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Study of the mechanisms underlying the fresh-state response of cementitious materials modified with nanoclays

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highlights

- \blacktriangleright Nanoclays have an immediate stiffening effect on fresh cement paste, little over time.
- \triangleright Stiffening due to flocculation, not adsorption by nanoclays leading to loss of free water.

 \blacktriangleright Clay effect on effective solid volume fraction, ϕ_f , and maximum packing density, ϕ_M .

 \triangleright Describe paste flow properties and SCC formwork pressure response through ϕ_f and ϕ_M .

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ABSTRACT

The fresh-state properties of concrete can be tailored to meet the needs of specific applications, such as reduced formwork pressure for self-consolidating concrete (SCC), which requires high flowability, and slipform paving concrete, which requires compactibility during casting followed by enhanced green strength immediately after placement. Small additions of clays (especially nanoclays) have been found to be very effective in achieving these properties. The purpose of this study was to examine concentrated cement–clay systems at multiple scales to gain a better understanding of the mechanisms underlying this behavior. This was done by investigating the macroscopic flow properties over time through a shear rheological approach, where the viscosity evolution under a constant applied shear rate and a measure of structural rebuilding over time were obtained. The measured flow parameters were used to describe the microstructure in the plastic state through the Krieger–Dougherty model. This was then tied in with results obtained previously through scanning laser microscopy, a direct measure of the microstructure by focus beam reflectance method (FBRM). The dominating mechanisms behind the stiffening behavior were determined, and ultimately used to explain the response of SCC for lab-scale simulations of formwork pressure. The effective solid volume fraction and maximum packing density were found to be useful parameters in doing so, and helped to explain why nanoclays are an effective mineral admixture in modifying SCC for reducing formwork pressure.

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1. Introduction

The continued advancement of self-consolidating concrete (SCC) hinges on a firm understanding of its fresh-state properties. Through innovative processing, conventional SCC can be designed to achieve high flowability, passing ability, and stability, all of which enhance construction productivity. The fresh-state properties of SCC can be further tailored to meet the needs of specific applications, such as reduced formwork pressure and slipform paving concrete, which require compactibility during casting followed by enhanced green strength (strength of freshly-cast concrete in

the plastic state) immediately after placement. Small additions of clays (especially nanoclays) have been found to be very effective in achieving these properties. Although this has been linked to increases in fresh-state stiffness (resistance to flow) caused by the clays, no direct correlations have been made so the mechanisms are still not well understood. This requires examining the clay-cement system at multiple scales.

Previous studies relating to this topic have provided insight into how clays affect the microstructural properties of fresh cement pastes. Results of shear and compressive rheology and scanning laser microscopy tests have shown that clays increase flocculation strength and floc size [\[1,2\].](#page-8-0) Although this explains the apparent increase in stiffness caused by the clays, it is not sufficient in explaining how it translates to the reduction in SCC formwork pressure observed in lab-scale tests. It is necessary to provide a link

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between the microstructural properties and the response of the materials designed for this application. The purpose of this study was to do so by investigating the macroscopic flow behavior over time of cement pastes with and without small nanoclay additions. This was done through a shear rheological approach, where the viscosity evolution under a constant applied shear rate and a measure of structural rebuilding over time were obtained. The measured flow parameters were used to describe the microstructure in the plastic state through the Krieger–Dougherty model. This was then tied in with results obtained previously through scanning laser microscopy, a direct measure of the microstructure by focus beam reflectance method (FBRM). The dominating mechanisms behind the stiffening behavior were determined, and ultimately used to explain the response of modified SCC mixes designed for reduced formwork pressure during casting.

2. Background

Very small additions of clays (up to 1.5% by mass of cement) lead to significant changes in the fresh-state properties of cement-based materials, which may be tailored to meet the needs of specific applications. A study on extruded cement-based materials demonstrated that cellulose ethers can be partially replaced by two types of kaolinite clays [\[3\]](#page-8-0). At additions as low as 0.15% by mass, mixes were able to flow through the die at sufficiently low pressures and exhibit shape stability upon exiting. In studies that simulated slipform (SF) paving with a lab-scale minipaver, various types of clays were found to be an integral mineral admixture in low compacting energy concrete, SF-SCC, mix design [\[4–7\]](#page-8-0). SF paving is widely used for concrete pavement construction, where placing, consolidating, and finishing of fresh concrete are done in one continuous process. SF-SCC makes it possible to eliminate external vibration from this process. Not only does this make it less energy intensive but it helps to avoid problems associated with over-consolidation, such as segregation and significant reduction of entrained air, which lead to reduction in durability and extensive longitudinal cracking along the path of the vibrators. SF-SCC containing fly ash and superplasticizers exhibited high fluidity and smooth surface finish but poor shape stability. A significant improvement was observed with small additions of clays, where straight edges were achieved after placement with little compromise to initial flowability and compactibility.

The motivation behind the present study was to better understand the response of clay-modified SCC designed for the application of reducing formwork pressure. Although the use of SCC is becoming more prevalent in pre-cast plants, there is still apprehension when extending it to cast-in-place applications. This is partially attributed to the high lateral pressure SCC exerts on formwork from increased fluidity and faster casting rates compared to vibrated concrete. There was a study that investigated the effect of various mineral admixtures on SCC formwork pressure response [\[8\]](#page-8-0), where the formwork pressure evolution of SCC concrete mixes were measured using a formwork pressure device (depth, $D = 15$ cm and height, $H = 30$ cm), shown in Fig. 1. This setup simulated the lateral pressure induced on formwork during casting, allowing for different casting speeds and heights. Vertical pressure is applied directly to the surface of the concrete using a steel piston loaded by a closed-loop hydraulic testing frame and the lateral pressure is recorded by a transducer. (The details of the test setup can be found elsewhere [\[8\].](#page-8-0)) The results for SCC with 0% (NC0) and 0.33% (NC0.33) nanoclays are shown in Fig. 2. Although both mixes exhibited the same initial flowability with a slump flow of 60 ± 2 cm (which fulfills the requirement for SCC), it is apparent that even at small dosages clays can lead to significantly reduced

Fig. 1. Formwork pressure device with a depth $D = 15$ cm and height $H = 30$ cm.

Fig. 2. Formwork pressure response of SCC with and without nanoclays (both mixes had an initial slump flow of 60 ± 2 cm) [\[8\].](#page-8-0)

formwork pressure. However, to explore the mechanisms behind this behavior was not within the scope of that study.

To most effectively tailor fresh-state properties for application through the use of nanoclays, it is necessary to understand the mechanisms underlying the stiffening behavior. Possible mechanisms include capillary suction [\[9\]](#page-8-0), hydration mechanisms, water adsorption, particle packing, and flocculation. However, hydration can be considered to be negligible due to the time scale of the observed behavior (within 2 h after initial cement–water contact). The contribution of dry clay on particle packing is unlikely a significant factor due to the small dosage (less than 1% by mass of cement). And capillary suction is thought to be an effect of flocculation. This leaves flocculation and water adsorption, both of which can be studied through flow parameters and the Krieger–Dougherty equation, which will be explained in the

proceeding section. Previous studies have demonstrated that clays increase flocculation strength and floc size. As for water adsorption, if the water content is kept constant, the inclusion of any nanoparticles (due to their high specific surface area) would reduce free water and result in increased yield stress and viscosity [\[10\].](#page-8-0) Therefore it is important to evaluate whether water adsorption by the nanoclays could be a dominating mechanism.

SCC formwork pressure has been linked to degree of thixotropy (a decrease in viscosity under shear followed by a recovery once the stress is removed [\[11\]\)](#page-8-0), where increased thixotropy leads to decreased formwork pressure [\[12,13\].](#page-8-0) Although thixotropy likely plays a role and the influence of clays on this property deserves attention, this present study focused on the overall stiffening behavior, without separating reversible and irreversible changes.

3. Theoretical background for data analysis

3.1. Krieger–Dougherty equation

The Krieger–Dougherty model [\[14\]](#page-8-0) describes the relationship between viscosity and solid volume fraction for dispersed suspensions:

$$
\frac{\eta}{\eta_c} = \left(1 - \frac{\phi}{\phi_M}\right)^{-[\eta]\phi_M} \tag{1}
$$

where η is the apparent viscosity of the suspension, η_c is the apparent viscosity of the continuous phase (or liquid phase), ϕ is the concentration of solids by volume, $[\eta]$ is the intrinsic viscosity of the suspension, and ϕ_M is the maximum solids concentration by volume, or maximum packing density. ϕ_M depends on particle size distribution and shape, while $[\eta]$ depends on particle shape – both are shear dependent. Derivations of the model can be found elsewhere [\[15,16\].](#page-8-0) It is apparent from the equation that an increase in ϕ would lead to higher viscosity, assuming ϕ_M and $[\eta]$ remain constant.

Cement pastes are highly concentrated, highly flocculated suspensions. Flocculation leads to entrapped water within particles, resulting in a change in the solid volume fraction. To take this into consideration, Soua et al. [\[17\]](#page-8-0) proposed the following modified model, which is an extension of the Krieger–Dougherty equation for flocculated suspensions:

$$
\frac{\eta}{\eta_c} = \left[b \left(1 - \frac{\phi_f}{\phi_M} \right) \right]^{-2} \quad \text{for} \quad \tau > \tau_y \tag{2}
$$

where ϕ_f is the effective volume fraction (floc volume fraction), ϕ_{fM} is the maximum packing density of the flocs, b is an adjusting factor, τ is applied shear stress, and τ_y is yield stress. And it follows that at sufficiently high shear $\phi_f \rightarrow \phi$ and $\phi_{fM} \rightarrow \phi_M$ as the flocs continue to break down and release entrapped water.

The flow behavior of cement paste was found to be well-described by the Krieger–Dougherty model [\[18,19\],](#page-8-0) although the parameters may not have the same physical significance as they do for non-flocculated systems. For cement paste, the maximum packing density is of great interest because it can provide a measure of the degree of flocculation in the system, with higher degree of flocculation leading to lower ϕ_M . Tregger et al. used the modified model to evaluate the influence of clays on flocculation strength, the resistance of flocs to break down under a given shear condition [\[1,20\].](#page-8-0) Results showed that cement paste with small clay additions yielded lower ϕ_M values compared to neat cement paste, indicating that clays lead to a more flocculated structure by increasing flocculation strength. This study was limited to obtaining the properties of the paste shortly after mixing. One of the main objectives of the present work is to see how the clays are affecting ϕ_M , as well as ϕ , over time.

The evolution of packing density is mainly due to flocculation or agglomeration, the process of which may be simply stated as the transition from single cement particles into flocs. The well-known theory of packing density of non-crowding binary particles shows the relationship between the fractional solid volume of one component and the maximum packing density of the entire system. The details of the theory can be found elsewhere [\[21\]](#page-8-0) and will briefly be explained here. In a binary system, there are two components (small grains and large grains) where each component i has a packing density, α_i . The fractional solid volume of each component, $[\phi_i]$, is defined as:

$$
[\phi_i] = \frac{\phi_i}{\phi_M} \tag{3}
$$

where ϕ_i is the partial volume of component *i* and ϕ_M is the packing density of the binary system. There are two possible conditions. The first is where the large grains are fully packed and the remaining interstitial space is composed of the small grains. Following the convention $r_1 > r_2$, the packing density for this condition can be represented by the following expression:

$$
\phi_{M,1} = \frac{\alpha_1}{1 - [\phi_2]} \tag{4}
$$

As $[\phi_2]$ increases (volume of small grains increases), the packing density of the system increases. The second condition is when the small grains are fully packed:

$$
\phi_{M,2} = \frac{\alpha_2}{1 - (1 - \alpha_2)[\phi_1]} \tag{5}
$$

Here, it is apparent that as $[\phi_1]$ increases (volume of large grains increases), ϕ_M increases. Fig. 3 shows the packing density of the system as a function of $[\phi_2]$.

Extending this model to a cement–water system, the two components can be considered to be single cement particles (r_2) and flocs (r_1) : $r_2 < r_1$. It follows that in the highly flocculated state cement paste will follow the $\phi_{M,1}$ curve whereas in the highly dispersed state it will follow the $\phi_{M,2}$ curve. Considering that single cement particles follow a particle size distribution (PSD) and flocs vary in size, this is a simplified assumption. Still, the model can be used to extract useful qualitative information about ϕ_M from the results of FRBM tests. FRBM is a particle size analysis measurement technique that provides information about particle chord length distribution in real time [\[22\].](#page-8-0) It is a direct measure of the microstructure of cement paste in its plastic state, whereas in

Fig. 3. Packing density of a binary system, ϕ_M , as a function of $[\phi_2]$.

shear rheological methods it is described indirectly using the Krieger–Dougherty model. In a previous study FBRM tests were conducted on cement pastes with and without clays subjected to shear and the evolution of average floc size were compared [\[2\]](#page-8-0).

4. Experimental methods and materials

4.1. Materials

Type I ordinary Portland cement and tap water were used in all paste mixes. A purified magnesium aluminosilicate, or palygorskite, was selected for this study, as it exhibited the most enhanced effect in previous studies compared to other clays. For example, in the flocculation study by Tregger et al., this clay yielded the lowest ϕ_M value, indicating that it led to the greatest increase in flocculation strength. They are nanoclays that have been chemically exfoliated to preserve their uniform shape and size while removing all impurities (such as quartz and swelling clays). As received, they are highly agglomerated – the scanning electron microscopy (SEM) image, Fig. 4a shows that the agglomerates are on the micron scale. However, in the highly dispersed state they are nanosized particles with a needle-like shape $(1.75 \mu m)$ in length and 3 nm in diameter) [\[23\],](#page-8-0) Fig. 4b.

4.1.1. Water adsorption of nanoclays

In the dispersed state, the high specific surface area of the nanoclays in combination with their small particle size can result in a significantly high adsorption capacity. The water adsorption of the nanoclay was measured through a pressure filtration setup used in similar work on alite (C_3S) [\[19\]](#page-8-0), shown in Fig. 5. A sample of dry nanoclay was placed on a fine filter (0.45 µm membrane) over which a layer of water was carefully poured so as not to disturb it. A constant negative pressure was applied at the bottom by a vacuum, which pulled the water through the nanoclays to form a cake and consolidate it. Water that was not retained was filtered out into an Erlenmeyer flask. The water over the cake was kept at a constant level and the pressure was applied until a constant flow rate was achieved (after

Fig. 4. SEM image of nanoclay in the (a) agglomerated state and (b) dispersed state [\[31\]](#page-8-0).

Fig. 5. Schematic of filtration setup to measure water adsorption of nanoclay.

approximately 90 min) at which point the cake reached a state of equilibrium (full adsorption of the nanoclays and full saturation of the cake). At sufficiently high pressures, it is assumed that the cake is consolidated enough to contain only the nanoclays and the adsorbed water. A portion of the cake was then carefully removed, the wet mass (m_{wet}) was recorded and then oven dried at 105 °C for 3 h to evaporate off all of the water. The sample was placed in a desiccator for 30 min before the dry mass (m_{dry}) was recorded. From the wet and dry mass, the water adsorption of the clays was calculated:

Adsorption(
$$
\%
$$
) = $\frac{m_{\text{wet}} - m_{\text{dry}}}{m_{\text{dry}}} \times 100\%$ (6)

This was performed at increasing applied pressures until the adsorption values converged. The average of three measurements was taken to be the representative water adsorption at each applied pressure. The results are shown in Fig. 6. At lower pressures, the cake was not fully consolidated and contained excess water. At higher pressures, the values converged to approximately 200% by dry mass of nanoclay, which agrees with reported values by the manufacturer [\[23\]](#page-8-0) and other literature [\[24\]](#page-8-0).

4.2. Dispersion

The nanoclays used in this study were hydrophilic, which made them easily dispersible in water. To determine the influence of dispersion on fluidity, the minislump flow of cement pastes were measured and compared for different mixing methods. Pastes were prepared either by a small planetary mixer or a household blender for high shear mixing. The mixing protocols are shown in [Table 1.](#page-4-0) In the case of the planetary mixer, the nanoclays were either undispersed and added asreceived in the dry, agglomerated form (P1) or pre-dispersed in water at high shear in a blender prior to mixing (P2). The mini-slump flow tests were performed immediately after mixing based on ASTM C 1611 [\[25\]](#page-8-0), where a Hägermann cone

Fig. 6. Water adsorption of nanoclays at various applied pressures as measured in the filtration test.

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Table 1 Mixing protocols.

Protocol	Mixer	Mixing procedure
P ₁	Planetary	(1) Add cement (and nanoclay) to bowl. Mix for 2 min at speed 1 (2) Add water, mix for 1 min at speed 1, scrape bowl, mix for 2 min at speed 1, scrape bowl, mix for 4 min at speed 2
P ₂	Planetary	(1) Add nanoclay and water to blender. Blend for 2 min
		(2) Add nanoclay-water suspension to dry cement in bowl (3) Step 2 of protocol P1
	Blender	(1) Add water (and nanoclay) to blender. Blend for 1 min
		(2) Add half of cement. Blend for 15 s (3) Add rest of cement, blend for 15 s, scrape. Blend until total mixing time is 3 min

with 100 mm lower diameter, 70 mm upper diameter and 50 mm height was used. One slump flow measurement was taken for each mix. All paste mixes had a volume fraction of 0.43 and a nanoclay addition of 0% or 0.5% by mass of cement.

4.3. Viscosity evolution

A shear rheological method was used to obtain the viscosity evolution of cement pastes under a constant shear condition, which will be referred to as the constant rate (CR) protocol herein. All shear rheology tests were performed in a temperature-controlled rheometer with a coaxial cylinder geometry set at room temperature. The CR protocol is shown in Fig. 7. A constant shear rate of 300 s^{-1} was applied to paste samples for 60 min, from which the viscosity evolution was obtained.

The mix compositions tested are shown in Table 2. A plain cement paste mix (NC0) and a mix with 1% addition of nanoclays by mass of cement (NC1) were tested, both with a volume fraction of 0.43. In addition, each mix was adjusted for water adsorption of the clays. NC1H is NC1 with an additional 10 g of water to compensate for the amount adsorbed by the 5 g of nanoclay while NC0H is NC0 with 10 g less water. (200% water adsorption by mass of clay was assumed according to the results of the filtration test.) For the CR test, pastes were prepared by following protocol B in Table 1, subjected to high shear mixing so that they could be placed in the rheometer as quickly as possible after initial cement and water contact. This was considered to be important because the adsorption rate of the nanoclays was unknown. This test was performed to determine whether water adsorption leading to a loss of free water was a dominating factor in the increase in stiffness caused by the clays. Three measurements were taken for each mix.

Fig. 7. Applied shear rate and measured viscosity plotted against time for CR protocol.

Table 2

4.4. Specific rebuilding energy (SRE)

In the CR protocol, the material was subjected to a constant high shear. To investigate the stiffening behavior when it is left at rest, the specific rebuilding energy (SRE) protocol, developed by Ferron et al., was applied [\[26\].](#page-8-0)

The applied shear rate for the SRE protocol is shown in Fig. 8. The paste sample was subjected to a pre-shear at a high rate (600 s^{-1}) for 10 min to break down its inherent microstructure until an equilibrium state was reached. Then the shear was linearly ramped down to $0 s⁻¹$ over 30 s to obtain an equilibrium curve. The mix was at rest for the remainder of the test, with the exception of a hysteresis loop (linearly ramped up to 300 s⁻¹ and back down to 0 s⁻¹ over 30 s) applied every 15 min up to 60 min. The area between each loading curve and the equilibrium curve between $100 s^{-1}$ and $200 s^{-1}$ is defined as the SRE, as shown in Fig. 9. Also given in the following expression:

$$
SRE = \int_{\dot{\gamma}=100}^{\dot{\gamma}=200} (\tau_{\text{Loading}} - \tau_{\text{Equil}}) d\dot{\gamma}
$$
 (7)

Fig. 8. Applied shear rate for SRE protocol.

Fig. 9. Shear stress versus shear rate curves obtained from SRE protocol.

where τ is shear stress and $\dot{\gamma}$ is strain rate. The rate at which SRE increases provides a measure of structural rebuilding of the material over time. Cement pastes with (NC0.3) and without (NC0) a 0.3% nanoclay addition by mass of cement were tested. For the SRE test protocol P2 was followed. Three measurements were taken for each mix.

In addition, the flow curve parameters, yield stress and viscosity, were obtained from the equilibrium curve to compare the initial state of the samples. A modified Bingham equation was used [\[27\]](#page-8-0):

$$
\tau = \tau_{y} \left[1 - \exp \left(3 \cdot \frac{\dot{\gamma}}{\dot{\gamma}_{\rm crit}} \right) \right] + \mu \dot{\gamma}
$$
 (8)

where τ_y is yield stress, $\dot{\gamma}_{\text{crit}}$ is critical strain rate corresponding to τ_y , and μ is plastic viscosity.

5. Experimental results

5.1. Dispersion

To determine the influence of dispersion, the slump flow of cement pastes were measured and compared for various mixing methods. The results are shown in Table 3. As expected, the plain cement paste (NC0) experienced higher fluidity when mixed in a blender than in a planetary mixer, due to a higher degree of deflocculation. On the other hand, in the case of the cement paste with 0.5% addition of clay (NC0.5), the opposite was observed – the mix prepared in the blender was much stiffer. In addition, the mix also experienced a significant decrease in fluidity when the clays were pre-dispersed in water with a blender prior to mixing in a planetary mixer with the cement. The results illustrate that high degree of dispersion of the clays maximize their effect, likely due to increased surface interactions between the nanoclays and the cement particles. Also, the stiffening effect of the clays occurs immediately after placement.

5.2. Viscosity evolution

Table 3

Fig. 10 shows the results of the CR protocol. Fig. 10a is the viscosity evolution up to 5 min, during which each mix reached equilibrium. In evaluating the effect of adjusting for water adsorption of the clays, if loss of free water is a governing stiffening mechanism, mixes NC0 and NC1H and mixes NC1 and NC0H should have exhibited similar viscosity evolutions. The plot shows that the viscosity at equilibrium of these mixes did not coincide: η_{NC1H} > η_{NC0} and η_{NC1} > η_{NC0H} . Also, the curves follow different trends – both clay mixes show a significant decrease in viscosity while the neat cement pastes do not. It is apparent that adjusting the water content was not sufficient in negating the effect of the clays on stiffening, thereby making it unlikely that water adsorption leading to loss of free water is a dominating factor.

The behavior of the clay mixes can be explained using Eq. [\(2\),](#page-2-0) which shows viscosity to be directly related to effective volume fraction. After the material is placed and at rest, ϕ_f increases instantaneously, indicated by the high viscosity measured at the start of the protocol upon introduction of shear. This also shows that the effect of the clays on stiffness occurs rapidly, within 5 min after initial cement–water contact. Viscosity decreases as

Fig. 10. Viscosity evolution (a) up to equilibrium (5 min) and (b) after equilibrium (up to 60 min).

the material continues to be sheared, indicating decreasing ϕ_f . This is likely due to deflocculation, during which changes occur in the solid and water content. It follows that flocculation induced by the clays is the dominating mechanism underlying the stiffening behavior.

After equilibrium, at which point flocs cannot be broken down any further under the given shear condition, viscosity increased for all pastes, as shown in Fig. 10b. As expected, η_{NC1} > η_{NC0} at equilibrium due to the increase in flocculation strength induced by the clays ($\phi_{M,NC1}$ < $\phi_{M,NC0}$). However, both mixes experienced a similar rate of stiffening over time. [Table 4](#page-6-0) shows the change in viscosity of all mixes from equilibrium to final (60 min). The values are all very close, indicating that the stiffening over time under the constant shear condition was not influenced by the clays. Referring back to the Krieger–Dougherty equation, Eq. [\(1\),](#page-2-0) changes in viscosity may be attributed to changes in the parameters ϕ and/or ϕ_M . Increasing viscosity can result from increasing ϕ , decreasing ϕ_M , or a combination of both. The results of the CR protocol alone are not sufficient in determining which parameter is governing this behavior over time, but they do show that it is attributed to the properties of the cement particles (e.g. adsorption or flocculation behavior), not of the clays.

Table 4

Change in viscosity from equilibrium to final (60 min) of pastes under constant shear.

Mix	Δ Viscosity (Pa-s)
NC ₀	0.192
NC ₁	0.175
N C 1 H	0.18
NCOH	0.176

5.3. Structural rebuilding

The results of the SRE protocol are shown in Figs. 11 and 12. It is apparent that the clay mix experienced greater yield stress and viscosity at equilibrium, as shown in [Table 5](#page-7-0) and Fig. 11. However, the two mixes went onto experience the same SRE rate, indicating similar structural rebuilding (Fig. 12). A change in the water or solids content would lead to a change in the SRE rate. For instance, mixes with higher water content would experience lower SRE rates due to a reduction in coagulation forces and interparticle collisions. Therefore, the results show that the clays did not have a significant effect on the water or solids content after equilibrium and the mechanism behind structural rebuilding must be linked to the cement particles. This agrees with the results of the constant applied shear rate protocol. The clays have an immediate effect on thixotropy, likely due to flocculation, but then have little influence over time.

6. Discussion

In the proceeding section, the results of the present study (viscosity evolution and structural rebuilding) will be tied in with those obtained in previous FBRM and formwork pressure studies that were performed by other researchers. The aim is to gain a better understanding of the effect of clays on the fresh state by observing clay-modified systems at multiple scales.

6.1. FBRM and linear packing density model

In a separate study, the microstructure of concentrated cement– clay suspensions was directly examined through scanning laser microscopy, where an FBRM probe was used to measure the floc size evolution under different shear stresses [\[2\]](#page-8-0). Cement pastes with and without clay (1.5% by mass of cement) were subjected to alternating cycles of low (40 rpm), high (400 rpm), and low (40 rpm) mixing intensities. The clay was sepiolite, a hydrated magnesium silicate. Sepiolite and palygorskite (the nanoclay used

Fig. 11. Equilibrium flow curve obtained from SRE protocol.

Fig. 12. SRE plot of pastes with and without a 0.3% nanoclay addition.

in the present study) are both phyllosilicates, and thereby share similar physical and physico-chemical properties [\[28\]](#page-8-0). The effect of sepiolite on the flocculation behavior of cement paste should reflect that of palygorskite.

The results are shown in Fig. 13, where floc size is given by average chord length. It is apparent that the clays increase floc size – the cement paste with clay exhibited higher mean chord length throughout the cycles. However, during the low intensity cycles, both pastes experienced a similar evolution in chord length. Although the clay mix started off the cycle more flocculated at around 2500 s, the rate of floc growth was similar to that of the plain cement paste. This suggests that the clays had an immediate effect on the degree of flocculation (upon introduction or removal of high shear) but had little influence over time, during which the rate of flocculation appears to be dominated by the cement phase. This was observed in the CR and SRE tests of the present study, as well.

Extending the linear packing density model to the results of the FBRM test, information about ϕ_M can be extracted. Upon removal of high shear, both pastes exhibited an immediate increase in average floc size. Therefore, it can be assumed that during the low

Fig. 13. FBRM results showing the evolution of mean chord length for pastes with and without a clay addition [\[2\].](#page-8-0)

Table 5

Equilibrium flow curve parameters obtained from SRE protocol.

shear cycle both pastes are in a highly flocculated state and will follow the $\phi_{M,1}$ curve in [Fig. 3.](#page-2-0) Over this cycle, the average floc size increases at a similar rate for both pastes. This translates to decreasing $[\phi_2]$ (implying increase of flocs) on the ϕ_{M_1} curve, which corresponds to decreasing packing density. Referring back to the Krieger–Dougherty equation, decreasing ϕ_M will result in increasing viscosity, as was measured in the CR protocol after equilibrium.

6.2. Application

The results of the shear rheology tests were supplemented with those of the FBRM test, leading to a better understanding of how nanoclays affect flow behavior. The immediate increase in stiffness upon removal of shear is due to flocculation, an incremental jump in ϕ_f induced by the clays. With the reintroduction of shear, the material will experience a significant decrease in ϕ_f but continue to exhibit higher viscosity compared to neat paste due to increased flocculation strength and floc size, $\phi_{M,NC1}$ < $\phi_{M,NC0}$. The increase in stiffness over time (i.e. $\Delta \eta$ and Δ SRE), however, is similar for both pastes with and without clays. This may be attributed to decreasing ϕ_M governed by the cement particles. Here, this behavior will be correlated with the response of clay modified SCC mixes for formwork pressure.

6.3. Formwork pressure

As aforementioned, in a previous study the formwork pressure response of SCC with 0% (NC0) and 0.33% (NC0.33) clays were measured, results of which are shown in [Fig. 2.](#page-1-0) The basic concrete mix proportions are given in Table 6. As shown in [Fig. 2](#page-1-0), the applied vertical pressure, σ_V , followed a step-wise loading protocol, where the load was increased incrementally every .5 h and held. In an actual casting situation, σ_V would increase gradually as SCC is being poured in the formwork before leveling off, as shown in Fig. 14, where the slope of σ_V is representative of casting rate. A two-function model proposed by Kwon et al. [\[29\]](#page-8-0) and Kim et al. [\[30\]](#page-8-0) can describe the measured lateral pressure, σ_L , from parameters obtained through a step-wise loading protocol by introducing a loadingtime dependency. In this model, the pressure ratio between σ_V and σ_L is expressed by two functions as follows:

$$
\Delta \sigma_L(t, t') = \alpha(t, t') \beta(t') \Delta \sigma_V(t')
$$
\n(9)

where α is delayed response, β is instantaneous response, t is current time, and t' is time of loading. Further, while the concrete is plastic (before final set), α and β can be represented in terms of delayed coefficient a and instantaneous coefficient b, both of which are material parameters:

$$
\alpha(t, t') = 1 - a^2 \cdot t' \cdot (t - t') \tag{10a}
$$

$$
\beta(t') = 1 - b \cdot t' \tag{10b}
$$

Table 6

The basic composition of SCC in formwork pressure study expressed in kg to produce 1 m^3 of concrete.

Ingredient	Water	Cement	Fly ash	Sand	Gravel
Amount	182	338	154	879	824

Fig. 14. Two function model for formwork pressure of concrete [\[30\].](#page-8-0)

In the CR and SRE protocols, an equilibrium state was established for a given constant shear rate, which provided a well-defined initial reference condition. In the loading protocol followed in the formwork pressure study, each step-wise increase in vertical pressure is an increase in volumetric strain. At each applied constant volumetric strain rate, a new equilibrium state is established. (It is assumed that the material can reach equilibrium very rapidly due to the high applied load.) This allows for the formwork pressure response to be described through ϕ_M .

The delayed and instantaneous response of NC0 and NC0.33 are shown in [Fig. 15a](#page-8-0) and b, respectively. Upon increase in volumetric strain, NC0.33 experienced lower lateral pressure compared to NC0, as shown in the evolution of β in [Fig. 15](#page-8-0)b. The $\beta(t')$ curve for NC0.33 is steeper, indicating NC0.33 experienced increasingly reduced σ_L at each incremental increase in σ_V compared to NC0. This can be attributed to the lower ϕ_M induced by the nanoclays. Similarly, a lower ϕ_M led to higher η at equilibrium in the CR protocol and higher yield stress and dynamic viscosity at equilibrium in the SRE protocol. As the volumetric strain was held constant over .5 h, lateral pressure decreased at a similar rate for both SCC mixes, as shown in the evolution of $d\alpha(t, t')$ $d(t-t')$ in [Fig. 15a](#page-8-0). This occurred because the rate at which ϕ_M decreased under constant loading was governed by the properties of the cement and fly ash particles, not the nanoclays. This correlates well with the flow behavior measured in this study, where clays had an immediate effect but have little or no influence over time.

For the design of formwork, the maximum formwork pressure, σ_{max} , is a critical value. The maximum formwork pressure depends only on placement rate and material properties (i.e. unit weight of concrete and coefficients a and b), not time. In a formwork pressure study by Kim et al. [\[30\],](#page-8-0) which used the two-function model to characterize a variety of different SCC mixes, a much greater variation was found for coefficient b than a. Also, the variation in a had a negligible effect on σ_{max} while the variation in b had a dominant one during the first few hours of casting. It is apparent from [Fig. 15](#page-8-0)b that the instantaneous increase in ϕ_M induced by the nanoclays significantly affects coefficient b. It follows that nanoclays have the capacity to greatly reduce σ_{max} , which explains why they are effective in modifying SCC for the application of reduced formwork pressure.

In the present study, the influence of clays on the flow behavior of systems with cement was investigated. Since SCC systematically incorporates fly ash, viscosity modifying agents and superplasticizers, the interaction of clays with these constituents and the effect on rheology must be explored, as well. Work is currently being done and will be reported in the future.

Fig. 15. (a) Delayed response and (b) instantaneous response.

7. Conclusions

Examining concentrated cement–clay systems at multiple scales gave insight into how the clays affect fresh-state properties. The effective solid volume fraction and maximum packing density were useful parameters in describing how the clays influence the macroscale flow properties of pastes over time. It is apparent that the clays have an immediate effect on stiffness, and the governing mechanism is flocculation, not water adsorption leading to loss of free water. However, under a constant shear rate, the clays have little or no influence over time – stiffening is dominated by changes in ϕ_M caused by the powder particles such as cement and/or fly ash. The parameters ϕ_f and ϕ_M were ultimately used to explain the response of SCC concrete for lab-scale simulations of formwork pressure, from which it was apparent why clays are an effective mineral admixture in modifying thixotropy of SCC for this application.

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