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Zhou, Jian; Al-Omari, I.A.; Liu, J. Ping; and Sellmyer, David J., "Structure and magnetic properties of SmCo_{7-x}Ti_x with TbCu₇-type structure" (2000). *David Sellmyer Publications*. 63. https://digitalcommons.unl.edu/physicssellmyer/63

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Structure and magnetic properties of $SmCo_{7-x}Ti_x$ with TbCu₇-type structure

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The SmCo_{7-x}Ti_x, x=0-0.56 bulk samples are prepared by arc melting. X-ray diffraction indicates that samples with 0.2 < x < 0.4 form a single disordered TbCu₇-type structure phase and other minor phases appear for other values of *x*, which indicates that Ti helps stabilize the 1-7 phase. The lattice parameters ratio (*c/a*) increases with increasing Ti concentration. Room temperature saturation magnetization and Curie temperature decrease with increasing *x*. X-ray diffraction and magnetization measurements on aligned samples show that all samples studied have uniaxial anisotropy. The anisotropy field is found to increase with increasing *x* reaching a maximum of 175 kOe at x=0.28 and then decreases for higher values of *x*. This anisotropy field is 20% higher than that of the same compound with Th₂Zn₁₇-type structure. © 2000 American Institute of Physics. [S0021-8979(00)21308-X]

I. INTRODUCTION

The rare-earth (RE) transition-metal intermetallic compounds have been widely investigated for many years, among them the Sm–Co series compounds with 1-5 and 2-17 crystal structures. These compounds have been used as sintered and bonded permanent magnets since the 1960s.^{1–3} Recently, attention has been focused on the TbCu₇-type structure Sm–Co intermetallic compounds because of their potential as high anisotropy magnetic materials.^{4–6} Both SmCo₇ and Sm₂Co₁₇ can have the 1-7 type structure when prepared appropriately. However, SmCo₇ is unstable unless a third doping element such as Ti, Zr, Cu, etc. is added to substitute Co.

TbCu₇-type crystal structure could be indexed according to the CaCu₅-type structure with significant deviations of the lattice constants and the x-ray peaks' intensities. That is because partial of the Ca sites are randomly substituted by a pair of Cu in CaCu₅-type structure. In Sm-Co compounds, SmCo₇ and Sm₂Co₁₇ can form the TbCu₇-type structure. Th₂Zn₁₇ structure can be formed if the substitution is ordered. Khan' found that Sm-Co with composition 1:5.4 has the TbCu₇-type structure above 1100 °C. The magnetic properties of these TbCu7-type Sm-Co alloys have been investigated by a few researchers. Saito et al.4 studied Sm₂(CoMn)₁₇ with TbCu₇-type disordered structure and found that the anisotropy constant (K_{u1}) is about 1.2–1.4 times larger than that for compounds with Th₂Zn₁₇-type structure. Recently, Huang *et al.*⁶ studied $\text{SmCo}_{7-x}\text{Zr}_x$ with TbCu₇-type disordered structure and found that at T= 300 K, these compounds exhibit anisotropy field as high as 180 kOe for x = 0.5. Lefevre *et al.*^{8,9} also reported the existence of TbCu7-type structure in Sm-Co-Zr ternary alloys prepared by arc melting. Disordered SmCo7 compound with TbCu7-type structure plays the role of a bridge between $SmCo_5$ and Sm_2Co_{17} compounds. It is natural to think that the 1:7 composition with 1-7 type structure may keep the merits of SmCo_5 such as high anisotropy and of $\text{Sm}_2\text{Co}_{17}$ such as large magnetization. However, a study of $\text{SmCo}_{7-x}\text{Zr}_x$ by Huang *et al.*⁶ showed that substituting Zr for Co increase the anisotropy field and decrease the magnetization. In this work, we focus our attention on the effects of Ti substituent on the structure and magnetic properties of $\text{SmCo}_{7-x}\text{Ti}_x$ alloys.

II. EXPERIMENT

A series of samples based on RE-(Co,Ti) composition 1:7 were prepared by arc melting. $\text{SmCo}_{7-x}\text{Ti}_x$ with x=0, 0.1, 0.2, 0.3, 0.35, 0.42, 0.56, and $\text{SmCo}_{6.5}$ (for comparison) with at least 99.9% pure elements were melted in a watercooled copper boat under flowing argon. An extra amount of Sm (8%–10%) was added to balance the loss of Sm due to evaporation. The alloys were melted several times to insure homogeneity. The same samples were annealed at different temperatures between 650 and 950 °C in a sealed quartz tube under argon pressure. All these samples were wrapped separately in Ta foils to prevent oxidation. The as-cast material was ground in a mortar and the fine powder particles (<38 μ m) were subsequently mixed with epoxy and aligned in a magnetic field of 12 kOe for 24 h for magnetic anisotropy measurements.

The crystal structures of the samples are determined by using x-ray diffraction. Loose powder samples were measured to determine the saturation magnetization at room temperature. Vibrating sample magnetometer (VSM) with a high temperature oven was used to measure the Curie temperature. A SQUID magnetometer with maximum field of 55 kOe was used to measure the magnetization of the aligned samples.

III. RESULTS AND DISCUSSION

Figure 1 shows the x-ray diffraction pattern of the ascast $\text{SmCo}_{7-x}\text{Ti}_x$ and $\text{SmCo}_{6.5}$. The pattern for SmCo_7

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FIG. 1. X-ray diffraction patterns for $\text{SmCo}_{7-x}\text{Ti}_x$ compounds.

sample shows two phases, SmCo₇ with the TbCu₇-type structure and Sm₂Co₁₇ with TbCu₇-type structure. This indicates that the SmCo₇ is unstable without the doping element, which is consistent with the results of Huang et al.,⁶ while Yang et al.⁵ reported single phase SmCo₇ with TbCu₇-type structure for samples prepared by mechanical alloying. For comparison, a composition SmCo_{6.5} with 1-7 structure is prepared and presented with the results. X-ray diffraction for samples with x = 0.2 - 0.35 show the existence of TbCu₇-type structure. This means that Ti can help the formation of the stable structure. For x > 0.4, other minor phases, mainly SmCo₃, appear with the 1-7 type structure. This behavior is similar to that of the $SmCo_{7-x}Zr_x$ series reported by Huang *et al.*⁶ Table I shows the changes in the lattice parameters (a)and (c) due to the doping element Ti in $SmCo_{7-x}Ti_x$ compounds. The lattice parameter a decreases with increasing Ti concentration, from 4.935 Å at x=0 to 4.900 Å at x=0.56, while c increases from 4.013 Å at x=0 to 4.134 Å at x =0.56 and hence the ratio (c/a) keeps increasing. X-ray diffraction for samples annealed at temperature between 650 and 950 °C shows the Th₂Zn₁₇-type structure. This indicates that annealing does not help the formation of a single TbCu₇-type structure.



FIG. 2. M-T measurement of SmCo_{7-x}Ti_x compounds under 1000 Oe applied field.

VSM magnetic measurements show that all the samples studied are ferromagnetic. The room temperature saturation magnetization (M_s) values for the different Ti concentrations are listed in Table I. It can be seen that M_s decreases with increasing x, where it drops from 102 emu/g for x =0-84 emu/g for x=0.56. Figure 2 shows the M-T curves of $\text{SmCo}_{7-x}\text{Ti}_x$. In this figure, T_{c1} and T_{c2} indicate the Curie temperatures for the 1:7 phase the 2:17 phase. The Curie temperature for the 1:7 phase decreases from 780 °C for x =0 to 710 °C for x = 0.56. For x > 0.42, measurements show trace of a small amount of SmCo₃, which support our x-ray diffraction results. For x values between 0.2 and 0.35, a single 1-7 magnetic phase presents. The decrease in the Curie temperature with increasing x indicates a weakening of the magnetic interactions. The metastable SmCo7 alloy decomposes into SmCo₅ and Sm₂Co₁₇ above 700 °C.

Field dependence of magnetization is investigated in the directions parallel (M_{\parallel}) and perpendicular (M_{\perp}) to the aligning magnetic field direction (easy axis direction). Figure 3 shows the magnetization curves for SmCo_{6.65}Ti_{0.35}. Other measurements for different values of x showed similar be-

TABLE I. Lattice parameters (a) and (c), (c/a) ratio, phases present, Curie temperature (T_c) , saturation magnetization (M_s) , and anisotropy field (H_A) of SmCo_{7-x}Ti_x compounds as a function of titanium concentration (x).

x	a (Å)	c (Å)	c/a	Phases	<i>T</i> _{<i>c</i>1} (°C)	M _s (emu/g)	H _A (kOe)
0	4.935	4.013	0.813	1:7+ 2:17	780	102	120
0.21	4.920	4.060	0.825	1:7	756	96	156
0.28	4.897	4.056	0.828	1:7	745	94	175
0.35	4.882	4.076	0.832	1:7	742	92	168
0.42	4.872	4.074	0.835	1:7+ 1:3	733	89	160
0.56	4.900	4.134	0.842	1:7+ 1:3	710	84	



FIG. 3. Typical M-H loops for SmCo_{6.65}Ti_{0.35} aligned sample at a temperature of 300 K.

havior. Extrapolation of M_{\perp} and M_{\parallel} curves were used to estimate the anisotropy field (H_A) . Table I shows the H_A values of TbCu₇-type SmCo_{7-x}Ti_x as a function of x. H_A increases with increasing Ti concentration reaching a maximum of 175 kOe at x = 0.28 and then decreases for larger values of x. These results are similar to the results by Huang et al.⁶ for SmCo_{7-x}Zr_x compounds and comparable to the results of Satyanarayana *et al.*¹⁰ for $Sm_2Co_{17-x}Ti_x$ compounds, where H_A increases from 90 to 125 kOe by increasing x from 0 to 0.5. This effect is important especially in the preparation of commercial $Sm(CoM)_{7+z}$ hard magnetic materials. The H_A value of the 1-7 structure structure $SmCo_{6.79}Ti_{0.21}$ is 20% larger than our H_A value for the same sample with the 2-17 type structure after annealing at 750 °C. The increase in H_A is due to the change in the anisotropy constant K_{u1} which is caused by the preferential occupation of the dumbbell sites by the doping elements as suggested by Deportes *et al.*^{11,12}

IV. CONCLUSIONS

Samples of TbCu₇-type structure SmCo_{7-x}Ti_x have been prepared and their crystal structures were determined by x-ray diffraction. Doping Ti can help stabilize TbCu₇-type structure in these alloys. The doping element is found to decrease T_c and M_s for all values of x, while increasing the anisotropy field for small values of x. An anisotropy field as high as 175 kOe is obtained in these compounds which is about 20% higher than that for the same samples with 2-17 type structure. The basic results in this article help the understanding of the effects of Ti substitution on the structure and magnetic properties of magnetic materials for high temperature applications.

ACKNOWLEDGMENT

The authors would like to thank the AFOSR for the financial support under Grant No. AFOSR-F49620-98-1-0098.

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