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The electronic structure and band hybridization of Co/Ti doped BaFe₁₂O₁₉

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Abstract: We identify contributions to the valence band of the nanosized BaFe_{12−2}*x*Co_{*x*}Ti_xO₁₉ barium ferrite particles, from the cobalt and titanium dopants. Resonant photoemission results show that cobalt and titanium dopants strongly hybridize with the barium ferrite matrix. Fano resonances are identified in the valence band region, at the Ba $(5s)$, Ti $(3p)$ and Co $(3p)$ thresholds, and there are significant contributions to the density of states, residing within 2 eV of the Fermi level, from barium and/or iron.

Keywords: Doping ferrites, Phase equilibria of ferrites

1. Introduction

There has been a significant thrust to develop nanocrystallites of magnetite (Fe₃O₄) and other ferrites [1], [2], [3] and $[4]$ on the multigram scale for application as ferrofluids [5] and [6] and magnetic recording media. The magnetic anisotropy of the BaFe_{12−2*x*}Co_{*x*}Ti_{*x*}O₁₉ barium ferrite particles strongly depends on the extent of cobalt and titanium doping [6], so that the bonding and hybridization of the dopants with the barium ferrite matrix is of some interest.

Superlattices, notably of oxides, with unusual electronic and magnetic properties as well as charge transfer in complex oxides have been investigated by photoemission and resonant photoemission [7], [8], [9], [10], [11], [12], [13] and [14]. While ultraviolet photoemission suffers from the drawback that the technique is an extremely surface sensitive technique, resonant photoemission can be exploited to some effectiveness to ascertain charge localization [7] and [8] and provide some insight as to bonding configuration [11], [12], [13], [14] and [15]. We have undertaken to use photoemission, with some theoretical support, to explore the local environment of the doping species (Co/Ti) within nanosized BaFe $_{12}O_{19}$ particles.

2. Experimental and theoretical methodology

The nanocrystalline barium ferrite BaFe12−2*x*Co*x*Ti*x*O19 particles were prepared by the glass crystallization method, as described elsewhere [3]. A compressed pellet was prepared suitable for ultra-high vacuum photoemission. The samples were prepared by a sequence of sputtering an annealing to remove contamination from the large surface area of these samples (120 m^2/g), but care was taken to prevent sintering and excessive preferential segregation, as determined from the shallow Ba 5p, Ti 3p, Co 3p and Fe 3p core levels. The average particle size of the BaFe12−2*x*Co*x*Ti*x*O19 particles was 9 nm in diameter, with a saturation magnetization of 22.8 Am²/kg and coercive field of 1.1 kA/m.

The angle-resolved photoemission spectra were obtained from the polarized synchrotron radiation dispersed by a 3 m toroidal grating monochromator, at the Center for Microstructures and Devices (CAMD) in combination with hemispherical electron energy analyzer with photoelectrons collected normal to the surface, as described elsewhere [16]. The combined resolution of the electron energy analyzer and monochromator, for the spectra shown herein, is 120– 150 meV for high kinetic photon energies (50–120 eV), but there is higher resolution at the lower photon energies (about 80 meV for photon energies of 25–40 eV). With the highly plane polarized synchrotron light, photoemission experiments were undertaken with a light incidence angle of 45° with respect to the surface normal (s + p polarization) and with the photoelectrons collected along surface normal. All binding energies are referenced to the Fermi level, as determined from clean tantalum in contact with the sample.

To model the various contributions to the photoemission, we undertook theoretical calculations of the density of states using the FEFF8 code [17], whose suitability to obtain detailed information on the electronic structure has been shown in a number of previous studies, e.g. [18], [19] and [20]. This self-consistent approach (FEFF8) are real space Green's function (RSGF) calculations, which include full multiple scattering (FMS) from atoms within a cluster of a given size, using the spinor relativistic approximation for the solutions of the Dirac equation. Naturally, a cluster approach limits the resolution of the density of states calculations. This limitation is evident in **Fig. 1b**, as the density

Fig. 1. (A) Ba L_{III} -XANES spectra of experimental (dashed) and calculated using FEFF8 code (solid) $\text{BaFe}_{12}\text{O}_{19}$, (B) The partial p — (dashed line) and *d* — (solid line) projected density of states for (doped) bariumhexaferrite as obtained by the FEFF8 code. The position of the vertical lines labeled 1–5 corresponds to the valence band photoemission features displayed in **Fig. 2**, plotted relative to the Fermi energy as estimated by FEFF8.

of states calculation does not reproduce the band-gap of the material correctly, which just appears as a dip in the density of states plot. Nonetheless, these model calculations determine the position of the various bands sufficiently well to aid in the assignment of the photoemission features. These calculations take in account the photoemission final-state somewhat better than is the case in ground state calculations as the Fermi energy position is estimated in the presence of a screened core hole.

The good agreement between calculated and experimental barium L_{III} -XANES spectra, as shown in **Fig. 1a**, is a "validation" of the efficacy of these calculations. XANES spectra were collected at the XMP (X-Ray Microprobe) beamline at the Center for Microstructures and Devices (CAMD). For the XANES spectra, monochromatic light was obtained by using a double crystal monochromator of Lemonnier type [21], equipped with Si(311) crystal pair $(d = 3.276$ Å) and the overall resolution was $0.8-1.0$ eV.

3. Resonant photoemission

In the valence band region, there are several occupied bands that are readily distinguished at 1.8, 3.2, 4.8, 7 and 10.8 eV, as seen in **Fig. 2**. More careful analysis suggests that the feature at 7 eV is in fact a result of contributions from two broad photoemission features at about 6.3 and 7.6

Fig. 2. Valence band photoemission spectra of nanocrystalline barium ferrite BaFe12−2*x*Co*x*Ti*x*O19 particles as a function of photon energy across the barium 5s core level threshold. The spectra were collected along normal to the surface at room temperature. (b) The various contributions to the spectra are indicated by the deconvolution of the spectra taken at 29 eV photon energy.

eV binding energy, thus explaining the weak shoulder, at higher binding energies, of the main photoemission feature, as indicated in **Fig. 2**.

As seen in **Fig. 3** and **Fig. 4**, the barium $5p_{1/2}$ and $5p_{3/2}$ shallow core levels are observed at 17.5 and 15.6 eV, respectively, are shifted from the expected 17.2 eV and 14.9 eV binding energies [22], as is expected from strong oxida-

Fig. 3. Valence band photoemission spectra of nanocrystalline barium ferrite BaFe12−2*x*Co*x*Ti*x*O19 particles as a function of photon energy across the titanium 3s and cobalt 3p core level thresholds. The spectra were taken at room temperature, with the photoelectrons collected along normal to the surface.

Fig. 4. Photoemission spectra taken at room temperature after prolonged heating followed after several short heating/sputtering cycles: (1) prolonged heating, 10 h annealing at 685 K, (2) after brief heating, (3) after sputtering. Spectra of the Co 3p and Fe 3p shallow cores (right panel), showing enhancement of Co 3p and/or Co and Ba Auger with annealing while the enhancement of the valence band 1.8 eV binding energy feature is shown at left. The photon energy was 115 eV.

tion and charge transfer. Similar increases in binding energies are observed with the other shallow core levels, again characteristic of the oxide. Thus the resonant photoemission threshold photon energies could be as much as 0.5 to 1 eV greater than the 'nominal' binding energy.

In spite of the limitations of photoemission, synchrotron based resonant photoemission provided evidence of pronounced resonances in the valence band spectra of a compress of Co/Ti doped nanosized BaFe₁₂O₁₉ particles. The valence band resonances occur at photon energy characteristics of the Ba 5s and Ti 3p core to bound excitation energies, at 30.3 and 32.6 eV photon energies respectively, as seen in **Fig. 1.** In the region of 30 eV photon energy, these resonant enhancements are most pronounced in the valence band features at 1.8, 7 and 10.8 eV binding energy, while the feature at 4.8 eV binding energy undergoes a 'delayed' resonance at 32 eV photon energy.

The valence band resonances at Ba 5s threshold, obtained in the vicinity of 30 eV photon energy are summa-

Fig. 5. The intensities of the photoemission features at 10.8 eV (a), 7.7 eV (b) and 1.8 eV (c) binding energy for nanocrystalline barium ferrite BaFe_{12−2*x*}Co_{*x*}Ti_{*x*}O₁₉ particles, as a function of photon energy.

rized in **Fig. 5**. The intensities, with changes in photon energy, exhibit the characteristic Fano resonance line shape, but this can occur in photoemission of the transition metals without invoking a super-Koster Kronig type resonance photoemission [23].

Additional enhancements of the valence band features at about 1.8 and 7 eV binding energy occurs at about 53 eV photon energy while a strong enhancement of the band at 4.8 eV occurs at about 62 to 66 eV photon energy, as indicated in **Fig. 3**.

4. Atomic contributions to the valence band features

Because of the strong resonant behavior, we can make some assignments of elemental contribution to the various bands observed in the occupied valence band structure of cobalt and titanium doped BaFe_{12−2*x*}Co_{*x*}Ti_{*x*}O₁₉ nanocrystallites. The valence band feature at 1.8 eV must contain both barium and transition metal iron and cobalt contributions as this feature is resonantly enhanced by excitations from the Ba 5s core (nominally at 30.3 eV, for unoxidized Ba) as well as at 60–62 eV, by excitations from the Co $3p_{1/2}$ and $3p_{3/2}$ cores (nominally at 58.9 eV, for unoxidized Co). This is consistent with our model calculations, which suggest presence of Ba 5d and oxygen 2p contributions with strong Fe and Co contributions, but much smaller Ti contributions to the density of states close to the Fermi level (**Fig. 1b**, vertical line labeled 1).

The strong feature at about 4.8 eV binding energy does not strongly resonate across the Ba shallow core levels but does undergo a photoemission enhancement at photon energies in the range of 62–66 eV. This tends to indicated that this band has both cobalt and titanium contributions. The broad range of photon energies, where photoemission enhancement occurs, tends to suggest that there are multiple resonances contributing to the valence band photoemission, and titanium 3s (nominally at 58.7 eV, for unoxidized Ti) and cobalt 3p (nominally at 58.9 eV, for unoxidized Co) lie very close in binding energy. As the core level shifts, due to oxidation which tends to separate the core level binding energies, this will tend to broaden the width of the photoemission resonance, as is observed in the vicinity of 62 eV photon energy. The "delayed" photoemission resonance of this feature at 4.8 eV binding energy at 32 eV, instead of 30 eV photon energy is consistent with the assignment of some titanium weight to this band (the Ti 3p shallow core occurs at nominally at 32.6 eV, for unoxidized Ti). Here theory suggests that the density of states in this region of the valence band is principally due to Ti 3d and Co 3d (**Fig. 1b**, vertical line labeled 3).

The features at about 7 and 10.8 eV binding energy show resonant enhancements in the valence band spectra at 30 eV photon energy as do the barium shallow 5p cores, and must therefore contain barium contributions. As the feature at 7 eV binding energy also undergoes a resonant enhancement at 53 eV photon energy, by excitations from the Fe $3p_{1/2}$ and $3p_{3/2}$ cores (nominally at 52.7 eV, for unoxidized Fe), we believe this feature also contains some iron contributions. On this basis, we must conclude that barium is strongly hybridized to iron mediated by the oxygen as both the bands at 1.8 and 7 eV binding energy contain both barium and iron contributions. Strong Fe 3d and Co contributions in this region are found in the calculated density of states **(Fig. 1b**, vertical line labeled 4) but little barium. If this calculated density calculation is indeed representative, then the only possible explanation of the photoemission enhancement at a barium core edge would be that the barium must be in close proximity to iron to permit extra-atomic resonant enhancements. Such extra-atomic excitation, although generally a weaker contributions to the resonant photoemission process than intra-atomic excitations, would explain the greater than expected Fe 4p contributions to the 1.8 eV photoemission feature and the stronger than expected barium contributions to the photoemission features at 7 and 10.8 eV binding energy.

Choosing a photon energy above both the Ba 4d thresholds (nominally at 92.6 eV and 89.9 eV, for unoxidized Ba) and the Fe 3s threshold (nominally at 91.3 eV, for unoxidized Fe), we can enhance the iron and barium contributions to the valence band photoemission spectra, as seen in the spectra taken at a photon energy of 115 eV. This is evident from a comparison of the spectra in **Fig. 4** (115 eV incident photon energy) with **Fig. 3** (54 to 70 eV photon energies). The feature at 1.8 eV is strongly enhanced, and the resonances at the barium and cobalt core threshold suggest that barium may also be strongly hybridized with cobalt.

Both cobalt and titanium contribute to the band at 4.8 eV binding energy and although both are dopants. Uniform placements of cobalt and titanium dopants throughout each crystallite, therefore seems unlikely. Surface/interface segregation of cobalt and titanium is consistent with the annealing studies discussed below (vide infra). While inhomogeneous Co/Ti concentrations in different particles cannot be excluded on the basis of the data presented here, this alone cannot explain these results nor the segregation results discussed below. If cobalt is in close proximity to titanium, then, as with iron and barium, extra-atomic resonant enhancement of largely Ti 3d bands (at 4.8 eV binding energy) can occur at the cobalt edge, as suggested by the calculated partial density of states (**Fig. 1b**, the vertical line labeled 3).

The assignments of atomic contributions to the various valence band photoemission features are in generally good agreement with the *l* projected density of state calculations obtained using the FEFF8-code performed on the system, whose results are shown in **Fig. 1b**. The valence band photoemission features displayed in **Fig. 2**, corresponds to features in the model calculations indicated by the position of vertical lines 1–5, plotted relative to the Fermi energy as estimated by FEFF8, in **Fig. 1b**. The energy positions of structures 1–4 found in the photoelectron spectra can be easily assigned to prominent features in the *l* and *d* projected density of states plot, which in turn leads to a straightforward verification of the results obtained by photoelectron spectroscopy.

The only major discrepancy is the energy position of the barium core levels (the 5th line in **Fig. 1b**), which can be explained by the fact that this density of states is not related to a valence band feature but a shallow core. It is thus subject to stronger relaxation to lower energies due to the presence of a core hole in the calculations. The core-shell separation is treated on a different theoretical level by the FEFF8 code.

5. Segregation

Although no magnetically "dead" surface layer was indicated in prior studies of this system [6], some surface segregation of some of the constituents of these nanocrystallites is indicated. This is abundantly clear from the changes in the density of states closest to the Fermi level, as seen in **Fig. 4**. Extensive annealing (at 685 K for 10 h in vacuo) of the sample leads to enhancement of the feature at 1.8 eV binding energy, indicating that barium and cobalt are enriched near the surface region, while iron is depleted at the surface compared to the bulk BaFe_{12−2*x*}Co_{*x*}Ti_{*x*}O₁₉. This is evident from the shift in the spectral peaks near 60 eV from the broad feature at 61 eV (unoxidized Co 3p is nominally at 58.9 eV binding energy, while unoxidized Ti 3s is nominally at 58.7 eV) to features at 58 and 56 eV binding energy (unoxidized Fe 3p is nominally at 52.7 eV binding energy). Such segregation, following extensive annealing, does not require a magnetically dead layer at the surface of the crystallites as ferromagnetic Co doped titanium oxide is known.

While there has been some speculation concerning a dead layer in cobalt and titanium doped barium ferrites, from magnetic modeling [24], such a layer would be less than 1 nm to be consistent with the photoemission results presented here. The limited mean free path of the photoelectrons indicates that the segregation layer is quite thin, even after extensive annealing, consistent with the very short argon ion sputtering time needed to remove the segregation layer. Even if the segregation leads to a material that is nominally paramagnetic, solutions to the Landau–Ginzburg equation [25] indicate that a layer so thin $(< 1$ nm) would have an induced magnetization by proximity.

6. Summary

We find that barium and iron contribute the greatest density of states near the Fermi level in nanocrystalline barium ferrite BaFe12−2*x*Co*x*Ti*x*O19 particles. The strong hybridization of the barium and iron, mediated by oxygen, is expected, but the lack of uniform placement of cobalt and titanium needs further study. The segregation process in this system, with annealing, is under investigation, but may well provide insight into the alkaline earth titinate to $Fe₃O₄$ interfaces [26].

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