



De Assis, G. C., Skovroinski, E., Leite, V. D., Rodrigues, M. O., Galembeck, A., Alves, M. C. F., ... De Oliveira, R. J. (2018). Conversion of "Waste Plastic" into Photocatalytic Nanofoams for Environmental Remediation. *ACS Applied Materials and Interfaces*, 10(9), 8077-8085.  
<https://doi.org/10.1021/acsami.7b19834>

Peer reviewed version

Link to published version (if available):  
[10.1021/acsami.7b19834](https://doi.org/10.1021/acsami.7b19834)

[Link to publication record in Explore Bristol Research](#)  
PDF-document

This is the author accepted manuscript (AAM). The final published version (version of record) is available online via ACS at <https://pubs.acs.org/doi/10.1021/acsami.7b19834> . Please refer to any applicable terms of use of the publisher.

## University of Bristol - Explore Bristol Research

### General rights

This document is made available in accordance with publisher policies. Please cite only the published version using the reference above. Full terms of use are available:  
<http://www.bristol.ac.uk/pure/about/ebr-terms>

## Conversion of “waste plastic” into photocatalytic nanofoams for environmental remediation

Geovânia Cordeiro de Assis, Euzébio Skovroinski, Valderi Duarte Leite, Marcelo Oliveira Rodrigues, André Galembeck, Mary Alves, Julian Eastoe, and Rodrigo Jose de Oliveira

*ACS Appl. Mater. Interfaces*, **Just Accepted Manuscript** • DOI: 10.1021/acsami.7b19834 • Publication Date (Web): 20 Feb 2018

Downloaded from <http://pubs.acs.org> on February 21, 2018

### Just Accepted

“Just Accepted” manuscripts have been peer-reviewed and accepted for publication. They are posted online prior to technical editing, formatting for publication and author proofing. The American Chemical Society provides “Just Accepted” as a service to the research community to expedite the dissemination of scientific material as soon as possible after acceptance. “Just Accepted” manuscripts appear in full in PDF format accompanied by an HTML abstract. “Just Accepted” manuscripts have been fully peer reviewed, but should not be considered the official version of record. They are citable by the Digital Object Identifier (DOI®). “Just Accepted” is an optional service offered to authors. Therefore, the “Just Accepted” Web site may not include all articles that will be published in the journal. After a manuscript is technically edited and formatted, it will be removed from the “Just Accepted” Web site and published as an ASAP article. Note that technical editing may introduce minor changes to the manuscript text and/or graphics which could affect content, and all legal disclaimers and ethical guidelines that apply to the journal pertain. ACS cannot be held responsible for errors or consequences arising from the use of information contained in these “Just Accepted” manuscripts.

1  
2  
3  
4  
5  
6  
7  
8  
9  
10  
11  
12  
13  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23  
24  
25  
26  
27  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

# Conversion of “Waste Plastic” into Photocatalytic Nanofoams for Environmental Remediation

*Geovania C. de Assis*<sup>§Δ</sup>; *Euzébio Skovroinski*<sup>‡</sup>; *Valderi D. Leite*<sup>§</sup>; *Marcelo O. Rodrigues*<sup>†</sup>; *André Galembeck*<sup>‡</sup>; *Mary C.F. Alves*<sup>§</sup>; *Julian Eastoe*<sup>‡\*</sup>; *Rodrigo J. de Oliveira*<sup>§\*</sup>

<sup>§</sup> Departamento de Química, Universidade Estadual da Paraíba, 58429-500, Brazil.

<sup>‡</sup> Programa de Pós-Graduação em Ciência de Materiais, Universidade Federal de Pernambuco, 50740-540, Brazil.

<sup>‡</sup> Departamento de Química Fundamental, Universidade Federal de Pernambuco, 50740-540, and CETENE, MCTI, 50740-545, Brazil.

<sup>†</sup> Instituto de Química, Universidade de Brasília, 70297-400, Brazil.

<sup>‡</sup> School of Chemistry, University of Bristol, BS8 1TS, UK.

## Abstract

Plastic debris is a major environmental concern, and to find effective ways to reuse polystyrene (PS) presents major challenges. Here it is demonstrated that polystyrene

1  
2  
3 foams impregnated with SnO<sub>2</sub> are easily generated from plastic debris, and can be  
4 applied to photocatalytic degradation of dyes. SnO<sub>2</sub> nanoparticles were synthesized by a  
5 polymeric precursor method, yielding specific surface areas of 15 m<sup>2</sup>/g after heat  
6 treatment to 700 °C. Crystallinity, size and shape of the SnO<sub>2</sub> particles were assessed by  
7 X-ray diffraction (XRD) and transmission electron microscopy (TEM), demonstrating  
8 the preparation of crystalline spherical nanoparticles with sizes around 20 nm. When  
9 incorporated into PS foams, which were generated using a thermally induced phase  
10 separation (TIPS) process, the specific surface area increased to 48 m<sup>2</sup>/g. These  
11 PS/SnO<sub>2</sub> nanofoams showed very good efficiency for photodegradation of Rhodamine  
12 B, under UV irradiation, achieving up to 98.2% removal. In addition the PS/SnO<sub>2</sub>  
13 nanofoams are shown to retain photocatalytic activity for up to five reuse cycles.  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23  
24  
25  
26  
27

28 Keywords: environmental remediation, nanocatalysis, dye degradation, porous catalyst,  
29 polymer waste, plastic reuse.  
30  
31  
32

### 33 1. INTRODUCTION

34  
35 A wide range of toxic and hazardous substances, including synthetic dyes, are  
36 continuously being released into wastewaters, mostly due to lack of effective treatment  
37 methods<sup>1</sup>. Dyes are used by various industries, however, recent studies have  
38 demonstrated that they are responsible for serious damage to aquatic ecological  
39 systems<sup>2</sup>. Rhodamine B (RhB) is a fluorone type organic dye belonging to the xanthene  
40 family, and studies suggest that RhB is carcinogenic, as well as a chronic reproductive  
41 and neuro toxin to humans and animals<sup>3</sup>, and to a lesser extent an allergen<sup>4</sup>. Therefore  
42 development of methods for degradation of dyes like RhB in industrial effluents are of  
43 high importance<sup>5</sup>.  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

1  
2  
3 Recently, numerous new approaches have been developed for removing pollutants  
4 from wastewater<sup>6</sup>. Advanced oxidation processes (AOP)<sup>7</sup> hold particular promise for  
5 degradation of organic contaminants. Among these AOP methods, heterogeneous  
6 photocatalysis is very attractive offering the potential to turn organic contaminants into  
7 mineral salts and end products such as CO<sub>2</sub> and H<sub>2</sub>O<sup>8-11</sup>. Various metal oxides, such as,  
8 TiO<sub>2</sub>, ZnO, ZnS, WO<sub>3</sub>, CdS, SnO<sub>2</sub>, among others, can be used as semiconductors for  
9 heterogeneous photocatalysis and degradation<sup>12</sup>. In addition, nanostructured materials  
10 have been widely studied owing to beneficial properties, such as high electrical  
11 conductivity, high transparency in the visible region, high thermal, mechanical and  
12 chemical stability, and for potential applications in optically transparent materials and  
13 liquid crystal displays<sup>13</sup>, catalysts, gas sensors<sup>14</sup> and solar cells<sup>15</sup>.

14  
15  
16  
17  
18  
19  
20  
21  
22  
23  
24  
25  
26 Catalytic nanoparticles, such as SnO<sub>2</sub>, are generally supported on substrates to  
27 facilitate separation and reuse from the reaction medium<sup>16</sup>. A variety of catalyst support  
28 morphologies have been studied, including fibers, three dimensional networks and other  
29 complex shapes<sup>17</sup>. In this regard, foams appear to be very promising supports for  
30 photocatalysts, since they present different length scales from nano- to millimeter,  
31 facilitating dispersion and support of nanoparticles, whilst also providing large surface  
32 areas in mechanically robust 3-D network structures of macroscopic dimensions. A  
33 procedure for obtaining micro- and nano-structured polymeric foams was introduced by  
34 Aubert and Clough<sup>18</sup>, by mixing polystyrene (PS) with a mixture of cyclohexane  
35 (CH)/benzene. Eastoe *et al*<sup>19</sup> also developed foams of surfactant-stabilized CaCO<sub>3</sub>  
36 nanoparticles using thermally induced phase separation (TIPS) from cyclohexane  
37 dispersions. Based on these previous reports it is of interest to see if semiconductor-  
38 containing foams can offer increased efficiency of organic pollutant photocatalytic  
39 degradation. Additionally, there is a need to find useful applications for waste  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

1  
2  
3 polystyrene. The aim of this article is to combine these two issues to explore reuse of PS  
4  
5 for the development of PS/SnO<sub>2</sub> based nanofoams, and then to evaluate these  
6  
7 photocatalytic systems for potential applications in environmental remediation of RhB  
8  
9 by photo-degradation.  
10

## 11 12 13 2. EXPERIMENTAL SECTION

### 14 15 2.1 Materials

16  
17 Citric Acid (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>·H<sub>2</sub>O) (Cargill / 99.5% purity), Ethylene Glycol  
18  
19 (HO·CH<sub>2</sub>·CH<sub>2</sub>·OH) (Avocado / 99.0% purity), nitric acid (HNO<sub>3</sub>) (Dinâmica / 65.0%  
20  
21 purity), tin chloride (SnCl<sub>2</sub>·2H<sub>2</sub>O) (Dinâmica / 98.0 to 103.0% purity), cyclohexane  
22  
23 (CH<sub>2</sub>, C<sub>6</sub>H<sub>12</sub>) (Sigma-Aldrich) and polystyrene (PS) obtained in the form of pellets sold  
24  
25 in a local market, were used as received.  
26  
27

### 28 29 2.2 Synthesis of nanoparticles of SnO<sub>2</sub>

30  
31 The SnO<sub>2</sub> nanoparticles were synthesized by the polymeric precursor method<sup>20</sup>.  
32  
33 Initially, 319.122 g of citric acid was dissolved in 1,000 mL of water and then, 125.000  
34  
35 g of tin chloride was added to the solution under stirring. The molar ratio of tin (II)  
36  
37 chloride dihydrate to citric acid was 1:3. After complete dissolution, 22 mL of  
38  
39 ammonium hydroxide aqueous solution was added to the mixture. The system was  
40  
41 cooled in an ice/water mixture to facilitate the precipitation of tin citrate. After  
42  
43 separation, 8.000 g of the tin citrate was dissolved at room temperature in 2.000 mL of  
44  
45 H<sub>2</sub>O and, to complete the dissolution, 8.000 mL of nitric acid were slowly added.  
46  
47 Ethylene glycol was added to the previous solution, in a ratio of 40:60 with respect to  
48  
49 the total amount of citric acid and then stirred at 70 °C to occur release of NO<sub>x</sub> gases,  
50  
51 leading to a reduction in resin weight by about one half. Polymerization occurs through  
52  
53 an esterification reaction between citrate and ethylene glycol, thereby obtaining a  
54  
55  
56  
57  
58  
59  
60

1  
2 polyester with homogeneously distributed metal ions. After the reaction, a polymer  
3 resin was formed and this resin was then placed in a muffle furnace EDG-01 at a  
4 temperature of 350°C for 2 hours under an air atmosphere, to partially remove organic  
5 matter. In this step, the polymer chains present in the resin were decomposed yielding a  
6 so called expanded resin precursor, being similar to a sponge material of dark  
7 coloration, which was removed from the furnace and powdered in a mortar. The powder  
8 precursor material was then treated at temperatures of 700, 800 and 900°C for 2 hours  
9 in an air atmosphere and named as SnO<sub>2</sub>-700, SnO<sub>2</sub>-800 and SnO<sub>2</sub>-900.  
10  
11  
12  
13  
14  
15  
16  
17  
18

### 19 2.3 Synthesis of SnO<sub>2</sub> impregnated polystyrene nanofoams (PS/SnO<sub>2</sub> nanofoams)

20 The PS/SnO<sub>2</sub> nanofoams were obtained by a TIPS process<sup>21</sup> combined with  
21 lyophilization<sup>22</sup>. The PS content in cyclohexane mixture ranged from 1.0, 2.5 to 5.0%  
22 (w/v) and the concentration of SnO<sub>2</sub> in the nanofoams was also varied (2.5 and 5.0%,  
23 w/w relative to PS mass). The samples were labeled as PS(X)/SnO<sub>2</sub>(Y), where X is the  
24 percentage (w/v) of PS in CH and Y is the percentage (w/w) of SnO<sub>2</sub> relative to PS.  
25 PS/CH mixture was sonicated for 3 hours in an ultrasonic bath to complete polystyrene  
26 dissolution in the solvent. For example, to prepare the colloidal dispersion for  
27 PS(2.5)/SnO<sub>2</sub>(5) nanofoams, 0.0625 g of SnO<sub>2</sub> was added to 5 ml of 2.5% PS solution,  
28 and then kept for 2 hours in an ultrasound bath, to allow the formation of the colloidal  
29 dispersion. The dispersions were frozen using a Peltier system, (Fig. S1 in Supporting  
30 Information) at -10 ° C (the freezing point of CH is + 6 ° C). After 10 minutes of  
31 cooling, a vacuum pump was connected to remove the CH solvent by lyophilization for  
32 30 minutes. This technique results in the removal of the frozen CH solvent based on  
33 sublimation<sup>23</sup>. Tests were performed to find the optimal time for obtaining the foams.  
34 The low pressure freeze-drying led to cyclohexane removal by sublimation in the  
35 sample leaving behind a solid foam with a hard "skeleton".  
36  
37  
38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

## 2.4 Characterization

All samples were characterized by X-Ray Diffraction (XRD) analysis in a Miniflex diffractometer 300 of RIGAKU with Cu Ka radiation, 2 kVA power, voltage of 30 kV and current of 30 mA. Scans were made in the range of  $2\theta$  between 10 and 80 ° with a step of 0.02 ° and speed of 2°s<sup>-1</sup>. The B.E.T. surface area of the samples was estimated by N<sub>2</sub> adsorption analysis at 77 K in a surface area analyzer from Quanta Chrome Instruments, New Model 2200E. The samples were treated at 80 °C for 3 h under vacuum until reaching a pressure lower than 10 μm Hg. Transmission electron microscopy was conducted using a FEI Tecnai 20 instrument, with acceleration voltage of 200kV, at the Centro de Tecnologias Estratégicas do Nordeste - CETENE.

## 2.5 Evaluation of Photocatalytic Activity

The photocatalytic activity of SnO<sub>2</sub> and PS/SnO<sub>2</sub> nanofoams was evaluated by photodegradation of Rhodamine B (RhB). Photocatalytic tests varying RhB concentration were carried out at 1.0x10<sup>-4</sup>, 1.5x10<sup>-5</sup> and 3.0x10<sup>-5</sup>M. Tests were also conducted to observe the influence of catalyst weight, being investigated the concentrations of 0.1, 0.4 and 0.8 g/L. To find an optimal pH for the photocatalytic activity, tests were made at pH 4.0, 7.0 and 8.0 using appropriate aliquots of 0.1 M NaOH solution, and 0.1 M HCl solution. After the addition of PS/SnO<sub>2</sub> photocatalyst, the system was kept in the dark for 30 minutes at 25 °C, to ensure adsorption/desorption equilibrium of RhB on the photocatalyst surface and then irradiated with 3 × Phillips 15 W lamps (45 W), emitting UV radiation  $\lambda = 254$  nm. The lamps were set in a closed photocatalytic chamber of dimensions 60 cm (height) x 80 cm (width) x 90 cm (height). Aliquots were collected for spectrophotometric analysis every 10 minutes during 70 min under irradiation. Assays were performed in triplicate and samples were stored in test

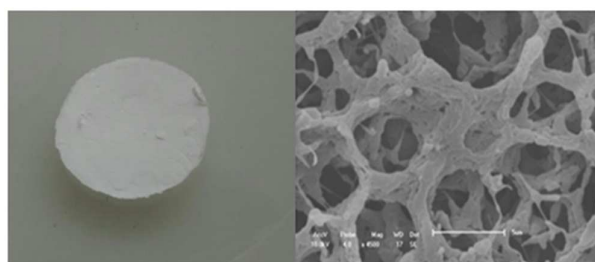


1  
2  
3 tubes covered with aluminum foil to prevent light exposure, prior to spectrophotometric  
4  
5 analysis in the UV-visible region using a Shimadzu UV-2550 spectrometer.  
6  
7  
8  
9

### 10 11 3. RESULTS AND DISCUSSION

#### 12 13 3.1 PS foams

14  
15 The foams made from PS solutions at an initial concentration of 1.0% showed no  
16  
17 structure or strength, and resulted in fragile materials not suitable for this work. On the  
18  
19 other hand, the foams obtained from PS concentrations of 2.5 and 5.0% were compact,  
20  
21 and robust, so these were used to make PS/SnO<sub>2</sub> nanofoams (Figure 1).  
22  
23



33  
34 **Figure 1.** Left: Foam of PS (2.5), rigid and compact with average size of 3 cm. Right:  
35  
36 PS foam SEM image showing porous structure (bar scale = 5 μm).  
37  
38

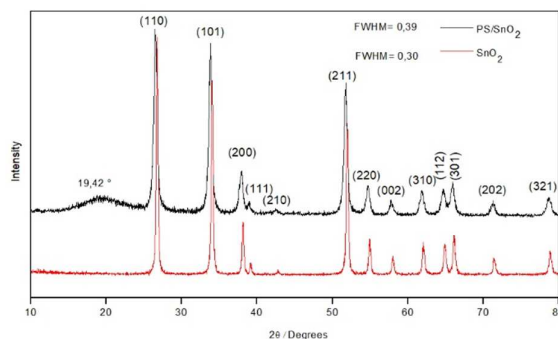
39 During freeze-drying, given the samples remain below the glass transition  
40  
41 temperature of PS and melting point of CH, the solvent is removed to give porous  
42  
43 structures, with voids resulting from the removal of the CH solvent. The temperature at  
44  
45 which the dispersion in cyclohexane solvent is annealed for freezing is a primary factor  
46  
47 in the preparation of foams. The mixture should be at or above the theta temperature at  
48  
49 which mixtures form stable dispersions. At the theta temperature, the polymer chains  
50  
51 behave as ideal chains, having no preferential interactions with solvent or between chain  
52  
53 segments, behaving similarly to the melt state<sup>18,24</sup>. For temperatures below theta the  
54  
55 systems separate into two phases, one polymer rich and another polymer poor, which  
56  
57  
58  
59  
60

1  
2  
3 results in poor distribution of polymer in the porous structure of the final material<sup>18,25-27</sup>.  
4  
5 The theta temperature for PS/CH solutions is 36 °C<sup>18</sup>, and as described by Aubert and  
6  
7 Clough, the porous foam structures form in such a way (below theta) that the mixtures  
8  
9 would separate into two liquid phases, one PS-rich and other CH-rich. However, if these  
10  
11 mixtures are very quickly quench-frozen at temperatures below the melting point of CH  
12  
13 (+6 °C), nuclei of the liquid PS/CH-rich phase will become entrapped inside the frozen  
14  
15 CH-rich phase. Once a frozen CH-rich phase is nucleated in the freeze-drying process,  
16  
17 more CH migrates from the liquid PS-rich phase to the CH-rich phase, enriching further  
18  
19 the polymer concentration. That means the porosity of the final material will be a  
20  
21 function of the polymer concentration in the PS-rich domains. The final state of the  
22  
23 system is a CH-free solid porous PS material with an open cell-like foam structure<sup>18</sup>.  
24  
25 Another way of thinking about this freeze-drying process is that the frozen CH solvent  
26  
27 acts as a porogen to produce porous materials<sup>28</sup>.  
28  
29

### 30 3.2 SnO<sub>2</sub> nanoparticles and PS/SnO<sub>2</sub>

31  
32  
33 Figure 2 shows the XRD patterns for PS(2.5)/SnO<sub>2</sub>(5)-700 nanofoams and for the  
34  
35 SnO<sub>2</sub>-700 nanoparticles to highlight possible changes resulting from the incorporation  
36  
37 of nanoparticles into the polymeric foams. Being an amorphous polymer polystyrene  
38  
39 does not give rise to any significant XRD peaks<sup>29</sup>. X-ray diffraction from the  
40  
41 PS(2.5)/SnO<sub>2</sub>(5)-700 system proves tin oxide is incorporated in the nanofoam, and also  
42  
43 shows changes related to the presence of polystyrene. Comparing before and after  
44  
45 impregnation in PS foams, observation of the tetragonal phase (JCPDS - 88-0287)  
46  
47 indicates SnO<sub>2</sub> in the nanofoams<sup>30-32</sup>, which is observed for all samples. The band over  
48  
49 the region between 10 and 19.42° indicates formation of a non-crystalline structure,  
50  
51 which, according to Botan *et al.*<sup>33</sup>, is characteristic of polystyrene, additionally there is  
52  
53 small peak broadening for SnO<sub>2</sub>, as shown in the *Full Width at Half Maximum* (FWHM)  
54  
55  
56  
57  
58  
59  
60

values calculated from the XRD pattern. The amorphous polymer structure was maintained during the TIPS process, whereas the FWHM broadening for SnO<sub>2</sub> can be attributed to a better dispersion of the SnO<sub>2</sub> decreasing the agglomeration. This observation suggests possible changes in photocatalytic activity may be observed (see below), due to increased surface area.



**Figure 2.** X-Ray diffractograms of PS(2.5)/SnO<sub>2</sub>(5)-700 nanofoams compared to bare SnO<sub>2</sub>-700.

The FWHM values were obtained using the PeakFit software and the crystallite sizes were calculated using the Scherrer equation<sup>34</sup>, calculated using the most intense peak of in the XRD (110). The results are summarized in Table 1, showing that nanoparticles were obtained in all cases and their sizes increase as the initial samples are heated to higher temperatures.

Table 1: Structural and electronic parameters for SnO<sub>2</sub> and PS/SnO<sub>2</sub>

SnO <sub>2</sub> (T °C)	Crystallite Size (D <sub>c</sub> , nm)	FWHM – SnO <sub>2</sub>	S <sub>B.E.T</sub> (m <sup>2</sup> /g) – SnO <sub>2</sub>	Band gap (eV) – SnO <sub>2</sub>	S <sub>B.E.T</sub> PS(2.5)/SnO <sub>2</sub> (5)- 700
700	22.7	0.39	15.2	3.0	48.7

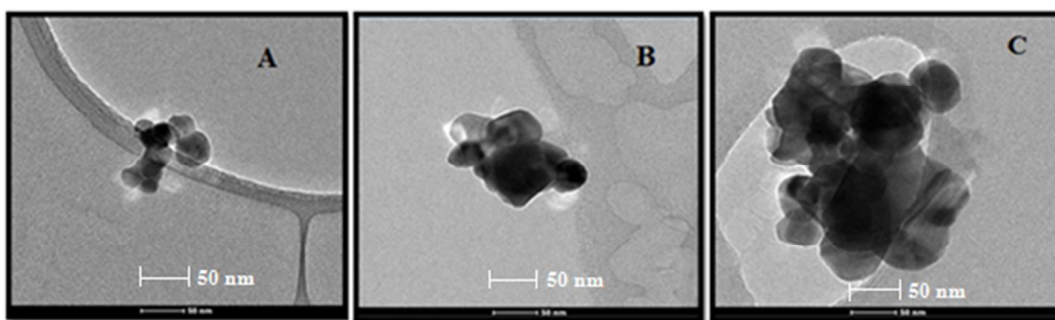
<b>800</b>	27.0	0.33	14.9	2.9	34.8
<b>900</b>	30.8	0.29	11.6	2.8	28.1

The results of the FWHM and crystallite size values were obtained from the XRD patterns and are presented in Table 1 (see also Fig. S2 in Supporting Information). It is observed that increasing temperature promotes crystallite growth and lowers FWHM values. This is consistent with temperature-induced crystallite nucleation occurring initially, and then growth with increasing temperature (due to high coalescence). As a consequence, the material is of higher crystallinity, according to the changes in FWHM values as a function of temperature, and smaller surface areas as indicated by the BET analyzes. According to Table 1, it is observed that increasing calcination temperature leads to lower surface area. The higher specific surface area at 700 °C may be attributed to more defects in SnO<sub>2</sub>, which leads to tiny pores. At 900 °C these pores collapse, resulting in a smaller specific surface area<sup>35</sup>. Majumdar *et al.*<sup>36</sup> also conducted studies involving other synthetic methods for SnO<sub>2</sub>, and concluded that heat treatment influences directly surface properties, just like as observed here. At high temperatures the studied oxides showed a significant decrease in surface area. It is suggested that calcination at 800° C and 900 ° C has led to further contraction of the porous structure, where some pores join to form larger voids resulting in reduction of surface area. These data corroborate the micrographs shown in TEM (see below), which indicate that one of the reasons for the change in surface area is the formation of aggregates/agglomerates with increasing calcination temperature, and the concurrent increase in particle size. Nonetheless, surface area values are still higher in the nanofoams than for the bare oxides. Studies conducted by Rumyantseva *et al.*<sup>37</sup> confirmed similar results to this study. They assessed the effect of heat treatment on the specific surface area and pore

1  
2  
3 volume for SnO<sub>2</sub> nanocrystalline powders and found that oxide calcined at lower  
4  
5 temperatures resulted in greater surface area compared to higher temperature treatment.  
6

7 The band gap energies for each sample were calculated using Tauc methodology<sup>38</sup>.  
8  
9 The observed increase in absorbance with temperature is attributed narrowing of the  
10  
11 SnO<sub>2</sub> band gap<sup>28</sup> owing to bigger particles sizes and surface roughness<sup>39</sup>, leading to a  
12  
13 higher absorbance for SnO<sub>2</sub>-900 (Fig. S3 in Supporting Information).  
14

15  
16 Figure 3 shows HRTEM images for the SnO<sub>2</sub>-700, 800 and 900. They show that the  
17  
18 oxide powder contains nanoparticles with sizes varying roughly between 20 and 80 nm,  
19  
20 depending on the heat treatment temperature. For agglomerated particles it was not  
21  
22 possible to obtain accurate size measurements, however, the effect of temperature on  
23  
24 particle size is clear<sup>35</sup>.  
25



37  
38 **Figure 3.** HRTEM images of the annealed SnO<sub>2</sub> nanoparticles at A) 700 B) 800, and C)  
39  
40 900 °C. (Size bar = 50 nm)  
41

42  
43 The same trends in nanoparticle size can be seen comparing these HRTEM results  
44  
45 with crystallite sizes values obtained from X-Ray Diffraction.  
46

### 47 3.3 Photocatalysis

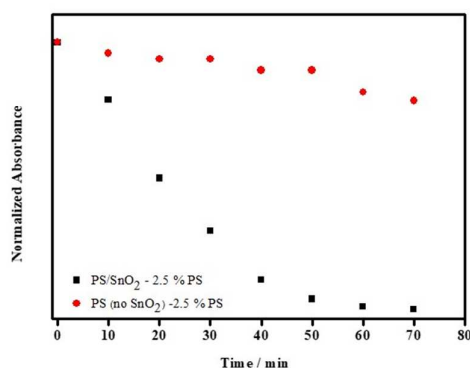
#### 48 3.3.1 Effects of polystyrene concentration

49  
50  
51 Initially, the influence of RhB concentration was investigated, and it was observed  
52  
53 that low concentrations favored photocatalytic degradation (Fig. S4 in Supporting  
54  
55 Information). Hu et al.<sup>40</sup> presented similar results for photocatalytic RhB degradation as  
56  
57  
58  
59  
60

1  
2  
3 a function of concentration. Within 300 minutes, RhB degradation decreased from 94 to  
4  
5 70% when the initial RhB concentration increased from 10 to 25 mg. L<sup>-1</sup>, consistent  
6  
7 with an “inner filter-type” effect leading to a lower rate of photocatalytic degradation<sup>40</sup>.  
8  
9 Tests were carried out to find the optimum PS level for RhB degradation, and Figure S4  
10  
11 shows photodegradation of RhB (1.5x10<sup>-5</sup>mol/L) using PS(2.5)/SnO<sub>2</sub>(2.5)-700 and  
12  
13 PS(5)/SnO<sub>2</sub>(2.5)-700 nanofoams.

14  
15  
16 According to Fig. S5 in Supporting Information the photocatalytic degradation of  
17  
18 RhB in the presence of PS(2.5)/SnO<sub>2</sub>(2.5)-700 appeared to be more effective compared  
19  
20 to the PS(5)/SnO<sub>2</sub>(2.5)-700 system. This could be owing to greater surface coverage of  
21  
22 SnO<sub>2</sub> nanoparticles by PS at the higher level decreases accessibility to SnO<sub>2</sub> active sites,  
23  
24 which in turn reduces photoactivity<sup>41</sup>.

25  
26  
27 As shown in Figure 4, for a foam in the absence of SnO<sub>2</sub> nanoparticles there is only a  
28  
29 maximum of 25% degradation after 70 min, showing that the presence of SnO<sub>2</sub> particles  
30  
31 is needed for good photocatalytic performance. An increase in reaction rate leading to  
32  
33 an almost complete RhB degradation (97.1%) in 70 minutes was observed with  
34  
35 PS(2.5)/SnO<sub>2</sub>(2.5)-700 nanofoam.



36  
37  
38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52 **Figure 4.** Photocatalytic degradation of RhB for PS(2.5)/SnO<sub>2</sub>(2.5)-700 in comparison  
53  
54 to only PS(2.5).  
55

1  
2  
3 Thus, the presence and amount of SnO<sub>2</sub> greatly influence photocatalytic degradation.  
4  
5 Regarding the mass concentration of photocatalyst Fig. S6 in Supporting Information  
6  
7 shows that 0.4 g / L is the most appropriate of those tested. According to studies by  
8  
9 Leea and Adesina<sup>42</sup>, an increase in catalyst mass may cause light scattering and hence  
10  
11 reduce the catalytic activity.  
12  
13  
14

### 15 3.3.2 Photodegradation pathway

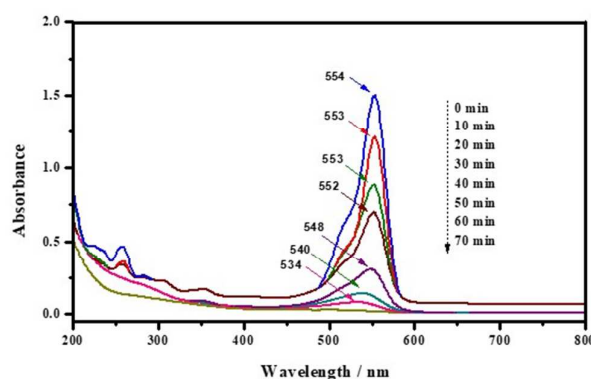
16  
17 The photocatalytic degradation of RhB was studied as a model reaction to assess these  
18  
19 photocatalytic nanofoams for applications in environmental remediation. Changes in the  
20  
21 main UV-vis absorption spectra as a function of time and appropriate concentrations  
22  
23 were followed to track evolution of the RhB degradation products shown in Supporting  
24  
25 Information.  
26  
27

28  
29 To confirm that the photocatalytic activity is due to the photocatalyst only, we carried  
30  
31 out blank experiments without a catalyst under light (photolysis)<sup>56</sup>, as shown in Fig. S7  
32  
33 in Supporting Information. It was observed that after 70 min, 20% of RhB was  
34  
35 degraded. One explanation for this behaviour is that RhB dye can absorb UV radiation,  
36  
37 and the energy is sufficient to break bonds in the chromophore groups. Photolysis,  
38  
39 however, represents a small fraction of the photodegradation process and does not lead  
40  
41 to effective and efficient degradation.  
42  
43

44 The spectral changes during the photocatalytic degradation of RhB under UV  
45  
46 irradiation using SnO<sub>2</sub> particles are shown in Figure 5. As can be seen, the characteristic  
47  
48 absorption band of RhB decreased rapidly, simultaneously with a hypsochromic shift in  
49  
50 the absorption maximum from 554 to 534 nm. Previous studies have also reported  
51  
52 similar changes using NaBiO<sub>3</sub> as photocatalyst<sup>43</sup>. The sharp decrease in the maximum  
53  
54 absorption results from cleavage of the chromophore, whereas the peak displacement is  
55  
56  
57  
58  
59  
60

1  
2  
3 related to a N-de-ethylation mechanism: RhB is considered completely N-de-ethylated  
4 when displacement of the absorption maximum moves from 554 to 498 nm<sup>44</sup>. Thus, it  
5 was concluded that N-de-ethylation of RhB does occur in this case, however, it is not  
6 the predominant mechanism. It can be seen in Figure 5 that absorbance decreases more  
7 quickly than the maximum wavelength is shifted. This behavior clearly indicates that  
8 the chromophore cleavage mechanism predominates over N-de-ethylation when using  
9 SnO<sub>2</sub> as photocatalyst.  
10  
11  
12  
13  
14  
15  
16

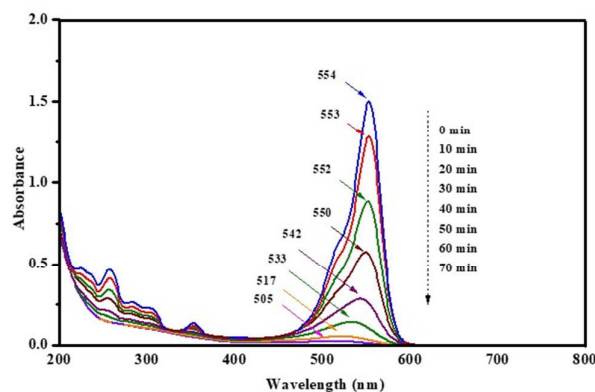
17  
18 The effect of temperature treatment on catalyst performance for RhB  
19 degradation was evaluated (discussed below), and kinetic parameters for SnO<sub>2</sub>-700,  
20 SnO<sub>2</sub>-800 and SnO<sub>2</sub>-900 samples are listed in Table 2. Catalysts treated at a lower  
21 temperature showed higher activity, and this can be understood as result of smaller  
22 nanoparticle size and subsequently larger surface area. Therefore, SnO<sub>2</sub>-700 was  
23 selected to make PS/SnO<sub>2</sub> nanofoams for applications testing. It was noted that  
24 changing pH lead to no significant effect on dye degradation, although a small decrease  
25 in degradation efficiency at higher pH was noted (Fig. S8 in Supporting Information).  
26  
27 Hence pH 4.0 was selected as the optimum condition for the reaction medium.  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37



38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51 **Figure 5.** UV-vis absorption spectra of RhB degradation in the presence of bare SnO<sub>2</sub>  
52 as a function of time (catalyst dosage: 0.4 g.L<sup>-1</sup>, C<sub>RhB</sub>:1.0x10<sup>-5</sup> mol/L, pH: 4.0)  
53  
54  
55  
56  
57  
58  
59  
60



1  
2  
3 The spectral changes during RhB photodegradation in the presence of PS/SnO<sub>2</sub>  
4 nanofoams are shown in Figure 6. The N-de-ethylation caused a shift in peak  
5 wavelength, resulting from the formation of a series of intermediate N-de-ethylation  
6 wavelength, resulting from the formation of a series of intermediate N-de-ethylation  
7 products of RhB, i.e., the ethyl groups were removed one by one, resulting in gradual  
8 changes in the peak maximum wavelength into the blue region<sup>45</sup>. The dye solution color  
9 disappeared (after 60 min irradiation), indicating that the chromophore was fully  
10 degraded. The generated products were identified as shown in Fig. S9 and S10 in  
11 Supporting Information, consistent with three N-de-ethylations. It can be observed that  
12 PS/SnO<sub>2</sub> nanofoams caused a higher N-deethylation in comparison with SnO<sub>2</sub>-only  
13 suspensions.  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23  
24  
25  
26  
27  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38  
39



40  
41 **Figure 6.** UV-vis absorption spectra of RhB up to 70 minutes in the presence of  
42 PS(2.5)/SnO<sub>2</sub>(2.5)-700 nanofoams (catalyst dosage: 0.4 g.L<sup>-1</sup>, C<sub>RhB</sub>:1.0x10<sup>-5</sup> mol/L,  
43 pH: 4.0).  
44  
45  
46

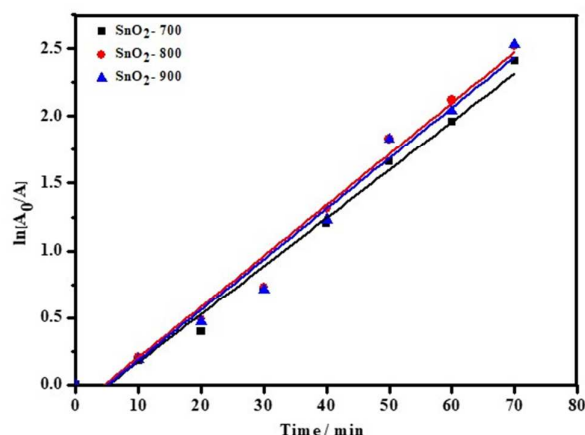
47 Comparing the PS/SnO<sub>2</sub> and SnO<sub>2</sub> systems, the degradation mechanisms might differ,  
48 probably due to a stronger adsorption of RhB on the surface of PS/SnO<sub>2</sub> than on bare  
49 SnO<sub>2</sub>, as discussed below. In general, these results showed that the PS(2.5)/SnO<sub>2</sub>(2.5)-  
50 700 systems promote two competitive processes for the photodegradation of RhB, N-  
51 de-ethylation or cleavage of the chromophore. If both pathways occur simultaneously  
52  
53  
54  
55  
56  
57  
58  
59  
60

1  
2  
3 this can enhance the rate of RhB degradation into colorless byproducts, and eventually  
4 to CO<sub>2</sub> and H<sub>2</sub>O<sup>46</sup>.

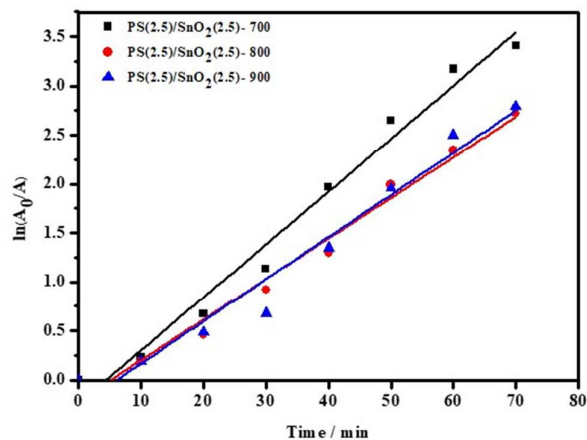
5  
6  
7 For SnO<sub>2</sub> only nanoparticle suspensions, the N-de-ethylation route was relatively less  
8 effective compared to cleavage, as noted. In contrast, when using PS/SnO<sub>2</sub>  
9 photocatalyst nanofoams, it was observed that, in addition to chromophore cleavage, a  
10 N-de-ethylation route opens up, consistent with the hypochromic shift to the blue region  
11 of the spectrum. Ortelli *et al.*<sup>45</sup> investigated degradation mechanisms of RhB, observing  
12 that the surface interactions semiconductor/dye dominate when the catalyst is well  
13 dispersed, leading to a degradation fully driven by OH• free radicals. In contrast, when  
14 the catalyst is supported or immobilized, substrate/dye interactions govern the N-de-  
15 ethylation process, driven by OH• radicals adsorbed on the surface.  
16  
17  
18  
19  
20  
21  
22  
23  
24  
25

### 26 3.3.3 Kinetics of RhB degradation

27  
28 To better understand dye photodegradation the reaction kinetics were studied.  
29 Considering that the radiation source used has constant flux, and the number of  
30 photocatalyst active sites is solely dependent on the surface area, pseudo first-order  
31 kinetics are expected, provided there is no catalyst poisoning during the process.  
32  
33  
34  
35  
36  
37



38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53 **Figure 7.** RhB photodegradation kinetics for SnO<sub>2</sub> treated at various temperatures  
54 (catalyst dosage: 0.4 g.L<sup>-1</sup>, C<sub>RhB</sub>: 1.0x10<sup>-5</sup> mol/L, pH: 4.59 (unadjusted))  
55  
56  
57



**Figure 8.** RhB photodegradation kinetics for PS/SnO<sub>2</sub> nanofoams at various SnO<sub>2</sub> temperature treatments (catalyst dosage: 0.4 g.L<sup>-1</sup>, C<sub>RhB</sub>:1.0x10<sup>-5</sup> mol/L, pH: 4.60 (unadjusted))

Figure 7 and 8 show plots of  $\ln(A_0/A)$ , s vs reaction time, for the system with bare SnO<sub>2</sub> and PS/SnO<sub>2</sub> nanofoams, respectively.  $A_0$  is the RhB initial absorbance and  $A$  is the absorbance at the time  $t$ . (Assuming the Lambert-Beer law applies, absorbance is directly related to concentration.) Comparing all the nanofoams the PS(2.5)/SnO<sub>2</sub>(2.5)-700 shows a higher photodegradation rate constant of 0.054 min<sup>-1</sup>, as can be seen in Table 2, and this can be attributed to greater surface areas in nanoparticle-impregnated nanofoams SnO<sub>2</sub>-700. Figures S11 and S12 in Supporting Information, show plots of  $A_0/A$  for degradation of RhB using SnO<sub>2</sub> only photocatalyst and SnO<sub>2</sub>/PS foams, respectively. Comparison of the results obtained here with the available literature is not simple, since the process of photodegradation of dyes with UV and semiconductors depends on several factors, such as nature and concentration of the organic contaminant, concentration and type of semiconductor, light source, lamp power and intensity, pH, temperature, etc.<sup>47</sup>. In general, each piece of work employs different conditions,

however, given these limitations an appraisal of the performance of different system is made below.

Zhu et al.<sup>48</sup>, studied the photocatalytic activity of SnO<sub>2</sub> microspheres for RhB degradation. They also observed pseudo-first order kinetics, and rate constants can be compared with those presented here. Zhu et al. analyzed different pH values, 2.91, 6.18 and 9.05, giving rate constants 0.0082, 0.0273 and 0.0195 min<sup>-1</sup>, respectively. According to<sup>48</sup> these results may be associated with N-de-ethylation of RhB, since, as shown in Fig. 8, N-de-ethylation cannot be totally avoided and will occur in parallel with the direct degradation pathway. In addition, it should be noted that RhB photodegradation with two different catalysts (SnO<sub>2</sub> and PS/SnO<sub>2</sub>) shows two changes: a rapid decrease in absorbance, and also a hypochromic shift of  $\lambda_{\max}$ , which was confirmed by Watanabe<sup>49</sup> as being due to the N-de-ethylation of RhB, where the product is RhB ( $\lambda_{\max} = 498$  nm). In the present work, with two types of different catalysts (SnO<sub>2</sub> and PS/SnO<sub>2</sub>), the blue shift was 15 and 44 nm at pH 4.0 consistent with the state of the catalyst dispersion affecting the photocatalytic degradation.

**Table 2.** Values of the degradation rate constants and the extent of degradation for all the systems studied under optimal experimental conditions.  $r^2$  were all >0.97.

Photocatalyst	S <sub>B.E.T</sub> (m <sup>2</sup> /g)	pH	k (min <sup>-1</sup> )	% (Photobleaching)
SnO <sub>2</sub> -700	15.16	4.0	0.037	94.0
SnO <sub>2</sub> -800	14.86	4.0	0.036	91.8
SnO <sub>2</sub> -900	11.63	4.0	0.034	88.1
PS(2.5)/SnO <sub>2</sub> (2.5)-700	48.69	4.0	0.054	98.2
PS(2.5)/SnO <sub>2</sub> (2.5)-800	34.83	4.0	0.041	98.1
PS(2.5)/SnO <sub>2</sub> (2.5)-900	28.12	4.0	0.043	96.1

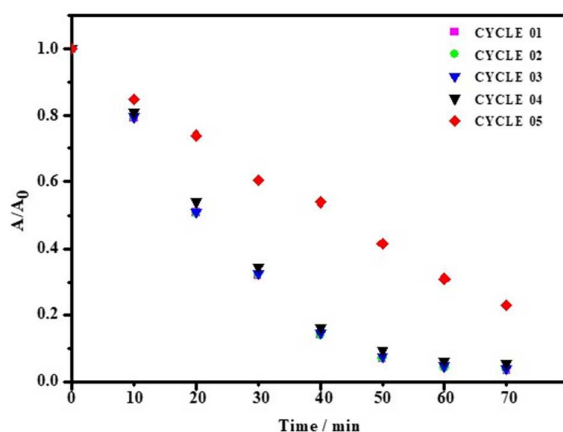
Clearly, the degradation efficiency using nanofoams was significantly improved over the bare nanoparticles, since nanofoams provide higher surface area and porosity, and also allow easy recovery of the catalyst. This structure can affect the transport pathways, allowing dye molecules to reach active sites, meaning that porosity can improve the photocatalytic efficiency<sup>50</sup>. These results are comparable, and even better, than some reported ones, as shown in Table 3. Note that it is not easy to compare these results, since besides RhB:SnO<sub>2</sub> molar ratio, irradiation power is an important factor, and the 60 W lamp used here is smaller than some others reported, such as 300 W for the work of Zhu et. al<sup>48</sup>.

**Table 3.** Kinetic constants obtained here (best system) compared to other literature reports.

System	RhB/SnO <sub>2</sub> (mol:mol)	k (min <sup>-1</sup> )	Irradiation	Ref.
SnO <sub>2</sub> - nanoporous (photochemical synthesis)	1:19	0.027	UV tube-like lamp. (254 nm, 8 W)	CHEN <i>et al.</i> , (2014) <sup>51</sup>
SnO <sub>2</sub> - microspheres (solvothermal method)	1: 10	0.0273	high pressure mercury lamp (300 W)	ZHU <i>et al.</i> , (2014) <sup>48</sup>
SnO <sub>2</sub> supported in mesoporous TiO <sub>2</sub> (sol-gel method)	1:15	0.010	UV lamp.	Abdel-Messih <i>et al.</i> (2013) <sup>52</sup>
SnO <sub>2</sub> /polystyrene (polymer precursors)	1: 88	0.054	UVC germicidal lamp.(254 nm, 60 W)	This work

### 3.3.4 Catalyst reuse

1  
2  
3 A large amount of disposable PS by-products cause serious environmental problems,  
4 this waste has been dubbed “white pollution”. Polystyrene waste is not economically  
5 viable for recycling, but can be employed to remove other pollutants present in various  
6 environments such as water. If the produced material proves highly effective for  
7 degradation of pollutants, like dyes, it could have important industrial applications in  
8 water remediation systems. An attractive aspect of the work described here is the  
9 potential to reuse waste PS in environmental remediation applications<sup>53</sup>. To further  
10 investigate the reuse of PS(2.5)/SnO<sub>2</sub>(2.5)-700 nanofoams, they were subjected to five  
11 successive use cycles for photodegradation of rhodamine B. Figure 9 shows the changes  
12 in catalyst effectiveness after five reuse cycles.  
13  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23



24  
25  
26  
27  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38  
39 **Figure 9.** Reuse of a PS(2.5)/SnO<sub>2</sub>(2.5)-700 nanofoam photocatalyst for RhB  
40 degradation.  
41  
42  
43  
44  
45  
46

47 The first run showed 98.3% of RhB degradation; cycle 2, 98.3%; cycle 3 showed  
48 97.8% degradation; cycle 4, 96.5% and the fifth degraded 82% of the initial dye.  
49 Unfortunately, there was a decrease in nanofoam activity from the fifth reuse. These  
50 results indicate that nanofoams exhibit stable photocatalytic activity, which is a very  
51 important factor for practical applications<sup>44</sup>, and also demonstrate the feasibility of  
52  
53  
54  
55  
56  
57  
58  
59  
60

1  
2  
3 waste polystyrene reuse for environmental remediation of effluent dyes. It is possible  
4 that successive reuse and washing brings about a modification of the interfacial  
5 structure, altering its capacity for catalysis. An alternative reason is UV-induced  
6 photodegradation of the PS structure. Altin and Sokmen<sup>53</sup>, also carried out study of  
7 reuse of catalysts based on polystyrene. The photocatalytic properties of the produced  
8 material were tested for degradation and removal of methylene blue (MB). It was noted  
9 the TiO<sub>2</sub>-PS activity decreased nearly 50% for MB removal after 5 cycles, and the TiO<sub>2</sub>-  
10 PS catalysts were still effective albeit with lower dye removal rates.  
11  
12  
13  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23  
24  
25  
26  
27  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

#### 4. CONCLUSIONS

Polystyrene was successfully converted from waste commercial pellets into nanofoams using a thermally induced phase separation (TIPS) method. The incorporation of SnO<sub>2</sub> nanoparticles generated composite nanofoams, being novel photocatalysts which were assessed for potential environmental applications by following degradation of a model pollutant dye RhB. The functional nanofoams made from SnO<sub>2</sub> nanoparticles annealed at 700°C resulted in higher photocatalytic activity in this application than if treated at a higher temperature of 900 °C. This temperature effect was accounted for in terms of a decrease in effective nanofoam surface area after annealing at the higher temperature. Irradiating RhB solutions in the presence of PS/SnO<sub>2</sub> nanofoam photocatalysts resulted in a final degradation of 98.2%, which is better than reported in literature for SnO<sub>2</sub> nanoparticles<sup>54</sup>. This improved photocatalytic performance of the PS/SnO<sub>2</sub> nanofoams was attributed to larger surface areas and greater porosities compared to SnO<sub>2</sub> nanoparticles. The porous nanofoam structures favor interactions between dye molecules and OH radicals on the foam surfaces, leading to dye degradation, principally by N-de-ethylation<sup>45</sup>. As such, porous structures offer more catalytically active sites, and hence improved surface stability of electrons and holes to generate hydroxyl radicals, which may react to degrade contaminants. Additionally, the porous channel structure may facilitate better transport of contaminant molecules to reach catalytically active sites<sup>55</sup>. Reaction kinetics analyses reinforce that incorporation of the SnO<sub>2</sub> nanoparticles into the PS foams improves RhB photodegradation, with rate constants as high as 0.054 min<sup>-1</sup>, being similar/higher than other reported systems, which are sometimes chemically more complex than these nanofoams<sup>48,51,52</sup>. Reuse of the PS/SnO<sub>2</sub> nanofoams was demonstrated over five photocatalytic cycles, showing retention of activity for up to 4 cycles, but loss of



1  
2  
3 efficiency only after the fifth cycle. This proof of principle work offers an interesting  
4  
5 two-fold environmental advantage: firstly, the feasibility to reuse polymeric waste to  
6  
7 fabricate new materials, which secondly, can be employed to tackle another  
8  
9 environmental issue, water treatment. New possibilities are being studied, with  
10  
11 incorporation of catalysts (gold, silver, gold/palladium) and adsorbents (dolomite) to  
12  
13 evaluate new catalytic transformations and removal of pollutants from wastewater and  
14  
15 underground water.  
16  
17  
18  
19

## 20 ASSOCIATED CONTENT

### 21 22 **Supporting Information.**

23  
24  
25  
26 The Supporting Information is available free of charge on the ACS Publications website  
27  
28  
29 at DOI: 10.1021/acsami.xxxxxx.  
30

31  
32 X-ray diffractograms of SnO<sub>2</sub>, crystallite size of SnO<sub>2</sub>, Diffuse reflectance spectra of  
33  
34 SnO<sub>2</sub>, RhB photodegradation related to SnO<sub>2</sub> concentration, RhB photodegradation  
35  
36 related to PS concentration, RhB photodegradation related to catalyst mass, RhB  
37  
38 photodegradation related to pH, Relation between RhB maximum wavelength and  
39  
40 absorbance with time for SnO<sub>2</sub> and PS/SnO<sub>2</sub>, A/A<sub>0</sub> vs time plots of RhB  
41  
42 photodegradation over SnO<sub>2</sub> and PS/SnO<sub>2</sub> catalysts.  
43  
44  
45  
46

## 47 AUTHOR INFORMATION

### 48 49 **Corresponding Author**

50  
51  
52 Rodrigo José de Oliveira  
53

54  
55 +55 83 99816 2342, [rodrigo@cct.uepb.edu.br](mailto:rodrigo@cct.uepb.edu.br)  
56  
57

**Corresponding Author**

Julian Eastoe

Phone +44-117-928-9180, [Julian.Eastoe@bris.ac.uk](mailto:Julian.Eastoe@bris.ac.uk)

**Present Address**

<sup>Δ</sup> Instituto de Química e Biotecnologia, Universidade Federal de Alagoas, Brazil.

**ORCID**

Rodrigo José de Oliveira: 0000-0002-8753-2661

Julian Eastoe: 0000-0001-5706-8792

**Author Contributions**

All authors have given approval to the final version of the manuscript and contributed to the final version of the manuscript.

**ACKNOWLEDGEMENTS**

RJdeO acknowledges CAPES agency for financial support, University of Bristol UK for a Benjamin Meaker Research Fellowship, CETENE for transmission electron microscopy images, and IQ-UnB for surface area analysis and XRD analysis. GCdeA acknowledges CAPES agency for a student scholarship.

**REFERENCES**

- (1) Chowdhury, S.; Balasubramanian, R. Graphene/semiconductor Nanocomposites (GSNs) for Heterogeneous Photocatalytic Decolorization of Wastewaters Contaminated with Synthetic Dyes: A Review. *Appl. Catal. B Environ.* **2014**,

- 1  
2  
3 160–161, 307–324.  
4  
5 (2) Srinivasan, A.; Viraraghavan, T. Decolorization of Dye Wastewaters by  
6  
7 Biosorbents: A Review. *J. Environ. Manage.* **2010**, *91*, 1915–1929.  
8  
9 (3) Nagaraja, R.; Kottam, N.; Girija, C. R.; Nagabhushana, B. M. Photocatalytic  
10  
11 Degradation of Rhodamine B Dye under UV/solar Light Using ZnO Nanopowder  
12  
13 Synthesized by Solution Combustion Route. *Powder Technol.* **2012**, *215–216*,  
14  
15 91–97.  
16  
17 (4) Pascariu, P.; Airinei, A.; Olaru, N.; Olaru, L.; Nica, V. Photocatalytic  
18  
19 Degradation of Rhodamine B Dye Using ZnO–SnO<sub>2</sub> Electrospun Ceramic  
20  
21 Nanofibers. *Ceram. Int.* **2016**, *42*, 6775–6781.  
22  
23 (5) Saharan, V. *Advanced Oxidation Technologies for Wastewater Treatment: An*  
24  
25 *Overview*; 2015.  
26  
27 (6) Ben Ali, M.; Barka-Bouaifel, F.; Sieber, B.; Elhouichet, H.; Addad, A.;  
28  
29 Boussekey, L.; Férid, M.; Boukherroub, R. Preparation and Characterization of  
30  
31 Ni-Doped ZnO–SnO<sub>2</sub> Nanocomposites: Application in Photocatalysis.  
32  
33 *Superlattices Microstruct.* **2016**, *91*, 225–237.  
34  
35 (7) Glaze, W. H.; Kang, J.-W.; Chapin, D. H. The Chemistry of Water Treatment  
36  
37 Processes Involving Ozone, Hydrogen Peroxide and Ultraviolet Radiation. *Ozone*  
38  
39 *Sci. Eng.* **1987**, *9*, 335–352.  
40  
41 (8) Townsend, T. K.; Sabio, E. M.; Browning, N. D.; Osterloh, F. E. Photocatalytic  
42  
43 Water Oxidation with Suspended Alpha-Fe<sub>2</sub>O<sub>3</sub> Particles-Effects of Nanoscaling.  
44  
45 *Energy Environ. Sci.* **2011**, *4*, 4270–4275.  
46  
47 (9) Gaya, U. I.; Abdullah, A. H. Heterogeneous Photocatalytic Degradation of  
48  
49 Organic Contaminants over Titanium Dioxide: A Review of Fundamentals,  
50  
51 Progress and Problems. *J. Photochem. Photobiol. C Photochem. Rev.* **2008**, *9*, 1–  
52  
53  
54  
55  
56  
57  
58  
59  
60

- 1  
2  
3 12.  
4  
5 (10) Kohtani, S.; Makino, S.; Kudo, A.; Tokumura, K.; Ishigaki, Y.; Matsunaga, T.;  
6  
7 Nikaïdo, O.; Hayakawa, K.; Nakagaki, R. Photocatalytic Degradation of 4-N-  
8  
9 Nonylphenol under Irradiation from Solar Simulator: Comparison between BiVO  
10  
11 4 and TiO<sub>2</sub> Photocatalysts. *Chem. Lett.* **2002**, *31*, 660–661.  
12  
13 (11) Robinson, T.; McMullan, G.; Marchant, R.; Nigam, P. Remediation of Dyes in  
14  
15 Textile Effluent: A Critical Review on Current Treatment Technologies with a  
16  
17 Proposed Alternative. *Bioresour. Technol.* **2001**, *77*, 247–255.  
18  
19 (12) Yasmina, M.; Mourad, K.; Mohammed, S. H.; Khaoula, C. Treatment  
20  
21 Heterogeneous Photocatalysis; Factors Influencing the Photocatalytic  
22  
23 Degradation by TiO<sub>2</sub>. *Energy Procedia* **2014**, *50*, 559–566.  
24  
25 (13) Benhaoua, B.; Abbas, S.; Rahal, A.; Benhaoua, A.; Aida, M. S. Effect of Film  
26  
27 Thickness on the Structural, Optical and Electrical Properties of SnO<sub>2</sub>: F Thin  
28  
29 Films Prepared by Spray Ultrasonic for Solar Cells Applications. *Superlattices*  
30  
31 *Microstruct.* **2015**, *83*, 78–88.  
32  
33 (14) Das, S.; Jayaraman, V. SnO<sub>2</sub>: A Comprehensive Review on Structures and Gas  
34  
35 Sensors. *Prog. Mater. Sci.* **2014**, *66*, 112–255.  
36  
37 (15) Enesca, A.; Baneto, M.; Perniu, D.; Isac, L.; Bogatu, C.; Duta, A. Solar-  
38  
39 Activated Tandem Thin Films Based on CuInS<sub>2</sub>, TiO<sub>2</sub> and SnO<sub>2</sub> in Optimized  
40  
41 Wastewater Treatment Processes. *Appl. Catal. B Environ.* **2016**, *186*, 69–76.  
42  
43 (16) Navalon, S.; Dhakshinamoorthy, A.; Alvaro, M.; Garcia, H. Metal Nanoparticles  
44  
45 Supported on Two-Dimensional Graphenes as Heterogeneous Catalysts. *Coord.*  
46  
47 *Chem. Rev.* **2016**, *312*, 99–148.  
48  
49 (17) Plantard, G.; Goetz, V.; Correia, F.; Cambon, J. P. Importance of a Medium's  
50  
51 Structure on Photocatalysis: Using TiO<sub>2</sub>□coated Foams. *Sol. Energy Mater. Sol.*  
52  
53  
54  
55  
56  
57  
58  
59  
60

- 1  
2  
3 *Cells* **2011**, *95*, 2437–2442.
- 4  
5 (18) Aubert, J. H.; Clough, R. L. Low-Density, Microcellular Polystyrene Foams.  
6  
7 *Polymer (Guildf)*. **1985**, *26*, 2047–2054.
- 8  
9 (19) Steytler, D. C.; Robinson, B. H.; Eastoe, J.; Ibel, K.; Dore, J. C.; MacDonald, I.  
10  
11 Effects of Solidification of the Oil Phase on the Structure of Colloidal  
12  
13 Dispersions in Cyclohexane. *Langmuir* **1993**, *9*, 903–911.
- 14  
15 (20) de Melo, D. S.; Santos, M. R. C.; Santos, I. M. G.; Soledade, L. E. B.; Bernardi,  
16  
17 M. I. B.; Longo, E.; Souza, A. G. Thermal and Structural Investigation of  
18  
19 SnO<sub>2</sub>/Sb<sub>2</sub>O<sub>3</sub> Obtained by the Polymeric Precursor Method. *J. Therm. Anal.*  
20  
21 *Calorim.* **2007**, *87*, 697–701.
- 22  
23 (21) Pavia, F. C.; La Carrubba, V.; Piccarolo, S.; Brucato, V. Polymeric Scaffolds  
24  
25 Prepared via Thermally Induced Phase Separation: Tuning of Structure and  
26  
27 Morphology. *J. Biomed. Mater. Res. Part A* **2008**, *86A*, 459–466.
- 28  
29 (22) Lu, X.; Wei, A.; Fan, Q.; Wang, L.; Chen, P.; Dong, X.; Huang, W. Macroporous  
30  
31 Foam of Reduced Graphene Oxides Prepared by Lyophilization. *Mater. Res.*  
32  
33 *Bull.* **2012**, *47*, 4335–4339.
- 34  
35 (23) Nireesha, G. R.; Divya, L.; Sowmya, C.; Venkateshan, N.; Babu, M. N.;  
36  
37 Lavakumar, V. Lyophilization/freeze Drying—an Review. *Int. J. Nov. trends*  
38  
39 *Pharm. Sci.* **2013**, *3*, 87–98.
- 40  
41 (24) Alger, M. *Polymer Science Dictionary*; Springer Netherlands: Dordrecht, 2017.
- 42  
43 (25) He, Z.; Zhu, Z.; Li, J.; Zhou, J.; Wei, N. Characterization and Activity of  
44  
45 Mesoporous Titanium Dioxide Beads with High Surface Areas and Controllable  
46  
47 Pore Sizes. *J. Hazard. Mater.* **2011**, *190*, 133–139.
- 48  
49 (26) Wu, H.; Ma, J.; Zhang, C.; He, H. Effect of TiO<sub>2</sub> Calcination Temperature on the  
50  
51 Photocatalytic Oxidation of Gaseous NH<sub>3</sub>. *J. Environ. Sci.* **2014**, *26*, 673–682.
- 52  
53  
54  
55  
56  
57  
58  
59  
60

- 1  
2  
3 (27) Muthuvinayagam, A.; Melikechi, N.; Dennis Christy, P.; Sagayaraj, P.  
4  
5 Investigation on Mild Condition Preparation and Quantum Confinement Effects  
6  
7 in Semiconductor Nanocrystals of SnO<sub>2</sub>. *Phys. B Condens. Matter* **2010**, *405*,  
8  
9 1067–1070.  
10
- 11 (28) Yu, J.; Wang, G.; Cheng, B.; Zhou, M. Effects of Hydrothermal Temperature and  
12  
13 Time on the Photocatalytic Activity and Microstructures of Bimodal Mesoporous  
14  
15 TiO<sub>2</sub> Powders. *Appl. Catal. B Environ.* **2007**, *69*, 171–180.  
16  
17
- 18 (29) Moraes, S. B. de; Botan, R.; Lona, L. M. F. Synthesis and Characterization of  
19  
20 Polystyrene/layered Hydroxide Salt Nanocomposites. *Quim. Nova* **2014**, *37*, 18–  
21  
22 21.  
23
- 24 (30) Salavati-Niasari, M.; Mir, N.; Davar, F. Synthesis, Characterization and Optical  
25  
26 Properties of Tin Oxide Nanoclusters Prepared from a Novel Precursor via  
27  
28 Thermal Decomposition Route. *Inorganica Chim. Acta* **2010**, *363*, 1719–1726.  
29  
30
- 31 (31) Wan, X.; Ma, R.; Tie, S.; Lan, S. Effects of Calcination Temperatures and  
32  
33 Additives on the Photodegradation of Methylene Blue by Tin Dioxide  
34  
35 Nanocrystals. *Mater. Sci. Semicond. Process.* **2014**, *27*, 748–757.  
36  
37
- 38 (32) Sankar, C.; Ponnuswamy, V.; Manickam, M.; Mariappan, R.; Suresh, R.  
39  
40 Structural, Morphological, Optical and Gas Sensing Properties of Pure and Ru  
41  
42 Doped SnO<sub>2</sub> Thin Films by Nebulizer Spray Pyrolysis Technique. *Appl. Surf.*  
43  
44 *Sci.* **2015**, *349*, 931–939.  
45
- 46 (33) Botan, R.; Gonçalves, N. A.; Moraes, S. B. de; Lona, L. M. F.; Botan, R.;  
47  
48 Gonçalves, N. A.; Moraes, S. B. de; Lona, L. M. F. Preparation and Evaluation of  
49  
50 Polystyrene (PS) – Layered Double Hydroxide (LDH) ZnAl –  
51  
52 Organofunctionalized with Laurate/palmitate Nanocomposites. *Polímeros* **2015**,  
53  
54 *25*, 117–124.  
55  
56  
57  
58  
59  
60

- 1  
2  
3 (34) Malagù, C.; Carotta, M. C.; Giberti, A.; Guidi, V.; Martinelli, G.; Ponce, M. A.;  
4 Castro, M. S.; Aldao, C. M. Two Mechanisms of Conduction in Polycrystalline  
5 SnO<sub>2</sub>. *Sensors Actuators B Chem.* **2009**, *136*, 230–234.  
6  
7  
8  
9 (35) Hao, Y.; Jiaqiang, X. U. Preparation, Characterization and Photocatalytic  
10 Activity of Nanometer SnO<sub>2</sub>. *Int. J. Chem. Eng. Appl.* **2010**, *1*, 241.  
11  
12  
13 (36) Majumdar, S. The Effects of Crystallite Size, Surface Area and Morphology on  
14 the Sensing Properties of Nanocrystalline SnO<sub>2</sub> Based System. *Ceram. Int.* **2015**,  
15 *41*, 14350–14358.  
16  
17  
18  
19 (37) Rumyantseva, M. N.; Gaskov, A. M.; Rosman, N.; Pagnier, T.; Morante, J. R.  
20 Raman Surface Vibration Modes in Nanocrystalline SnO<sub>2</sub> : Correlation with Gas  
21 Sensor Performances. *Chem. Mater.* **2005**, *17*, 893–901.  
22  
23  
24  
25 (38) Wood, D. L.; Tauc, J. Weak Absorption Tails in Amorphous Semiconductors.  
26 *Phys. Rev. B* **1972**, *5*, 3144–3151.  
27  
28  
29  
30 (39) He, Z.; Zhu, Z.; Li, J.; Zhou, J.; Wei, N. Characterization and Activity of  
31 Mesoporous Titanium Dioxide Beads with High Surface Areas and Controllable  
32 Pore Sizes. *J. Hazard. Mater.* **2011**, *190*, 133–139.  
33  
34  
35  
36 (40) Hu, L.; Yuan, H.; Zou, L.; Chen, F.; Hu, X. Adsorption and Visible Light-Driven  
37 Photocatalytic Degradation of Rhodamine B in Aqueous Solutions by  
38 Ag@AgBr/SBA-15. *Appl. Surf. Sci.* **2015**, *355*, 706–715.  
39  
40  
41  
42 (41) Wang, C.; Shao, C.; Zhang, X.; Liu, Y. SnO<sub>2</sub> Nanostructures-TiO<sub>2</sub> Nanofibers  
43 Heterostructures: Controlled Fabrication and High Photocatalytic Properties.  
44 *Inorg. Chem.* **2009**, *48*, 7261–7268.  
45  
46  
47  
48 (42) Lea, J.; Adesina, A. A. The Photo-Oxidative Degradation of Sodium Dodecyl  
49 Sulphate in Aerated Aqueous TiO<sub>2</sub> Suspension. *J. Photochem. Photobiol. A*  
50 *Chem.* **1998**, *118*, 111–122.  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

- 1  
2  
3 (43) Yu, K.; Yang, S.; He, H.; Sun, C.; Gu, C.; Ju, Y. Visible Light-Driven  
4 Photocatalytic Degradation of Rhodamine B over NaBiO<sub>3</sub>: Pathways and  
5 Mechanism. *J. Phys. Chem. A* **2009**, *113*, 10024–10032.  
6  
7  
8  
9 (44) Moraes, S. B. de; Botan, R.; Lona, L. M. F. Synthesis and Characterization of  
10 Polystyrene/layered Hydroxide Salt Nanocomposites. *Quim. Nova* **2014**, *37*, 18–  
11 21.  
12  
13  
14  
15 (45) Ortelli, S.; Blosi, M.; Albonetti, S.; Vaccari, A.; Dondi, M.; Costa, A. L. TiO<sub>2</sub>  
16 Based Nano-Photocatalysis Immobilized on Cellulose Substrates. *J. Photochem.*  
17 *Photobiol. A Chem.* **2014**, *276*, 58–64.  
18  
19  
20  
21  
22 (46) Chen, G.; Ushida, T.; Tateishi, T. Development of Biodegradable Porous  
23 Scaffolds for Tissue Engineering. *Mater. Sci. Eng. C* **2001**, *17*, 63–69.  
24  
25  
26 (47) Byrappa, K.; Subramani, A. K.; Ananda, S.; Rai, K. M. L.; Dinesh, R.;  
27 Yoshimura, M. Photocatalytic Degradation of Rhodamine B Dye Using  
28 Hydrothermally Synthesized ZnO. *Bull. Mater. Sci.* **2006**, *29*, 433–438.  
29  
30  
31  
32 (48) Zhu, Z. F.; Zhou, J. Q.; Wang, X. F.; He, Z. L.; Liu, H. Effect of pH on  
33 Photocatalytic Activity of SnO<sub>2</sub> Microspheres via Microwave Solvothermal  
34 Route. *Mater. Res. Innov.* **2014**, *18*, 8–13.  
35  
36  
37  
38 (49) Watanabe, T.; Takizawa, T.; Honda, K. Photocatalysis through Excitation of  
39 Adsorbates. 1. Highly Efficient N-Deethylation of Rhodamine B Adsorbed to  
40 Cadmium Sulfide. *J. Phys. Chem.* **1977**, *81*, 1845–1851.  
41  
42  
43  
44 (50) Zhu, C.; Lu, B.; Su, Q.; Xie, E.; Lan, W. A Simple Method for the Preparation of  
45 Hollow ZnO Nanospheres for Use as a High Performance Photocatalyst.  
46 *Nanoscale* **2012**, *4*, 3060–3064.  
47  
48  
49  
50  
51 (51) Chen, W.; Sun, F.; Zhu, Z.; Min, Z.; Li, W. Nanoporous SnO<sub>2</sub> Prepared by a  
52 Photochemical Strategy: Controlling of Specific Surface Area and Photocatalytic  
53  
54  
55  
56  
57  
58  
59  
60



- 1  
2  
3 Activity in Degradation of Dye Pollutants. *Microporous Mesoporous Mater.*  
4  
5 **2014**, *186*, 65–72.  
6  
7 (52) Abdel-Messih, M. F.; Ahmed, M. A.; El-Sayed, A. S. Photocatalytic  
8  
9 Decolorization of Rhodamine B Dye Using Novel Mesoporous SnO<sub>2</sub>–TiO<sub>2</sub>  
10  
11 Nano Mixed Oxides Prepared by Sol–gel Method. *J. Photochem. Photobiol. A*  
12  
13 *Chem.* **2013**, *260*, 1–8.  
14  
15 (53) Altın, İ.; Sökmen, M. Preparation of TiO<sub>2</sub>-Polystyrene Photocatalyst from Waste  
16  
17 Material and Its Usability for Removal of Various Pollutants. *Appl. Catal. B*  
18  
19 *Environ.* **2014**, *144*, 694–701.  
20  
21  
22 (54) Ji, X.; Bai, C.; Zhao, Q.; Wang, A. Facile Synthesis of Porous SnO<sub>2</sub> Quasi-  
23  
24 Nanospheres for Photocatalytic Degradation of Rhodamine B. *Mater. Lett.* **2017**,  
25  
26 *189*, 58–61.  
27  
28 (55) Chen, H.; Zhang, D.; Zhou, X.; Zhu, J.; Chen, X.; Zeng, X. Controllable  
29  
30 Construction of Ordered Porous SnO<sub>2</sub> Nanostructures and Their Application in  
31  
32 Photocatalysis. *Mater. Lett.* **2014**, *116*, 127–130.  
33  
34  
35  
36  
37  
38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

## Table of Contents Graphic

