

# **Controlled release of chlorhexidine from a HEMA-UDMA resin using a magnetic field**

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## **Abstract**

**Objectives:** To functionalise novel chlorhexidine (CHX) particles with iron oxide ( $\text{Fe}_3\text{O}_4$ ) nanoparticles and control their release kinetics in a dental resin using an external magnetic field.

**Methods:**  $\text{Fe}_3\text{O}_4$  nanoparticles were synthesized and incorporated into spherical CHX particles and the powder was freeze dried. Resin disc specimens were produced using a UDMA-HEMA resin mixed with freeze dried spherical  $\text{Fe}_3\text{O}_4$ -CHX particles (5 wt. %), which were placed into a Teflon mould (10 mm diameter  $\times$  1 mm depth) and covered with a Mylar strip. A MACS magnet was left in contact for 0 mins (Group 1), 5 mins (Group 2) or 10 mins (Group 3) and the resin discs subsequently light cured (Bluedent LED pen, Bulgaria) for 60 s per side. The resin discs were immersed in deionised water at various time points up to 650 h. UV-Vis absorbance was used to determine the CHX content. CHX released for each time point was determined. The functionalized CHX particles and resin discs were characterized using TEM, TGA, EDX and SEM.

**Results:**  $\text{Fe}_3\text{O}_4$  nanoparticles (20 nm) incorporated into the spherical CHX particles, led to a mean (SD) particle size reduction from 17.15 (1.99)  $\mu\text{m}$  to 10.39 (2.61)  $\mu\text{m}$ . The presence of  $\text{Fe}_3\text{O}_4$  nanoparticles in the spherical CHX particles was confirmed with SEM, EDX, and TGA. SEM of group 1 resin discs (no magnetic exposure) showed functionalized CHX spheres were homogeneously distributed within the resin discs. For resin discs which had magnetic exposure (5 or 10 mins) the particles started to cluster nearer the surface (Group 2: 43.7%, Group 3: 57.3%), to a depth of 94  $\mu\text{m}$ . UV-Vis absorbance revealed Group 1 resin discs had a cumulative CHX release of 4.4% compared to 5.9 % for group 2 and 7.4% for group 3 resin discs, which had magnetic exposure (5, 10 mins).

## **1. Introduction**

Antibacterial agents and their delivery are of great importance in medicine and dentistry, since a wide range of bacterial infections are still the major reasons for recurrent/ persistent infections despite the use of antibiotics [1, 2]. Biomaterial implants used in modern medicine for functional restorations are also susceptible to infections, which can lead to failure [3]. Increased application of implants in dentistry has led to significant numbers of patients developing peri-implantitis (47.1% prevalence with mean functional loading of 10 years) [4]. These painful infections are caused by anaerobic bacteria and lead to bone loss, exposure of titanium implant threads and ultimately implant removal. Painful surgical treatment, smoothing of the implant [5] and cleaning together with antibiotic therapy can be carried out, but many of these treatment strategies have poor outcomes [6]. The site of bacterial infections can also be largely inaccessible to antimicrobial agents used in the oral cavity, especially in periodontal pockets and proximal/marginal areas of composite restorations, which are susceptible to bacterial micro-leakage. This can lead to the establishment of bacterial biofilms causing secondary caries, inflammation and degradation of the polymer composite [7]. Bacteria may become resistant if the antibacterial agents cannot penetrate the biofilm and if sufficient drug is not available at the infection site [8, 9]. Therefore efficient delivery and penetration of the antibacterial agent to the exact site of infection is highly desirable. In some infection, enhanced antimicrobial drug delivery might be achieved by using a non-invasive external magnetic force to improve drug navigation to the infection site. Magnetic field navigated drug delivery is based on the use of magnetic nanoparticles such as magnetite, strontium ferrite, manganese ferrite and others [10] embedded into different nano-/micro- carriers or used directly, and various therapeutic agents have been loaded for magnetic targeted delivery [11-14]. Recent studies have demonstrated that by using a magnetic field superparamagnetic iron oxide nanoparticles are able to target infection sites, inhibit several bacterial functions and penetrate biofilms; thereby overcoming

the therapeutic barrier often encountered when using traditional antibiotics or other antibacterial agents [3, 9]. In dentistry, magnetic nanoparticles have been incorporated into polymeric scaffolds or cement composites to enhance cell adhesion and osteogenic differentiation [15-17] and have been navigated inside dental tubules via an external magnetic field for treating dental hypersensitivity [18]. Magnetic nanoparticles are also combined with other components to enhance their antibacterial performance. Mahmoudi et al., reported core-shell nanoparticles consisting of a superparamagnetic core and silver shell, which showed enhanced antimicrobial activities and excellent penetration of biofilms when an external magnetic field was applied [19].

CHX is a bis-biguanide antiseptic and disinfectant extensively used in medicine and dentistry [20, 21]. The current authors previously developed a novel formulation of CHX with a controllable crystal size [22] and release behavior when incorporated into a dental resin [23]. Functionalisation of the novel CHX particles with gold nano-rods resulted in the ability to produce a near-infrared light (NIR) responsive CHX release [24]. With the ability to move CHX to an infection site using an external magnetic field, the combination of magnetic nanoparticles with CHX crystals could extend the range of antibacterial applications and improve efficacy. Therefore we aim to functionalise the novel CHX particles with iron oxide nanoparticles ( $\text{Fe}_3\text{O}_4$ ) and investigate the possibility of moving the resulting CHX composite in a dental resin with an external magnetic field. There are many clinical situations where a targeted drug release would be desirable in restorative dentistry, periodontology and medicine. Magnetic responsive CHX formulations may be particularly useful in developing magnetic targeted antibacterial materials.

## **2. Materials and Methods**

### **2.1 Magnetic nanoparticles synthesis**

Iron oxide nanoparticles ( $\text{Fe}_3\text{O}_4$ ) were synthesized according to a well-established Massart's co-precipitation method [25]. Briefly, 2.35g  $\text{FeCl}_3$  (Fluka, 44944, Lot: 30607125) and 0.86g  $\text{FeCl}_2$  (Fluka, 44939, Lot: 24606139) were added in 40 ml  $\text{H}_2\text{O}$  in a three-neck flask, and then placed in an oil bath and heated up to 80 °C in an argon atmosphere. The mixture was next stirred using a magnetic stirrer (VWR Stirrer, USA), at a rate of 800 rpm, whilst 5 ml  $\text{NH}_4\text{OH}$  (Sigma-Aldrich, UK, lot: 320145) was added slowly with a syringe. Heating was maintained at 80 °C for 30 mins and then 2 ml of 0.5g/ml citric acid (Sigma-Aldrich, UK, 27490, Lot: 23405C03) was introduced. The temperature was next raised to 95 °C and held for 90 mins. The product was cooled and dialysed against  $\text{H}_2\text{O}$  in a 14 kDa cut-off membrane (Sigma-Aldrich, UK, D9527) for one week. The  $\text{Fe}_3\text{O}_4$  nanoparticles were then characterized using transmission electron microscopy (TEM) (JEOL-JEM 2010, USA) at acceleration voltage of 200 kV.

### **2.2 $\text{Fe}_3\text{O}_4$ -Chlorhexidine sphere synthesis**

Spherical CHX particles (SCPs) were functionalized with  $\text{Fe}_3\text{O}_4$  nanoparticles and prepared by mixing 200  $\mu\text{l}$  of  $\text{Fe}_3\text{O}_4$  nanoparticle suspension with 1 ml of 0.33M  $\text{CaCl}_2$  (Sigma-Aldrich, UK, C8106, Lot: SLBF7416V). The mixture was added to 1 ml of 15 mg/ml CHX-diacetate solution (Sigma-Aldrich, UK, C6143, Lot: 19H0417). The mixtures were shaken for 1 min, and then centrifuged at 2000 rpm for 1 min (Eppendorf centrifuge, 5417C, Germany). To reduce the dissolution of the particles, the precipitates were washed three times with 0.33M  $\text{CaCl}_2$  solution. All the supernatants were collected for UV-Vis absorbance (Wavelength = 254 nm). The functionalized CHX spheres ( $\text{CHX}/\text{Fe}_3\text{O}_4$ ) were freeze dried (ScanVac Cool Safe Freeze Drying, Denmark) at -107 °C, 0.009 mBar for 1 day. The proportion of CHX in the

powders was calculated using both UV-Vis (Lambda 35, Perkin Elmer, USA) and thermogravimetric analysis (TGA Q50) as described in our previous work [23]. Control SCPs (without Fe<sub>3</sub>O<sub>4</sub> nanoparticles) were synthesized by co-precipitation of CaCl<sub>2</sub> and CHX-diacetate, as described in our previous study [26].

### **2.3 Preparation of functionalized chlorhexidine UDMA- HEMA resin discs**

UDMA-HEMA resin was prepared by mixing 64% urethane dimethacrylate (UDMA) (Esschem, UK, Lot: 591-22), 36% hydroxyethyl methacrylate (HEMA) (Aldrich, UK), 0.08% of N, N-dimethyl-P-toluidine (Acros Organics, UK) and 0.05% dimethylamino ethyl methacrylate (Aldrich, UK). The mixture was stirred at 800 rpm for 15 min (VWR Stirrer, USA). Finally, camphorquinone (Sigma-Aldrich, UK) was added at the proportion of 0.1% and the mixture was stirred for another 15 min. The viscous liquid resin was next mixed for 1 min with the freeze dried CHX/Fe<sub>3</sub>O<sub>4</sub> spheres (5 wt.% CHX content) using a Rotomix (120V/60Hz, 2850 rotations/min) (ESPE RotoMix, USA).

The resin mixture containing CHX/Fe<sub>3</sub>O<sub>4</sub> spheres was placed into a Teflon mould (10 mm in diameter × 1 mm depth), left for 10 mins and then cured through a Mylar film with a curing light (Bluedent LED pen, Bulgaria) (430-490nm, 600 mW/sq.cm) for 60 s on both sides (Group 1). To control the distribution of CHX/Fe<sub>3</sub>O<sub>4</sub> particles in the resin mixture, it was again placed into the Teflon mould and covered with a Mylar film. A MACS magnet (1.5 cm diameter, 0.5 cm depth, Miltenyi Biotech, UK) was left in contact for 5 mins (Group 2) or 10 mins (Group 3) and then the resin discs were light cured for 60 s per side. All the discs were next weighed on a microbalance (Salter Ander-180A weighing scale, UK) and the amount of CHX in each disc was calculated.

## **2.4 UV/VIS Spectroscopy**

The release kinetics of CHX from the resin discs was measured using UV-Vis absorption (Lambda 35, Perkin Elmer, USA). All the UDMA-HEMA resin discs containing CHX/Fe<sub>3</sub>O<sub>4</sub> spheres were kept in cuvettes containing 2 ml deionized water at room temperature. Specimen Groups 1, 2 and 3 (n=3 per group) were tested at time points of 1h, 3h, 5h, 15h, 25h, 40h, 65h, 95h, 140h, 205h, 275h, 350h, 500h and 650h. Solutions from each time interval were collected for the UV-vis absorbance test and replaced with fresh deionised H<sub>2</sub>O. The released CHX for each time point was determined according to an established calibration curve [23]. Residual CHX in all the supernatants was determined by measuring the UV absorption of the supernatants at 254 nm. Cumulative release for each of the groups was plotted and compared.

## **2.5 Scanning electron microscopy (SEM) characterization**

The spherical CHX particles, CHX/Fe<sub>3</sub>O<sub>4</sub> spheres and resin discs containing CHX/Fe<sub>3</sub>O<sub>4</sub> spheres (Groups 1-3) were characterised using SEM. The resin discs containing CHX/Fe<sub>3</sub>O<sub>4</sub> spheres were immersed in liquid nitrogen and fractured to analyse the distribution of spheres. All the samples were gold coated for 45s at 18 mA, 0.04 mBar using a sputter coater (SC7620, Emitech, UK) and viewed using a scanning electron microscope (FEI Inspect-F, USA), in the secondary electron imaging mode, with an accelerating voltage of 10 kv and spot size of 3.5. To identify the presence of Fe in the CHX/Fe<sub>3</sub>O<sub>4</sub> spheres, energy dispersive X-ray spectroscopy (EDX), (INCA, Oxford Instruments, High Wycombe, UK) was used with an accelerating voltage of 20 kV, working distance of 10 mm and spot size of 5. Back scattered SEM images were also used to illustrate the distribution of Fe in the CHX/Fe<sub>3</sub>O<sub>4</sub> spheres.

To determine the influence of Fe<sub>3</sub>O<sub>4</sub> nanoparticles on the size of spherical CHX particles, the Mean (SD) particle diameter of the spherical CHX particles and functionalized CHX particles

(CHX/Fe<sub>3</sub>O<sub>4</sub>) were measured using quantitative image analysis software (Nano Measurer, version 1.2) of the SEM images.

Resin discs containing CHX/Fe<sub>3</sub>O<sub>4</sub> spheres with 0 mins (Group 1), 5 mins (Group 2) or 10 mins (Group 3) magnetic exposure were characterized using SEM using the above protocol. To acquire a panoramic image, cross-sectional SEM images were taken continuously from one side of the resin disc to the other side and then all the individual images were assembled to produce a panoramic image. To analyze the distribution of CHX/Fe<sub>3</sub>O<sub>4</sub> spheres in the resin discs as a function of distance to the magnet, the panoramic images for each group (n=3 per group) were divided into 21 frames. Each frame was presented along the x-axis in 47 μm increments. The number of CHX/Fe<sub>3</sub>O<sub>4</sub> spheres in each of the frames was counted and summed to get the total number. The CHX/Fe<sub>3</sub>O<sub>4</sub> sphere distribution in the resin discs was plotted as a function of frame distance to the magnet side, with each point representing the percentage of CHX particles in each frame accordingly. The border of the frame next to the magnet was set as zero.

In order to assess the influence of magnetism on particle distribution the number of CHX/Fe<sub>3</sub>O<sub>4</sub> particles in each frame (n=3 in each of the three test groups) were statistically compared using a one way ANOVA (p<0.05, Tukey test, Sigma stat, version 2.03, SPSS Inc.).



## 2.6 Cytotoxicity Assay

The Cytotoxicity of CHX/Fe<sub>3</sub>O<sub>4</sub> spheres were evaluated with a standard MTT (3-[4,5 dimethylthiazol-2-yl]-2, 5 diphenyl tetrazolium bromide) assay using L929 fibroblast cell line (ECACC 85011425, UK). The MTT assay was carried out according to the protocol outlined in ISO 10993-5:2009 (Biological evaluation of medical devices - Part 5: Tests for in vitro cytotoxicity). Cells were cultured in Dulbecco's Modified Eagle's Medium (DMEM; Lonza, UK) supplemented with 10% fetal bovine serum (FBS) with 100 IU/mL penicillin, 100 µg/ml streptomycin, and 2 mmol/L glutamine (all from Invitrogen, UK) in a humidified incubator in 10% CO<sub>2</sub> in air at 37°C seeded in flat bottom 96-well microtiter plates (Sarstedt, Germany) at 10<sup>4</sup> cells per well with a final volume of 225µl and incubated for 22 hours. Following incubation, a series of 2 fold dilutions (0.000625 % to 0.08 %) of CHX/Fe<sub>3</sub>O<sub>4</sub>, commercial chlorhexidine diacetate (CHX) and pure Fe<sub>3</sub>O<sub>4</sub> nanoparticles in deionised water were prepared. 25µl of each CHX/Fe<sub>3</sub>O<sub>4</sub>, CHX and pure Fe<sub>3</sub>O<sub>4</sub> nanoparticle concentration was added into respective 96-well flat-bottomed microtiter plates to achieve a final CHX/Fe<sub>3</sub>O<sub>4</sub>, CHX and pure Fe<sub>3</sub>O<sub>4</sub> nanoparticle concentration range of 0.0000625 % to 0.008 %. 25µl of sterile deionized water were added into the control wells (cells only and medium only) and plates were incubated for 24 hours. Culture medium containing the treatments was then removed and 50 µL of 1 mg/mL tetrazolium salt MTT (Sigma-Aldrich, Gillingham, UK) was added to each well and incubated in 37 °C for 2 hours. Formazan crystals generated by mitochondrial enzyme activity were then dissolved by 100 µl of isopropanol and the intensity of the purple coloured reaction product was quantified by measuring the absorbance spectra with a plate reader at 570 nm. The absorbance of untreated cells was considered to be 100%. Relative cell viability was calculated as: (absorbance of treated cells/absorbance of untreated cells) x 100%. Finally, an inverted microscope (Nikon Eclipse TE2000-S, UK) was used to image the morphology of treated and untreated cells.

## 2.7 Antimicrobial Assay

Oral pathogenic bacterium, *Porphyromonas gingivalis* (strain W50) was used to evaluate the antibacterial activity of CHX/Fe<sub>3</sub>O<sub>4</sub>. *P. gingivalis* was grown on blood agar (Blood Agar Base No. 2; Oxoid, UK) plates in an anaerobic atmosphere (10% H<sub>2</sub>, 10% CO<sub>2</sub>, and 80% N<sub>2</sub>) at 37°C for 48 hours. The bacterial culture was harvested and suspended in brain heart infusion (BHI) broth (CM1135; Oxoid, UK) supplemented with 5µg/ml haemin and 5µg/ml menadione bisulphite. The bacterial culture was then grown for 24 hours anaerobically after which it was adjusted to an initial optical density (OD) of 0.1 at 600 nm (OD<sub>600nm</sub>) with BHI broth. A series of 2 fold dilutions (from 0.08 % to 0.000625 %) of CHX/Fe<sub>3</sub>O<sub>4</sub> in deionised water were prepared. 225µl of diluted bacteria and 25µl of each CHX/Fe<sub>3</sub>O<sub>4</sub> concentration was mixed into 96-well flat-bottomed microtiter plates to achieve a final CHX/Fe<sub>3</sub>O<sub>4</sub> concentration range of 0.008 % to 0.0000625 %. For control wells (bacterial suspension only and BHI only), 25µl of sterile deionized water were added. The plates were incubated for 24 hours anaerobically and OD was determined at 595 nm (OD<sub>595nm</sub>) to quantify bacterial growth. The lowest CHX/Fe<sub>3</sub>O<sub>4</sub> concentration with no detectable bacterial growth was recorded as the minimum inhibitory concentration (MIC). Additionally, MTT assay was carried out as described above to determine the viability of treated bacteria. Briefly, bacterial culture medium containing different CHX/Fe<sub>3</sub>O<sub>4</sub> concentrations were removed, 50 µL of 1 mg/mL tetrazolium salt MTT (Sigma-Aldrich, Gillingham, UK) was added to each well and incubated anaerobically for 2 hours. Formazan crystals were then dissolved with 100 µl of isopropanol and absorbance was measured at 570 nm. Bacterial viability was calculated as: absorbance of treated bacteria/absorbance of untreated bacteria x 100%.

### 3. Results

#### 3.1 Results of the TEM study

TEM image of synthesized  $\text{Fe}_3\text{O}_4$  nanoparticles is showed in Figure 1. It can be seen that the mean diameter (SD) of the nanoparticles is 13.1 (2.4) nm and they appeared in clusters.

#### 3.2 Results of the SEM study

SEM photomicrographs of the SCP and CHX/ $\text{Fe}_3\text{O}_4$  spheres is shown in Figure 2a, e and 2b, f. Spherical CHX particles were produced by co-precipitation of chlorhexidine diacetate and  $\text{CaCl}_2$ . Both samples showed a porous structure and particles were monodispersed (Figs. 2a-f). The mean (SD) diameter for SCP was 17.15 (1.99)  $\mu\text{m}$ , whilst the CHX/ $\text{Fe}_3\text{O}_4$  spheres had a mean (SD) diameter of 10.39 (2.61)  $\mu\text{m}$ . SEM photomicrographs at high magnification ( $\times 40,000$ ) indicated a dendritic structure for both particles (Figs. 2c-d), but  $\text{Fe}_3\text{O}_4$  nanoparticles appeared dispersed within the structure for the functionalized particle (Fig. 2d). SEM images in the back scattered mode illustrated the presence of Fe in the CHX/ $\text{Fe}_3\text{O}_4$  spheres (Fig. 2f), which was not present in the SCP (Fig. 2e) and peaks for Fe were clearly identified using EDX (Fig. 3).

The cross-sectional SEM images of resin discs (Groups 1-3) assembled to make panoramic images are presented in Figures 5a, b, c. Frames (indicated by white lines, first 3 numbered) are presented in 47  $\mu\text{m}$  increments and the position of the magnet (M) is marked to indicate the magnet side. For the Group 1 resin discs with no magnetic exposure (Fig. 5a), the CHX/ $\text{Fe}_3\text{O}_4$  spheres were homogeneously distributed throughout the resin discs. The Group 2 and 3 specimens (5 and 10 mins magnetic exposure) resulted in the clustering of many of the CHX/ $\text{Fe}_3\text{O}_4$  spheres at the magnet end (Figs. 5b, c). According to the numbers of magnetic CHX/ $\text{Fe}_3\text{O}_4$  spheres counted in each of the frames, their distribution was plotted as a function of distance from the magnet (Figs. 6 a, b, c). To gauge the effects of the magnetic field the

percentage of particles in the first three frames (nearest the magnet) and in the disc with no treatment are listed. Group 1 frames were; 1: 8.4 %, 2: 3.6 % and 3: 7.1 % (Fig 6a). Group 2 (5 min magnetic exposure) frames were; 1: 22.9%, 2: 20.8%, 3: 7.1% and Group 3 (10 min magnetic exposure) were 1: 38%, 2: 19.3% and 3: 5.5% (Figs 6 b, c). The results of the statistical analysis of particles in the SEM frames (1-3) were highly significant ( $p < 0.01$ ) and are given in Table 1. The power of the performed test with alpha was 0.050:1.0.

### **3.3 Results of the Thermo-gravimetric analysis**

The results of the TGA analysis are shown in Figure 4. The spherical CHX particles (SCP) had 8.5 wt.% remaining after increasing the temperature to 800 °C [24], and for the Fe<sub>3</sub>O<sub>4</sub> functionalized CHX spheres there was 33 wt.% remaining.

### **3.4 Release kinetics of CHX from resin**

The release of CHX from the HEMA-UDMA resin discs containing functionalized CHX/Fe<sub>3</sub>O<sub>4</sub> spheres (Groups 1-3), demonstrated a two stage process (Fig. 7). In the first stage there was a burst release (until 200 h) which was more rapid for Groups 2 and 3 when compared to Group 1. The second stage of the plots demonstrated a sustained release for all groups. Group 1 (no magnetic treatment) resulted in 4.4% CHX released, whilst Group 2 (5 min magnetic treatment) gave 5.9% CHX released and Group 3 (10 min magnetic treatment) resulted in 7.4% CHX released at 650 h.

### 3.5 Cytotoxicity Assay Results

The results of the cytotoxicity assay are presented in Figure 8a. Treatment with functionalized chlorhexidine spheres (CHX/Fe<sub>3</sub>O<sub>4</sub>) and commercial chlorhexidine diacetate (CHX) reduced cell viability in a dose-dependent manner *in vitro*. Overall, CHX/Fe<sub>3</sub>O<sub>4</sub> spheres demonstrated reduced cytotoxicity in comparison with CHX at the same concentration.

Relative cellular viability was reduced to approximately 50% in 0.0005% CHX treated cultures whilst the viability was above 80% in CHX/Fe<sub>3</sub>O<sub>4</sub> sphere treated cells. While viability remained >70% in CHX/Fe<sub>3</sub>O<sub>4</sub> sphere treated cells at 0.001% concentration, CHX treated cells demonstrated reduced viability (~30%) (Fig.8a). Accordingly, changes in cellular morphology (rounded, swelled, and loss of attachment to the wells) were also observed in both 0.0005% and 0.001% CHX treated cultures, whilst the normal morphology was maintained in CHX/Fe<sub>3</sub>O<sub>4</sub> sphere treated cultures (Fig. 8b-d).

### 3.6 Antimicrobial Assay Results

Figure 9 shows the antimicrobial activity of CHX/Fe<sub>3</sub>O<sub>4</sub> against *P.gingivalis*. The growth of *P. gingivalis* was completely inhibited by 0.0005% wt./vol. CHX/Fe<sub>3</sub>O<sub>4</sub> (MIC) (Figure 9a) as measured by culture optical density. The MTT assay also confirmed this MIC value with no viability being detected at 0.0005% wt./vol. CHX/Fe<sub>3</sub>O<sub>4</sub> (Figure 9b).

#### 4. Discussion

The antibacterial activities of metal nanoparticles such as gold, silver, copper, zinc and titanium and their applications against bacterial infections are well established [27]. In the current study  $\text{Fe}_3\text{O}_4$  nanoparticles with an mean (SD) diameter of 13.1 (2.4) nm (Fig. 1) were synthesized using a co-precipitation method to take advantage of their ferrimagnetic properties [28] and allow targeted CHX drug release. Magnetic nanoparticles are extensively used for targeted drug delivery and as diagnostic agents due to their excellent magnetic properties, superior biocompatibility and ease of functionalization [14]. Magnetite ( $\text{Fe}_3\text{O}_4$ ) is an inverse spinel structure with various particle morphologies that include spherical, octahedral or cubic structures [29]. The cubic structure is known to grow via a screw dislocation process [30]. In this study TEM of the precipitated  $\text{Fe}_3\text{O}_4$  nanoparticles appeared to show signs of spherical morphology, although this was indistinct. The shape, size and surface chemical functionalities of iron oxide nanoparticles have a profound effect on their antibacterial activities [3]. When tested, CHX/ $\text{Fe}_3\text{O}_4$  demonstrated antimicrobial activity against the periodontal pathogen *P. gingivalis* (Fig. 9). Importantly the cytotoxicity assay results showed that the CHX/ $\text{Fe}_3\text{O}_4$  treated cells maintained >90% viability at the concentration that inhibited bacterial growth (Fig. 8a). Furthermore,  $\text{Fe}_3\text{O}_4$  nanoparticle alone treated cells demonstrated >90% viability even in the highest treatment concentration tested (0.008%), whilst commercial CHX diacetate treated cells were not viable. This demonstrates that  $\text{Fe}_3\text{O}_4$  nanoparticles were not cytotoxic to the fibroblastic cells (Fig. 8a). Magnetic targeted superparamagnetic iron oxide nanoparticles have been shown to be active against *Staphylococcus epidermidis* biofilms and may be an effective alternative to antibiotics for antibiotic-resistant strains [3]. They also appear more effective in penetrating and inhibiting biofilms compared with traditional antibiotics [9]. The antibacterial mechanisms associated with metal nanoparticles include their electrostatic attraction to the negatively charged bacterial cell membrane, their high surface area to volume particle ratio [31]

and production of a reactive oxygen species, ensuring intimate membrane contact and leading to functional disorder of the bacterial cells [3]. The dynamic series of interfacial interactions between nanoparticles and biological structures are however very complex [32].

The crystallization of the CHX compound onto the Fe<sub>3</sub>O<sub>4</sub> nanoparticles in the present study is advantageous as it presents an opportunity for a dual and tailored antibacterial effect. Enhanced antibacterial activity might also be obtained through the release of Fe<sub>3</sub>O<sub>4</sub> nanoparticles through the dissolution of CHX from the sphere structure. Such a possibility would be valuable in treating antibiotic resistant bacterial biofilms but remains to be demonstrated.

Functionalization of the CHX spheres using the Fe<sub>3</sub>O<sub>4</sub> nanoparticles appeared to dramatically decrease the mean (SD) CHX particle diameter from 17.15 (1.99) μm for the particles with no functionalization (Figs. 2a, 2e) to 10.39 (2.61) μm for the CHX/Fe<sub>3</sub>O<sub>4</sub> spheres (Figs. 2b, 2f). However, the spherical morphology and structure appeared unchanged (Figs. 2c-f). These metal ions may also act as sites for CHX crystallization, where the co-precipitation system encourages surface crystallization. Previous work indicated that functionalizing the CHX spheres with gold nanorods resulted in correlation between the number of nanoparticles added and the mean CHX particle diameter and number ( $r^2=0.98$ ) [24]. It is hypothesized that this might be possible with a number of metal ions including Fe<sub>3</sub>O<sub>4</sub>. The present work was directed towards the magnetic properties associated with ferrimagnetic domains in the Fe<sub>3</sub>O<sub>4</sub> structure (tetrahedral and octahedral sub-lattices) [29], enabling a magnetic responsive drug release behavior. The Fe<sub>3</sub>O<sub>4</sub> nanoparticles identified in this study are known to interact with each other and problems have been posed with functionalizing these particles for different applications [33]. Domains of the excess Fe<sub>3</sub>O<sub>4</sub> nanoparticles were identified trapped within the CHX crystal structure (Figs. 2d, 2f, 3), which enhanced the magnetic effect and allowed the CHX/Fe<sub>3</sub>O<sub>4</sub> particle movement.

The spherical CHX particles produced by co-precipitation of  $\text{CaCl}_2$  and chlorhexidine diacetate had a very high drug content ( $> 90$  wt.%). After TGA analysis (Fig. 4) the remaining weight of the spherical CHX particles was 8.5 wt.%, which was ascribed to the presence of  $\text{CaCl}_2$  and chlorhexidine diacetate/degradation products [23]. When the CHX particles were functionalized with  $\text{Fe}_3\text{O}_4$  nanoparticles, the remaining weight increased to 33 wt.% (Fig. 4), which was due to the non-decomposing metal nanoparticles at high temperature. Therefore 24.5 wt.% of  $\text{Fe}_3\text{O}_4$  nanoparticles were calculated to have been incorporated into the CHX particles and this proportionally reduced CHX content may be the reason for the reduced cytotoxicity observed against fibroblastic cells in comparison with CHX at the same concentration (Fig. 8).

Magnetic nanoparticles are largely incorporated within carriers together with drugs resulting in a low drug loading rate [34]. The current strategy overcomes this problem as it was possible to grow the CHX drug (67 wt.%) directly onto the surface of metal nanoparticles and trap them within its structure (Figs. 2d, 2f). There are many studies using magnetic nanoparticles for targeted delivery, but this is the first study to integrate magnetic nanoparticles directly into drug crystals. This method promotes intimate contact between the metal nanoparticles and CHX crystals that can improve both plasmonic [24] and magnetic sensitivity, broadening the range of drug delivery methods. This also allows the potential of varying the  $\text{Fe}_3\text{O}_4$  nanoparticle content to tailor these properties and to control the size, number and CHX/ $\text{Fe}_3\text{O}_4$  sphere ratio via particle crystallisation.

Functionalization of the CHX spheres allowed the particles to be moved in the viscous HEMA-UDMA resin in response to application of an external magnetic force (Figs. 5b, 5c). The resin sample without magnetic exposure presented a homogeneous distribution of CHX/ $\text{Fe}_3\text{O}_4$  spheres across the SEM cross sectional samples (Figs. 5a, 6a), which were evenly distributed by the resin mechanical mixing prior to curing. There was also no statistical difference ( $p>0.05$ )



between the number of spheres present in SEM frames 1-3 indicating this to a depth of 141 microns. Following magnetic exposure (5 and 10 mins) there was a significant increase ( $p < 0.01$ ) in the number of CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres to a depth of 94 microns from the surface (Fig 5b, 5c, 6b, 6c) nearest to the magnet. This represented 43.7%- 57.3% of the CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres in the samples (Figs. 6 b, 6c). At a depth >94 microns (Frames 3, Figs. 5b, 5c) from the surface nearest the magnet the CHX/ Fe<sub>3</sub>O<sub>4</sub> sphere number was reduced and not significantly different to the CHX/ Fe<sub>3</sub>O<sub>4</sub> sphere distribution (frame 3, group 1) with no magnetic exposure (Table 1,  $P > 0.05$ ). Clearly with magnetic exposure there was bulk movement of the CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres towards the magnet. The strength of the magnetic field was a key factor, and the magnetic field currently used was 400 mT at the magnet surface. Magnetic field strength has been shown to drop rapidly as a function of distance from the surface for Fe<sub>3</sub>O<sub>4</sub> nanoparticles incorporated in microcapsules (1.3  $\mu\text{m}$  diameter) and dispersed in cell suspensions [35]. The current work necessitated the movement of the larger mean (SD) 10.39 (2.61)  $\mu\text{m}$  diameter CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres through a viscous HEMA-UDMA resin and with a Mylar film (1 mm depth) used in the disc fabrication process, further reducing the magnetic fields influence. This would explain the residual CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres remaining in the SEM frames furthest away from the magnet (Figs 6 b, 6c). Further work is required on the type of magnet/magnetic field strength and duration on the movement of the drug particles to optimise this process. More efficient synthesis of cubic nanoparticles could also affect their magnetic sensitivity [36], allowing a stronger magnetic field to be used to reduce the exposure time. Guardia et al [37] suggested that cube-shaped (polydispersity <20%) Fe<sub>3</sub>O<sub>4</sub> nanoparticles (19 nm) were in the transition between superparamagnetic and ferromagnetic particles, which yields high magneto-thermal properties. The magnetic attraction of CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres and their directed movement through a dental polymer or gel is extremely desirable, as it allows the drug to be moved to the infection site to potentially kill the bacteria more efficiently. Antibacterial agents of this kind

navigated by an external magnetic field resulted in an 8-fold higher antibacterial effectiveness in comparison to using antibiotics [3]. The current CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres might therefore be effective in sites in the oral cavity inaccessible to current antimicrobial agents, such as deep periodontal pockets or sites susceptible to secondary caries.

The release kinetics of CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres from the HEMA-UDMA resin demonstrated a two-stage process. Utilising a magnetic field (5 and 10 mins) to draw the CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres to the surface appeared to have the effect of inducing a more rapid drug release in the initial stages (until 200 h) from 2.9% (no magnetic field) to 4.0% (5 min) or 4.5% (10 min) (Fig. 7). Near surface drug entrapment and rapid leaching in water explains this burst effect [38, 39]. Inhibition of the polymerisation process by the increased CHX content in the near specimen surface (94 µm) and the presence of residual monomers /unreacted hydrophilic components encouraging water sorption of the polymer [40, 41]. After a 650 h sustained CHX release the overall drug content was also increased from 4.4% to 7.4% (10 min magnetic exposure). Most of the CHX incorporated in the Group 1 resin discs remained in the resin due to the homogeneous distribution of CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres throughout the sample interior (Group 1, Fig 5a), which limited the volume fraction of CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres available for rapid dissolution and diffusion mechanisms near the surface [41]. Synthesis of spherical CHX particles (without functionalization) by co-precipitation of CaCl<sub>2</sub> and chlorhexidine diacetate indicated that the particles were stable, less soluble and with lower CHX release rates than the chlorhexidine diacetate counterpart when dispersed in resin [23]. This can be attributed to Cl<sup>-</sup> ions in their structure [42] and has been linked to the CHX release kinetics [22]. Following application of a magnetic field for 5-10 mins the CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres (43.7%- 57.3%) were drawn to a depth of 94 microns from the surface of the polymer (Fig 5b, 5c, 6b, 6c). The diffusion of CHX is determined by the water droplets formed around the drug particles, which

form pathways or channels linked to the surface [43]. It is therefore not surprising that the resin discs with more CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres distributed near the surface had higher release rates. Mechanisms for CHX/ Fe<sub>3</sub>O<sub>4</sub> sphere dissolution appeared to be via the central part of the particle which was more prevalent for spheres nearer the resin surface than the interior. This dissolution behavior was thought to be associated with the penetration and intimate contact between the resin and the interior CHX/ Fe<sub>3</sub>O<sub>4</sub> sphere structure [23]. The magnetic CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres embedded in resin could therefore also be further triggered with an alternating magnetic field to induce a responsive release [33, 44, 45], which will be explored in future work.

The functionalized CHX spheres (CHX/ Fe<sub>3</sub>O<sub>4</sub>) have a number of potential applications and this is the first study to demonstrate novel magnetic functionalized CHX particles, where their distribution in a resin can be manipulated with an external magnetic field to alter the CHX release kinetics. These particles could be incorporated into dental filling materials and used to restore exposed implant surfaces or dispersed into maxillo-facial/ denture lining materials, allowing magnetically directed and drug responsive properties. These magnetic CHX/ Fe<sub>3</sub>O<sub>4</sub> spheres dispersed in gels and varnishes etc. are particularly attractive for the treatment of persistent periodontal or biomaterial-implant-associated infections.

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## List of Figures

- Figure 1, TEM image of  $\text{Fe}_3\text{O}_4$  nanoparticles.
- Figure 2, SEM photomicrographs of: (a, c, e) spherical chlorhexidine particles and; (b, d, f) CHX/ $\text{Fe}_3\text{O}_4$  spheres.
- Figure 3, EDX of CHX/ $\text{Fe}_3\text{O}_4$  spheres indicating peaks for Fe.
- Figure 4, TGA for the spherical chlorhexidine spheres and CHX/ $\text{Fe}_3\text{O}_4$  spheres.
- Figure 5, Cross-sectional SEM photomicrographs of HEMA-UDMA resin discs with  $\text{Fe}_3\text{O}_4$ /CHX spheres: (a) without magnetic field treatment; (b) with magnetic field treatment for 5 min; (c) with magnetic field treatment for 10 min.
- Figure 6, Plots of the distribution of  $\text{Fe}_3\text{O}_4$  /CHX spheres in HEMA-UDMA resin disc as a function of magnetic field treatment. (a) without magnetic field treatment; (b) with magnetic field treatment for 5 min; (c) with magnetic field treatment for 10 min.
- Figure 7, Release kinetics of chlorhexidine from the HEMA-UDMA resin disc as a function of magnetic field treatment. black line= without magnetic field treatment; red line= with magnetic field treatment for 5 min; blue line= with magnetic treatment for 10 min.
- Figure 8, (a) Relative viability of fibroblast cells following treatment with  $\text{Fe}_3\text{O}_4$  nanoparticles, CHX and CHX/ $\text{Fe}_3\text{O}_4$  spheres (results averaged from three independent experiments). Light microscopy images of cell morphology in; (b) untreated-0%; (c) CHX/ $\text{Fe}_3\text{O}_4$  (0.001%) and; (d) Chlorhexidine diacetate (0.001%).
- Figure 9 (a) Growth of *P. gingivalis* was determined by measuring OD 595nm at 0 h after 24 h. (b) Relative viability of *P. gingivalis* following 24 h treatment with CHX/ $\text{Fe}_3\text{O}_4$  (results averaged from three independent experiments and shown as mean SD).

**Figure 1**

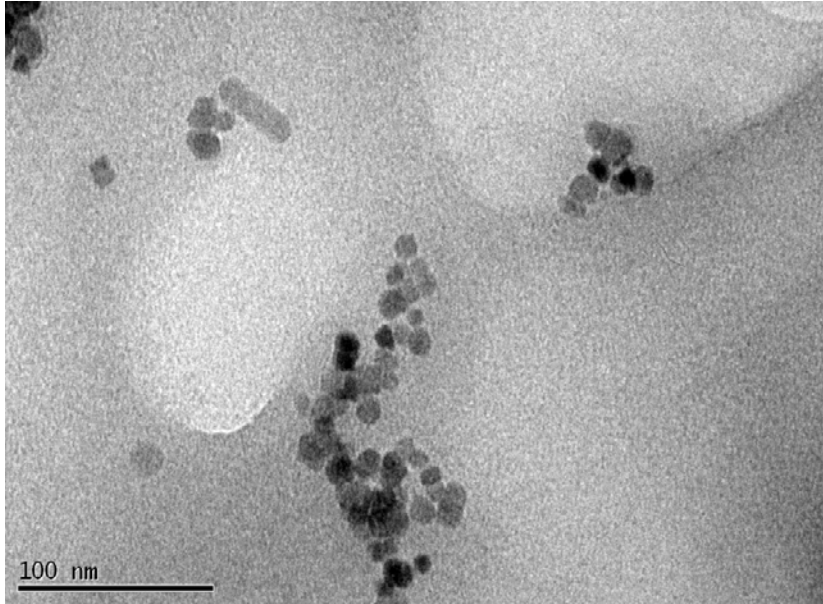


Figure 2

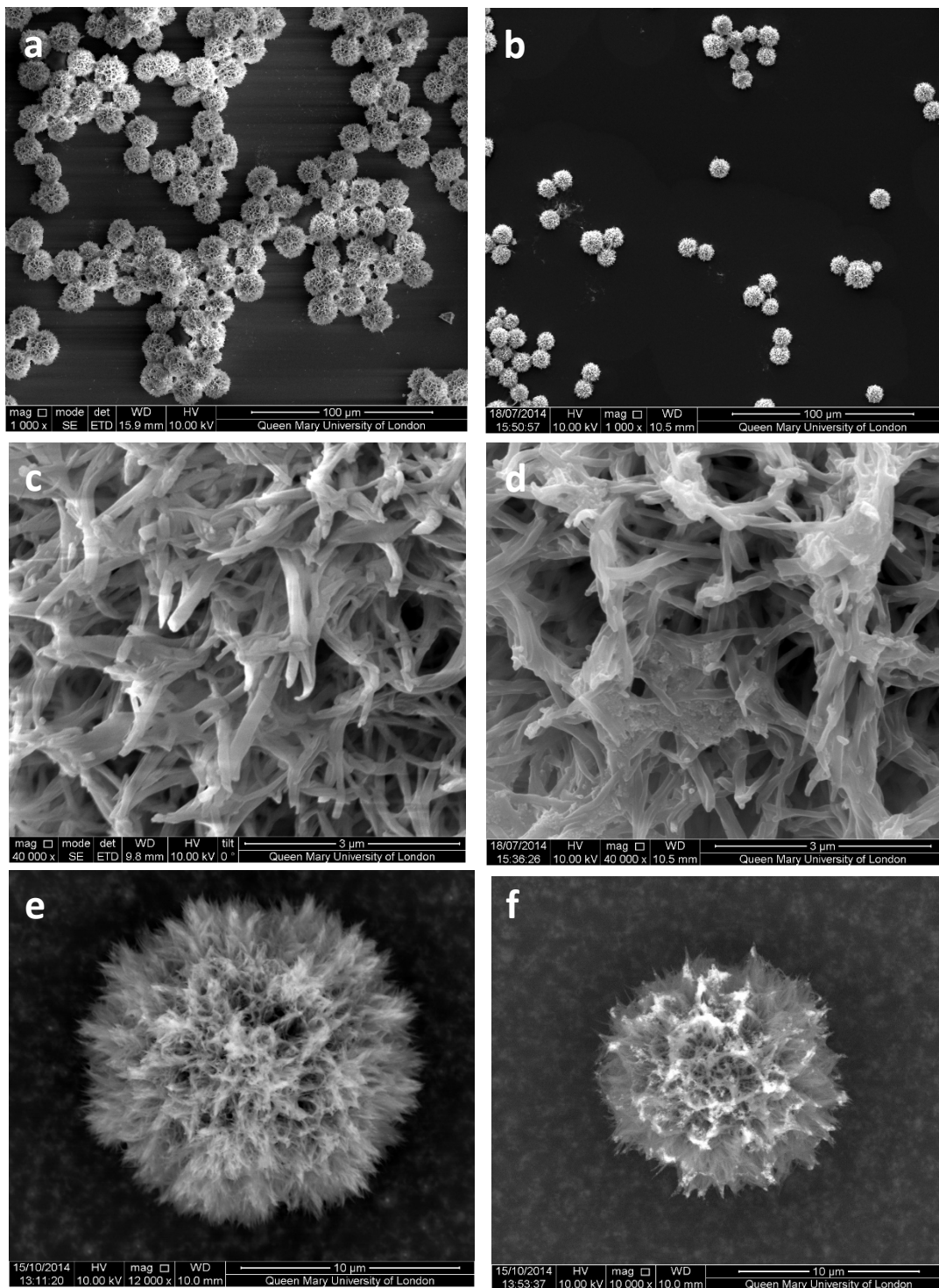
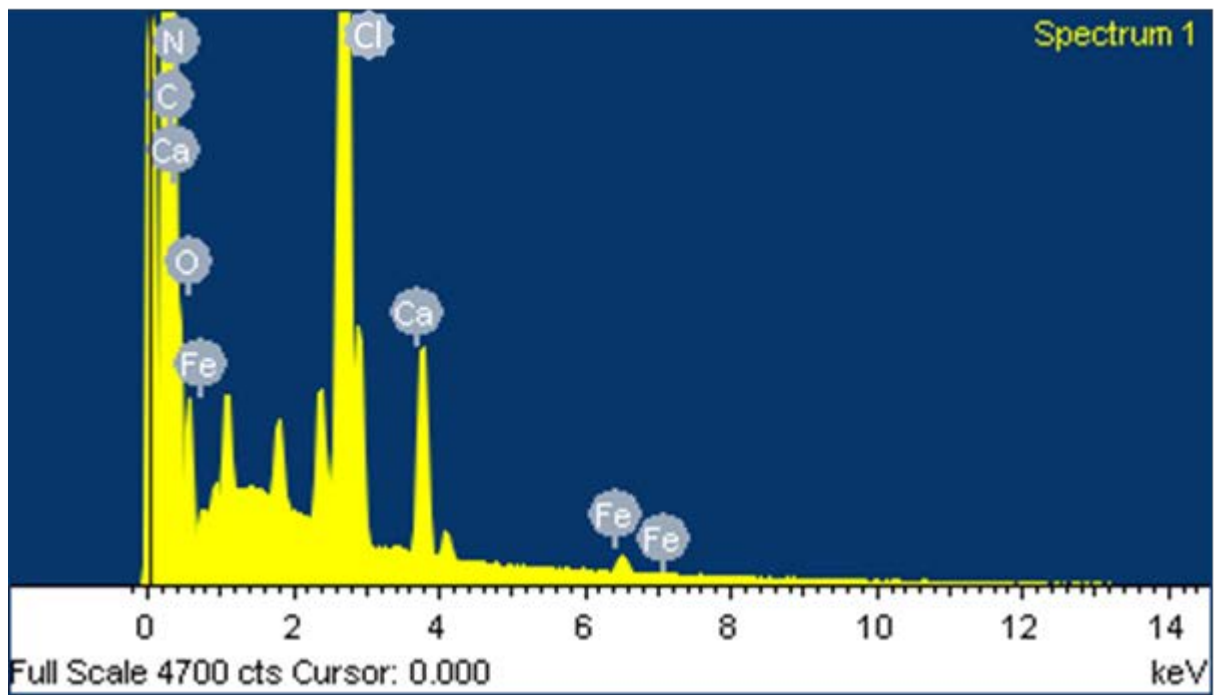
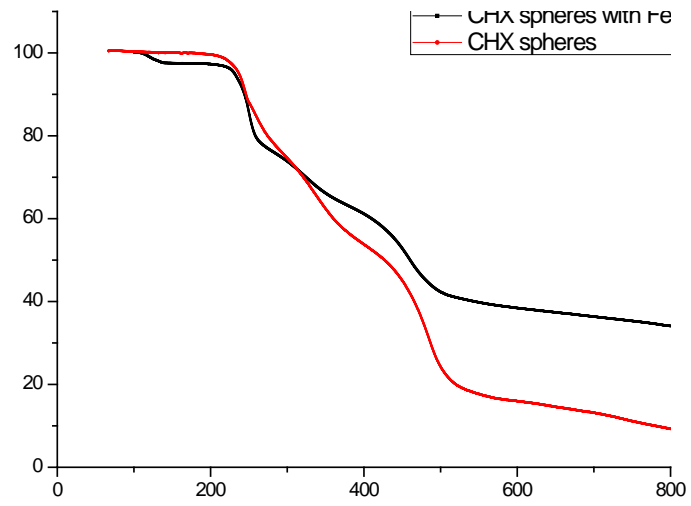


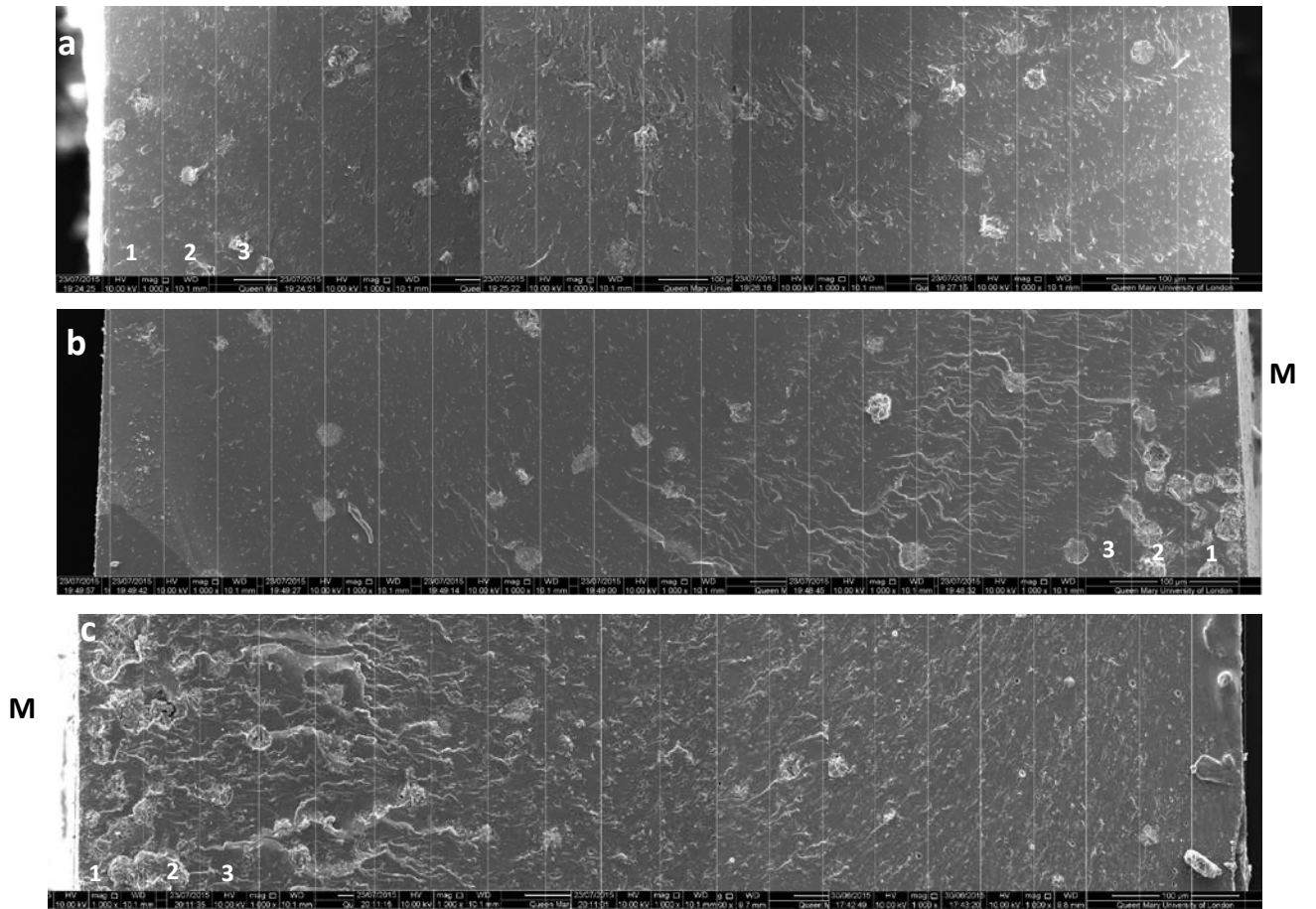
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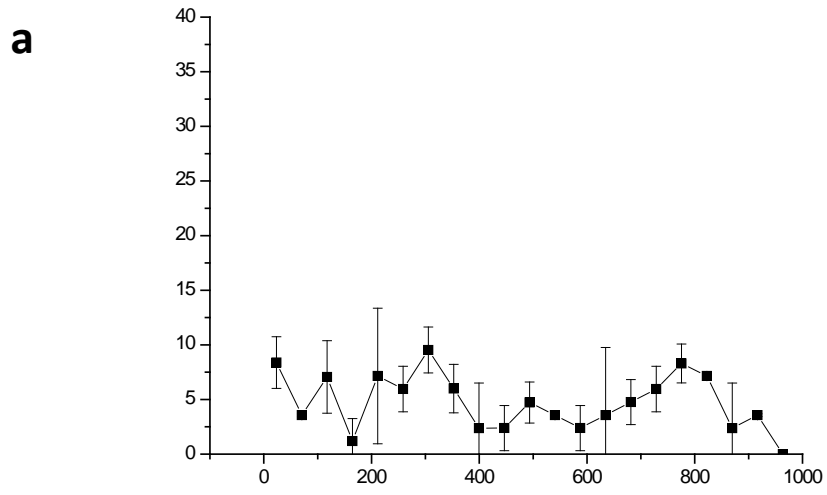
**Figure 4**



**Figure 5**

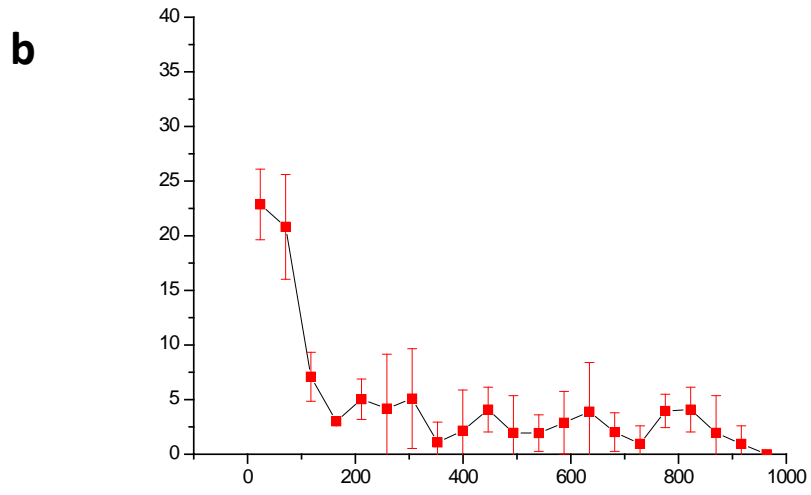


**Figure 6a**





**Figure 6b**



**Figure 6c**

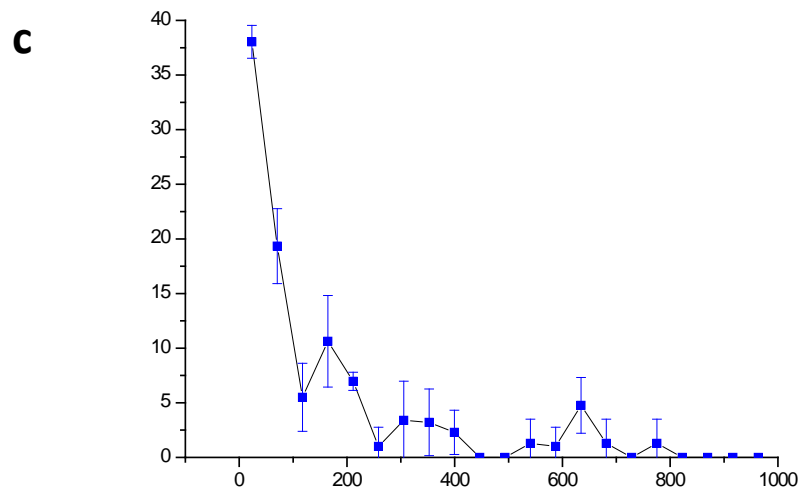
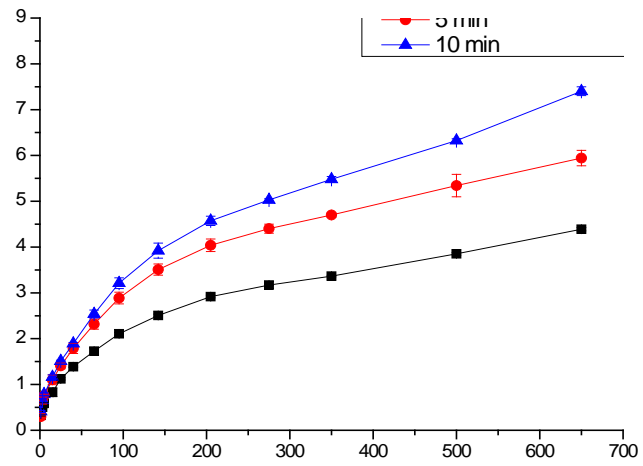
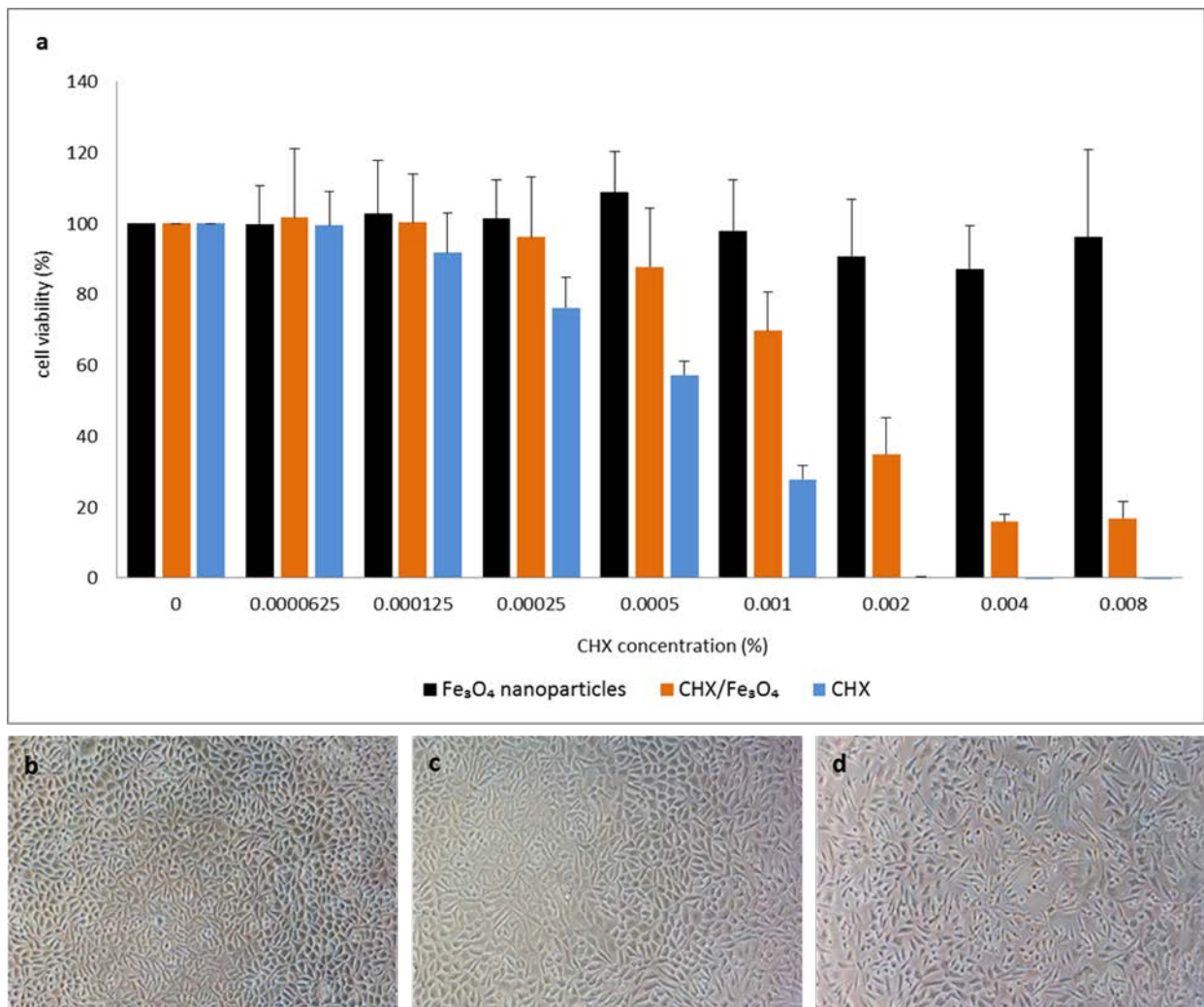


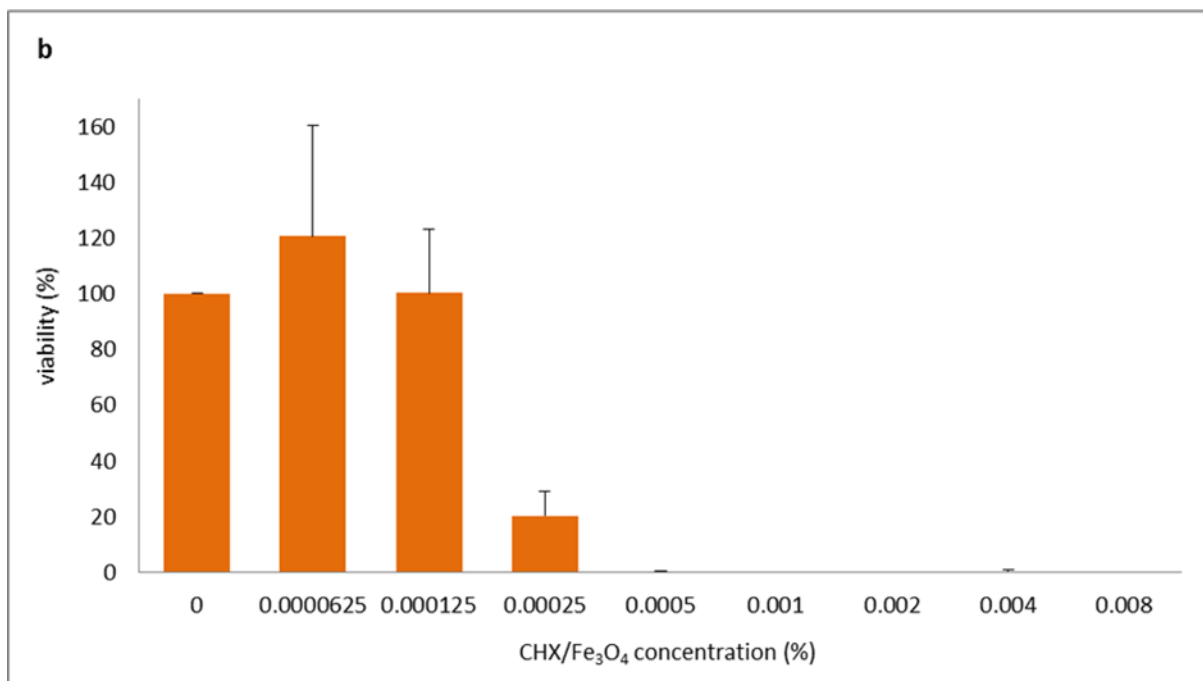
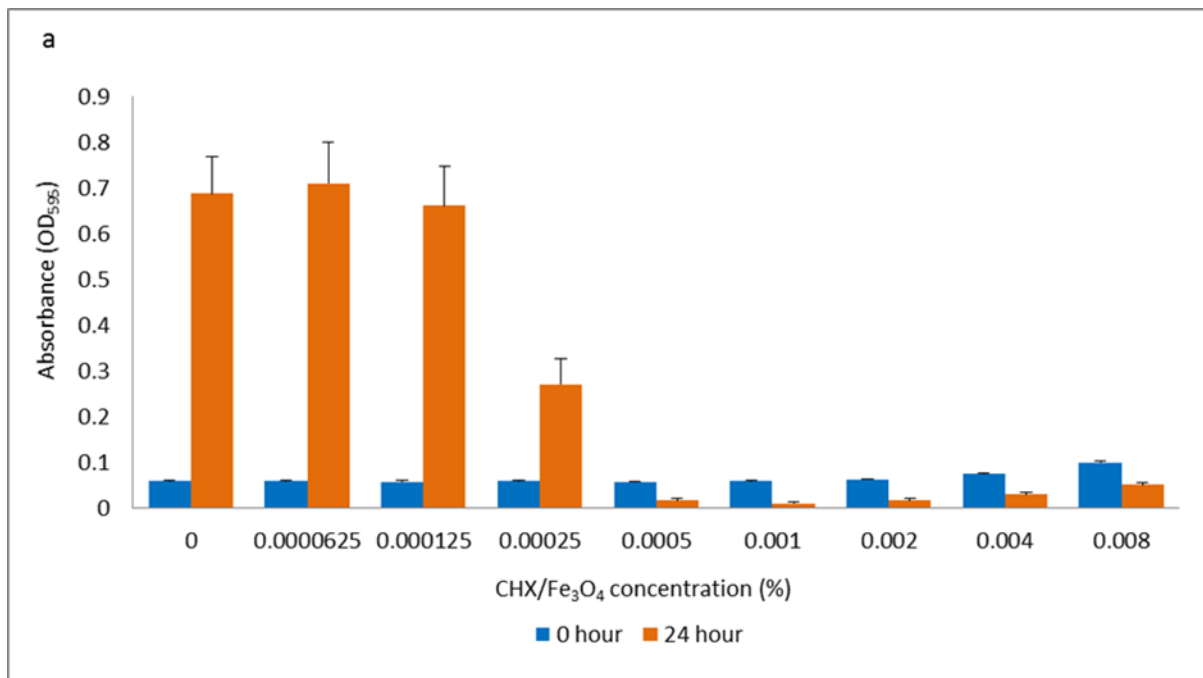
Figure 7



**Figure 8**



**Figure 9**



## List of Tables

Table 1, Mean (SD) CHX/Fe<sub>3</sub>O<sub>4</sub> particle distribution in the cross-sectional SEM photomicrographs, frames 1-3 (figures based on n=3 panoramic SEM images per test group).

**Table 1**

<b>SEM Frame Sections</b>	<b>Mean Particle Number (SD) (Group 1)</b>	<b>Mean Particle Number (SD) (Group 2)</b>	<b>Mean Particle Number (SD) (Group 3)</b>
<b>1</b>	<b>2.3 (0.6)</b> <sup>a,c,e,f</sup>	<b>7.7 (1.5)</b> <sup>b,d</sup>	<b>11 (1.0)</b> <sup>d</sup>
<b>2</b>	<b>1.0 (0.0)</b> <sup>a,c,f</sup>	<b>7.0 (2.0)</b> <sup>b</sup>	<b>5.7 (1.5)</b> <sup>b,e</sup>
<b>3</b>	<b>2.0 (1.0)</b> <sup>a,c,f</sup>	<b>2.3 (0.6)</b> <sup>c,e,f</sup>	<b>1.7 (1.2)</b> <sup>f</sup>

\*Significant differences are indicated by different superscript letters between groups (P<0.05).  
(Figures are based on n=3 panoramic images per test group).