50 (2015) 32–48

Contents lists available at ScienceDirect

CALPHAD: Computer Coupling of Phase Diagrams and Thermochemistry

journal homepage: [www.elsevier.com/locate/calphad](http://www.elsevier.com/locate/calphad)

Accelerated exploration of multi-principal element alloys for structural applications

O.N. Senkov\*, J.D. Miller, D.B. Miracle, C. Woodward

Air Force Research Laboratory, Materials and Manufacturing Directorate, Wright-Patterson AFB, OH 45433, USA





# HIGH-THROUGHPUT EXPERIMENTS



## Stage 1

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### STAGE 0

Calculated Phase Diagrams

### STAGE 1

Structure-Insensitive Properties

### STAGE 2

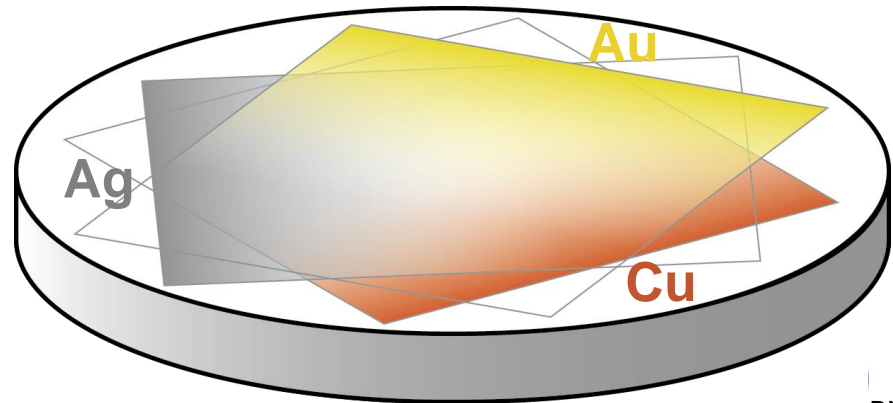
Structure-Sensitive Properties

**Materials libraries with controlled composition gradients**

- Density, modulus,  $T_m$ , **environmental resistance**

**Environmental resistance is probably the most impactful test**

- Possibility to reject many candidates early in the evaluation process
- Fast and conceptually straightforward
- Has not yet been demonstrated





# HI THRU-PUT EXPERIMENTS



## Stage 2

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### STAGE 0

Calculated Phase Diagrams

### STAGE 1

Structure-Insensitive Properties

### STAGE 2

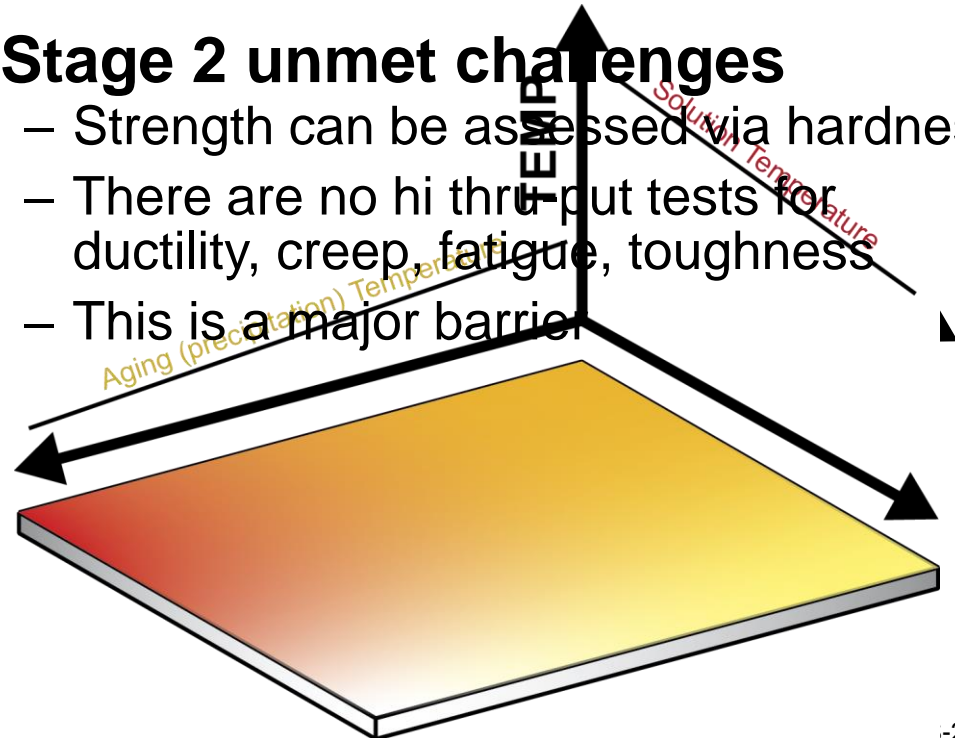
Structure-Sensitive Properties

#### Fixed-composition materials libraries with controlled microstructure gradients

- Strength, tensile ductility, creep, fracture, fatigue
- Libraries are conceptually simple

#### Stage 2 unmet challenges

- Strength can be assessed via hardness
- There are no hi thru-put tests for ductility, creep, fatigue, toughness
- This is a major barrier





# MAJOR ACCOMPLISHMENTS

## Four HEA Core Effects



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**Present data & analyses do not support an observable effect of configurational entropy on preferred formation of 1-phase alloys, solid solution phases or simple crystal structures**

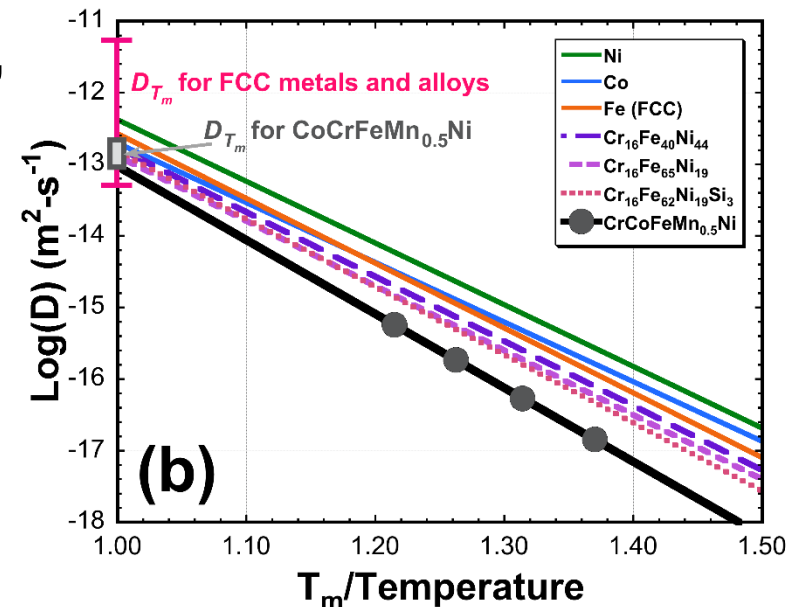
- Two direct experiments and two computational studies do not support the ‘high entropy’ hypothesis
- Six issues bias observations toward 1-phase/ SS/ ‘simple’ microstructures
- A simple ‘structure in – structure out’ (SISO) analysis links observed alloy microstructures to the structures of the constituent elements

**Limited data show that MPEA diffusion is not unusually ‘sluggish’**

- In the same range as FCC elements and conventional alloys

**Insufficient data to evaluate the ‘lattice distortion’ hypothesis**

**The ‘cocktail effect’ is a colorful phrase for ‘non-linearity’ of elemental combinations in general**





# MAJOR ACCOMPLISHMENTS



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**Introduced and established the concept of exploring the vast compositional realm of multi-principle element alloys.**

**Establish importance – not dominance – of configurational entropy as a control variable in phase selection.**

**Define/explore 7 new alloy families and 122 new base alloys, emphasizing 3d transition metal and refractory metal alloys.**

**Expand the known 3d transition metal FCC phase field and discover up to 10 other extended 1-phase fields.**

**Establish composition/ microstructure/ properties relations for 3d transition metal MPEAs**

**Expand initial ideas to include multi-phase microstructures, functional materials and high-throughput evaluations.**

**High throughput calculations identify over 200 alloy bases as potential high temperature structural materials.**



# RECOMMENDATIONS



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## **Continue to explore and develop MPEAs for structural use**

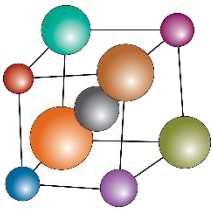
- Expand the range of alloy families, possibilities have barely been scratched
- Include both SS and precipitation strengthened microstructures
- Expand thermo-mechanical processing efforts to control microstructure

## **Apply strategy linking high-throughput computations & experiments to develop high temperature MPEAs**

- High throughput experiments for structural materials requires new tools

## **Establish basic scientific concepts**

- Strengthening models
- Lattice distortion effects



# THANKS!



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# Crystal structures at RT and at T<sub>m</sub>

All but 2 allotropic transformations convert to BCC at T<sub>m</sub>

**Table 3**

Elemental properties and the number of multi-component alloys that use the indicated elements.

Element	Atomic Number	Used in # of alloys	Structure at RT*	Structure at T <sub>m</sub> *	Element	Atomic Number	Used in # of alloys	Structure at RT*	Structure at T <sub>m</sub> *
H	1				Rh	45	1	A1 (FCC)	
Li	3	7	A2 (BCC)		Pd	46	3	A1 (FCC)	
Be	4	1	<b>A3 (HCP)</b>	<b>A2 (BCC)</b>	Ag	47	1	A1 (FCC)	
B	5	3	– (hR105)		Cd	48		A3 (HCP)	
C	6	7	A3 (HCP)		In	49		A6 (tl2)	
N	7				Sn	50	17	A5 (tl4)	
O	8				Sb	51		A7 (hR2)	
Na	11		A2 (BCC)		Te	52		A8 (hP3)	
Mg	12	8	A3 (HCP)		Cs	55		A2 (BCC)	
Al	13	263	A1 (FCC)		Ba	56		A2 (BCC)	
Si	14	15	A4 (cubic) <sup>1</sup>		La	57		<b>A3 (DHCP)</b>	<b>A1 (FCC)</b>
P	15		– (oC8)		Ce	58		<b>A3 (HCP)</b>	<b>A2 (BCC)</b>
S	16		– (oF128)		Pr	59		<b>A3 (DHCP)</b>	<b>A2 (BCC)</b>
K	19		A2 (BCC)		Nd	60	1	<b>A3 (DHCP)</b>	<b>A2 (BCC)</b>
Ca	20		<b>A1 (FCC)</b>	<b>A2 (BCC)</b>	Sm	62		<b>C19 (hR3)</b>	<b>A2 (BCC)</b>
Sc	21	1	<b>A3 (HCP)</b>	<b>A2 (BCC)</b>	Eu	63		A2 (BCC)	
Ti	22	101	<b>A3 (HCP)</b>	<b>A2 (BCC)</b>	Gd	64	2	<b>A3 (HCP)</b>	<b>A2 (BCC)</b>
V	23	55	A2 (BCC)		Tb	65	2	<b>A3 (HCP)</b>	<b>A2 (BCC)</b>
Cr	24	307	A2 (BCC)		Dy	66	2	<b>A3 (HCP)</b>	<b>A2 (BCC)</b>
Mn	25	79	<b>A12 (cI58)</b>	<b>A2 (BCC)</b>	Ho	67		A3 (HCP)	
Fe	26	335	<b>A2 (BCC)</b>	<b>A2 (BCC)</b>	Er	68		A3 (HCP)	
Co	27	292	<b>A3 (HCP)</b>	<b>A1 (FCC)</b>	Tm	69	1	A3 (HCP)	
Ni	28	344	A1 (FCC)		Yb	70		<b>A1 (FCC)</b>	<b>A2 (BCC)</b>
Cu	29	178	A1 (FCC)		Lu	71	2	A3 (HCP)	
Zn	30	7	A3 (HCP)		Hf	72	5	<b>A3 (HCP)</b>	<b>A2 (BCC)</b>
Ga	31		A11 (oC8)		Ta	73	13	A2 (BCC)	
Ge	32	1	A4 (cubic) <sup>1</sup>		W	74	2	A2 (BCC)	
As	33		A7 (hR2)		Re	75		A3 (HCP)	
Se	34		A8 (hP3)		Os	76		A3 (HCP)	
Rb	37		A2 (BCC)		Ir	77		A1 (FCC)	
Sr	38		<b>A1 (FCC)</b>	<b>A2 (BCC)</b>	Pt	78		A1 (FCC)	
Y	39	6	<b>A3 (HCP)</b>	<b>A2 (BCC)</b>	Au	79	2	A1 (FCC)	
Zr	40	33	<b>A3 (HCP)</b>	<b>A2 (BCC)</b>	Hg	80		A10 (hR1)	
Nb	41	29	A2 (BCC)		Tl	81		<b>A3 (HCP)</b>	<b>A2 (BCC)</b>
Mo	42	49	A2 (BCC)		Pb	82		A1 (FCC)	
Ru	44	1	A3 (HCP)		Bi	83		A7 (mC4)	