



FACULDADE DE ENGENHARIA DA UNIVERSIDADE DO PORTO

## **Silicone breast implants: Experimental analysis of failure mechanisms**

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Laughter is timeless, imagination has no age, and dreams are forever.

Walt Disney (1901-1966)



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## Abstract

Silicone breast implants have been in use for nearly six decades. Some adverse effects have tainted their history enlightening the complex factors involved in the interaction of the device with the human body. One relevant event happened in March 2010, regarding the Poly Implant Prothèse (PIP) breast implants. It was assumed the use of industrial-grade instead of certified medical-grade silicone was responsible for reportedly higher rates of implant rupture *in vivo*. Thus, the main goal of this study was to find an explanation for the seemingly higher rates of rupture for PIP breast implants compared to other manufacturer of breast implants. Through an analysis of explanted implants combined with control implants (virgin), we can determine the various factors related to ruptured implants. In order to identify the problems inherent to PIP breast implants an extensive experimental protocol was developed according to the international standards for mammary implants (ISO 14607), involving the determination of tensile stress-strain properties (ISO 37), and biological evaluation of medical devices (ISO 10993). Twenty two (22) explanted PIP implants and fourteen (14) controls were studied using a broad combination of mechanical testing (tensile and fatigue tests), chemical analysis by fourier transform infrared spectroscopy (FTIR), surface characterization by scanning electron microscopy (SEM), and *in vitro* degradation tests. The obtained results are reported in six main papers which attempt to answer the research question: “Why do implants fail?” The first is a review paper followed by five cross-sectional studies, each one with different aims in order to evaluate the implants’ rupture.

Due to the evidence collected it was possible to demonstrate the heterogeneous nature of the PIP shell. Shell thickness varied significantly for PIP implants, exceeding

the manufacturer's specifications. It was observed that thinner thicknesses are likely to have a lower strength and a higher probability of failure. In general, there were significant differences between intact and ruptured implants: ruptured implants were thinner (0.73 mm vs. 0.91 mm) and weaker (7.42 MPa vs. 9.59 MPa) compared with intact implants. These results point to a reduced ability of the ruptured implants (shells), to withstand mechanical stresses. By comparing the results with the other tested brand (Brand X), thickness variation was within manufacturer's specifications. Moreover, in the analysis of PIP implant ruptures by electron microscopy, features normally associated to fatigue phenomena were found. These features detected in explanted implants constitute a significant finding since, as far as the author is aware, they have not been identified in the previously literature, indicating that fatigue can be at the origin of breast implants ruptures. In FTIR analysis, no spectral deviations were observed during implantation time, which suggests a lack of chemical degradation.

Therefore, through this research we can conclude that the thickness variation and fatigue phenomena, besides the material properties were the main implications for the failure of the eleven ruptured implants. The findings should be considered as a relevant parameter during the manufacturing process, for quality assurance purposes.

## Resumo

Os implantes mamários de silicone são usados há quase seis décadas. Alguns efeitos adversos marcaram a sua história com eventos, mostrando que existem fatores complexos envolvidos na interação do implante com o corpo humano. Um desses eventos aconteceu em Março de 2010 relativamente aos implantes Poly Implant Prothèse (PIP). Assume-se que o uso de um silicone de grau industrial em vez de um silicone de grau médico certificado, foi o responsável pelas elevadas taxas de rotura dos implantes *in vivo*. Assim, o objetivo principal deste estudo foi encontrar uma explicação para as elevadas taxas de rotura dos implantes mamários PIP em comparação com outro fabricante. Através de uma análise dos implantes explantados combinada com os implantes de controlo (vírgens), pudemos determinar os vários fatores relacionados com a rotura destes implantes. Para identificar os problemas inerentes aos implantes mamários PIP foi desenvolvido um protocolo experimental de acordo com as normas internacionais para implantes mamários (ISO 14607); para os ensaios de tração (ISO 37); e para avaliação biológica de dispositivos médicos (ISO 10993). Vinte e dois (22) implantes PIP explantados e catorze (14) de controlo foram estudados numa ampla combinação de testes mecânicos (ensaios de tração e fadiga), análise química por espectroscopia infravermelho de transformação de fourier (FTIR), caracterização de superfície por microscopia eletrónica de varredura (SEM), e teste de degradação *in vitro*. Os resultados obtidos são relatados em seis diferentes artigos que tentam responder à pergunta: “Porque é que os implantes falham?” O primeiro é um artigo de revisão seguido por cinco estudos transversais, cada um com objetivos diferentes, a fim de avaliar a rotura dos implantes

Os resultados permitiram verificar a existência de uma natureza heterogênea no invólucro dos implantes PIP. A espessura do invólucro variou significativamente nos implantes PIP, excedendo as especificações do fabricante. Observou-se que espessuras mais finas tendem a ter uma menor resistência e conseqüentemente uma maior probabilidade de falha. Em geral, a comparação entre os implantes com rotura e os intactos apresentaram diferenças estatisticamente significativas: os implantes com rotura eram mais finos (0.73mm vs 0.91mm) e mais fracos (7.42MPa vs. 9.59MPa) comparativamente com os intactos. Estes resultados apontam para uma capacidade reduzida dos implantes com rotura (no invólucro) para suportar tensões mecânicas. Comparando os resultados com a outra marca de implantes testada (Marca X), a variação da espessura esteve de acordo com as suas especificações. Na análise dos implantes com rotura, através da microscopia, foram encontradas estrias que normalmente estão associadas a fenómenos de fadiga. Estas estrias constituem uma descoberta significativa, uma vez que, tanto quanto é do conhecimento do autor, não estão identificadas na literatura, indicando assim que a ocorrência de fenómenos de fadiga pode estar na origem das roturas dos implantes mamários. Na análise FTIR, não foram observados desvios nos espectros do material durante o tempo de implantação, o que sugere uma falta de degradação química.

Assim, através desta investigação podemos concluir que os fenómenos da variação da espessura e fadiga, para além das propriedades do material, foram as principais implicações para a falha dos onze implantes com rotura. Estes resultados devem ser considerados como um parâmetro relevante durante o processo de fabrico, para fins de garantia de qualidade.

# Contents

Abstract .....	ix
Resumo.....	xi
Contents .....	xiii
List of Figures .....	xix
List of Tables.....	xxvii
List of Abbreviations .....	xxix
List of Symbols.....	xxvii
<b>Chapter I</b>	
<b>Introduction .....</b>	<b>3</b>
1. Motivation.....	3
2. Objectives.....	5
3. Thesis Outline .....	6
References .....	10
<b>Chapter II</b>	
<b>Background Literature Review .....</b>	<b>11</b>
1. Evolution of Breast Implants.....	13
2. Mechanical Interaction between Tissue and Implants .....	16
3. Mechanisms of Implant Failure.....	20
References .....	26
<b>Chapter III</b>	
<b>Research Methodology .....</b>	<b>33</b>
References .....	42

## Chapter IV

<b>Review Article</b> .....	45
<b>Biomechanical Properties of Breast Tissue, a State-of-the-art Review</b>	
<b>Article 1</b> .....	47
Abstract .....	49
1. Introduction.....	51
2. Characterization of Soft Tissues – Basic Concepts .....	56
3. Experimental Techniques to Characterize Breast Tissue.....	62
3.1 In vivo Techniques.....	62
3.2. <i>Ex vivo</i> Techniques .....	65
4. Mechanical Properties of Breast Tissue.....	67
5. Discussion and Conclusions .....	78
References.....	86

## Chapter V

<b>Original Articles</b> .....	97
<b>Mechanical Performance of Poly Implant Prosthesis (PIP) Breast Implants a Comparative Study</b>	
<b>Article 2</b> .....	99
Abstract .....	101
1. Introduction.....	103
2. Material and Methods .....	104
2.1.Breast Implants Collection.....	104
2.2.Mechanical Testing Protocol .....	104
2.2.1. Samples Preparation .....	105
2.2.2. Testing Procedure .....	108
2.3 Statistical Analysis.....	109
3. Results.....	109
3.1. Appearance of Explanted Implants .....	110
3.2. Mechanical Testing.....	112
3.2.1 Breast Implants Shell Strength – Global Overview.....	112
3.2.2 Thickness Variation .....	120

4. Discussion.....	123
5. Conclusions.....	127
References.....	129
<b>Breast Implants Rupture Induced by Fatigue Phenomena</b>	
<b>Article 3 (Letter Communication) .....</b>	<b>133</b>
References.....	137
<b>A Morphologic Analysis of Rupture of Poly Implant Prosthesis (PIP) Breast Implants</b>	
<b>Article 4 .....</b>	<b>139</b>
Abstract .....	141
1. Introduction.....	143
2. Material and Methods .....	144
2.1 Materials .....	144
2.2. Scanning Electron Microscopy (SEM) Analysis .....	144
2.3 Fatigue Test.....	145
3. Results.....	146
3.1. Visual Inspection of Implants .....	146
3.2. SEM Analysis of Shells and Failure Regions .....	147
3.3 Fatigue Tests Results .....	151
4. Discussion.....	153
5. Conclusions.....	157
References.....	158
<b>Intact vs Ruptured Breast Implants. A Woman-centric Paired Analysis</b>	
<b>Article 5 .....</b>	<b>163</b>
Abstract.....	165
1. Introduction.....	167
2. Material and Methods .....	168
2.1 Clinical Data .....	168
2.2 Testing Protocol.....	171
2.3 Fourier Transform Infrared Spectroscopy (FTIR) Characterization.....	171
2.4 Statistical Analysis.....	172

3. Results.....	172
3.1. Shell Properties of Intact vs Ruptured Breast Implants .....	175
3.2. Chemical Characterization of Silicone Shells and Gels .....	180
4. Discussion.....	181
5. Conclusions.....	184
References.....	186
<b><i>In vitro</i> Degradation of Polydimethylsiloxanes for Breast Implant Applications</b>	
<b>Phenomena</b>	
<b>Article 6</b> .....	191
Abstract.....	193
1. Introduction.....	195
2. Material and Methods .....	196
2.1 Degradation test .....	197
2.2 Mechanical Test.....	197
2.3 Morphological Characterization .....	198
2.4 Surface Characterization by Fourier Transform Infrared Spectroscopy (FTIR) .....	198
2.5 Statistical Analysis.....	198
3. Results.....	199
3.1. Mass Loss During <i>In Vitro</i> Aging.....	199
3.2 Mechanical Properties Analysis.....	200
3.3. SEM Analysis .....	202
3.4. ATR – FTIR Analysis.....	204
4. Discussion.....	206
5. Conclusions.....	207
References.....	209
<b>Chapter VI</b>	
<b>Integrated Discussion</b> .....	213
<b>Chapter VII</b>	
<b>Conclusions</b> .....	227



**Chapter VIII**

**Limitations and Recommendations for Future Works ..... 233**



# List of Figures

## Chapter I – Introduction

Figure 1. Exploratory steps to evaluate the ruptures causes..... 6

## Chapter II – Background Literature Review

Figure 1. Breast Implants Evolution. a) First Generation, b) Second Generation, and c) Third Generation. Adapted from [7, 8]..... 15

Figure 2. Modern Generation with round shape a) The textured surface and b) cohesive gel is visible (the arrow points to the cohesive gel) ..... 16

Figure 3. Explanatory scheme of deformations over time between the breast tissues and the implant for two different cases. The lines represent the behaviour between the tissue and implants. For case a) the breast has an excellent tissue support, high-stiffness/ low-compliance/high-resilience breasts, and so it's very likely to maintain the postoperative result in the long term. In case b) the breast has poor tissue support, low stiffness/ high compliance/low-resilience, so are at risk for intense creep deformation when loaded with large or high-projecting implants. Adapted from [17].  
..... 19

## Chapter III – Research Methodology

Figure 1. Exploratory steps to evaluate the ruptures causes, with description of the methodology used.....35

Figure 2. Schematics of the experimental procedure: (a,b,c) implant segmentation into 12 segments, (d) example of sample preparation for tensile tests of a segment.....35

Figure 3. Scheme divided by articles about the number of implants and samples were analysed ..... 41

## **Chapter IV – Review Article**

### **Article 1- Biomechanical Properties of Breast Tissue, a State-of-the-art Review**

Figure 1. Anatomy of Breast ..... 52

Figure 2. Stiffness in different soft tissue. Adapted from [30]..... 56

Figure 3. Mechanical behaviour of linear elastic and hyperelastic materials..... 58

Figure 4 The dashed is a hysteresis loop and shows the amount of energy lost (as heat) in a loading and unloading cycle. Adapted from [31]. ..... 59

Figure 5. (a) Unconfined compression, (b) Confined compression and (c) Indentation test ..... 61

Figure 6. An overview of elasticity imaging methods. Adapted from [32, 66].....65

Figure 7. Behaviour of breast tissue at different levels of pre-load compression. Adapted from [32, 34,85]..... 76

## **Chapter V – Original Articles**

### **Article 2- Mechanical Performance of Poly Implant Prosthesis (PIP) Breast Implants a Comparative Study**

Figure 1. Classification of the Shell damage. a) Hole shaped damage; b) V- shaped split; c) Split and d) Gross damage, in this case the shell and the cohesive gel were totally separated.. ..... 106

Figure 2. Schematics of the experimental procedure. a) Regions of the implant; b) Implant segmentation into 12 segments. To ensure traceability of each segment over the

implant, each segment was labelled with a number (1 to 12); c) example of sample preparation for tensile tests; d) Tensile testing equipment .....	107
Figure 3. Scheme of the sample thickness measurement. Yellow-edges of the samples; red-control area.....	108
Figure 4 This figure is a direct comparison between implants with a large and small rupture. <b>a</b> Example of the aspect of the shell and gel in an explanted implant with split rupture (posterior location and implanted for 40 months); <b>b</b> V-shape split that shows a yellowish coloration and calcifications in gel and shell (anterior and equatorial location and implanted for 46 months).....	111
Figure 5. Material behaviour of the control implants (stress at 266% of strain). Average values of measures are expressed as mean (M) ± standard deviation (SD). A total of 24 samples in anterior region and 12 samples in posterior region are represented in contour plot for ControlPip02. For ControlBrand X02, 23 samples are shown in anterior and 20 in posterior regions.....	113
Figure 6. The box plot shows the stress (MPa) at 266% of strain between Controls (PIP and brand X). Values are presented as median (horizontal line within box), 25-75th percentile (box) and T-bars (range to the minimum or maximum values).....	114
Figure 7. Stress (MPa) of two explanted implants with different characteristics - type of rupture, implant colour and duration of implantation. PIP01 was implanted for 46 months, and had a V-shaped rupture and yellowish appearance (56 samples). PIP08 was implanted for 64 months, and had a hole rupture and clear appearance (57 samples).....	114
Figure 8. The box plot compare the stress (MPa) between three regions of implant (Anterior, equatorial, and posterior) for three groups of implants .....	115

Figure 9. The box plot compares the stress (MPa) at different levels of strain between PIP implants (explanted and control). Values are presented as median (horizontal line within box), 25-75th percentile (box) and T-bars (range to the minimum or maximum values).....117

Figure 10. a) Correlation between stress (MPa) at 266% of strain and duration of implantation of ruptured PIP implants ( $r = 0.56$ ;  $n=11$ ;  $P = 0.0053$ ); b) Correlation between stress and year of implantation ( $r=-0.681$ ;  $n=11$ ;  $P=0.0208$ ).....118

Figure 11. Pearson correlation between stress (MPa) at 266% of strain and thickness (mm) of controlPip01 breast implants (coefficients in Table 4).  $R^2$  is the coefficient of determination. Total of 60 samples, from which 24 were in the anterior, 24 were in the equatorial and 12 in the posterior regions.....121

Figure 12. Example of Pearson correlation between stress (MPa) with 266% of strain and thickness (mm) of explanted implant PIP04 (Coefficients are reported in Table 4).  $R^2$  is the coefficient of determination. A total of 58 samples, from which 22 were in the anterior region, 24 were in the equatorial and 12 in the posterior regions.... 122

Figure 13. Example of Pearson correlation between stress (MPa) with 266% of strain and thickness (mm) of ControlBrand X01 implant (Coefficients are reported in Table 4). A total of 60 samples (24 in anterior; 12 in equatorial and 24 in posterior regions)..... 122

**Article 3- Breast Implants Rupture Induced by Fatigue Phenomena**

Figure 1. Fatigue fracture surface a) schematic representation [4], b) micrograph (75x) of implant samples from the hole rupture site, and c) micrograph (200x) of implant

samples from the split rupture site;the arrow points to the fatigue crack origin.....	136
<b>Article 4- A Morphologic Analysis of Rupture of Poly Implant Prosthesis (PIP)</b>	
<b>Breast Implants</b>	
Figure 1. Samples Geometries. a) uniaxial sample and b) biaxial sample.....	146
Figure 2. Two implants with different ruptures. a) Gross Damage with 140mm of rupture size; b) V-shape split with 80mm .....	147
Figure 3.SEM micrographs (75x), view of v-shape split rupture in four implants.....	148
Figure 4. SEM micrographs (75x), view of gross damage of the implant shell.....	149
Figure 5.Small ruptures. a) Hole rupture and b) Split rupture .....	150
Figure 6. SEM micrographs of small ruptures - a,c) 75x; b,d) 200x. The same type of striations appears in all implants. a,b) Split rupture and c,d) Hole rupture. The arrows point in the direction of crack initiation point.. .....	150
Figure 7. SEM micrographs (a) 75x and b) 200x) in biaxiais samples, the striations are visible. The dashed lines are to emphasize part of striations.....	152
Figure 8. a) schematic representation of radial tearing lines or ridges, and propagation of a fatigue crack in parallel planes (see de arrow); b) micrograph of biaxial samples, the radial tearing lines and the fatigue crack in parallel planes (see arrow) are visble. ....	152
Figure 9. SEM micrographs of the a) inner surface (500x) and b) outer surface (75x) of the sample IMGHC-TX-S-265 with damage.....	153

## Article 5- Intact vs Ruptured Breast Implants. A Woman-centric Paired Analysis

Figure 1. Shell Aspect: a) Control implant; clear aspect. b) Intact implant; clear aspect and no macroscopic changes. c) Ruptured shell; yellowed shell.....	174
Figure 2. Different gel conditions: a) Clear gel, non-cohesive (oily). b) Yellow gel, non-cohesive. c) Clear gel, cohesive single mass .....	175
Figure 3. Comparison of Tensile Strength (MPa) for implant shells grouped according with gel conditions. Values are presented as median (horizontal line within box), 25-75th percentile (box) and T-bars (range to the minimum or maximum values).....	176
Figure 4 Tensile strength (MPa) of explants and controls (3 groups) as a function of implantation time (months). .....	178
Figure 5. Comparison between intact and ruptured implants, per patient. The colour variation represents the tensile strength along the shell. I= Intact implant; R= Ruptured Implant; IT= Implantation Time and RT= Rupture type .....	180
Figure 6. FTIR spectra of gel and shell extracted from intact, ruptured and control implants .....	181

## Article 6- *In vitro* Degradation of Polydimethylsiloxanes for Breast Implant

### Applications Phenomena

Figure 1. Experimental results of weight loss for Brand 1 during the degradation period under buffer solutions.....	199
Figure 2. Experimental results of weight loss for Brand 2 during the degradation period under buffer solutions .....	200



Figure 3. Example of tensile test results during the degradation in two buffer solutions:  
a) and b) for Brand 1 and lot 1.2; c) and d) for Brand 2. Blue and red lines are used  
to represent the stiffening of the shell ..... 202

Figure 4. SEM micrographs of the outer surface of a) Brand 1 implants (x75 to lot 1.1 and  
x200 to lot 1.2), and b) Brand 2 implants (x75 and x200) over two time points (0 and  
12 Weeks) under pH 7.4 and a pH 4.0 solutions ..... 203

Figure 5. FTIR spectra of Brand 1 (a) and Brand 2 (b) soaked in different solutions and  
different degradations stages ..... 205



# List of Tables

## Chapter IV – Review Article

### Article 1- Biomechanical Properties of Breast Tissue, a State-of-the-art Review

Table 1. Mechanical tests for the breast tissue reported in literature, grouped according with vital state of the subject ( <i>in vivo</i> / <i>ex vivo</i> ) and testing technique.....	69
Table 2. A Summary of the results from mechanical testing of <i>ex vivo</i> breast tissue.....	74
Table 3. A summary of results from <i>in vivo</i> magnetic resonance elastography for breast tissue.....	78

## Chapter V – Original Articles

### Article 2- Mechanical Performance of Poly Implant Prosthesis (PIP) Breast Implants a Comparative Study

Table 1. Clinical characteristics and implant rupture status .....	110
Table 2. Multi-factor ANOVA analysis results. Homogeneous Groups regarding regions (stress values at 266% strain) for all implants. The table is organized into subgroups, and within each column, the levels contain a group of means within which there are no statistically significant differences. ....	116
Table 3. Multi-factor ANOVA analysis results. Homogeneous Groups regarding stress (at 266% strain) for PIP implants (explanted vs control). The table is organized into	

subgroups, and within each column, the levels contain a group of means within which there are no statistically significant differences.....	119
Table 4. Correlation analysis using Pearson Correlation for all implants. ....	123
<b>Article 4- A Morphologic Analysis of Rupture of Poly Implant Prosthesis (PIP) Breast Implants</b>	
Table 1. Information about the control PIP breast implants.....	151
<b>Article 5- Intact vs Ruptured Breast Implants. A Woman-centric Paired Analysis</b>	
Table 1. Clinical and demographic information.....	170
Table 2. Classification of gel and shell condition.....	174
Table 3. Multi-factor ANOVA results regarding gel condition for explanted implants.....	176
Table 4. Multi-factor ANOVA analysis results regarding all implants, and shell thickness.....	177
Table 5. Tensile strength and thickness comparison between intact and ruptured explanted implants per patient (data expressed as mean±standard deviation).....	179
<b>Article 6- <i>In vitro</i> Degradation of Polydimethylsiloxanes for Breast Implant Applications Phenomena</b>	
Table 1. Statistical analysis of mechanical properties for breast implant samples in different stages of degradation (0 and 12 weeks). Bold type indicates significant differences ( $p < 0.05$ ).....	201

## List of Abbreviations

ATR	Attenuated Total Reflectance
BMI	Body Mass Index
CEMUP	Materials Centre of the University of Porto
CETRIB	Tribology, Vibrations and Industrial Maintenance
CT	Computed Tomography
DCIS	Ductal Carcinoma in Situ
IDC	Invasive Ductal Carcinoma
D4	Octamethylcyclotetrasiloxane
FDA	Food and Drug Administration (USA)
FEM	Finite Element Method
FTIR	Fourier Transform Infrared Spectroscopy
INFARMED	National Authority of Medicines and Health Products
INEGI	Institute of Science and Innovation in Mechanical and Industrial Engineering
MRI	Magnetic Resonance Imaging
MRE	Magnetic Resonance Elastography
OCT	Optical Coherence Elastography
PET	Positron Emission Tomography
PIP	Poly Implant Prothèse
TGA	Therapeutic Good Administration (Australia)

SEM	Scanning Electron Microscopy
SN	Serial Number
SPECT	Single Photon Emission Computed Tomography
PDMS	Polydimethylsiloxanes
US	Ultrasound
Th	Thickness

## List of Symbols

	Stress
	Strain
$E$	Young's Modulus
	Poisson's ratio
$q$	Load density
$a$	Radius of loaded area (piston)
$w$	Maximum displacement in the load direction
$K$	Conversion factor that depends on the indenter's geometry
$\epsilon_a$	Strain in loading direction (axial)
$\epsilon_d$	Strain in lateral direction
$p_i$	Initial weight of the sample
$p_f$	Weight after different degradation stages





# Chapter I

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Introduction



## **1. Motivation**

For decades, women have undergone breast implant surgery either for health or aesthetic reasons. Some adverse effects have tainted the history of such breast implants illustrating the complex factors involved in the interaction between the device and the human body. Understanding the adverse outcomes is a key factor to improve the safety of breast implants. The major concern to the patient and to the plastic surgery community is the lifespan of a breast implant.

The lifespan of silicone implants was initially presumed to be unlimited, but it was later demonstrated that the silicone elastomer has a finite lifespan and that silicone implants age and eventually fail [1]. The normal lifespan of breast implants is ten or more years. The implant failure rates depends on the definition of implant failure, the manufacture, the population, and the diagnostic method used [1-3].

This work started after the media reports in 2010 about the Poly Implant Prothèse (PIP) failures. The aim was to understand the mechanisms and failure rate of these implants. In March 2010, the French medical device regulatory agency (AFSSAPS) suspended the marketing, distribution and use of all silicone implants produced by Poly Implant Protheses (PIP) due to serious concerns about the quality of the applied material [4]. The controversy surrounding the PIP breast implants caused heightened anxiety and extensive publicity regarding breast implant safety in humans. Based on peer-reviewed published studies the probability of rupture for PIP implants was estimated to be of 14.5% to 31 %, 10 years after implantation, whereas other silicone breast implants brands have

been reported with a rupture rate of 1.1 to 11.6%, considering the same follow-up period [5-11].

In order to identify the problems inherent to breast implants, it is necessary to analyse the explanted implants. Thus, an extensive experimental protocol was developed to analyse PIP explanted and control implants (PIP and other brand), in order to identify the causes of implant failures, rupture mechanism, and the interaction between the implants and breast tissue. The effects of implantation time on the durability of implant shells should be analysed by studying the implants according to type, so that explants can be compared with the controls. This is necessary because the strength of implant shells can vary considerably as a function of manufacturer, implant type, and lot-to-lot variability for the given type. For this reason data for the control implant, PIP and other brand (Brand X) are also presented with the explant data in order to understand main reason for the rupture causes.

This research was carried out with the collaboration of Dra. Maria da Luz Barroso and Dra. Diana Costa Santos of Department of Plastic Surgery of the Hospital Center of Gaia, Portugal. The collaboration involved the supply of all explanted breast implants, demographic information about patients, clinical information and the necessary contacts for the acquisition of the control implants (virgin) - PIP and another brand (Brand X). PIP sealed controls were supplied by the National Authority of Medicines and Health Products (INFARMED, Portugal). Brand X was supplied by manufacturers' representatives in Portugal.

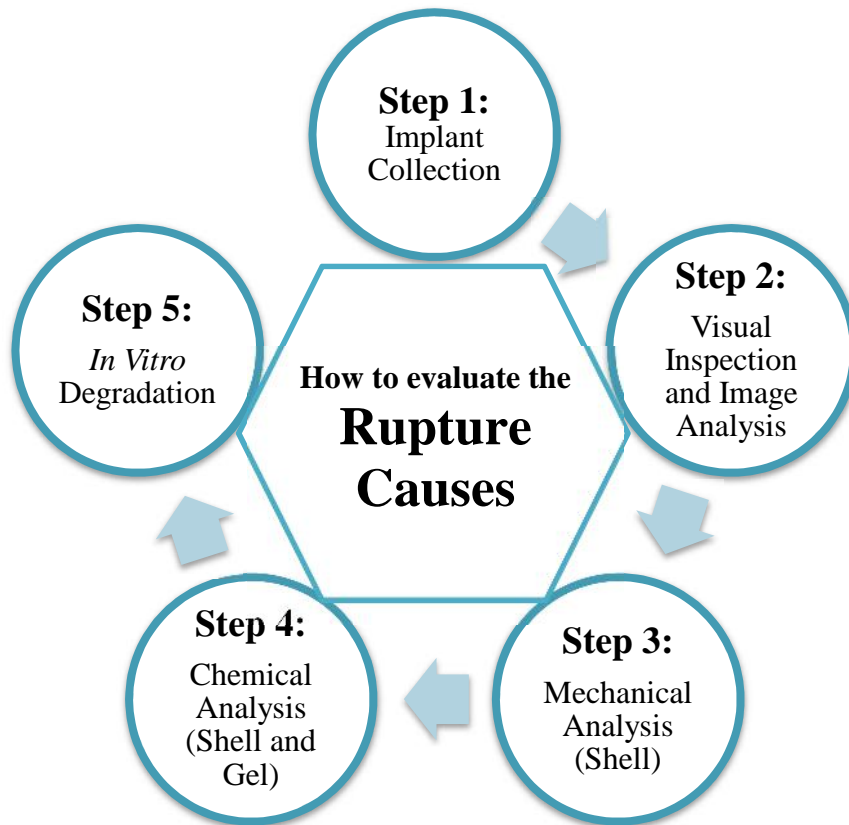
## 2. Objectives

The main goal of this study was to find an explanation for the seemingly higher rates of rupture for PIP breast implants compared to other breast implants brands.

To understand and contextualize this problem, the following specific objectives were defined:

- Review the mechanical properties of breast tissue.
- Characterize the ruptured breast implants as a function of patient's demographic and clinical data.
- Analyse and describe the rupture characteristics of explanted implants.
- Develop a protocol to analyse the shell integrity of explanted implants.
- Provide details of the ruptured shell region and its relation with breast implant failure.
- Characterize the shells and gels (intact and ruptured implants) by chemical surface analysis.
- *In vitro* evaluation of the effect of material properties during implantation time, under physical/chemical conditions of the human body (temperature and pH).
- Validate the experimental results for explanted implants by comparison with available control implants.

Figure 1 illustrates schematically the sequence of activities carried out to collect the experimental data.



**Figure 1.** Exploratory steps to evaluate the ruptures causes.

### 3. Thesis Outline

This thesis' structure is based on published/submitted journal articles and is organized in 8 main Chapters.

Chapter I introduces the thesis subject, and is divided into 3 sub-sections. Sub-section 1 describes motivation for the development of the work. The objectives of the project are presented in the second sub-section. The last section shows the thesis outline with a brief explanation of its structure.

Chapter II provides a background on the breast implants and the principal concerns related to them. The first sub-section starts by addressing the evolution of the material and design of breast implants over the years. Afterwards, a brief description of the

relationship between tissues and breast implants is presented, as well as possible complications associated with this interaction (sub-section 2). Sub-section 3 reviews the main causes/implications of breast implants' rupture.

Chapter III describes the experimental methodology adopted for this thesis.

Chapters IV (Review Article) and V (Original Articles) are composed of the articles written during the project, depicting in greater detail the obtained results. These chapters were designed to achieve the main purpose of this thesis, comprised of one review article and five original studies, each one with different aims. The sequence of articles is organized as follows:

**Chapter IV - Article 1:**

***Title:*** Biomechanical Properties of Breast tissue, a State-of-the-art Review.

***Authors:*** Nilza Ramião, Pedro Martins, Rita Rynkevic, António A. Fernandes, Maria da Luz Barroso, Diana C. Santos

***Published in:*** Biomechanics and Modeling in Mechanobiology, 2016:15(5); 1307-23 doi: 10.1007/s10237-016-0763-8.

**Chapter V- Article 2:**

***Title:*** Mechanical Performance of Poly Implant Prosthesis (PIP) Breast Implants a Comparative Study

***Authors:*** Nilza Ramião, Pedro Martins, Maria da Luz Barroso, Diana C. Santos, Francisco Pereira, António A. Fernandes

***Published in:*** Aesthetic Plastic Surgery, 2017, doi: 10.1007/s00266-017-0776-4

**Chapter V - Article 3:**

**Title:** Breast Implants Rupture Induced by Fatigue Phenomena

**Authors:** Nilza Ramião, Pedro Martins, Maria da Luz Barroso, Diana C. Santos, António A. Fernandes

**Published in:** Journal of Plastic, Reconstructive & Aesthetic Surgery, 2017, doi: 10.1016/j.bjps.2017.01.002

**Chapter V - Article 4:**

**Title:** An Experimental Analysis of Shell Failure in Breast Implants

**Authors:** Nilza Ramião, Pedro Martins, Maria da Luz Barroso, Diana C. Santos, António A. Fernandes

**Published in:** Journal of the Mechanical Behavior of Biomedical Materials, 2017, doi: 10.1016/j.jmbbm.2017.04.005

**Chapter V - Article 5:**

**Title:** Intact vs Ruptured Poly Implant Prothèse (PIP) Breast Implants. A Woman-centric Paired Analysis

**Authors:** Nilza Ramião, Pedro Martins, Maria da Luz Barroso, Diana C. Santos, António A. Fernandes

**Submitted to an International Journal:** Journal of Plastic, Reconstructive & Aesthetic Surgery



**Chapter V - Article 6:**

***Title:*** *In vitro* Degradation of Polydimethylsiloxanes for Breast Implant Applications

***Authors:*** Nilza Ramião, Pedro Martins, Maria da Luz Barroso, Diana C. Santos, António A. Fernandes

***Published in:*** Journal of Applied Biomaterials & Functional Materials, 2017, doi: 10.5301/jabfm.5000354

Chapter VI presents an integrated discussion of the obtained results and the thesis' main contributions.

Chapter VII and VIII summarizes the main conclusions and possible pathways for future research.

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# Chapter II

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Background Literature Review



## 1. Evolution of Breast Implants

Breast augmentation is the third most performed aesthetical surgical procedure in the world, with 1.348.197 surgeries having been performed in 2015 [1]. Breast implants have been an essential tool in the global plastic surgeon's cosmetic and reconstructive set of skills since their invention by Cronin and Gerow in the early 1960's [2]. Since then, the development of new materials, manufacturing, design and surgical technical improvement have continued to evolve. Five main generations of silicone breast implants have been introduced to the market over the last 60 years [3]. The Breast implant material is polydimethylsiloxane (PDMS), and over the years gels with different amounts of cross-linking – and thus different properties – have been used. PDMS is the basis for both the breast implant silicone gel and shell. PDMS *“is an oily, sticky liquid with a viscosity that increases as the average chain length (molecular weight) is increased”* [4]. The breast implants are created from liquid components during the formation of the shell, to which are added a selected amount of “nano-particles” of amorphous “fumed” silica (SiO<sub>2</sub>) filler [5]. Adding the “nano-particles” results in a silicone rubber with improved strength, and increased elongation before failure as tensile loading is increased [4].

The gels are consisted of polymeric networks of swollen cross-linked PDMS. The extent of cross linking and amount of fluid added to the gel accounts for the wide variety of viscosities and cohesivities of various generation silicone gel implants [6]. In contrast, the shell has *“much greater crosslinking, very little fluid and the addition of amorphous silica for strength”* [6].

There are several generations of breast implants, presenting an evolution and development of their shell (outer) and gel (fill) material.

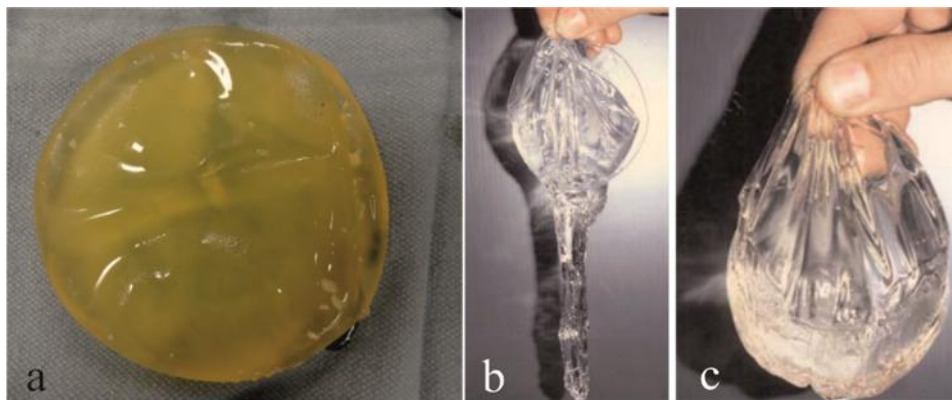
The first generation implants had a thick shell (0.75 mm) and a highly viscous silicone fill material resulting in firm and durable implants [6-8]. An example of this implant is visible in Figure 1 a). The firmness of the gel was related to the relative amount of highly cross-linked material in the gel. The gel contained about 50% highly cross-linked silicone and about 50% low molecular weight chains for first generation [7]. Rupture rates were low, but most women developed very firm breasts within a year of their surgery, due to the capsular contracture and calcification, nearing 100% in implants in place greater than 10 years [9, 10].

Second generation implants (Figure 1 b)) were designed to create a softer and more natural feel. As a result of these design changes, the implants had a much thinner (0.13 mm) and softer shell, and less viscous silicone (thin and watery) [7]. The gel contained only about 80% low molecular weight chain and 20% highly cross-linked silicone [7]. As a result of these design changes, second generation was recognized as ineffective in reducing contracture, thus resulting in a more fragile device. This generation increased the rupture rates to as high as 60% [11, 12]. After these problems, modifications were made for further improvement, resulting in the subsequent third generation implants.

Third generation implants, see Figure 1 c), had a more durable, thicker (0.30 to 0.50 mm) (high performance) and multilayered shell that significantly reduced rupture and silicone bleed [13, 14]. The fill was a much more cohesive gel, contained larger particle size and increased cross linking to decrease diffusion [13]. Some studies have indicated that third-generation implants have demonstrated to be much more durable than second-generation implants [13, 15]. However, there was no proof of any relationship between tissue silicone levels and capsular contracture [15, 16].

More recently, fourth and fifth generation implants have been developed [8]. These implants have a more cohesive gel filler combined with thicker shells, and different surfaces (both textured and smooth) [3, 8]. In order to minimize the amount of free low molecular weight molecules available to pass into the surrounding tissues through the silicone shell, the gels of the latest generation of implants are highly cross-linked [4].

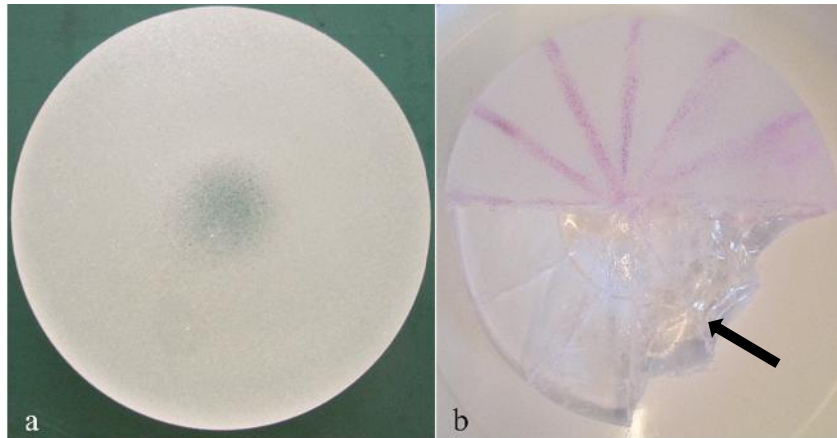
The textured shell was developed with the aim to reduce capsular contracture, since the tissues adhere better to rough surfaces. Also, the post-surgical mechanical behaviour of implants filled with silicone gel is more similar to natural breast tissue.



**Figure 1.** Breast Implants Evolution. a) First Generation, b) Second Generation, and c) Third Generation. Adapted from [7, 8].

Modern generation breast implants can be divided into different categories, such as surface (textured, micro-textured and smooth), shape (round or anatomical), fill (cohesive gel or saline), and implant dimensions (height/width, projection and volume). These implant characteristics allow individualization for each patient depending on the patient's tissue quality/quantity and tissue-based bio-dimensional assessment. The choice of breast implant is dependent on the specific surgical indication along with patient and surgeon preferences [6].

Based on the review about the breast implant evolution, the implants used in the various studies throughout this thesis were: modern generation, textured shell, round shape and with cohesive gel (Figure 2). All of them had different volumes.



**Figure 2.** Modern Generation with round shape a) The textured surface and b) cohesive gel is visible (the arrow points to the cohesive gel).

## **2. Mechanical Interaction between Tissue and Implants**

After augmentation or reconstruction, there is a need to increase the existing knowledge of the breast to improve existing protocols of clinical examination, like sensibility in the decision process, diagnostic and surgical planning. For these reasons, a comprehensive knowledge of the mechanical properties of the breast tissues is important for studying the effect of plastic and oncoplastic surgery techniques for breast reconstruction as well as for design of cosmetic breast implants.

Breast augmentation may cause complications in the long-term. For instance, postoperative tissue stretch deformities may appear. These complications are responsible for many potential risks, such as breakdown of the inframammary fold, permanent tissue atrophy, visible implant edges, and sensory loss [17].



In augmentation mammoplasty one of the complications is the malposition or displacement of the prosthesis. This phenomenon can be seen when the patient's breast and skin are very soft and easily stretchable [18,19]. The displacement of the prosthesis was not noted in patients with firm or normal breasts. Therefore, it is important to evaluate the elasticity of the skin and other breast tissue (glandular and fat tissue) before designing the placement of the breast implants. These factors highlight the importance of measuring the elasticity of breast skin for surgery [18]. A survey performed among aesthetic plastic surgeons [20] found that skin elasticity ranked first among the vital preoperative considerations in breast augmentation.

In fact, the relationship between breast tissues and the mammary implant is a reciprocal stress and strain [17]. A detailed understanding of breast tissue dynamics is important for a long lasting and rewarding breast augmentation. Hence, a literature review of the investigations that have been made on mechanical behaviour of different breast tissues was done in by Ramião et al. [21].

Many of the factors affecting the results of the final implant shape are not well understood by breast implant manufacturers, nor by the plastic surgery community. Some of these factors are the effect that the implant shape (round vs anatomic) has on the final breast shape, how the implant distributes its volume in the tissues and how the implant and breast tissue change over time [22].

Ramião et al. [21] showed that the mechanical behaviour is different among the tissues, and is highly dependent on the tissue preload compression level. For instance, when the woman is subject to a breast augmentation, the breast is loaded with an implant which undergoes instant deformation [17]. On the other hand, the stretched breast exerts a load on the implant [17]. This deformation and load will not remain constant over time because, among other factors, the augmented breast and the implant will undergo a

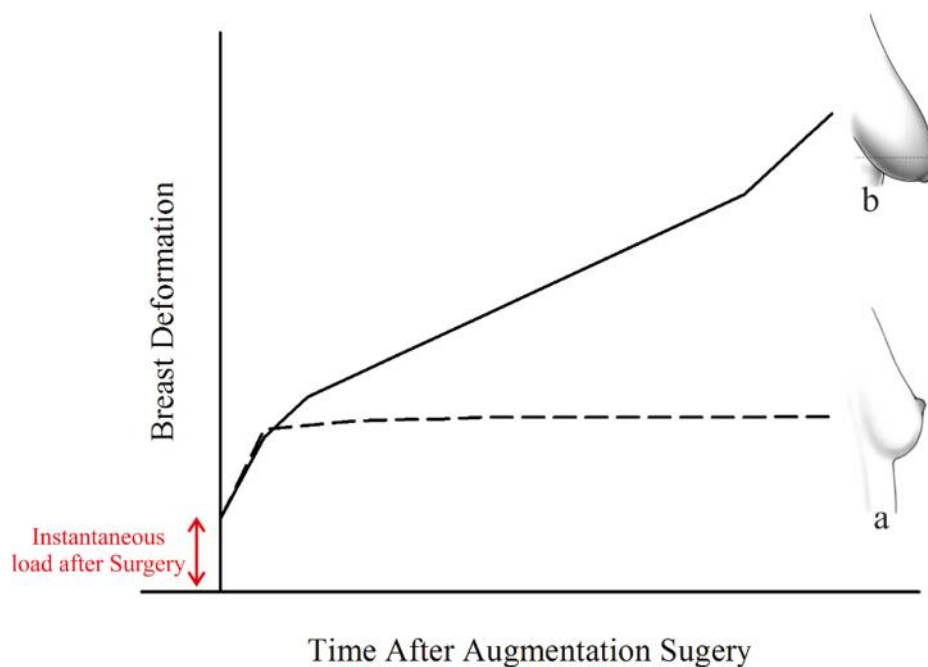
variable amount of creep deformation. This can be associated to the distribution of various tissues during women's life cycle, which undergoes periodical changes that depend on factors such as age, menstrual cycle, pregnancy/lactation, weight, body mass index, hormone therapy and menopause. Such alterations are expected to have an effect on the biomechanical properties of breast tissue. Summarily, the creep deformation of the augmented breast is dependent on three factors, the implant size (volume), the ever-changing supporting structure of the breast (coopers ligament) and aging (time).

Several authors showed that implant soft tissue dynamics affect the short and long-term results of augmentation in both primary and reoperation cases [17, 22-26]. For the best performance of breast augmentation, two concepts should be considered: compliance/stiffness and resilience/creep.

For instance, women who need an augmentation mastopexy, because they suffer from empty hypoplastic breasts after pregnancy/nursing, may have poor tissue support, low-stiffness/ high compliance/ low-resilience breasts. These women are also at risk for intense creep deformation when loaded with large or high-projecting implants [17] (Figure 3). This type of breast, with poor tissue support, accepts large/high-projecting implants. However, due its low resilience, it might easily undergo to a substantial amount of creep deformation over time (Figure 3) [17]. Therefore, the choice of an implant with a medium volume and projection is fundamental to avoid creep deformation. This reasoning also applies to women with small breasts, excellent tissue support, and high-stiffness/low-compliance/high-resilience breasts. This type of breast is not easily stretched to the point of accepting a large/high-projecting implant due to its excellent tissue support and, for the same reason, will not easily soften with time after the augmentation [17]. Therefore, in these cases a "tight" breast augmentation will remain tight for a long time because of its low compliance and high resilience. Furthermore, a

stable long-term result is very predictable in these cases because postoperative long term creep deformation will be low [17].

Questions about how to evaluate mechanical tissues at the breast level, as well as whether or not they can influence the rupture of the implants, have to be discussed and worked out in the future. Henceforth, future research should be undertaken in order to expand this area, and to guide surgeons and implant manufacturers to provide patients with safer and longer-lasting good results from implant-based breast augmentations.



**Figure 3.** Explanatory scheme of deformations over time between the breast tissues and the implant for two different cases. The lines represent the behaviour between the tissue and implants. For case a) the breast has an excellent tissue support, high-stiffness/ low-compliance/high-resilience breasts, and so it's very likely to maintain the postoperative result in the long term. In case b) the breast has poor tissue support, low stiffness/ high compliance/low-resilience, so are at risk for intense creep deformation when loaded with large or high-projecting implants. Adapted from [17].

### 3. Mechanisms of Implant Failure

A considerable amount of literature is available in the area of chemical and physical characterisation of breast implants in general, as well as PIP implants. Considerable research has been conducted on the mechanical properties of implant shells in an effort to find an explanation for the rates of rupture. Current section reports the results of breast implant physicochemical characterisation studies that have been conducted [27].

Proposed causes for implant rupture include:

- damage from surgical instruments;
- shell swelling;
- fold flaw;
- trauma to the implant such as a force to the chest or closed capsulotomy;
- physical and chemical features of the initial implant;
- the implantation procedure and site;
- time since the implantation and associated material degradation;
- personal-specific factors such as lifestyle, and sportive activities or accidental mechanical overexposure.

For a long time, some published studies have indicated that implant failure can be explained on the basis of *in vivo* degradation of the shell's mechanical properties when exposed to a biological environment [28-31]. A negative correlation between implant duration and mechanical resistance was found, suggesting a shell degradation.

The other cause for implant failure that was explored in the literature was focused on the time-dependent phenomenon of the silicone shell's swelling. The swell phenomenon is described as a decrease in shell strength due to migration of silicone fluid from the gel into the shell.

Various papers by Brandon and colleagues [32-36] report the mechanical properties of various generations of implants manufactured by Dow Corning. The authors confirmed the degradation in mechanical properties due to swelling but showed that the crosslink density remained constant. In 2006 [37], they postulate failure at the site of implants folds as an etiology of implant rupture, and reported that implant folding is thought to be more common in the presence of capsular contracture of long duration.

More recently, Necchi et al. [38] studied the failure of 100 implants of a recent generation, comparing intact (n=67) and ruptured (n=33) implants that were implanted during 6 to 13 years. The gel was carefully removed from the shell, and the shell gently wiped using isopropyl alcohol-moistened Kimwipes (Kimberly–Clark Corp., USA) from explanted and control implants. Then, three samples for the tensile test (Die C half-scale, see ASTM D 412-06ae2 Standard, 2006) were cut. The tests were carried out in displacement control at a cross-head velocity of 250 mm/min. A preload of 0.2N was imposed to uniform the specimens' initial stress state. After tests, the authors found a significant decrease in shell strength of ruptured versus non-ruptured implants, and compared this change to shell swell based on increased fraction of extraction in the ruptured implants.

Handel et al. [39] reported that the most common cause of implant rupture is damage caused by surgical instruments during placement. In this article, Mentor and Allergan data shows that 50–64% of ruptured implants were reported to be damaged by surgical instruments. In addition, the authors suggest there is a critical need to implement uniform statistical methodology using follow-up data only through the patient's last magnetic resonance imaging scan, as rupture rates can vary greatly depending on the statistical methodology selected. Brando et al. [40] confirmed the results of Handel et al.

[39] through micrograph images. These authors showed micrographs that clearly demonstrate that the shell striations present patterns similar to a scalpel cut.

Since March 2010 until the time of writing, most studies about implant ruptures have focused on the French brand PIP. Several studies have attempted to delineate specific flaws in the implants, and extrapolate the points at which quality control may have failed. However, there were few physical and chemical studies on PIP implants, but there is a large number of retrospective studies.

There are still some questions in the published literature regarding the specific flaw in the PIP implants that lead to their failure. This is in part due to a lack of readily accessible information regarding the specific processes used in the manufacture of the PIP implants [41]. A major challenge in analysing data concerning PIP implants is that there appears to have been variations in the quality of the implants between different batches [27] that were released to the market.

There was no indication from the available data that the demographic profile of patients (women) who have had PIP breast implants differ from women with implants from other manufacturers [27].

Based on peer-reviewed published studies, the probability of rupture for PIP implants is estimated to be around 14.5% to 31% after 5 to 10 years of implantation, while other silicone breast implants have been reported with a rupture rate of 1.1 to 11.6% after 10 years of implantation [42-47].

Therapeutic Goods Administration (TGA) [48] analysed explanted and control (new) PIP implants. For control PIP implants, the TGA presented that the mechanical properties of shells meet the requirements of applicable international standards. However, explanted PIP implants exhibited a decrease of the tensile strength of the shell comparatively with control implants.

Yildirimer et al. [49] compared 18 explanted PIP implants (15 intact and 3 ruptured implants) with four medical-grade silicone control implants. These implants were subjected to mechanical tests and Fourier transform infrared spectra (FTIR). Mechanical properties were assessed according to the standard ISO 37:2005. For tensile testing, dog bone-shaped pieces type 3 (shaft length 20 mm, width 4mm, six pieces per implant) were obtained from the shell. Uniaxial tension was applied to either end of the specimen at an extension rate of 100mm/min until failure. FTIR spectra were obtained on a Jasco FT/IR 4200 spectrometer (Jasco, Great Dunmow, UK) equipped with a diamond attenuated total reflectance (ATR). A total of six pieces per shell or gel were analysed. Spectra were produced from an average of 20 scans at  $4\text{cm}^{-1}$  resolution over a range of  $600\text{-}4000\text{cm}^{-1}$  wave numbers. The authors showed that PIP silicone shells have significantly weaker mechanical strength, when compared to medical grade controls. However, it must be taken into account that the control implants were from another brand, as it is desirable to compare the properties of the explants with those of lot-matched controls from the same brand. FTIR analysis demonstrated changes that suggest degradation of the Si-O-Si cross-links of the silicone in the PIP shells to Si-OH; such change correlated to the implantation time. The explanted implants gels showed a variable consistency, ranging from highly cohesive to soft, non-adhesive and prone to breakdown on manual handling compared with gels taken from intact PIP and control implants. The probable reason for this is that *in vivo* exposure of the silicone gel leads to degradation and cross-link scission.

Swart et al. [50] compared 19 explanted ruptured PIP implants with two control PIP implants. Specimen preparation and investigations followed the protocol for “*analysis of breast implants*” published by Brandon et al. [32-36] with the exception being that mechanical properties were assessed according to the international standard for mammary implants (ISO 14607:2007). Five dog-bone shaped specimens were cut from each implant

shell using a die described in ISO 37:2005. The authors observed that key properties, such as shell thickness displayed significant variation within sample and between samples of PIP implants. They found areas on the surfaces of nearly all the explanted devices where the absolute minimum thickness of the shell was below 0.57 mm, which was the minimum specified by the manufacturer. This higher variability could partly explain the higher early rupture rates of PIP implants.

Schubert et al. [51] compared 23 explanted PIP implants (13 textured and 10 micro-textured) with 2 different brands. The mechanical properties of the shell were analysed according to DIN 53504. Small dog-bone shaped specimens (shape S3) were chosen to allow an investigation of the homogeneity of the mechanical properties. Each specimen was stretched to failure at a constant crosshead speed yielding of 20 mm/min. One new implant from each of the two brands was analysed together with five explants. The authors reported that the tensile strength properties differ significantly between textured and micro-textured PIP implants, and in particular, a micro-textured shell seems to be more resilient.

Amoresano et al. [52] analysed 16 explanted implants from different manufacturers. The implant shells have been examined following the standard procedure reported in ISO 104302:2012. Three to five samples for each implant has been cut into a dog-bone shaped, 50mm long, from the implant shells, and measurements between 0 and 4MPa were performed using a tensile tester. The study revealed that the PIP implants had the greatest weakening compared to all other samples. PIP implants showed breaking points at the lowest strain value with respect to all others, which again was further indication of poor mechanical properties. However, it is necessary to note that the experimental protocol was different from other studies, and analysed only two PIP implants out of 16.



Beretta and Malacco [53] compared the filler materials of a ruptured explanted PIP with a control McGhan implant and a sample of technical-grade silicone. The authors analysed these samples using rheological techniques, attenuated total reflectance infrared spectroscopy, nuclear magnetic resonance, gas chromatography coupled to mass spectrometry and flow injection electrospray mass spectrometry. They found significant amounts of cholesterol and a lack of cross-linkages from gel of PIP implants, which corresponded with the lack of cohesiveness compared to medical-grade silicone.

In conclusion, given the lack of standardized screening and reporting, as well as the multiple implant generations and manufacturers available, it is difficult to reach comparable rupture rates across implant types and manufacturers. Thus, the significant differences in methodologies, experimental protocols, variations in patient evaluation, as well as variable length of implantation duration reported in rupture data, make it difficult to directly compare rupture rates across manufacturers and implant types (surface, shape, volume, etc.).

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# Chapter III

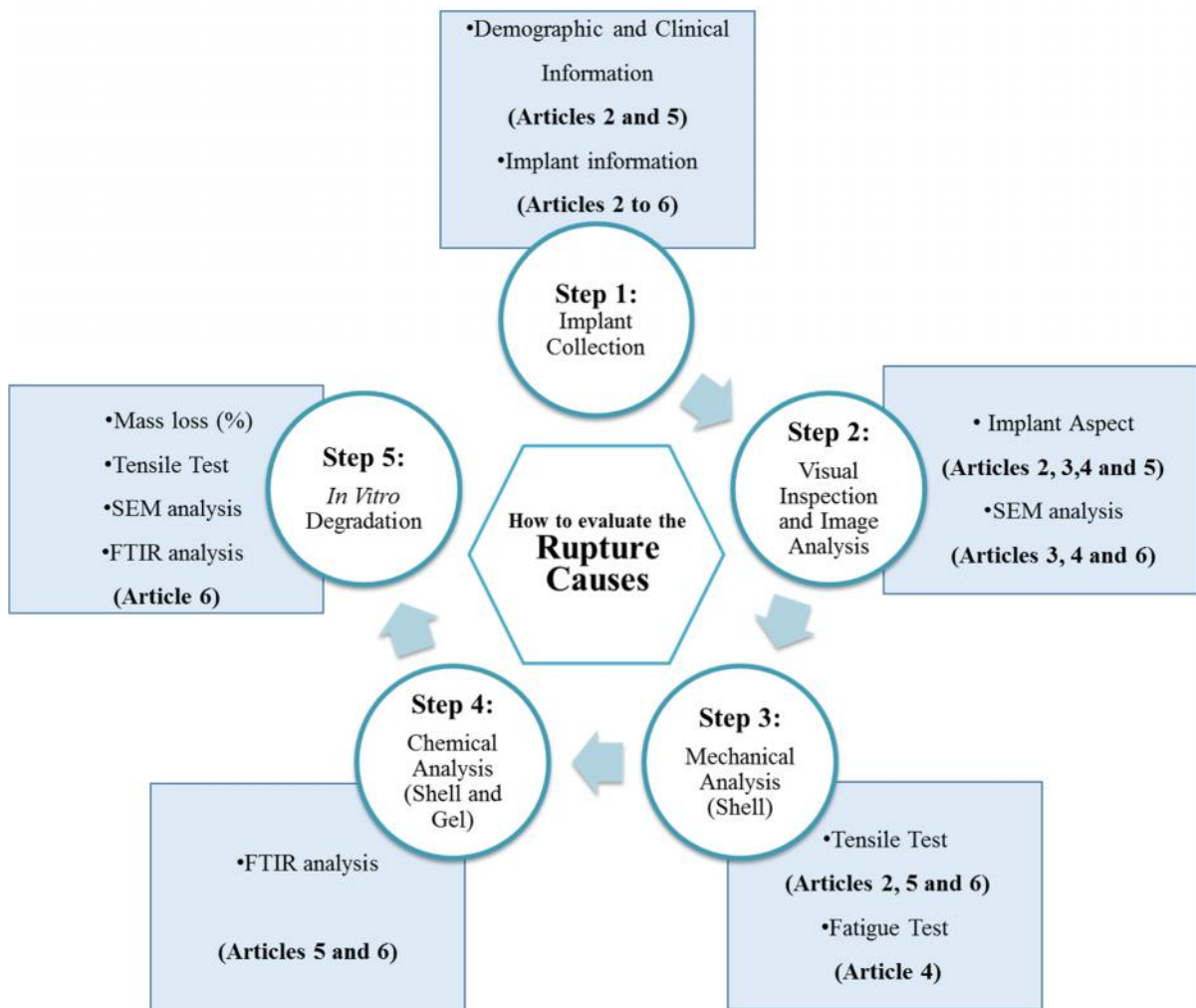
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Research Methodology



## Research Methodology

The approach taken to investigate this project mainly consist in the elaboration of experimental protocols to analyse breast implants in order to further understand the cause of their rupture. The summary of the design methodology is found in Figure 1.



**Figure 1.** Exploratory steps to evaluate the ruptures causes, with description of the methodology used.

The durability and useful life of a breast implant is of major concern to both the patient and the plastic surgery community. The influence of complex effects in the properties of breast implants, both chemical and physical, are discussed in this thesis. A wide variety of methods were used to determine the changes occurring in PIP implants when compared with another brand (Brand X). A confidentiality agreement was signed for Brand X, and all obtained data can only be used for academic purposes. Explanted implants (intact and ruptured) were directly compared to available control implants. Explants and controls were studied by a broad combination of mechanical testing, chemical and surface analysis. The experimental protocols were established according to the standard ISO 14607 (Non active surgical implants – mammary implants – particular requirements), and the Food and Drug Administration (FDA) guidance document [1] for breast implants. The following methodologies were delineated to fulfil the defined objectives:

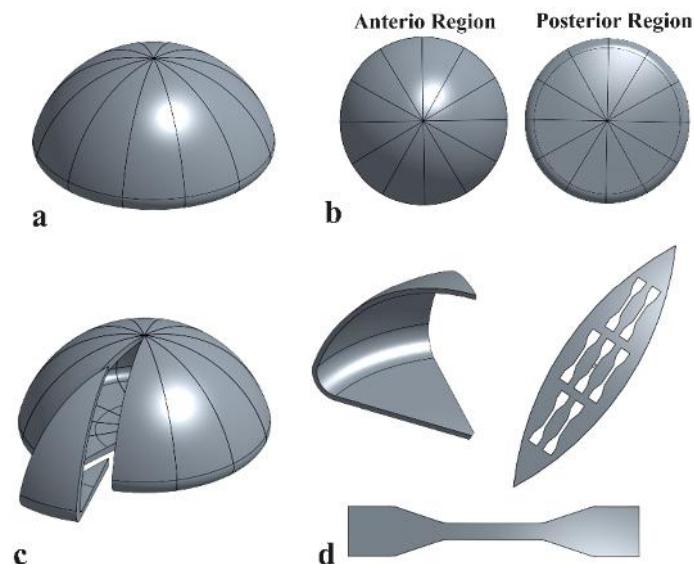
- Demographics and baseline characteristics were collected at the beginning of the study. This information was divided in two groups: (1) patient demographic and baseline characteristics (per patient basis), and (2) Implants' baseline characteristics and surgical baseline characteristics (per implant basis). For the first group, detailed information was collected regarding age, height, weight, profession, pathologies/ chronic diseases, prior surgical interventions (date of surgery and surgical intervention) and explantation surgery data (date and reason for surgery). The second group's gathered information was about implant surface, shape, volume, region of rupture, implant aspect, implant position (e.g., retromuscular, subglandular) and implantation time. This information is described in **Articles 2 and 5**, included in Chapter V. Detailed data about implant surface, shape and

volume is presented in **all Articles (2 to 6)** for the main characterization of the implant.

- After the collection and organization of demographics and baseline characteristics, the explanted implants were analysed. The explants were manually disinfected using alcohol wipes. Following disinfection, the explants were visually examined, and the presence and appearance of any shell rupture (hole, split or v-shaped), discoloration, opacity or other unusual features were recorded. The shell and gel integrities were classified in relation to shell damage and gel condition, according to the criteria of the Department of Health Therapeutic Goods Administration (TGA) [2]. This classification can be seen in **Articles 2, 3, 4 and 5** included in Chapter V. All of them had the purpose of characterizing implant ruptures through visual inspection and relate them with physical-analytical methods that are described below.

- To analyse the integrity of the implant shell, the mechanical testing protocol was developed according to the international standard for mammary implants (ISO 14607:2007), and the standard for rubber, vulcanized or thermoplastic for determination of tensile stress-strain properties (ISO 37:2005). The samples' preparation presented by Schubert et al. [3] was adopted. Each implant was characterized in all its regions (anterior, equatorial and posterior). Each of the shells was divided into 12 segments counted clockwise from 1 to 12 (Figure 2 a, b and c) and from each of the segments between five and nine specimens (depending on implant size and shape) were prepared (Figure 2 d). Following this procedure, each implant provided a minimum of sixty (60) samples. For these samples, type 4 (ISO

designation) dog-bone shaped specimens (shaft length 12 mm, width 2 mm) were used. The mechanical properties were obtained from uniaxial tensile data. Before the tensile test, the samples were subjected to a 0.25N preload, which allowed to control the initial geometry and loading conditions, therefore contributing to the reproducibility of the experimental procedure. The samples were tested until failure, at a constant displacement rate of 20mm/min. This experimental protocol was used in the majority of the published articles, and we can see their results in **Articles 2, 5 and 6**.



**Figure 2.** Schematics of the experimental procedure: (a,b,c) implant segmentation into 12 segments, (d) example of sample preparation for tensile tests of a segment.

- Scanning electron microscopy (SEM) was used to analyse the ruptured PIP breast implants. Several samples were cut from the rupture region for examination by SEM at CEMUP (University of Porto, Portugal). Samples were coated with an Au/Pd thin film, by sputtering, using the SPI Module Sputter Coater equipment, during 120 s and using a 15 mA current. This analysis is described in **Articles 3 and 4**.

- The research made to determine the causes and mechanisms of rupture of PIP explanted breast implants led to evidence that fatigue phenomena may be associated with the initiation of implant rupture. In order to verify this phenomena, a fatigue test protocol was carried out to simulate a mechanism of fatigue crack growth. Two parameters were used in the fatigue test: displacement and frequency. The displacement amplitude was 15mm, equivalent to ~20% strain in the narrow region of the specimen. This displacement was calculated by the tensile tests carried out in control implants in **Article 2** and **5**. The samples were tested at 1 Hz because it is similar to that of walking or a beating heart [4]. This approach is described in the **Article 4** included in Chapter V.

- The characteristics of the degradation of the material are directly linked to the breakage of crosslinks, and can be seen by Fourier Transform Infrared Spectroscopy (FTIR). Thus, the chemical composition of the gel and shell of explanted implants (intact and ruptured) were analysed by FTIR equipped with a diamond attenuated total reflectance (ATR) accessory. Spectra were acquired over 20 scans with wavelengths ranging from 600 to 4000 $\text{cm}^{-1}$ , with a resolution of 4  $\text{cm}^{-1}$ . This analysis is described in **Article 5** included in Chapter V.

Several studies [5-12] pointed, as primary factor for *in vivo* implant shell failure, the material degradation/change of properties during implantation time. This degradation involves multiple physical and/or chemical processes. Although biodegradation effects had been identified through the explanted implants, *in vitro* biodurability testing of PDMS breast implants is not well documented in the open literature. The purpose of this study was to characterize the degradation of the breast implant under physical/chemical

conditions of the human body (temperature and pH): a 'normal' physiology (pH = 7.4) and an inflammatory process (pH = 4.0), in order to understand whether the material was degraded or not with the conditions imposed. This approach is described in **Article 6** included in Chapter V, which comprises the following stages:

- *In vitro* degradation process was conducted in accordance with ISO 10993 "Biological evaluation of medical devices". To study *in vitro* degradation processes of the implant shell, Phosphate Buffer Solution pH 7.4 (PBS) and Potassium hydrogen phthalate buffered (HP) with pH 4.0 were used as a model of biological fluid (normal and inflammation). Samples were kept in the buffered solutions at 37°C during twelve (12) weeks. After the degradation process began, a batch of samples was removed periodically from the thermal bath for their weight to be measured.

- Analysis of the mechanical performance of the shell after degradation (stage 0 and 12 week) followed the mechanical protocol described above.

- Analysis of the surface, via SEM, followed the protocol described above. In this case, the shell surfaces were analysed to verify if there has been any changes due to degradation.

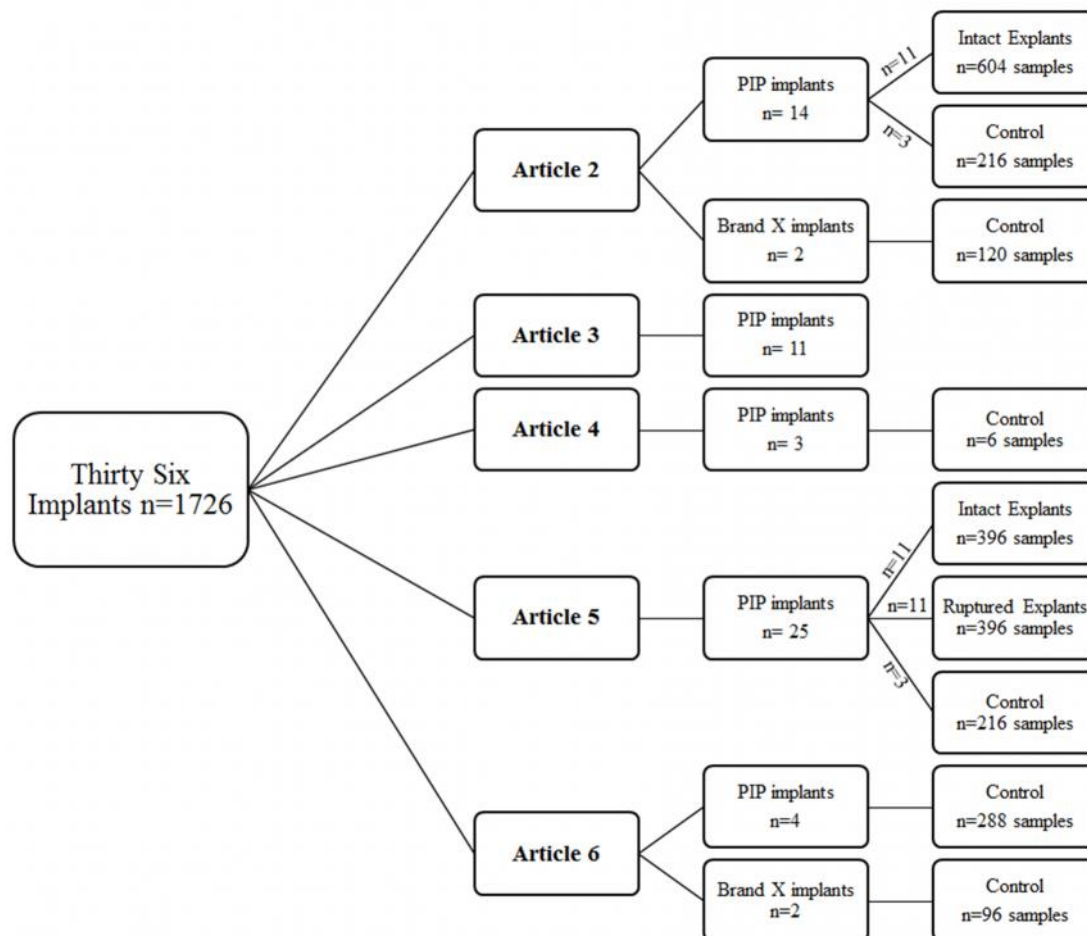
- Analysis of the chemical composition FTIR, followed the protocol described above.



Following the description of the performed methodology, a brief description of the number of implants and samples studied in this work is presented below.

In this thesis, a total of thirty six (36) implants from two different brands were tested (PIP and Brand X implants). Twenty two (22) explanted PIP breast implants were analysed: eleven (11) had intact shells and the remaining eleven (11) had ruptured shell. Ten (10) PIP implants and four (4) Brand X were used as control. All implants were from different lots.

A total of 1726 samples were removed from the implants' shell in order to complete this research. The mechanical tests (described above) were the main tests to analyse the strength of implant material. According to the main objectives for each articles/studies, the samples removed from each group are shown in Figure 3.



**Figure 3.** Scheme divided by articles about the number of implants and samples were analysed.

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# Chapter IV

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Review Article



## Article 1

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### **Biomechanical Properties of Breast Tissue, a State-of-the-art Review**

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## **Abstract**

This paper reviews the existing literature on the tests used to determine the mechanical properties of women breast tissues (fat, glandular and tumoral tissue) as well as the different values of these properties. The knowledge of the mechanical properties of breast tissue is important for cancer detection, study and planning of surgical procedures such as surgical breast reconstruction using pre-surgical methods and improve the interpretation of clinical tests.

Based on the data collected from the analysed studies some important conclusions were achieved: (1) the Young's modulus of breast tissues is highly dependent on the level of tissue pre-load compression (2) the results of these studies clearly indicate a wide variation in moduli not only among different types of tissue but also within each tissue type. These differences were most evident in normal fat and fibroglandular tissues.

**Keywords:** Breast tissue, Mechanical properties, Compression loading, Elasticity modulus.



## 1. Introduction

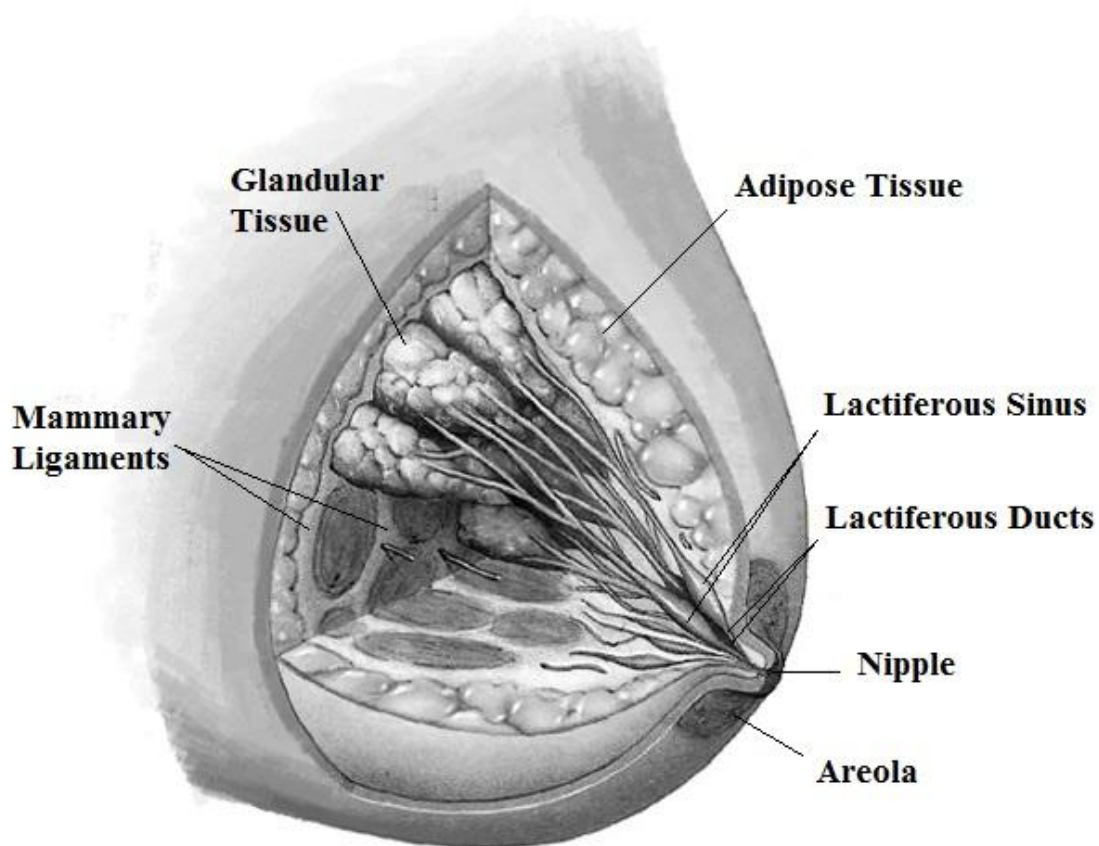
The breast is an important organ in women's body, since it contains glandular tissue essential for the production and secretion of milk. The breast is a heterogeneous structure containing different tissue layers (Figure 1), however, the predominant types of tissue within the breast are fat and glandular tissue. Each breast contains 15–25 lobes of compound glands that are embedded in fibrous and adipose tissues. These lobes, each containing an excretory duct that drains into the lactiferous sinus, radiate from a central nipple-areolar complex.

The breast is firmly attached to the skin and underlying structures by fibrous bands referred to as suspensory ligaments (Cooper's ligaments), which provide the functions of support, hold the breasts in place and contribute to determine the shape and contour of the breast.

The distribution of various tissues during women's life cycle undergoes cyclic changes that depend on factors like age, menstrual cycle, pregnancy / lactation, hormone therapy, menopause, among others. Some of these changes have profound effects in tissue's structure and morphology.

Such alterations are expected to have an effect on the biomechanical properties of breast tissue. For instance, a stretching of Cooper's ligaments and a weakening of the coupling between the breast and the surrounding tissues are observed with increasing age. An important property of soft tissues is their intrinsic elasticity, which may change under the influence of pathophysiologic processes, such as tumor development [1]. Breast cancer is the most common female disease worldwide, currently affecting approximately

1.38 million women per year. Worldwide, breast cancer comprises about 25% of all types of cancer in women [2]. Their prognosis and survival rates depend mostly on the type and stage of breast cancer. Early detection leads to a more effective treatment and improvement of the survival rate [3,4].



**Figure 1.** Anatomy of Breast.

In this context, the mechanical properties of breast tissue play a prominent role in the research related to several clinical, pre-clinical, as well as current applications such as self-diagnosis through palpation. These applications include cancer detection, mechanics of injury, surgical simulators and tumor motion tracking during surgeries. For these, engineering has contributed to improve clinical examination protocols, through improvements on the diagnosis, surgical planning and decision supporting tools. Some

studies are based on biomechanics' concepts using finite element modeling (FEM). These numerical models are differentiated, primarily, by how the breast geometry is discretized, application of boundary conditions, and/or knowledge of the breast tissue material properties. In most studies, large deformations were considered, and information on patient-specific breast morphology and on elastic-tissue properties was required. To improve the outcome of breast needle biopsy, Azar et al. [5] developed a model of the breast to predict tissue deformation during the procedure, and Carter et al. [6] presented a model that can potentially be applied for image guided surgery.

Roose et al. [7] presented a computational model capable of simulating the postoperative shape of the breast with up to 1cm accuracy after a subglandular breast implantation.

Unlu et al. [8] developed and tested a new computerized finite element method (FEM) based on a 3D non rigid registration of PET and MR breast images. This simple method was proposed to facilitate the nonrigid registration of MRI or CT images of any type of soft-tissue to their molecular counterparts such as those obtained using PET and SPECT.

There are several challenges associated with localization of suspect lesions in the breast in an MRI exam. These difficulties include patient positioning, visibility of the lesion that may fade after contrast injection, menstrual cycles, and lesion deformation. Stewart et al. [9], Azar et al. [5], Samani et al. [10], Carter et al. [6], Unlu et al. [8] and Pathmanathan et al. [11] are examples of some authors that developed patient-specific finite element (FE) breast models obtained from diagnostic MR images, with potential for patient-specific therapeutics.

The ideal approach to achieve high quality patient-specific simulations should include the *in vivo*, and ideally non-destructive estimation of the mechanical properties.

Han et al. [12] performed *in vivo* material parameter estimation by ultrasonic indentation tests on breast tissues. Another *in vivo* experimental technique was used by Buijs et al. [13], consisting of a model to predict target displacements using a combination of ultrasound elastography and finite element (FE) modeling. This technique can help pre-operative planning of minimally invasive surgical interventions.

While significant research has been conducted to develop techniques to measure the elastic modulus of breast tissues, little research has been focused on their hyperelastic mechanical behaviour.

Following, the ability of FE models to predict *in vivo* behaviour strongly depends on the accuracy of the mechanical properties of tissue components. An accurate breast model has proved to be very difficult to implement, due to several difficulties associated to breast tissues: (1) the complex morphology; (2) the patient-specific variability, (3) the highly nonlinear (hyperelastic) mechanical behaviour and (4) the difficulty of measuring elastic properties of different types of tissues in the breast [10,14-18].

However, in order to make the biomechanical models predict more realistically *in vivo* behaviour and help improving clinical and pre-clinical applications (for example, cancer detection), several authors studied the mechanical properties of the different breast tissues. Thus, throughout this article we provide a review about the techniques to measure the mechanical properties *in* and *ex vivo*. These measurement types are subdivided in two categories *in vivo* and *ex vivo*. Regarding *ex vivo* techniques, it is given a comprehensive review of the test protocols (compressive tests) used in various studies as well as the obtained biomechanical results [10,14,19-22]. Considering *in vivo* testing the approach is elastography, a noninvasive by nature and based on an imaging technique [23-27].

All these studies have the purpose of simulating and assisting the diagnostic methods used in clinical breast examination. In most breast examination methods,

compression is applied to help detecting lesions, and sometimes it becomes necessary to apply higher compression loads to investigate stiffer regions. This situation creates the need to know more about the mechanical properties of breast tissue under compression. Despite the importance of compressive loading and its contribution to the characterization of tissue in applications of cancer detection, there are few studies that have focused on the mechanical behaviour of breast tissue in response to compression.

Widespread adoption of such techniques (*in vivo* and *ex vivo*) associated with biomechanical modeling and imaging techniques of the breast have the potential to significantly reduce the numbers of misdiagnosed breast cancers and enhance surgical planning for patient treatment.

The main objective of this review is gathering the mechanical properties of breast tissues (adipose, glandular, and tumor), available to date in the literature, through different *in vivo* and *ex vivo* tests, enabling as well the identification of the relationship between tissue properties and pathological mechanics.

An accurate knowledge of the breast tissues' mechanical properties allows realistic simulations by finite element modeling and improvement in clinical exams for breast cancer (screening, diagnosis and monitoring tests), thus it opens possibilities for medical applications such as surgery planning and surgery outcome simulation.

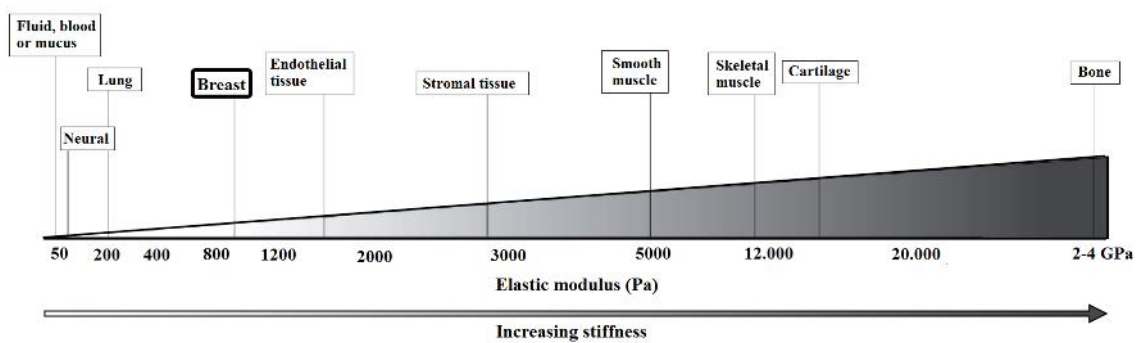
This paper is organized in five sections that evolve from the simplest concepts of breast tissue characterization to the state of the art testing techniques used. Section 2 details the characterization of breast tissues, and addresses the main experimental challenges involved. Section 3 analyses the differences and specificities of *in vivo* and *ex vivo* mechanical techniques. Section 4 summarizes the mechanical experimental results of each breast tissue, and Section 5 includes a discussion and reached the conclusions.

## 2. Characterization of Soft Tissues – Basic Concepts

This section presents fundamental concepts to understand the biomechanical studies presented. The biomechanical properties of tissue (ex. stiffness/elastic modulus), vary markedly between organs and tissues, and are inherently related to tissue function (Figure 2).

Breast tissue has a unique rheology and optimum biomechanical properties, changing over the course of development in response to function (as during mammary gland lactation) or in pathological situations (such as tumours). Although breast tumours are much stiffer than normal breast, the material properties of breast tumours remain significantly softer than those of muscle or bone [28].

An important characteristic of breast tissue is their nonlinearity at high deformation [29]. For example, the tensile response of breast tissue exhibits a nonlinear stiffening while undergoing high deformations.



**Figure 2.** Stiffness in different soft tissue. Adapted from [30].

The mechanical characteristics of soft tissues, consists, in general, of a complex combination of elastic and viscous components [31]. This combination controls the deformation of tissue [32].

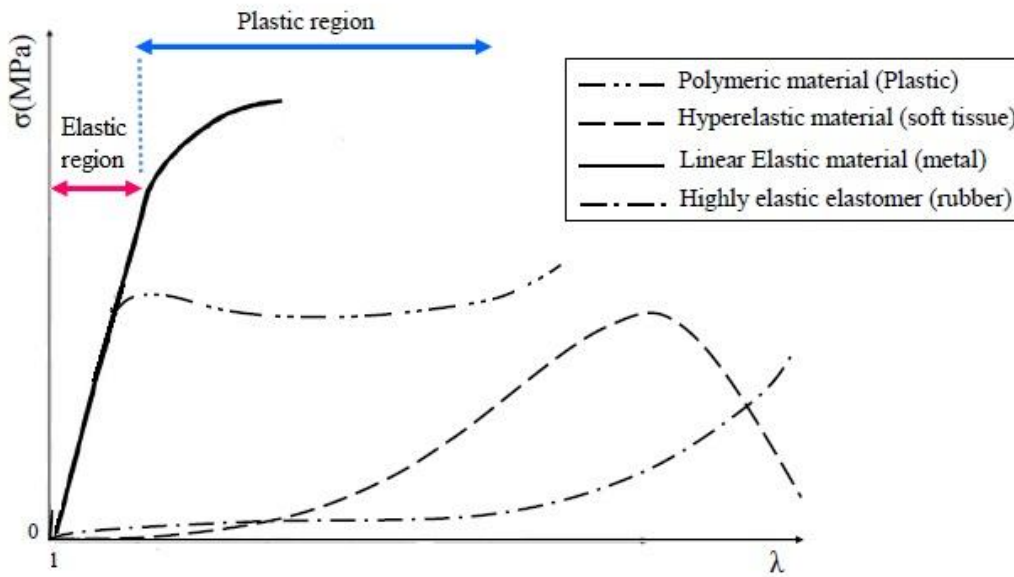


Regarding the classic elasticity theory, this represents the linear relation between stress (  $\sigma$  ) and strain (  $\epsilon$  ), given by Hooke's Law:  $\sigma = E \epsilon$  . In this case, the constant of proportionality ( $E$ ) represents the elasticity modulus, which is the slope of the stress-strain curve in the linear section (see Figure 3) – corresponding to the elastic region – and constitutes the mechanical parameter which indicates the stiffness of a material [31]. To characterize the tissue stiffness there are three types of elastic modulus defined by the tensile (Young's modulus), shear (shear modulus) and volumetric elasticity (bulk modulus) respectively.

The Young's modulus is the most commonly used to quantify stiffness, and will be used throughout article to characterize breast tissue. Therefore, according to the experimental protocol for measuring the mechanical properties of breast tissues mentioned by several authors [19,22,33,34] Young's modulus,  $E$ , is analyzed using equation (1) [19].

$$E = \frac{2(1-\nu^2)q}{w} \quad (1)$$

where  $\nu$  is the Poisson's ratio,  $q$  is the load density (force per unit area),  $a$  is the radius of the loaded area, and the  $w$  is the maximum displacement in the direction of the load. The most common mechanical analysis performed is the indentation test discussed ahead in this section.



**Figure 3.** Mechanical behaviour of linear elastic and hyperelastic materials.

Poisson's ratio ( $\nu$ ), measures transversal deformation relative to the longitudinal direction of load application and is defined as follows:

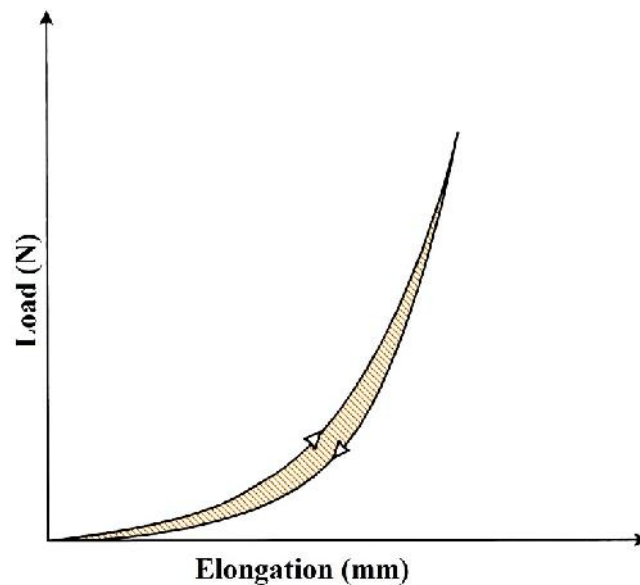
$$\nu = -\frac{\varepsilon_d}{\varepsilon_a}, \quad (2)$$

where  $\varepsilon_a$  is the strain in loading direction (axial) and  $\varepsilon_d$  is the corresponding strain in lateral direction. The Poisson's ratio is an intrinsic parameter of a material, and it is unique for different materials. For soft tissues which are quasi-incompressible due to their high (incompressible) fluid content, Poisson's ratio is  $\sim 0.5$ .

Mechanical properties of a tissue are also dependent on time and strain history. For this reason, stress- strain curves during loading and unloading do not follow the same path, and loading-unloading cycles are always different from one to the other, usually displaying a hysteresis effect, shown in Figure 4. This can be related to the viscoelastic phenomenon taking place when the load-deformation (stress-strain) diagram curve suffers a path deviation [31].

The effect of viscoelasticity results mainly from shear contact between collagen fibers, the proteoglycans and elastin component of ground substance.

Shear stress causes energy dissipation due to the recovery of the tissue after elongation or contraction, a behaviour that creates a hysteresis cycle, during loading and unloading stages of the test [35]. Moreover there is also microscopic sliding among collagen fibers while the tissue undergoes axial stresses [36,37].



**Figure 4.** The dashed is a hysteresis loop and shows the amount of energy lost (as heat) in a loading and unloading cycle. Adapted from [31].

All tests presented in this review contain repeated loading and unloading of the tissue sample which can reduce hysteretic effects, and can also soften the tissue. Pre-conditioning involves the repeatedly loading and unloading of the tissue so that a steady state is achieved for a given load cycle [31]. Pre-conditioning was performed by Krouskop et al. [19], Samani et al. [22], Samani and Plewes [10] and Wellman et al. [20]. Both Wellman et al. [20] and Krouskop et al. [19] found viscous effects to be negligible, although Wellman et al. [20] did note that some long time scale force relaxation was

likely to occur. This behaviour results from complex interactions of collagen fibers, elastin, proteoglycans and water within the tissue, and can provide us with additional insight into the composition of tissues and allow us to build sophisticated models of tissues.

The structure and mechanical properties are quite different in various soft tissues, vary significantly from one individual to another and can take different values whether measured *in vivo* or *ex vivo*.

Thus, according to the main structure of the breast tissues and the main objectives of each study, several authors opted by the compression [38] (unconfined or confined) and indentation tests [19,20,22,33,34].

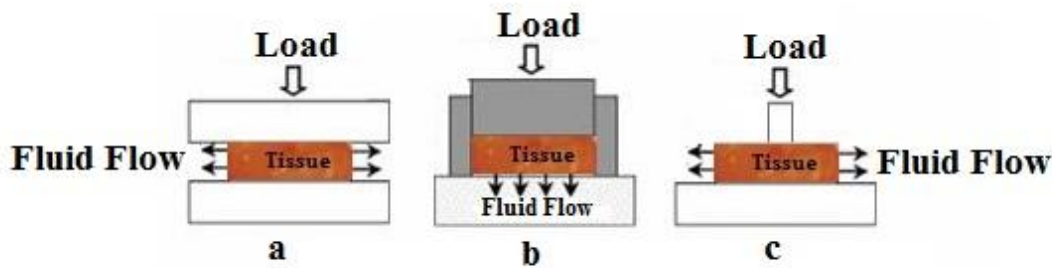
The unconfined compression consists on the application of a compressive load on the specimen which is fixed between two plates (as seen in Figure 5a). The compressor size should not be inferior to the size of the material sample tested. The tissue is then deformed in a direction parallel to the applied force (lateral).

Confined compression is similar to unconfined compression (see Figure 5b), but in this case the specimen is additionally constrained in the radial direction to the applied load. The specimen is placed in a chamber and a constant compressive load is applied to it. The additional constrain avoids free lateral tissue deformation developing a lateral pressure. On the tissue sample interstitial fluid can only flow axially through the surface into the filter in chamber, shown as in Figure 5 (b).

Indentation test is similar to compression test, a procedure where an indenter applies a compressive load to the tissue with a cylindrical, typically plane-ended or spherical-ended indenter. The resulting deformation of the external surface is recorded. The slope relating stress with strain (force-displacement) represents the compressive Young's

modulus ( $E$ ) shown in equation (1). In this case, the fluid flow outside the indenter-tissue contact point is possible in both lateral and axial directions.

The indentation machine uses a linear motor programmed to apply a user defined displacement. Therefore, deformations can be obtained directly from the test parameters. The load, applied with the indenter and the contact area indenter-tissue are known parameters. The main difference between indentation and compression test (see Figure 5c) is that indenter surface is smaller than the specimen testing surface.



**Figure 5.**-(a) Unconfined compression, (b) Confined compression and (c) Indentation test.

In punch indentation experiments the piston has a diameter of around 5mm which allows a tissue of a homogenous type to be tested, although the technique presented in Samani et al.[22] is suitable for samples which contain both normal and pathological tissue.

When breast tissue is compressed the strain increases rapidly corresponding to the elimination of free fluid. The elastic modulus becomes progressively higher with increasing strain [39].

The practical implication of this mechanical behaviour is that breast tissue needs to be statically preloaded and accurate measurement requires small increments in stress (i.e. in the linear region), to obtain reproducible and useful values of Young's modulus. In

addition, the testing conditions as well as tissue characteristics must be specified (for example: age of the tissue, its temperature, type of test...).

### **3. Experimental Techniques to Characterize Breast Tissue**

This section reviews the available literature on the stiffness of the breast tissues obtained experimentally. The discussion is divided in two parts: (3.1) *in vivo* techniques and (3.2) *ex vivo* techniques. With *in vivo* techniques tissues are tested with small strains or loads and all the changes induced are reversible. In contrast, *ex vivo* protocol involves larger strains or loads inducing non-reversible changes to the tissues. The relationships between the strain (a measure of deformation) and the stress (internal pressure) are reviewed for each breast tissue type [10,14,19-22,34,40-48]. Several authors have reported that mammary tissue has a nonlinear mechanical response [8,14,10,19,21,22,41-47,49-56]: the *in vivo* (elastography) and *ex vivo* (compression and punch indentation tests) testing methods found in literature are summarized in Table 1. The mechanical properties of the breast constituents have been characterized considering linear elastic Young's moduli to quantify the stiffness.

#### ***3.1 In vivo Techniques***

Palpation is an effective method for breast cancer diagnosis. It's a technique based on a qualitative assessment of the low-frequency stiffness of tissue, which is primarily useful to detect relatively large and superficial tumors.

However, this technique is not sufficiently sensitive with cases where the tumor is too small and/or its location deep in the body, precludes its detection and evaluation by palpation. In some cases, the examiner may be inexperienced or the signs are not clear,

so the result of palpation becomes doubtful. Therefore, other qualitative methods are required to detect the presence of abnormalities.

To help detect relatively large and superficial tumors, an imaging technique, called elastography, has been developed from the late 1980s to the early 1990s [51,57-59]. This technique provides quantitative information on tissue stiffness and is characterized by estimations of the elastic modulus [21,40,44,46,60].

Elastography helps estimating or assessing the non-invasively changes in the mechanical properties of the tissues under compression at a microscopic level [45]. This technology can be understood as imaging-based counterpart to palpation, commonly used by physicians to diagnose and characterize diseases.

Elastography is characterized by having a higher degree of specificity and sensitivity. It has the ability to detect the type of abnormality (tumor benign or malignant) and separate it from healthy tissues, for example separating the tumor from the adipose and fibroglandular tissue [49].

This technique provides insight into the elastic properties of biological tissues (when applying a small axial uniform compression) and of the strains resulting on site [51, 52, 60]. The goal of this technique is to create an image of the distribution of physical parameters related to the mechanical properties of the tissue by measuring the response or strain of the tissue resulting from the applied stress. The elasticity imaging methods consist in applying some form of stress or mechanical excitation to the tissue, and measuring the tissue response to this stimulus, and from this response calculating parameters that reflect the mechanical properties (see Figure 6) [50].

There are different approaches to elastography, either quasi-static or dynamic (transient and harmonic). The following references provide a comprehensive overview of these techniques [51, 61-65].

There are different methods of elastography depending on the tissue response measurement, namely ultrasonography/compression, MR and optical (Figure 6).

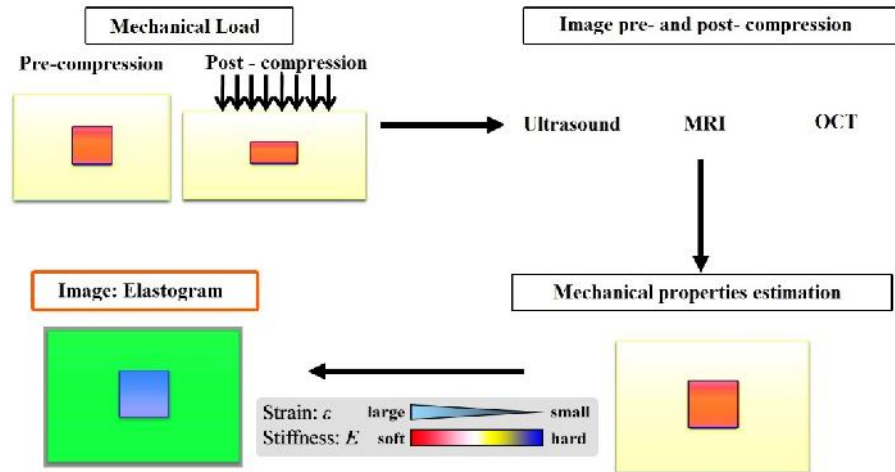
After the measurement of tissue responses to applied stress, using the elasticity imaging processes and the acquired data, it's possible to estimate the mechanical properties of the tissue. Typically, soft tissue is assumed to be isotropic, linear elastic, and Hookean when elasticity imaging techniques are employed.

Some studies intend to characterize the type of abnormality and increase the specificity of elastography, in an attempt to avoid unnecessary biopsies [55].

Changes in the stiffness of soft tissues are generally associated with the presence of pathology; malignant or benign breast tumors are usually stiffer than normal breast tissues, and malignant breast tumors are significantly stiffer when compared to benign tumors. Thus, the mechanical properties of breast tissues, measured by elastography, can help to detect the presence of abnormality in the breast (sensitivity), and also to classify the type of the detected abnormality (specificity) [55].

Depending on the particular elastography technique used, there is a range of false-negatives [33]. These errors could be reduced or minimized with a better understanding of the testing conditions; such investigations can be carried out using *ex vivo* techniques. As pre-compression, required to initiate elastography, influences the test results, an improved knowledge of its influence is critical. A modern and more accurate definition of pre-compression was provided by Umemoto et al. [34], “pre-load compression”.





**Figure 6.** An overview of elasticity imaging methods. Adapted from [32, 66].

### 3.2. *Ex vivo* Techniques

Other researchers have proposed mechanical tests to measure the mechanical properties of *ex vivo* tissue immediately after it is removed from the body [10,19,20,22, 43].

For example, Krouskop et al. [19] measured the elastic modulus of pathological breast tissues (fibrous, fat, glandular, carcinomas, intraductal carcinomas, and invasive ductal carcinomas) submitted to a uniaxial compressive force with pre-load compression levels of 5% and 20%, respectively. These tissues were tested with a sinusoidal load at three frequencies: 0.1, 1.0, and 4.0 Hz. The strain rate used during compression testing was selected so that viscoelastic effects were negligible. Wellman et al. [20] adopted a similar experimental methodology, but tested more breast tissue types. Wellman et al. [20] in their study used a test instrument for uniaxial compression and punch indentation of tissue, which applies repeated loads on the sample. Sarvazyan et al. [38, 43] studied the elastic properties of breast tissues through uniaxial compression test. The authors tested 20 specimens of postoperational material (adipose, fibroglandular and tumor

tissue) under compression between two plates. The intent of these studies was to characterize the viscoelastic behaviour and to confirm whether or not the tissue could be modeled as an elastic material within the frequency range of interest.

Samani et al. [10, 22] developed a complex system to measure the elastic modulus of normal breast and tumorous tissue (without the need to remove the tumors) from slices obtained after surgery. For normal tissue Samani et al. [67] developed a technique where small block shape specimens were indented and the resulting force–displacement slope was converted to the Young’s modulus using an FE model. For tumours, Samani and Plewes [68] (used a technique where tumours remained within tissue slices. The tumour is surrounded by normal tissues. The sample is indented and the resulting force displacement slope converted to the Young’s modulus iteratively using the tissue slice FE model. One major improvement of this technique is that tumorous tissue may be tested imbedded on normal tissue. The comparison between experimental (experimental phantom) and FE simulation (numerical phantom) data has associated errors. As the authors recognize, an ideal solution would include 3D MRI or CT scan, so that the naturally occurring variations in tumour shape and density can be reflected on the FE simulation (detailed FE mesh). This may be the reason why the authors state that they obtained a smaller error while analyzing larger tumours. A relevant thumb rule pointed by the authors is the 1(thickness):4(slice diameter) ratio of tissue slice dimensions. This is the minimal thickness: diameter ratio that allows a comparison between experimental results and the FEM simulation of a semi-infinite body. For practical purposes, if the tumours are small (less than a few centimeters) the errors due to the deformation of the surrounding (normal) tissues will be significant.

Matsumura et al. [33] and Umemoto et al. [34], measured the elastic property, young’s moduli, from surgically-resected breast tissue by material testing machine

(Instron) with 3mm cylindrical indenter. The breast tissues samples (glandular, adipose and tumour tissue) were cut and soaked in saline and heated for 5 minutes in the thermostatic chamber maintaining 45C° temperature. Then the samples were removed from the saline and placed on the testing stage which is kept under 37C° temperature (the surface of the sample is kept moist with saline). The authors used different compression protocols, with compression starting from zero-compression (zero-stress – 0kPa) up to a compression strain of 30% (50% in the case of fat or gland) with compression speed of 1mm/min.

In several studies of breast tissue, samples were properly preserved according to standard preservation procedures. Often the time gap between collection and testing did not exceed the period of two hours. There were no measurable changes in the data obtained after allowing the specimens to sit for periods up to two hours [19, 22].

The experimental techniques used to estimate the biomechanical properties of breast tissues during the last decades are summarized in Table 1.

#### **4. Mechanical Properties of Breast Tissue**

Over the past decades, several research works were performed to characterize the biomechanical properties of soft tissues, which are subject to some degree of mechanical activity [31]. However, very limited quantitative information is available on the biomechanical properties of soft tissues, which do not have an active mechanical function such as the breast [22].

Several studies [8,19,20,22,30,69] have quantified the mechanical properties of the breast constituents using Young's moduli to relate the stiffness to the type of tissue. These studies have shown that tumors are much stiffer than normal breast tissues. This occurs

because the tumoral tissues undergo collagen remodeling which leads to stiffening. According to Lopez et al. [70] “*the oriented, thickened collagen fibers along whose mammary gland tumor cells have been seen to migrate are indeed a source of the ECM stiffening*”. In order to develop tractable mathematical models, from which material properties can be extracted, several researches [5,19,20,47,70-73] assumed that the different types of tissue (fat, glandular and fibrous tissue) can be modeled as homogeneous, and that their behaviour under compression is approximately isotropic and nearly incompressible [31]. Since soft biological tissue is predominately composed of water - an incompressible fluid - it is considered incompressible [31]. With these assumptions, it is possible to model the behaviour of the tissue using a single elastic or shear modulus. Several authors [5, 19,20, 22, 69, 74] considered that the incompressibility condition imply that the Poisson ration is 0.5, which means if compressive load is applied in axial direction the material expands in other two directions with a ratio of 0.5 with respect to the compression axis [75].

Under these assumptions, Sarvazyan et al. [38] presented results in which they measured the elastic modulus of 168 *ex vivo* specimens of normal, fibroadenomatous and cancerous breast tissues. Reported Young’s modulus values ranged from 2.0 kPa for normal tissue and 15.0 kPa for invasive ductal carcinomas. However, these authors did not describe in detail their measurement system, so it is hard to identify the source of the observed differences.

Sarvazyan et al. [43] reported a study of 150 specimens of normal, fibroadenomatous and cancerous tissues. They showed that fibroadenomas are typically 4 times as stiff as normal tissue, while tumors can be as much as 7 times stiffer.

**Table 1.** Mechanical tests for the breast tissue reported in literature, grouped according with vital state of the subject (*in vivo* / *ex vivo*) and testing techniques.

<b>Mechanical tests</b>	<b>Experimental Condition</b>	<b>Author</b>
<b>Compression/ultrasound Elastography</b>	<i>In vivo</i>	J. Ophir et al. [51]
		Garra et al. [60]
		Hiltawsky et al. [76]
		Thomas et al. [72]
		Sinkus et al. [40-42, 77]
		Plewes et al. [78]
		McKnight et al. [44]
<b>Magnetic resonance elastography</b>	<i>In vivo</i>	Van Houten et al. [46]
		Manduca et al. [54]
		Kruse et al. [47]
		Lorenzen et al. [79]
		Siegmann et al. [80]
		Lawrence et al. [81]
		Xydeas et al. [82]
<b>Optical coherence tomographic elastography</b>	<i>In vivo</i>	Cheng et al. [83,84]
		Srivastava et al. [45]
		Krouskop et al. [19]
<b>Uniaxial compression and punch indentation</b>	<i>Ex vivo</i>	Sarvazyan et al. [38,43]
		Wellman et al. [20]
		Samani et al. [10, 14, 22, 67]
		Umemoto et al. [34]
		Matsumura et al. [33]

Krouskop et al. [19] measured the elastic moduli of 142 *ex vivo* samples of normal and pathological breast tissues. The study concluded that the Young's moduli of the breast tissues is highly dependent on the level of tissue pre-load compression used in the measurement, in other words, the moduli increased significantly with additional

compression. For example, at 5% pre-load compression strain he found that the ratio of the elastic modulus of cancerous tissue to that of fat was 5:1, while at 20% pre-load compression strain the ratio grew to 25:1. The same authors observed that the modulus of adipose breast tissue is relatively constant over the range of loadings studied. For the ductal carcinoma the modulus is low at low strain; it is indistinguishable from fat at the low strain range but at the high strain range, the modulus is larger than any of the normal tissues. The invasive ductal carcinomas are very stiff and the modulus of this tissue is higher than the other tissues at both strain ranges tested. In conclusion, the modulus dependency on pre-load compression confirms that the nonlinear elastic behaviour is often observed in biological tissues [19,20]. Krouskop et al. [19] found that cancerous tissue is not only much stiffer than adipose and normal glandular tissue, but displays a higher non-linear increase in stiffness. Recent studies by Barr and Zhang [85] contradict the results of Krouskop et al. [19]. This evidence demonstrated that different levels of pre-load compression applied to adipose tissue lead to different mechanical behaviour. The results obtained by Krouskop et al. [19] most likely occurred because the samples were not confined to a limited volume, as seen in normal breast tissue. Wellman et al. [20] studied the stiffness of 26 samples of adipose tissue, 7 of fibroglandular tissue, 1 of ductal carcinoma in situ (DCIS) and 25 of invasive ductal carcinoma (DCI). The authors reported a wide scatter in the mechanical behaviour of the various tissue samples tested. For example, at 1% pre-load compression strain, it was found that the stiffness ratio of cancerous tissue to that of the other normal tissues was 10:1, while at 15% pre-load compression strain the ratio increased to approximately 50:1. Comparing the stiffness of the cancerous tissue with adipose and normal glandular tissue the study concludes that cancerous tissue is 10 times as stiff as normal fat at 1% strain, and more than 70 times as

stiff at 15% strain, while the stiffness in the cancerous tissue to glandular is more than 2.5 times as stiff at 1% strain and approximately 5 times as stiff at 15% strain.

Samani et al. [22] developed two different methods to measure tissue elasticity. The authors tested 169 *ex vivo* breast tissue samples, including fat, fibroglandular tissue as well as benign and malignant breast tumor types. They reported that fat and fibroglandular tissues exhibit identical mechanical properties, with a Young's modulus of 3.25 kPa under small strains. Tumor tissues data obtained by Samani et al. [22] show a substantially higher Young's modulus than fibroglandular tissue, compared to the data of Sarvazyan et al. [43]. Moreover, the authors observed a general increase in the elastic modulus with more invasive cancers, when compared with other type of tumours. Thus, for high-grade invasive ductal carcinomas were the stiffest tumours exhibiting a Young's modulus approximately 13 fold larger than either fat or fibroglandular tissue, with other tumours types demonstrating a 3–6-fold increase in tissue stiffness. In Table 2, it is noted that the values the standard deviation is high in some cases, e.g. high-grade IDC (12.47). This can be attributed to a number of factors including having a small statistical sample (for example in high-grade IDC has only 9), systematic errors associated with the measurement techniques and used FE models, tissue heterogeneity and finally to the variability of tissue stiffness during menstrual phase and for different age groups. Although there is a similarity with the results of Sarvazyan et al. [43], there is no correlation with data from Baki [74] and Krouskop et al. [19]. In general, the authors obtained smaller Young's modulus values compared to the values obtained by Krouskop et al. [19], which makes clear the Young's modulus variation observed between the studies. These disagreements may arise due to the method of pre-load compression chosen and preparation of samples by these two studies. Examples of these differences

can be: using substantially larger compression forces for preloading; and ignoring tissue specimen heterogeneity.

Matsumura et al. [33], measured the elastic moduli, with different pre strain, of 60 *ex vivo* samples of normal and 27 pathological breast tissues. The authors verified non-linearity in tissue elasticity and difference of Young's modulus in all tissues depending on compression level. For example, DCIS revealed larger stiffness than normal fat or gland under a slight stress, but the relation between them changed when stress increased. The IDC and mucinous carcinoma exhibited significantly larger Young's moduli than normal tissues (fat or gland) and the DCIS. The authors also verified that the elasticity of IDC varies over a wide range of compression.

More recently Umemoto et al. [34] measured the elastic moduli of the 87 surgical tissues, including 33 lesions and normal tissues (fat: 29 locations, mammary gland: 24 locations). As seen in Table 2, the Young's moduli of breast tissues differed under conditions of light stress (<1 kPa), and, in ascending order of their elasticity, the tissues were fat, normal gland and ductal carcinoma in situ (DCIS) and invasive ductal carcinoma (IDC). The rates of increase in elasticity of normal breast tissues with respect to a stress axis from 0.0 to 1.2 kPa are significantly larger than those of malignant tissues, especially in IDC; the Young's moduli of normal breast tissues increase to the point where they come close to or exceed those of malignant tissues. The authors also verified significant difference in non-linearity between DCIS and IDC, especially in the stress-elastic modulus relationships under the minimal stress conditions. The authors concluded that the Young's modulus relationship between normal breast tissues and malignant tumors dramatically changes as stress is applied because of the non-linear properties (see Figure 7). Table 2 summarizes the results of mechanical properties of the *ex vivo* breast tissue obtained by different authors.



As can be seen in Sec. 3.1, breast tissue's elastic modulus can be measured *in vivo* using magnetic resonance elastography. Lawrence et al. [81], were among the first to study *in vivo* breast MRE. A total of nine healthy female volunteers have been evaluated, and demonstrated that MRE is feasible and can adequately illuminate the breast tissues with shear waves and can characterize biomechanical properties of glandular tissue ( $2.45 \pm 0.2$  kPa) and fat tissue ( $0.43 \pm 0.07$  kPa).

Kruse et al. [47] presented preliminary results from an *in vivo* MRE exam of a patient with a biopsy-proven carcinoma. Showed that a localized area roughly two to three times stiffer than the surrounding fibrous tissues corresponds to a biopsy proven cancerous tumor.

Similarly, Sinkus et al. [40] reported that carcinoma exhibits an anisotropic elasticity distribution while the surrounding benign tissue appears isotropic. The results obtained *in vivo* revealed increases in stiffness of roughly two to three times between background tissue and lesions. Van Houten et al. [46] separated the properties of the adipose and fibroglandular tissue within the breast by manual segmentation. The authors concluded that the adipose tissue has lower Young's moduli compared to other tissues. Srivastava et al. [45] measured the mechanical properties for a normal, malignant and benign breast tissue. These authors reported that Young's modulus for malignant breast tissue samples are approximately four times higher than that of the normal tissues, while for benign tissue samples it is about two times higher than that of the normal samples. The data reported is, consistent with previous studies, like Krouskop et al. [19], Wellman et al. [20] and Samani et al. [22].

**Table 2.** A Summary of the results from mechanical testing of *ex vivo* breast tissue.

Authors	Young's Modulus (kPa) mean (STD)				
	Pre-strain (compression)	Normal fat tissue	Normal Glandular tissue	Tumor tissue	
			DCIS	IDC	
Krouskop et al. [19] (Loading frequency (Hz) of 0.1 to 4)	5% pre-load compression	18 (7) to 22 (12)	28 (14) to 35 (14)	22 (8) to 26 (5)	106 (32) to 112 (43)
	20% pre-load compression	20 (8) to 24 (6)	48 (15) to 66 (17)	291 (67) to 307 (78)	558(180) to 460(178)
	1% Strain	4.8 (2.5)	17.5 (8.6)	71.2 (0.0)	47.1(19.8)
	15% Strain	17.4 (8.4)	Fibroglandular sample	2162 (0.0)	1366.5(348.2)
Samani et al [10, 68]	5% Compression	3.25 (0.9)	3.24 (0.61)	16.38	L: 10.4 (2.6); M: 19.99 (4.2) H:42.5(12.47); Data is provided for low, medium and high-grade IDC
	Not given	5 (0.0)	50 (0.0)	100 (0.0) to 5000 (0.0) for palpable nodule	
Sarvazyan et al. [43]	Not Given	1.0 (0.5) data is given for a combined fatty and fibroglandular sample		3.5 (0.5)	10.0(1.9)
Matsumura et al. [33]	0-0.2 Stress	0.7 (0.2)	0.8 (0.2)	3.4 (1.3)	11.5 (8.4)
	1.0-1.2 Stress	17.3 (4.8)	15.4 (3.9)	15.6 (2.0)	27.0(9.2)
	0-0.2 Stress	0.69 (0.19)	0.73 (0.18)	5.25 (0.46)	13.82 (9.60)
Umemoto et al. [34]	1.0-1.2 Stress	19.08 (4.99)	16.99 (4.92)	16.15 (4.24)	30.5 (11.46)

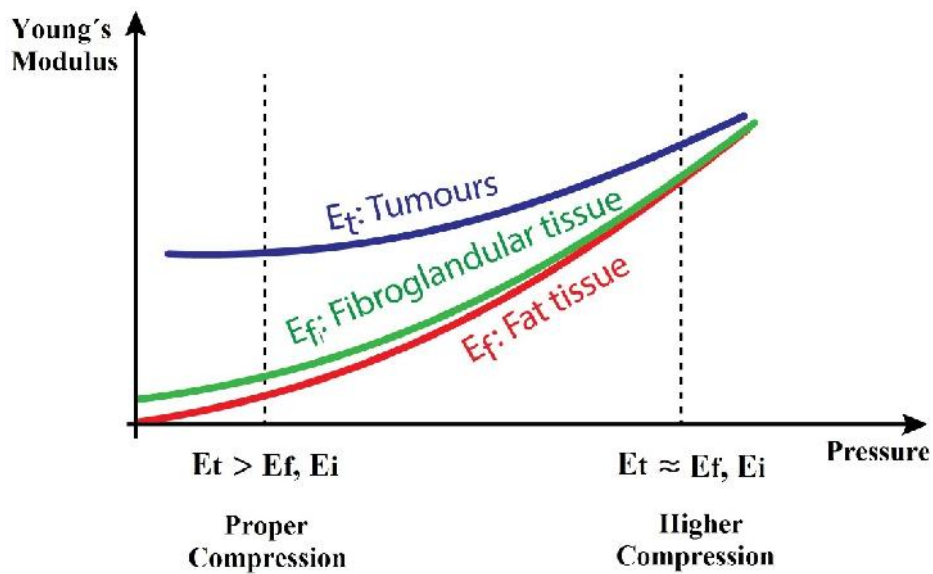
McKnight et al. [44] studied six healthy volunteers and six patients with biopsy-proven palpable breast malignancies, and concluded that the average shear stiffness of the tumors was 33 kPa (range = 18–94 kPa), which was about four times greater than that of adipose tissue (mean = 8 kPa, range = 4–16 kPa) in breast cancer patients. In the healthy volunteers, the mean value for adipose tissue was  $3.3 \pm 1.9$  kPa, which is less than their fibroglandular tissue ( $7.5 \pm 3.6$  kPa).

Xydeas et al. [82] studied viscosity and elasticity of breast tissues in five patients with six malignant lesions, eleven patients with benign lesions, and four patients with no lesions using MRE. The aim of the study was to investigate the potential value of MRE to improve the differentiation between benign and malignant tumors. The mean elasticity parameters were: breast cancer ( $3.1 \pm 0.7$  kPa), fibroadenoma ( $1.4 \pm 0.5$  kPa), fibrocystic changes ( $1.7 \pm 0.8$  kPa) and surrounding tissue ( $1.2 \pm 0.2$  kPa). According to the study, malignant tumors documented higher values of elasticity than benign corresponding to signal intensity and morphologic data. Table 3 summarizes some results from *in vivo* MRE elastography.

Sayed et al. [26] used multi-compression 3D ultrasound elastography and demonstrated the ability of the technique to better diagnose stiff masses inside breast tissue. The results obtained *in vivo* revealed the target mass was approximately 6.3 times stiffer than the background soft tissue. These results were compared with biopsy diagnosis, and showed a good agreement with biopsy outcomes.

It should be noted that normally the stress distribution is not uniform within the body and the tissue elasticity is nonlinear. According with tissue nonlinearity, the Young's modulus tends to increase when the compression is intensified as shown in Figure 7.

A recent study tested four regions of pre-load compression (Region: A 0-10%; B 10-25%; C 25-40%; D >40%) that explain clinical elastographic results [85]. It was concluded that, when the degree of compression is slight, 10% tissue compression approximately, the difference in the Young's modulus between breast tissue and tumor tissue is large and consequently the tumor tissue is clearly identified on a relatively low-strain region. But for high compression levels (about 40%), the stiffness of the breast tissue will increase, and the difference from the tumor tissue will be smaller. It is recommended that all clinical images are obtained approximately at a level of 10% pre-load compression.



**Figure 7.** Behaviour of breast tissue at different levels of pre-load compression. Adapted from [32,34,85].

To counter this effect Cheng et al. [84] developed a preliminary study with a novel non-compressive breast MRE setup. This study was performed with seven healthy female volunteers and one female patient with a biopsy-proven invasive ductal carcinoma. For the seven volunteers the stiffness of tissue ranged from 0.25 to 0.41 (mean = 0.33) kPa

for adipose tissue, and from 0.46 to 0.9 (mean = 0.64) kPa for glandular tissue. For the other patient the stiffness of adipose tissue was  $0.41 \pm 0.10$  kPa and of glandular tissue was  $0.90 \pm 0.18$  kPa. The invasive ductal carcinoma was stiffer,  $1.42 \pm 0.17$  kPa, as shown in table 3. The invasive ductal carcinoma is about 3 times stiffer than the adipose tissue and 1.5 times stiffer than the glandular tissue.

Based on the data collected from the analyzed studies, the following conclusions were achieved:

-The stress–strain curves of the breast tissues, describing the mechanical behaviour of the tissue under different stress levels, follow an exponential behaviour, with malignant masses showing a steeper curve than the benign tissues.

-The moduli of elasticity of the fibrous, glandular and cancerous tissue are significantly higher than the adipose tissue, and are not constant along the studied strain variations. The fat tissue behaviour is closer to linear than all other tissues measured.

-There was [19,20] a dependency of the mechanical properties with the technique used: if an image of the elastic modulus distribution throughout the breast, was obtained at one strain level and then the strain level was doubled, the whole of the tissue compressed would suffer an increase in stiffness. Thus, the Young's modulus of breast tissues is highly dependent on the level of tissue pre-load compression, and the relative stiffness is a good predictor of histological diagnosis.

The results of these studies clearly indicate a wide variation in moduli not only among different types of tissue but also within each tissue type. These differences were most evident in normal fat and fibroglandular tissues.

The research works reviewed, used different techniques for estimating the tissue stiffness distribution within a breast. However there is surprisingly little available

information in the literature on the mechanical properties that would allow conclusions about the histological nature of the tissue directly from the estimated stiffness.

**Table 3.** A summary of results from *in vivo* magnetic resonance elastography for breast tissue.

Authors	Elastic modulus (kPa) mean (STD)		
	Normal fat tissue	Normal glandular tissue	Tumour tissue
Kruse et al. [47] (Frequency of 100Hz)	15-25	30-45	50-75 for carcinoma
Sinkus et al. [40,42] (Frequency of 60Hz)	0.5-1	2-2.5	3.5-4 for carcinoma
Mcknight et al. [44] (Frequency of 100Hz)	3.3	7.5	25
Houten et al. [46]	23.5 (4.03)	26.6 (4.49)	-
Lawrence et al. [81] (Frequency of 50-100Hz)	0.43 (0.07)	2.45 (0.2)	-
Cheng et al. [84] (Without compression)	0.41 (0.10)	0.90 (.018)	1.42(0.17) for ductal carcinoma
Xydeas et al. [82] (Frequency of 65Hz)	1.2 (0.2)	1.2 (0.2)	3.1 (0.7) for breast
Srivastava et al. [45]		4.17 (0.074)	16.45 (1.103) for invasive ductal carcinoma

## 5. Discussion and Conclusions

One of the main motivations for evaluating the mechanical properties of breast tissue is its potential for disease assessment applications. Normally, the tissue tends to stiffen with disease. These modifications result in a restructuring of the normal tissue components, which manifests itself as a change in the elastic modulus of tissue - the

common mechanical property used for evaluation [86]. Thus, the studies performed have been focused on the measurement of breast tissue stiffness through the elasticity moduli. On the other hand, a comprehensive knowledge of the mechanical properties for glandular and adipose tissue is not yet available in the literature, which explains the lack of recent articles in this review. Although all these studies were made to visualize the distribution of stiffness within the breast, there are few studies on the mechanical properties aimed at understanding the histological nature of the tissue directly from the estimated stiffness.

Table 2 and 3 show the mechanical properties range reported by different authors. As can be seen in the tables, there is a significant variability which makes it difficult to use statistical data to model the individual properties of the breast. This variability is highly dependent on several types of pre-load compression. In summary, Young's moduli of normal breast tissue increased dramatically with increasing compression. As for the DCIS (ductal carcinoma in situ) specimens, their elastic moduli becomes close to those of normal breast tissues, as the stress applied increases under higher compression. Moreover, these studies showed a general increase in the elastic modulus associated with more invasive carcinoma. As a consequence, Young's moduli measured for invasive carcinoma specimens exhibited greater variation than those for normal tissues. Variation must have its roots on the complex pathologic structure of the tissue i.e., the heterogeneous mixture of cellular and fibro stromal components. Thus, the elasticity measurements clearly indicate that each tissue in the breast exhibits different non-linear characteristics in stress versus elastic modulus relationships under light compression conditions. Characterization of the mechanical behaviour of the breast requires a combination of experimental techniques, specialized software particularly regarding the compression levels used. This approach is of capital importance to predict deformations accurately using biomechanical simulation models such as FEM models.

By analyzing Table 2, the results Krouskop et al. [19] were clear relatively to: Young's moduli variation between different tissues of the breast, and Young's moduli increase with the initial condition of strain applied (i.e., percentage of pre-load compression). In comparison, the Young's moduli measured for adipose, normal gland, DCIS and IDC in several authors, such as, Samani et al. [22], Matsumura et al. [33] and Umemoto et al. [34] tended to be smaller than those reported by Krouskop et al. [19]. The observed disagreements in this case are attributed to the fact that, in their measurement, Krouskop et al. [19] applied substantial pre-load compression of 5% and 20%, and consequently the authors did not describe their initial stress conditions fully. However, it can be speculated that the differences in Young's moduli were originated from the different stresses applied to the specimen. It was also presumed that the stress used in studies described by Matsumura et al. [33] and Umemoto et al. [34] was lower than the stress used by Krouskop et al. [19].

Similar results to Krouskop et al. [19] have been shown by Wellman et al. [20]. Both studies obtained higher Young's when compared with the other studies. Matsumura et al. [33] and Umemoto et al. [34] used a similar protocol testing (same stress), which found very similar results for the different breast and tumor tissues. They concluded that the results revealed a reduction or inversion in the difference of Young's moduli between normal and tumour tissues with increasing stress.

By comparing the results by Samani et al. [22] with those reported by Sarvazyan et al. [38], it was verified that some of the results were in accordance while others show Young's modulus generally smaller. For example, Samani et al. [22] reported Young's modulus values of approximately 3.25 kPa and 19.99 kPa for normal tissues and IDC, respectively, which are fairly well compared with the 2.0 kPa and 15.0 kPa that obtained by Sarvazyan et al. [38]. However, the results described by the other authors in Table 2



are different when comparing with Sarvazyan et al. [38] and Sarvazyan et al. [43]. So, it is important to refer that Sarvazyan et al. [38] and Sarvazyan et al. [43] did not reported details of their measurement system, thus it is hard to speculate the source of the observed disagreements.

The data reported in Table 3 is, consistent with previous studies, like Krouskop et al. [19], Wellman et al. [20], Samani et al. [22], Matsumura et al. [33] and Umemoto et al. [34]. The results of these studies indicate a wide variation in elastic moduli not only among different types of tissue but also within each tissue type.

Although the studies from Kruse et al. [47] and McKnight et al. [44] used a similar elasticity imaging technique (frequency at 100Hz), the results for the various tissues were different. For example, Kruse et al. [47] reported values of approximately 15 kPa and 50 kPa for fat tissues and tumor, respectively, which are different compared with the 3.0 kPa and 25.0 kPa obtained by McKnight et al. [44]. Xydeas et al. [82], Sinkus et al. [40] and Lawrence et al. [81] shown a similar results for normal tissue (fat and glandular tissues). It is important to note that the variability of the results reported by MRE elastography may be associated with the variability in the test procedure (such as the different shear wave frequencies applied).

Until now researchers have used different approaches to estimate the mechanical properties of soft biological tissue. The differences in stiffness between normal and abnormal breast tissue have been recognized for a long time [51]. To analyze large deformations (ex. pre-strains up to 20%) the *ex vivo* tests are the most suitable. However, *in vivo* data is only collected under small pre-strain conditions and often the pre-strain used is not recorded. Considering this limitation, *in vivo* data is of limited usefulness for modelling large deformations of the breast. Often, the force information is discarded during the test to estimate the mechanical properties of the tissue because it is applied as

an adjunct to existing imaging modalities. Thus, it is necessary to establish a method to measure large deformations of tumor tissue and its nonlinear elastic behaviour with accuracy. Typically these techniques make images of the tissue at two different applied loads and compute a displacement field from them. This displacement field is then used to infer the stiffness of the tissue, from assumptions made about the stress field. Basically the pre-load compression has a considerable effect on the quality and results of elastography. For example, the breast elastography is very susceptible to pre-load compression because the chest wall acts as a hard posterior surface, allowing for substantial pre-load compression when scanning. The effects of pre-load compression are significant in the breast and can easily affect test outcomes (benign versus malignant). A clear example is referred by Matsumura et al. [33], which showed that DCIS cannot be sometimes easily detected (i.e. false negatives) at excessive breast compression in clinical exam on elastography. This limitation highlights the need to quantify the preload for compression magnitude (strain or stress) in various breast and tumor tissues. Umemoto et al. [34] understood this need and measured the compression magnitude through loaded stress on the tissue sample in compression test. Thereby showing quantitatively the relationship between the magnitude of compression and tissue elasticity (Young's modulus) in the target lesion.

Therefore, the importance of nonlinear responses of soft tissue to compressive loads in clinical breast examination highlights the need for launching a comprehensive study on the hyperelastic characterization of *ex vivo* and *in vivo* soft tissues, to enhance clinical approaches including the detection of breast cancer.

Despite all available results from compression experiments, until now there is no data available about the material properties of the breast under uniaxial or biaxial tensile loading conditions (because of its fragile constitution). Recently, Sommer et al. [87]

performed tests in human abdominal adipose tissue by biaxial tensile and triaxial shear tests. This experimental attempt to understand the anisotropy in the properties of the adipose tissue produced promising results. Adipose tissue was characterized as a nonlinear, anisotropic and viscoelastic soft biological material. These tests are a new approach to study the breast adipose tissue. None of the experiments reported takes into account the pre-strain caused by gravity, hydration and tissue fibers. It is a limitation of *ex vivo* tests, contrary to *in vivo* tests where the measurements are performed in the natural state, i.e., the blood supply and interstitial fluids are present. When compared to the *ex vivo* tests, *in vivo* tests are used as a diagnostic tool that help assessing the changes in the mechanical properties of the tissues under compression in a simple and non-invasive way. As they can separate tumors from adjacent healthy tissues and distinguish if the tumor is malignant or benign according to their stiffness, these tests are valuable for characterizing the mechanical properties of the different breast tissues due to their high degree of specificity and sensitivity [40,44-47,50,53,54].

The high Young's modulus variability reported by several studies is directly correlated with the use of different mechanical tests, experimental conditions (*in vivo* or *ex vivo*), different pre-load compression, tissue heterogeneity and systematic errors associated with the measurement techniques. Some of these errors may be introduced due to the blood supply and interstitial fluids absence during the tests, although these are efforts to keep the samples hydrated. The other source of error could be associated to the location where the tissue samples were removed. In addition, it is expected to see different measures of firmness in the same tissue.

Regarding the different pre-load compressions, the Young's modulus between breast tissue and tumor tissue is large when the degree of compression is slight. However, when the compression is too strong, the stiffness of the breast tissue will increase, and the

difference from the tumor tissue will be smaller [32,85]. This compression effects partly explains the inconsistencies in the reported stiffness values of breast tissues from different methods in the literature [32,44,67,85,88,89].

Another feature is that the mechanical properties of breast tissues differ between individuals and over time due to the variability in breast morphology, hormonal status, age, and physiological condition [45]. Despite of in the majority of the studies the authors refer a range of age of the samples, there was a lack of information regarding some important factors such as pre or post menopause, menstrual cycle and so on, which could have influence in the experimental results. For example, Lorenzen et al. [90] found that fibroglandular tissue roughly doubled in stiffness during the menstrual cycle. Therefore, future studies should include these factors in order to understand the variations of breast tissue in the several stages of the women's life.

Glandular, adipose and fibrous tissues are the main tissues of the breast that have been studied to estimate their mechanical properties. There are no studies to date of the suspensory cooper's ligaments (they provide support and hold the breasts in place). So, it is necessary to develop techniques to test the suspensory cooper's ligaments. This effort can contribute to establish a methodology based on the finite element method to simulate a realistic 3D model of the breast. Thus, all knowledge on the mechanical properties of the breast tissue is important for studying the effect of plastic and oncoplastic surgery techniques in breast reconstruction, as well as for design of cosmetic breast implants.

In conclusion, it was possible to verify that the difference in mechanical behaviour between tissues, provides useful information with potential impact on clinical diagnosis. The development in experimental protocols led to an improvement of clinical diagnosis. In order words, better experimental protocols led to a refinement of the compression

magnitude to apply in clinical examination. This procedure is fundamental to avoid false-negatives.

The mechanical tests of soft biological tissues require a test system suitable to the specificity of these materials. The determination of mechanical properties can be used to: correlate the mechanical behaviour with pathology (e.g. cancer) or with population characteristics (age, menopause, lactation, etc...) and to simulate the biomechanics of the breast tissue. Further research is therefore needed to: (1) integrate the etiological factors influencing the biomechanical proprieties of breast tissues, such as age, body mass index or hormonal status (menopause); (2) characterize all tissues, including the suspensory cooper's ligaments; (3) build experimental set-ups that includes *in vivo* and *ex vivo* testing in order to validate the results; (4) standardizing the experimental protocol, in order to analyse samples from the same breast location; (5) controlling the amount of pre-load compression (for instance, test two levels of pre strain, a proper and a higher level used in clinical breast examination). Because the pre-load compression is a substantial factor in obtaining accurate results.

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## Conflict of interest statement

None declared.

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# Chapter V

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Original Articles



## Article 2

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### **Mechanical Performance of Poly Implant Prosthesis (PIP) Breast Implants a Comparative Study**

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## **Abstract**

**Background:** There is societal concern regarding potential health problems associated with breast implants. Much of this distrust climate was a reaction to the Poly Implant Prosthesis (PIP) scandal. Studying the mechanisms of implant rupture is an important step for their improvement. The mechanical behaviour of breast implant shells were studied on explanted and virgin implants. Implants from both PIP and another brand (brand X), currently in the market, were considered.

**Methods:** To study the mechanical behaviour of the shell, a total of 940 samples, from 11 explants and 5 control implants were analysed. The experimental protocol follows the ISO standards for shell integrity and determination of tensile stress-strain properties. Pearson correlation analyses and the multi-factor ANOVA statistical tests were performed using mechanical test data.

**Results:** Both PIP control and explants had significant variations of stress ( $P=0.0001$ ) and shell thickness ( $P=0.000$ ) throughout the implant. The stress was found directly related to shell thickness. Shell thickness varied significantly for PIP implants, exceeding the manufacturer's specifications. Regarding the other brand, thickness variation was within manufacturer's specifications.

**Conclusions:** The heterogeneous nature of PIP implants was confirmed. The implant shell thickness should be considered as a relevant parameter during the manufacturing process, for quality control purposes. These results may contribute to dispel mistrust and doubt surrounding breast implants, among the medical community and patients.

**Keywords:** Breast implant, Poly Implant Prosthesis (PIP), Mechanical behaviour;  
Implant thickness.

## 1. Introduction

Overall safety, durability, long-term stability, and aesthetic value of the breast implants has been a matter of interest and concern for a long time [1-6]. However, events surrounding PIP breast implants in 2010, have renewed the debate among the medical community and PIP breast implant users about the safety of breast silicone implants, and its effect in patients' health.

In Portugal, the health regulator entity (INFARMED) banned the use of PIP implants in 2010. The removal rate of this type of implants (revision surgery) started to increase from 2011 onwards, due to a wider media coverage, increasing 60% in 2012 [7].

Several studies on biodurability of breast implant shells, concluded that it is influenced by different factors, such as damage during surgery, material ageing following implantation, open and closed capsulotomy, trauma injuries (shocks from car accidents), manufacturing defects, fatigue, shell wrinkling, needle biopsy or hematoma aspiration, and patch detachment [8]. In particular, for PIP breast implants, problems included: ageing of implant materials following implantation (degradation of material), variability in the design and manufacturing process, the quality of surgical procedure for implantation, and the quality of raw materials [9-11]. Following these studies, it was suggested that PIP implants have not been manufactured according to medical device standards [10-12]. These led regulatory authorities [9] and public opinion to associate the brand with a higher risk of several types of ruptures.

The available literature on PIP implant rupture is supported by limited experimental data. In particular, the mechanical characterization, when available, is based on a small

number of tests per implant, typically between 1 and 20 [4, 11, 13-15]. To fill this gaps in the available mechanical characterization, a protocol to characterize all regions of the implant (anterior, equatorial and posterior) is proposed. The experimental protocol included a detailed analysis of the different implant regions. Moreover, PIP implants were compared with another commercially available brand.

There is a wide variation of tensile behaviours of implant shells over different brands. Inter-brand variations may be linked to different curing processes during manufacture [11, 16, 17]. Even for different batches of the same brand there are noticeable variations of mechanical behaviour over the same deformation range, as found by this investigation and by other researchers [8]. To compare implants from different brands, the experimental protocol adopted for this work consisted on tensile analysis over fixed strain points of the test sample – 33%, 66%, 133% and 266% length increase. These strain points were common to all tested implants, thus providing a robust comparison basis.

## **2. Material and Methods**

### **2.1. Breast Implants Collection**

Eleven explanted PIP breast implants were collected from February 2012 to July 2013. The implants were implanted between 2004 and 2010. Three PIP virgin implants were obtained from the Portuguese health regulator entity, and were used as controls. Two implants from another brand (brand X) commonly used in Portugal, were also used as controls.

### **2.2. Mechanical Testing Protocol**

The mechanical testing protocol was designed according to the international standard for mammary implants (ISO 14607:2007), and the standard for rubber,



vulcanized or thermoplastic for determination of tensile stress-strain properties (ISO 37:2005). The samples preparation was performed using the protocol developed by Schubert et al. [18].

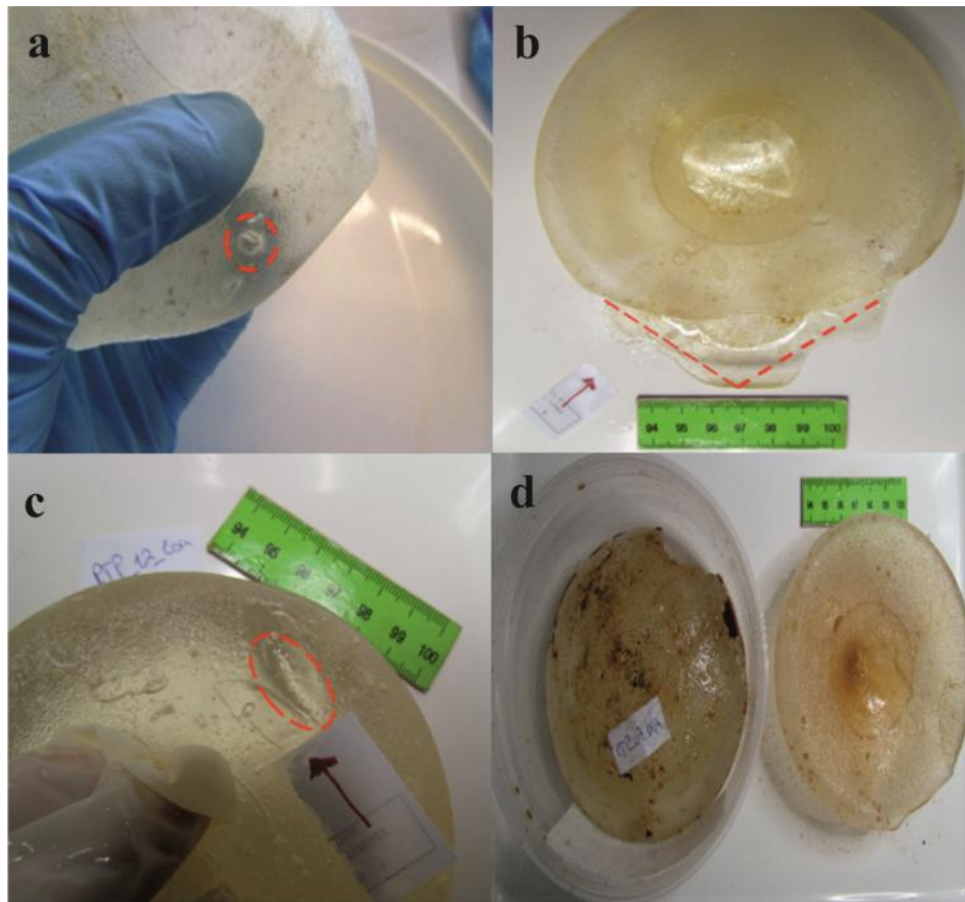
A total of sixteen (16) implants were tested. The samples removed from each group were the following:

- 11 PIP explanted implants – 604 samples;
- 3 PIP control implants (virgin) – 216 samples;
- 2 Brand X control implants (virgin) – 120 samples;

Each sample was analysed using the same experimental techniques as explained in sections 2.2.1 and 2.2.2.

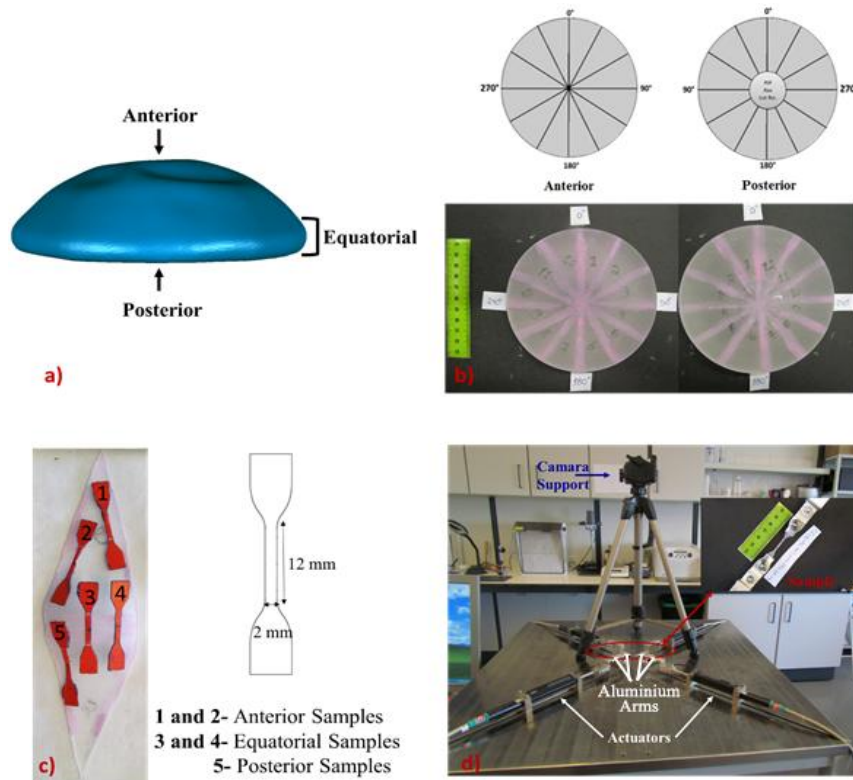
### **2.2.1. Samples Preparation**

The explants were disinfected manually using alcohol wipes. Following disinfection, the explants were visually examined and the presence and appearance of any shell rupture (hole, split or V-shaped), discoloration, opacity or other unusual features were recorded for future comparison. The shell integrity was classified in relation to the shell damage and gel condition, according to the Department of Health Therapeutic Goods Administration (TGA) [19] criteria. Figure 1 illustrates the types of damage observed by explants.



**Figure 1.** Classification of the Shell damage. a) Hole shaped damage; b) V- shaped split; c) Split and d) Gross damage, in this case the shell and the cohesive gel were totally separated.

A careful analysis of the implants, the rupture location and aspect was carried out. Each shell was subdivided into 12 segments. This segmentation enables a wide mapping of the mechanical properties of the implant shell material. Each segment provided, between five and nine specimens, depending on the implant size and shape. Following this procedure, each implant provided a minimum of sixty specimens. Figure 2 a), b) and c) illustrates schematically the preparation of the specimens.

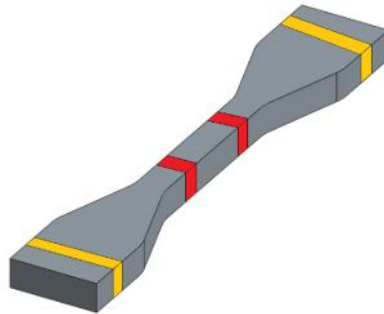


**Figure 2.** Schematics of the experimental procedure. a) Regions of the implant; b) Implant segmentation into 12 segments. To ensure traceability of each segment over the implant, each segment was labelled with a number (1 to 12); c) example of sample preparation for tensile tests; d) Tensile testing equipment.

Dog-bone shaped samples type 4 (shaft length 12 mm, width 2 mm) as illustrated in Figure 2 c) were cut from the original shell, using a cutting dye to obtain a standard geometry. These geometrically controlled samples facilitate calculations and increase reproducibility.

Prior to the tensile test, the initial dimensions of all samples were measured, using a digital calliper (Powerflex Profi; accuracy 0-100mm  $\pm 0.02$  / 100-150mm  $\pm 0.03$ ). The thickness of each sample was measured four times in two different locations, corresponding to the central area and the two fillet areas, respectively (see Figure 3). Maximum, minimum, and absolute minimum thickness values were recorded for each

sample. Only the thickness of the failure region was considered to calculate the tensile stress.



**Figure 3.** Scheme of the sample thickness measurement. Yellow-edges of the samples; red-control area.

### 2.2.2. Testing Procedure

The mechanical properties were obtained from uniaxial tensile data. The equipment used in this study is a prototype, shown in Figure 2 d). This equipment, has four perpendicular aluminium alloy arms, connected to four actuators and two load cells (with 50N capacity).

Before the tensile test, the samples were subjected to a 0,25N preload. The preload guaranteed a pre-testing controlled initial geometry and loading conditions, contributing to the reproducibility of the experimental procedure. All samples were tested using a displacement rate of 20mm/min in one direction. The mechanical behaviour of shell material was evaluated by comparing the stress ( ) at different strain levels (33%, 66%, 133% and 266%).

All tests were recorded in video by a camera positioned over the testing equipment, depicted in Figure 2 d). The video was used to validate the test, since anomalous occurrences such as slippage or significant misalignment could be easily detected.

## **2.3 Statistical Analysis**

The data was analysed using SPSS version 20 (SPSS, Inc, Chicago, Illinois). One-Way ANOVA was used to compare the different groups considered. The multi-factor ANOVA was used to analyse the stress variance (MPa) over different factors. Different implants and different regions were considered. This analysis helped to determine which factors had a statistically significant effect on the stress at different levels of strain. Pearson correlation [20] was used to study the influence of shell thickness variation on the mechanical properties. The same statistical test was used to evaluate the influence of the duration of implantation on the stress. Average values of the measured or calculated data were expressed as mean (M)  $\pm$  standard deviation (SD). Generally, a statistical significance was defined as  $P < 0.050$ .

## **3. Results**

Regarding the explanted implants, the most frequent reason for an implant removal was the patient's fear of future complications and possible implant rupture. The 11 explanted implants analysed were ruptured in the shell. All had textured shell, round shape and volumes ranging between 210 and 310 cc (see Table 1). The duration of implantation varied from 36 to 95 months, with average of  $57.36 \pm 19.96$  months. The rupture of these implants happened in a period lower than 10 years (120 months) of implantation and were removed in 2012. Five virgin implants, being three PIP and two from brand X, with round shape, textured shell surface and volumes ranged between 170 and 415 cc were used as controls. All control implants belong to different lots.

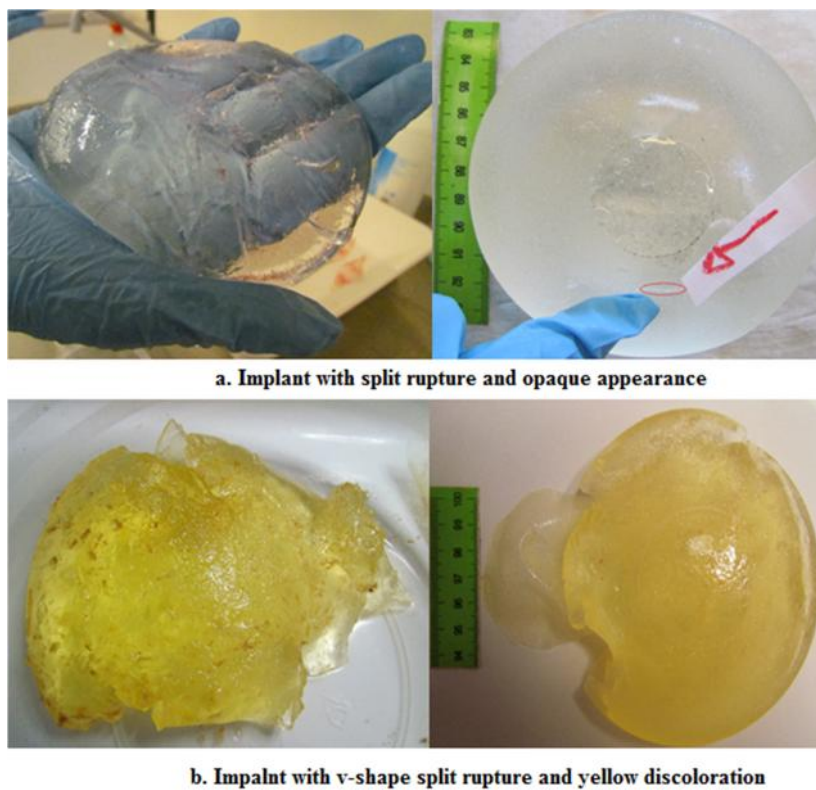
**Table 1.** Clinical characteristics and implant rupture status

<b>Explanted Implants</b>	<b>Volume (cc)</b>	<b>Year of implantation</b>	<b>Duration of Implantation (months)</b>	<b>Type of Rupture</b>	<b>Rupture Location</b>	<b>Rupture Size</b>	<b>Colour</b>
Pip01	290	2008	46	V-split	Anterior and equatorial	Large	Yellow
Pip02	245	2007	52	Hole	Posterior	Small	Clear
Pip03	210	2007	61	Hole	Equatorial	Small	Clear
Pip04	310	2008	39	V-Split	Equatorial and Posterior	Large	Yellow
Pip05	290	2007	56	V-Split	Equatorial and Posterior	Large	Yellow
Pip06	250	2008	40	Split	Posterior	Small	Clear
Pip07	250	2008	50	Split	Posterior	Small	Clear
Pip08	230	2007	64	Hole	Posterior	Small	Clear
Pip09	245	2009	36	Gross Damage	Anterior, equatorial and Posterior	Large	Yellow
Pip10	260	2007	95	Gross Damage	Anterior, equatorial and Posterior	Large	Yellow
Pip11	225	2005	92	V-Split	Equatorial and Posterior	Large	Yellow

### **3.1. Appearance of Explanted Implants**

According to TGA rupture classification [19], the ruptures observed were v-shaped split (n=4), hole (n=3), and split (n=2) and gross damage (n=2). A visual inspection of

PIP ruptured implants showed a significant yellowing in shell and gel for implants with v-shaped split and gross damage. The yellowish appearance may be related with more abrupt ruptures (large ruptures) followed by gel contact with the tissue (see Figure 4). Some ruptured implants showed cohesive gel leakage. Others had a liquefied, non-cohesive gel, leaking from the shell easily. Results indicate that the posterior region (n=9) dominates the location of the rupture, when compared with other regions; equatorial (n=5) and anterior (n=3).



**Figure 4.** This figure is a direct comparison between implants with a large and small rupture. **a** Example of the aspect of the shell and gel in an explanted implant with split rupture (posterior location and implanted for 40 months); **b** V-shape split that shows a yellowish coloration and calcifications in gel and shell (anterior and equatorial location and implanted for 46 months).

## **3.2. Mechanical Testing**

A total of 940 individual samples were tested. The explanted PIP implants provided a total of 604 samples, from which 255 samples were in the anterior region, 230 were in the equatorial region and 119 in the posterior region. The number of samples removed from the posterior region was lower since it was the most affected by ruptures.

From PIP control implants 216 samples were prepared; 88 anterior samples, 78 equatorial samples and 50 posterior samples. 120 samples were prepared from brand X implants; 48 anterior samples, 24 equatorial samples and 48 from the posterior region.

The study aimed to analyse the stress difference among regions, depending on different levels of strain, and identify the weaker regions in each implant.

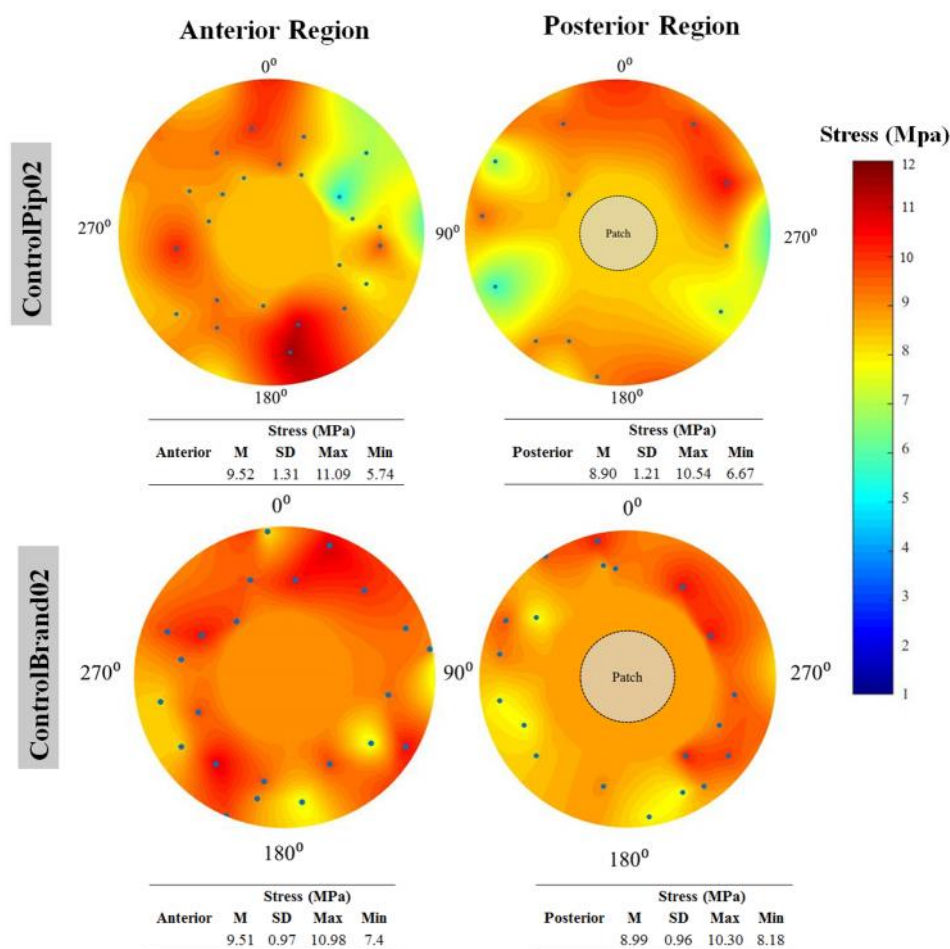
### **3.2.1 Breast Implants Shell Strength – Global Overview**

To obtain an implant-wide stress variation mapping, a custom contour plot was developed and coded in MatLab. Regarding the control implants, Figure 5 represents the stress along the shell at a 266% strain in three regions of both PIP control and brand X implants. Sample's location and stress level (MPa) are shown by blue dots and colour variation, respectively. It can be observed that the brand X implant, shows a uniform stress variation among the regions (anterior and posterior), in contrast with the PIP control implant. In fact, the brand X shows a 9.51MPa ( $\pm 0.97$ ) stress average for the anterior region and 8.99MPa ( $\pm 0.96$ ) for the posterior region, compared to 9.52MPa ( $\pm 1.31$ ) and 8.9MPa ( $\pm 1.21$ ) respectively for PIP implants, which has a higher standard deviation than brand X. Figure 6 gives a statistical overview of the control groups results variation for

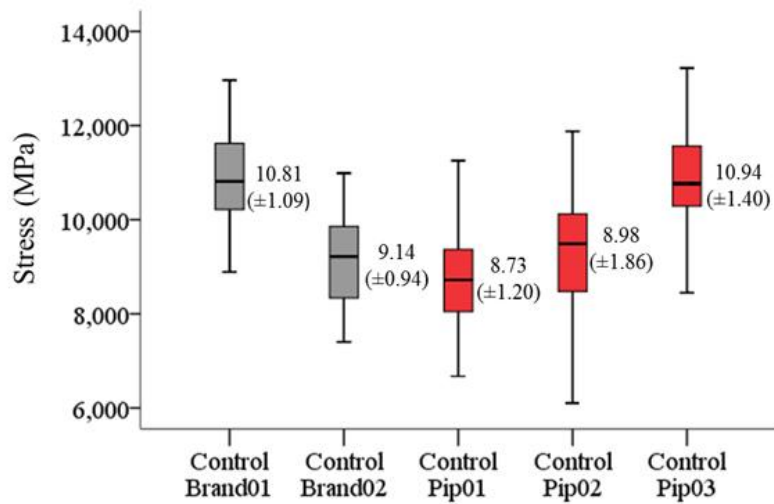


both manufactures controls. As can be observed PIP controls show higher standard deviations than brand X controls.

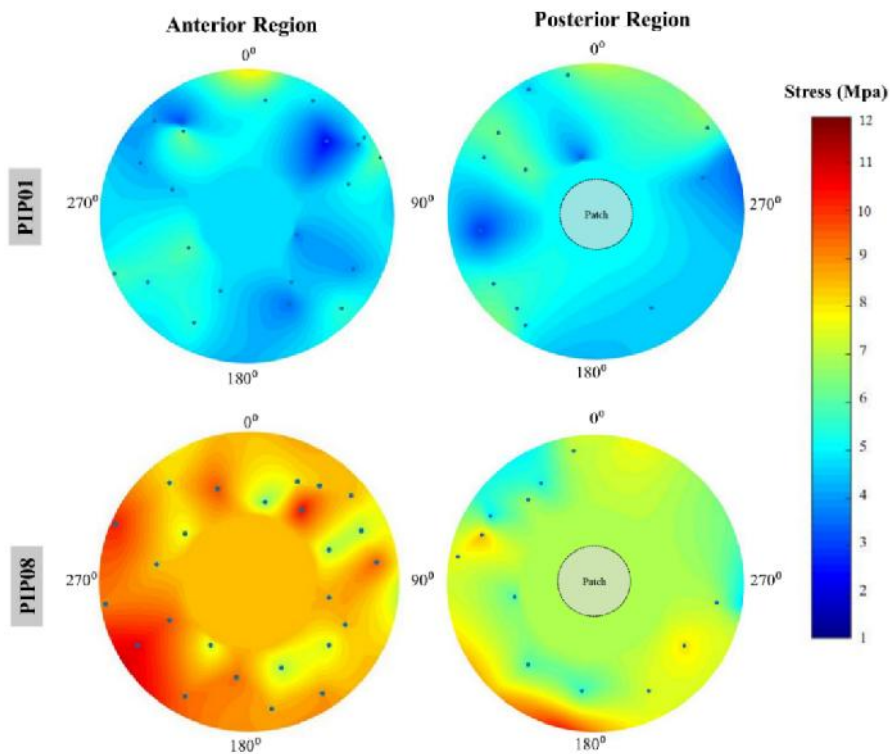
In Figure 7 the variation within the explants group can be easily observed. For instance, PIP01 has a lower mechanical resistance (stress with range 4.47 for posterior region to 5.87 MPa for anterior region) compared to PIP08 (range 6.72 to 5.93 MPa). These two implants show that the contour plot helps to visualize the mechanical behaviour of each implant, using the stress values along the shell.



**Figure 5.** Material behaviour of the control implants (stress at 266% of strain). Average values of measures are expressed as mean (M) ± standard deviation (SD). A total of 24 samples in anterior region and 12 samples in posterior region are represented in contour plot for ControlPip02. For ControlBrand X02, 23 samples are shown in anterior and 20 in posterior regions.



**Figure 6.** The box plot shows the stress (MPa) at 266% of strain between Controls (PIP and brand X). Values are presented as median (horizontal line within box), 25-75th percentile (box) and T-bars (range to the minimum or maximum values).

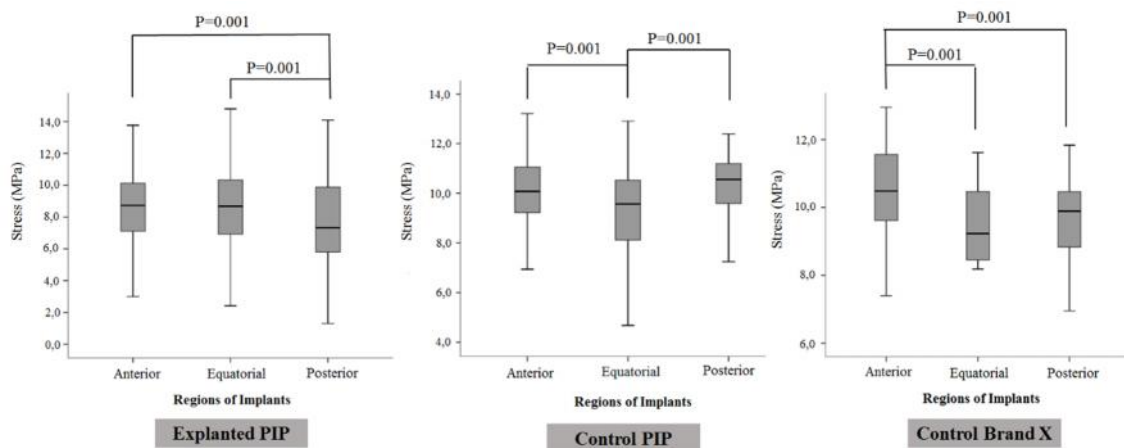


**Figure 7.** Stress (MPa) of two explanted implants with different characteristics - type of rupture, implant colour and duration of implantation. PIP01 was implanted for 46 months, and had a V-shaped rupture and yellowish appearance (56 samples). PIP08 was implanted for 64 months, and had a hole rupture and clear appearance (57 samples).

To verify if there was a significant variation among regions and implants, a multi-factor ANOVA analysis was conducted over all available data (940 mechanical tests).

Table 2 presents the regions showing significantly different stress results. Regarding the PIP explants, the regions with higher damage, corresponded to those with lower stresses (see Table 1). The result showed statistical significance ( $P < 0.050$ ), as it can be seen in Figure 8.

Control groups from both brands showed different results. For the PIP control group, the equatorial region showed statistically significant ( $P < 0.050$ ) differences from the anterior and posterior regions. For the brand X control group, the anterior region showed statistically significant ( $P < 0.050$ ) differences from the equatorial and posterior regions.



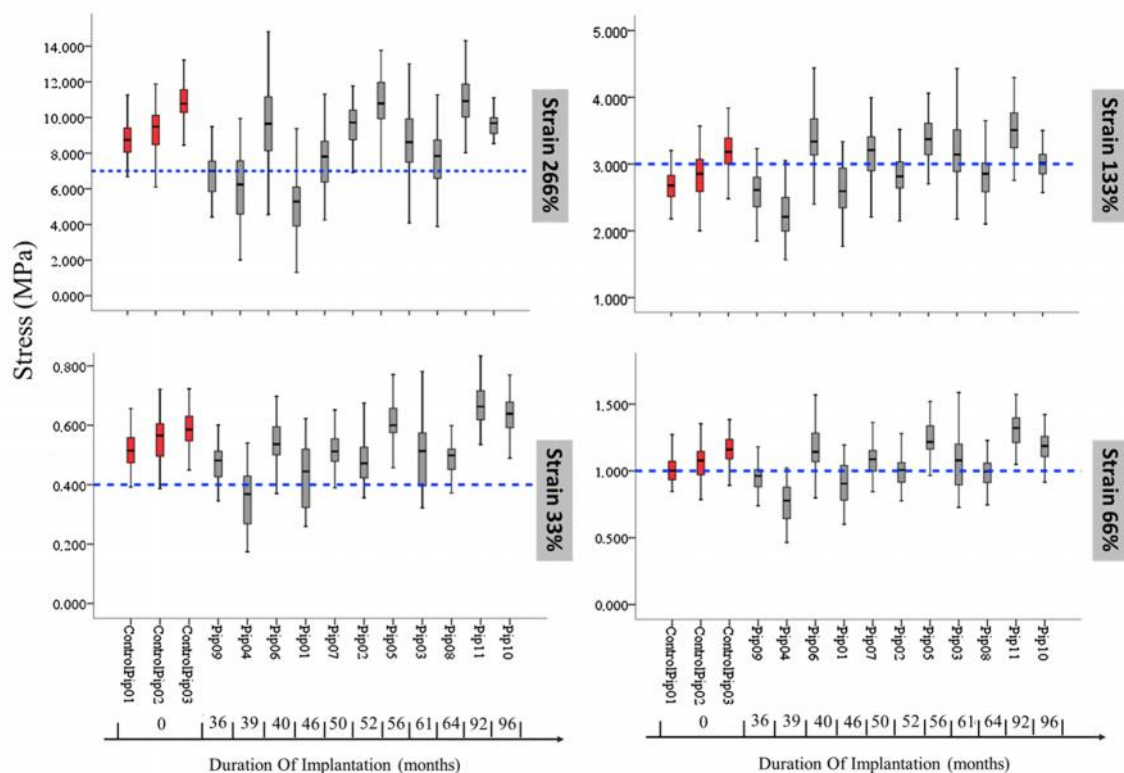
**Figure 8.** The box plot compare the stress (MPa) between three regions of implant (Anterior, equatorial, and posterior) for three groups of implants.

**Table 2.** Multi-factor ANOVA analysis results. Homogeneous Groups regarding regions (stress values at 266% strain) for all implants. The table is organized into subgroups, and within each column, the levels contain a group of means within which there are no statistically significant differences.

Mean vs standard deviation									
Explants Pip			Control Pip			Control other Brand X			
Region	n	Subgroups	Region	n	Subgroups	Region	n	Subgroups	
Posterior	119	7.67(±2.65)	Equatorial	78	9.37(±2.30)	Equatorial	24	9.44(±1.08)	
Equatorial	230	8.57(±2.50)	Anterior	88		Posterior	48	9.60(±1.25)	
Anterior	255	8.58(±2.24)	Posterior	50	10.35(±1.24)	Anterior	48	10.50(±1.44)	
	<i>P</i>	1		<i>P</i>	1		<i>P</i>	0.887	1

Explanted PIP implants showed statistically significant differences ( $P=0.000$ ) in relation to PIP control. Table 3 shows the results of the multiple comparison procedure to determine which PIP implants have a statistically significant effect on stress. For instance, the shell of the controlPip03 implant had a similar stress behaviour to Pip05 and Pip11; this means that there are no statistically significant differences between them ( $P=0.999$ ).

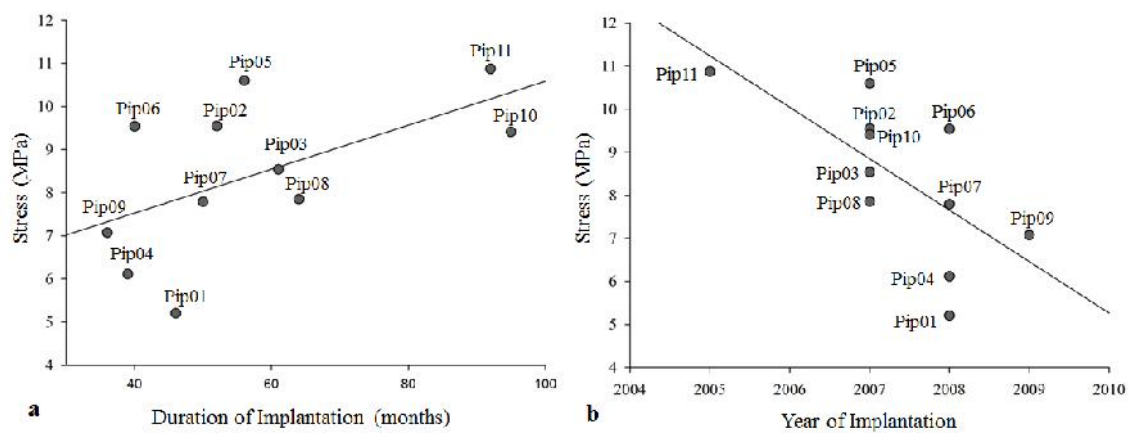
Figure 9 shows the variation of stress at different levels of strain, among the implants. The stress level differences among implants remains coherent throughout the analysed strain range, 0 to 266%. The dispersion (difference between minimum and maximum stress) increases with strain, as expected with simple tension experimental data.



**Figure 9.** The box plot compares the stress (MPa) at different levels of strain between PIP implants (explanted and control). Values are presented as median (horizontal line within box), 25-75th percentile (box) and T-bars (range to the minimum or maximum values).

For each explanted implant, a positive correlation was found between the duration of the implantation and stress (at 266% strain).

The influence of the implantation year on the shell material was analysed. Stress was negatively correlated with year of implantation ( $r = -0.681$ ,  $n = 11$ ,  $P = 0.0208$ ), as shown in Figure 10.



**Figure 10.** a) Correlation between stress (MPa) at 266% of strain and duration of implantation of ruptured PIP implants ( $r = 0.56$ ;  $n = 11$ ;  $P = 0.0053$ ); b) Correlation between stress and year of implantation ( $r = -0.681$ ;  $n = 11$ ;  $P = 0.0208$ ).

**Table 3.** Multi-factor ANOVA analysis results. Homogeneous Groups regarding stress (at 266% strain) for PIP implants (explanted vs control). The table is organized into subgroups, and within each column, the levels contain a group of means within which there are no statistically significant differences.

Implants	n	Mean vs standard deviation					
		1	2	3	4	5	6
Pip01	58	5.20(±1.70)					
Pip04	59	6.11(±1.83)	6.11(±1.83)				
Pip09	48		7.07(±1.50)	7.07(±1.50)			
Pip07	60			7.79(±1.87)	7.79(±1.87)		
Pip08	57			7.85(±1.70)	7.85(±1.70)		
Pip03	59			8.54(±1.86)	8.54(±1.86)	8.54(±1.86)	
ControlPip01	60			8.73(±1.18)	8.73(±1.18)	8.73(±1.18)	
ControlPip02	60			8.98(±1.80)	8.98(±1.80)	8.98(±1.80)	
Pip10	44			9.41(±1.35)	9.41(±1.35)	9.41(±1.35)	
Pip02	57			9.54(±1.70)	9.54(±1.70)	9.54(±1.70)	
Pip06	60						10.60(±1.91)
Pip05	59						10.87(±1.42)
Pip11	53						10.94(±1.34)
ControlPip03	96						
<i>P</i>		0.163	0.106	0.415	0.123	0.081	0.999

### 3.2.2 Thickness Variation

During sample preparation, it was noticed that shell thickness displayed significant variation within and among the samples, especially regarding PIP implants.

The average shell thickness of the PIP explanted breast implants ranged between 0.62 and 1.07 mm. The maximum and minimum thickness values recorded were 1.30 mm and 0.44 mm, respectively. Variation was also observed in the PIP control implants, with a 0.59mm minimum thickness and 1.15mm maximum. The brand X implants showed smaller thickness variation between the regions, with minimum thickness of 0.55mm and maximum of 0.98mm.

The thickness variation (%) was calculated with equation,  $Th\% = \frac{(m_{Th} - m_{Th})}{m_{Th}} \times 100$ , where  $m_{Th}$  is the mean thickness per implant, while  $m_{Th}$  and  $m_{Th}$  are the minimum and maximum thickness per implant.

The thickness variation for PIP explanted implants reached values between 30.93% and 65.22%. For the PIP control group, thickness variation increases, ranging from 31.71% to 85.26%. Regarding the brand X, thickness variation reached inferior values, between 30.97% and 41.49%.

To study a possible influence of the sample thickness on stress, the stresses at 33%, 66%, 133% and 266% strain were analysed for all tested samples. A noticeable influence of thickness over stress was found only for 266% strain.

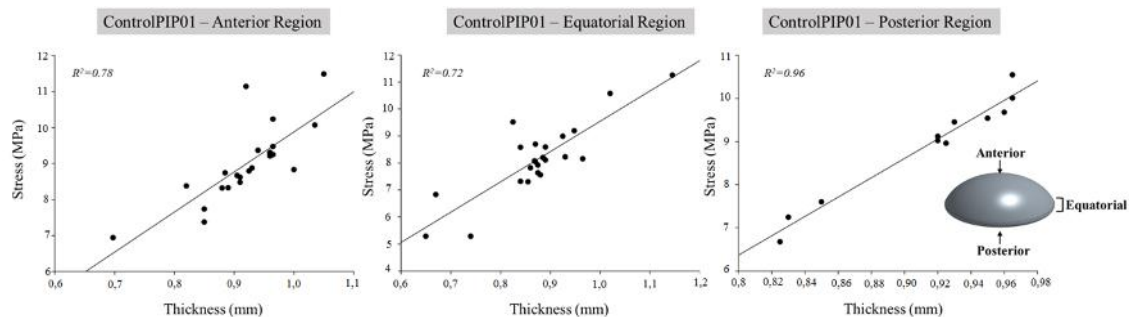
For PIP control implants, a strong correlation was observed between thickness and stress, using the Pearson coefficient (Table 4). Taking as example controlPip01 implant (Figure 11) the results per region were: anterior [r=0.89, n=24,  $P<0.05$ ], equatorial [r=0.84, n=24,  $P<0.05$ ] and posterior [r=0.98, n=12,  $P<0.001$ ].



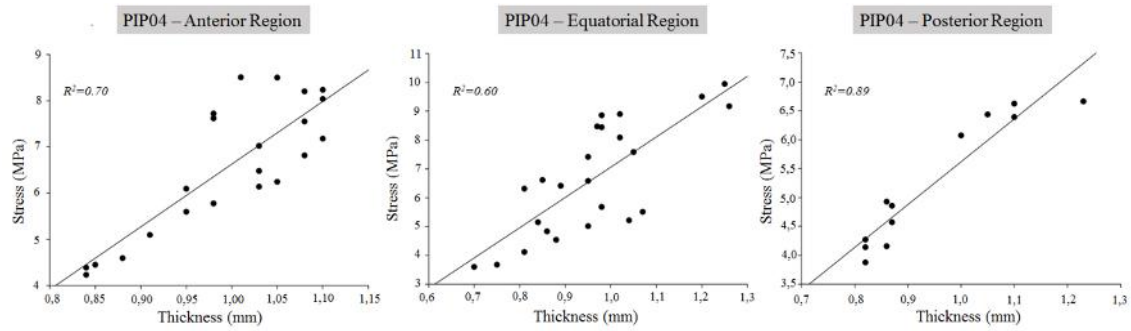
For the PIP explanted implants, a good correlation was also observed (Table 4). Taking as example PIP04 implant (Figure 12) the results per region were: anterior [ $r=0.83$ ,  $n=22$ ,  $P<0.001$ ], equatorial [ $r=0.88$ ,  $n=24$ ,  $P<0.001$ ] and posterior [ $r=0.64$ ,  $n=12$ ,  $P<0.01$ ].

These findings support the hypothesis that a thickness increase, leads to a stress increase. Therefore, a possible reason for rupture may be the non-homogeneity of the shell and its lower thickness.

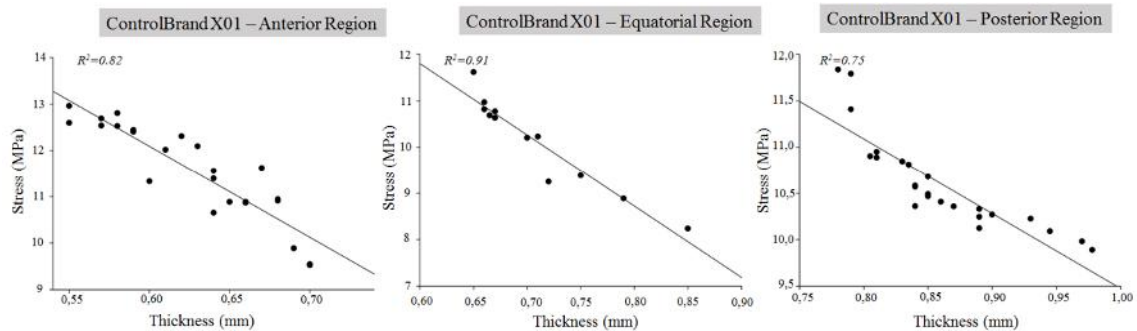
Regarding brand X implants, a strong negative correlation was observed between the regions (Table 4). The results per region were: anterior region [ $r=0.98$ ,  $n=24$ ,  $P<0.001$ ], equatorial [ $r=0.98$ ,  $n=24$ ,  $P<0.001$ ] and posterior region [ $r=0.98$ ,  $n=12$ ,  $P<0.001$ ] (Figure 13).



**Figure 11.** Pearson correlation between stress (MPa) at 266% of strain and thickness (mm) of controlPIP01 breast implants (coefficients in Table 4).  $R^2$  is the coefficient of determination. Total of 60 samples, from which 24 were in the anterior, 24 were in the equatorial and 12 in the posterior regions.



**Figure 12.** Example of Pearson correlation between stress (MPa) with 266% of strain and thickness (mm) of explanted implant PIP04 (Coefficients are reported in Table 4).  $R^2$  is the coefficient of determination. A total of 58 samples, from which 22 were in the anterior region, 24 were in the equatorial and 12 in the posterior regions.



**Figure 13** Example of Pearson correlation between stress (MPa) with 266% of strain and thickness (mm) of ControlBrand X01 implant (Coefficients are reported in Table 4). A total of 60 samples (24 in anterior; 12 in equatorial and 24 in posterior regions).

**Table 4.** Correlation analysis using Pearson Correlation for all implants.

Implants	Correlation	Anterior Region	Equatorial Region	Posterior Region
		r	r	r
ControlPip01	vs. Th	0.89*	0.84*	0.98***
ControlPip02	vs. Th	0.83*	0.97***	0.98*
ControlPip03	vs. Th	0.90**	0.95***	0.97***
ControlBrand X01	vs. Th	-0.91***	-0.95***	-0.87***
ControlBrand X02	vs. Th	-0.83***	-0.45***	-0.79***
Pip01	vs. Th	0.82***	0.77	0.94*
Pip02	vs. Th	0.78	0.72*	0.53*
Pip03	vs. Th	0.85***	0.77	0.94*
Pip04	vs. Th	0.83***	0.88***	0.64**
Pip05	vs. Th	0.82*	0.90***	0.82*
Pip06	vs. Th	0.82*	0.85***	0.47***
Pip07	vs. Th	0.89***	0.86*	0.46
Pip08	vs. Th	0.87*	0.90***	0.97***
Pip09	vs. Th	0.90***	0.87***	0.95**
Pip10	vs. Th	0.56**	0.86*	0.96*
Pip11	vs. Th	0.91*	0.95	0.90**

r is the Pearson correlation coefficient; : Stress (MPa) with a 266% strain and Th: Thickness  
 \* $P < 0.05$ ; \*\* $P < 0.01$ ; \*\*\* $P < 0.001$

## 4. Discussion

The present study, was focused on the analysis of breast implants, regarding the mechanical behaviour of the implant shell. The study included the effects of duration of implantation, differences between regions and between the two brands studied.

There were significant differences between deformation ranges of virgin and explanted implants. The largest maximum strain over the 940 individual samples tested, was close to ~266%. This limit strain was considered as well as 33%, 66% and 133%.

Since stress effects were stronger at 266% strain, all statistical analyses were carried out at this strain level.

The implantation interval of eleven explanted PIP implants (between 2005 at 2012) is similar to previous studies [1,7,15,21-23]. These studies showed that PIP breast implants produced in the period 2001-2010 had a higher probability of rupture and earlier rupture than breast implants from other manufacturers. These considerations are supported by the current study through the negative correlation between stress and years of implantation ( $r=-0.681$ ;  $P=0.02$ ) (Figure 10 b). This is particularly evident for implants produced between 2007 and 2008, which points to potential problems related to the quality assurance of the material or the manufacturing process.

Another important factor was that all the explanted implants had a life time lower than 10 years (ruptured after 3 to 8 years of implantation). Based on peer-reviewed published studies, rupture rates for PIP implants ranging from 14.5% to 31%, after 5 to 10 years of implantation, were reported [1,24-27]. In contrast, the rupture prevalence for the other manufacturers ranges from 1.1 to 11.6% [1, 28]. However, the methodology and rupture definitions varies among studies. In the present work, the TGA rupture classification was followed [19].

The appearance analysis of the ruptured implants shows different types of damage, as illustrated in Figs. 1 and 4. Most ruptures showed large damaged areas. No explanation was found for the damage variability (type and extension). Thus further work is needed to understand the significance of the location and rupture appearance.

Discoloration of the silicone gel was presented in explanted implants, mainly those with abrupt ruptures, with varying degrees of yellowing. It has been reported that PIP implants are softer and more likely to have yellow discoloration than other implants [9]. Yellowing is not unique to PIP implants. It has been attributed to a higher liquidity of the

silicone and to a higher tendency of cholesterol absorption into the implants [12,14,15,29]. A lower cohesiveness of the gel from ruptured implants was also verified. Previous studies pointed to the gel's reduced viscoelasticity [22], and the *in vivo* exposure of the silicone (leading to hydrolytic degradation and cross-link scission [15,30]) as the main reasons.

Using multifactor ANOVA (Table 3) all PIP implants (3 controls and 11 explants) were compared. The results show the stress variation between implants. The variability observed may be associated with lot-to-lot differences [8]. Ideally, it is desirable to compare the properties of the explants with those of lot-matched controls. For this study, it was not possible to control the implant lot since all implants were kindly donated for research purposes.

The different results obtained for PIP explants, may also be associated with external factors. Among these, loading induced by normal daily activities (which cannot be controlled and/or fully understood) may generate damage in the shell due to fatigue effects. Moreover, there are clinical observations that suggest the presence of two distinct subpopulations of PIP implants. Those manufactured from suboptimal industrial silicone, that are more susceptible to rupture and those containing 'normal' appearing silicones [9, 15].

For brand X group, there are stress differences among the implants (Figure 6), however the variations between implants maximum and minimum stresses are smaller. This seems to point to tighten quality control level of the manufacturing process.

There was high rupture incidence in the posterior region for explanted implants. The statistical analysis confirms that the posterior region is significantly different from other regions ( $P=0.001$ ), as illustrated in Table 2 and Figure 8. However, for PIP controls the equatorial region showed a lower stress when compared with the other two regions.

For the brand X (Table 2), the regional data are consistent with the PIP control implants. The breast implant material is a polydimethylsiloxane (PDMS) silicone rubber, which can have a wide range of structures and properties, depending on a range of molecular weights and crosslinking. The typical manufacturing process of breast implants shells uses a technique of immersion of a mould in a liquefied PDMS batch. The process is done manually leading to regional property differences over the implant shell, due to “*differences in forming temperature, pressure, and other variables*” [24].

Globally, explants did not show a statistically significant change of properties over time (Figure 10 (a)). Our results are consistent with the study made by Swarts et al. [11]. However, Yildirimer et al. [15], reported that the mechanical properties’ decreased while the implantation time increased. Even so, it must be taken into account that Yildirimer et al. [15] analysed a small number of samples, a total of 18 PIP implants, three of which were ruptured at explantation. In contrast, in this study, there was a total of 11 explanted implants (all with rupture) with a total of 604 samples analysed.

The literature, reports several factors that may affect the integrity of the shell, one of them being the different shell thickness along the implant [31]. This variability was observed in the present study. It was found that different areas of the shell had different thicknesses on nearly all the PIP explanted implants. The minimum thickness average was 0.62mm and the maximum 1.07mm.

Although the average thickness of all PIP samples (820 samples) falls within the manufacturer’s specifications (range from 0.5 and 1.0 mm) [32], in some implant regions the minimum thickness was below 0.5mm and the maximum above 1mm. The data is in accordance with the recent studies reported by TGA testing updates, ANSM, Swarts et al. [11,13,33].

For brand X, the thickness variation was in compliance with the manufacturer's specifications. Statistically, for higher stresses, there were lower thicknesses (Figure 13) [34].

For PIP implants, the variation of thickness may be related to the possibility of failure, since the minimum stress is associated to the minimum thickness measured in all the implants (Figures 11 and 12). The variability of the results for PIP implants agrees with the findings of Swart et al. [11]. Those authors establish a link between manufacturing techniques and properties variability. Implant production may require finishing by hand dipping in silicone and pushing on to a bed of salt ('lost-salt' process) before curing [11, 17]. The authors suspected of a lack of quality control at this stage of manufacture [11].

In the present study, the good correlation between thickness and stress may partly explain the higher early rupture rates of PIP implants. Moreover, it may indicate an inconsistency in the manufacturing process and/or raw material selection.

## **5. Conclusions**

The mechanical properties of the shell from breast implants were studied on PIP explanted and control implants (PIP and brand X). The normal lifespan of breast implants is of 10 or more years (120±months), however we have seen a higher prevalence of rupture in the average implantation period of 57.36 months (approximately 5 years). This temporal shift points to structural and integrity problems.

The data analysis carried out confirms that there is an uptake of lipophilic molecules into the implant gel as revealed by visual inspection. It also demonstrates that the physical characteristics of the PIP implant are variable, and have a strong relationship with the

shell thickness (thickness variations). It was observed that for PIP implants the thinner thicknesses are likely to have a lower strength and a higher probability of failure. The posterior region had lower strength than other regions. This may indicate poor quality in both design and the manufacturing processes and is supported by the heterogeneous nature of PIP implant quality, as reported by other authors [9,10,11,15].

The study results show that PIP implants were of substandard biodegradability quality. The main implications of this research are that PIP manufacturer should have paid greater attention to implant design, material selection and as the results point, the manufacturing process. By doing so, tighter tolerances for the shell thickness and material resistance can be guaranteed.

The brand X analysed can be a good example of quality control in breast implants. These results may contribute to dispel fears among the medical community and patients about the reliability of breast implants.

## **Acknowledgements**

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## **Declaration of Conflicting Interests**

None declared.

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**Article 3**  
**(Letter Communication)**

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**Breast Implants Rupture Induced by Fatigue Phenomena**

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The research to study the causes and mechanisms of rupture of explanted breast implants (Poly Implant Prosthesis – PIP), led to evidence that fatigue phenomena may be associated to the initiation of implant ruptures. This finding has not been reported in the literature as far as the authors are aware.

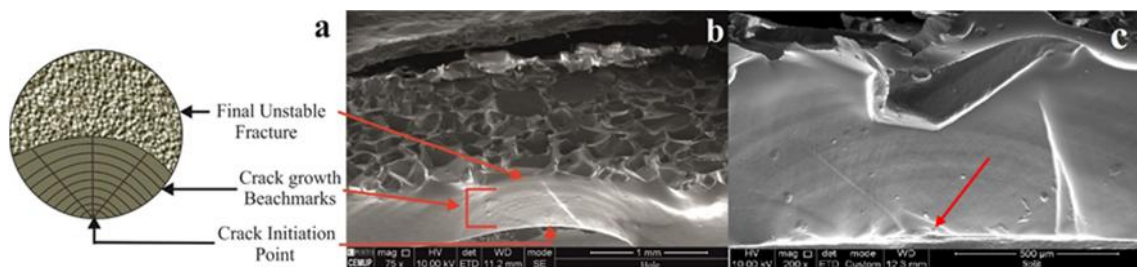
Eleven ruptured implants, explanted in the Department of Plastic Surgery of the Hospital Center of Gaia, Portugal, were analysed. The shell samples, cut from the ruptured region were examined by SEM (Scanning electron microscopy) at CEMUP (University of Porto, Portugal).

The implants were classified in four groups, according to the type of shell rupture (hole, split, v-shaped and gross damage), following the methodology introduced by the Department of Health Therapeutic Goods Administration (Australia) [1]. Four explants had a v-shape split, three had a hole, two a split and the remaining two presented gross damage. The cause of large ruptures (v-shape split and gross damage) could not be identified. Implants with small ruptures (hole or split) showed fractographic features commonly found in fatigue processes, as illustrated in Figure 1.

This suggests that implants, subjected to cyclic loads, can initiate fatigue cracks from pre-existing defects (microstructural inhomogeneities such as inclusions, pores and among others [2,3]). As can be observed in Figure 1, a fractographic analysis reveals crack initiation and microscopic crack growth, as illustrated by the “beachmarks” present in the rupture surface. The “beachmarks” have a concave shape, typically associated to the crack initiation origin (see red arrow in Figure 1). They are due to nonuniform crack propagation induced by variation of the externally applied loads (e.g running, walking and other) [4]. The “beachmarks” represent one of the most well known morphologic feature of fatigue crack surfaces. The final unstable fracture (rupture) occurs when the crack length exceeds a critical depth.

The fractographic features found, illustrated in Figure 1, were also simulated in a laboratory setting, subjecting specimens from a control implant to a fatigue test under constant amplitude loading conditions. The features identified in some ruptured implants (explanted), and in specimens (control) tested under laboratory controlled conditions, indicate that fatigue phenomena can be the cause of some ruptures.

The findings suggest that fatigue phenomena should be taken under consideration by the implant manufacturers when characterizing the mechanical behaviour of the shell, for homologation purposes. To better understand this type of failure and to establish its relative importance among other implant rupture mechanisms, further research is required. The information will lead to potentiate safer and more compliant products.



**Figure 1.** Fatigue fracture surface a) schematic representation [4], b) micrograph (75x) of implant samples from the hole rupture site, and c) micrograph (200x) of implant samples from the split rupture site; the arrow points to the fatigue crack origin.

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**An Experimental Analysis of Shell Failure in Breast Implants**

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## **Abstract**

Breast implant durability and the mechanisms of rupture are important topics in the medical community, for patients, manufactures and regulatory medical agencies. After concerns about the Poly Implant Prosthesis (PIP) implants, the need for understanding the adverse outcomes and the failure mode to improve the breast implants increased. The objective of this research is to analyze and describe the rupture characteristics of failed explanted PIP implants to study the modes and causes of rupture.

Eleven explanted PIP implants were analysed by visual inspection and scanning electron microscopy (SEM). To simulate hypothetical ruptures caused by cyclic mechanical stress (fatigue) in the implant shell, two control implants were submitted to fatigue tests, and analysed with SEM.

None of the samples from the explants showed any damage that could be associated to surgical instruments. However, for small ruptures (either Hole or split) striations were found, which normally appear due to fatigue phenomena.

In the context of this work, the striations found in explants constitute a significant finding as they point to the occurrence of fatigue phenomena associated with mammary implants rupture. This research, also demonstrates that rupture surface analysis of explanted breast implants has the potential to become a useful indicator for assessing implant rupture mechanisms.

**Keywords:** Poly Implant Prosthesis (PIP), Failure mechanism, Scanning electron microscopy (SEM), Fatigue.



## 1. Introduction

Breast implant rupture has been an important topic for the plastic surgery community, regulatory agencies and particularly for patients [1-3]. Concerns about the safety of silicone implants were intensified since the 2010 scandal involving Poly Implant Prothèse (PIP) manufacturer. Recent studies concluded that the probability of early rupture (life time lower than 10 years [4-5]) is higher for PIP implants. Based on published studies, rupture rates for PIP implants ranged from 14.5 to 31 % after 5 to 10 years of implantation [4,6-9], while other implants showed a rupture rate from 1 to 11.6% [4-5].

The failure of breast implants is influenced by different factors: material ageing following implantation; surgical procedure quality (e.g., inadvertent damage); manufacturing defects; shell wrinkling; patch detachment, among others [10]. Another factor that possibly explains implant failure is the loading frequency imposed to the breast due to daily activities such as running and walking. This mechanism is called mechanical fatigue. Trauma injuries, such as shocks from car accidents [11] may play a role in the damage mechanism that causes implant failures.

Current literature states that a large percentage of PIP implants ruptures are possibly related to the shell quality over a large number of batches [1, 12]. This may point to a considerable variability in the manufacturing process.

Even though the literature indicates the manufacturing process as a principal factor leading to PIP implants failure, few studies have been conducted to characterize the type of failure (e.g. damage due to surgical instruments, cyclic loading/fatigue). Therefore, the

objective of this study is to determine the causes of rupture by analyzing failed explanted breast implants.

## **2. Material and Methods**

### **2.1 Materials**

To understand how and why PIP implants show significant rates of premature failure, eleven ruptured implants, explanted in the Department of Plastic Surgery of the Hospital Center of Gaia, Portugal, were analysed. Sealed controls were supplied by the National Authority of Medicines and Health Products (INFARMED, Portugal).

The explants underwent a disinfection procedure, following the health regulatory authority's standard procedure [1-2].

The diagnostic techniques for explants were: Stage 1 - visual inspection; Stage 2 - scanning electron microscopy (SEM)); Stage 3 - Mechanical testing.

The last two stages will be described in sub-chapters: 2.2 Scanning electron microscopy analysis and 2.3 Fatigue test.

During Stage 1, failure regions, shell rupture (hole, split or v-shaped), discoloration, opacity and other features were recorded. In this work, the methodology reported by the Department of Health Therapeutic Goods Administration (Australia) [3] was followed.

### **2.2. Scanning Electron Microscopy (SEM) Analysis**

After inspection and disinfection of explants, several samples were cut from the rupture region for examination by SEM at CEMUP (University of Porto, Portugal).



Virgin (control) implants were used to simulate the implant rupture caused by a cyclic mechanical stress in the implant shell.

Fractographic studies of fatigue cracks were conducted with SEM to identify characteristic features of crack initiation and growth. Such analyses provide data on local deformation, loading conditions, crack initiation, and propagation path leading to fracture.

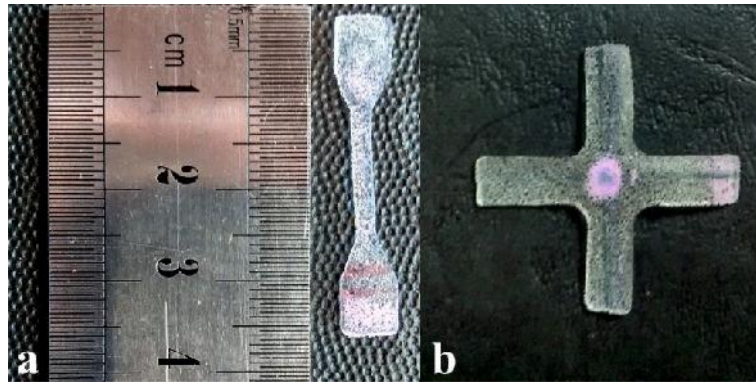
For SEM analysis, samples were coated with an Au/Pd thin film, by sputtering, using the SPI Module Sputter Coater equipment, for 120 s and with a 15 mA current.

## **2.3 Fatigue Test**

Fatigue tests were carried out to simulate a mechanism of fatigue crack growth, particularly the fractographic features in the implant material (Polydimethylsiloxane). The samples were fatigue loaded in a mechanical testing prototype (uniaxial/biaxial), with two load cells with 50N capacity, developed at INEGI Biomechanics Laboratory.

The main fatigue test parameters were waveform, frequency, force or displacement levels, loading mode, and test duration [11]. The samples were tested at 1 Hz because it is similar to that of walking or a beating heart [11]. The displacement amplitude was 15mm, equivalent to ~20% strain in the narrow region of the specimen. This displacement was used following tensile tests carried out in control implants. Two sample geometries were used. One a dog bone-shaped type 4 (shaft length 12 mm, width 2 mm). The other a biaxial geometry, 5x5 mm central square region, to induce similar stresses on both axes, as illustrated in Figure 1. The biaxial test tries to mimic the planar stresses occurring on the implanted shell, due to cyclic loading, which should be closer to real life.

A defect was introduced in the center of the sample, using a needle of 2.3  $\mu\text{m}$  diameter tip.



**Figure 1.** Samples Geometries. a) uniaxial sample and b) biaxial sample.

### **3. Results**

#### **3.1. Visual Inspection of Implants**

Eleven ruptured explants and three control implants were analysed to characterize modes and causes of implant failure.

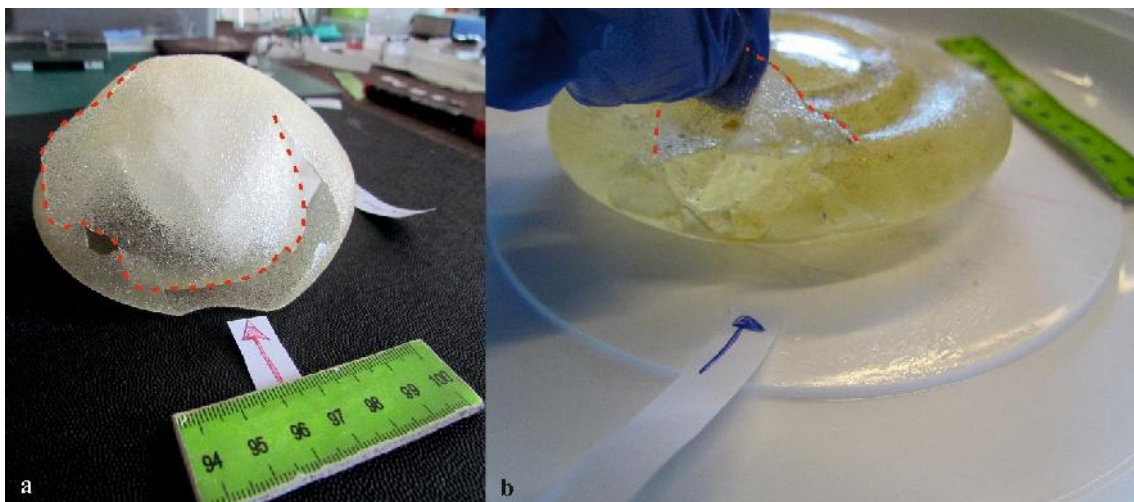
All the explants had round shape, textured shell and volumes ranging between 210 and 310 cc. These implants were implanted on a period ranging from 6 to 95 months, with an average of  $57.36 \pm 19.96$  months. Three round shaped and textured control implants were used in this study. One implant with a volume of 265cc and two with 365cc.

Following the nomenclature of TGA [3] to describe implant rupture, four explants had a v-shape split, three had a hole, two a split and the remaining two presented gross damage.

### 3.2. SEM Analysis of Shells and Failure Regions

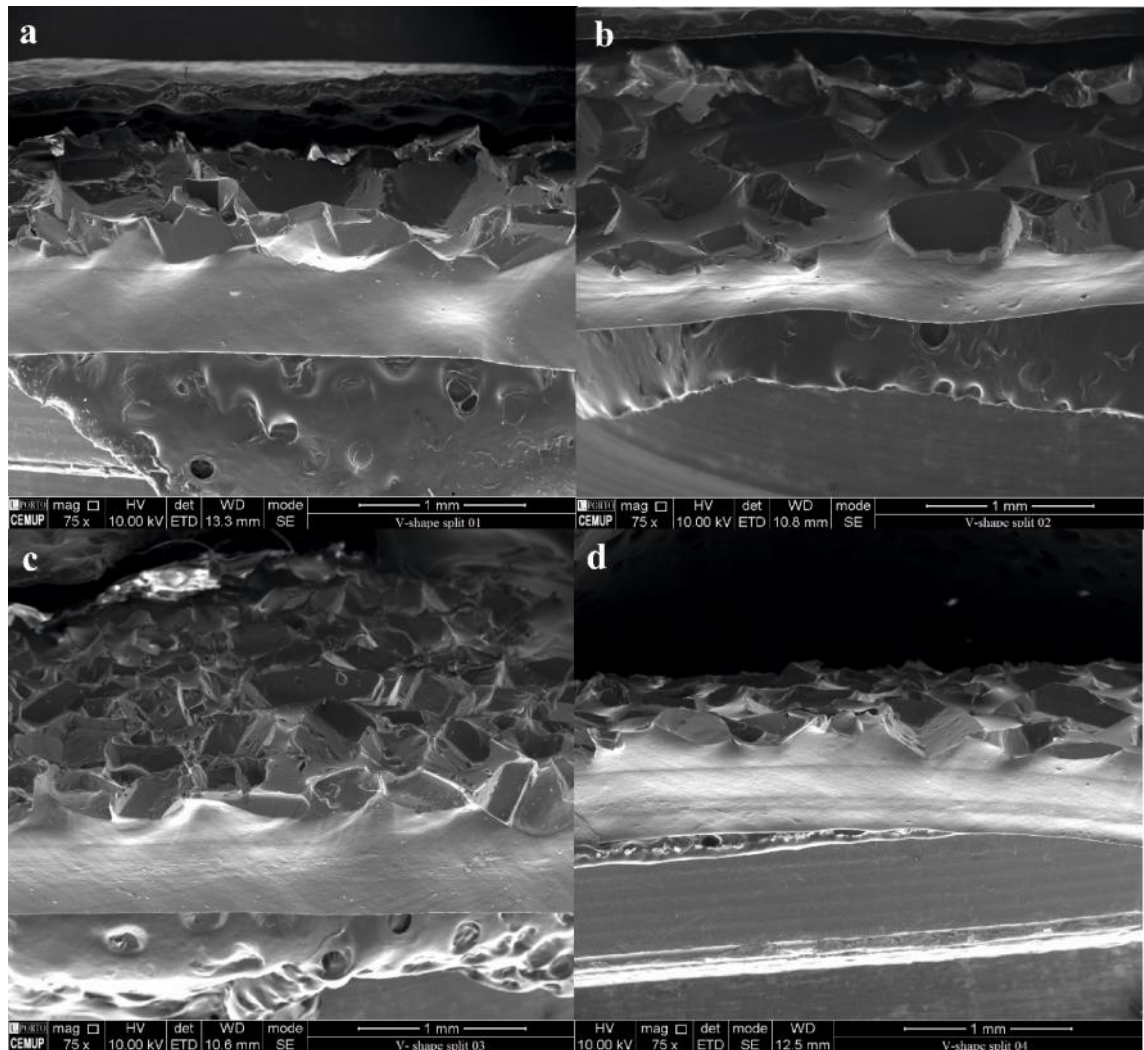
Implant failure was analysed through SEM images from the rupture site at the cross section (magnifications of 75x and 200x).

The four implants with v-shape split ruptures had volumes between 225 and 310cc, yellow coloration and large ruptures that covered an extensive area. Rupture size varied from 80mm to 140mm. Four to six samples were removed (depending on rupture size), to enable the SEM analysis. Figure 2 shows examples of gross damage and v-shape rupture.



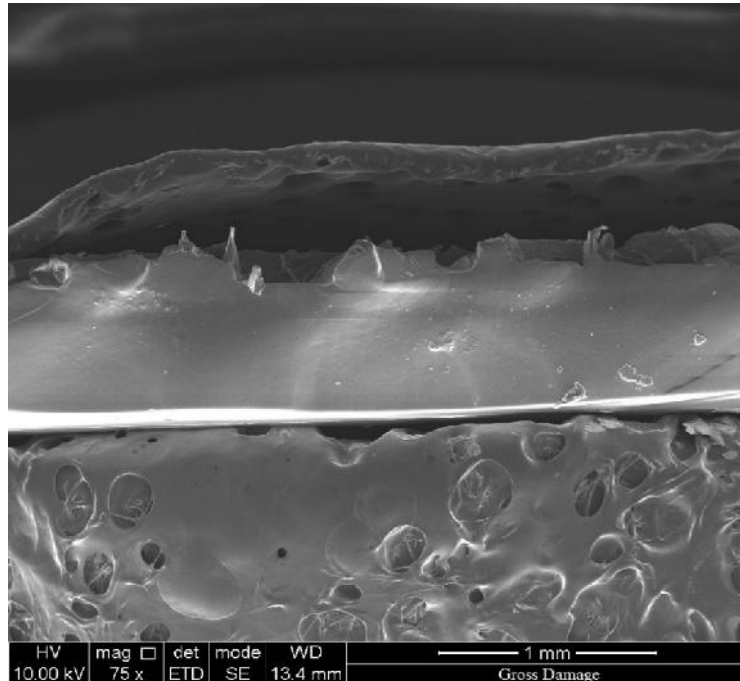
**Figure 2.** Two implants with different ruptures. a) Gross Damage with 140mm of rupture size; b) V-shape split with 80mm.

It was not possible to identify the origin of the rupture through SEM analysis. The cross-sectional images of four implants (with v-shape split) were inconclusive, as seen in the Figure 3.



**Figure 3.** SEM micrographs (75x), view of v-shape split rupture in four implants.

It was also impossible to identify the origin of the implant rupture with gross damage, as shown in Figure 4. The implants with gross damage had volumes of 245cc and 260cc; they present extensive shell rupture, covering all regions of the implant. The same procedure, used in the v-shaped split implants, was used for sample collection on the gross damaged implants.

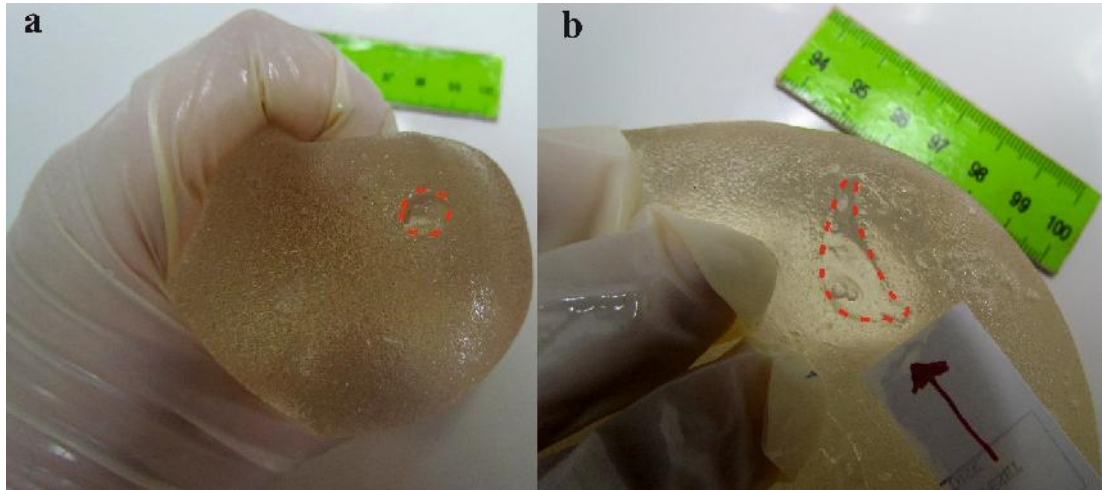


**Figure 4.** SEM micrographs (75x), view of gross damage of the implant shell.

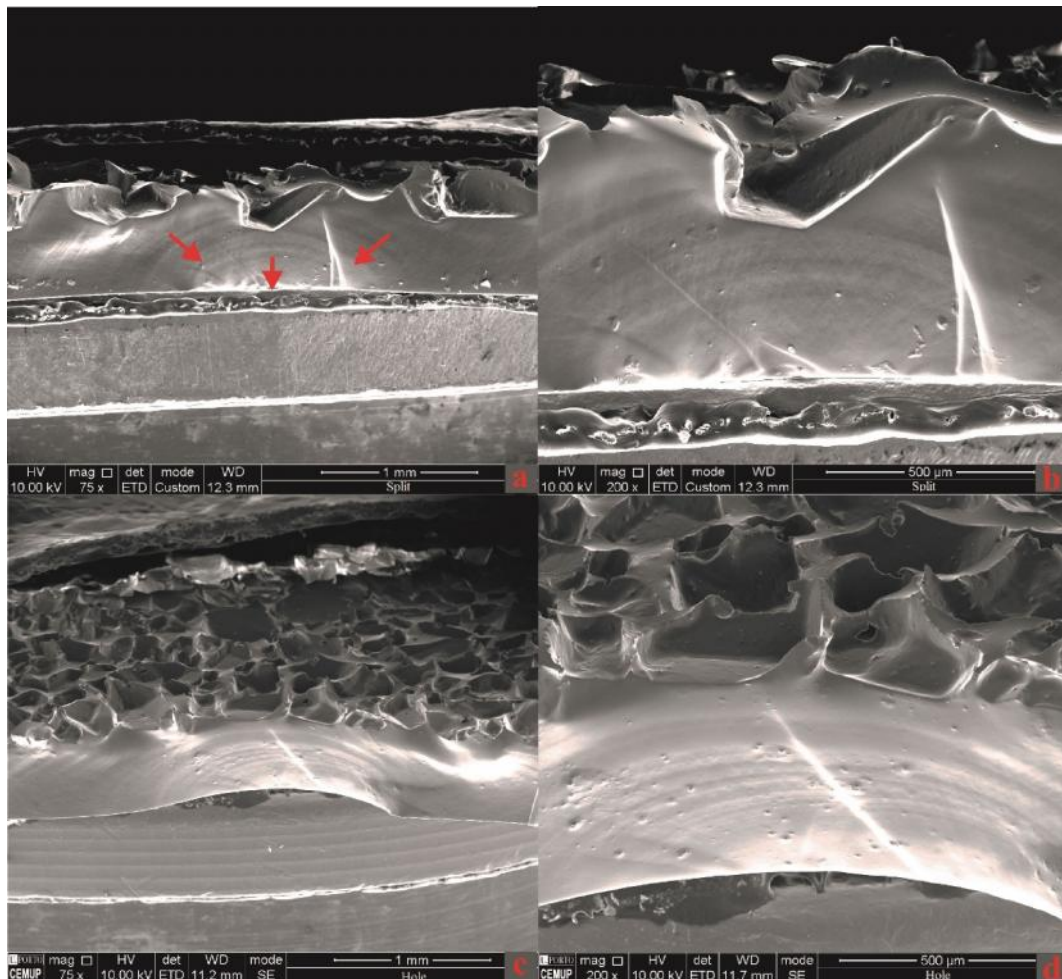
The implants with small ruptures had volumes between 210 and 250cc. They are almost translucent, as shown in Figure 5. Due to the small size of the damaged area, it was analysed one sample from each implant (taken from the ruptured area). These samples (with hole or split) provide more conclusive results. In Figure 6, the damage starting point is clearly visible. Moreover, there is visible striation normally associated to fatigue [11, 13-14], although fatigue crack growth can occur without striation formation [13]. Macroscopic marks such as “beachmarks” can be formed by thousands of striations. Each striation is formed due to one load cycle, although not all load cycles produce a striation.

In the context of this work, this constitutes a significant finding as it points (with a high degree of certainty) to the occurrence of fatigue phenomena associated with breast implants rupture.





**Figure 5.** Small ruptures. a) Hole rupture and b) Split rupture.



**Figure 6.** SEM micrographs of small ruptures - a,c) 75x; b,d) 200x. The same type of striations appears in all implants. a,b) Split rupture and c,d) Hole rupture. The arrows point in the direction of crack initiation point.

### 3.3 Fatigue Tests Results

The features identified in the crack surfaces of ruptured implants with small defects, shown in Figure 6, seem to indicate that the defects grow by a fatigue mechanism.

Hypothetically, fatigue failure may be one of the mechanisms involved in implant rupture. To study this effect, automated fatigue crack growth tests were conducted on control implants samples.

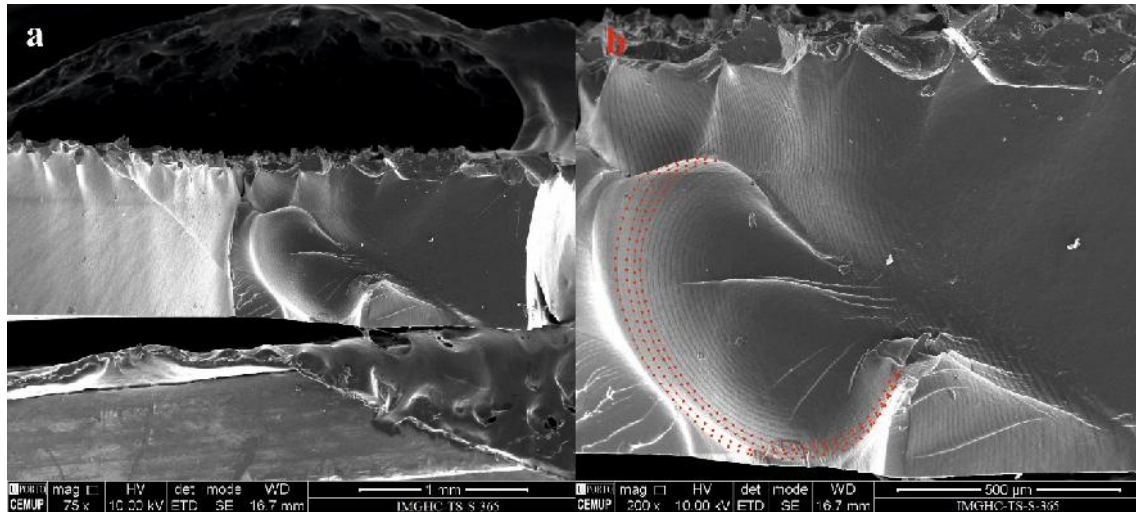
This technique provides information on the relative strength of different breast implants, through standard fatigue test conditions applied to implant shell samples. The failure surfaces of the fatigued shell samples were examined using SEM, and the details of both the inside and outside surfaces of the shell at the failure location were described. The samples failed after ~10.000 cycles.

Two round, textured controls (volume of 365cc) were used, to simulate the effect of fatigue on the implant shell. Table 1 describes the information available about each implant.

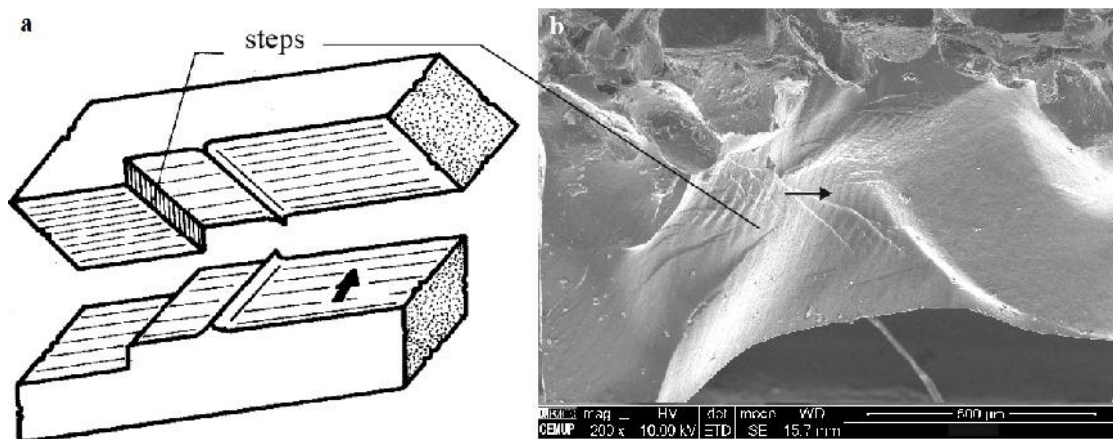
**Table 1.** Information about the control PIP breast implants

Explanted Implants	Reference	Volume (cc)	Lot	SN	Type of test	Type of Sample	Number of Samples
Control01	IMGHC-TX-S-365	365	24709	095	Fatigue Growth	Uniaxial	4
Control02	IMGHC-TS-S-365	365	24809	616	Fatigue Growth	Biaxial	2

Figure 7 and 8 shows biaxial samples' SEM results. The striations observed are strong indicators that a fatigue process took place as a consequence of the cyclic loading. These results suggest that striations do appear in the implant shell material as a consequence of fatigue processes (Figure 6).



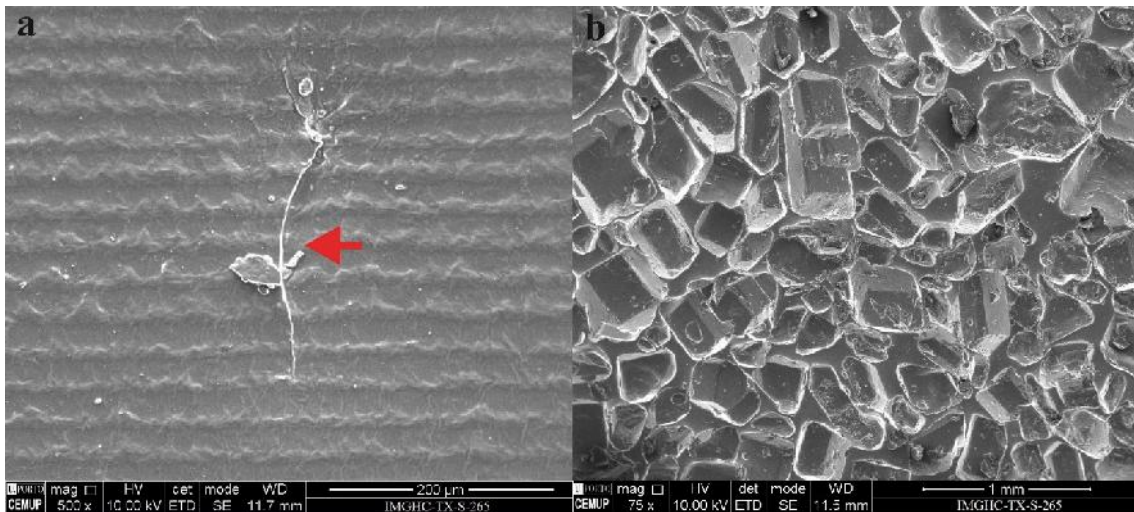
**Figure 7.** SEM micrographs (a) 75x and b) 200x) in biaxiais samples, the striations are visible. The dashed lines are to emphasize part of striations.



**Figure 8.** a) schematic representation of radial tearing lines or ridges, and propagation of a fatigue crack in parallel planes (see de arrow); b) micrograph of biaxial samples, the radial tearing lines and the fatigue crack in parallel planes (see arrow) are visible.



As shown in Figure 9, by looking towards the inner shell surface, the needle defects were clearly observed. The texture structure on the outer shell surface, makes the shell damage difficult to visualize. The inner shell surface also showed parallel markings or grooves (Figure 9-b), seen in all analysed surfaces.



**Figure 9.** SEM micrographs of the a) inner surface (500x) and b) outer surface (75x) of the sample IMGHC-TX-S-265 with damage.

## 4. Discussion

The physical characteristics of PIP implants were analysed. Both ruptured and virgin implants were studied. Control implants were intentionally damaged using a needle. These samples were subjected to fatigue tests (uniaxial and biaxial). The research aimed at explaining device failure mechanisms, in particular those associated to fatigue induced crack, propagation and ultimately rupture.

The explants were examined and classified according to their rupture characteristics. Yellow color appeared in implants with abrupt ruptures, whereas implants with small ruptures were translucent. According to literature, the yellow color may be caused by the ingress of biological fluid (after implantation) into the implant and by a

strong tendency for cholesterol absorption into the implants [12, 15-18]. Each of the explants, had an appearance in agreement with known types of damage namely, hole, v-shape split and gross damage. This variability was verified by other authors [3, 19].

Figure 6 showed one type of striation commonly found in fatigue processes [20]. It suggests that implants can be subjected to cyclic loading that may lead to rupture. A typical metal fatigue crack surface shows the three phases of a fatigue process, as shown in Figures 6: crack initiation, crack growth and final unstable fracture.

The crack initiation is composed by nucleation and microscopic growth of the crack. When this occurs inside a material, it is normally associated to the presence of defects and microstructural inhomogeneities (for example, inclusions and pores) [11, 14].

“Beachmarks”, are a characteristic feature of fatigue crack surfaces. These marks appear during crack propagation and show the crack front position and its propagation direction history throughout fatigue. “Beachmarks” are associated to crack growth rate variations, stops/ accelerations due to load variations induced by internal or external causes [13]. For instance, changes in load or the degradation process over a certain period of time [11, 13-14] are sources of “beachmarks”.

The final unstable fracture occurs when the crack length exceeds a critical depth. Critical depths depend on the toughness and resistance of the material and on the loading conditions [13-14, 21].

The striation of explanted implants differs from the laboratory simulated ones. In laboratory conditions, the striations are equidistant, as load cycles were imposed at a constant frequency. In contrast, there were significant variations of the distances between explanted implants striations. A possible explanation is the random (potentially aperiodic) nature of the cyclic mechanical loadings due to repetitive voluntary activities of the woman.

The striations are common in a large number of metallic materials [22], where it is possible to identify fatigue marks. Fatigue striations are also visible in different [23-28] (other than metallic) materials. Regarding implant shell materials, the authors could not find studies that correlate the striations with the fatigue process. Therefore, it was necessary to determine whether these materials generated similar fatigue markings on fracture surfaces produced under cyclic loading conditions.

In Figure 6, the crack initiation started from the inside (smooth) to the outside (textured) of the shell. This may indicate that the rupture happened due to the quality of the material (e.g, inclusions and pores), or due to significant body motion (physical activities, trauma, among others). As large cyclic stresses weaken the implant, as fatigue damage accumulates, the corresponding crack propagation may result in failure. If implant shell rupture does occur due to a fatigue process, the following possibilities should be considered: the cyclic loading that ultimately led to crack propagation and implant rupture, must have occurred after implantation; there was a shell defect, either prior to implantation or inflicted (accidentally) during implantation that acted as the nucleation agent for the fatigue process. Since the whole implant history is unknown and a detailed (microstructural) analysis of the implant is not available, there is no definitive evidence on the two possibilities stated. To confirm fatigue as a source of the striations (Figures 6), laboratory fatigue tests were carried out on virgin implant shell samples. The fatigue test results shown in Figure 7 shows the fatigue striations in a regular pattern.

Figure 8 shows “step” striations. They correspond to different point of crack propagation, at different planes that can connect to each other [13].

The authors could not find any other studies in the available literature, showing the existence of fatigue striation in explanted breast implants.

Other researchers have studied how fatigue affects breast implants using a laboratory apparatus in which flat plates cyclically compressed the implants [29-32]. Marotta et al. [30] concluded ‘that a primary mechanism for rupture must be the progressive cyclic mechanical stress induced creation and enlargement of tears in weakened silicone fluid swollen silicone elastomeric shells’. Brandon et al. [29], described the morphological features of the fatigue failure surfaces of smooth and textured saline-filled breast implants. These studies showed that some fatigue processes can induce the shell rupture in breast implants.

Two different modes of failure were observed depending on the magnitude of the cyclic load, corresponding to a different number of fatigue cycles at failure. The rupture modes found in Brandon et al. [29] were similar to those found on samples of the explanted implants in the current study. Both ruptures were small, one was a tear and other a pinhole. More recently, Haws et al. [33], analysed the surgical techniques to better understand the etiology of implant rupture, suggesting the occurrence of flex fatigue.

The parallel marking lines observed in inner shell surfaces (Figure 9) were also verified by Swarts et al. [15] and on PIP- Technical Report [34]. Swart et al. [15] suggested that the marks can be caused by the manufacturing forms used to shape the shell. As shown in Figure 9, this type of material makes difficult to determine the origin of rupture, even when the outer shell surface has visible rupture. Therefore, if there are some instrument damage during surgery or if the material is already damaged, it will be difficult for the surgeon to identify the rupture.

In summary, further research is important to analyse breast implants failure, and understand the mechanisms that generate the shell damage, with potential to improve the manufacture process.

## 5. Conclusions

The SEM analysis used in this research allow the classification of the failure mode, and may be an important tool in the diagnosis of implant failure mechanisms.

In summary, the implant failure may be related to implant handling before the surgical procedure, the implantation procedure, *in vivo* processes (e.g., abrasion or breast biopsy), the explantation procedure, and *in vivo* cyclic loading that may induce fatigue damage in the implant.

This study suggests that the fatigue damage can be potential cause of *in vivo* failure. The implant accumulation of *in vivo* cyclic loading over the years is unavoidable. To better understand this type of failure and to establish its relative importance among other implant rupture mechanisms, further research is required.

This manuscript emphasizes the need to analyze the explanted implants after rupture. Data collection at the time of explantation by a surgeon or appropriate healthcare provider at the explant site is recommended. Thus, information such as - reason(s) for the device explantation; the presence of any shell defects; type of rupture; extent of implant rupture (intracapsular, extracapsular, or migrated gel); any discoloration, opacity of the shell and in the filler; and whether the rupture occurred before or during explantation (if applicable) - would be (potentially) helpful for the product's improvement.

The existence of information about the breast implant after rupture, may potentiate the development and improvement of safer and more compliant products, being at the same time a significant tool for future scientific research and product monitoring.

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## Article 5

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### **Intact vs Ruptured Poly Implant Prothèse (PIP) Breast Implants. A Woman-centric Paired Analysis**

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Surgery*



## **Abstract**

**Background:** Despite many studies that evaluated breast implants rupture, there is no consensus over causes and incidence. Most studies lack a multifactor analysis of risk causes associated with breast implant rupture. To fill this gap, an experimental protocol was developed to compare ruptured and intact Poly Implant Prothèse (PIP) breast implants from the same woman. These conditions guarantee that physical/biological variables are the same for each woman.

**Methods:** Twenty-two PIP explants (eleven intact and eleven ruptured) and three control PIP implants were analysed. The mechanical properties of ruptured and intact implants were compared in terms of brand, lot, implantation time, and demographic conditions.

**Results:** In general, there were statically significant differences between intact and ruptures PIP implants. Ruptured implants were thinner (0.73mm Vs 0.91mm) and weaker (7.42MPa Vs 9.59MPa) than intact implants. The same was observed for each woman, using a paired analysis of the intact Vs ruptured implants.

**Conclusions:** Intact and ruptured implants have distinct mechanical behaviours, and thickness variations. According with authors' understanding of the problem, these differences may be associated with the typical manufacturing process of breast implant shells. The results stress the importance of a thorough control of the shell thickness. Given its relevance, shell thickness should be used as a control quality measure for homologation purposes.

**Keywords:** Breast Implants, Explanted Implants, Ruptured Implants, Mechanical Properties.

## 1. Introduction

Breast implant rupture is one of the principal complications and concerns surrounding mammary prosthesis implantation. It represents the main cause of implant removal. Usually, the rupture of silicone breast implants from recent generations, does not produce a change in volume. Consequently, the patient does not know that a rupture has occurred. Therefore, one of the principal concerns about breast implants is to understand the cause of rupture and how to avoid it. Several papers have studied breast implant failure [1-8]. Their fundamental contributions were: failure reasons; shell-gel interaction; and change of material properties during implantation.

Several studies [2-3,6,9-11] pointed as primary factor for *in vivo* implant shell failure, the change of mechanical properties over time. These results showed that shell strength, toughness and elasticity decrease with implantation time for all implant generations (first, second or third generation). However, other studies pointed the shell swelling, by the inner gel, as the main factor responsible for the shell decay [4-5,9,12-15]. Recently, the concerns about the Poly Implant Prothèse breast implants renewed the discussion about implant failure causes. According to some authors [8,16,17] shell rupture may be related to deficiencies in manufacturing techniques, associated with shell thickness variation.

However, besides the problems listed above, breast implants can fail for other reasons: shell damage caused by surgical instruments; open or closed capsulotomy; shell wrinkling; mechanical pressure during mammography; needle biopsy or hematoma aspiration; and cyclic fatigue or friction between tissues and implant [15,18].

Despite many studies that evaluated breast implants rupture, there is no consensus over causes and incidence. Most studies lack a multifactor analysis of risk causes associated with breast implant rupture. The rupture causes may depend on multifactors such as the patient, the implant, biological and environmental factors. Due to literature shortcomings an experimental protocol to study the rupture causes was developed. Paired analysis of the mechanical properties of ruptured and intact PIP explanted implants per patient was conducted. The mechanical properties were then discussed on the light of the demographic conditions (same patient, age, BMI, physical activity, surgery, implantation time, implant position, among others). The intact and ruptured implants were analysed and compared by manufacturer and lot.

## **2. Material and Methods**

Patient and implant data are summarized in Table 1. Details regarding samples preparation, the mechanical testing protocol and chemical analysis (FTIR) are presented in the following sections.

### **2.1 Clinical Data**

This study includes clinical information about 11 women undergoing a removal and replacement of breast implants (2 per woman) between February 2012 to July 2013, at the Department of Plastic Surgery of the Hospital Centre of Gaia, Portugal. The 22 implants removed were analysed and linked to their clinical information (Table 1). The clinical information was obtained through computerised and paper clinical files. In some



cases, patients were called to obtain missing data. The available clinical data is summarized in Table 1.

The information collected included the following parameters:

- Patient sociodemographic data (age, body mass index (BMI), profession, pathologies/ chronic diseases);
- prior surgical interventions (date of surgery, surgical intervention);
- explantation surgery data (date, reason for surgery);
- explanted implant characterization (shape, surface, volume, region of rupture, aspect of implant, position of the implants and implantation time).

BMI was evaluated according to the classification of the WHO (World Health Organization) [19] for: underweight (BMI<18); normal (BMI>18<25) and overweight (BMI>25<30). The physical activity of each patient was assessed according to working activity: Manual labor/significant physical effort or work without significant physical activity. Three virgin implants (same brand), obtained from the National Authority of Medicine and Health Products (INFARMED), were used as controls. Code Pa# correspond to intact and ruptured implant per patient.

Before the mechanical tests the explants were classified in relation to the shell damage and gel condition, according to the Department of Health Therapeutic Goods Administration criteria [20].

**able. 1** Clinical and demographic information

Patient	Age (Years)	BMI	Profession	Pathologies/ chronic diseases	Surgical intervention	Implant Position	Duration of Implantation (months)	Local of Rupture in Breast
Pa#01	38	Normal	HPA	No Pathology	BAM	Retromuscular	46	Left
Pa#02	34	Normal	LPA	No Pathology	BAM	Subglandular	52	Left
Pa#03	40	Missing Data	Missing Data	No Pathology	Missing Data	Retromuscular	61	Right
Pa#04	54	Overweight	LPA	Breast, colon cancer	SM+ Implant	Retromuscular	39	Left
Pa#05	52	Overweight	LPA	No Pathology	BAM	Subglandular	56	Right
Pa#06	44	Normal	HPA	No Pathology	BAM	Subglandular	40	Left
Pa#07	31	Overweight	HPA	No Pathology	MI	Subglandular	50	Left
Pa#08	65	Normal	LPA	No Pathology	BAM	Subglandular	64	Right
Pa#09	65	Normal	LPA	Asthma sinusitis rhinitis	BAM	Subglandular	36	Right
Pa#10	68	Missing Data	Missing Data	Dyslipidemia, Cancer	SM+ Implant	Retromuscular	95	Left
Pa#11	56	Normal	HPA	Smoker	BAM	Retromuscular	92	Right

BMI: Body Mass Index; HPA: high physical activity; LPA: low physical activity; BAM: Bilateral Augmentation Mammoplasty;

SM: Subcutaneous Mastectomy; MI: Mastopexy with implants

## **2.2 Testing Protocol**

The experimental protocol, was developed according with the ISO standards for the analysis of breast implant materials: ISO 14607:2007 and ISO 37:2005. The 22 explants (eleven intact and eleven ruptured) and 3 control implants were prepared for testing by removing the gel from shell. The sample preparation and testing followed the protocol previously developed by the authors [21].

The samples were tested until failure with a displacement rate of 20mm/min, in one direction. It was used a mechanical testing machine prototype (with biaxial capabilities), developed at INEGI Biomechanics Laboratory (Porto, PT). To guarantee a controlled initial geometry and loading conditions, the samples were subjected to a 0.25N preload. The main mechanical property considered in this study was the tensile strength (stress at break) ( $\sigma_{max}$ ).

## **2.3 Fourier Transform Infrared Spectroscopy (FTIR) Characterization**

The chemical composition of the surfaces and gels of twenty two implants were analysed using a Cary 630 FTIR Spectrometer (Agilent Technologies, USA) equipped with a diamond attenuated total reflectance (ATR) accessory. The tests were carried out at INEGI's tribology laboratory (CETRIB): spectra were acquired over 20 scans with wavelengths ranging from 600 to 4000 $\text{cm}^{-1}$ , with a resolution of 4  $\text{cm}^{-1}$ . A background scan was performed before each sample measurement.

## 2.4 Statistical Analysis

Data was analysed using SPSS version 20 (SPSS, Inc, Chicago, Illinois) and p values <0.05 were significant. Descriptive statistic, mean  $\pm$  standard deviation, were calculated for all outcome measurements. Data normal distribution was verified with Kolmogorov- Smirnov and Shapiro-Wilk tests. A comparison between the two implant groups (intact and ruptured) was performed by means of independent-samples t test and Manny-Whitney U test. One-Way ANOVA test was used to compare the tensile strength of the shell, with gel condition and Implant status at explantation time.

## 3. Results

The sample size comprised of 11 women with ages ranging between 68 and 31 years years ( $\bar{A} = 49.73 \pm 14.64$  years). All primary implantation surgeries took place between 2005 and 2008. Most surgeries had aesthetic purposes, and were bilateral augmentation mammoplasties (n=7).

Between 2012 and 2013, 22 breast implants were collected. All implants, 11 intact and 11 ruptured, were from the same manufacturer. All patients underwent imaging diagnostic exams (ultrasound and/or MRI) with suspected implant rupture.

The explants had volumes ranging between 210cc and 310cc. The mean duration of implantation was 57.36 months ( $\pm 19.96$ ), ranging from 36 to 95 months, see Table 1. The silicone gel-filled implants were round shaped and textured.

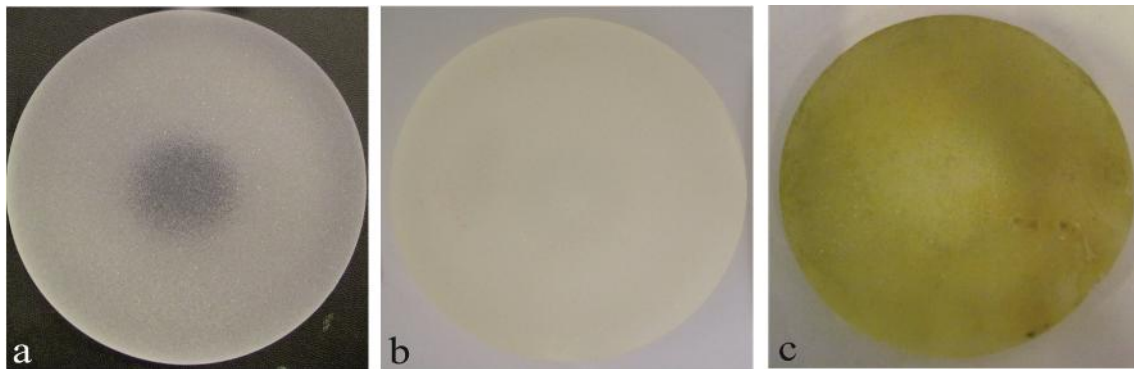
The control implants had a round shape, textured shell surface and volumes ranging between 205 and 415 cc. All control implants were from different lots.

The appearance of shell and gel, for intact and ruptured implants is summarized in Table 2, and Figures 1 and 2. Using the TGA classification [20] the gel condition of the intact explants was clear (n=7), opaque (n=4) and extremely sticky (n=11). In same case, it was difficult to separate the gel from the shell. Most of the shells were clear (n=8), while others were yellowed (n=3). Regarding the ruptured implants, the appearance and colour varied according with the type of rupture. The ruptures observed were V-shape split (n=4), hole (n=3), and split (n=2) and gross damage (n=2). For large ruptures (V-shape split and gross damage), there was a significant yellowing of both shell and gel. The gel, easily leaking from the ruptured shell, had a liquefied (oily) and non-cohesive (non-uniform) aspect. There were traces of blood in the gel. The yellowish appearance may be related with the contact between gel and tissue [6]. Explants with small ruptures (hole and split) showed clear gels and shells. Upon shell removal, the gel from Pa#02 and Pa#03 remained in a cohesive single mass, as shown in Figure 2-c. The remaining explants (Pa#05/06/07) had liquefied and non-cohesive gels. All control implants showed a clear aspect and a cohesive gel, as expected.

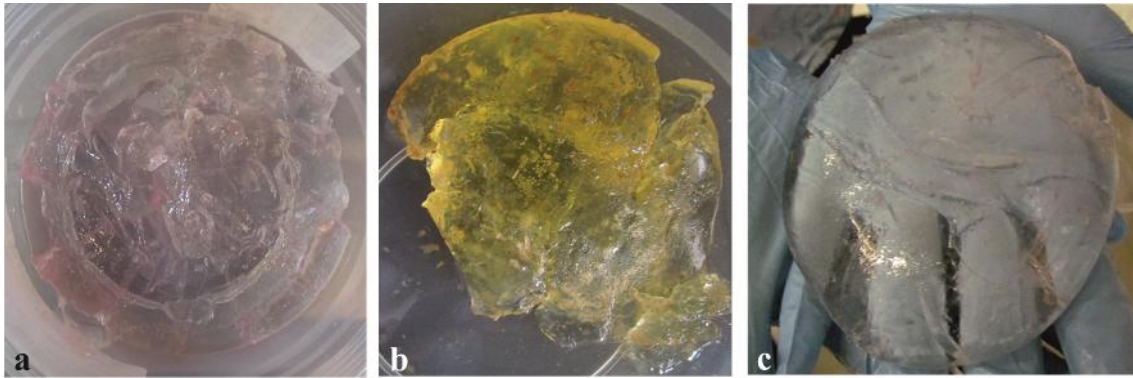
**Table 2.** Classification of gel and shell condition.

Variable	Gel Colour		Gel aspect		Shell Aspect		Rupture type	Rupture Size
	Intact	Ruptured	Intact	Ruptured	Intact	Ruptured		
Pa#01	Clear	Yellow	ExtS	NonU, oily	Clear	Yellow	V- Split	Large
Pa#02	Clear	Clear	ExtS	CohSM	Clear	Clear	Hole	Small
Pa#03	Clear	Clear	ExtS	CohSM	Clear	Yellow	Hole	Small
Pa#04	Opaque	Yellow	ExtS	NonU, oily	Yellow	Yellow	V- Split	Large
Pa#05	Clear	Yellow	ExtS	NonU, oily	Clear	Yellow	V- Split	Large
Pa#06	Clear	Clear	ExtS	NonU, oily	Clear	Clear	Split	Small
Pa#07	Clear	Clear	ExtS	NonU, oily	Clear	Clear	Split	Small
Pa#08	Opaque	Clear	ExtS	NonU, oily	Yellow	Clear	Hole	Small
Pa#09	Opaque	Yellow	ExtS	NonU, oily	Clear	Yellow	GrosDam	Large
Pa#10	Opaque	Yellow	ExtS	NonU, oily	Yellow	Yellow	GrosDam	Large
Pa#11	Clear	Yellow	ExtS	NonU, oily	Clear	Yellow	V-split	Large

ExtS: Extremely sticky; NonU: Non-uniform (non-cohesive); CohSM: cohesive single mass; GrosDam: gross damage



**Figure 1.** Shell Aspect: a) Control implant; clear aspect. b) Intact implant; clear aspect and no macroscopic changes. c) Ruptured shell; yellowed shell.



**Figure 2.** Different gel conditions: a) Clear gel, non-cohesive (oily). b) Yellow gel, non-cohesive. c) Clear gel, cohesive single mass.

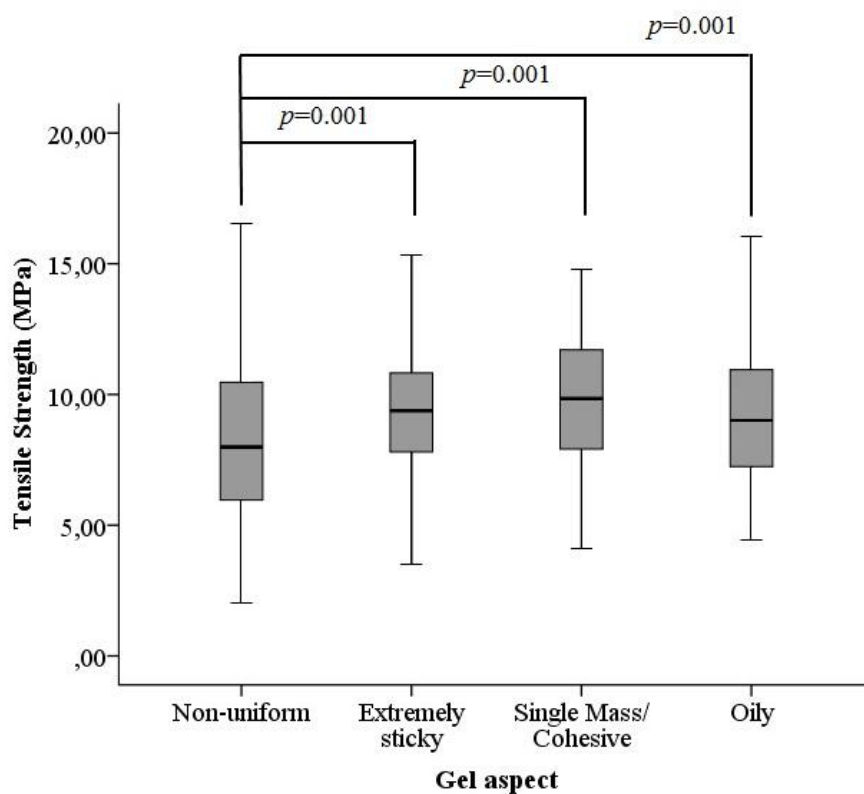
### **3.1. Shell Properties of Intact vs Ruptured Breast Implants**

In the present study, 25 breast implants (22 explanted and 3 controls) were analysed. A total of 1008 samples were removed from all the shell implants: a total of 396 samples were collected from the ruptured explanted implants, another 396 came from intact explanted implants, and 216 from the control implants (virgin).

As seen in Table 2 and Figure 2, the aspect of the gel varied with the integrity of explants. According to the available literature, there is a connection between gel cohesiveness and shell deterioration [11,16]. The relation between gel condition and shell resistance was statistically analysed. The non-uniform gels (lower cohesiveness), corresponded to shells with lower tensile strength (Table 3). This relation was statistically significant ( $p < 0.05$ ), as can be seen in Table 3 and Figure 3.

**Table. 3** Multi-factor ANOVA results regarding gel condition for explanted implants.

Mean vs standard deviation			
Gel Aspect	n	Subgroups	
		Tensile Strength (MPa)	
		1	2
Non-uniform	216	8.24 MPa ( $\pm 2.05$ )	
Oily	108	9.22MPa ( $\pm 2.33$ )	
Extremely Sticky	396	9.59MPa ( $\pm 2.10$ )	
Single Mass/ Cohesive	72	9.65MPa ( $\pm 2.02$ )	
	<i>p</i>	1.00	0.623



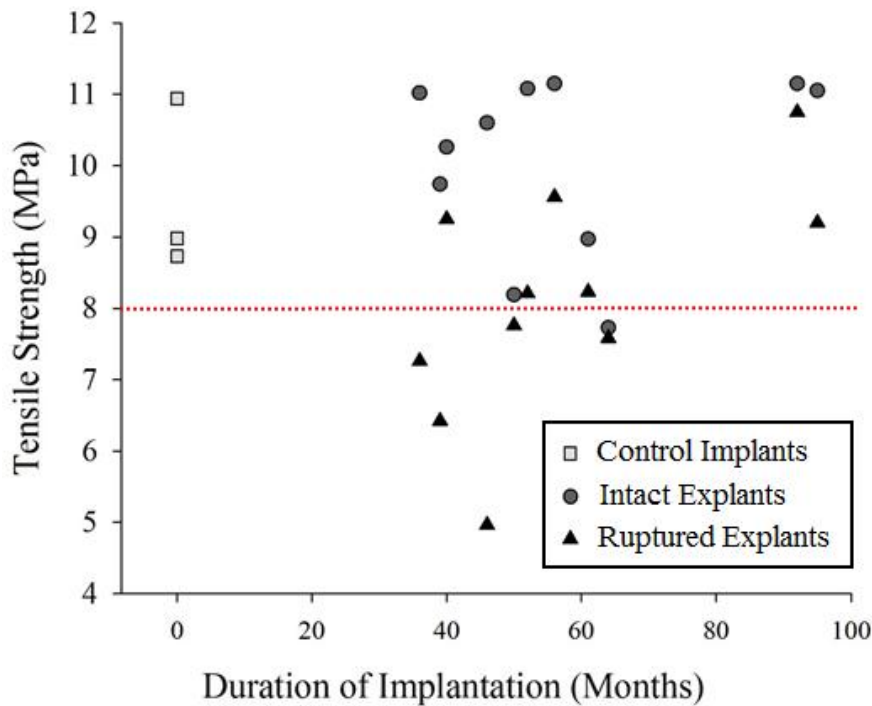
**Figure 3.** Comparison of Tensile Strength (MPa) for implant shells grouped according with gel conditions. Values are presented as median (horizontal line within box), 25-75th percentile (box) and T-bars (range to the minimum or maximum values).



The three implant groups were analysed in terms of tensile strength and thickness, see Table 4. Ruptured implants showed statistically significant differences ( $p=0.000$ ), in relation to the tensile strength of intact and control implant shells. Regarding the thickness, all implants differ from each other. It must be noticed that damaged implants had lower thickness, see Table 4. Figure 4 summarizes all tensile strength results (mean), for the three groups as a function of implantation time (months). Control data is plotted at time zero. The implantation times of explants vary from 36 to 92 months. Figure 4 shows the variation of tensile strength within explants and control groups. There is no time-dependent degradation in the mean shell tensile strength during 95 months of implantation.

**Table. 4** Multi-factor ANOVA analysis results regarding all implants, and shell thickness.

Mean vs standard deviation								
Tensile Strength (MPa)				Thickness (mm)				
Implants	n	Subgroups		Implants	n	Subgroups		
		1	2			1	2	3
Ruptured	396	7.42( $\pm$ 2.65)		Ruptured	396	0.73( $\pm$ 0.10)		
Control	216	9.56 ( $\pm$ 1.62)		Control	216	0.84( $\pm$ 0.09)		
Intact	396	9.59 ( $\pm$ 2.37)		Intact	396	0.91 ( $\pm$ 0.11)		
	<i>p</i>	1.00	0.992		<i>p</i>	1.00	1.00	1.00



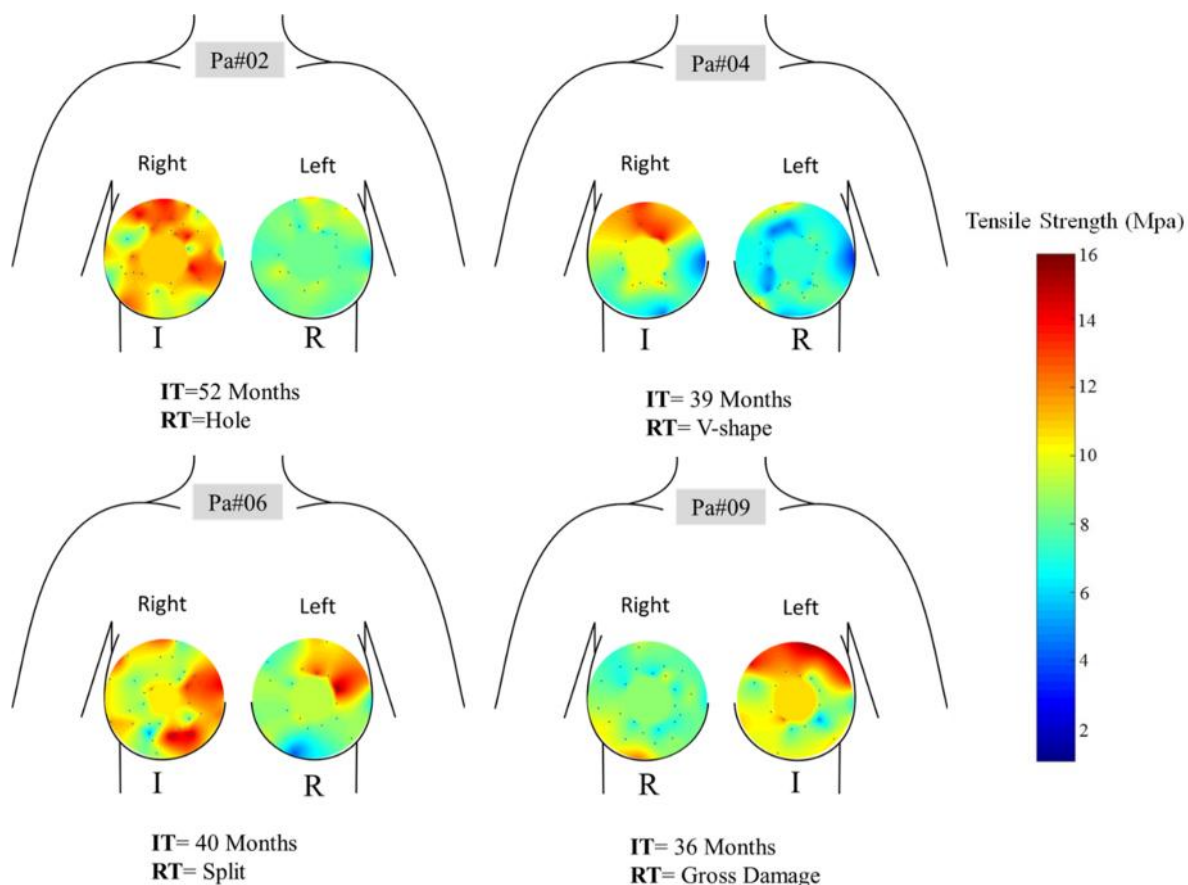
**Figure 4.** Tensile strength (MPa) of explants and controls (3 groups) as a function of implantation time (months).

The mechanical properties of the two types of explanted implants (same woman, ruptured and intact implant) were analysed according to the same demographic conditions. Table 5 showed the results of tensile strength, and thickness. It is clear that the thickness is smaller for implants with damage, with statistically significant differences ( $p=0.000$ ) in relation to intact implants (except for Pa#11). For tensile strength, statistical differences between intact and ruptured implants were found (Table 5).

To characterize the material behaviour along the shell, a contour plot was developed. Figure 5 shows the resistance of the material over the shell. The intact implants (right implant, except for Pa#06) had higher strength than ruptured implants. Higher strength is linked to red “stains”, and lower strength with blue “stains”.

**Table. 5** Tensile strength and thickness comparison between intact and ruptured explanted implants per patient (data expressed as mean±standard deviation).

Variable	n	Tensile Strength (MPa)	<i>p</i>	Thickness (mm)	<i>p</i>
Pa#01	Intact	36	10.06±1.84	<b>0.000</b>	0.94±0.10
	Ruptured	36	4.96±1.73		0.79±0.05
Pa#02	Intact	36	11.08±2.16	<b>0.000</b>	0.94±0.10
	Ruptured	36	8.21±1.36		0.73±0.07
Pa#03	Intact	36	8.97±1.59	0.077	1.0±0.14
	Ruptured	36	8.23±1.93		0.62±0.05
Pa#04	Intact	36	9.74±1.90	<b>0.000</b>	0.99±0.06
	Ruptured	36	6.42±1.98		0.70±0.10
Pa#05	Intact	36	11.15±2.03	<b>0.003</b>	1.00±0.05
	Ruptured	36	9.56±2.10		0.78±0.08
Pa#06	Intact	36	10.26±1.98	0.134	0.93±0.10
	Ruptured	36	9.25±2.33		0.73±0.11
Pa#07	Intact	36	8.19±1.36	0.295	0.91±0.09
	Ruptured	36	7.76±2.00		0.82±0.11
Pa#08	Intact	36	7.73±1.85	0.719	0.89±0.09
	Ruptured	36	7.58±1.61		0.67±0.09
Pa#09	Intact	36	11.02±2.05	<b>0.000</b>	0.81±0.06
	Ruptured	36	7.26±1.62		0.59±0.05
Pa#10	Intact	36	11.05±2.20	<b>0.000</b>	0.82±0.07
	Ruptured	36	9.20±2.02		0.69±0.11
Pa#11	Intact	36	11.15±2.03	0.409	0.81±0.07
	Ruptured	36	10.75±2.04		0.80±0.05



**Figure 5.** Comparison between intact and ruptured implants, per patient. The colour variation represents the tensile strength along the shell. I= Intact implant; R= Ruptured Implant; IT= Implantation Time and RT= Rupture type.

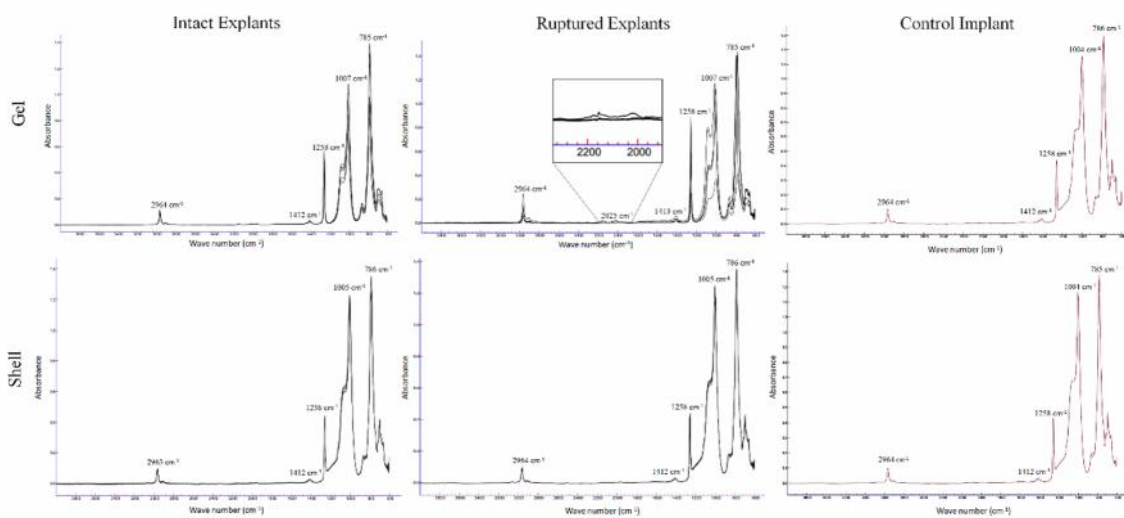
### 3.2. Chemical Characterization of Silicone Shells and Gels

FTIR spectroscopy was performed on silicone shells and gels of intact and ruptured explanted implants.

FTIR analysis revealed that all implants (intact and ruptured) were made of almost identical materials, since all spectra were similar (Figure 6). The peak at  $785\text{ cm}^{-1}$  reflects Si-C bonds ( $800\text{--}760\text{ cm}^{-1}$ ); at  $1007\text{ cm}^{-1}$  represents Si-O-Si bonds ( $1100\text{--}1000\text{ cm}^{-1}$ ); at  $1258\text{ cm}^{-1}$  corresponds to Si-CH<sub>3</sub> bonds ( $1280\text{--}1250\text{ cm}^{-1}$ ); at  $2964\text{ cm}^{-1}$  corresponds

to vibrations of the Si-OH bonds; at  $1420\text{ cm}^{-1}$  is correlated with asymmetric deformation vibration within the silicone compound (Figure 6).

The gel samples' analysis of ruptured implants showed a new peak, due to a mesoporous silica group ( $2025\text{-}2030\text{ cm}^{-1}$ ), detected at  $2025\text{ cm}^{-1}$  (see zoom box in Figure 6) [22]. This peak was found in the Pa#10 ruptured implant. The implant belonged to a patient with breast cancer and dyslipidemia (Pa#10).



**Figure 6.** FTIR spectra of gel and shell extracted from intact, ruptured and control implants.

## 4. Discussion

The present study analysed explanted implants from recent generation. The mechanical properties of ruptured and intact implants were compared in terms of brand, lot, implantation time, and demographic conditions (same patient, age, BMI, physical activity, surgery, and among others).

The appearance analysis of the intact and ruptured implants show different colours and aspects of shells and gels (Figures 1 and 2 and Table 2). These results agree with Necchi et al. [5].

The color of shell and gel for ruptured implants varied according to the type of rupture. Implants with abrupt ruptures, showed yellowing of shell and gel. Conversely, the gels and shells from the small ruptures were clear. The available literature agrees that PIP implants had a higher tendency for cholesterol absorption, which made them softer and more likely to present yellow discoloration than other implants [6, 20, 21, 23, 24]. Lower gel cohesiveness (broken and oily) was seen in ruptured implants. Previous studies suggested as a cause, the gel's reduced viscoelasticity [25], and the *in vivo* exposure of the silicone (leading to hydrolytic degradation and cross-link scission [6,26,27] According to Brandon et al. [16] and Bodin et al. [11] the broken gel and lower cohesiveness of the ruptured implants may increase the level of non-cross-linked silicone. This process may induce the shell progressive deterioration.

The present research found statistically significant differences of the tensile strength, between non-uniform and uniform gels of explanted implants (Table 3 and Figure 3). These findings along with other parameters must be taken in account to explain early shell ruptures. Regarding the explants, no direct correlation could be established between the gel and shell integrity and implantation time. This result is in contradiction with other publications [18], which reported a change of gel properties with implantation time.

The thickness of intact implants was (on average) higher compared to ruptured implants (Table 4 and 5). In the literature, shell thickness variations have already been identified as one of several factors that may affect the integrity of the shell [8,21]. In a previous work the authors have concluded that shell thickness may play a role in the process of implant failure, with thinner implants displaying lower tensile strengths [21].

For the analysis of intact and ruptured implants, the mechanical properties of shell samples were compared. There were significant differences in material resistance (tensile strength) (Tables 4 and 5).

These results point to a reduced ability of the ruptured implants (shells), to withstand mechanical stresses. This may be one of the possible causes of the failure observed. These results are consistent with the study made by Necchi et al. [5], and contradict the results reported by Brandon et al. [4,13,28]. Given the size and specificity of the analysed implant sample (brand, lot, shape, surface, etc.), it cannot be established a definite connection between the mechanical properties and rupture probability. However, implants with small damages (Tables 2 and 5) did not show significant differences of the tensile strength when compared with intact implant. These results may be related to fatigue phenomena, occasionally identified as the mechanism of implant rupture, reported by the authors on a previous study [29].

FTIR analysis revealed that all implants (explants and controls) were made of similar materials. All tested samples' spectra were very similar, and their spectroscopic profiles were found almost superimposable [30,31]. These findings suggest lack of chemical degradation during the implantation time. The peak ( $2025\text{ cm}^{-1}$ ) found in patient Pa#10 spectra, shows that the gel did absorb mesoporous silica compounds. These compounds are widely used as catalysts for drug delivery reagents and imaging [32,33]. This evidence agrees with the known clinical history of Pa#10, which includes oncologic pathologies (Table 1). Despite this evidence being found in a single implant, it highlights the possibility of bioaccumulation and tissue contamination of the implant materials (shell and gel). Further studies are needed to study chemical impact the mechanical properties, due to bioaccumulation [27,34].

## **5. Conclusions**

Three groups of implants were tested and compared to investigate the causes of breast implant failure. This study has unique characteristics, as it compares for the first time (as far as the authors' knowledge goes) rupture and intact implants from the same woman. These conditions guarantee that physical/biological variables are the same for each patient (pair of intact and ruptured implants). The results show that the gel cohesion must be involved in the long-term durability of implants, as well as the thickness of the shell material. Intact and ruptured implants have distinct mechanical behaviors, as significant differences in the tensile strength were observed. Despite each implant pair have endured the same environment (the patient body), for the same time, there were significant differences between them (intact vs ruptured). These differences (in integrity and mechanical behavior) may be linked with several factors. According with authors' understanding of the problem [21,29], these differences may be associated with the typical manufacturing process of breast implant shells. The manufacturing of implant shells uses a technique based on immersion of a positive mould in a liquefied Polydimethylsiloxane (PDMS) batch. The process is done manually leading to regional property differences over the implant shell. Regarding the eleven ruptured implants, resistance appears to be associated with implant thickness. The results stress the importance of a thorough control of the shell thickness. Given its relevance, shell thickness should be used as a control quality measure for homologation purposes.



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## **Declaration of Conflicting Interests**

None declared.

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## Article 6

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### ***In vitro* Degradation of Polydimethylsiloxanes for Breast Implant Applications Phenomena**

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## **Abstract**

The durability of breast implant material is associated to the failure probability, increasing with time since implantation. The current study avoids the bias introduced by biological factors, to systematically investigate the degradation over-time of shell materials. The fundamental physical/chemical conditions were maintained (temperature and pH), to decouple biological aspects from the degradation process.

Six virgin implants of two brands were submitted to: *in vitro* degradation process; mechanical testing of shell materials; surface change analysis (via Scanning electron microscopy - SEM); and chemical composition analysis by Fourier Transform Infrared Spectroscopy (FTIR).

FTIR results show that, the principal chemical bonds of the material remained intact after 12 weeks of degradation. Apparently implants shell's structure remained unchanged. Despite this observation, there were statistically significant differences between strain at failure at different time points for the shell of both brands, translated into a stiffening of the material overtime.

Material stiffening is reported as an indicator of material degradation. This altered mechanical behaviour added to the mechanical friction from tissue-tissue/tissue-implant and to the external mechanical loading (physical activity) may alter the material performance in women's body. Ultimately these changes may affect the implant durability. Further works are needed to understand the biological aspects of the degradation process and their impact on implants durability.

**Keywords:** Breast Implants, Degradation Process, Polydimethylsiloxanes (PDMS).



## 1. Introduction

The Polydimethylsiloxanes (PDMS) is the basis of both gel and shell of breast implants. The shell is produced from liquid components, and an amorphous “fumed” silica (SiO<sub>2</sub>) filler. SiO<sub>2</sub> is added to make high-performance silicone rubber for many purposes, including for the enhancement of mechanical properties [1-5]. The gel is a weakly crosslinked material which forms a three-dimensional polymer network. Crosslink occurs due to the reaction of vinyl groups present in the copolymer chains (dimethyl- and methylvinyl-siloxane) [3-5]. Increasing PDMS crosslinking degree, can lead to stronger and stiffer shell and gel materials [4]. The aging of implant material, usually associated among other factors to slow degradation, involves multiple physical and/or chemical processes.

A main risk of breast implant failure, is the material degradation during implantation time [1-3,6-9]. Several causes for degradation of the breast implant, were identified in the literature:

- additional crosslinking to embrittle the silicone [2];
- degradation or weakening of the Si-O-Si cross-links in the shells to Si-OH [9,10,11];
- degradation leading to the production of dimethyl siloxanes (e.g octamethylcyclotetrasiloxane (D4)) [2];
- swelling of the silicone elastomer shell by silicone fluid from the gel [2,7,12];

- silicone degradation produced by prolonged contact with lipids [3,8,13,14];
- degradation of the base polymer to a lower average molecular weight material [2].

Although biodegradation effects had been identified, *in vitro* biodegradability testing of PDMS breast implants are not well documented in the open literature. Most of the work done by breast implant producers for implant development, is confidential.

The purpose of this research was to characterize the breast implant degradation, under physical/chemical conditions of the human body (temperature and pH): a 'normal' physiology (pH = 7.4) and an inflammatory process (pH = 4.0). This study comprises the following stages: (I) *in vitro* degradation process; (II) analysis of the mechanical performance of the shell; (III) analysis of the surface (via Scanning electron microscopy - SEM); (IV) analysis of the chemical composition by Fourier Transform infrared spectroscopy (FTIR).

## 2. Material and Methods

Six virgin implants of two different brands were tested. All implants were round shaped, with textured surface and low profile. The degradation study was carried out taking into account,

- inter-brand comparison. Different brands compared at the same time point.
- intra-brand comparison. For each brand, the same implant lot was used for all time points.

## 2.1 Degradation test

Degradation process was conducted according with ISO 10993 "*Biological evaluation of medical devices*". To study *in vitro* degradation processes of the materials two buffer solutions were used: phosphate buffered saline with pH 7.4 (Ref. P4417 Sigma-Aldrich®), and Potassium hydrogen phthalate buffered (Ref. 82560 Sigma-Aldrich®) with pH 4.0. The degradation process was carried out during twelve weeks at controlled temperature, of 37°C. Every week a batch of samples was removed from the thermal bath, and samples weight were measured. Weight loss (%) was calculated using equation (1),

$$\% m \quad l_t = \frac{p_f - p_i}{p_i} * 100 \quad (1)$$

where  $p_i$  is the initial samples' weight and  $p_f$  their weight after different degradation stages (0-12 weeks).

## 2.2 Mechanical Test

The mechanical properties were evaluated at the end of each stage of degradation. These properties were obtained from uniaxial tensile testing data, through a prototype developed at INEGI Biomechanics Laboratory (Porto, PT). The experimental protocol follows the ISO standards for shell integrity (ISO 14607:2007) and determination of tensile stress-strain properties (ISO 37:2005). The equipment used, has alloy aluminium arms, connected to actuators and two load cells with 50N capacity. Before the tensile test, samples were subjected to a 0.25 preload, to guarantee a pre-testing controlled initial geometry. The samples were tested until failure, at a constant displacement rate of 20mm/min.

## **2.3 Morphological Characterization**

SEM analyses were carried out in a JEOL JSM 6301F/ Oxford INCA Energy 350/Gatan Alto 2500 microscope (Tokyo, Japan) at CEMUP (University of Porto, Portugal). This technique was used to analyse the morphology evolution due to degradation. Samples were analysed at four time points - 0, 4, 8 and 12 weeks.

Samples were coated with an Au/Pd thin film, by sputtering, using the SPI Module Sputter Coater equipment, for 120 s and with a 15 mA current.

## **2.4 Surface Characterization by Fourier Transform Infrared Spectroscopy (FTIR)**

The chemical composition of the surfaces were analysed using a Cary 630 FTIR Spectrometer (Agilent Technologies, USA) equipped with a diamond attenuated total reflectance (ATR) accessory. The tests were carried out at INEGI's tribology laboratory (CETRIB). Each spectrum was acquired over 20 scans with wavelengths ranging from 600 to 4000 $\text{cm}^{-1}$ , with a resolution of 4  $\text{cm}^{-1}$ . A background scan was performed before each sample measurement.

## **2.5 Statistical Analysis**

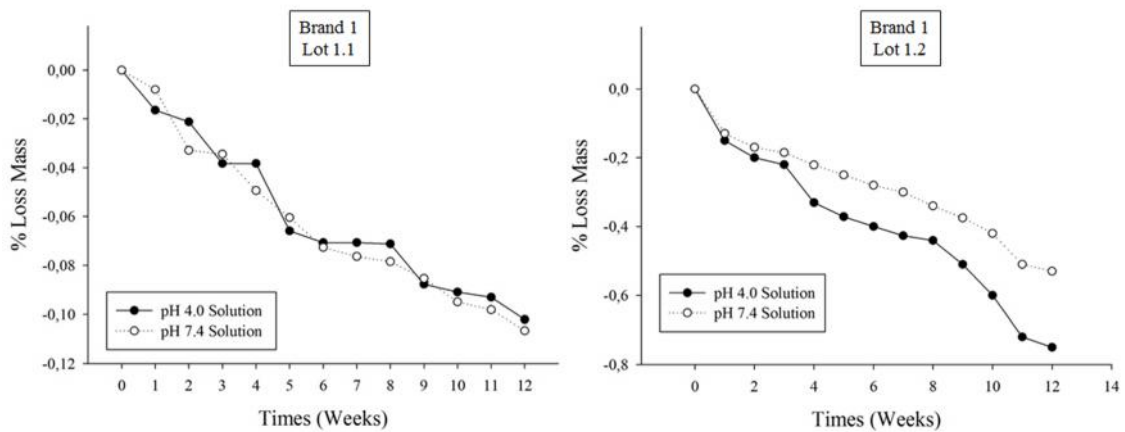
The statistical analysis was performed using IBM SPSS Statistics software version 20.0, with the significance level set at  $p < 0.05$ . Data normal distribution was verified with Kolmogorov- Smirnov and Shapiro-Wilk tests. The statistical differences in the mechanical properties among groups were assessed using independent-samples  $t$  test and

Manny-Whitney U test. The groups considered were Brand 1 and lot 1.1, Brand 1 and lot 1.2, and Brand 2. Each group was controlled at two time points (0 and 12 weeks), while subjected to *in vitro* aging using different buffer solutions - pH 7.4 and pH 4.0.

### 3. Results

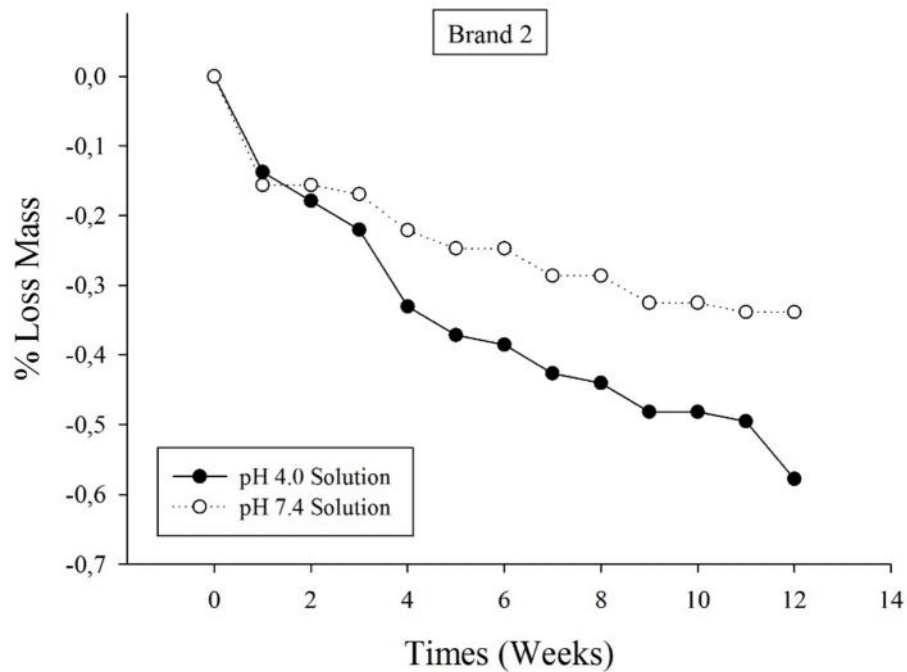
#### 3.1. Mass Loss During *In Vitro* Aging

This study includes 4 implants from brand 1: 2 of the implant from one lot (ref. lot 1.1) and more 2 implants from another lot (ref. lot 1.2). Lot 1.1, after 12 weeks of degradation, lost 0.10% mass soaked in pH 7.4 solution and 0.11% in pH 4.0 solution (Figure 1); lot 1.2 lost 0.52% mass, soaked in pH 7.4 solution and 0.75% in pH 4.0 solution (Figure 1). The initial pH of the buffer solutions did not change during the degradation period.



**Figure 1.** Experimental results of weight loss for Brand 1 during the degradation period under buffer solutions.

Brand 2 aged in a pH 7.4 solution from week 11 onwards, displays a mass loss rate of 0.34% which appears to be an asymptotic convergence (Figure 2). Implants soaked in pH 7.4 solution lost 0.34% in mass and lost 0.58% in pH 4.0 solution.



**Figure 2.** Experimental results of weight loss for Brand 2 during the degradation period under buffer solutions.

### 3.2 Mechanical Properties Analysis

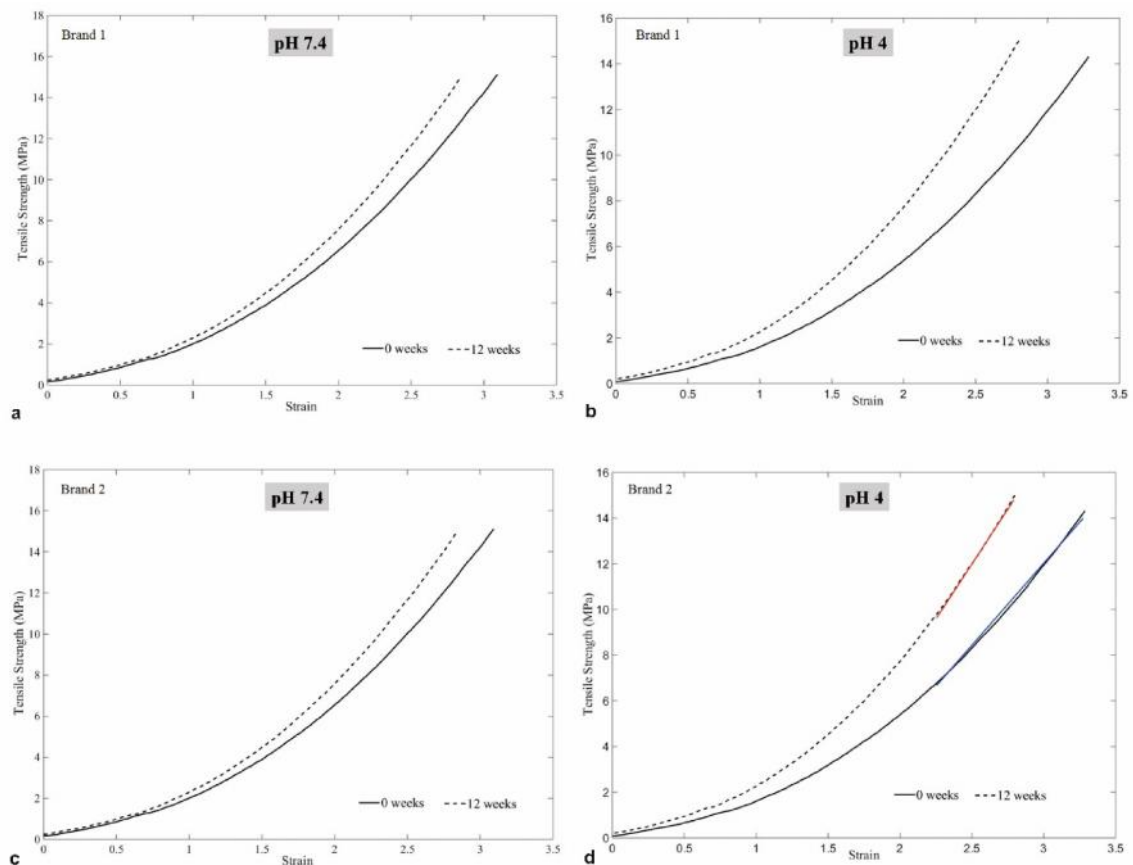
To analyse the material behaviour during different degradation stages (weeks), a total of 384 samples were tested under uniaxial loading. For comparison purposes a subsample corresponding to 0 (beginning) and 12 (end) weeks was considered after the degradation process. The mechanical properties of each tested implant and the statistical results are presented in Table 1.



**Table 1.** Statistical analysis of mechanical properties for breast implant samples in different stages of degradation (0 and 12 weeks). Bold type indicates significant differences ( $p < 0.05$ ).

Variable	Weeks	N	Tensile Strength (MPa)	$p^1$	Strain at failure	$p^2$		
Brand 1	pH 7.4	0	6	10.34 ± 1.39	0.496	3.06 ± 0.25	<b>0.004</b>	
		12	6	11.22 ± 1.87		2.50 ± 0.07		
	Lot 1.1	pH 4.0	0	6	11.68 ± 1.56	0.217	2.98 ± 0.01	0.575
			12	6	10.07 ± 1.5		2.76 ± 0.26	
	Lot 2.1	pH 7.4	0	9	12.28 ± 1.56	0.245	2.87 ± 0.34	0.171
			12	9	13.32 ± 2.08		2.85 ± 0.14	
		pH 4.0	0	9	12.43 ± 1.89	0.811	2.91 ± 0.38	<b>0.031</b>
			12	9	12.66 ± 2.04		2.77 ± 0.10	
Brand 2	Lot 2	pH 7.4	0	18	15.01 ± 1.15	0.164	3.08 ± 0.24	<b>0.003</b>
			12	18	15.58 ± 1.34		2.96 ± 0.04	
	pH 4.0	0	18	14.45 ± 0.91	0.211	3.11 ± 0.31	<b>0.030</b>	
		12	18	14.99 ± 1.52		2.88 ± 0.10		

There were no statistically significant differences in the tensile strength for every group analysed (Figure 3). However, the strain at failure results for Brand 1 in Lot 1.1 under pH 7.4 solution and for Lot 1.2 under pH 4.0 solution were significant different ( $p < 0.05$ ). The strain at failure of Brand 2 samples showed statistically significant differences ( $p < 0.05$ ) under both solutions. It is worth of notice that strain at failure decreased between 0 and 12 weeks (Table 1, and Figure 3).

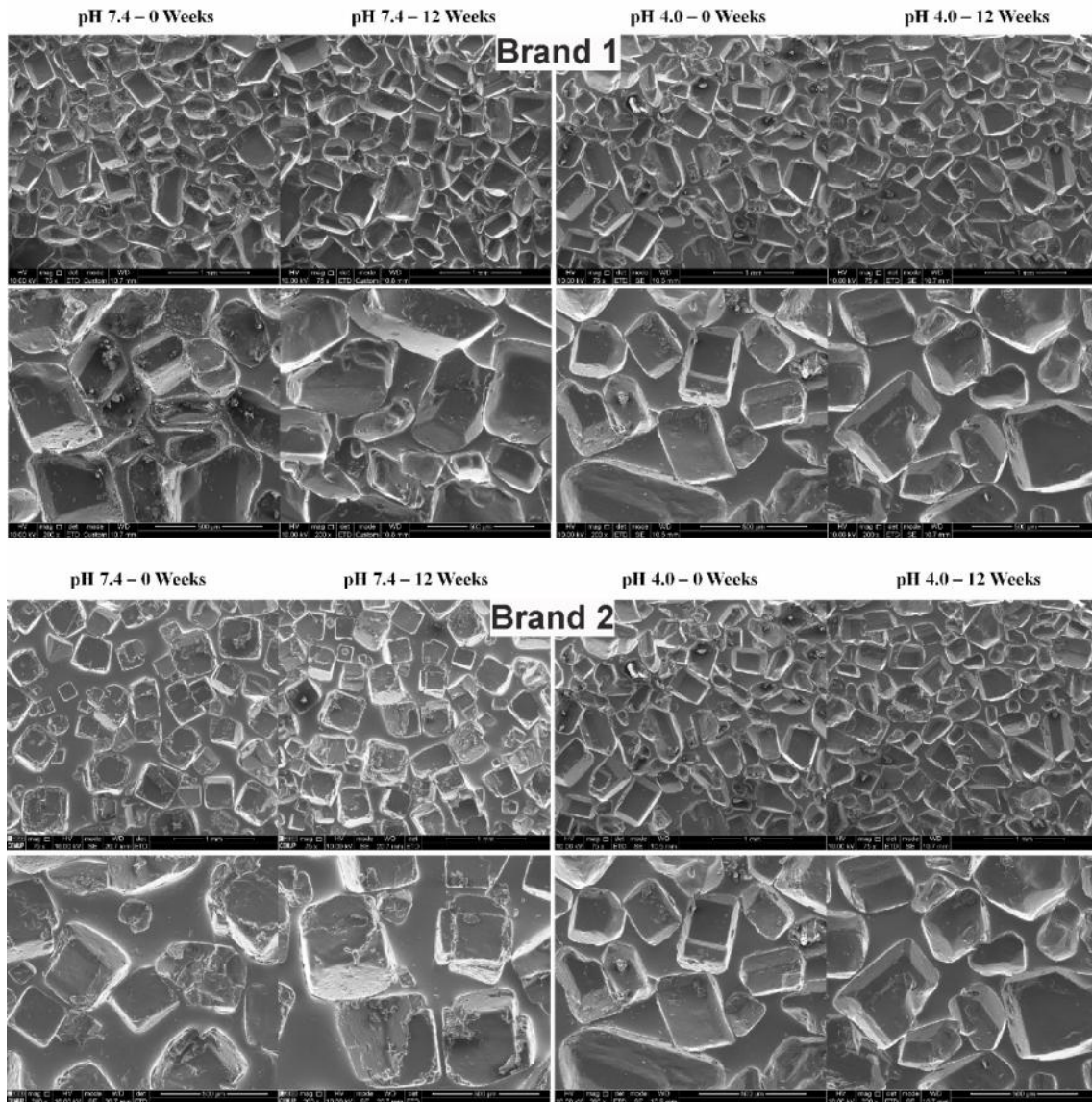


**Figure 3.** Example of tensile test results during the degradation in two buffer solutions: a) and b) for Brand 1 and lot 1.2; c) and d) for Brand 2. Blue and red lines are used to represent the stiffening of the shell.

### 3.3. SEM Analysis

The SEM was used to analyse the material surface. Before degradation (0 weeks), implants from the same lot showed similar surface morphology.

Brand 1 and Brand 2 showed similar surface morphology over different degradation stages for pH 7.4 and pH 4.0 buffer solutions (Figure 4). The textured outer surface of Brand 1 showed a morphology different than Brand 2 (Figure 4). Brand 1 reveals larger surface structures extending over several hundred micrometres, whereas Brand 2 structures are squarer and with larger gaps between each other.

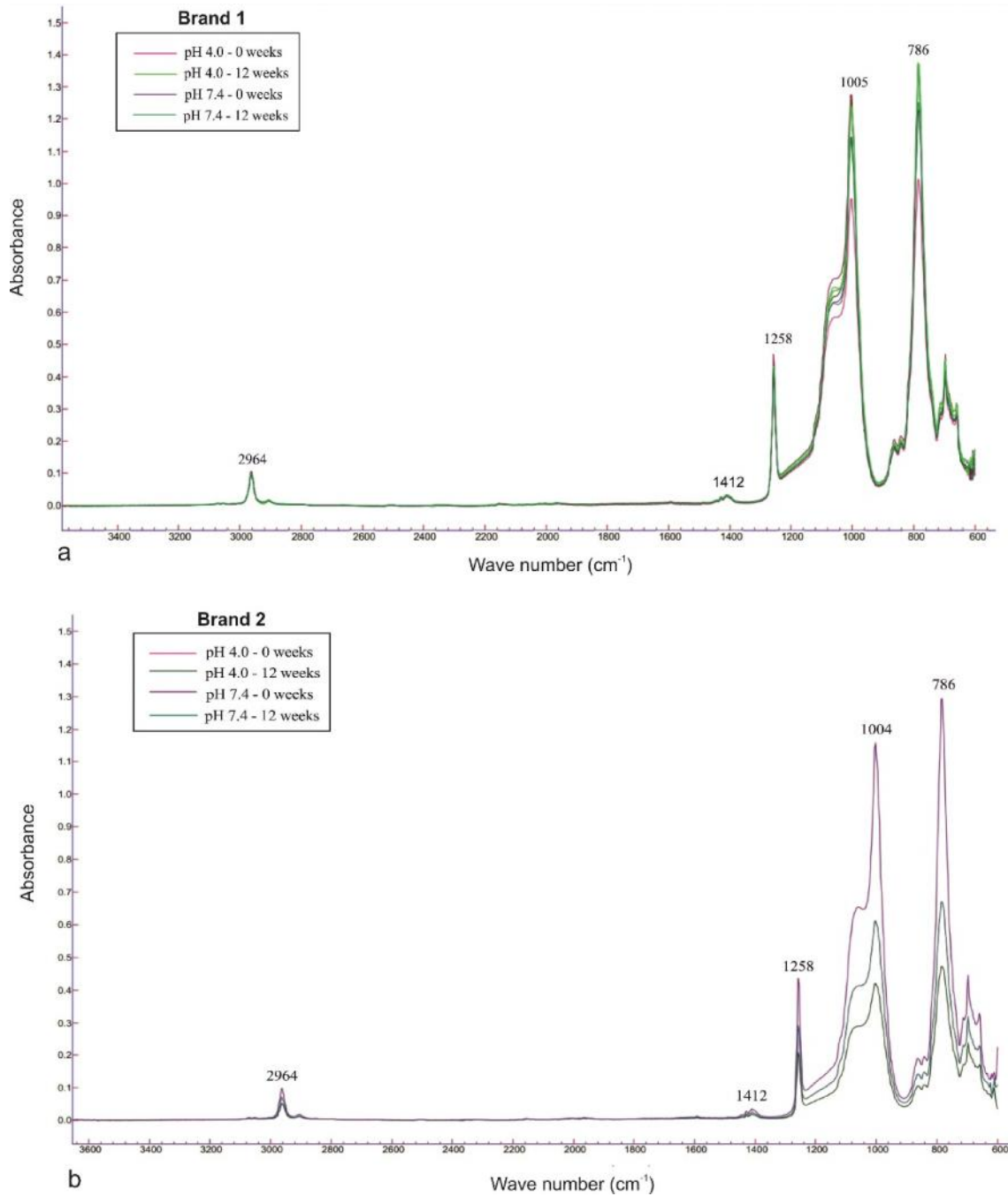


**Figure 4.** SEM micrographs of the outer surface of Brand 1 implants (x75 to lot 1.1 and x200 to lot 1.2), and Brand 2 implants (x75 and x200) over two time points (0 and 12 Weeks) under pH 7.4 and a pH 4.0 solutions.

### **3.4. ATR – FTIR Analysis**

Figure 5, show the FTIR spectra for all implants over different stages and solutions degradation of pH 7.4 and pH 4.0. All FTIR spectra showed that samples were of the same type of material similar to a PDMS spectrum as found in the literature [8]. There was no evidence of significant chemical structure modifications.

The wave numbers ( $\text{cm}^{-1}$ ) showed significant peaks (for material identification) at  $786\text{cm}^{-1}$ , corresponding to Si–C bond vibrations ( $800\text{--}760\text{ cm}^{-1}$ ). The peaks at  $1005\text{cm}^{-1}$  and  $1004\text{cm}^{-1}$  correspond to the stretching vibrations of Si–O–Si bonds, with a broad band in the region  $1100\text{--}1000\text{ cm}^{-1}$  (polymer backbone). The peak at  $1258\text{ cm}^{-1}$  is associated with Si–CH<sub>3</sub> bonds ( $1280\text{--}1250\text{ cm}^{-1}$ ), and the weak band at  $1412\text{ cm}^{-1}$  is correlated with asymmetric deformation vibration. A peak at  $2964\text{ cm}^{-1}$  corresponds to vibrations of the Si–OH bonds. The variation in the absorbance seen in Figure 5 is likely due to the differences in elasticity and thickness of samples.



**Figure 5.** FTIR spectra of Brand 1 (a) and Brand 2 (b) soaked in different solutions and different degradations stages.

## 4. Discussion

The durability and useful life of a breast implant continues to be a subject of intense interest and debate to both the patients and the plastic surgery community. To evaluate the potential impact of *in vivo* degradation on the mechanical properties of implant shells, an experimental protocol including uniaxial tension testing and *in vitro* degradation was carried from two implants brands. The observation that implants shells are sensitive to degradation in the body has been demonstrated for a long time on several studies [2,6,9,12,15-17]. Several authors showed a negative correlation between implant duration and mechanical resistance [1,2,6,7,9,16,17]. Furthermore, Yildirimer et al. [9] using FTIR, found evidence of degradation of the Si-O-Si cross-links to Si-OH on silicone shells, which may be related to inflammation.

However, Brandon et al. [12,18,19], Wolf et al. [20], and Swart et al. [21], showed that there was no time-dependent degradation in the shell tensile strength over years of implantation. These authors considered that implant failure is highly correlated with the implantation/explantation procedures, trauma of the breast, or manufacturing defects.

This assessment of the problem does not have into consideration the baseline of implant shell properties, since all previous studies were conducted on explanted implants (*in vivo*) [1,2,6,7,9,12,15-17,18-21]. The current study avoids the bias introduced by biological factors, to systematically investigate the degradation over-time of shell materials. The fundamental physical/chemical conditions were maintained (temperature and pH), in an effort to decouple biological aspects from the whole degradation process. The overall material degradation differs from a person to person, from tissue to tissue and over time for the same person [22]. In the present study found evidence of mass loss over

the degradation period (Figures 1 and 2); however the tensile strength of the shell material was not significantly affected as observed in Table 1, and Figure 3.

There was a statistically significant strain at failure reduction for the shell of both brands after a degradation period of 12 weeks (Table 1). This is due translated into a stiffening of the shell, as illustrated by the blue and red lines in Figure 3d. Material stiffening is reported as an indicative factor of material degradation [23]. Considering Brand 2 (Figure 3d) for a 2.5 of strain, the stress after 12 weeks of degradation is higher than initial stress.

SEM results revealed that the surface morphology did not show any differences after 12 weeks of degradation. Apparently implants shell's structure remained unchanged.

FTIR results (Figure 5) indicated the presence of the same type of PDMS for implant material in all samples, which agrees with literature [8,9,24-26]. No spectral deviations were observed during the degradation period, which suggests a lack of chemical degradation.

## **5. Conclusions**

In this study the authors attempted to decouple the biological aspects from the whole degradation process, by maintaining the human body physical/chemical conditions i.e., temperature and pH.

FTIR results show that, the principal chemical bonds of the material remained intact after 12 weeks of degradation. Despite this observation, there were statistically significant differences between the strain at different times points, which translated into a stiffening of the material overtime. This change may alter the mechanical friction tissue-tissue and/or tissue-implant, affections significantly the performance of the implant material in

the women's body when subjects to external mechanical loading such as physical activity and other. Ultimately these changes may affect the implant durability.

Further work is needed to understand the biological aspects of the degradation process especially under longer testing periods. In particular aspects such as oxidation (due to oxidants produced by tissues) and enzymatic degradation should be analysed. Larger degradation periods may also shed some light on fundamental degradation mechanisms such as weight loss, which was not fully understood with the current study.

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## **Declaration of Conflicting Interests**

None declare.



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# Chapter VI

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Integrated Discussion



## **Integrated Discussion**

In the last few decades, breast augmentation and reconstruction have evolved significantly due to the development of new materials and surgery technical improvement. Despite the improvements in implant design and manufacturing technologies throughout the years, the long-term reliability of implants and the phenomena involved in their failure are not completely clarified yet [1,2]. The same applies to plastic and oncoplastic surgical techniques for breast reconstruction. For instance, the stress-strain correlation between the implants and tissue, the implications the properties may have on final implant shape and the influence they may have on the implant rupture are not well understood by the plastic surgery community nor by breast implant manufacturers [3]. Therefore, researching the mechanical behaviour of both breast tissues and implant material is “vital” to understand the long-term results of the breast augmentation procedure.

A recent Poly Implant Prosthèse (PIP) breast implants scandal revived concerns about the mechanical stability of implanted silicones. As such, the main goal of this study focuses on explaining the seemingly higher rates of rupture for PIP breast implants compared to other breast implants brands, by presenting original studies to corroborate the importance of analysing explanted implants. The studies show that experimental evaluation and testing may contribute to characterize the causes, the physical process and the biomechanical process of implant failure. Hence, based on the main objectives and methodologies adopted, the most relevant findings of this research are discussed below.

Focusing on currently available data, there is no indication that the demographic profile of women who have had PIP breast implants differs from those with implants from other manufacturers [4]. In this sense, the silicone breast implants may fail, regardless of the manufacturer. Based on peer-reviewed published studies, the probability of rupture for PIP implants is estimated to be around 14.5% to 31 % after 5 to 10 years of implantation [5-10], while other silicone breast implants have been reported with a rupture rate of 1.1 to 11.6% after 10 years of implantation [5,11]. Data suggests that PIP implant rupture after 10 years is higher than that of modern generation implants, but comparable to that of older generation implants [12]. These studies showed that PIP breast implants produced during 2001-2010 presented a higher probability of rupture and earlier rupture than implants from other manufacturers [5,6,10,13-17]. Our findings are in the line with these conclusions, since the twenty-two studied explants had been implanted between 2005 and 2012. Furthermore, the explanted implants, especially the eleven considered with rupture, had a lifetime lower than 10 years (ruptured after 3 to 8 years of implantation). This temporal shift can be associated with structural integrity problems.

The visual appearance, related to cohesiveness and other qualitative parameters such as clarity and oiliness, is highly variable in PIP implants (intact, ruptured and controls) (**Articles 2, 4 and 5**). Thus, for PIP and Brand X controls most gels appeared to be firm (cohesive). However, for explanted implants, gels appeared relatively less firm (non-cohesive) and oily, regardless of the time of implantation. Additionally, lower gel cohesiveness (broken and oily) was seen in ruptured implants. The main reason for a lower cohesiveness of the gel is that *in vivo* exposure of silicone leads to hydrolytic degradation and cross-link scission [13,18]. Furthermore, according to TGA the "milk fluid" aspect witnessed in explanted PIP breast implants with ruptures was found to be predominantly an emulsion of water and silicone [19].



Another finding, discussed in **Articles 2, 4 and 5**, was the colour of both shell and gel for ruptured implants varied according to the type of rupture. There is unanimous agreement in the literature that the data evaluated for PIP implants contained a higher tendency for cholesterol absorption, which made it softer and more likely to present yellow discoloration than other implants [4,13,17,20,21]. However, in this study (**Article 2**) the analysis of explanted implants for Brand X to verify if they presented the same variation of cohesiveness and colour was not undertaken since no ruptured implants were available. In this context, and according to the literature, implants from other manufacturers can be clear, yellow, opaque or oily, but not non-cohesive [22], which may be considered as additional evidence of the variability between PIP implants.

The results of **Article 5** show that gel cohesion may be involved in the long-term durability of implants shells. According to Brandon et al [1] and Bodin et al. [23] the broken gel and lower cohesiveness of the ruptured implants may increase the level of non-cross-linked silicone that accentuates the mechanism of theorized shell diffusion and accentuate the process of shell progressive deterioration. More systematic and independent analyses should be performed on a larger sample of implants, to confirm the results obtained, for example through cohesivity testing.

To explore the ruptures in breast implants, we developed an experimental protocol for characterizing the entire implant shell. One of the key assumptions of all studies on the mechanical properties of silicone breast shells is that silicone properties are the same along the specimen under investigation, which may limit the conclusions. For instance, in **Articles 2 and 5** through the schematic representation of a colour map (Figures 5 and 7 in **Article 2**, and Figure 5 in **Article 5**) shows the variation of tensile strength over the shell. It is evident that the shell is inhomogeneous for each implant, and the mechanical properties vary from anterior, equatorial and posterior region. This variability is

associated with the typical manufacturing process of breast implant shells. The manufacturing process of implant shells uses a technique based on immersion of a mould in a liquefied PDMS batch. The process is done manually leading to regional property differences over the implant shell, due to differences in forming temperature, pressure, and other variables [24]. The available literature on implant rupture is supported by limited experimental data. The mechanical characterization, when available, is based on a small number of tests per implant, typically between 1 and 20 [13,19,21,25-28]. In our opinion, it is a small sampling to characterize the implant material behaviour along the shell. The approach shown in **Articles 2, 5 and 6** includes a far superior number of samples, as up to now a minimum of sixty samples per implant have been tested. This method enables a wide mapping of the mechanical properties of the implant shell material, as the location of each individual sample was known and, therefore, mapping of the mechanical properties per sample location was possible, as seen in **Articles 2, 5 and 6**.

Tensile testing provides a measure of the compliance of a particular sample with specified quality criteria, and the quality of the sample is considered to be reflective of the batch from which it is drawn. The mechanical tests performed along the different articles of this thesis present a variation in the tensile strength between explanted and controls implants. One of this variation is presented in **Article 5**, where a comparison between intact and ruptured implants showed a shell tensile strength of 7.42MPa for ruptured implants, compared with 9.59MPa for intact implants. These results point to a reduced ability of the ruptured implants (shells) to withstand mechanical stresses. It can be suggested that the resistance of the implants associated with other factors may be one of the possible causes of the failure observed.

Both this work and the literature presented shell thickness variations, which is one of several factors that may affect the integrity of the shell, as well as the variation in quality parameters from PIP implants [4,19,27]. Although the average thickness of all PIP samples (a total of 1216 samples – 396 from intact implants, 604 from ruptured implants and 216 from control implants) falls within the manufacturer's specifications (range from 0.5 and 1.0 mm), in some implant regions the minimum thickness was below 0.5mm and the maximum above 1mm. It was found that different regions of the shell had different thicknesses on nearly all the PIP implants. Brand X exhibited a smaller thickness variations among the regions, and this variation was in compliance with the manufacturer's specifications. In **Article 2** evidence that a thinner shell thickness for PIP implants was more likely to have a lower strength and a higher probability of failure is reported. **Article 5** showed that intact implants were thicker (0.91mm) when compared to ruptured ones (0.73mm), which suggest that thickness can be related to the possibility of failure. Furthermore, this suggests inconsistencies in the manufacturing process that could explain higher rates of rupture.

Microscopy techniques provided details of the ruptured shell region and could be used to determine the cause of breast implant failure, as seen in **Articles 3** and **4**. Results suggest that fatigue damage can be a potential cause of *in vivo* failure. All implant with small ruptures (hole and split) presented features commonly found in fatigue processes. Likewise, in **Article 6**, implants with small damages did not show significant differences of the tensile strength when compared with intact implant, which may be related to the fatigue phenomena found in **Article 3** and **4**. Shell failure may not be directly related to a decrease in mechanical strength but to fatigue effects. Implant failure may be associated with loading induced by normal daily activities, in addition to the presence of implant defects and microstructural inhomogeneities (e.g., inclusions and pores) that may

generate damage in the shell due to fatigue mechanisms. The features detected in explants were verified by fatigue testing performed in laboratory conditions, and should be viewed as a starting point for further work on fatigue test crack growth. The microscopy features suggest that fatigue phenomena should be taken into consideration by manufacturers when characterizing the mechanical behaviour of the shell, for homologation purposes.

Literature states that silicone exposure (from both shell and gel) to a strongly acidic environment and to bacterial contamination during the implantation procedure, results in hydrolytic degradation of cross-links and chain scission of the polymeric backbone [14,17,21,29-31]. Kaali et al. [30] showed that inflammation processes (i.e capsular contracture) coupled with bacterial contamination during implantation further accelerates silicone degradation. Scala et al. [31] demonstrated a temporal association between exposure to bodily fluids with or without inflammatory surroundings and silicone degradation. Yildirimer et al. [13] reported spectroscopic changes between  $3200\text{ cm}^{-1}$  and  $3600\text{ cm}^{-1}$ , and between  $1525\text{ cm}^{-1}$  and  $1760\text{ cm}^{-1}$ , attributed to Si-OH bonds from silicone degradation and to protein-like bands, respectively. A co-existing 'protein-like' spike in the spectroscopy suggests that this degradation may be due to the presence of a bio-film, which may be related to inflammation [13]. Furthermore, absorption bands seem to be associated with water and proteins from breast late periprosthetic fluid (LPF) infiltrated inside the implants [13,17,32]. Recently, Amoresano et al. [21] showed signs of carboxylic acid that can be ascribed to lipid infiltration.

In **Article 5** the shell and gel chemistry analysis for explanted implants by FTIR indicated the presence of the same type of PDMS for implant material in all samples. No spectral deviations were observed during implantation time, and we did not find significant differences between the intact and ruptured implant. These findings suggest a lack of chemical degradation during the implantation time. However, the new found peak

( $2025\text{cm}^{-1}$ , see Figure 6 in **Article 5**) shows that the gel is able to absorb mesoporous silica compounds, which may be related with medication taken by the patient at the time of implant rupture. This was confirmed with the collected clinical information, since the implant was explanted from a 68 year woman with breast cancer and dyslipidemia. Even if this peak was found in a single implant, it highlights the possibility of bioaccumulation and tissue contamination in the implant material (shell and gel). This opens the door to further studies, namely those focusing on the chemical impact to the mechanical properties, due to bioaccumulation.

Material aging may due to the contact with biological tissues may be associated to implant failures as reported by some authors [13,25,33-35]. In this context, it was decided to carry out an *in vitro* degradation analysis as detailed in **Article 6**. *In vitro* degradation tests concluded that chemical bonds of the material remained intact after 12 weeks of degradation, and the implants' shell structure (studied with SEM analysis) apparently remained unchanged. However, there was a stiffening of the material along that period for both PIP implant (described as Brand 1 in **Article 6**) and Brand X (described as Brand 2 in **Article 6**). Material stiffening is reported as an indicator of degradation: this mechanical behaviour change combined with mechanical friction from tissue-tissue/tissue-implant and with external mechanical loading (physical activity), may alter the material performance in women's bodies, and in consequence affect implant durability. Further work is needed to understand the biological aspects of the degradation process, especially under longer testing periods, as well as to emphasize the relevance of studies between the interaction of the tissue and the implant.

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# Chapter VII

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Conclusions



## Conclusions

The main purpose of this thesis was to contribute for the explanation of the seemingly higher rates of rupture for PIP breast implants compared to other breast implants. To achieve meaningful contributions on this domain, a multidisciplinary effort between engineers and medical doctors was required. After the thesis' conclusion, it can be assumed that the experimental methodology adopted proved to be an useful tool for the analysis of explanted implants. The physical and chemical processes in the methodology are able to reveal why breast implants fail. The most relevant conclusions of this research are summarized below:

The heterogeneous nature of PIP implants was confirmed in **Article 2**. This study demonstrated that the physical characteristics of the PIP implant are variable, and have a strong relationship with the shell thickness (thickness variations). By comparison, the Brand X can be a good example of quality control in breast implants, by not showing these variations. This may contribute to dispel fears among the medical community and patients about the reliability of breast implants.

**Articles 3** and **4** demonstrated that, with the proper background and experience in analysing ruptured breast implant shells, the features at the failure site can be correctly interpreted and the corresponding failure mechanisms can be diagnosed. Breast implant failure may be related to several factors, such as: implant handling before the surgical procedure, the implantation procedure, *in vivo* processes (e.g., abrasion or breast biopsy),

the explantation procedure, and *in vivo* cyclic loading that may induce fatigue damage in the implant. Analysing the microscopy images of the eleven ruptured implants, the study concluded that fatigue damage can be a potential cause of *in vivo* failure.

**Article 5** has unique characteristics because, to the author's best knowledge, it compares rupture and intact explanted implants from the same woman for the first time. These conditions guarantee that physical/biological variables are the same for each patient (pair of intact and ruptured implants). Although all implant pairs endured the same environment (the patient body), the results showed that there were statistically relevant differences between intact and ruptures implants. Ruptured implants were thinner (0.73 mm vs 0.91 mm) and weaker (7.42 MPa vs 9.59 MPa) than intact implants. The most important observation based on the present study is that the mechanical weakness of the shell has to be considered one of the main mechanisms of breast implant failure.

Finally, in **Article 6** the authors attempted to simulate the *in vivo* degradation, by decoupling the biological aspects from the whole degradation process. This was achieved by maintaining the human body physical/chemical conditions i.e., temperature and pH. The data analysis showed that the material stiffening may be an indicator of material degradation. However, despite this observation the results showed the need to improve the developed degradation process. Improving this process should allow for better understanding of the biological aspects of the degradation process (especially under longer testing periods) and their impact on implants' durability.

In conclusion, the thesis results point to several possible causes of rupture for the explanted implants, these may explain the high rupture rates of the PIP implants compared

to other brands. Regarding the eleven implants with rupture it can be concluded that the implants' resistance is associated with several factors, but the thickness variation and fatigue phenomena, were identified as the main reasons leading to failure. The findings suggest that fatigue phenomena and thickness should be taken into account by manufacturers when characterizing the mechanical behaviour of the shell, for homologation purposes. To better understand this type of failure and its relevance among other implant rupture mechanisms further research is required. This is the kind of information that will potentiate safer and longer lasting products.

In conclusion, this thesis should not be viewed as a finalized work, but rather as the beginning of several research avenues.





# Chapter VIII

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Limitations and Recommendations for Future  
Works



## **Limitations and Recommendations for Future Works**

During the design of the study, we tried to obtain an equal number of implants for each of the considered groups, which was not achieved. The main reason for the lack of implant parity in the PIP and Brand X control groups was virgin implant availability. From all the partners in the study, only Brand X gave the proper consent to study their implants, both in small numbers and with anonymity. PIP implants were obtained through Portuguese regulatory authorities (INFARMED), and we accepted the number and type of implants gently provided. Due to these constraints, to properly complete the work it would be necessary to have larger samples of Brand X or other brands currently used.

Another important point was the lack of explanted Brand X implants, which was due to the fact that no explants from Brand X were available during the implant collection period (over a year). The relative abundance of PIP explants over the collection period is mainly explained by the concerns expressed over the potential health issues linked with PIP implants, which is not the case for Brand X. Thus, inter-brand explant comparison is impossible at this stage, and will be a topic for future studies. Additionally, to strengthen the present work, the acquisition of a larger number of explanted implants (e.g. PIP) related with demographic information and surgical procedures would be needed to improve the understanding of failure mechanisms. Likewise, the description of implant position in the chest cavity and the rupture spot in the implant by surgeons would be important to compare with the data in **Articles 2, 3, 4 and 6**. This approach could be used to verify if the position and contact with the breast tissue influences rupture, and if the

mechanical behaviour of tissues and internal forces during static postures and dynamic activities have any direct effects on the rupture.

Due to differences in cohesiveness between explanted implants and controls, cohesivity testing (ISO 14607- Annex D) shall be performed to measure both the rheological properties and the integrity of the gel.

In **Articles 3** and **4**, the features identified in some ruptured explanted implants and in specimens (control) tested under laboratory controlled conditions indicate that fatigue phenomena can be the cause of some ruptures. Since these findings have not been reported in the literature, to the best of the author's knowledge, this is a matter for future studies. More fatigue crack growth tests performed in laboratory are required to better understand this type of failure.

According to the *in vitro* degradation study performed (**Article 5**), further work is needed to understand the biological aspects of the degradation process, especially for longer testing periods. In particular, aspects such as oxidation (due to oxidants produced by tissues) and enzymatic degradation should be analysed. Larger degradation periods may also shed some light on fundamental degradation mechanisms such as weight loss and material stiffening, which was not fully understood in **Article 5**.

Considering our experience in the material behaviour of PIP breast implants, allied with the sensitivity of the group to study soft tissues, a study of the soft tissue pocket as well as the implant could be performed to better understand the relation between both mechanical behaviours, as there certainly is an interdependence/interaction (mechanical, chemical, etc.) between implant and capsule. Phenomena such as capsule contracture would be better understood if both structures were studied.

Likewise, it would be important to study the different breast tissues. Despite many studies that characterize the mechanical behaviour of breast tissue, most studies lack a

multivariate analysis. Because of the shortcomings referenced in the **review Article 1** further research is needed to:

- (1) integrate the etiological factors influencing the biomechanical proprieties of breast tissues, such as age, body mass index or hormonal status (menopause);
- (2) characterize all tissues, including the suspensory cooper's ligaments;
- (3) build experimental set-ups that includes *in vivo* and *ex vivo* testing in order to validate the results;
- (4) standardize the experimental protocol, in order to analyse samples from the same breast location;
- (5) control the amount of pre-load compression (for instance, test two levels of pre strain, a proper and a higher level used in clinical breast examination).

After this approach, it is important to relate the mechanical interaction between implants and breast tissue, as for instance, a mechanical knowledge of the factors affecting breast deformation and shape along time (e.g., the properties of skin) is necessary.



