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Predicting Microstructure-Property Relationships in Structural Materials via Multiscale Models Validated by In-Situ Synchrotron Observation

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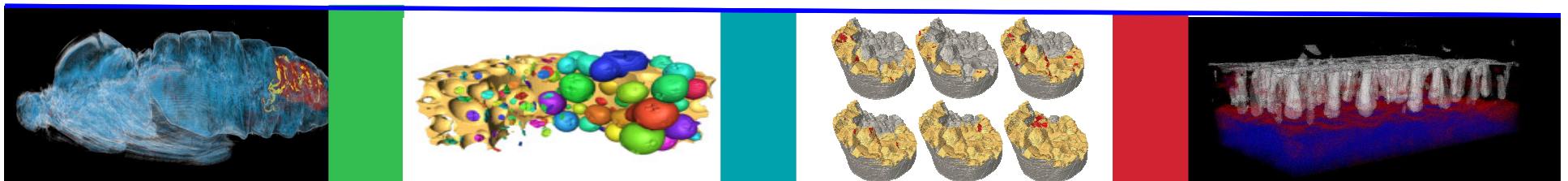
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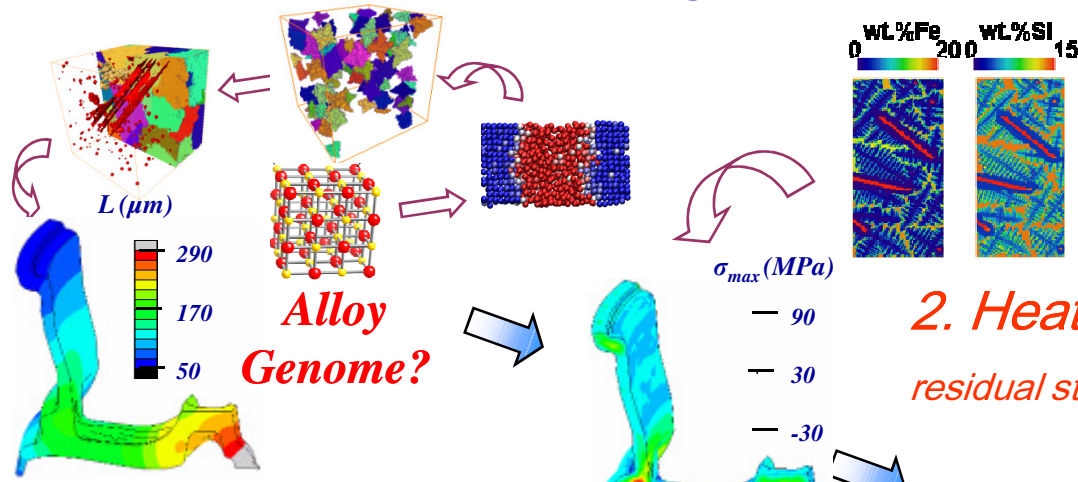
*Predicting Microstructure-Property
Relationships via multiscale models...
Nature versus Nature in the MGs...
validated by synchrotron observations*
Peter D. Lee, Lang Yuan, Chedtha Puncreobutr, S. Karagadde
Manchester X-ray Imaging Facility
www.mxif.manchester.ac.uk



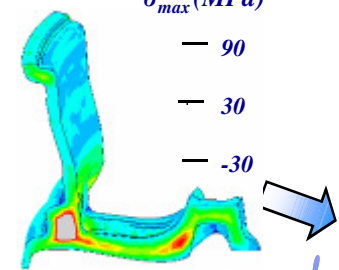
Why simulate Microstructural Evolution (ICME)?

To track genome evolution across length scales and processes

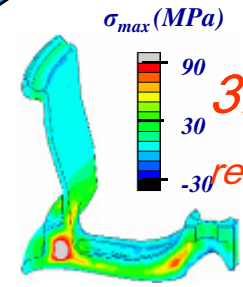
1. Casting:
Alloy/Microstructure dependent properties



2. Heat Treatment:
residual stresses

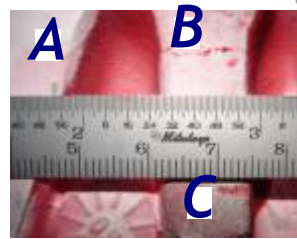
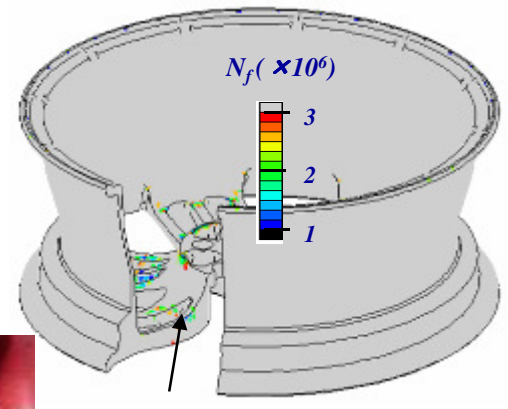


3. Machining:
residual stresses



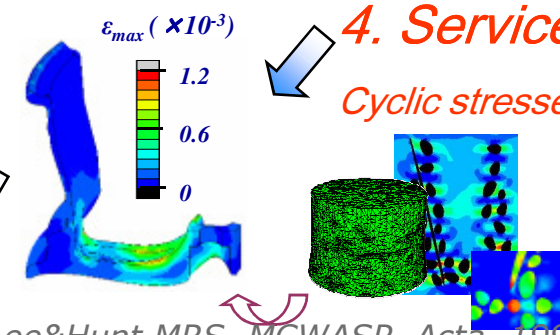
Multiscale, Through Process Genome Nurturing!

5. Component Performance:
Fatigue Life Prediction



Crack Initiation

4. Service:
Cyclic stresses/strains



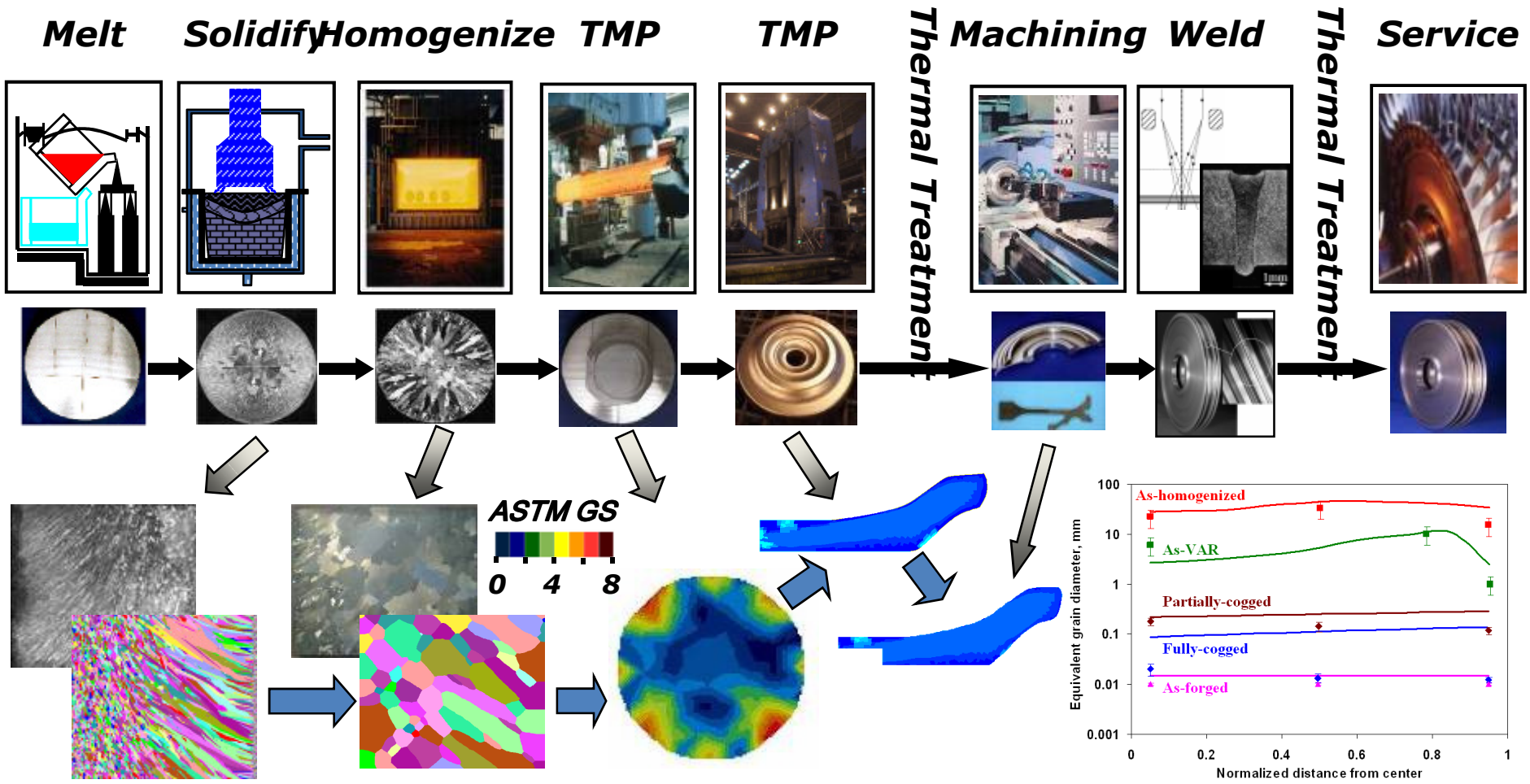
Lee&Hunt, MRS, MCWASP, Acta, 1994-7
Lee et al, Mat. Sci. Eng. A,, 2004
Maijer, Lee et al, Met. Trans, 2004
John Allison, ICME. JOM, 2006

I've modelled alloy processing for 25 years - What lessons have I learned?

- 1. When using multi-scale, through process modelling (or ICME), there are sufficient unknown parameters one can tune, you usually get the answer you want...**
- 2. For structural materials, it is not only the innate alloy properties, but also how you manufacture the component that matters**

I.e. Nurture can be more important than Nature if you want to get the most out of a Material's Genome.

Typical number of Nurturing steps...



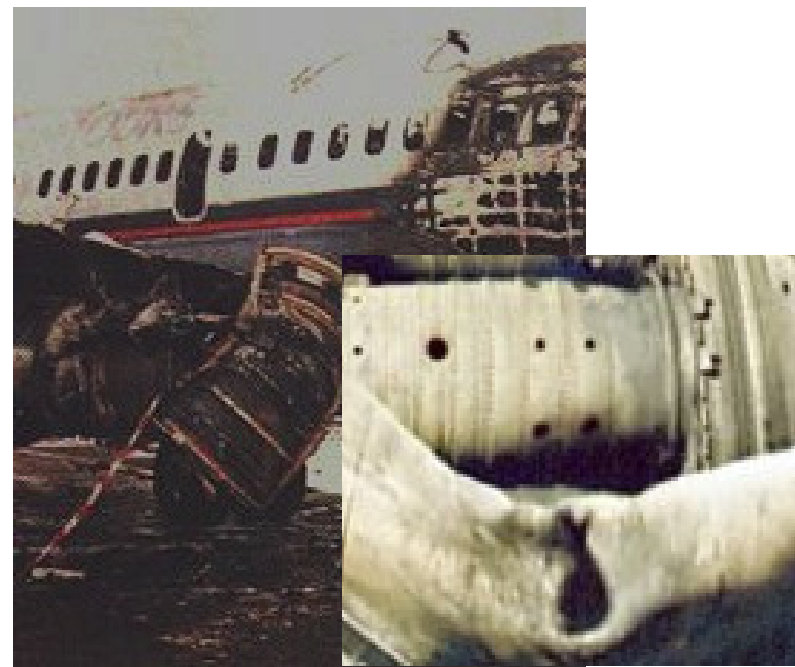
**With Rolls-Royce; Special Metals & Wyman-Gordon
Univ. of Cambridge (Tin) and Birmingham (Ward)**

Kermanpur, Tin, Lee, JOM 56(3) 2004, 72-78. or Tin, Lee, et al Met. Trans. A., 2005.

*Evolution of the
Material Genome
During Nurturing!*

Is optimising *Nature*, then providing good *Nurturing* enough?

- 1989 Kegworth air crash, caused by fan blade loss, manufacturing defect
- 1985, Manchester, failed combustor weld repair - porosity

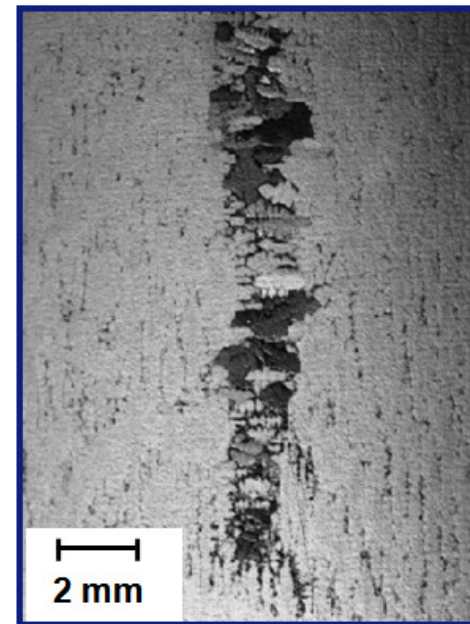
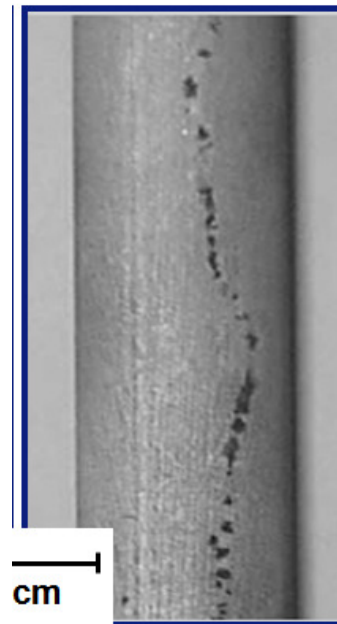


Lesson 3 - Lifing is often limited by a deviant microstructural feature, rather than the average, even though it may have the same genome...

My Conclusion...

The Materials Genome Project needs to map out not only the average behaviour, but also the distribution in behaviour, including the rebels

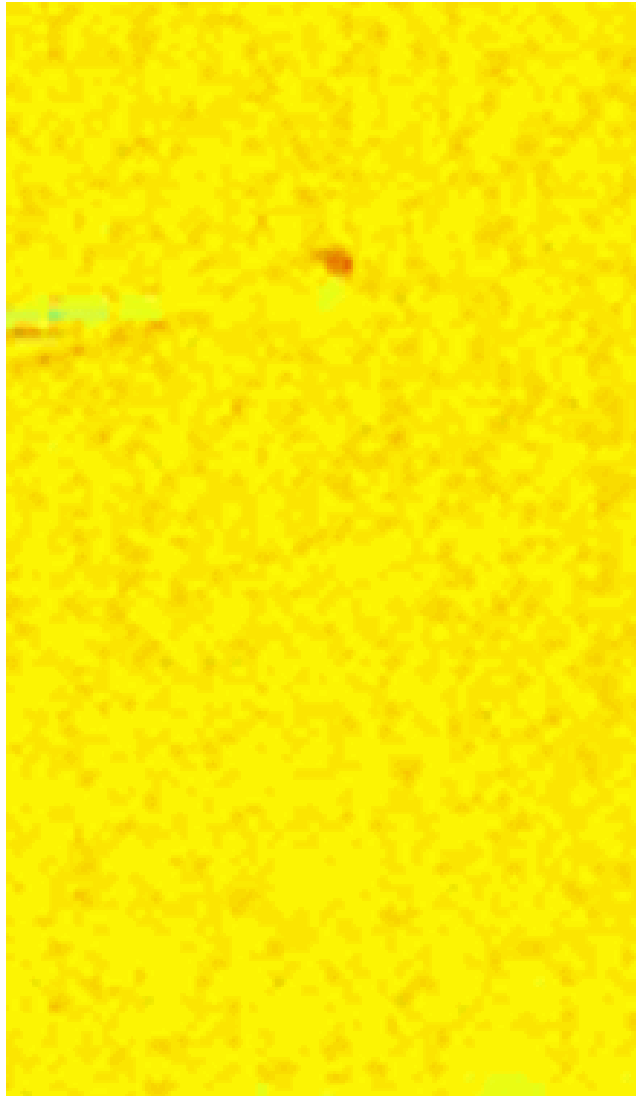
Example 1: Predicting deviant microstructures in Ni-based SX turbine blades blades: or *the Freckle Rebel*



Beckermann, Flemings Symposium, 2001

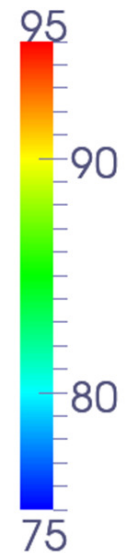
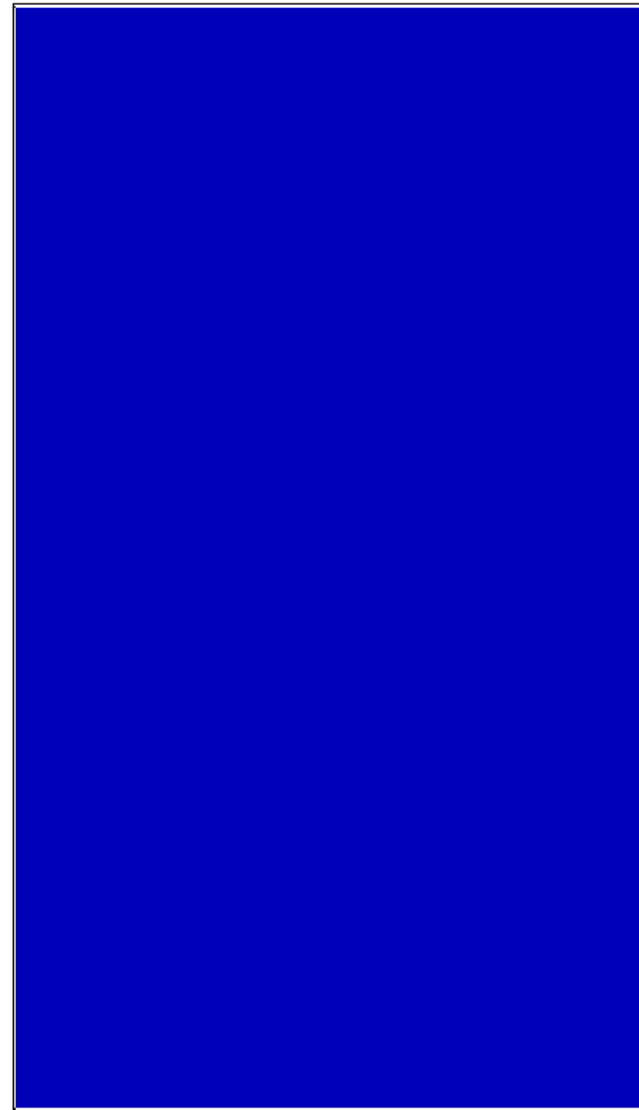
Solidification of Ga-25wt%In alloy, $G = 0.5\text{K/mm}$, $R = 8.1\ \mu\text{m/s}$

X-ray In-situ observation,
Courtesy HZDR, DE



*N Shevchenko et al 2012
Mater. Sci. Eng. 33 012035*

μ MatIC Simulation



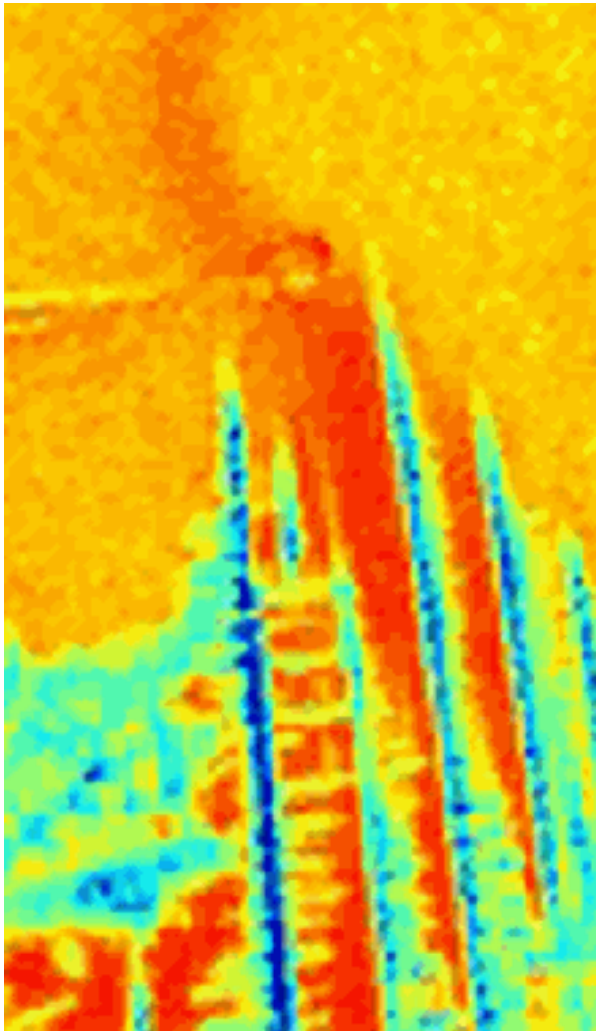
wt% Ga

www3.imperial.ac.uk/advancedalloys/

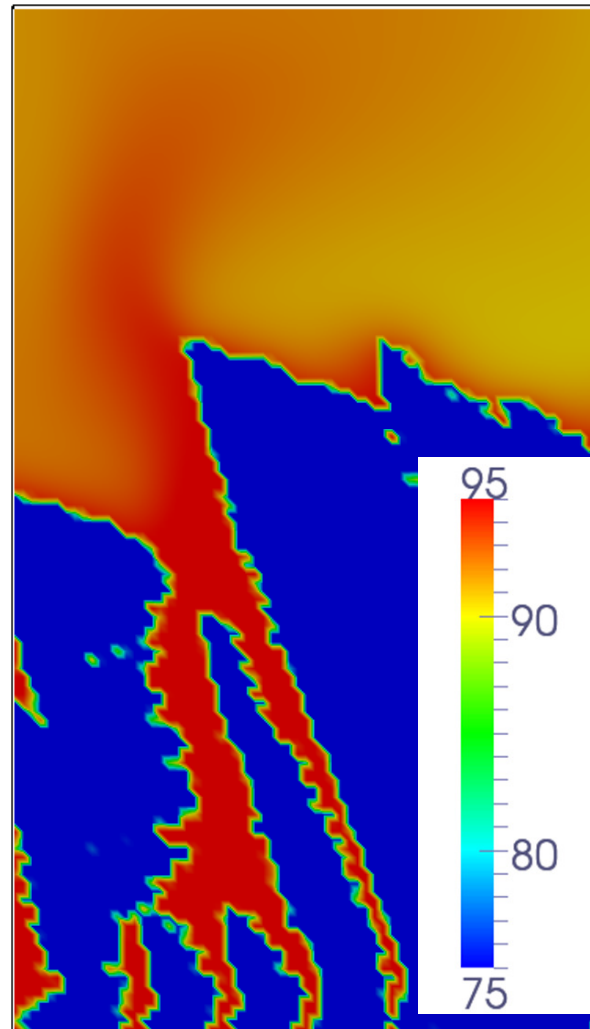
Solidification of Ga-25wt%In alloy, G .5K/mm, R 8.1 $\mu\text{m/s}$

X-ray In-situ observation,
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μMatIC Simulation



*N Shevchenko et al 2012
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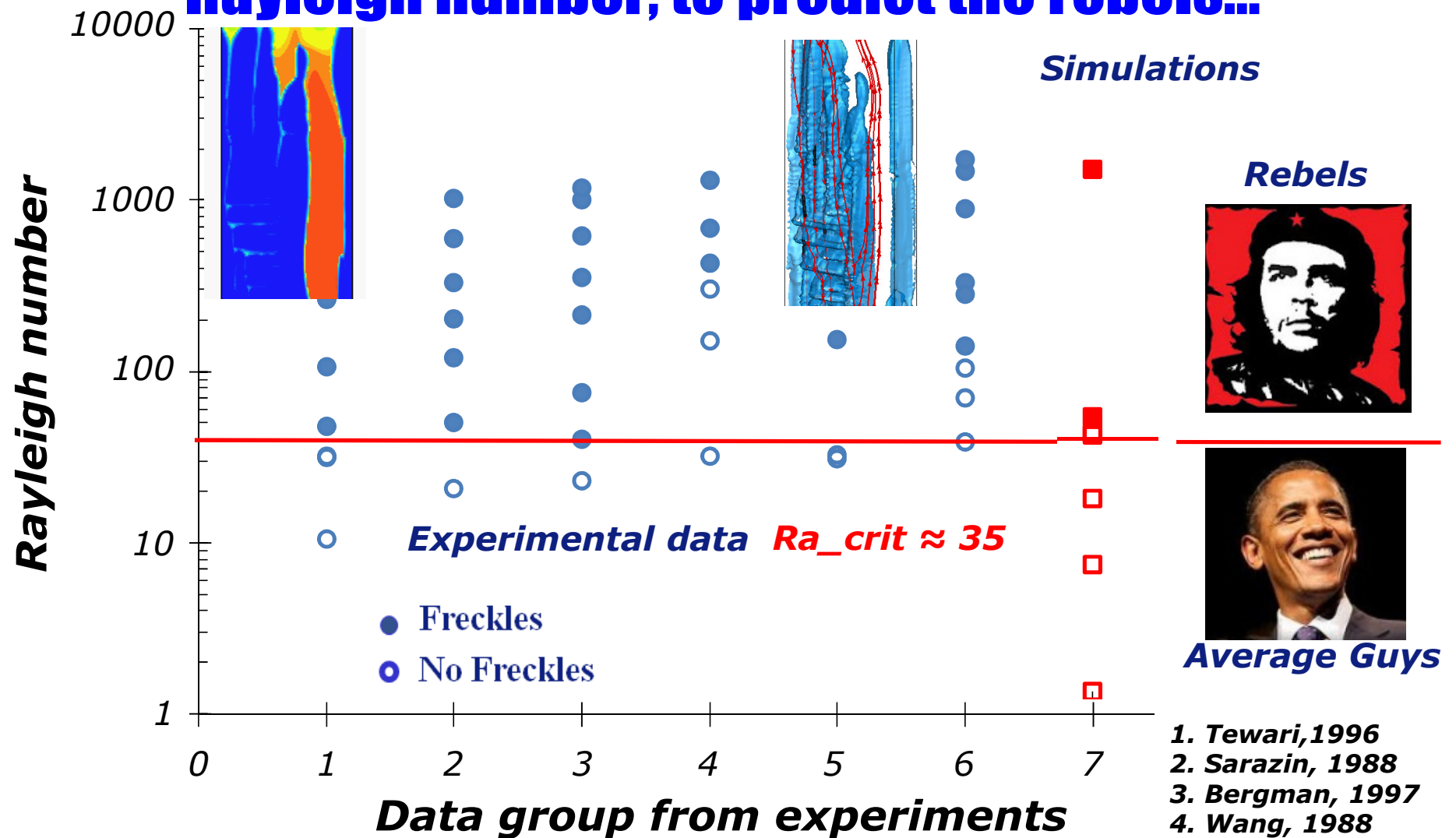


wt% Ga

www3.imperial.ac.uk/advancedalloys/

*Upwards
liquid flow
increases
solute
concentration
in the channel
with local
remelting,
remelting
secondary
arms and
stopping
primary arms*

To scale the microstructural model to a macroscopic level, we can approximate it via the Rayleigh number, to predict the rebels...



Yuan L and Lee PD, Acta Mater., 2012

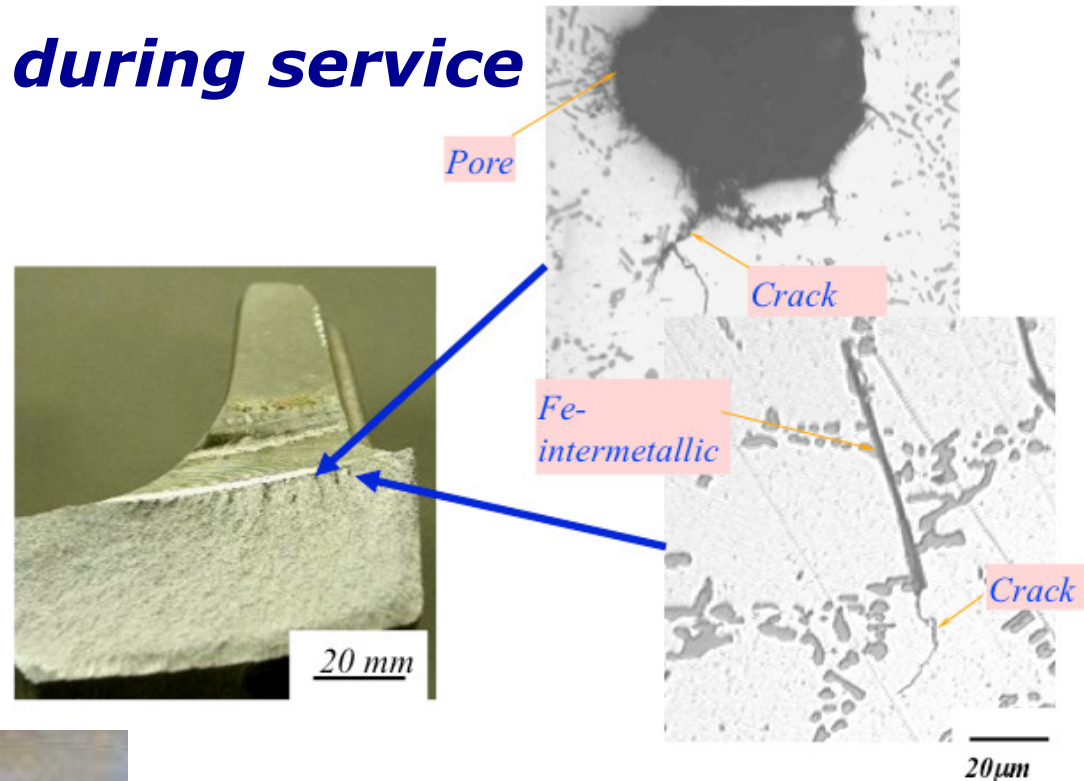
1. Tewari, 1996
2. Sarazin, 1988
3. Bergman, 1997
4. Wang, 1988
5. Streat, 1974
6. Sarazin, 1990

Example 2

**Understanding why eating your
spinach is not always good for
your strength, or
*predicting Fe-intermetallic Rebels...***

Why worry about deviant microstructures like pores and Fe-intermetallics?

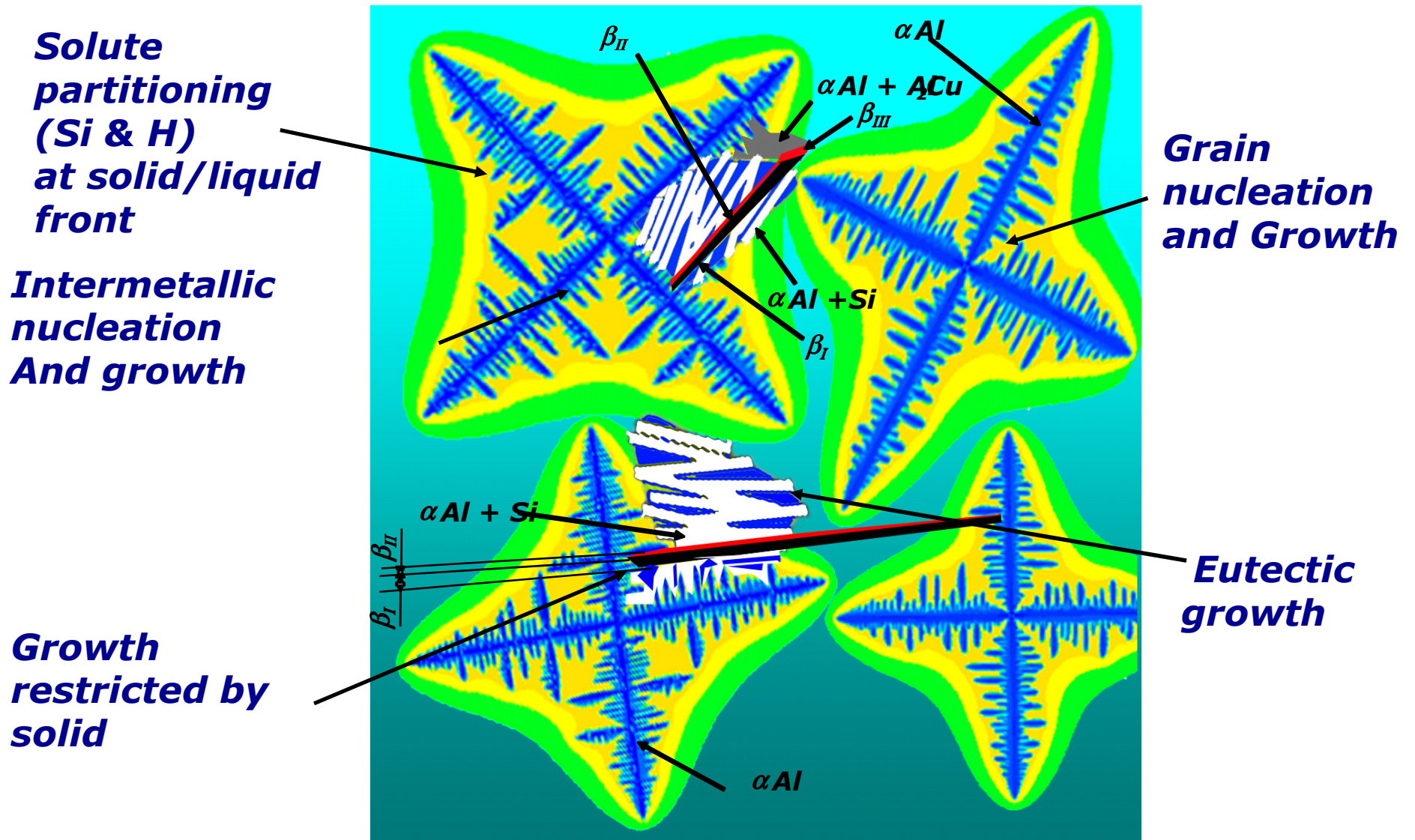
They initiate failure during service



And they alter formability during manufacturing

Yi, J.Z. et al. Met. Trans. 34A, 2003.

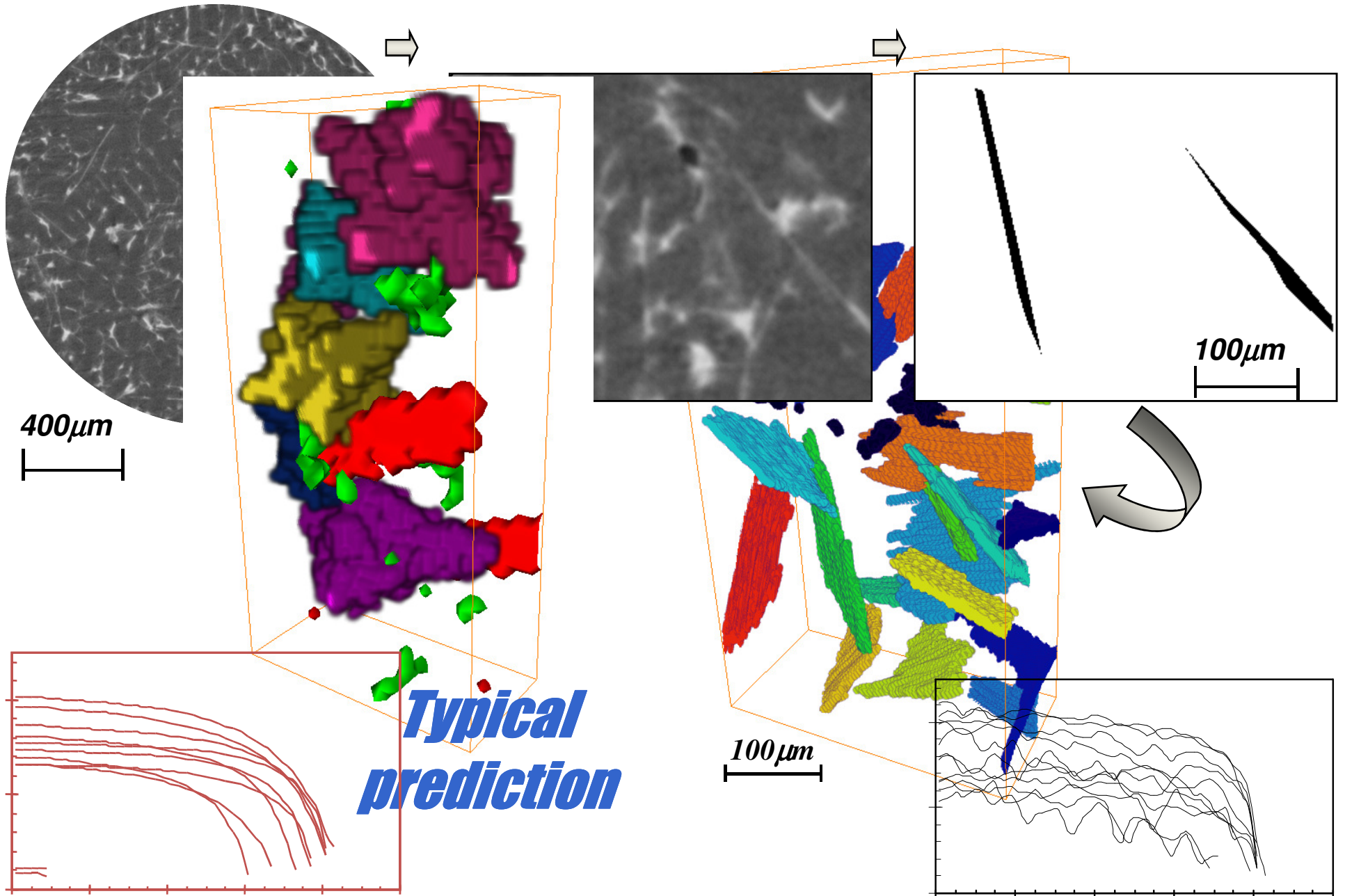
What needs to be simulated?



For speed we model with $10\mu\text{m}$ elements, and approximated anisotropy

- ~~Nucleation: nanometres~~ → *Empirical fn*
- ~~Dendrite tip radius: 1 micron~~ → *$F_s > 0.5$, approximate...*
- ~~Coarsening: 10 microns~~ → *Borderline...*
- ~~Solute diffusion: 10 microns~~ → *$F_s > 0.5$, ~Scheil btwn dendrites*
- H diffusion: 100 microns ✓
- Pores: 10-100' s microns ✓
- Intermetallics: 10-100' s microns ✓
- Grain size: 100-1000 microns ✓

Synchrotron CT Characterization of Fe Intermetallic Morphology compared to model prediction

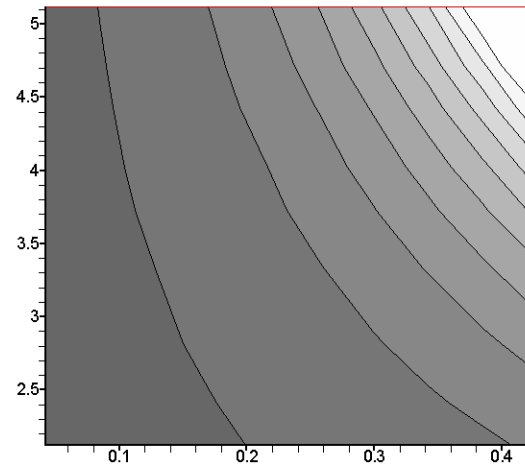
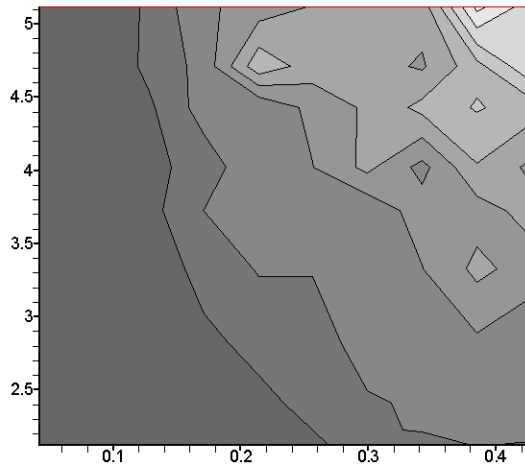


How do we capture the genome and span scales? Via model-based constitutive equations

**1000's micromodel
predictions**

Regression Fit

**Solidification Time
($\ln t_s$)**



**Maximum
Intermetallic Length
Microns**

1000



0

Initial Fe Content C_{Fe}^{init}

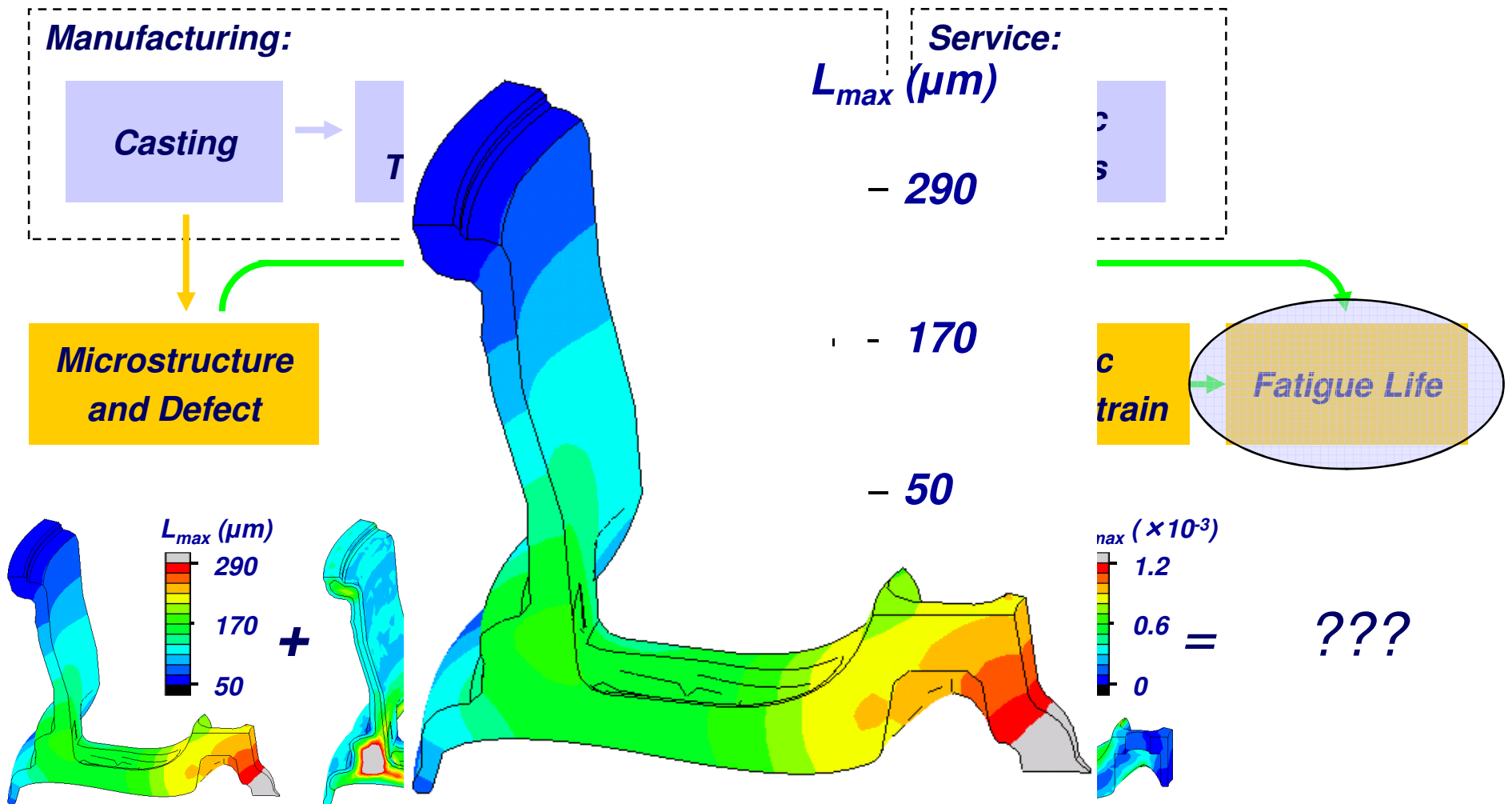
$$\ln L_{\max} = b_0 + b_1 \ln t_s C_{Fe}^{init}$$

and statistical variation

Challenges:

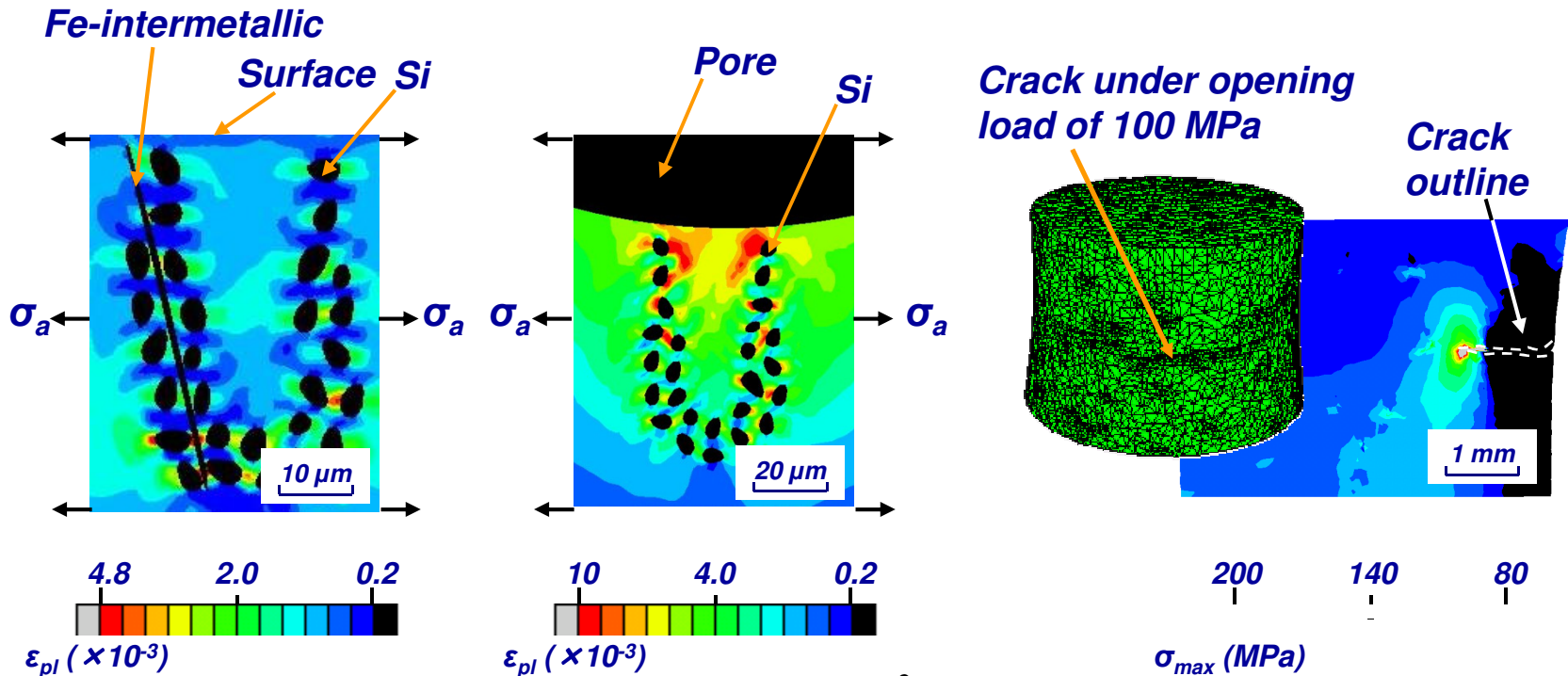
- 1. Improved statistical tracking of multi-variant distributions**
- 2. Fitting highly coupled phenomena (e.g. Pressure)**

Coupling deviant microstructure to Lifting



Li et al, MCWASP 2006, MMTA 2007

Crack Initiators



$$N_f = N_i + N_p = \frac{C_0}{\lambda_2} \left[\frac{1}{k_\sigma \sigma_a} \left(k_0 + \frac{\alpha}{\sqrt{\lambda_2}} \right) \right]^{\frac{2}{\beta}} + C_1 \left(\epsilon_{max} \frac{\sigma_a}{\sigma_y} \right)^{-s \times t} \left(a_f^{-t+1} - a_i^{-t+1} \right)$$

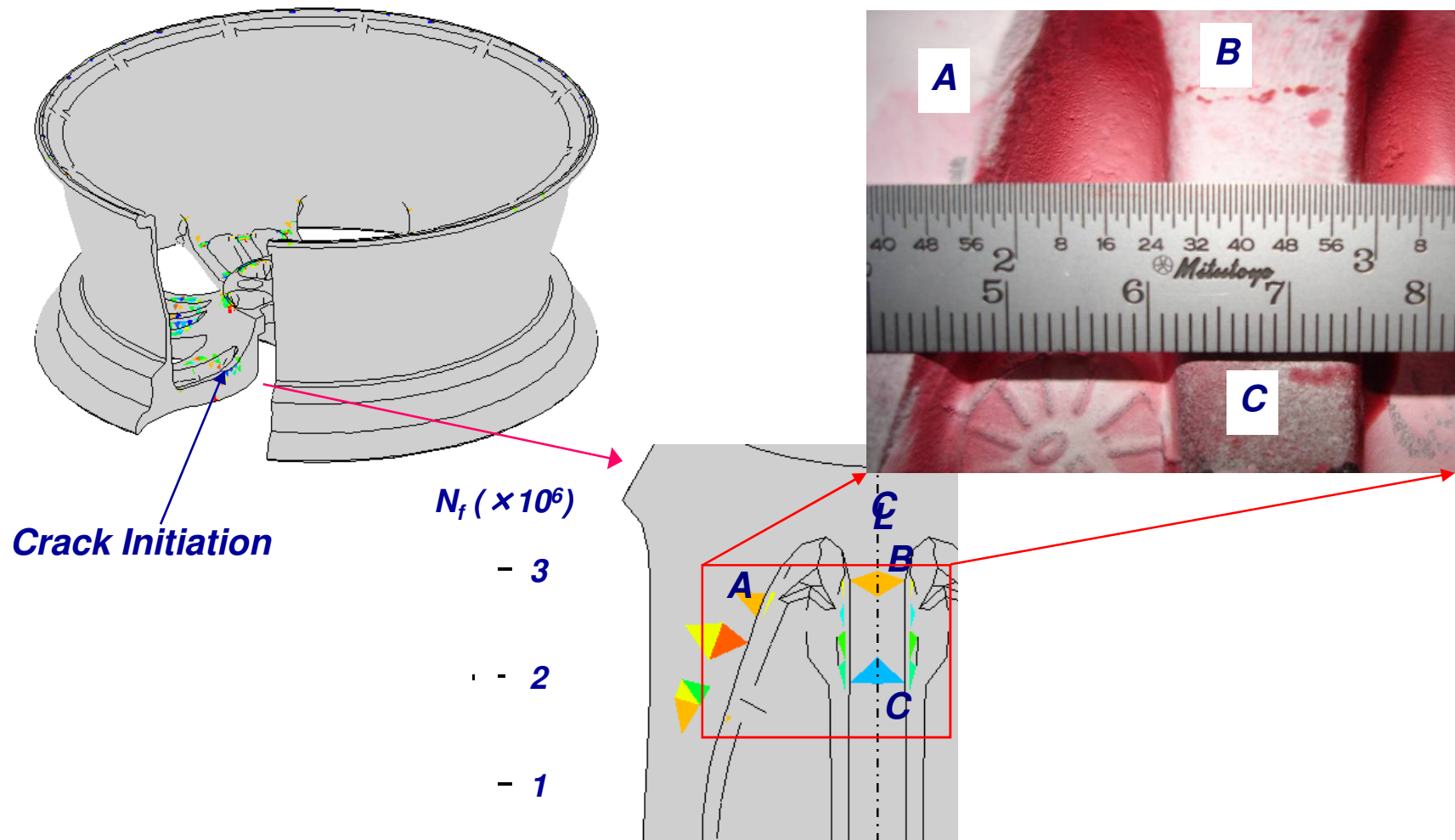
Maximum size of deviant microstructures: Pores or Fe-rich intermetallics - L_{max}

Gao YX et al., Acta Mat, 2005
Li P et al., AEM, 2006

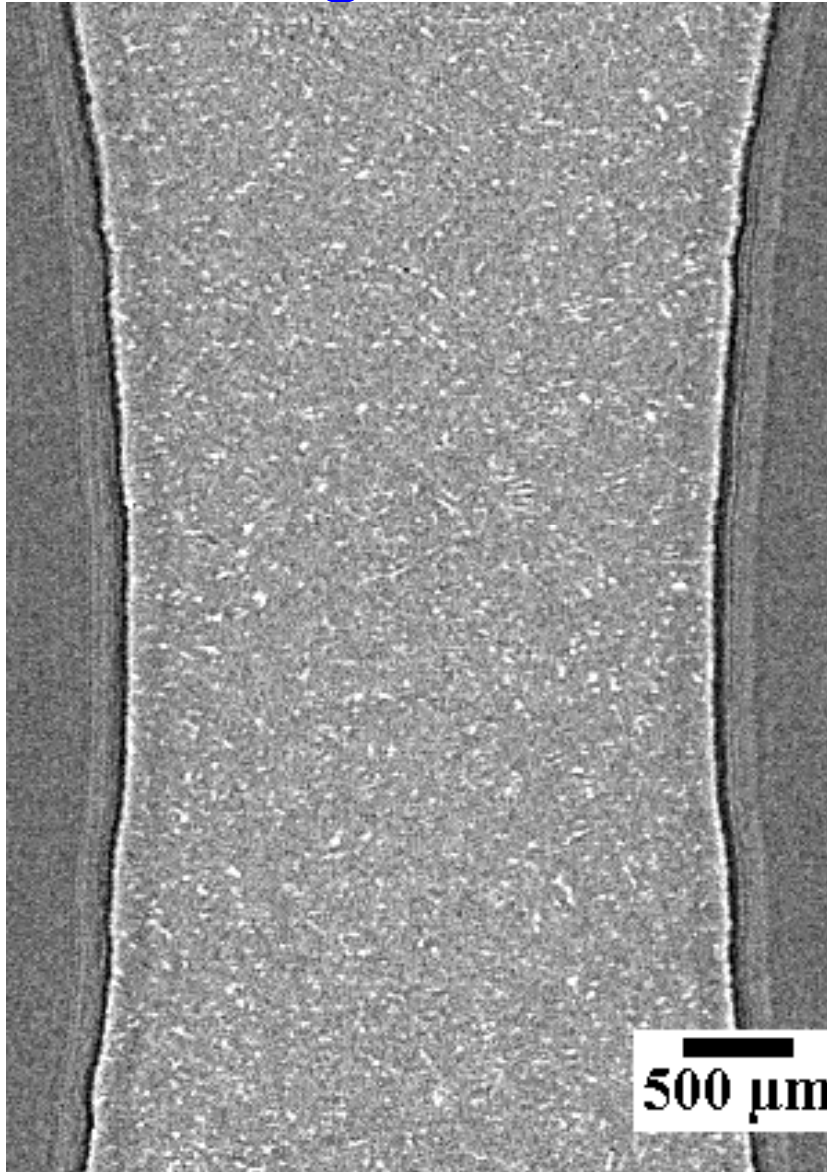
Challenges: models for flow stress of each phase, interface strength and adding debonding model, etc...

Does it work?

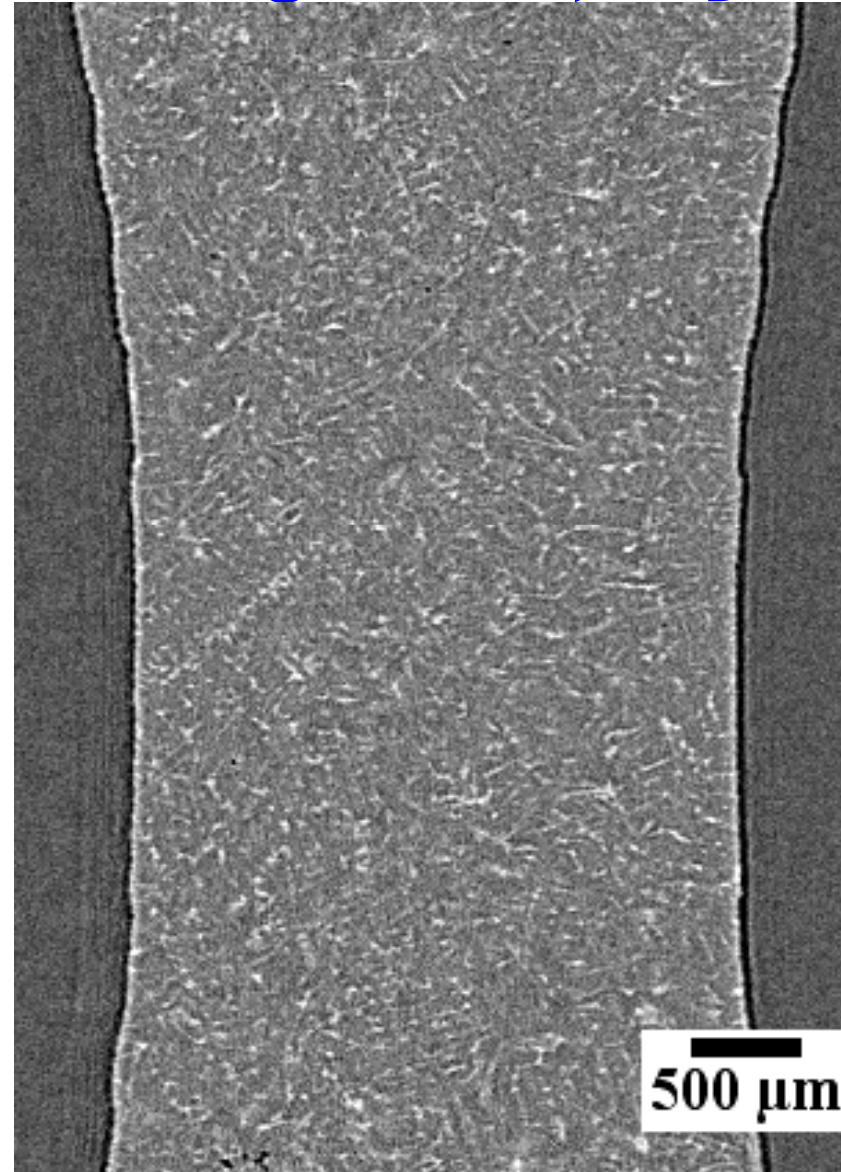
Accurate prediction of failure location was achieved (deviant feature - pores).



In situ observation shows Fe reduces hot forming/increases hot-tearing of A319, why?



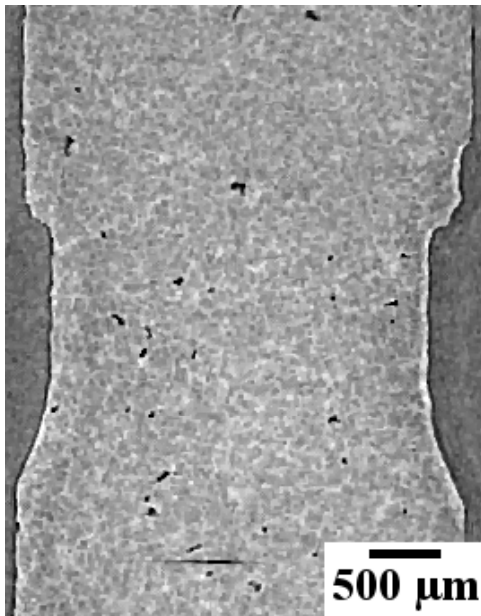
0.2wt.%Fe



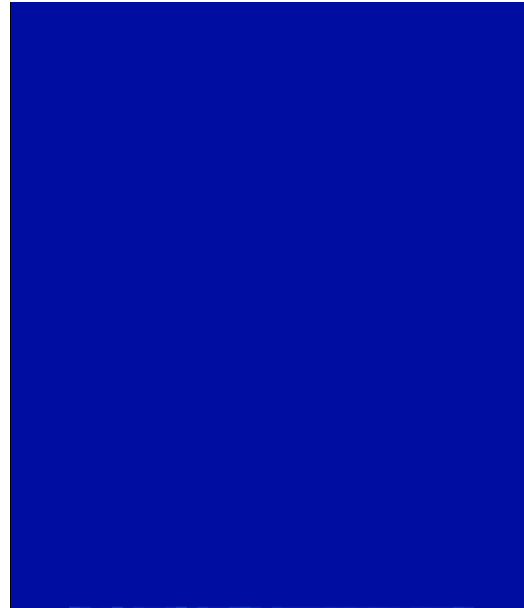
0.6wt.%Fe

Comparison of analysis techniques

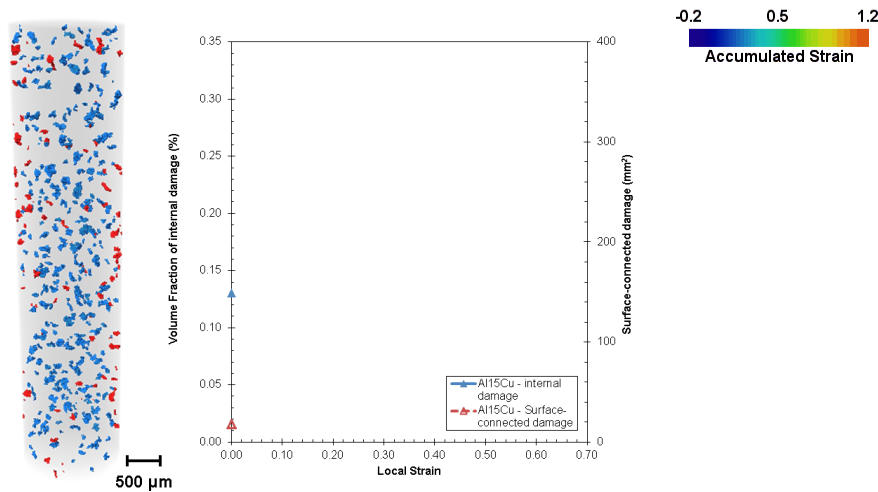
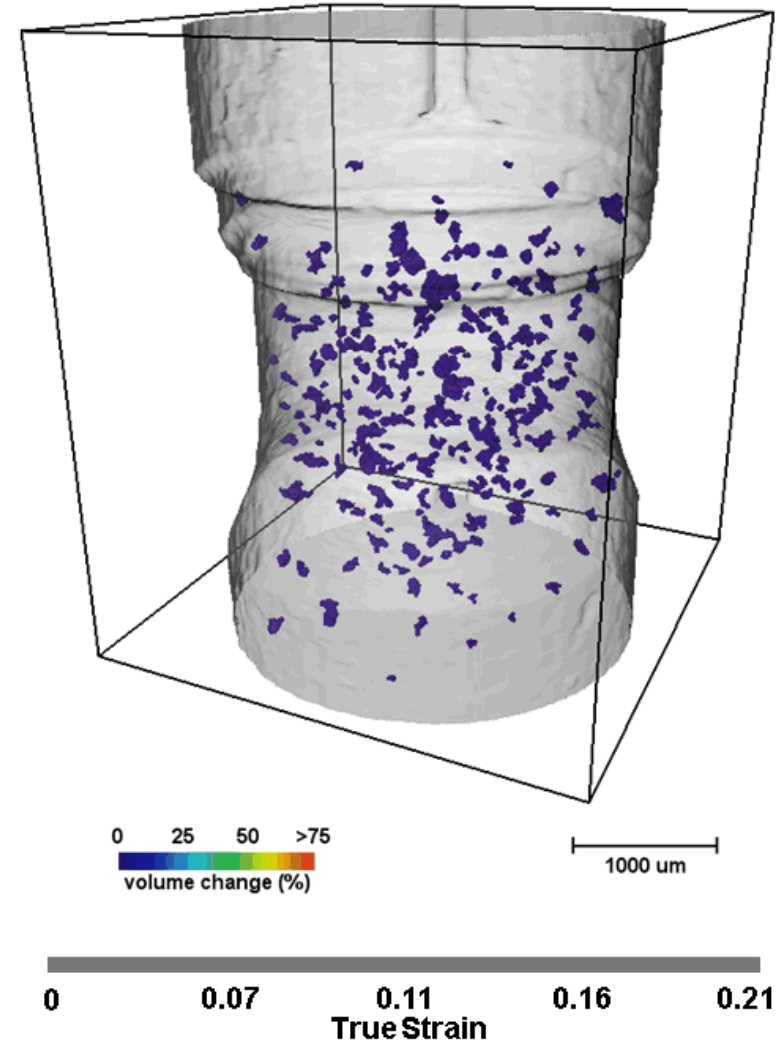
Imaging



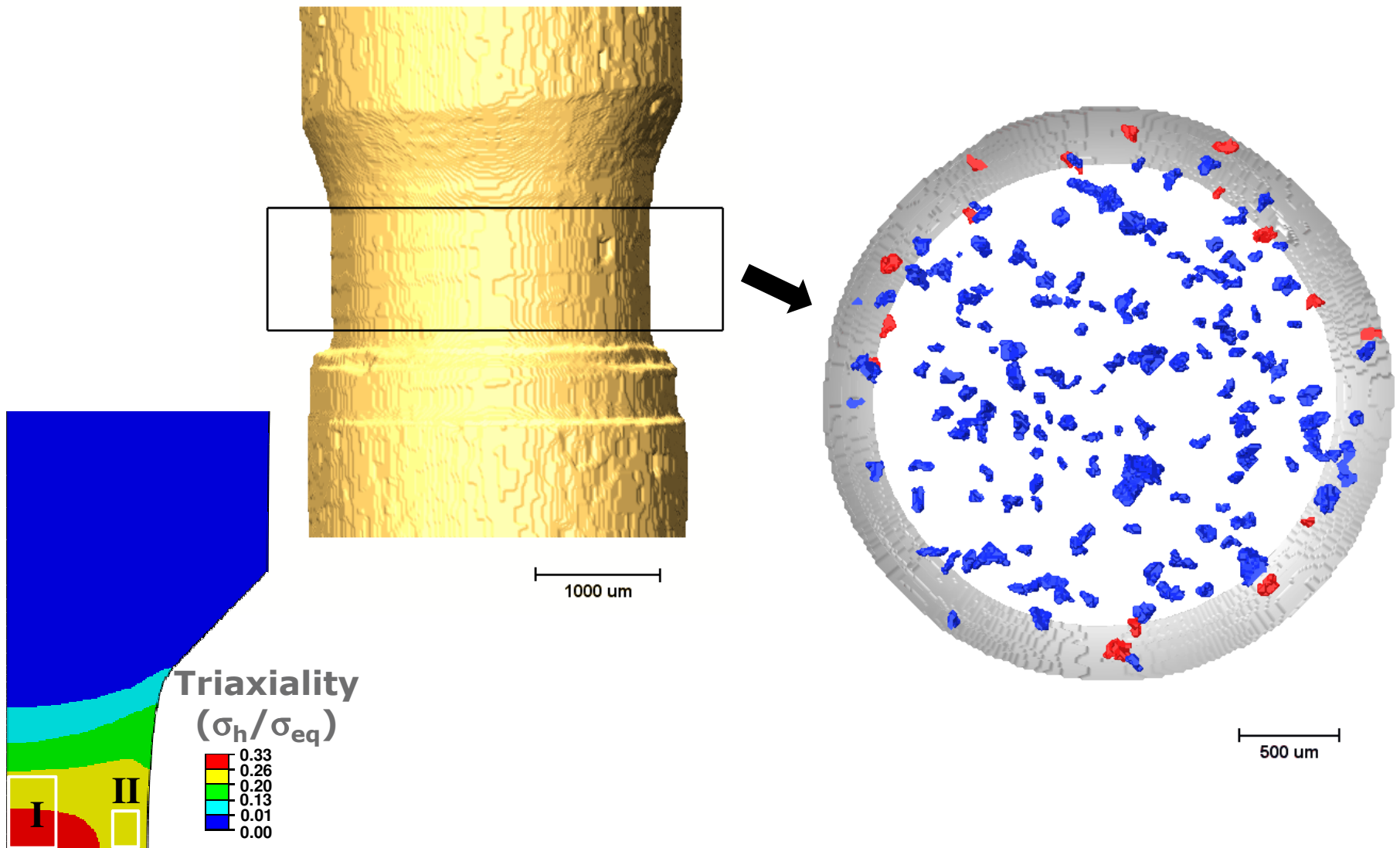
DVC



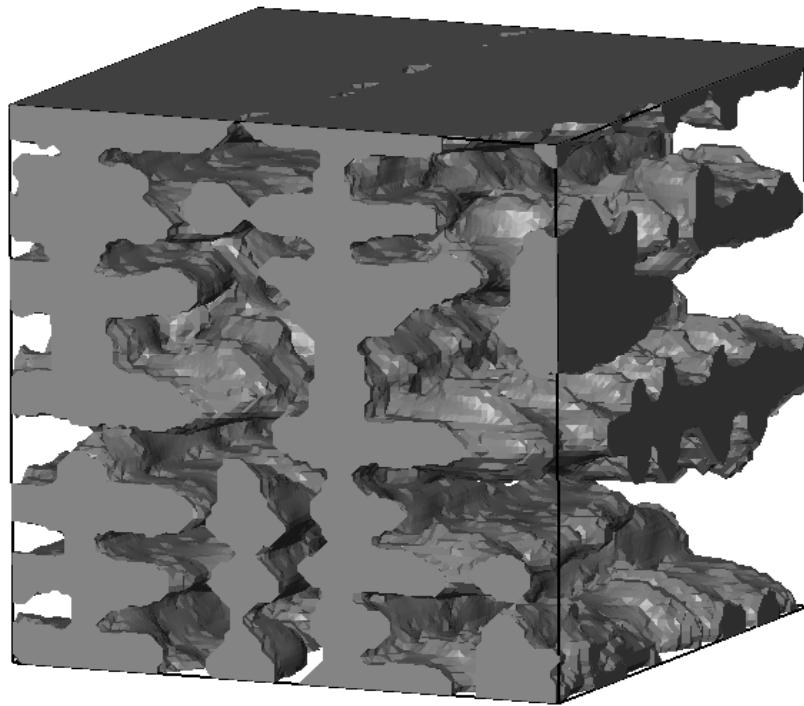
Tracked Quantification



We can directly compare to predicted influence of triaxiality on localisation of damage

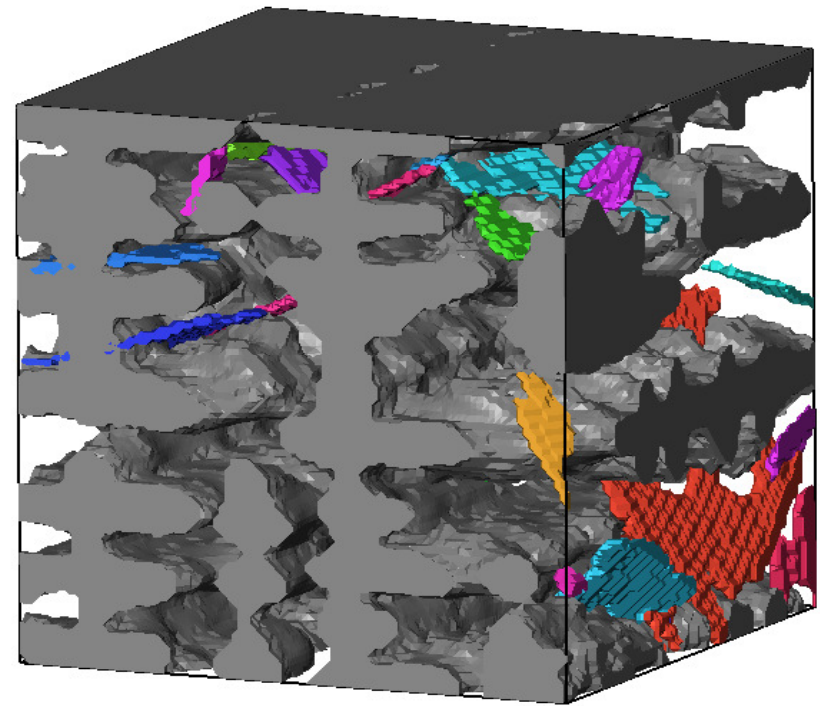


1st hypothesis, the intermetallics reduce interdendritic flow – we can use image based modelling to directly simulate the flow



275 μm

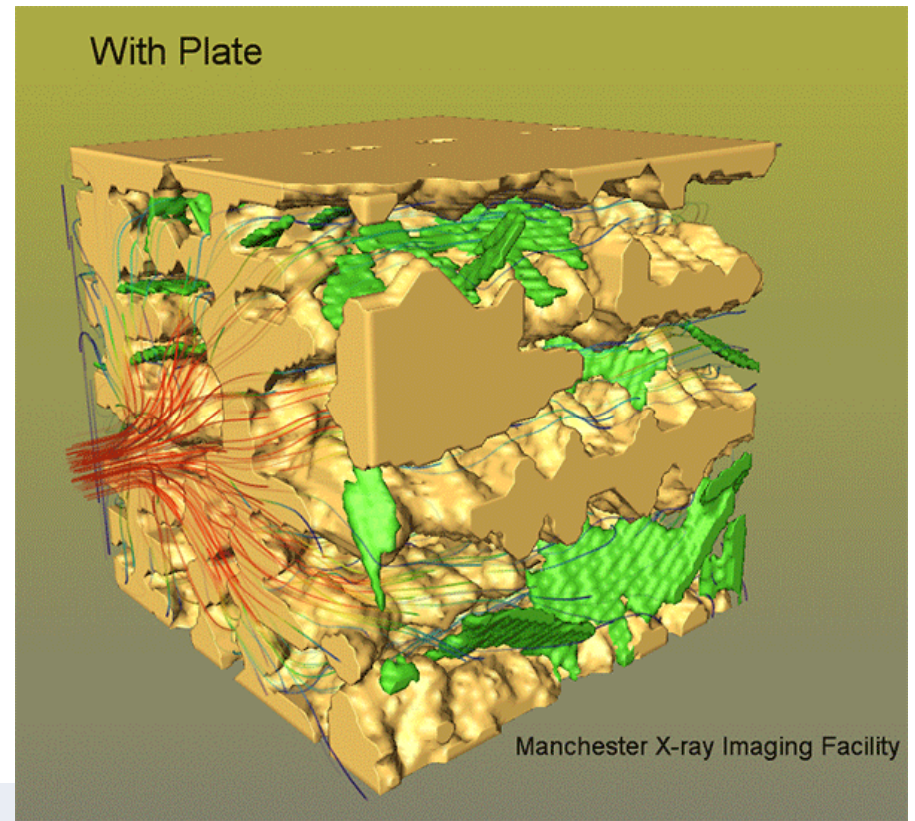
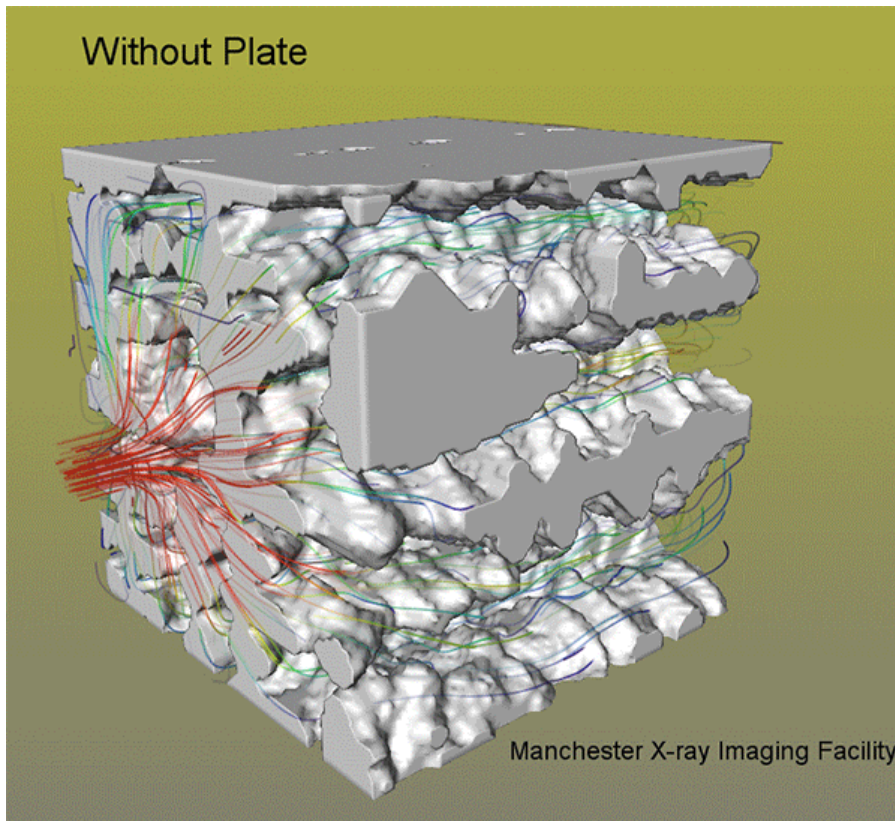
Dendrite



275 μm

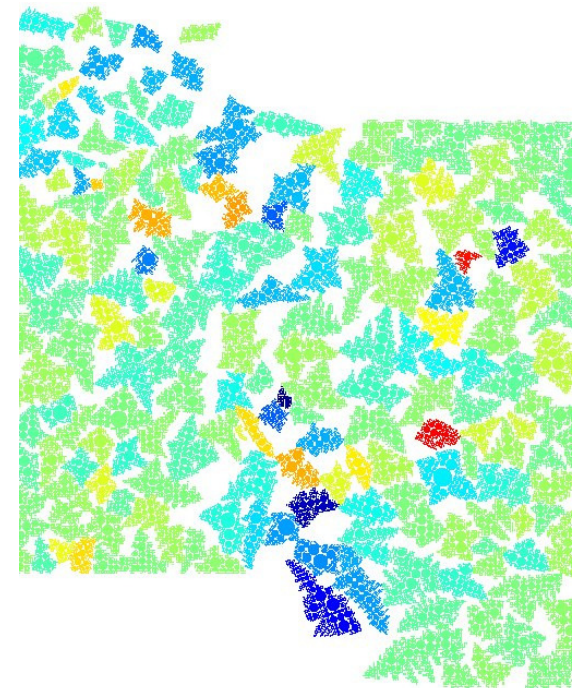
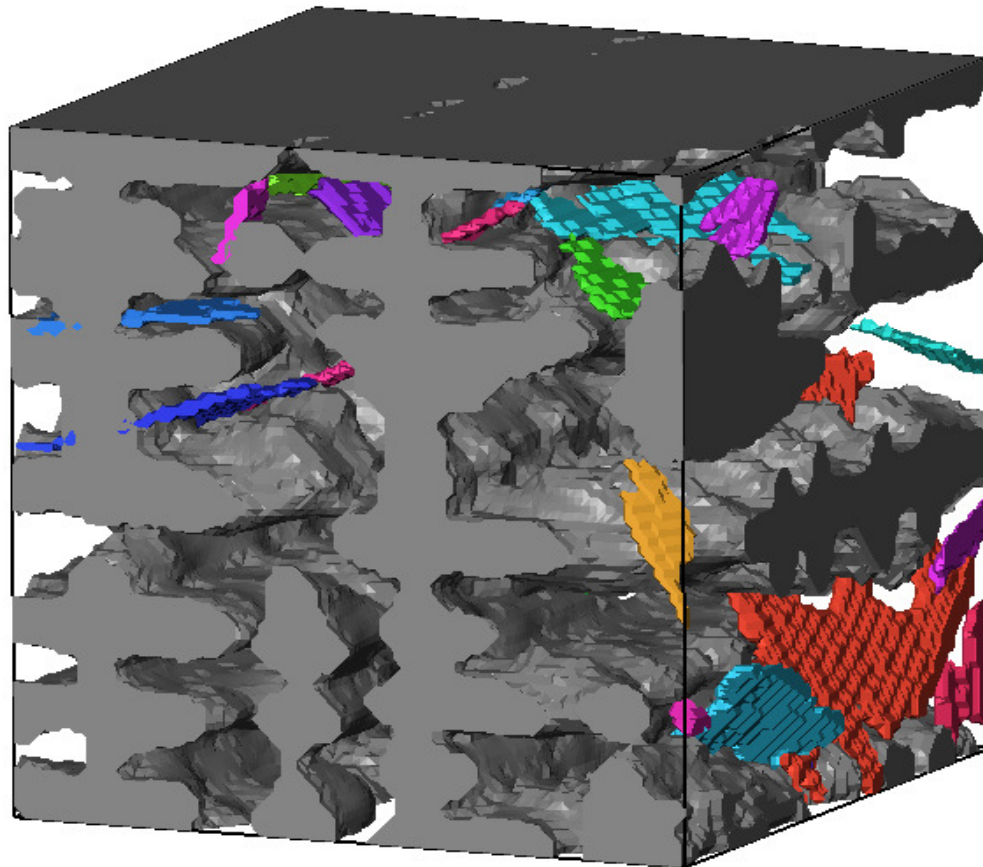
Dendrite + Intermetallics

Flow Simulation results – <10% reduction in flow



Geometry file	k [μm^2]	k [d]	Input pressure [Pa]	Output pressure [Pa]	Flow rate [$\mu\text{m}^3.\text{s}^{-1}$]	Viscosity [Pa.s]
Without Plate	178.5231	180.88853	130000	100000	4.87E+11	0.001
With Plate	167.14487	169.35954	130000	100000	4.86E+11	0.001

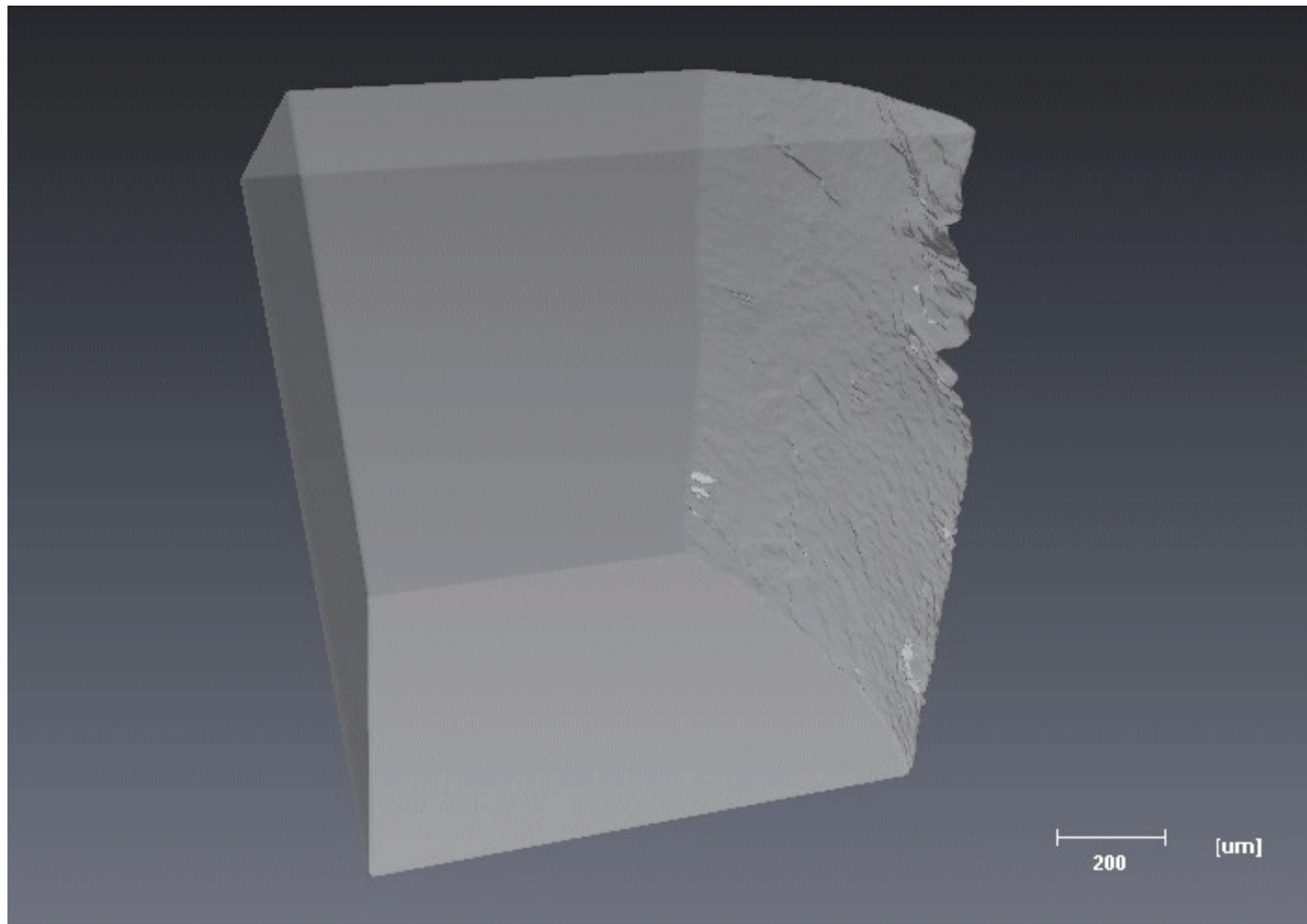
**Synchrotron imaging with direct simulations
demonstrated flow is only a minor effect, so now
we need another hypothesis!**



275 μm

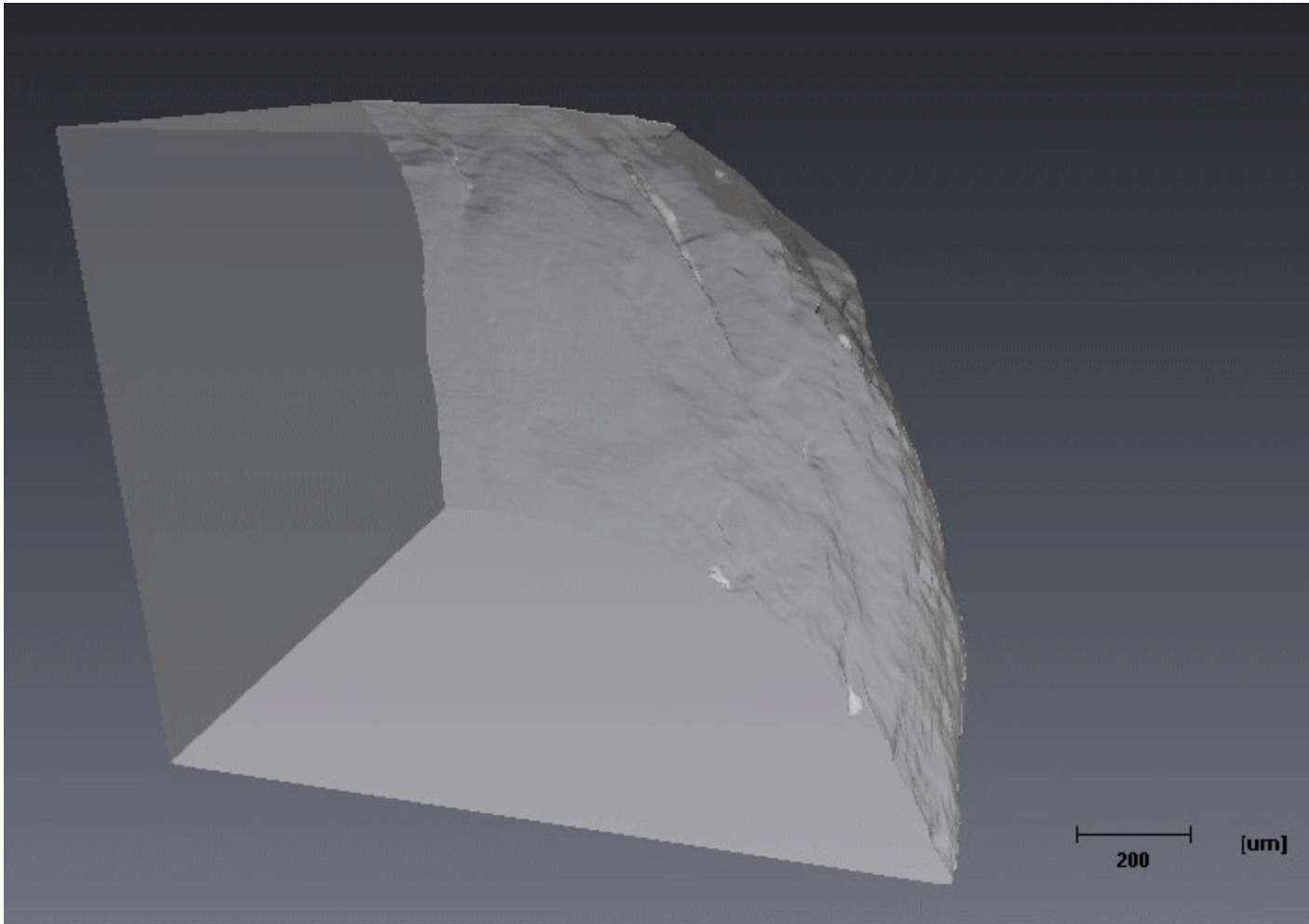
However, the synchrotron observations helped answer other questions:

- 1. when/where do the intermetallics nucleate, and**
- 2. do pores nucleate on intermetallics**



***Cooled at
3° C/min***

Do pores nucleate on intermetallics?



Cooled at 3° C/min

Conclusions

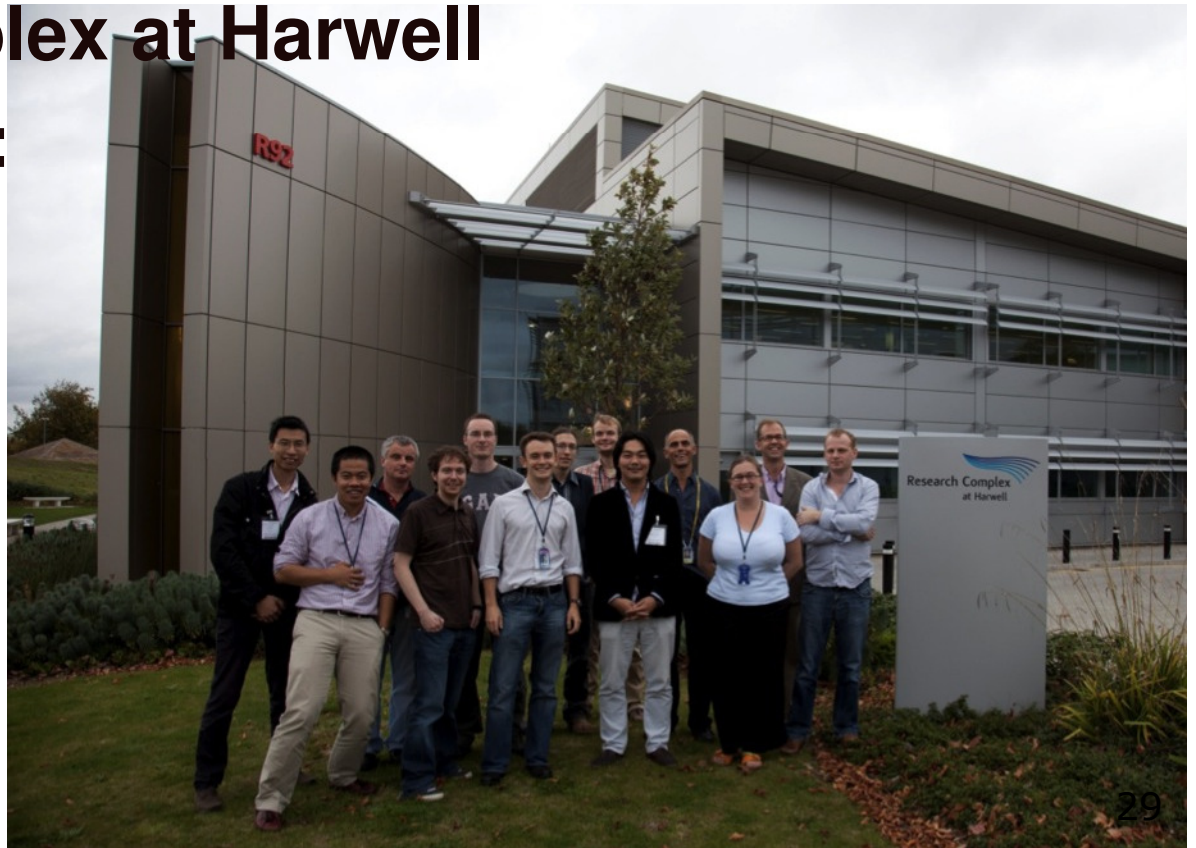
In situ observation shows us the kinetics of microstructural formation, clarifying dominate mechanisms and causality

Nurture can matter as much as Nature

We need to look out for the rebels when during the Materials Genome Initiative

Acknowledgements

- Ford, Tata Steel, GE
- Diamond Light Source & I12 Team
- EPSRC (grant EP/I02249X/1)
- Research Complex at Harwell
- The MXIF Team:
 - UoM
 - RCaH
 - Diamond
 - Imperial College



Questions?