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Infrared absorption of closely-aligned heterostructures of monolayer and bilayer graphene with hexagonal boron nitride

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We model optical absorption of monolayer and bilayer graphene on hexagonal boron nitride for the case of closely-aligned crystal lattices. We show that perturbations with different spatial symmetry can lead to similar absorption spectra. We suggest that a study of the absorption spectra as a function of the doping for almost completely full first miniband is necessary to extract meaningful information about the moiré characteristics from optical absorption measurements and to distinguish between various theoretical proposals for the physically realistic interaction. Also, for bilayer graphene, the ability to compare spectra for the opposite signs of electric-field-induced interlayer asymmetry might provide additional information about the moiré parameters.

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Graphene and hexagonal boron nitride (h-BN) are two representatives of a new class of materials, the twodimensional atomic crystals.^{1,2} Both are made of atoms arranged in a single layer of hexagons. However, while graphene, made of carbon, is a gapless semiconductor with a linear dispersion relation for electrons in the vicinity of the two inequivalent corners of the hexagonal Brillouin zone,³ h-BN is an insulator with a band gap of 6eV.^4 It is now possible to place graphene flakes on *h*-BN substrates⁵ or build stacks with alternating graphene and h-BN crystals.⁶ Because of the small difference in lattice constants, heterostructures with closely-aligned crystal structures of the two materials, form a moiré pattern in STM images⁷ and the dispersion of graphene electrons is reconstructed into minibands as a result of the periodic moiré perturbation. $^{8-13}$

theoretical¹⁴⁻²² Despite considerable and $experimental^{23,24}$ effort, little is known about the actual van der Waals interaction between graphene and h-BN. While considering only the inversion-symmetric scalar potential was enough to explain the qualitative features of the initial STM measurements,⁸ a more complex model involving three inversion-symmetric terms¹² was used to model some of the transport and capacitance measurements.^{9,25} The same approach was used to demonstrate the importance of the latticemixing perturbation term to explain optical absorption measurements.²⁴ However, recent observation of topological currents in graphene/h-BN heterostructure²⁶ means that the moiré perturbation contains a significant inversion-asymmetric mass term²⁷ while recent theoretical papers suggest that the inversion-asymmetric part of the moiré perturbation might be comparable to the inversion-symmetric part.^{16,18,20-22} In this article, we use four models^{12,13,17,22} developed to describe the effect of the *h*-BN substrate on graphene electrons to investigate the infrared optical absorption of monolayer (MLG) and bilayer (BLG) graphene on h-BN. We show that physically different moiré perturbations can lead to similar absorption spectra, limiting the amount of information about the moiré perturbation that can be extracted from the spectrum. However, we suggest that careful examination of the absorption spectra for doping in the vicinity of four holes per moiré unit cell will provide insight into the character of the moiré perturbation and an experimental technique for distinguishing between the various theoretical proposals for the moiré perturbation. Moreover, we show that for BLG on *h*-BN (where, in contrast to MLG on *h*-BN, the optical absorption has not been investigated previously 24,28), it is important to consider the trigonal warping of the electronic band structure when interpreting the absorption spectra, as it leads to shifts of the miniband edges and hence of some of the spectral features. We also discuss the BLG/h-BNabsorption spectra in the presence of an interlayer asymmetry due to a perpendicular electric field.

For a MLG/BLG crystal with a lattice constant $a_{\rm G} = 2.46\text{\AA}$ placed on top of a *h*-BN flake with a lattice constant $a_{h\text{-}\rm{BN}} = (1+\delta)a_{\rm G}$, the lattice mismatch $\delta = 1.8\%,^7$ together with a possible misalignment of the two lattices given by the angle θ , create for small $\theta \lesssim 2^\circ$ a periodic structure shown schematically in Fig. 1(a). This planar superlattice can be described using a set of reciprocallattice vectors $\mathbf{b}_n = \hat{\mathbf{R}}_{n\pi/3} \left[1 - (1+\delta)^{-1} \hat{\mathbf{R}}_{\theta} \right] \left(0, \frac{4\pi}{\sqrt{3a}} \right), n = 0, 1, \ldots, 5$, where $\hat{\mathbf{R}}_{\varphi}$ stands for anticlockwise rotation by angle φ and $b = |\mathbf{b}_n| \approx \frac{4\pi}{\sqrt{3a}} \sqrt{\delta^2 + \theta^2}$.¹² Note that this set both rotates by $\phi(\theta)$ and changes its size as a function of θ , see Fig. 1(b), which is especially important when considering the zone-folding of the trigonally warped band structure of BLG.²⁹

In order to obtain the perturbed band structure and wave functions, and hence to compute the absorption spectra, we use a recently proposed phenomenological model constructed to describe the perturbation caused by the closely-aligned h-BN substrate on graphene electrons.¹² Within this description, all symmetry-allowed terms containing the dominant first harmonics of the moiré perturbation are added to the effective Dirac-like Hamiltonian describing electrons in MLG in the vicinity of one of the two corners (valleys) K/K' of the hexagonal Brillouin zone. This model has been also applied to the BLG/*h*-BN heterostructure²⁹ by assuming that only the bottom layer is affected by the *h*-BN substrate. The resulting Hamiltonians $\hat{H}^{\rm MLG}$ and $\hat{H}^{\rm BLG}$ for electrons in monolayer and bilayer graphene, respectively, can be written as^{12,13,29}

$$\hat{\boldsymbol{H}}^{\text{MLG}} = \boldsymbol{v}\boldsymbol{\sigma} \cdot \boldsymbol{p} + \boldsymbol{\delta}\hat{\boldsymbol{H}}
\delta\hat{\boldsymbol{H}} = (\boldsymbol{u}_{0}^{+}\boldsymbol{f}_{+} + \boldsymbol{u}_{0}^{-}\boldsymbol{f}_{-}) + \tau_{z}\sigma_{z}(\boldsymbol{u}_{3}^{+}\boldsymbol{f}_{-} + \boldsymbol{u}_{3}^{-}\boldsymbol{f}_{+})
+ \tau_{z}\boldsymbol{\sigma} \cdot [\boldsymbol{l}_{z} \times \nabla(\boldsymbol{u}_{1}^{+}\boldsymbol{f}_{-} + \boldsymbol{u}_{1}^{-}\boldsymbol{f}_{+})] + \Delta\tau_{z}\sigma_{z},
\boldsymbol{f}_{+} = \sum_{n} e^{i\boldsymbol{b}_{n}\cdot\boldsymbol{r}}, \boldsymbol{f}_{-} = i\sum_{n}(-1)^{n}e^{i\boldsymbol{b}_{n}\cdot\boldsymbol{r}},$$
(1)

$$\hat{\boldsymbol{H}}^{\text{BLG}} = \begin{pmatrix} \boldsymbol{v}\boldsymbol{\sigma} \cdot \boldsymbol{p} + \boldsymbol{\delta}\hat{\boldsymbol{H}} + \boldsymbol{u} & \hat{\boldsymbol{T}} \\ \hat{\boldsymbol{T}}^{\dagger} & \boldsymbol{v}\boldsymbol{\sigma} \cdot \boldsymbol{p} - \boldsymbol{u} \end{pmatrix},
\hat{\boldsymbol{T}} = \frac{1}{2}\gamma_{1}(\tau_{z}\sigma_{x} - i\sigma_{y}) + \frac{1}{2}\boldsymbol{v}_{3}(\sigma_{x} + i\tau_{z}\sigma_{y})(\boldsymbol{p}_{x} + i\tau_{z}\boldsymbol{p}_{y}).$$

The MLG Hamiltonian \hat{H}^{MLG} above is written in the basis of the Bloch states on sublattices $\{\phi(A), \phi(B)\}$ in the K valley and $\{\phi(B), -\phi(A)\}$ in K', whereas for the BLG Hamiltonian \hat{H}^{BLG} the basis is $\{\phi(A), \phi(B), \phi(\tilde{A}), \phi(\tilde{B})\}$ near K and $\{\phi(B), -\phi(A), \phi(\tilde{B}), -\phi(\tilde{A})\}$ near K'. In the BLG case, the tilde denotes the sublattice in the top layer. We also use two sets of Pauli matrices σ_i , $\boldsymbol{\sigma} = (\sigma_x, \sigma_y)$, and τ_i acting in the sublattice and valley space, respectively. The terms $v\boldsymbol{\sigma} \cdot \boldsymbol{p}, v \approx 10^6 \text{m/s},^{30}$ arise due to the electron hopping from a carbon atom to one of the three nearest neighbors in the same layer. In MLG, this gives rise to the conical electronic dispersion relation. In BLG, due to the interlayer coupling block \hat{T} , which includes the direct interlayer coupling $\gamma_1 = 0.38 \text{eV}^{31}$ the electronic dispersion is roughly parabolic at energies $\epsilon \ll \gamma_1.^{32}$ Also present in \hat{T} is the skew interlayer coupling, $v_3 \approx 0.1 v^{31}$ which induces a trigonal distortion in the low energy band structure. We also include the interlayer asymmetry u which can be induced by apply-

TABLE I: Values in meV of parameters from Eq. (1) for the four models used in this work to calculate the absorption spectra of MLG/*h*-BN and BLG/*h*-BN heterostructures. The given values all correspond to perfectly aligned ($\theta = 0$) structures.

model	u_0^+	u_1^+	u_3^+	u_0^-	u_1^-	u_3^-	Δ	Ref.
(1)	11	-21	-18	0	0	0	0	12,13
(2)	0	0	0	-11	-21	-18	0	12,13
(3)	1.26	0.7	-0.36	8.98	-7.31	-5.63	3.74	17,19
(4)	2	21	-0.06	5.2	-42	-5.9	5.3	22

ing electric field perpendicular to BLG and which opens a band gap in the electronic spectrum.^{32,33} Finally, the moiré perturbation $\hat{\delta H}$ consists of four terms: the first describes a simple potential modulation, the second the local A-B sublattice asymmetry due to the substrate, and the third the spatial modulation of hopping between the A and B sublattices. The final term, characterised by the parameter Δ , describes a non-zero average value of the sublattice asymmetry. Within each of the first three contributions to the moiré perturbation, the first term inside the round bracket, characterised by the dimensionless parameters u_i^+ , i = 0, 1, 3, describes the inversion symmetric part of the perturbation. Correspondingly, the second term in each round bracket, characterised by one of the dimensionless parameters u_i^- , i = 0, 1, 3, represents the inversion asymmetric part of the perturbation. Note that in the case of BLG, the perturbation enters only in the part of the Hamiltonian describing the

bottom layer since, because of the exponential decay of

(a)



FIG. 1: (a) Example of the graphene/h-BN moiré superlattice (the lattice mismatch has been enlarged for visual purposes). The black dots denote the carbon atoms in graphene, whereas the red and green correspond to the boron and nitrogen in h-BN. The yellow rhombus depicts the superlattice unit cell. (b) Hexagonal Brillouin zone (BZ) of monolayer and bilayer graphene with the valley coordinate system p_x, p_y together with an enlarged view of the supercell Brillouin zone (sBZ) and its own symmetry points κ and μ .

the $2p_z$ orbital wave function with increasing distance³⁴, we assume that the effect of *h*-BN on the top layer of graphene, twice as far from the substrate as the bottom one, can be neglected.³⁵

The position-dependent moiré perturbation terms we wrote in Eq. (1) represent all symmetry-allowed terms generated by the first harmonic of the moiré within the continuum model. Various physical effects, like strain generated due to mutual relaxation of graphene and h-BN lattices 23 or electron-electron interaction 15 , contribute to the currently unknown values of the parameters $\{u_i^+, u_i^-\}$ and Δ . Here, we use values as provided by four models published previously. All the sets, summarised in Table I, were originally obtained for a MLG/h-BNheterostructure with no angular misalignment ($\theta = 0$). The first two, 12,13 (1) and (2), result from representing h-BN as a lattice of point-charges. Model (1) assumes that the inversion-symmetric part of the perturbation is dominant, whereas for model (2) only the inversionasymmetric part is non-zero. The strength of the perturbation is then governed by a single parameter, which we set by requiring that $\sum_{i}(|u_i^+| + |u_i^-|) = 50$ meV for both of the models. Parameters for the model $(3)^{17,19}$ have been obtained using DFT. Model $(4)^{22}$ borrows the information on the stacking-dependent adhesion energies from DFT and uses them as input in an analytic description of the low-energy band structure taking into account the elastic energy of graphene.³⁷ We point out that both models (3) and (4) explicitly include the influence on the electrons of deformations of the graphene lattice arising due to its structural relaxation on top of h-BN. Such relaxation minimises the total energy of the crystal through the interplay between lowering local van der Waals interaction energy but increasing the elastic energy by introducing local strains. The evidence of such strains in graphene was found in Ref. 23. Note that it is known that the dominant effect of deformations in graphene crystals is the generation of a gauge field 38 with sublattice and valley structure identical to the third term in $\delta \hat{H}$ in Eq. (1). Indeed, according to Ref. 22 and model (4), the inhomogeneous, moiré-periodic strain in graphene lattice generates significant parameters u_1^+ and u_1^- , see Table I. This is however not the case according to model (3). Deformations also induce terms in $\delta \hat{H}$ with higher harmonics of the moir. However, these only lead to second order corrections to the energy of the lowest minibands.

In order to plot the optical absorption spectra shown in this work, we calculate the absorption coefficient $g(\omega)$ for incident light with frequency ω and polarisation $\boldsymbol{e} = (e_x, e_y)$,

$$g(\omega) = \frac{8\pi\hbar}{c\omega S} \Im \sum_{\boldsymbol{p},\lambda,\lambda'} \frac{f_{\boldsymbol{p}\lambda'} - f_{\boldsymbol{p}\lambda}}{\hbar\omega + \epsilon_{\boldsymbol{p}\lambda} - \epsilon_{\boldsymbol{p}\lambda'} + i\eta} M_{\alpha\beta}^{\lambda\lambda'} e_{\alpha}^* e_{\beta},$$

where $\alpha, \beta = x, y, \epsilon_{p\lambda}$ stands for the miniband energy found by diagonalisation of the appropriate Hamiltonian in Eq. (1), $f_{p\lambda}$ are the occupation numbers, S is the normalisation area of the miniband plane wave states and η is the broadening of the energy states (which, unless stated otherwise, we take to be $\eta = 3 \text{meV}$). The indices λ and λ' represent the remaining quantum numbers: spin, valley, and miniband index. We also numerically compute the eigenstates of the Hamiltonians in Eq. (1), in order to calculate the matrix elements of the current operator

$$M^{\lambda\lambda'}_{\alpha\beta} = \langle \boldsymbol{p}\lambda | \hat{\boldsymbol{j}}^*_{\alpha} | \boldsymbol{p}\lambda' \rangle \langle \boldsymbol{p}\lambda' | \hat{\boldsymbol{j}}_{\beta} | \boldsymbol{p}\lambda \rangle$$

where $\hat{\boldsymbol{j}}_{\alpha} = \partial \hat{\boldsymbol{H}} / \partial \boldsymbol{p}_{\alpha} = ev\sigma_{\alpha}$ are the Dirac current operators. In the absence of a moiré perturbation, the monolayer absorption is flat with a value of $\pi e^2 / \hbar c$,³⁹ while the bilayer shows a pronounced peak at $\hbar \omega = \gamma_1$.^{40,41}

In Fig. 2, we show optical absorption spectra calculated for all the four perturbation models in Table I for both MLG (top row) and BLG (bottom row). For the spectra in the left column, we assume that the structure is neutral (chemical potential μ is at the Dirac point for MLG and the neutrality point for BLG). The spectra in the right column are for *p*-doped structures with hole density $n_0 = \frac{b^2\sqrt{3}}{2\pi^2}$ equivalent to four holes per moiré unit cell ($n_0 \approx 2.5 \times 10^{12} \text{cm}^{-2}$ for $\theta = 0$). If the first valence miniband is isolated from the rest of the spectrum (for example, by a secondary Dirac point somewhere at the edge of the sBZ as suggested by experiments,^{9,10}) this hole density corresponds to the full filling of this miniband. As a result, optical transitions at low frequencies are sensitive to the details of the reconstructed miniband spectrum.

For the neutral structures, one expects the absorption spectra to be modified in the frequency range $\hbar\omega_{\rm MLG} \approx$ $\hbar vb$ for MLG and $\hbar\omega_{\rm BLG} \approx \frac{(\hbar vb)^2}{\gamma_1}$ for BLG, as these energies correspond to optical transitions from the reconstructed electronic states at the boundary of sBZ in the valence band to matching states on the conduction-band side. However, the first three models show only weak modulation of the absorption spectra, both for MLG and



FIG. 2: Absorption spectra for MLG and BLG on *h*-BN with hole density n = 0 or $n = n_0 = \frac{b^2 \sqrt{3}}{2\pi^2}$, equivalent to filling of one separated miniband with holes. For clarity, the consecutive spectra have been shifted by $3\frac{\pi e^2}{hc}$ along the *y* axis.

BLG. As a result, despite significant differences in the weighting of the parameters u_i^+ and u_i^- in each of the sets and important differences in the underlying physics (e.g. presence or lack of inversion symmetry), the presented spectral features do not offer a clear method of distinguishing between the perturbation models. For example, in a recent experimental work,²⁴ the optical conductivity data was fitted using model (1) and the results interpreted as a confirmation of the importance of the sublattice mixing term, in particular u_1^+ . However, our results indicate that distinction between the influences of u_1^+ and u_1^- or, as an alternative example, between models (1) and (2), requires more careful analysis. In fact, potentially very different combinations of the moiré perturbation parameters might lead to similar spectra for particular choices of chemical potential. However, if model (4) is the physically relevant description, then the presence of strong absorption peaks at $\hbar\omega \approx 0.3 \text{eV}$ for the monolayer, and $\hbar\omega \approx 0.1 \text{eV}, 0.5 \text{eV}$ for the bilayer will be experimentally distinguishable. This is also evidenced in the right column in Fig. 2 where we model spectra of MLG and BLG structures p-doped with hole density n_0 . Again, the optical spectra for models (1) and (2) look very similar, with the most significant feature being a low-energy peak indicating optical transitions between the second and first miniband in the valence band. This peak, although higher and shifted to even lower frequencies, appears also in the spectrum for model (3). Again, model (4) yields the most distinctive spectrum.

We now move beyond a "single spectrum" picture, which concentrated on analysing the absorption spectrum for a given carrier density, and discuss changes in optical spectra as a function of increasing hole doping $n \approx n_0$. In Fig. 3, we show the evolution of optical absorption for the four moiré perturbation models as the doping is varied from $0.8n_0$ to $1.2n_0$. For this density range, absorption spectra at low energies are dominated by features due to the moiré perturbation (for an unperturbed MLG/BLG spectrum with chemical potential μ , optical transitions with energy $\epsilon < 2\mu$ are forbidden due to Pauli blocking). For MLG for both models (1) and (2) [panels (a) and (b)], the low-energy peak due to transitions connecting states at the boundary of the sBZ in the first and second valence miniband grows in size with n increasing up to $n \sim 1.1 n_0$. A shoulder also develops for $n \approx n_0$. However, the spectrum for model (1) displays a clear side feature to the right of the main peak which appears when filling the second miniband which is a clear experimental indicator of strong inversion-symmetric perturbation. For model (3), the peak decreases rapidly after the hole density increases beyond n_0 and splits into a double-peak feature. In comparison, for model (4), the height of the low-energy peak does not change much as a function of increasing hole density. Its width increases as the first miniband is fully filled and then decreases to roughly the same shape. However, similarly to model (1), a second smaller peak appears at slightly higher frequencies.

For the BLG spectra, we notice a peak at lower frequencies than for monolayer, at ~ 50 meV. For model (1), this peak grows for hole densities lower than n_0 and then decreases for $n > n_0$. This is in contrast to the behaviour for model (2), for which this peak remains of similar height for $n = 0.8n_0$ and $n = 0.9n_0$ but increases in the vicinity of n_0 and remains high with further increase of n. This is a second clear way to experimentally distinguish model (1) from model (2). For model (3), the low-energy absorption peak grows for the hole density increasing from $n = 0.8n_0$ to $n = 0.9n_0$ but decreases already for $n = n_0$ and remains small with further increase of the hole density. The spectrum obtained for model (4) is the only one for BLG that displays more clearly distinguished features. First of all, the low-energy absorption peak does not change much with $n \leq n_0$. However, for $n > n_0$ this peak splits into two, with the one at higher energy decaying quicker than the one at lower energy as the hole density is increased. Moreover, a second peak above 0.1 eV exists for $n \leq n_0$.

An interesting open question is the dependence of the electronic moiré perturbation effects on the misalignment



FIG. 3: Absorption spectra for structures hole-doped to $0.8n_0, 0.9n_0, n_0, 1.1n_0$, and $1.2n_0$. For clarity, the consecutive spectra have been shifted by $3\frac{\pi e^2}{\hbar c}$ along the y axis. For each spectrum, the dashed vertical line indicates the 2μ threshold for Pauli blocking below which the absorption for an unperturbed MLG/BLG would be zero.



FIG. 4: Absorption spectra for BLG/*h*-BN structures [models (1) and (2) only] with various angular misalignments between the crystalline axes of the two crystals. For clarity, the consecutive spectra have been shifted by $2\frac{\pi e^2}{hc}$ along the *y* axis.

between the crystalline directions of graphene and *h*-BN. Making use of the dependence of u_i^+ and u_i^- on θ predicted for models (1) and (2),^{12,13} we show in Fig. 4 the absorption spectra for the BLG/*h*-BN heterostructure for those two models (see Ref. 28 for the MLG/*h*-BN heterostructure) and a range of misalignment angles θ . Because the perturbation parameters decrease with increasing misalignment, the spectral features due to the moiré weaken as well. However, the shift of the spectral features toward higher frequencies is visible as the characteristic energy of the moiré, $\hbar v b$, increases with θ . In particular, for misalignment $\theta \sim 1^\circ - 1.5^\circ$, the peak at the energy $\sim \gamma_1$ due to optical transitions from the lowenergy bands to the high-energy split bands⁴⁰ is modified because at these angles $\frac{v b}{\gamma_1} \approx 1$.

One of the unique features of BLG is the possibility of opening a bandgap in the electronic spectrum by breaking layer symmetry with external electric field applied perpendicular to the graphene layers.^{32,33} Band gaps $\Delta_{\rm g} \approx |u| \approx 200 {\rm meV}$ have been opened with high electric fields.^{42,43} The interlayer asymmetry u breaks inversion symmetry but preserves electron-hole symmetry³³ and hence, in the absence of the moiré perturbation, absorption in the case of asymmetry u is the same as for -u.⁴¹ In comparison, a generic graphene–h-BN interaction described by the parameters in Eq. (1) breaks electron-hole symmetry and opens a gap Δ_0 quadratic in the moiré parameters u_i^+, u_i^- .²⁹ As a result, for both the moiré perturbation and interlayer asymmetry, the gap at the neutrality point is $\Delta_{\rm g} \approx |u+\Delta_0|$, yielding different gaps for u and



FIG. 5: Absorption spectra for BLG/h-BN heterostructures [models (1) and (2) only] in the presence of the interlayer asymmetry u.

-u. This is seen in Fig. 5 where we show optical absorption spectra calculated for BLG/h-BN heterostructures with moiré perturbation effects described by models (1) and (2) in Table I and in the presence of |u| = 50 meVand |u| = 100 meV. At frequencies close to zero, the absorption falls to zero⁴⁴, indicating presence of a gap in the spectrum. This gap is larger for u < 0 because for both models $\Delta_0 < 0$. Because of that difference in the band gap at the neutrality point, although the general features of the spectra for u and -u are similar, those for u < 0 are consistently shifted to higher frequencies. We suggest that, if Δ_0 is experimentally significant in BLG, then comparing optical absorption spectra for opposite interlayer asymmetries u and -u should reveal the sign of Δ_0 and through the dependence of Δ_0 on $u_i^+, u_i^$ potentially some more information about the moiré perturbation characteristics.

Finally, in Fig. 6 we present comparison of BLG optical absorption spectra calculated taking into account or neglecting the interlayer skew coupling term parametrised by the velocity v_3 in Eq. (1). For this example, we used parameters from model (4) and assumed chemical potential to lie between the valence and conduction bands. For $v_3 = 0$, the electronic spectrum has circular symmetry and so all states with the same magnitude of momentum $p = |\mathbf{p}|$ contribute to the absorption at the same frequency. However, nonzero trigonal warping shifts the energy of an electronic state by amount depending on the polar angle of \mathbf{p} . As a result, both the position and intensity of the spectral features change. For that reason, all spectra in Figs. 2-5 for the BLG/*h*-BN heterostructure were carried out with $v_3 = 0.1v$.

In summary, we have discussed the optical absorption spectra for MLG and BLG on closely aligned *h*-BN using four models developed previously to describe the effect of the substrate on graphene electronic dispersion. We showed that, unfortunately, physically very different moiré perturbations can lead to similar absorption spectra. A detailed investigation of the spectra for samples with (almost) completely filled first miniband will provide more information about the details of the electronic perturbation due to moiré. For BLG on *h*-BN, additional information can be obtained by measuring absorption spectra in double-gated devices. However, for this sys-



FIG. 6: Absorption spectra for BLG/h-BN heterostructure with moiré parameters according to model (4) in Table I without (black solid line) and including the effects of trigonal warping.

tem it is important to include the trigonal warping of the electronic band structure during theoretical analysis.

During the final stages of writing of this paper, we became aware of the work in Ref. 45, which calculates the optical conductivity for MLG on h-BN within the framework of model (3) in the THz regime.

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