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## Co-localised Raman and force spectroscopy reveal the roles of hydrogen bonds and $\pi$ - $\pi$ interactions in defining the mechanical properties of diphenylalanine nano- and micro-tubes

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An integrated atomic force and polarized Raman microscope were used to measure the elastic properties of individual diphenylalanine (FF) nano- and micro-tubes and to obtain quantitative information regarding the inter-molecular interactions that define their mechanical properties. For individual tubes, co-localised force spectroscopy and Raman spectroscopy measurements allowed the calculation of the Young's and shear moduli ( $25 \pm 5$  GPa and  $0.28 \pm 0.05$  GPa, respectively) and the contribution of hydrogen bonding network to the Young's modulus (~17.6 GPa). The  $\pi$ - $\pi$  interactions between the phenyl rings, dominated by T-type arrangements, were estimated based on previously published X-ray data to only 0.20 GPa. These results provide experimental evidence obtained from individual FF tubes that the network of H-bonds dominates the elastic properties of the FF tubes. © 2014 AIP Publishing LLC. [http://dx.doi.org/10.1063/1.4885090]

Self-assembled amyloid-like structures have unique physical and chemical properties, providing opportunities for engineering nano- and micro-scale biomaterials.<sup>1,2</sup> The formation of fibrils has also been associated with numerous pathologic conditions such as Alzheimer's and Creutzfeldt-Jacob diseases.<sup>3–5</sup> The diphenylalanine (FF) (L-Phe-L-Phe) peptide is the core recognition motif for self-assembly of the  $\beta$ -amyloid peptide associated with Alzheimer's disease.<sup>6</sup> This peptide has been used as a building block for fabrication of nano- and micro-tubular structures (Fig. 1) that have been proposed for biosensing, bioimaging, drug delivery applications, as well as for 3D tissue culture scaffolds.<sup>7</sup> Previous studies showed that the elastic moduli of protein fibrils vary over four orders of magnitude.<sup>8</sup> For some amyloid fibrils, values of the Young's moduli as high as 13 GPa have been measured.<sup>8</sup> Theoretical models indicated that the back-bone interactions, which are dominated by the hydrogen-bonding network and modulated by side-chain interaction, were crucial in stabilizing the structure of the fibrils and defining their high rigidity.<sup>8</sup> Similarly, the high density of H bonds between the nanoscale crystalline monomers in silkworm and spider silks was shown to provide high elastic moduli, similar to amyloid-fibrils.9

The mechanical properties of FF tubular structures have been measured by atomic force microscopy (AFM) indentation measurements and bending beam models.<sup>10,11</sup> The values for the Young's moduli from these studies were reported to be 19 GPa (Ref. 10) and  $27 \pm 4$  GPa,<sup>11</sup> and shear modulus of  $0.15 \pm 0.02$  GPa.<sup>11</sup> Recent density functional theory studies described the structure of the FF fibrils in terms of an array of peptide nanotube backbones with interpenetrating "zipperlike" aromatic interlocks. It was suggested that the Young's modulus in the longitudinal direction had a dominant contribution from the H bonds linking the peptide backbones, while dispersive interactions, mostly  $\pi$ - $\pi$  interactions, had a contribution higher than  $\sim 50\%$  to the Young modulus in the directional perpendicular to the tube axis ( $\sim 8.8$  GPa).<sup>12</sup>

In this study, an integrated atomic force-Raman microscope was used to perform co-localised force spectroscopy and polarised Raman micro-spectroscopy (PRMS) measurements on individual FF nano- and micro-tubes. Such a setup allows measurement of the Young's and shear modulus by the bending beam method described and co-localised



FIG. 1. Schematic description of the diphenylalanine (FF) porous tubular structures. (a) FF building block; (b) cross-section of the porous FF tubes showing the peptide cylinders connected by the phenyl rings; (c) FF cylinders formed by the peptide backbones. In (b) and (c), the peptide backbone is highlighted by a purple line, and the side chain (phenyls) denoted by a green annulus.

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polarized Raman spectra that reveal the molecular structure, orientations, and force constants for particular vibrational modes. Therefore, the Young's modulus and the contributions from the hydrogen bonded to the mechanical strength can be measured experimentally on individual FF tubes.

All peptide solutions were prepared by initially solubilising lyophilised L-diphenylalanine (FF) peptide (Sigma Aldrich) in 1,1,1,3,3,3-hexafluoro-2-propanol (HFIP) (Sigma Aldrich) to give a stock solution of 100 mg/ml. Stock solutions were further diluted to a working concentration of 2 mg/ ml using ultrapure water (pH 6, resistivity 18.2 M $\Omega$  cm). Copper transmission electron microscope (TEM) grids with window size of 19 × 19  $\mu$ m (G1000HS, Gilder grids, UK), which have empty windows, were utilized as a support for the tubes. Using a pair of tweezers, the TEM grids were pulled gently through a solution of FF tubes so that tubes attached on one side. Immediately after this, the grids were deposited onto a quartz microscope slide (SPI supplies) and air-dried.

All measurements were performed using an in-house polarized Raman micro-spectrometer and integrated atomic force microscope (NanoWizard II, JPK).<sup>13</sup> The instrument allows for co-localised force spectroscopy and polarised Raman spectroscopy measurements for individual FF tubes. Each AFM tip (AC240TS, Asylum) was calibrated for the sensitivity [nm/V] by acquiring a force curve from a clean glass coverslip and linearly fitting the contact region. The spring contact of the cantilever [N/m] was subsequently determined by fitting a Lorentzian curve to the thermal resonance of the tip away from a surface. This calibration allows the voltage setpoint for the AFM to be converted to an equivalent force setpoint. The polarised Raman spectra were recorded using a similar experimental setup and methods as reported previously.<sup>14</sup> The Raman spectra were recorded while force measurements were performed-during which the tip was kept at the corresponding force setpoint value for 50 s, while the Raman spectra were recorded for 45 s within this time window.

The Young's and shear moduli of a single tube can be determined by pressing sections of a tube at the half-way point of the suspended length  $L^{11}$ 

$$\frac{1}{E_b} = \frac{1}{E} + \frac{48f_S I_S}{GA} \frac{1}{L^2},$$
 (1)

where *E* and *G* are the Young's and shear moduli of the tube, respectively,  $f_s$  is the form factor of the tube,  $I_s$  is the second moment of area, *A* is the cross-sectional area, and *L* is the suspended length. The term  $E_b$  is the bending modulus and is written in terms of the force applied by the AFM tip, and the deflection as

$$\frac{1}{E_b} = 192 \frac{\delta I_S}{FL^3}.$$
 (2)

For each section of a suspended tube, an AFM image was acquired in order to measure L and the diameter D(which will then give  $I_S$ ). The deflection  $\delta$  was obtained by finding the difference in the piezo-stage z-position for the same force setpoint (60 nN) between measurements on and away from the midpoint of the suspended tube, as seen in Fig. 2(b). Effects due to deformation and indentation have been excluded, as these been shown to occur only at sufficiently elevated temperatures.<sup>15</sup>

Prior to obtaining co-localised force spectroscopy and Raman spectroscopy data from individual FF tubes, the elastic moduli of four FF tubes (two of which have diameter larger than 1500 nm, and two have diameter of roughly 800 nm) were measured using the bending beam model to establish the accuracy of this method (Fig. 2). For each tube, the values of *E* and *G* were calculated by linearly fitting the results shown in Fig. 2(b). The force spectroscopy results indicate that the Young's and shear moduli for this particular FF nanotube are  $25 \pm 4$  GPa and  $0.28 \pm 0.05$  GPa, in agreement with previous reports.<sup>11</sup> The elastic moduli for the FF tubes are significantly higher than for other cross- $\beta$  amyloid structures investigated previously (2–14 GPa).<sup>8</sup>

To provide insight into the contributions of the H bonding network and side-chain interactions to the elastic moduli of the FF tubes, Raman spectra and force spectra were acquired for individual FF tubes. Fig. 3 shows an example of co-localised AFM force spectra and polarised Raman spectroscopy measurements for a typical FF tube. After the collection of force spectra that allowed calculation of the Young's modulus, polarised Raman spectra were acquired from the same FF tubes while using the AFM tip to apply forces in the range of 0–240 nN.

The band assignment for the Raman spectra of FF nanotubes has been reported earlier based on experimental and



FIG. 2. Typical example of using the bending beam model to measure the Young's and shear moduli of FF tubes. (a) Composite micrograph showing the length of a single isolated tube on a copper grid, with accompanying AFM images for six suspended sections of differing lengths *L*. (b) Force spectra measured at different sections (blue-extend, red-retract), and at the fixed end of the tube (black-extend, blue-retract). (c) Plot of  $1/E_b$  versus  $1/L^2$  for the measurement of the Young's and shear moduli using Eq. (1).

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density functional theory.<sup>14</sup> In particular, the 1249 cm<sup>-1</sup> Raman band was assigned to the amide III band, which corresponds to the combination of C-N stretching and N-H inplane bendings.<sup>14,16,17</sup> The polarization dependence of the amide III band is noticeable in Fig. 3 as the intensity in the XX configuration is higher than in the ZZ configuration, indicating that the principal axis of the Raman tensor of the amide III vibration is oriented perpendicular to the axis of the nanotubes (the first index denotes the laser polarization and the second index represents the direction of the analyzer, while the excitation laser and the Raman back-scattered radiation propagated in the Y direction). The frequency of the amide III band depends on the properties of the backbone, peptide bond, and conformation of peptides (Ramachandran angles) as well as hydrogen bonding at the N-H and C = O. Theoretical studies have showed that hydrogen bonding at N-H site can increase the vibration frequency of the amide III band by  $\sim 22 \text{ cm}^{-1}$ .<sup>18</sup>

Experimental evidence of blue-shifting of the amide III band upon H bonding at the N-H site has been reported, both for  $\alpha$ -helical and  $\beta$ -sheet polypeptides.<sup>19</sup> In addition, comparison between experimental and theoretical calculations of the Raman spectra of polypeptides allowed the estimation of the force constants corresponding to H bonds based on the frequency of the amide III band.<sup>20–22</sup> Theoretical calculations showed that for  $\beta$ -sheet polypeptides, such as  $\beta$ -(Ala)<sub>n</sub> strong H bonds (N-H...O angle ~165°, H...O distance ~1.76 Å) led to amide III band at ~1243 cm<sup>-1</sup>, which corresponds to a force constant of 15 N/m.<sup>21,22</sup> On the other hand, a lower frequency of the amide III band in the Raman spectrum of  $\beta$ -(GlyI)<sub>n</sub> (1234 cm<sup>-1</sup>) was correlated with weaker H bonding (N-H...O angle ~134°, H...O distance ~2.12 Å) and a lower value of the force constant of ~12.5 N/m.<sup>20</sup> FF



FIG. 3. Example of performing simultaneously polarized Raman and force spectroscopy measurements on the FF tube for which the elastic moduli measurements were presented in Fig. 2. The direction Z is along the FF tube and the excitation laser and the Raman back-scattered radiation propagated in the Y direction.

tubes have no H bonds at the C=O sites;<sup>14</sup> therefore, the amide III band at 1249 cm<sup>-1</sup> indicates strong H bonding corresponding to an estimated force constant of  $f_{Am} = 15$  N/m. These results agree with the geometrical results obtained from X-ray data for the FF crystals indicating a N-H...O angle of ~154° and H...O distance of ~1.86 Å.<sup>12</sup> Fig. 3 also shows that no shifts in the frequency of the amide III were detected in the polarised Raman spectra when forces in the range of 0–240 nN were applied to the FF tubes. These results indicates that bending of the FF tubes led did not change the force constants associated to the H bonding, suggesting a strong contribution of the tubes.

Apart from the H bonds associated to the N-H sides of the amide bonds, there are an additional six H bonds per unit cell associated to the head-to-tail interactions  $NH_3^+...^{-}OOC$  (Fig. 4). Based on the X-ray data of FF crystals, these bonds are characterised by  $a \sim 144^{\circ}$  N-H...O angle and  $\sim 1.89$  Å



FIG. 4. Schematic description of the inter-molecular interactions in porous FF tubes. (a) The network of H bonds provides a strong contribution to the elastic modulus in the longitudinal direction (Z axis). (b) The high-density of phenyl rings leading to  $\pi$ - $\pi$  interactions (both T-shaped and parallel displaced) increases the elastic moduli both in the Z and X-Y directions.

This article is copyrighted as indicated in the article. Reuse of AIP content is subject to the terms at: http://scitation.aip.org/termsconditions. Downloaded to IP: 92 232 188 255 On: Wed, 09 Jul 2014 11:00:17 H...O distance,<sup>12</sup> suggesting a lower force constant, of  $f_{HT} = 13.87$  N/m compared to the H bonds corresponding the amide bonds. Using the values for the force constants for the H bonding, the contribution of the H bonding network in to the Young's modulus of FF nano- and micro-tubes can calculated by using a simple network of springs linking the units cells along the tube axis

$$E_{HB} = f_{Am}h\zeta_{Am} + f_{HT}h\zeta_{HT} = 17.6 \,\text{GPa}, \qquad (3)$$

where h = 5.38 Å distance between the ring-like structures of six molecules (or the unit cell along the tube axis), and  $\zeta_{Am} = \zeta_{HT} = 6/(23 \times 23)$  Å<sup>-2</sup> is the density of the H bonds.

The value of  $\sim 17.6$  GPa for the H bonding contribution is in close agreement to the theoretically estimated value of  $\sim$ 17 GPa attributed to the back-bone H bonding network in cross- $\beta$  amyloids.<sup>8</sup> In addition to H bonding interactions, Fig. 4 shows that interactions between the phenyl rings can also produce a considerable contribution to the elastic modulus in the Z direction. It is known that the benzene dimer can form two distinct stable configurations, commonly termed Tshaped and parallel-offset interactions.<sup>23</sup> Fig. 5(b) shows a cross-section of the atomic model for the FF tubes (built using supplementary material from Azuri et al.<sup>12</sup>), with the phenyl interactions split into green boxes and blue triangles depending on whether they are directly between two backbone rings, or between three rings, respectively. Fig. 4(b) shows the arrangement for these sections along the tube axis (hydrogens removed for clarity). The interactions can be viewed as combinations of T-shaped and parallel offset interactions in both groups of phenyl rings. The interaction strength can be estimated from the potential curves for benzene dimers<sup>24</sup> and using the fact that there are 12 phenyl side chains in the unit cell, 6 in a green box region and 6 in the blue triangular region. For the T-shaped interactions in the green boxed regions, the equilibrium potential and separation (measured from molecular model<sup>12</sup>) are -2.1 kcal/mol and  $h_{\text{T-green}} = 5.0 \text{ Å}$ , respectively. Modelling the interaction as a harmonic oscillator gives a spring constant of  $f_{\text{T-green}} = 0.11 \text{ N/m}$ . Similarly, for the phenyl rings in the blue triangular regions, the equilibrium potential and separation are -2.6 kcal/mol and  $h_{\text{T-blue}} = 4.6$  Å, leading to a force constant of  $f_{\text{T-blue}} = 0.17 \text{ N/m}$ . For the parallel-offset configuration of the phenyl rings, the energy can be estimated using the potential function reported by McGaughey et al.,<sup>25</sup> for a centroid distance measured to be  $h_{\text{offset}} = 5.38 \text{ Å}$  (similar to the green and blue boxed regions). This leads to an energy of -0.4 kcal/mol, which in turn gives a force constant of  $f_{\rm offset} = 0.048$  N/m. Using these values, the contribution of the  $\pi$ - $\pi$  interactions to the Young's modulus is

$$E_{SC} = (6h_{T-green}f_{T-green} + 6h_{T-blue}f_{T-blue} + 12h_{offset}f_{offset})/A,$$
(4)

where  $A = 23.89 \times 23.89$  Å is the area of the unit cell. From the T-shaped interactions, the Young's modulus is 0.15 GPa, and the contribution of the parallel offset interactions is 0.05 GPa, providing a total contribution to the Young's modulus of FF tubes of 0.20 GPa. Knowles *et al.*<sup>8</sup> estimated the range of the side-chain contributions to the Young's modulus for cross- $\beta$  amyloid fibrils using a surface-tension model, which resulted in a range of 0.03–0.13 GPa. The value of 0.20 GPa calculated here is significantly higher than the upper limit specified for amyloid fibrils.<sup>8</sup>

However, this contribution is still almost two orders of magnitude lower than the contributions from the intermolecular H bonding. Nevertheless, the estimated Young's modulus of 17.8 GPa is significantly lower than the experimentally measured modulus of 25 GPa from the bending beam model. First, neglecting the anisotropy of these materials may cause the bending beam model to give an overestimate of the Young's modulus. Second, the calculations of the backbone and side-chain interactions may not take into account all aspects of intermolecular interactions contributing to the elastic moduli, such as, for example, the residual water molecules inside the pores, which simulations have shown to be stabilizing.<sup>26</sup>

The integrated atomic force and polarized Raman microscope (AFM-PRM) was used to measure the elastic moduli of individual diphenylalanine nano- and micro-tubes and to obtain quantitative information regarding the inter-molecular interactions that define the mechanical properties. Colocalised force spectroscopy and Raman spectroscopy measurements allowed the calculation of the Young's and shear moduli ( $25 \pm 5$  GPa and  $0.28 \pm 0.05$  GPa, respectively) and the contribution of hydrogen bonding network to the Young's modulus (~17.6 GPa). The  $\pi$ - $\pi$  interactions between the phenyl rings have an estimated total contribution to the Young's modulus of FF tubes of 0.20 GPa. Although this contribution is significantly larger compared to side-chain interactions in other amyloid fibrils, it is clear that the elastic properties of the FF tubes are dominated by the backbone H bond network.

The combination of AFM-force and polarized-Raman spectroscopy in a single instrument allows for quantitative co-localised mechanical and structural measurements on individual FF nanotubes. In principle, this technique could be applied to other nanoscale peptide fibrils for similar measurements at the scale of single fibrils.

- <sup>2</sup>I. W. Hamley, Angew. Chem. Int. Ed. 46, 8128 (2007).
- <sup>3</sup>D. B. Teplow, Amyloid 5, 121 (1998).
- <sup>4</sup>M. Goedert and M. G. Spillantini, Science **314**, 777 (2006).
- <sup>5</sup>E. D. Roberson and L. Mucke, Science **314**, 781 (2006).
- <sup>6</sup>M. Reches and E. Gazit, Science **300**, 625 (2003).
- <sup>7</sup>X. Yan, P. Zhu, and J. Li, Chem. Soc. Rev. **39**, 1877 (2010).
- <sup>8</sup>T. P. Knowles, A. W. Fitzpatrick, S. Meehan, H. R. Mott, M. Vendruscolo,
- C. M. Dobson, and M. E. Welland, Science 318, 1900 (2007).
- <sup>9</sup>S. Keten, Z. Xu, B. Ihle, and M. J. Buehler, Nature Mater. 9, 359 (2010).
- <sup>10</sup>N. Kol, L. Adler-Abramovich, D. Barlam, R. Z. Shneck, E. Gazit, and I. Rousso, Nano Lett. 5, 1343 (2005).
- <sup>11</sup>L. Niu, X. Chen, S. Allen, and S. J. B. Tendler, Langmuir 23, 7443 (2007).
- <sup>12</sup>I. Azuri, L. Adler-Abramovich, E. Gazit, O. Hod, and L. Kronik, J. Am. Chem. Soc. **136**, 963–969 (2014).
- <sup>13</sup>C. S. Sweetenham, M. Larraona-Puy, and I. Notingher, Appl. Spectrosc. **65**, 1387 (2011).
- <sup>14</sup>B. Lekprasert, V. Korolkov, A. Falamas, V. Chis, C. J. Roberts, S. J. B. Tendler, and I. Notingher, Biomacromolecules **13**, 2181 (2012).
- <sup>15</sup>V. L. Sedman, L. Adler-Abramovich, S. Allen, E. Gazit, and S. J. B. Tendler, J. Am. Chem. Soc. **128**, 6903 (2006).
- <sup>16</sup>S. A. Asher, A. Ianoul, G. Mix, M. N. Boyden, A. Karnop, M. Diem, and R. Schweitzer-Stenner, J. Am. Chem. Soc. **123**, 11775 (2001).
- <sup>17</sup>A. V. Mikhonin, S. V. Bykov, N. S. Myshakina, and S. A. Asher, J. Phys. Chem. B **110**, 1928 (2006).

<sup>&</sup>lt;sup>1</sup>T. Aida, E. W. Meijer, and S. I. Stupp, Science 335, 813 (2012).

- <sup>18</sup>N. S. Myshakina, Z. Ahmed, and S. A. Asher, J. Phys. Chem. B 112, 11873 (2008).
- <sup>19</sup>M. C. Chen and R. C. Lord, J. Am. Chem. Soc. **96**, 4750 (1974).
- <sup>20</sup>W. H. Moore and S. Krimm, Biopolymers **15**, 2439 (1976).
- <sup>21</sup>W. H. Moore and S. Krimm, Biopolymers **15**, 2465 (1976).
- <sup>22</sup>J. F. Rabolt, W. H. Moore, and S. Krimm, Macromolecules 10, 1065 (1977).
- <sup>23</sup>C. A. Hunter and J. K. M. Sanders, J. Am. Chem. Soc. **112**, 5525 (1990).
- <sup>(1990).</sup>
  <sup>24</sup>M. O. Sinnokrot and C. D. Sherrill, J. Phys. Chem. A 108, 10200 (2004).
- <sup>25</sup>G. B. McGaughey, M. Gagne, and A. K. Rappe, J. Biol. Chem. 273, 15458 (1998).
- <sup>26</sup>C. Guo, Y. Luo, R. Zhou, and G. Wei, ACS Nano 6, 3907 (2012).