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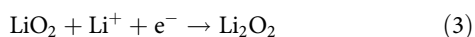
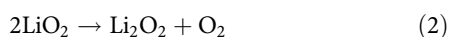
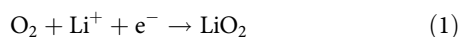
Kinetics of lithium peroxide oxidation by redox mediators and consequences for the lithium–oxygen cell

Yuhui Chen¹, Xiangwen Gao¹, Lee R. Johnson ² & Peter G. Bruce ¹

Lithium–oxygen cells, in which lithium peroxide forms in solution rather than on the electrode surface, can sustain relatively high cycling rates but require redox mediators to charge. The mediators are oxidised at the electrode surface and then oxidise lithium peroxide stored in the cathode. The kinetics of lithium peroxide oxidation has received almost no attention and yet is crucial for the operation of the lithium–oxygen cell. It is essential that the molecules oxidise lithium peroxide sufficiently rapidly to sustain fast charging. Here, we investigate the kinetics of lithium peroxide oxidation by several different classes of redox mediators. We show that the reaction is not a simple outer-sphere electron transfer and that the steric structure of the mediator molecule plays an important role. The fastest mediator studied could sustain a charging current of up to 1.9 A cm^{-2} , based on a model for a porous electrode described here.

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The rechargeable aprotic lithium–O₂ (air) battery operates by the reduction of O₂ at the positive electrode forming Li₂O₂ on discharge, with oxidation of Li₂O₂ taking place on charge^{1–10}. Li₂O₂ is an insulating and insoluble solid^{11–16}. Ether-based electrolytes, such as dimethoxyethane (DME) and tetraethylene glycol dimethyl ether (tetraglyme), have been used as the basis of electrolyte solutions in most Li–O₂ cells, because of their relative stability towards reduced oxygen species. However, they cannot dissolve LiO₂, the intermediate in the reduction of O₂ to Li₂O₂,



resulting in LiO₂ being adsorbed on the electrode surface, and resulting in the growth of Li₂O₂ films on the electrode, leading to low rates, low capacities and early cell death^{14,17}. The problem is exacerbated by the formation of Li₂CO₃ between Li₂O₂ and carbon, the latter is usually employed as the material for the porous positive electrode¹⁸. Use of redox mediators (RMs) on discharge, such as 2,5-di-tert-butyl-1,4-benzoquinone (DBBQ), which are reduced at the electrode surface on discharge and then go on to reduce O₂ to Li₂O₂ in solution, can help to mitigate these problems, but result in the formation of Li₂O₂ disconnected from the electrode surface and therefore electronically isolated during charging¹⁹. This introduces the need for a redox mediator to be employed on charging that can oxidise Li₂O₂^{20–33}. Such mediators are molecules capable of oxidation at the surface of the pores in the porous positive electrode on charging and then transfer of holes to the electronically isolated Li₂O₂ particles within the pores. As a result, Li₂O₂ is oxidised and O₂ released, the mediator molecule being reduced in the process and returning to the electrode surface for the cycle to be repeated.

Suitable oxidation mediators must have a redox potential above that for O₂/Li₂O₂: 2.96 V vs. Li⁺/Li, a sufficiently high heterogeneous rate constant for electron transfer at the electrode surface to support the required charging rates, a highly reversible redox process such that the cycle may be carried out many times, and of course not only be capable of oxidising Li₂O₂, but with a sufficiently high rate to sustain the required charging current³⁴. Stability of the mediators, especially on long-term cycling, is also an important challenge and recent work has considered the design of more stable redox mediators for cycling²²; however, very little is known about the factors affecting the reaction between oxidation mediators and Li₂O₂. It is often assumed that a mediator with a high redox potential has fast kinetics for the oxidation of Li₂O₂, but this is not necessarily so³³. Alternatively, the kinetics of Li₂O₂ oxidation by a mediator could be linked to the kinetics of its own redox process, but this would only be the case if both were outer-sphere electron transfer processes. Importantly, little experimental evidence exists about the kinetics of Li₂O₂ oxidation by redox mediators, yet their use and such kinetics are crucial to the operation of the Li–O₂ cell.

Here, we investigate the kinetics of Li₂O₂ oxidation by several classes of redox mediators, which differs in the *E*⁰ (standard redox potential) and *k*⁰ (standard heterogeneous electron transfer rate constant) values, to ascertain the factors that control the rate of Li₂O₂ oxidation by the mediators.

Results

Apparent rate constants (*k*_{app}) of mediators. Apparent rate constants (*k*_{app}) for Li₂O₂ oxidation by the redox mediators were

determined using scanning electrochemical microscopy (SECM). Details of the cell and procedures used are given in the Methods section. In brief, SECM feedback approach curves at a Li₂O₂ disk, composed of a pressed pellet of commercial Li₂O₂ with a diameter of 12 mm, were recorded and apparent rate constants, *k*_{app}, for Li₂O₂ oxidation were obtained by fitting to the theoretical feedback approach curves developed by Cornut et al.^{35–37}. When recording an approach curve, the SECM tip was held at a sufficiently positive potential such that a steady-state current was obtained for the oxidation of the redox mediator. The tip approaches the Li₂O₂ disk and at small separation distances, the mediator oxidised at the tip diffuses to the Li₂O₂ disk where it oxidises Li₂O₂, regenerating itself and contributing to a feedback loop, while concurrently, diffusion of the mediator to the tip is blocked by the surface. The balance of the two alter the current at the SECM tip, *i*_T, and the faster the kinetics of Li₂O₂ oxidation by the mediator the greater the current, see Supplementary Figure 1. As we do not know the mechanism by which the mediators oxidise the lithium peroxides surface, we can only obtain an apparent rate constant (*k*_{app}) based on the feedback response; however, this provides a comparison between the different mediators and indicates the overall rate capability.

Fig. 1 shows the oxidation mediators studied. They are in three classes, amines, nitroxy and thiol compounds, chosen because they are classes of compounds known to exhibit reversible redox processes and include several of the compounds that have been used as oxidation mediators in Li–O₂ cells, such as tris[4-(diethylamino)phenyl]amine (TDPA), 2,2,6,6-tetramethyl-1-piperidinyloxy (TEMPO) and 10-methylphenothiazine (MPT)^{20–22}. *k*_{app} for Li₂O₂ oxidation by the mediators are presented in Supplementary Table 1. The standard redox potential, *E*⁰, and standard heterogeneous electron transfer rate constant, *k*⁰, were measured for each mediator using cyclic voltammetry, as described in the Methods section. The diffusion coefficients, *D*, were obtained from the steady-state current at an ultramicroelectrode (UME), also as described in the Methods section. The values for each of the three parameters are also given in Supplementary Table 1. Three additional mediators, tetrathiafulvalene (TTF), ferrocene (FC) and 5,10-dimethylphenazine (DMPZ), which do not belong to the above three classes, but have been commonly used as oxidation mediators, were also studied and are listed in Supplementary Table 1^{23,24,38}. The standard redox potentials are all positive for the O₂/Li₂O₂ reaction. The diffusion coefficients vary by no more than a factor of 3. The *k*⁰ for the mediators themselves are all relatively high, ranging from 0.007 to 0.078 cm² s^{−1}, sufficiently so to support an areal current density over 200 mA cm^{−2} at an overpotential of 60 mV, based on the true surface area of the pores and therefore more than sufficient to sustain an areal current density suitable for a Li–O₂ cell. Of course, this does not take into account the kinetics of Li₂O₂ oxidation required to sustain the current, which will be discussed below after the presentation of the rates of Li₂O₂ oxidation. The assumptions regarding the porous cathode structure and the approach used to make this estimate are described in the Supplementary Note.

Before considering the kinetics of the mediator oxidation in more detail, we first determine the surface composition of the disk and the possibility of passivation with, for example, Li₂CO₃. A disk of Li₂O₂ was immersed in 1 M LiTFSI in tetraglyme for 3 h and then examined by time of flight secondary ion mass spectrometry (TOF-SIMS), alongside a disk that was not exposed to the electrolyte solution. As shown in Fig. 2, for both disks, the major peaks are from Li₂O₂⁺, with the secondary peaks being ascribed to Li₂CO₃. These results show that although there is some Li₂CO₃, even on the surface of the pristine disk, a significant proportion of the surface remains as Li₂O₂ even after 3 h of

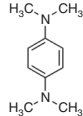
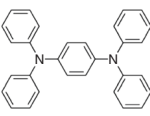
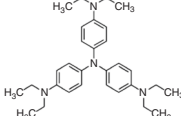

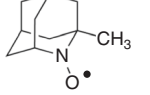
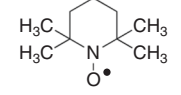
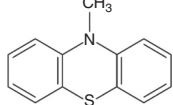
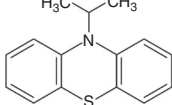
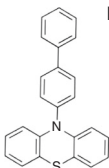
	<p>TMPD</p> 		<p>DPAB</p> 		<p>TDPA</p> 
k_{app} ($\times 10^{-3}$ cm s $^{-1}$)	0.30	>	0.13	>	0.025
	<p>AZO</p> 		<p>MAZO</p> 		<p>TEMPO</p> 
k_{app} ($\times 10^{-3}$ cm s $^{-1}$)	7.9	>	6.7	>	3.6
	<p>MPT</p> 		<p>PPT</p> 		<p>BPPT</p> 
k_{app} ($\times 10^{-3}$ cm s $^{-1}$)	2.36	>	1.82	>	0.82

Fig. 1 Structures of the oxidation mediators and their kinetics of Li_2O_2 oxidation. Comparison of the apparent rate constants (k_{app}) for the reaction between the redox mediators and Li_2O_2 grouped by structure

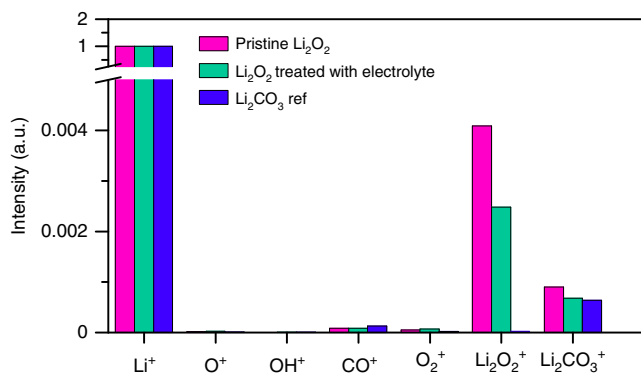


Fig. 2 TOF-SIMS of Li_2O_2 disks before and after treating with electrolyte of 1M LiTFSI in tetraglyme. Both Li_2O_2 disks show signal of Li_2O_2^+ and Li_2CO_3^+ whereas the Li_2CO_3 disk shows little signal of Li_2O_2^+ , confirming the presence of Li_2O_2 on the surface of disk after treating with the electrolyte

exposure to the electrolyte, confirming that the disk is suitable for the SECM measurements. Note that the sensitivity of TOF-SIMS to different species varies, consequently it is not possible to quantify the relative amounts of Li_2O_2 and Li_2CO_3 by simply comparing the areas under the peaks. Instead, the disk was etched until the signal from Li_2O_2 was constant, therefore corresponding to the bulk peroxide, i.e., 100% Li_2O_2 . Comparing this signal with that for Li_2O_2 at the surface indicated that approximately 35% of the disk surface was Li_2O_2 .

A disk of Li_2CO_3 was investigated with SECM using TEMPO as the oxidation mediator, as it has a sufficiently high potential to oxidise Li_2CO_3 and shows fast kinetics with the Li_2O_2 disk. The

results are shown in Supplementary Figure 2. The k_{app} for oxidation of Li_2CO_3 by TEMPO is four orders of magnitude lower than the data collected on the Li_2O_2 disk, indicating that even for mediators with sufficiently high potentials the contribution of Li_2CO_3 oxidation to the k_{app} is very small. The dominant reaction for the range of mediators studied here, even taking account of partial coverage by Li_2CO_3 , is oxidation of Li_2O_2 .

It has been reported previously by us and by others that several of the redox mediators used in $\text{Li}-\text{O}_2$ cells to date exhibit some degree of decomposition^{24,39–41}. Assembling a cell with commercial Li_2O_2 and the oxidation mediators TTF and AZO, and then charging to a capacity of ~ 1 mAh results in notable decomposition of TTF and AZO as seen by ^1H NMR of the electrolyte, see Supplementary Figure 3. In the SECM experiments, only a small amount of charge, ~ 1 nAh, is passed, therefore the fraction of mediator that is decomposed is negligible.

Inner-sphere process for mediator oxidising Li_2O_2 . To explore the possible correlations between k_{app} and the electrochemical parameters of the redox mediators, E^0 and k^0 , plots of k_{app} vs. k^0 and E^0 and are presented in Figs. 3 and 4, respectively. There is no apparent dependence of k_{app} on k^0 , Fig. 3. The values of k^0 for the different redox mediators appear independent of the nature of the electrode used to measure them, as demonstrated by measuring these values at Au and glassy carbon electrodes, see Supplementary Figure 4 and Methods section, consistent with the RM^+/RM reactions occurring by outer-sphere electron transfer. If the oxidation of Li_2O_2 was also an outer-sphere electron transfer reaction, then k_{app} would be proportional to k^0 of the redox mediator (and hence the reorganisation energy of the RM and surrounding solution), or the rate of the reaction $\text{Li}_2\text{O}_2 \rightarrow \text{Li}_2\text{O}_2^+ + e^-$. Since there is no dependence of k_{app} on k^0 , the former

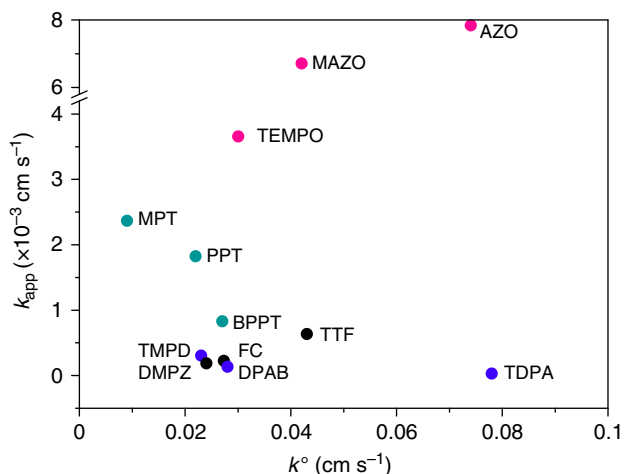


Fig. 3 Dependence of the apparent rate constant, k_{app} , on the heterogeneous electron transfer rate constant, k^0 , of the mediators. Amines, nitroxy and thiol compounds are marked in blue, red and green

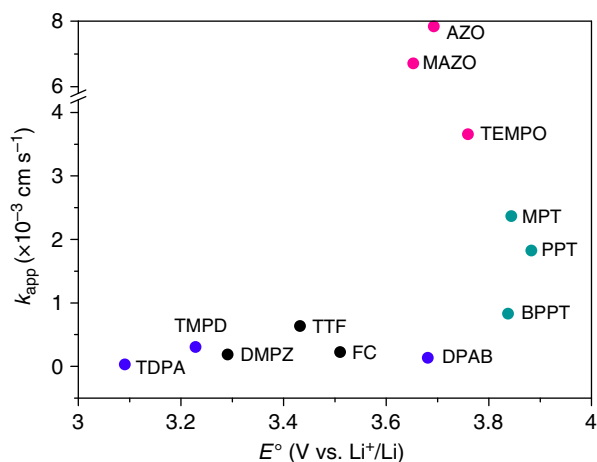


Fig. 4 Dependence of the apparent rate constant, k_{app} , on the redox potential, E^0 , of the mediators. Amines, nitroxy and thiol compounds are marked in blue, red and green

cannot be true. If the rate was limited by the electron transfer kinetics associated with the Li_2O_2 , then k_{app} would be invariant, which again is not the case. We conclude that oxidation of Li_2O_2 by the redox mediators is mainly an inner-sphere process, i.e., involves adsorption of the mediator on the peroxide surface. The values for k_{app} are one order of magnitude smaller than the corresponding k^0 values, indicating that the reaction of mediators oxidising Li_2O_2 is most likely to be the rate determining step of the entire charge process. This will particularly be true towards the end of charge when the surface area of the remaining Li_2O_2 is low. We estimate that a k_{app} from 2.5×10^{-5} to $7.9 \times 10^{-3} \text{ cm s}^{-1}$ in a $\text{Li}-\text{O}_2$ cell with a porous cathode filled with Li_2O_2 would provide an areal current density of 108 mA cm^{-2} to 1.9 A cm^{-2} using the same model for the porous cathode as above. The details are described in the Supplementary Note and Supplementary Figure 5. Although we note that this equivalent charging current varies with consumption of Li_2O_2 , it is sufficient to sustain the charging process, even for some of the slowest oxidation mediators investigated here.

Turning to the plot of k_{app} vs. E^0 , Fig. 4, it appears that the highest rates are observed for mediators with potentials above $\sim 3.6 \text{ V}$. However, potential per se is not the explanation for the high rate, as there are examples of mediators with a high potential but low rate, e.g., BPPT. From the experiment on the Li_2CO_3 disk using TEMPO, we know higher rates at high potentials are not due to the onset of Li_2CO_3 oxidation contributing to the overall surface oxidation kinetics. Different crystal facets of Li_2O_2 will have different oxidation potentials⁴². Mediators operating at higher potentials could oxidise these higher potential facets and hence access a greater Li_2O_2 surface area. However, the fact that the rates vary for different mediators above 3.6 V and several high potential mediators have relatively low k_{app} suggests that this alone cannot be the reason for high rate mediators having a relatively high potential. As discussed below, we believe an important factor controlling the rate of the mediators is the nature of the oxidising centre and the degree of its steric hindrance.

Considering the molecules presented in Fig. 1 and the k_{app} values shown in the figure, it is evident that the nitroxy radicals exhibit the fastest rates of Li_2O_2 oxidation. The thiol group also provides a high rate, in contrast to the amines that are all low rate. The chemistry of the redox centre appears to be an important factor for controlling the rate of oxidation, probably due to the interaction with Li_2O_2 surface. The oxidation rates decrease when the redox centre of the molecule is surrounded by bulky groups, Fig. 1. This suggests that a key factor influencing the kinetics of Li_2O_2 oxidation is the steric hindrance as the molecule approaches the surface of Li_2O_2 . The fastest kinetics is exhibited by 2-azaadamantane-*N*-oxyl (AZO), $7.9 \times 10^{-3} \text{ cm s}^{-1}$, which has the most exposed redox centre of all the redox mediators studied here. This observation is in accord with the lack of evidence for an outer-sphere reaction and provides direct evidence for Li_2O_2 oxidation proceeding by an inner-sphere mechanism.

Discussion

In conclusion, we have measured the rate constants for the oxidation of Li_2O_2 particles by a series of molecular mediators spanning standard redox potentials, E^0 from 3.1 to 3.9 V and standard heterogeneous rate constants for electron transfer, k^0 from 0.007 to $0.078 \times 10^{-3} \text{ cm s}^{-1}$. The surface of Li_2O_2 particles in a typical electrolyte solution, LiTFSI in tetraglyme, is partially covered by Li_2CO_3 , but the rate of Li_2CO_3 oxidation, a mediator that operates at 3.8 V, TEMPO, is four orders of magnitude lower than for Li_2O_2 , therefore Li_2O_2 oxidation dominates. There is no correlation between the variation of k^0 , the standard heterogeneous rate constant at the electrode surface for the mediators, and the rate of Li_2O_2 oxidation by the mediators, indicative of this not being an outer-sphere electron transfer process at the Li_2O_2 surface. There is evidence of Li_2O_2 oxidation rates depending on the nature of the oxidising molecule. Nitroxy radicals, especially those with low steric hindrances of access to the Li_2O_2 surface, exhibit the highest rates. Nevertheless, the mechanism of Li_2O_2 oxidation by molecular oxidants is still not well understood, and such understanding will be important in order to inform the design of optimised oxidation mediators. All mediators studied display kinetics sufficient to enable relatively high rates within a battery, charging current density exceeding 100 mA cm^{-2} . A mediator with a k_{app} of $7.9 \times 10^{-3} \text{ cm s}^{-1}$ can sustain an areal current density of up to 1.9 A cm^{-2} , based on the same model. It is important to note that stability is still a challenge for the $\text{Li}-\text{O}_2$ battery and here we observe significant mediator decomposition when passing large amounts of charge. More stable electrolytes and mediators are required to minimise side reactions and hence improve cycleability.

Methods

Materials preparation. Li_2O_2 and Li_2CO_3 disks were obtained by pressing Li_2O_2 powder (Aldrich) and Li_2CO_3 powder (Aldrich) with a die set in an Ar-filled glove box. Disks of 13 mm diameter and ~1 mm of thickness were prepared and served as substrate. A Au microelectrode (diam. 25 μm , CHI) served as an SECM probe tip. Prior to measurement, the Au tip was polished with a microelectrode beveller (Sutter) and checked with a microscope. A silver wire reference electrode (RE) and a platinum counter electrode (CE) were used. 2,2,6,6-tetramethyl-1-piperidinyloxy (TEMPO), 2-azaadamantane-*N*-oxyl (AZO), 1-methyl-2-azaadamantane-*N*-oxyl (MAZO), tris[4-(diethylamino)phenyl]amine (TDPA), 1,4-bis(diphenylamino)benzene (DPAB), *N,N,N',N'*-tetramethyl-*p*-phenylenediamine (TMPD), 10-methylphenothiazine (MPT), 10-isopropylphenothiazine (PPT), 10-(4-biphenyl)phenothiazine (BPPT), tetrathiafulvalene (TTF), ferrocene (FC) and 5,10-dimethylphenazine (DMPZ) are from Aldrich. 10 mM redox mediators are dissolved in 100 mM LiTFSI-tetraglyme electrolyte for electrolyte solution.

A Swagelok cell was assembled as reported previously⁴⁰, using a piece of gas diffusion layer electrode (GDL) as the positive electrode. A lithium super ionic conductor disc (LiSICON, Ohara) was used to protect Li metal as the negative electrode. A Li_2O_2 disk was placed between the GDL and the LiSICON essentially placing the cell in a discharged state. TTF and AZO were chosen as the oxidation mediators. The cell was charged by holding at 3.4 V for TTF and 3.7 V for AZO until 1 mAh charge passed prior to further chemical characterisations. For NMR analysis, the electrodes and separators were rinsed with 0.7 ml of CDCl_3 , and measurements were recorded on a Bruker spectrometer (400 MHz).

Electrochemical measurements. SECM experiments were performed with SECM bipotentiostat (CHI 920) in an Ar-filled glovebox. Prior to kinetics measurement, the NG factor of Au tip was determined by approaching a completely insulating surface and fitting the negative approach curve. The data processing and fitting process were described elsewhere^{35–37}. A dimensionless rate constant, κ , was obtained by data fit, which equals to $k_{\text{app}} r_0/D$, where r_0 is the radius of tip and D is the diffusion coefficient of redox mediators. D of various mediators were determined by measure steady-state current of a Au microelectrode with known radius r_0 , according to $i_{\text{ss}}=4nFDr_0C$.

The redox potential and heterogeneous electron transfer rate constants k^0 of redox mediators itself were determined using cyclic voltammetry (CV) measurements. The redox potential is determined by the centre of two redox peaks, which is measured in a 100-mM LiTFSI-tetraglyme solution with 10 mM of various mediators at a Au electrode. Partially charged LiFeO_4 (LFP) protected by a glass frit served as an RE and it gave a potential of 3.45 V vs. Li^+/Li as reported previously. A platinum wire served as a CE. The details of k^0 measurement are described elsewhere⁴³. Briefly, CVs were recorded at various scan rates, ranging from 0.05 to 10 V s^{-1} . ψ , a function of CV peaks separation, was plotted vs. root of scan rate and a linear fit was applied. k^0 was obtained from the slope of linear fit. The k^0 measurement was carried out at both Au and glassy carbon (GC) WEs. Due to the non-negligible resistance of ether-based electrolytes, an Ohmic overpotential correction was applied to account for the uncompensated resistance during CV measurements and a silver wire RE was used.

Characterisations. For the surface characterisations, the Li_2O_2 disk was immersed in 1 M LiTFSI-tetraglyme solution for 3 h prior to XPS and TOF-SIMS experiments. Both pristine disk and treated disk were characterised in an air-sensitive holder. To measure the TOF-SIMS of bulk Li_2O_2 , the data were recorded after 2 min etching.

Data availability. The data that support the findings of this study are available from the corresponding author upon reasonable request. Background data has been deposited in the Oxford Research Archive (ORA) at: <https://ora.ox.ac.uk/objects/uid:c23a0cc0-55b5-455f-bb68-a14d8ea2e3bc>.

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Author contributions

Y.C., X.G. and L.R.J. designed the experiments and analysed the data. Y.C. and X.G. performed the electrochemical measurements and characterizations. Y.C., X.G., L.R.J. and P.G.B. interpreted the data. P.G.B. wrote the paper.

Additional information

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