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## Structural and dielectric properties of KBa<sub>5</sub>TiNb<sub>9</sub>O<sub>30</sub> ferroelectrics

H Sharma, Kiran Kumari<sup>\*</sup> and H S Singh

University Department of Physics, Veer Kunwar Singh University, Arrah-802 301, India

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**Abstract** : Potassium barium titanium niobate (hereafter PBTN) having composition KBa<sub>5</sub>TiNb<sub>9</sub>O<sub>30</sub> has been synthesized by high temperature solid-state reaction technique. Room temperature X-ray diffraction (XRD) studies of the compound show that it has a tetragonal structure with lattice parameters a = b = 1243 Å and c = 410 Å Measurement of dielectric constant from liquid nitrogen temperature to 400°C suggests that the material is ferroelectric at room temperature and transforms into a paraelectric phase at around 290°C

Keywords : Tetragonal structure, dielectric constant, ferroelectric material

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Potassium barium titanium niobate (KBa<sub>5</sub>TiNb<sub>9</sub>O<sub>30</sub>) belongs to a ferroelectric oxide family of tungsten-bronze (TB) structure [1] of general formula AB<sub>5</sub>TiNb<sub>9</sub>O<sub>30</sub> (A = Na, K and B = Sr, Ba). The tungsten-bronze (TB) structure consists of a complex array of distorted BO<sub>6</sub> octahedra sharing corners in such a way that the different types of interstices ( $A_1$ ,  $A_2$ ,  $B_1$ ,  $B_2$  and C) are available for cation substitution [2]. The polar axis of most of the members of TB family is normally c-axis. A wide variety and range of compounds of tungsten-bronze (TB) type has been studied. Some niobates with TB structure such as barium sodium niobate and potassium lanthanum niobates [3] are quite attractive and interesting owing to their wide industrial applications. Studies of structural and dielectric properties of some ferroelectric oxides of TB structure [4,5] have been reported. The electrical conductivity measurements of a few compounds [6] also confirm the occurrence of offset near the transition temperature.

Present Address : Department of Physics, Dr SKSM College, Mothari-845 401

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A literature survey reveals that even though the compound has been suggested to be ferroelectric, no systematic X-ray and detailed dielectric studies have yet been reported. We, therefore, present in this paper preliminary structural and detailed electrical (dielectric constant ( $\epsilon$ ) and loss (tan  $\delta$ )) properties of the PBTN compound.

The polycrystalline samples of PBTN were prepared by a standard high-temperature solid-state reaction technique from the raw materials : TiO<sub>2</sub> (99 % s.d. fine chem Pvt. Ltd.), Nb<sub>2</sub>O<sub>5</sub> (99.9 % SMP), K<sub>2</sub>CO<sub>3</sub> (99.9 % SM Chemicals) and BaCO<sub>3</sub> (M/s Ultra Pure LOBA CHEMIE) in a suitable proportion. These oxides and corbonates were thoroughly mixed in an agate-mortar in alcohol for 10 h and dried. The dried powders were calcined in a crucible at 1050°C for 25 h. The process of calcination and mixing was repeated till the final homogeneous powder of PBTN was obtained. The formation and quality of the compound were checked with X-ray diffraction technique. Some of cylindrical pellets (of diameter 10.3 mm and thickness 2.3 mm) were made under the isostatic pressure ( $6 \times 10^7$  Kg/cm<sup>2</sup>) using a hydraulic press. The pellets were then sintered in air atmosphere at 1100°C for 10 h. After polishing and grinding, both the flat surfaces of some pellets were electroded with air drying silver paints for electrical measurements.

The X-ray diffractograms of the PBTN pellet samples were taken with  $CuK_{\alpha}$  radiation ( $\lambda = 1.5418$  Å) in a wide  $2\theta$  range ( $20^{\circ} \le 2\theta \le 70^{\circ}$ ) with Philips powder diffractometer (PW 1710 Holland).

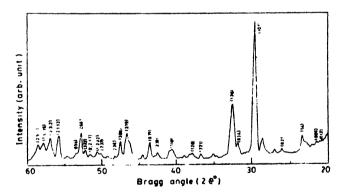
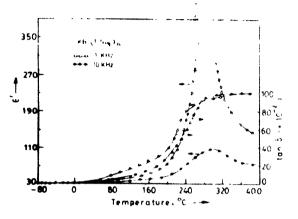


Figure 1. X-ray diffractogram of PBTN pellet sample.

The dielectric permittivity ( $\varepsilon'$ ) and loss (tan  $\delta$ ) were obtained on the sintered and electroded pellet samples as functions of frequencies (500 Hz to 10 KHz) and temperature (liquid nitrogen to 400°C) using G R 1620 capacitance measuring assembly. The reliability of the data was checked by repeating the experiments with different instruments (LCR-High tester, Hioki 5530, Japan) in the same physical conditions. Following the data given in Landolt-Bornstein tables [7], all the prominent peaks in Figure 1 were indexed using tetragonal unit cells. A good agreement between the observed and calculated *d*-values suggests that the choice of unit cells is correct and the corresponding lattice parameters are a = b = 12.43 Å and c = 4.10 Å



**Figure 2.** Variation of dielectric permittivity (t') and dielectric loss (tan  $\delta t$ ) with temperature at two frequencies 1 kHz and 10 kHz

Figure 2 shows that the variation of dielectric permittivity  $(\varepsilon')$  and dielectric loss (tan  $\delta$ ) with temperature at two frequencies 1 kHz and 10 kHz. It is evident from this figure that the dielectric permittivity increases gradually above room temperature and becomes maximum around 290°C. The dielectric loss on the other hand increases up to about 290°C and then levels off. The reason for levelling off may be due to increased ac conductivity of the sample. The peak in the dielectric permittivity  $(\varepsilon')$  around 290°C is indicative of ferroelectric phase transition

It is thus concluded that PBTN is tetragonal at room temperature and it may possess ferroelectric property with a ferroelectric phase transition temperature of 290°C which is quite high as compared to other members of the family

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