Electronic structure and bonding in skutterudite-type phosphides

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(Received 13 October 1995)

The electronic structures of the skutterudite-type phosphides CoP_3 and NiP_3 have been investigated by means of first-principles linear muffin-tin orbital–atomic sphere approximation band-structure calculations. The presence of P_4 rings in the skutterudite structure is of great importance in determining the nature of the electronic bands around the Fermi level, composed mainly of π -type molecular orbitals of these units. The metallic character found for NiP_3 should be ascribed to the phosphorus framework rather than to the metal atoms.

The skutterudite (i.e., $CoAs₃$ -type) structure is found for a number of binary transition-metal pnictides MX_3 (Ref. 1), with interesting electronic properties that suggest their use as advanced thermoelectric materials.^{2,3} Except for NiP₃, all other known compounds in this family $(MX_3$ with $M=Co$, Rh, Ir and $X = P$, As, Sb) are isoelectronic and have been found to behave as diamagnetic p -type semiconductors.³⁻¹³ While some of them seem to have substantial optical band gaps (0.45 and 1.4 eV for CoP₃ and IrSb₃, respectively),^{3,7} no optical gap is found for $CoAs₃$ and $CoSb₃$.⁷ NiP₃, the only known skutterudite^{14,15} of a group-10 metal, shows metallic behavior with Pauli-type paramagnetism. 13

Closely related crystal structures are found for the ternary compounds RM_4X_{12} (R =rare-earth metal, M =Fe, Ru, Os, and *X*=P, As, Sb). In these, the M_4X_{12} subnet adopts the skutterudite structure, $16-19$ while the rare-earth metal atoms occupy the cubic positions that are empty in the binary phases. The physical properties of the ternary phosphides $\text{LaM}_4\text{P}_{12}$ are specially interesting, being superconductors with transition temperatures $T_c=4.1$, 7.2, and 1.8 K for $M = Fe$, Ru, and Os,²⁰ respectively.

Although these compounds show interesting physical properties, relatively little is known about their electronic structure, probably because of the complex crystal structure with a relatively open unit cell. So far, only two bandstructure calculations have been reported for the skutteruditetype compounds. In the first study, 21 the electronic structures of $LaFe₄P₁₂$ and related compounds were analyzed by using the extended Hückel tight-binding (EHTB) method. Recently, self-consistent band-structure calculations, using the linear augmented plane-wave method for IrSb_3 , CoAs₃, and $CoSb₃$, have been published by Singh and Pickett.²²

In this contribution, we report first-principles bandstructure calculations using the linear muffin-tin orbital method in the atomic sphere approximation $(LMTO-ASA)$ (Refs. 23–25) to explore the band structure of CoP_3 and $NiP₃$.

CRYSTAL STRUCTURE

The nature of the highest occupied bands, responsible for the electronic properties of a material, is strongly dependent on its crystal structure. A detailed knowledge of the crystal structure is, therefore, indispensable before analyzing the details of the electronic structure. In the case of the skutterudite tails of the electronic structure. In the case of the skutterudite
structure (space-group $Im\overline{3}$, n. 204),²⁶ the conventional body-centered-cubic unit cell contains eight MX_3 units. Metal atoms are located on 8*c* sites, while the nonmetal ones occupy 24*g* sites. The latter positions depend on two parameters *y* and *z* that, together with the lattice parameter *a*, completely define the crystal. $27-30$ A simple way to describe the skutterudite structure is to deduce it as a distortion of the more symmetric ReO_3 structure, 31,32 formed by a simple cubic array of metal atoms octahedrally surrounded by six oxygen atoms. This structure can also be considered as an array of vertex-sharing octahedra [Fig. $1(a)$].

The characteristic X_4 rings in skutterudites can be obtained by displacing four of the nonmetal atoms located on parallel edges of a metal cube to its center, as shown in Fig. 1(b). Since each cube has 12 nonmetal atoms on its edges, the displacement must be done simultaneously in eight neighboring cubes. The new unit cell is thus eight times larger than the original ReO_3 unit cell. Since there are not enough nonmetal atoms to make X_4 rings at the center of each cube, the distortion leaves one out of each four metal cubes empty, as shown in Fig. $2(a)$. In the resulting skutterudite structure, each metal atom is in an octahedral environ-

FIG. 1. ReO_3 crystal structure (a) and distortion that leads to the skutterudite structure (b). Black and gray balls represent O and Re atoms, respectively.

FIG. 2. Skutterudite structure: Unit cell (a), coordination of the metal atoms and array of vertex-sharing octahedra (b), and arrangement of the X_4 rings in chains (c). Black and gray balls represent nonmetal and metal atoms, respectively. In (c), three mutually orthogonal chains have been highlighted for the sake of clarity.

ment [Fig. 2(b)] and each MX_6 octahedron is sharing vertices with six neighboring octahedra as in ReO_3 . The main differences between both structures are that, in the case of skutterudite, the MX_6 octahedra are distorted (local D_{3d} symmetry) with their relative orientations in the threedimensional array tilted. The X_4 rings in the skutterudite structure are arranged in mutually orthogonal linear arrays that run parallel to each crystallographic direction [Fig. $2(c)$].

COMPUTATIONAL METHODOLOGY

Band-structure calculations for CoP_3 and NiP_3 have been obtained by using the local-density-functional approximation $33,34$ and the scalar-relativistic linear muffin-tin orbital method in the atomic spheres approximation $(LMTO-ASA).^{23-25}$ In this method, the one-electron potential entering the Schrödinger equation is a superposition of overlapping spherical potential wells centered at the atomic positions. A spherical charge density inside the atomic spheres is also assumed. These approximations are specially well suited for highly compact crystal structures. In the case of CoP_3 and NiP_3 with the relatively open skutterudite structure, the use of only atom-centered spheres would result in substantial errors. To avoid this problem, it has been necessary to introduce interstitial spheres in order to choose sufficiently small radii for the spheres that result in small overlaps between them and at the same time fulfill the usual condition of maximizing the volume of the crystal occupied by the atomic spheres. The location and size of the atomic

TABLE I. Geometrical parameters used for the LMTO-ASAcalculations on CoP_3 and NiP_3 . S_R indicates the radius chosen for each atomic sphere. *E* represent empty spheres.

Class	$S_R(\AA)$	\boldsymbol{x}	y	Z.
Co, Ni $(8c)$	1.2748	0.2500	0.2500	0.2500
P(24g)	1.2954	0.0000	0.3482	0.1453
$E_1(2a)$	2.0686	0.0000	0.0000	0.0000
$E_2(12e)$	0.9938	0.0000	0.5000	0.3524
$E_3(24g)$	0.7714	0.0000	0.1799	0.3046

and interstitial spheres has been determined, using the superposition of Hartree atomic potentials.³⁴ Application of this procedure to CoP_3 and NiP_3 results in the sphere positions and radii shown in Table I. Together with the real atoms, our calculations included also 19 empty spheres on each primitive unit cell that, according to symmetry, belong to three nonequivalent classes. The same internal parameters and sphere radii have been used for both compounds. For the cell parameter a , the experimentally determined value (7.073 Å) for CoP_3 and 7.8192 Å for NiP₃) has been used in each case.

All calculations were performed within the tight-binding ~TB! minimal basis set representation of the LMTO-ASA

FIG. 3. LMTO-ASA band structures for CoP_3 (a) and NiP₃ (b). The horizontal dashed line indicates the Fermi level.

FIG. 4. Total and projected LMTO-ASA electronic density of states (in states/eV cell) for CoP₃ (a) and NiP₃ (b). The dashed line indicates the Fermi level.

method.²⁵ The structure constants were first generated for the short-range TB basis set, but afterwards the transformation to the standard orthonormal basis set has been performed. The down-folding technique²⁵ based on Löwdin's perturbation theory³⁵ has been applied to this latter set. Integrations in k space were performed using the tetrahedron method $36,37$ with a mesh of 512 points in the irreducible wedge of the first Brillouin zone.

Qualitative features of the electronic structure and bonding in these compounds have been analyzed by using tightbinding band-structure calculations within the extended Hückel framework (EHTB).³⁸⁻⁴⁰ The parameters used in these calculations are indicated in Ref. 41.

RESULTS AND DISCUSSION

The calculated $(LMTO-ASA)$ band structures of $CoP₃$ and $NiP₃$ are shown in Figs. 3(a) and 3(b), respectively. The general trends observed in both band-structure diagrams are qualitatively similar to those found by Singh and Pickett²² for IrSb₃, CoSb₃, and CoAs₃. The most important feature in the dispersion diagrams is the more or less well-defined pseudogap that appears just below the Fermi level (see Fig. 4). In CoP_3 , this pseudogap is crossed by a totally occupied single band that almost reaches the conduction band at the center of the first Brillouin zone. The calculated width of the indirect pseudogap is 1.26 eV, that of the indirect band gap (from Γ to a point on the Γ -*H* line) is 0.07 eV and that of the direct gap at Γ , 0.28 eV. Our calculated direct band gap is somewhat smaller than the value given by Ackermann and Wold^{\prime} for the optical gap in CoP₃ (0.45 eV), probably due to the well-known tendency of the local-density-approximationbased methods to underestimate this property. As already pointed by Singh and Pickett²² for IrSb₃, CoSb₃, the dispersion of the band in the pseudogap region of $CoP₃$ is remarkable: although it has a parabolic shape in a small region around the Γ point, it rapidly becomes linear in energy when leaving the center of the Brillouin zone. This effect is most evident in the Γ -*H* direction, where the crossover between parabolic and linear dispersion occurs for wave vectors of about 5% the distance to the zone boundary. The transport properties of the hole-doped material should thus be modified by this linear dispersion from those expected for standard semiconductor behavior.

The dispersion diagram for NiP_3 , although similar in its rough features to that found for CoP_3 , shows some distinct features. The larger dispersion of the bands reduces the pseudogap region to approximately 0.57 eV. In this case, the band in the pseudogap region penetrates the bottom of the conduction band. In this compound, the metal atoms provide one more electron per formula unit than in CoP_3 . These electrons occupy the bottom of the conduction band and are responsible for the metallic behavior of $NiP₃$.

Figures $4(a)$ and $4(b)$ show the calculated total and projected densities of states (DOS) for both compounds. The pseudogap region with low values for the DOS is clearly visible just below the Fermi level. The phosphorus projection can be roughly divided in three separate regions: the first one ranging from approximately -15 to -9 eV, the second one from -8 eV to the lower edge of the pseudogap, and the last one starting from the upper edge of the pseudogap. Since these results agree in their main features qualitatively well with those obtained from EHTB calculations (not shown in the figures), this method will be used as a qualitative tool in the analysis of bonding in these materials.

Our EHTB calculations indicate that the lowest region is due basically to the σ -bonding orbitals of the P₄ rings with large phosphorus 3*s* contributions. The second region is formed mainly by both the nonbonding and the bonding π -type molecular orbitals of the P₄ rings. The third region is due mainly to ring antibonding molecular orbitals, both of π -type (lowest part) and σ -type (region above 5 eV). It is also very important to note that a large contribution of the nonbonding and π -type bonding orbitals that have the proper spatial orientation to interact strongly with the metal *d* orbitals is found at the lowest part of this third region. EHTB calculations show for these states an important positive (bonding) overlap population for the P-P bonds.

The projection of the DOS for the metal atoms shows that for both CoP_3 and NiP_3 the main peak is found in the region directly below the pseudogap. According to these results, the *d* orbitals of the metal atoms seem to be almost completely filled. This indicates that bonding should be more covalent or metallic than ionic in agreement with the arguments that have been given by some authors based on different experimental evidences. $5,10,42-45$ The most relevant conclusion that can be drawn from these facts is that the bands responsible for the electrical conduction in $NiP₃$ are mainly centered on the phosphorous sublattice. This is clearly visible in our LMTO-ASA calculations, where, the contributions of phosphorous and nickel states to the DOS at the Fermi level are equal to 51.7 and 13.7 states/eV per unit cell, respectively. A more detailed picture of the nature of these states shows that they are mostly formed by nonbonding and π -type bonding orbitals of the P_4 rings. Coupling of these orbitals to neighboring rings in all three crystallographic directions is achieved through interaction with the *s* and *p* orbitals of the metal atoms.²¹

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FIG. 5. Phosphorous contribution (a) and interaction between phosphorous and metal p orbitals (b) in the highest occupied band for CoP_3 .

For CoP_3 , it is also interesting to describe the nature of the single band in the pseudogap region. Our calculations confirm the result obtained earlier by other authors, 21 indicating that it is basically formed by the π -type antibonding orbital of the P_4 rings shown in Fig. 5(a). The important dispersion of this band is due to mixing of these orbitals belonging to neighboring rings with the metal p orbitals $[Fig.$ $5(b)$.

CONCLUDING REMARKS

First-principles band-structure calculations for CoP_3 and $NiP₃$ show that both materials have similar electronic structures with predominantly covalent or metallic bonding. As a consequence, the highest occupied bands are mainly composed of phosphorous-centered orbitals. The electrical properties of phosphorous-rich late transition metal phosphides seem thus to be determined principally by the phosphorous sublattice rather than by the metallic atoms present in the structure. This finding is specially surprising for $NiP₃$, since it has been experimentally found to be a metallic conductor with Pauli paramagnetism. For d^6 metal skutterudites like CoP_3 , electrical properties should be determined by the existence of a pseudogap separating the valence and conduction bands. The low density of states in this pseudogap region is provided by a single occupied band that exhibits quasilinear dispersion relations in large parts of the symmetry lines joining Γ and the zone boundaries. To confirm whether the highest occupied bands originate largely from the phosphorous or the metal sublattices, it would be interesting to perform a spectroscopic investigation of the valence-band region for this family of compounds.

ACKNOWLEDGMENTS

The stay of V.Z. in Barcelona was made possible by a sabbatical grant of the Ministerio de Educación y Ciencia (Spain). M.L. thanks CIRIT (Generalitat de Catalunya) for a doctoral grant. The authors are grateful to U. Schönberger for many helpful comments on this work. Financial support was provided by DGICYT through Grant No. PB92-0655.

Metals, Metals Park, OH, 1986), Vol. 1.

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