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S. Kment University of Nebraska-Lincoln

H. Kmentova University of Nebraska-Lincoln

A. Sarkar University of Nebraska-Lincoln

Rodney J. Soukup University of Nebraska-Lincoln, rsoukup1@unl.edu

Natale J. lanno University of Nebraska-Lincoln, nianno1@unl.edu

See next page for additional authors

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Authors

S. Kment, H. Kmentova, A. Sarkar, Rodney J. Soukup, Natale J. Ianno, J. Krysa, Z. Hubicka, J. Olejnicek, C.

L. Exstrom, and S. A. Darveau

A NOVEL SOL-GEL ROUTE TO PINHOLE-FREE IRON SULFIDE THIN FILMS

S. Kment^{1, 2}, H. Kmentova^{1, 3}, A. Sarkar¹, R. J. Soukup¹, N. J. lanno¹, J. Krysa⁴, Z. Hubicka⁵, J. Olejnicek^{5,6},

C. L. Exstrom⁶ and S. A. Darveau⁶

¹Department of Electrical Engineering, University of Nebraska-Lincoln, Lincoln, NE, USA

²RCPTM, Faculty of Science, Palacky University, 17. listopadu 12, 77146 Olomouc, Czech Republic

³Institute of Chemical Process Fundamentals, Academy of Sciences of the Czech Republic, Prague, Czech Republic ⁴Institute of Chemical Technology - ICT Prague, Prague, Czech Republic

⁵Institute of Physics, Academy of Sciences of the Czech Republic, Prague, Czech Republic

⁶Department of Chemistry, University of Nebraska at Kearney, Kearney, NE, USA

ABSTRACT

The general purpose of the study is to fabricate and improve upon FeS₂ thin films which can be used as the photon absorber laver for a heteroiunction or homoiunction solar cell. This work deals with the preparation of the pyrite by an unconventional sol-gel approach. Thin pyrite films were prepared by sulfurizing the iron oxide films previously deposited through the sol-gel method using iron (III) chloride as a precursor. The structural, morphological, electronic and optical properties of the deposited films were determined using X-ray diffraction (XRD), Raman spectroscopy, scanning electron microscopy, Auger spectroscopy electron (AES), UV-Vis absorption spectroscopy, Hall effect and profilometry. The effects of annealing and sulfurization temperatures were studied. The work was also devoted to the research of sodium diffusion from the substrate due to the thermal treatment and its affect on the pyrite films functionality.

INTRODUCTION

In recent years, iron disulfide has attracted considerable attention as a candidate for alternative solar cell materials. The very high absorption coefficient, $\alpha = 6 \times 10^5 \text{ cm}^{-1}$ for λ = 633 nm [1] and the composition of abundant, cheap and non-toxic elements makes pyrite an interesting material for thin-film solar cells [2]. Iron disulfide has a bandgap energy of E_q = 0.95 eV, [3] somewhat less than the ideal. However, a thorough modeling of the potential of pyrite indicates that efficiencies of 20% are possible [4]. It is interesting to note that the model used here included a bandgap of only 0.8 eV, so potential efficiencies are even greater. Despite these favorable physical properties, the conversion efficiencies of pyrite thin film solar cells have been disappointing and an incentive for further study. In this laboratory pyrite thin films have previously been prepared by several methods such as sputtering, sulfurization of sputtered iron or iron oxide layers [5], and chemical bath deposition [6].

A simple method should be developed for the preparation of pyrite thin films by using non-vacuum, low-cost equipment. Thin films described in this paper have been obtained by sulfurization of layers prepared using a sol-gel method. It has several advantages: simplicity, treatment of large areas, easy control of film thickness, use of metallic and nonmetallic substrates and low cost.

The main advantages of thin film pyrite solar cells would be a low material consumption and a fast and low cost production process. It is still unknown which structure would be the most favorable, a pn homojunction, a p-i-nstructure or a pn heterojunction. A promising technique, the sol-gel method, for non-vacuum growth of pyrite thin films is reported here. The results stimulate further experiments and offer the opportunity for the preparation of solar cells from pyrite.

EXPERIMENTAL PROCEDURES

Preparation of thin films

First the sol-gel procedure was employed in the preparation of the Fe₂O₃ thin films which were further sulfurized in order to obtain pyrite thin films. Iron oxide thin films were synthesized according to the following procedure. Iron(III) nitrate (2.5 g) was dissolved in ethyl alcohol (10 ml) and used as a molecular precursor of Fe₂O₃. Then, 2 ml of propylene oxide was dropped slowly into the solution and after a few minutes the solution turned dark. The hydrolysis of inorganic Fe(III) solutions consists of (a) formation of low-molecular-weight species; (b) further condensation to form a red cationic polymer; (c) aging of the polymer causing conversion to oxide phases; (d) precipitation of oxide phases directly from lowmolecular-weight precursors. The prepared solution was used to spin coat previously cleaned substrates at 2200 rpm. By a single spin, thin (~ 50 nm) layers were obtained; whereas thicker layers were obtained by multiple coatings. After each spin coating, thermal treatments were performed raising the temperature at approximately 100 °C/h and keeping the temperature at different levels for 1 hour. Soda lime glass slides were used as the substrates.

Next, the calcinated layers prepared by the sol-gel method were sulfurized in an evacuated quartz tube using a Rapid Thermal Annealing (RTA) irradiation lamp as a heating source. The RTA system allows precise programming of a thermal annealing ramp. The temperature was increased at the rate of 150 °C/ min up to 450 °C. At this temperature

level, the samples dwelled for 15 minutes and then the RTA was turned off allowing the temperature to decrease slowly to ambient temperature. During the sulfurization, the samples were kept in a graphite boat together with sulfur powder. The sulfur amount of 1 g was the same for all the experiments. The boat was placed in the quartz tube, in which it was connected via a thermocouple to a station operating the RTA lamp. The tube was first evacuated and then filled with Argon. The influence of two parameters, temperature and time of sulfurization, was studied. The sulfurization temperature was varied between 400 °C and 500 °C (heating rate 150 °C/ min, 1 hour dwelling time in these experiments) and the sulfurization time was constantly decreased from 5 hours to 15 min at the temperature of 450 °C, the temperature which yielded the best results.

Thin Film Characterization

Raman spectra were recorded by means of a back scattering Raman microscope (Renishaw) with the excitation laser wavelength of 514 nm. X-ray diffraction patterns were taken on an XRD diffractometer - Rigaku D/ Max-B diffractometer (Cu K α radiation ~ 1.544 Angstroms, $2\Theta = 20 - 70^{\circ}$). The layers' morphologies and grain sizes were imaged using scanning electron microscopy (Hitachi S4700). The UV/Vis spectrophotometer (Perkin-Elmer, model Lambda 900) was employed for detecting absorption edges of the prepared samples. Chemical composition of the layers was analyzed by Auger Electron Spectroscopy – AES (Physical Electronics 560 AES/XPS).

RESULTS AND DISCUSSION

The sol-gel method appears to be a suitable candidate to produce high-quality pyrite layers. Fig. 1 shows the Raman spectrum of a five-layer α -Fe₂O₃ coating after sulfurization at different temperatures.



Fig. 1: Raman spectra of FeS_2 films sulfurized at different temperatures for one hour. In the graph, P - Pyrite, M – Marcasite and C – Carbon.

Using the temperature of 400 $^\circ$ C led to the chemical sulfurization of hematite phase film to pyrite FeS₂

(hereafter sample L1). Three basic Raman bands of pyrite, corresponding to the peaks at 339 cm⁻¹ \sim E_g, 374 cm⁻¹ \sim A_g and 426 cm⁻¹ \sim T_g, are observed in the spectrum (see Fig. 1). However, the chemical transformation is incomplete. This is demonstrated by the presence of the two peaks at around 1500 cm⁻¹, which are apparently attributed to the residual organic carbon coming from the sol-gel reagents. The low intensities of pyrite Raman peaks probably reflect the low extent of crystallinity of the sample. Contrary to L1, any peaks located at the position of carbon Raman bands are not seen in case of the sample sulfurized at 500 °C (sample L2). On the other hand different crystallographic structure corresponding to marcasite FeS₂ was revealed in this spectrum. The phase transformation is obviously due to the higher sulfurization temperature. The best result, in terms of desired pyrite crystal phase, was accomplished by performing the sulfurization at 450 °C. It is known that at this temperature, close to the sulfur boiling point, it is in its cyclic S8 form, which is the most suitable for the sulfurization processes.

Further, the effect of the sulfurizarion duration on the films properties was studied. The dwelling time was varied in the range between 15 min. and 6 hours. Somewhat surprising results were observed in Raman spectra shown in Fig. 2.



Fig. 2: Raman spectra of pyrite samples sulfurizing the hematite films at different time durations.

It is clearly seen from the graph that overall time of the sulfurization did not influence the crystallographic structure in any way. The three main Raman bands are present in every spectrum. The complete chemical transformation from the hematite to pyrite was achieved after 15 minutes. This is a very appealing condition for potential solar cell fabrication.

These films were also compared in terms of their morphological properties. The SEM images of the samples can be seen in Fig. 3. It is evident that increasing the time of sulfurization led to the formation of bigger grains aggregates, which resulted in films's greater roughness. Nevertheless, any significant defects in the films were not detected.



Fig 3: SEM images of the samples sulfurized at 450 °C for: (A) 15 min, (B) 1 h, (C) 3 h and (D) 6 h.

Furthermore the films were also observed using XRD analysis. The results correlate with the Raman spectroscopy data. The intensity of XRD peaks, which only appeared for the pyrite phase, is almost equal for all time durations, indicating the same extent of crystallization, see Fig. 4.

The sample with five layers is of interest because its thickness is in the appropriate range for photovoltaic applications. This film was annealed five times (once after each spin coating) at 600 °C to achieve the hematite structure and further sulfurized at 450 °C for 15 minutes.

The problem encountered in films of several layers was the incorporation of Na in the films. The more times the sol-gel material is annealed the more likely the Na has of diffusing from the soda lime glass into the substrate.



Fig. 4: XRD spectra of pyrite films prepared by sulfurizations of varying time duration.

This effect has been confirmed by glow discharge optical emission spectroscopy (GDOES) which showed considerable penetration of the Na into the FeS_2 (see Fig. 5). This problem is being studied by using various barrier coatings and different substrates.



Fig. 5: GDOES elemental depth profile of the film sulfurized for 15 min at 450 °C.

Another important diagnostic tool to determine the chemical composition of the pyrite films was AES. A typical depth profile Auger scan of the films revealed that the stoichiometry of the film was almost precisely 2 (S) : 1 (Fe), within the range of precision of the AES system and was constant through the entire film.

Because pyrite films are being studied for potential use in solar cells, their optical properties are among the most important parameters. The energy band gap value is

critical. The intersection of the trend line for the linear portion of the Tauc curve with the energy (x) axis denotes the band gap energy width. The band gap energy determined form this measurement was 0.93 eV, which is just slightly below the expected value of 0.95 eV [3].



Fig. 6: Tauc plot of the film sulfurized for 15 min at 450 °C. The intersection of the red line with energy axis denotes the band gap energy.

The photo-responses of the films were assessed using photoelectrochemical experiments of amperometry. For this purpose the system of conductive (Indium doped tin oxide) ITO substrate / anatase TiO_2 / FeS₂ was prepared. The system was illuminated with light corresponding to band gap of FeS₂., which served as the prime absorber sensitizing the anatase film. The system is conventional three electrode set-up of a working electrode (ITO/TiO2/FeS2), auxiliary electrode (Pt) and a reference electrode (Ag/AgCl in 3M KCl).



Fig. 7: Photocurrent curve – dependency on time at fixed potential of 0.6 V.

Fig. 7 shows the dependence of current density (photocurrent) on time under a fixed potential of 0.6 V. The rather low value of the photocurrent is probably due to the

high recombination rate of the carriers (electron/hole pair) and sulfur deficiency in the lattice.

CONCLUSION

The novel approach for the production of the pinhole free thin films of FeS₂ pyrite was described. The method is based on the sol-gel preparation of α -Fe₂O₃ (hematite) thin layers and their subsequent chemical sulfurization. The effects of thermal annealing of hematite films and temperature of the sulfurization on the properties of the films is discussed. It is apparent that this sol-gel method produces promising photovoltaic absorber material.

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