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# Influence of curing on pore properties and strength of alkali activated mortars

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6	ABSTRACT
7	The paper investigates the effect of wet/dry, wet and dry curing on the pore properties and
8	strength of an alkali activated cementitious (AACM) mortar. The pore characteristics were
9	determined from the cumulative and differential pore volume curves obtained by mercury
10	intrusion porosimetry. AACM mortars possess a bimodal pore size distribution while the
11	control PC mortar is unimodal. AACM mortars have a lower porosity, higher capillary pore
12	volume, lower gel pore volume and lower critical and threshold pore diameters than the PC
13	mortar which indicate greater durability potential of AACMs. Wet/dry curing is optimum for
14	AACM mortars while wet curing is optimum for the PC mortar. Shrinkage and retarding
15	admixtures improve the strength and pore structure of the AACMs.
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24	Keywords: Alkali activated cementitious material AACM, gel pores, capillary pores,
25	cumulative pore volume, differential pore volume, Porosity

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42	Notations:	
43	AACM	Alkali activated cementitious materials
44	PC	Portland cement
45	GGBS	Ground granulated blast-furnace slag
46	ITZ	Interfacial transition zone
47	p	Absolute applied pressure
48	r	Pore radius
49	у	Mercury surface tension (= 0.48N/m)
50	$\phi$	Mercury contact angle (= $140^{\circ}$ )

#### 51 **1.0 Introduction**

52 The use of alkali activated cementitious materials (AACM) in place of Portland cement (PC) 53 has been recognized to have great potential in construction applications. There is the need for 54 a viable alternative to PC because of the high carbon footprint generated during its production 55 with a huge energy demand, which is not sustainable in the future. The carbon footprint is 56 significant because of the large volume of Portland cement PC consumed worldwide, which 57 is ranked second after the volume of water [1]. To put this into perspective, for each tonne of cement produced an equivalent tonne of  $CO_2$  is emitted into the atmosphere. This translates to 58 the emission of 400 Kg of  $CO_2$  per 1 m<sup>3</sup> of concrete production [2]. In addition, the cement 59 industry consumes between 12 - 15% of the total industrial energy use [3]. The electric 60 61 energy consumption for the burning process during cement production is estimated to be 65 62 kWh/tonne while the thermal energy consumption for cement grinding is 2.72 GJ/tonne [3]. 63 Clearly, there is a dire need for reducing this carbon foot print and energy demand.

64 Limited knowledge is available in literature on the pore properties of AACMs and 65 geopolymers [4]. However, established knowledge on the pore properties of conventional 66 concrete [5] shows their critical importance in controlling the durability and strength of 67 concrete. The pore properties are equally important for AACMs and other porous ceramic materials. The refinement of concrete pore structure improves its compressive strength, 68 resistance to diffusion of deleterious substances such as chlorides and CO<sub>2</sub>, which affect its 69 70 durability [6]. These deleterious substances which cause corrosion of steel in concrete are 71 transported through the concrete pores by capillary absorption, hydrostatic pressure and 72 diffusion [7]. Diffusion of the ionic elements ( $Cl^{-}$  and  $Na^{+}$ ) is mainly through the pores of the 73 cement paste matrix and not through the interface between cement paste and aggregates [8]. 74 The interface between the cement paste and aggregates accounts for up to 50% of the total 76 from each other, thereby preventing the penetration of harmful elements through them [8]. 77 The little understanding of the pore properties of AACM concrete provided in current 78 literature suggests that the pore size distribution of AACMs is bimodal under all curing 79 conditions [2,4]. The pores of AACMs are separated into two zones (>  $1\mu m$  and <  $0.02\mu m$ 80 ranges) unlike a similar grade of PC matrix which is observed to be unimodal ranging 81 between  $0.01\mu m$  to  $0.1\mu m$  [2,4]. Literature suggests that the gel pores in AACMs are formed 82 during the polymerization of aluminosilicate gel during curing [9]. The extent of gel pores 83 formed under different curing regimes is not understood. The gel pores are defined to be 84 within the range of 0.005 to 0.01 µm based on PC concrete research [9]. The large capillary 85 pores which are orders of magnitude bigger than gel pores and are within the range of 0.01 86 µm to 100 µm based on PC concrete research [9]. Yue and Jiaqi [10] showed an inverse 87 relationship between the volume of gel and capillary pores as hydration progresses in PC 88 concrete. During the hydration process of concrete, the volume of capillary pores decreases 89 while the gel pores increases. This results in a lower cumulative pore volume in time because 90 the comparatively large capillary pores is partially occupied by the binder gel. Ultimately, a 91 denser microstructure evolves as the hydration progresses. The influence of curing on the 92 pore properties of AACMs such as the gel pores, capillary pores, critical and threshold pore 93 diameters are not defined in literature. These aspects of pore properties of AACMs are 94 reported in this paper.

volume of pores in hardened concrete but these were found to be discontinuous and isolated

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Pore refinement of PC concrete is achieved by high humidity (> 80% R.H) curing which provides prolonged hydration of cement at low or high temperatures [5]. In the case of AACMs, earlier research has shown a need for high temperature curing at 50 -  $80^{\circ}$ C, such as steam or dry heat, for optimum geopolymerization reaction [2,11]. More recent work uses ambient temperature (20 ±  $2^{\circ}$ C), which is practical on construction site, for curing AACMs 100 [12,13]. The optimum levels of relative humidity required for AACM curing are not 101 established. However, results indicate that "dry" curing at low relative humidity (e.g. 60% 102 R.H.) produces high strength for AACMs unlike PC concrete which has maximum strength 103 under wet curing (100% R.H.) [14,15]. This can be beneficial for practical use of AACMs 104 since insitu curing in construction does not provide idealized wet conditions. Practical site 105 conditions represent a balance between, wet, wet/dry and dry conditions by preventing 106 moisture loss at early age while concrete is exposed to ambient conditions of wetting and 107 drying in the longer term. The practical curing conditions wet/dry, dry and wet at ambient 108 temperature applicable in the field, were adopted in this investigation to determine the 109 benefits of early age moisture available for curing on the strength and pore properties of 110 AACMs.

A potassium-based activator used in AACMs reduces the mean pore diameter more than a sodium-based activator [4] while the total porosity of an alkali activated blast furnace slag (BFS) is reduced by the inclusion of a high modulus (more concentrated) activator and low water content in the mix [2]. The influence of chemical admixtures such as retarder and shrinkage reducing admixtures on the pore properties of AACMs is not known. This aspect together with the influence of activator dilution on the pore properties of AACM mortar is investigated.

118 Mercury intrusion porosimetry (MIP) is the common test method for investigating the 119 microstructure of concrete. This is performed by applying mercury under high pressure 120 through concrete pores. The method is based on the "non-wetting" property of mercury on the 121 walls of the concrete pores. Mercury intrusion into the concrete matrix is suitable for pores 122 within the range of 0.003 to 400  $\mu$ m [16]. This method is used for analysing the accessible 123 pores within the AACM and the control PC mortar samples in this investigation.

124 This paper is part of a comprehensive durability investigation of AACMs (mortar and 125 concrete) being undertaken by the authors. It characterises the basic pore-properties of the 126 material to provide a deeper understanding of the durability properties of reinforced AACM 127 concrete.

128 **2.0** Experimental programme

#### 129 2.1 Materials and mixes

130 The control PC mortar had a composition of 1: 2.1 (by weight) of CEM 1 cement to CEN 131 standard sand with a water/cement ratio of 0.47. The CEM 1 cement used is 42.5 Portland 132 cement and it was supplied by Frank Key group, Sheffield, UK. The PC mortar was produced 133 in accordance with BS EN 196-1:2016 [17]. The corresponding AACM 1 and 2 mortar mixes 134 comprised of GGBS binder, sodium silicate and hydroxide based activator, fine aggregate of 135 80% particle size passing 1mm sieve, liquid/binder ratio of 0.47 (alkali activator + water), a shrinkage reducing admixture SRA and retarder R42. The fresh AACM 1 and 2 mortar mixes 136 137 were designed to achieve a flow of about 15 cm using the flow table test method [18]. The 138 shrinkage reducing admixture SRA was added to reduce the shrinkage of AACMs while retarder R42 was added to increase the setting time. AACM 1 and 2 mixes were investigated 139 140 to provide optimum properties of the fresh and hardened material for practical applications. 141 However, AACM 1a and 2a mixes were also prepared with the same mix proportions but 142 without admixtures to provide data for direct comparison with the PC mix which also did not 143 contain admixtures. The mix compositions for AACM 1, 2, 1a, 2a and the control PC mortars 144 are presented in Table 1. Table 2 shows the chemical composition of Portland cement (PC) 145 and ground granulated blast-furnace slag (GGBS) binders used in the tests.

146 The average 28-day strength of the AACM and control PC mixes were designed to be fairly 147 similar under wet curing, based on trial mixes. Wet curing is the standard method for quality

148 testing of concrete [5]. The different curing methods adopted in this research are detailed in

149 section 2.2.

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Table 1: Composition of the AACM and control PC mortars

Mix	Binder (%)	Fine Agg.	Liquid	Liquid/Binder	Activator	R42	SRA
		(%)	(%)	Ratio	Dilution (%)	(% binder)	(% binder)
AACM 1	49	28.0	23.0	0.47	0	0.75	2.0
AACM 2	49	28.0	23.0	0.47	7.76	0.75	2.0
AACM 1a	49	28.0	23.0	0.47	0	-	-
AACM 2a	49	28.0	23.0	0.47	7.76	-	-
Control PC	28	59.0	13.0	0.47	-	-	-

151

Table 2: Chemical composition of Portland cement and GGBS binders

Chemical component	SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	Fe <sub>2</sub> O <sub>3</sub>	CaO	MgO	K <sub>2</sub> O	Na <sub>2</sub> O	TiO <sub>2</sub>	$P_2O_5$	MnO	SO <sub>3</sub>
PC (mass %)	11.1	8.35	3.16	64.2	2.09	1.19	0.227	1.88	2.01	2.14	3.64
GGBS (mass %)	28.6	12.4	5.7	42.3	6.1	0.8	0.4	1.78	< 0.1	0.3	0.08

152 Sodium silicate activator of molarity 6.5 mol/L and modulus 2% was used for the AACM 153 mixes to provide optimum viscosity for controlling workability and setting time [19]. The 154 molarity of NaOH activator used was 4.8 mol/L. The combined molarity of the activators was 155 at the lower end of values used by other researchers [20] for a similar activator combination. 156 The activator for AACM 2 mixes was diluted with water at 7.76% (Table 1). The retarder R42 is made from a blend of high grade Polyhydroxycarboxylic acid derivatives while the 157 158 shrinkage reducing admixtures (SRA) is made from Alkyl-ether. Each admixture contained 159 less than 0.1% chloride ion and 3.5% sodium oxide.

160 2.2 *Casting and curing* 

161 The GGBS binder and saturated surface dry fine aggregate were placed in a 12 litre, 3 speed 162 Hobart mixer. They were mixed for 30 seconds at the lowest speed (option-1) to avoid 163 dispersing the powder into the atmosphere. The liquid component containing the activator 164 and retarder R42 were slowly added to the mix. Mixing continued for 2 minutes until a 165 uniform texture was produced. The shrinkage reducing admixture SRA was then slowly added while mixing continued. The mortar was further mixed for 1 minute before stopping. 166 167 The control PC mortar and the AACM mixes without admixtures were prepared in a similar 168 manner without adding retarder R42 and shrinkage reducing admixture SRA. The AACM 169 and control PC mortars were cast in 75 x 75 x 75 mm steel cube moulds which had been lightly oiled to prevent the hardened mortar from sticking to the surface. Three mortar cubes 170 were cast for each mix. The specimens were left covered in the moulds with polythene sheets 171 for 24 hours at room temperature of  $20 \pm 2$  <sup>0</sup>C and a relative humidity of about 65%. The 172 173 specimens were demoulded 24 hrs after casting and were exposed to three different curing 174 regimes.

175 Three practical curing regimes (wet/dry, wet and dry), commonly applied in the construction 176 field, were adopted in this research work as shown in Table 3. Wet/dry curing involved placing the mortar cubes in water at a temperature of  $20 \pm 2^{\circ}C$  for 3 days immediately after 177 demoulding (24 hrs after casting), followed by dry curing in the laboratory air at a 178 179 temperature of  $20 \pm 2^{\circ}$ C and approximately 65% relative humidity for 24 days (total curing 180 period of 28 days). Wet curing was provided by placing the cube specimens in water at a temperature of  $20 \pm 2$  <sup>0</sup>C for 27 days immediately after demoulding. Dry curing of the 181 mortars was provided by exposing them in the laboratory air at a temperature of  $20 \pm 2^{0}$ C and 182 183 approximately 65% relative humidity for 27 days after demoulding. When cured in the 184 laboratory air (during wet/dry and dry curing), the specimens were securely covered with 185 polyethene sheets to prevent moisture loss from the concrete surface representing site practice where different methods can be used to prevent rapid drying of concrete such as applying 186 curing membranes or covering concrete surface with wet hessian. 187



Table 3: Curing methods used for the AACM and control PC mortars

Samples	Age(days)	Wet/dry	Wet	Dry

-		(Curing Medium)	(Curing Medium)	(Curing Medium)
75 mm cubes	0-3	Water	Water	Air
	3-28	Air	Water	Air
MIP samples	28-31	Oven (50 <sup>°</sup> C)	Oven (50 <sup>0</sup> C)	Oven (50 <sup>°</sup> C)
	31-34	Desiccator	Desiccator	Desiccator

#### 189 2.3 Test procedure

#### 190 2.3.1 Compressive strength

191 The compressive strength and density of the 75 mm mortar cubes were determined after 28 192 days curing under wet/dry, wet and dry regimes (Table 3). The compressive strength tests on 193 the cubes were conducted in accordance with BS EN 12390-3:2009 [21]. The density of the 194 cubes was determined according to BS EN 12390-7:2009 [22]. Three specimens were used to 195 determine the density and compressive strength of the 75 mm cubes. A loading rate of 3 196 MPa/min was applied during the compression testing and a post peak of 30% failure load was 197 programmed into the compression machine to prevent complete disintegration of the crushed 198 specimen. Samples for the Mercury intrusion porosimetry (MIP) tests were obtained from 199 these crushed samples.

# 200 2.3.2 Mercury Intrusion Porosimetry

Mercury intrusion porosimetry (MIP) test samples, with a weight of 1 - 2 g and average length of 1 cm, were obtained from the crushed 75 mm mortar cubes. Errors caused by hysteresis and entrapment of moisture during MIP testing was minimised by controlling the dimensions, mass and drying of all test samples [23] as described here. The mercury intrusion porosimetry (MIP) test samples were dried in an oven at a temperature of  $50^{\circ}$ C for 3 days (28- 31 days age) as shown in Table 3. Oven drying at a higher temperature than  $50^{\circ}$ C causes microcracking which may adversely affect the test results [24]. After oven drying for 3 days, 208 the samples were placed in a desiccator for another 3 days to reduce their temperature to 209 about  $20^{\circ}$ C. The desiccator had silica gel at the bottom to further assist with removing 210 absorbed water and preventing moisture migration from the air. The drying and cooling were 211 carried out to remove absorbed water within the mortar pore system, which can obstruct its 212 accessible porosity during MIP testing.

MIP testing was performed using a Pascal 140/240 Porosimeter which is in two parts. Pascal 140 applies pressure of up to 100 MPa and Pascal 240 applies pressure of up to 200 MPa to aid the intrusion of mercury through pore sizes down to 0.007  $\mu$ m. The Pascal 140/240 Porosimeter measures pore sizes within the range of 0.007 to 100  $\mu$ m. Its computer microprocessor translates the data collected on the applied pressures to pore radius using the Washburn equation (equation 1).

$$p = \frac{2y\cos\phi}{r}$$
 1

219 Where *p* is the absolute applied pressure; *r* is the pore radius; *y* is the mercury surface 220 tension (= 0.48N/m);  $\phi$  is the contact angle (= 140<sup>0</sup>).

The limitation of Washburn equation is the assumption that the pores in the concrete matrix are cylindrical in shape which has been criticised by researchers [23]. The graphs of pore sizes and pore distribution were obtained at the end of the mercury intrusion porosimetry analysis. The MIP analysis was performed on three test samples for each curing condition for the AACM and control PC mortars.

- 226 **3.0** Results and discussion
- 227 3.1 Compressive strength and density

228 The average value of the compressive strength and density of the three specimens tested per
229 mix had less than 5% variation.

230 3.1.1 Effect of curing regimes on density and compressive strength

The densities of the 75 mm mortar cubes at 28 days age are between 2.22 - 2.35 g/cm<sup>3</sup> for wet/dry curing, 2.10 - 2.23 g/cm<sup>3</sup> for dry curing and 2.07 - 2.15 g/cm<sup>3</sup> for wet curing. The corresponding 28day compressive strength for the AACM and control PC mortars under wet/dry, wet and dry curing (Table 3) are shown in Fig. 1.



235

Fig. 1: Compressive strength of AACM and control PC mortars under wet/dry, wet anddry curing at 28 days

238 The compressive strengths of AACM 1 are 70.9 MPa for wet/dry curing, 57.9 MPa for wet 239 curing, and 61.2 MPa for dry curing. The corresponding values for AACM 2 are 65.2 MPa, 240 46.4 MPa and 54.6 MPa. Similar trend is observed in AACM 1a and 2a. The wet/dry curing 241 method achieved the highest strength for all the AACM mortars. This curing method involved 3 days wet curing at  $20 \pm 2^{\circ}$ C followed by 24 days in the laboratory air ( $20 \pm 2^{\circ}$ C, 242 243 65% R.H.). The dry curing method of AACM 1 and 2 mortars (27 days curing in laboratory 244 air at  $20 \pm 2^{\circ}$ C, 65% R.H.) gave lower strength than the wet/dry method while wet curing (27) 245 days curing in water at  $20 \pm 2^{\circ}$ C) gave the least compressive strength. The wet/dry curing of AACM mixes gave the highest strength due to the formation of more crystalline 246 247 geopolymerisation products [2,11].

248 The effect of curing methods on the control PC mortar contrasts the AACM mortars by 249 providing the maximum compressive strength under wet curing. The availability of moisture 250 in the PC mortar supported cement hydration which produced more strength. The geopolymer 251 reactions in AACMs do not rely on moisture to the same extent as the hydration reactions in PC. The control PC mortar recorded the highest compressive strength of 51.4 MPa (Fig. 1) 252 253 under wet curing followed by 43.4 MPa under wet/dry curing, which is slightly higher than 254 42.3 MPa under dry curing as shown in Fig. 1. The results of the control PC mortar are 255 consistent with other research which shows a similar effect of curing conditions on the 256 strength of PC concrete [25,26]. The relative humidity in the PC capillary pores is maintained 257 above 80% when cured in water, which favours hydration reactions [5]. There is little loss of 258 strength when PC concrete is cured in a moist medium above 80% R.H.

The 28 day strengths of AACM 1, 2 (both with retarder and shrinkage reducing admixture) and PC mortar (without the admixture) under wet curing are 57.4 MPa, 52.3MPa and 51.4MPa respectively. The compressive strengths under wet curing of AACM mortars 1a and 2a (both without admixture) average 46.4 MPa and 46.2MPa respectively compared with 51.4MPa for wet cured PC mortar. The average strength of the AACM mixes is similar to the PC mortar (control) mix under wet curing whereas their strength is much higher under partially dry curing conditions (wet/dry and dry) which are encountered on site.

266 *3.1.2 Effect of activator dilution on compressive strength* 

Fig. 1 shows the effect of activator dilution on the compressive strength of the AACM mortars under wet/dry, wet and dry curing. The compressive strength decreases with increasing dilution of activator. For example, the compressive strengths of AACM 1 mortar were 70.9 MPa, 57.9 MPa and 61.2 MPa compared with 65.2 MPa, 46.4 MPa and 54.6 MPa for AACM 2 mortar (7.76% dilution) under wet/dry, wet and dry curing respectively (Fig. 1).

Activator concentration is an effective factor in the geopolymerisation process in AACM concrete. A reduction in strength has been reported when the activator concentration is not sufficient for the geopolymerisation reaction [11,27]. Similarly, high activator concentration will delay the AACM formations due to excessive cations, thereby limiting their mobility and potential to interact with the reactive pozzolanic species [27]. This reverse effect of strength reduction with increasing concentration of the alkali activator was, however, not observed in this study.

- 279 *3.2 Pore size distribution*
- 280 *3.2.1* Unimodal and bimodal pore distribution

The relationship between pore size and differential pore volume for AACM 1, 2 and the control PC concrete under wet/dry, wet and dry curing are shown in Figures 2, 3 and 4 respectively.





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Fig. 2: Pore size distribution for AACM 1, 2 and control OPC mortars under wet/dry curing





Fig. 3: Pore size distribution for AACM 1, 2 and control OPC mortars under wet curing





289

Fig. 4: Pore size distribution for AACM 1, 2 and control OPC mortars under dry curing

290 The figures show the range of pore diameters under which significant levels of differential 291 pore volumes are observed and the range when the differential pore volume is at or near zero. The diameter zones showing significant differential pore volume represent porosity whereas 292 293 the range indicating zero differential pore volume represents a non-porous zone. Based on 294 these criteria, it can be observed that the PC mortar has a unimodal pore distribution under 295 the three curing conditions shown in Figures 2, 3 and 4. The pore distribution is referred to as 296 unimodal when a single range of pore volume is observed within the differential pore volume 297 graphs [2,4]. The pore volume of the PC mortar falls within the range of 0.01 to 0.1 µm pore

298 diameter. Other studies on the microstructure of PC matrix also show a unimodal pore size 299 distribution with most of the pore volume within the range of 0.01 to 0.1 µm pore diameter 300 [2,4]. On the other hand, a double range of pore diameters with significant differential pore 301 volume which are separated by a diameter range with nearly zero differential pore volume is 302 categorized as a bimodal pore distribution [2,4]. These pore sizes are normally observed 303 between two separate zones of > 1  $\mu$ m and <0.02  $\mu$ m [2]. Figures 2, 3 and 4 show that 304 AACM 1 and 2 mortars fall under this category with significant porosity observed at  $> 1 \mu m$ 305 and  $<0.02 \mu m$  while there is little porosity between these pore size ranges.

The differences in the effects of wet/dry, wet and dry curing on the differential pore volumes over the pore diameter ranges in Figures 2, 3 and 4 have been quantified by determining the pore system parameters such as porosity and are discussed fully in section 3.3.

# 309 *Wet/dry curing*

310 The pore sizes in AACM 1 and 2 mortars subjected to wet/dry curing (Fig. 2) show a bimodal 311 pore size distribution. The first range of pore diameters showing significant differential pore 312 volumes in AACM 1 mortar is <0.02 µm while the second range is predominantly between 313 0.2 to 4.5  $\mu$ m. There is insignificant differential pore volume between 0.02 and 0.2  $\mu$ m 314 diameter. AACM 2 mortar shows a similar trend of bimodal pore distribution, the pore 315 diameters range from under 0.03 µm to greater than 0.2µm. On the other hand, the control PC 316 mortar shows a unimodal pore size distribution (Fig. 2) of diameter lesser than 0.3  $\mu$ m. The 317 bimodal distribution of pores in AACM 1 and 2 mortars extends to larger pore diameters than 318 the control PC mortar; however the large pore size zone is isolated due to the bimodal 319 distribution, which will affect porosity as discussed in section 3.3.

320 Wet curing

The bimodal pore size distribution in AACM 2 mortar is less pronounced under wet curing (Fig. 3) than under wet/dry (Fig. 2) or dry curing (Fig. 4). There is significant continuity of

323 pores between pore diameters 0.01 to 100 µm (particularly AACM 2) which is reflected by 324 the differential pore volume remaining slightly above zero in this pore diameter range. This 325 does not appear under both wet/dry and dry curing. Therefore some interconnection between 326 the gel pores ( $< 0.05 \,\mu$ m) and capillary pores (0.1 to 100  $\mu$ m) is likely in wet cured AACM 2 327 mortar. The interconnection is represented by the regular distribution of peaks throughout the 328 range of pore sizes 0.01 to 100 µm (particularly AACM 2). The less solid gel products 329 produced in AACM 2 mortar due to the higher activator dilution may be insufficient to block 330 the interconnecting pores. Another reason for pore continuity could be the leaching of alkali 331 cations into the curing solution thereby resulting in loss of alkali concentration needed for 332 geopolymerisation reaction [28]. A slight degree of hydration reactions may also be a likely 333 contributor to the interconnection of pores under wet curing in the AACM 2 due to the high 334 degree of moist curing.

# 335 Dry curing

336 AACM 1 and 2 mortars under dry curing (Fig. 4) show a bimodal pore size distribution 337 similar to wet/dry curing. The first range of pores in AACM 1 mortar are less than 0.05 µm 338 while the second range of the bimodal pore size distribution is greater than 0.1 µm and 339 extends to 100 µm diameter. AACM 2 mortar has slightly different pore ranges of less than 340 0.02 µm and greater than 0.1 µm and extends to 100 µm diameter. The control PC mortar has 341 a unimodal pore distribution between 0.01  $\mu$ m to approximately 2  $\mu$ m, the pore diameter 342 range is higher than under wet/dry and wet curing. The PC mortar shows significant 343 differential pore volume within the dip between the two peaks in Figure 4, unlike the AACMs 344 where the differential pore volume reaches near zero between the bimodal peaks.

345 3.3 Pore system parameters

346 Pore system parameters are frequently used in analytical and empirical property-347 microstructure relationship models [29,30]. These parameters are derived from the 348 cumulative porosity curves and logarithmic differential pore volume curves. They are classified as intrudable porosity  $\boldsymbol{\Phi}_{in}$ , critical pore diameter  $d_c$ , threshold pore diameter  $d_{th}$  and 349 350 porosity [29,30]. The location of  $\boldsymbol{\Phi}_{in}$  is shown on the cumulative pore volume curve for both PC and AACM mortars (Figures 5 and 6). The location of  $d_c$  and  $d_{th}$  is shown on the 351 corresponding differential pore volume curves in these figures. Two locations of  $d_c$  and  $d_{th}$ 352 are given on the bimodal graphs of AACM mortars. The porosity of cementitious material is 353 354 the percentage of pores in the total bulk volume of the mortar whereas intrudable porosity 355 represents only the pore volumes which are accesible to mecury intrusion [29]. The values of 356 these pore parameters are presented in Table 4. The porosity and pore volumes of AACM and 357 control PC mortars with and without shrinkage reducing admixture SRA and retarder R42 are 358 presented in Table 4.





Fig. 5: Definition of Pore System Parameters in OPC Mortar (Authors' data)





Fig. 6: Definition of Pore System Parameters in AACM Mortar (Authors' data)



	Mix	Curing	Porosity		Pore diar	Pore diameters (µm)		Pore volumes (%)	
			Intrudable	Porosity	Critical	Threshold	Gel	Capillary	
			$(mm^{3/g})$	(%)	$d_{c1}$	$d_{th1}$			
		Wet/dry	29.68	4.64	0.0073	0.013	0.60	4.04	
res	AACM 1	wet	38.14	6.53	0.0073	0.014	0.66	5.87	
nixtu		dry	53.44	9.90	0.0075	0.025	1.42	8.48	
n adı		Wet/dry	30.17	6.67	0.0073	0.021	0.98	5.69	
Witl	AACM 2	wet	45.48	8.02	0.0081	0.034	0.91	7.11	
		dry	59.13	10.70	0.0085	0.048	1.65	9.05	
s		Wet/dry	44.26	7.71	0.0081	0.018	0.26	7.45	
ture	AACM 1a	wet	51.66	9.05	0.0082	0.019	0.97	8.08	
dmix		dry	65.64	11.93	0.0084	0.032	1.92	10.01	
ut a		Wet/dry	46.92	9.14	0.0086	0.027	1.18	7.96	
'itho	AACM 2a	wet	53.83	10.13	0.0087	0.032	1.80	8.33	
H I		dry	68.05	11.69	0.0089	0.051	1.96	9.73	
		Wet/dry	81.62	14.02	0.049	0.35	10.83	3.19	
	Control PC	wet	68.16	13.30	0.016	0.28	8.6	4.70	
		dry	93.51	17.43	1.07	1.12	10.58	6.85	

364 *3.3.1* Intrudable pore volume

The volume of intrudable pores (intrudable pore volume) within AACM 1, 2 and the control PC mortar matrix was determined under wet/dry, wet and dry curing from the cumulative 367 pore volume curves as shown in Figures 7, 8 and 9 respectively. Figure 10 shows the



368 intrudable pore volume.



Fig. 7: Intrudable porosity for AACM 1, 2 and control OPC mortars under wet/dry curing



Fig. 8: Intrudable porosity for AACM 1, 2 and control OPC mortars under wet curing





Fig. 9: Intrudable porosity for AACM 1, 2 and control OPC mortars under dry curing



375

Fig.10: Intrudable pore volume for AACM 1, 2 and OPC mortars under wet/dry, wet anddry curing

378 Wet/dry curing

AACM 1 and 2 mortars have lower intrudable pore volume of 29.68 mm<sup>3</sup>/g and 30.17 mm<sup>3</sup>/g respectively compared with 81.62 mm<sup>3</sup>/g for the control PC mortar as shown in Figures 7 and 10. The application of wet/dry curing to AACM concrete was observed to enhance its resistance to chloride ingress under exposure to salt laden environment [15]. The initial 3 days wet curing followed by the 24 days dry curing in laboratory air under the wet/dry curing
method (Table 3) resulted in a lower intrudable pore volume in AACMs.

The intrudable pore volume for AACM 1 mortar is similar to the AACM 2 mortar (7.76% activator dilution) at 29.68 mm<sup>3</sup>/g and 30.17 mm<sup>3</sup>/g respectively under wet/dry curing as shown in Figures 7 and 10.

388 Wet curing

The wet curing of the control PC mortar resulted in an intrudable pore volume of 68.16 389  $mm^3/g$  (Figures 8 and 10) compared with 81.62  $mm^3/g$  and 93.51  $mm^3/g$  for wet/dry and dry 390 391 curing respectively (Figures 7 and 9). The wet curing method usually provides the best 392 mechanical and durability properties for PC concrete due to saturation of its pore spaces with 393 water which aid cement hydration. Powers [31] and Patel et al [32] observed that the 394 hydration of PC concrete is greatly reduced when the relative humidity within the pore spaces drops below 80%. Since both the wet/dry and dry curing methods exposed the control PC 395 396 mortar to laboratory air (R.H. 65%) before cement paste hydration was completed, it resulted 397 in more intrudable pores than under wet curing.

The intrudable pore volume of AACM 1 mortar ( $38.14 \text{ mm}^3/\text{g}$ ) under wet curing (Fig. 8) is more than ( $29.68 \text{ mm}^3/\text{g}$ ) under wet/dry curing (Fig. 7). AACM 2 mortar also shows a similarly higher intrudable pore volume under wet curing (Fig. 8).

401 Dry Curing

AACM 1 mortar has an intrudable porosity of 53.44 mm<sup>3</sup>/g compared with 93.51 mm<sup>3</sup>/g for
the control PC mortar under dry curing (Figures 9 and 10). The results presented in Figures 7,
8, 9 and 10 indicate that AACM 1 and 2 mortars possess significantly less intruded pore
volume than the control PC mortar under the three curing conditions.

406 3.3.2 Critical and threshold pore diameters

407 The critical and threshold pore diameters of a concrete matrix influence its durability 408 properties [29,30,33]. Lower values of these parameters represent enhanced durability 409 properties. AACM mortars under wet/dry curing had the lowest critical and threshold pore 410 diameters, followed by wet and dry curing. For example, AACM 1 had critical pore diameter 411 of 0.0073 µm both under wet/dry and wet curing and 0.0075 µm under dry curing (Table 4). 412 The corresponding threshold pore diameter was 0.013 µm, 0.014 µm and 0.025 µm. The higher dilution of alkali activator in AACM 2 increased the critical and threshold pore 413 414 diameters as shown in Table 4. This pattern is similar for the three curing regimes wet/dry, 415 wet and dry. The difference in values under the three curing conditions is more pronounced 416 for the AACM 2 mix than AACM 1.

417 On the other hand, PC mortar under wet curing has the lowest critical and threshold pore 418 diameters compared with wet/dry and dry curing. The critical pore diamters are 0.016  $\mu$ m, 419 0.049  $\mu$ m and 1.07  $\mu$ m under wet, wet/dry and dry curing respectively. The corresponding 420 threshold pore diameters are 0.28  $\mu$ m, 0.35  $\mu$ m and 1.12  $\mu$ m in Table 4. The pore blocking 421 effect in PC concrete was proposed by *Khatib and Mangat* [34] for the wet curing regime. 422 The availability of water during curing allowed for more hydration to take place resulting in 423 the formation of more calcium silicate gel thereby reducing the critical and threshold pores.

AACM mortars displayed lower critical and threshold pore diameters than PC concrete. Therefore, the durability properties of AACM mortars are expected to be superior to the PC mortar. Early results from a comprehensive durability study by the authors indicate that chloride diffusion in the PC mortar is greater than the AACM mixes [15]. The relationship between chloride diffusion and the critical and threshold diameters will be addressed in a future paper by the authors.

430 3.3.3 Porosity of AACM and PC mortar

The relationship between porosity and incremental pore diameter range of AACM 1 and the
control PC mortars under wet/dry, wet and dry curing are shown in Figures 11, 12 and 13
respectively.



435 Figure 11: Relationship between porosity and incremental pore diameter (µm) range for AACM 1 and

436 control OPC mortars under wet/dry curing.



437







Figure 13: Relationship between porosity and incremental pore diameter (μm) range for AACM 1 and
 control OPC mortars under dry curing.

443 The figures show that the porosity of AACM 1 mortar is distributed along a limited range of 444 pore diameters with more significant porosity at larger diameters. On the other hand, the 445 control PC mortar has its porosity distributed along the whole range of pore diameters with more significant porosity at smaller diameters. The AACM mortars show a distinctively large 446 447 volume of pores within the capillary pore zone (>0.16 µm) while PC mortars have a large 448 volume of pores within the gel pore zone (<0.16 µm). For example, the percentage of 449 capillary pore volume is 4.04% and 3.19% for AACM 1 and PC mortars respectively under 450 wet/dry curing. The corresponding gel pore volume is 0.60% and 10.83%



452 Figure 14: Effective porosity of AACM and control OPC mortars under wet/dry, wet and dry 453 curing.

451

Fig. 14 shows the effective porosity of AACM 1 and the control PC mortars under wet/dry, wet and dry curing. This is the summation of the incremental pore volumes in Figures 11, 12 and 13. The porosity of AACM 1 mortar is much lower than the control PC mortar despite the presence of larger pores in AACM 1 mortar. The porosity of AACM 1 mortar is 4.64%, 6.53% and 9.90% compared with 14.02%, 13.30% and 17.43% for the control PC mortar under wet/dry, wet and dry curing respectively. The porosity for the corresponding AACM 2 mortar is 6.67%, 8.02% and 10.70%.

The porosity of AACM mix 1a is 7.71%, 9.05% and 11.93% for wet/dry, wet and dry curing respectively. Each value is significantly lower than the corresponding value for PC mortar. AACM mix 1a did not incorporate any admixtures (SRA and R42) and, therefore, is directly comparable with the PC mortar. The porosity of AACM mix 2a is similarly lower than the PC mortar. The results confirm the lower porosity of the AACM mixes.

466 The wet/dry curing is optimum for AACM mortar while wet curing is best for the control PC467 mortar, the latter being a well-established fact.

468 RILEM TC 224 [2] reported that the total porosity (i.e. summation of both gel and capillary 469 pores) of AACM is somewhat similar or sometimes higher than comparative PC. The 470 contrary results of this study show that the total pore volume was higher in PC mortar than in 471 AACM mortar. Nevertheless, a higher capillary pore volume was observed in AACMs while 472 their gel pore volume was much lower than PC mortar. For example, AACM 1 and 2 mortar 473 has higher percentage of capillary pore volume of 4.04% and 5.69% respectively compared 474 with 3.19% for the control PC mortar under wet/dry curing (Table 4, Figures 11, 12 and 13). 475 On the other hand, the percentage of gel pore volume of 0.60% and 0.98% in AACM 1 and 2 476 respectively was much lower than 10.83% for PC mortar under wet/dry curing. A similar 477 trend is observed under wet and dry curing (Table 4, Figures 11, 12 and 13).

# 478 3.3.4 Strength-porosity relationship of AACM mortars

Strength and porosity data of AACM mixes 1, 2, 1a and 2a (Table 1) are considered in this section together with the data for similar AACM mixes which were prepared with other activator dilutions under 4%. The same mix proportions and test procedures outlined in the paper were used for these mixes. The strength-porosity relationship of all the AACM mortars under wet/dry, wet and dry curing together with the combined plot of wet/dry and dry curing is shown in Figure 15. The best fit lines provide a non-linear plot according to the following relationship proposed for porous materials by *Balshin* [35]

$$6 = 6_0 (100\% - P)^n \qquad 2$$

486 Where  $\mathfrak{G}$  = Compressive strength,  $\mathfrak{G}_0$  = Compressive strength of fully dense material at 0% 487 porosity, *P* = Porosity and *n* = Constant.





Figure 15: Strength- porosity relationship of AACM mortars under different curing.

490 A non-linear regression analysis of the data in Fig. 15 using equation 2 provided the491 following best-fit equation for the combined wet/dry and dry cured AACM mixes.

$$6 = 143.6(100\% - P)^{0.37}$$

492 With a coefficient of correlation, 
$$R^2 = 0.71$$

493 The corresponding relationship for the wet cured AACM mortars is given by the following494 equation:

$$6 = 136.7(100\% - P)^{0.42}$$

495 With a coefficient of correlation,  $R^2 = 0.68$ 

AACM mortar subjected to wet/dry curing had the lowest porosity and highest strength. The initial wet curing aided the production of more geopolymerisation while the subsequent dry curing resulted in increased compressive strength [11]. AACM mortar subjected to wet/dry and dry curing had a higher strength than wet curing in the same range of porosity as shown in Fig. 15. For example from the best-fit relationships, the compressive strength at a porosity 501 of 10% is 61.2 MPa and 52.7 MPa under wet/dry and wet curing respectively. This indicates 502 an additional effect to porosity which enhances the strength of dry cured AACMs. This can 503 be due to enhanced strength of the geopolymerisation products with dry curing, including 504 increased bond within the geopolymer structure. The initial wet curing also favours the 505 hydration reactions of any high calcium compounds in the AACM binders. Therefore, the 506 optimum curing for strength-porosity relationship in AACMs is achieved under wet/dry 507 curing.

#### 508 **4.0 Conclusions**

509 The paper presents an investigation on the effect of wet, wet/dry and dry curing on the pore 510 size distribution, porosity and strength of an alkali activated cementitious (AACM) mortar 511 and a comparative PC mortar. The AACM mixes were made with and without admixtures 512 (SRA and R42). The following conclusions can be drawn from the results of the study:

513 1) The wet/dry curing regime produces the highest compressive strength in AACM
514 mortars while it is wet curing for the control OPC mortar. For example, the 28 day
515 strength of AACM 1 mix under wet/dry, wet and dry curing was 70.9MPa,
516 57.9MPa and 61.2MPa respectively while it was 65.2 MPa, 46.4MPa and 54.6MPa
517 for AACM 2. The corresponding values of 43.4 MPa, 51.4 MPa and 42.3 MPa
518 were observed for OPC mortar.

519 2) AACM mortar develops a bimodal micropore distribution which is influenced by
520 the type of curing and the activator dilution. Wet/dry curing (3 days in water
521 followed by 24 days in air) provides an optimum pore structure for AACM. OPC
522 mortar develops a unimodal pore structure which is optimum under wet curing.

3) Higher activator concentration, within the range used, results in improved strength
and a more refined pore structure. For example, the strength of AACM mortar
under wet/dry curing with 0% activator dilution (AACM 1) is 70.9MPa compared

- with 65.2MPa for AACM mortar with 7.76% activator dilution (AACM 2). Their
  corresponding porosity is 4.64% and 6.67%.
- 4) Wet/dry curing of AACM mortar produces the lowest porosity and pore volume.
- The porosity of AACM mixes is much lower than the control OPC mortar for each curing condition. For example, AACM 1 mix under wet/dry, wet and dry curing had a porosity of 4.64%, 6.53% and 9.90% respectively. In comparison, the control OPC mortar under wet/dry, wet and dry curing had a porosity of 14.02%, 13.30% and 17.43% respectively, giving the lowest porosity under wet curing.
- 5) The threshold pore diameters of AACM mixes, which influence durability
- 535 properties, are at least an order of magnitude lower than for the control OPC mixes. 536 For example, the threshold diameters for AACM 1 mortar under wet/dry, wet and 537 dry curing are 0.013  $\mu$ m, 0.014  $\mu$ m, and 0.025  $\mu$ m respectively. The corresponding 538 values for the control OPC mortar are 0.35  $\mu$ m, 0.28  $\mu$ m, and 1.12  $\mu$ m.
- 539 6) The volume of gel pores, within the range of 0.005  $\mu$ m to 0.01  $\mu$ m, in AACM
- 540 mortars is less than the control OPC mortar. On the other hand, the volume of 541 capillary pores, within the range of 0.01  $\mu$ m to 100  $\mu$ m pore diameter, is higher in 542 AACM mortars. However, the total porosity (summation of both gel and capillary 543 pores) is higher in the control OPC mortar than in AACM mortars. For example, 544 the gel porosity in AACM 1 and OPC mortar is 0.60% and 10.83% respectively 545 while their corresponding capillary porosity is 4.04% and 3.19% under wet/dry 546 curing.
- 547 7) The inclusion of a shrinkage reducing and retarding admixture in AACMs
- 548 enhances strength and produces a more refined pore structure particularly under
  - 549 wet/dry and dry curing. AACM mortars, both with and without admixtures, have

- superior strength and a more refined pore structure than the control OPC mortarunder wet/dry and dry curing.
  - 8) The strength-porosity relationship of AACM mortars under combined wet/dry and dry curing is as follows: 6 = 143.6(100% P)<sup>0.37</sup> with a coefficient of correlation R<sup>2</sup> = 0.71. The relationship under wet curing is given by: 6 = 136.7(100% P)<sup>0.42</sup> with R<sup>2</sup> = 0.68. For any given porosity, the strength is lower under wet curing.

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# 559 **References**

- 560 [1] P.C. Aïtcin, Cements of yesterday and today concrete of tomorrow, Cem. Concr. Res. 30
- 561 (2000) 1349–1359. doi:10.1016/S0008-8846(00)00365-3.
- John L. Provis, J.S.J. van Deventer, Alkali-Activated Materials State-of-the-Art Report,
  RILEM TC 224-AAM, 2014.
- 564 [3] N.A. Madlool, R. Saidur, M.S. Hossain, N.A. Rahim, A critical review on energy use and
  565 savings in the cement industries, Renew. Sustain. Energy Rev. 15 (2011) 2042–2060.
- 566 doi:10.1016/j.rser.2011.01.005.
- 567 [4] P. Mangat, P. Lambert, Sustainability of alkali-activated cementitious materials and
- 568 geopolymers, in: Sustain. Constr. Mater., Elsevier Ltd, 2016: pp. 459–476. doi:10.1016/B978-
- 569 0-08-100370-1.00018-4.
- 570 [5] A.M. Neville, Properties of Concrete, Pearson Education Limited, 2011.

- 571 [6] R. Kumar, B. Bhattacharjee, Assessment of permeation quality of concrete through mercury
- 572 intrusion porosimetry, Cem. Concr. Res. 34 (2004) 321–328.
- 573 doi:10.1016/j.cemconres.2003.08.013.
- 574 [7] K.D. Stanish, R.D. Hooton, M.D.A. Thomas, Testing the Chloride Penetration Resistance of
  575 Concrete: A Literature Review, 1997.
- 576 [8] J.A. Larbi, Microstructure of the interfacial zone around aggregate particles in concrete,
  577 Heron. 39 (1993) 1–69.
- 578 [9] D. Mindess, Sidney; Young, J. Francis; Darwin, Concrete, Prentice Hall, Pearson Education,
  579 Inc. Upper Saddle River, NJ 07458, U.S.A., 2003.
- 580 [10] Y. Li, J. Li, Capillary tension theory for prediction of early autogenous shrinkage of self-
- 581 consolidating concrete, Constr. Build. Mater. 53 (2014) 511–516.
- 582 doi:10.1016/j.conbuildmat.2013.12.010.
- 583 [11] J.C. Petermann, A. Saeed, M.I. Hammons, Alkali-activated geopolymers: a literature review
  584 air force research laboratory materials and manufacturing directorate, 2010.
- 585 [12] T. Medina-Serna, S. Arredondo-Rea, J. Gómez-Soberón, C. Rosas-Casarez, R. Corral-Higuera,
- 586 Effect of curing temperature in the alkali-activated blast-furnace slag paste and their structural
- 587 influence of porosity, Adv. Sci. Technol. Res. J. 10 (2016) 74–79.
- 588 doi:10.12913/22998624/64021.
- 589 [13] G. Fang, W.K. Ho, W. Tu, M. Zhang, Workability and mechanical properties of alkali-
- 590 activated fly ash-slag concrete cured at ambient temperature, Constr. Build. Mater. 172 (2018)
- 591 476–487. doi:10.1016/j.conbuildmat.2018.04.008.
- 592 [14] A. Cwirzen, R. Engblom, J. Punkki, K. Habermehl-Cwirzen, Effects of curing: comparison of
- 593 optimised alkali-activated PC-FA-BFS and PC concretes, Mag. Concr. Res. 66 (2014) 315–
- 594 323. doi:10.1680/macr.13.00231.
- 595 [15] Olalekan.O. Ojedokun and Pal. S. Mangat, Chloride Diffusion in Alkali Activated Concrete,
- 596 in: II Int. Conf. Concr. Sustain., 2016: pp. 521–531.
- 597 [16] BS ISO 15901-1:2016 Evaluation of pore size distribution and porosity of solid materials by
   598 mercury porosimetry and gas adsorption. Mercury porosimetry, 2016.

- 599 [17] BS EN196-1, Methods of testing cement Part 1: Determination of strength, Eur. Stand.
  600 (2005) 1–33. doi:10.1111/j.1748-720X.1990.tb01123.x.
- 601 [18] BS EN 12350-5 Testing fresh concrete Part 5: Flow Table Test, Eur. Stand. (2009).
  602 doi:10.1007/s13398-014-0173-7.2.
- H. Jansson, D. Bernin, K. Ramser, Silicate species of water glass and insights for alkaliactivated green cement, AIP Adv. 5 (2015) 67167. doi:10.1063/1.4923371.
- 605 [20] V. Sathish Kumar, N. Ganesan, P. V Indira, Effect of Molarity of Sodium Hydroxide and
- 606 Curing Method on the Compressive Strength of Ternary Blend Geopolymer Concrete, IOP
- 607 Conf. Ser. Earth Environ. Sci. 80 (2017) 12011. doi:10.1088/1755-1315/80/1/012011.
- 608 [21] BS EN 12390-3:2009 Testing Hardened Concrete Part 3: Compressive Strength of Test
  609 Specimens, 2009.
- 610 [22] BS EN 12390-7:2009, Testing hardened concrete. Density of hardened concrete BSI British
- 611 Standards, in: n.d. https://shop.bsigroup.com/ProductDetail/?pid=00000000030164912
  612 (accessed June 21, 2018).
- [23] N. Hearn, R.D. Hooton, Sample mass and dimension effects on mercury intrusion porosimetry
  results, Cem. Concr. Res. 22 (1992) 970–980. doi:10.1016/0008-8846(92)90121-B.
- 615 [24] M.M. Reda Taha, A.S. El-Dieb, N.G. Shrive, Sorptivity: a reliable measurement for surface
  616 absorption of masonry brick units, Mater. Struct. Constr. 34 (2001) 438–445.
- 617 [25] M.I. Mousa, M.G. Mahdy, A.H. Abdel-Reheem, A.Z. Yehia, Self-curing concrete types; water
  618 retention and durability, Alexandria Eng. J. 54 (2015) 565–575.
- 619 doi:10.1016/J.AEJ.2015.03.027.
- 620 [26] K. Tan, O.E. Gjorv, Performance of concrete under different curing conditions, Cem. Concr.
- 621 Res. 26 (1996) 355–361.
- 622 [27] D. Khale, R. Chaudhary, Mechanism of geopolymerization and factors influencing its
- 623 development: A review, J. Mater. Sci. 42 (2007) 729–746. doi:10.1007/s10853-006-0401-4.
- 624 [28] A.C. Garrabrants, F. Sanchez, D.S. Kosson, Leaching model for a cement mortar exposed to
- 625 intermittent wetting and drying, AIChE J. 49 (2003) 1317–1333. doi:10.1002/aic.690490523.
- 626 [29] K.K. ALIGIZAKI, Pore structure of cement-based materials: testing, interpretation and

- 627 requirements, Taylor & Francis, Abingdon [England], 2006.
- 628 [30] H. Ma, Mercury intrusion porosimetry in concrete technology: Tips in measurement, pore
  629 structure parameter acquisition and application, J. Porous Mater. 21 (2014) 207–215.
- 630 doi:10.1007/s10934-013-9765-4.
- 631 [31] T.C. Powers, A discussion of cement hydration in relation to the curing of concrete, Highw.
- 632 Res. Board Proc. 27 (1948). https://trid.trb.org/view.aspx?id=102345 (accessed June 29,
- 633 2017).
- R.G. Patel, D.C. Killoh, L.J. Parrott, W.A. Gutteridge, Influence of curing at different relative
   humidities upon compound reactions and porosity in Portland cement paste, Mater. Struct. 21
- 636 (1988) 192–197. doi:10.1007/bf02473055.
- 637 [33] A.T.C. Guimarães, G. De Vera, F.T. Rodrigues, C. Antôn, M.A. Climent, Comparison
- 638 between Dcrit Considering the Abrupt Variation and Inflexion in the Concrete Mercury
- 639Intrusion Porosimetry Curve, Exp. Tech. (2015). doi:10.1111/ext.12002.
- 640 [34] J.M. Khatib, P.S. Mangat, Porosity of cement paste cured at 45C as a function of location
- relative to casting position, Cem. Concr. Compos. 25 (2003) 97–108. doi:10.1016/S0958-
- 642 9465(01)00093-2.
- 643 [35] Balshin M.Y., Relation of Mechanical Properties of Powder Metals and their Porosity and the
- 644 Ultimate Properties of Porous-Metal Ceramic Materials, 67 (1949) 831–834.