

Pedro Nuno de Jesus Francisco Freitas

Licenciado em Engenharia de Micro e Nanotecnologias

Conductive bridging RAM devices inspired on solidstate biopolymer electrolytes

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Orientador: Doutor Luís Miguel Nunes Pereira, Professor Auxiliar, Faculdade de Ciências e Tecnologia da Universidade Nova de Lisboa

Co-orientador: Doutora Asal Kiazadeh, Professora Auxiliar, Faculdade de Ciências e Tecnologia da Universidade Nova de Lisboa

Júri

Presidente: Arguente: Vogal: Professor Doutor Rodrigo Ferrão de Paiva Martins Professor Doutor Vítor Manuel Grade Tavares Professor Doutor Luís Miguel Nunes Pereira



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Faculdade de Ciências e Tecnologia

Universidade Nova de Lisboa

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ABSTRACT

This work reports the design, fabrication and characterization of metal-insulator-metal (MIM) structures acting as conductive bridging random access memory (CBRAM) devices using biopolymer insulator. Chitosan and hydroxypropyl cellulose (HPC) were deposited by spin coating in between evaporated Pt and Ag electrodes. CBRAM devices fabricated using chitosan as the insulating layer demonstrated retention times of up to 10⁵ s with an on/off ratio of 10² as well as enduring several program/erase cycles. Devices fabricated with HPC showed retention times of up to 10⁴ s with an on/off ratio of approximately 10⁶, and also showed stable device operation over several cycles.

Furthermore, the functionalization of chitosan with silver nanoparticles and its integration in the MIM structures were investigated, as well as the substitution of the e-beam evaporated Ag electrode by a screen printed Ag electrode.

Keywords: Chitosan, Hydroxypropyl cellulose, resistive switching, electrochemical metallization, conductive bridge, biopolymer.

RESUMO

Este trabalho reporta a fabricação e caracterização de estruturas metal-isolante-metal (MIM) que funcionam como dispositivos *conductive bridging random access memory* (CBRAM), usando biopolimeros como a camada isolante. Quitosano e hidroxipropilcelulose (HPC) foram depositados por *spin coating* entre contactos evaporados de Pt e Ag. Os dispositivos CBRAM fabricados com quitosano como camada isolante, mostraram valores de retenção de estado até 10⁵ s com um rácio *on/off* de 10², apresentando tambem durabilidade ao longo de vários ciclos de programação. Dispositivos fabricados com HPC apresentam uma retenção de estado até cerca de 40 min com um racio on/off de aproximadamente 10³, mostrando tambem operação estável ao longo de vários ciclos de operação.

Para além disso, a funcionalização do quitosano com nanopartículas de prata e a sua integração em estruturas MIM foram investigadas, bem como a substituição dos electrodos de Ag obtidos por evaporação de *e-beam* por electrodos de Ag impressos por *screen printing*.

Palavras-chave: Quitosano, Hidroxipropilcelulose, *resistive switching,* metalização eletroquímica, *conductive bridge,* biopolímero.

LIST OF ABBREVIATIONS

AE – Active electrode	
CBRAM – Conductive bridging random access memory	
CMC – Carboxymethyl cellulose	
CNP – Cellulose nanofiber paper	
CRF – Carbon rich filaments	
DMAc – Dimethylacetamide	
EC – Ethyl cellulose	
ECM – Electrochemical metallization	
EDS – Energy dispersive spectrometer	
FWMH – Full width at medium height	
HCS – High current state	
HPC – Hydroxypropyl cellulose	
IE – Inert electrode	
ITO – Indium tin oxide	
LCS – Low current state	
MC – Methyl cellulose	
MIM – Metal-insulator-metal	
MLC – Multi-level cell	
NMMO – N-methylmorpholine-N-oxide monohydrate	
NDR – Negative differential resistance	
NTP – Normal temperature and pressure	
NVM – Non volatile memory	
PCM – Phase change memory	
P/E – Program/erase	
PET – Polyethylene therephtalate	
PMU – Pulse measurement unit	
ReRAM – Resistive random access memory	
SEM – Scanning electron microscopy	
SMU – Semiconductor measurement unit	
SPE – Solid polymer electrolyte	
TCM – Thermochemical memory	
TFT – Thin film transistor	
VCM – Valence change memory	
WORM – Write once read many times	

WORM – Write once read many times

LIST OF SYMBOLS

- СС - Compliance current
- CC_{read} Read pulse compliance current
- CC_{reset} Reset pulse compliance current CC_{set} Set pulse compliance current
- Electric field Е
- ΔI - Current variance
- Ion/Ioff - Current On/Off ratio
- Reset pulse time treset
- tretention Retention time
- tset
- Set pulse time
 Read voltage
 Reset voltage
 Set voltage V_{read}
- Vreset
- Vset

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1. Motivation and objectives

1.1. Motivation

Nowadays, there is an increased urge towards the research of new and innovative materials and fabrication techniques in electronics. This holds especially true in the case of non-volatile solid-state memory storage, where most commercial solutions to date are based on FLASH technology. FLASH is based on the electrical functioning of silicon based floating gate transistors, and as such, it has several drawbacks, such as low write speeds (50 µs), low endurance (10⁵ cycles), high program/erase (P/E) voltages (20V), and the inherent scalability issue that CMOS technology is facing beyond the <10nm node[1]. As such, new types of non-volatile memory (NVM) storage, such as, ferroelectric RAM (FeRAM), magnetic (MRAM), and resistive RAM (ReRAM) have been gaining increased attention throughout this past few years in order to mitigate FLASH's shortcomings. In particular, ReRAM based devices is an ever-growing field of interest, due to its high operation speeds (1ns)[2], low operating voltages (0.6V)[3], great scalability potential (≤28nm)[4] and multi-level cell (MLC) operation[5]. Even though ReRAM is becoming something of a hot topic nowadays, the study of ReRAM switching mechanics and properties using different materials is still in its infancy, and there is a clear market opportunity towards the development of such devices, with a prospective market value for trending NVM technologies of 7B\$ by 2020 [6].

Conductive bridging random access memory (CBRAM) devices are inserted in the ReRAM category, and they stand out due to their bipolar electrochemical switching effects which are area independent, and as such, have a great scalability potential. The main requirement for CBRAM devices is the usage of an inert electrode, an electrochemically active electrode, and an insulator that exhibits high ionic conductivity. As the main requirement for the insulating material is its ionic conductivity towards the active electrode, there is a wide variety of materials that fit into this category, however, state-of-the-art CBRAM devices are mostly based on the usage of chalcogenide, oxide and bilayer structures deposited by physical or chemical vapor deposition techniques[7]–[9].

Another topic that is gaining relevance nowadays is the usage of biopolymer materials in electronics, whether it be simply as substrates for printable and solution based electronics[10], [11], or as functionalized materials towards electronics[10], due to their structural properties, biocompatibility and recyclability. As such, some reports have surfaced lately, in which the role of biopolymers as the insulating material in CBRAM devices is studied[12]–[20].

This work aims to bring new insights towards the usage of cost-effective materials and methods involving the usage of solution processed biopolymers as the insulating layer in CBRAM devices. Here, we investigate the usage of two biopolymers: chitosan and hydroxypropyl cellulose (HPC) as the insulating material in a metal-insulator-metal (MIM) structure using conventional electrodes deposited by physical vapor deposition (PVD), and its applicability towards CBRAM devices. We also explore the applicability of these devices towards printed electronics by studying the electrical behavior using screen printed electrodes. Furthermore, the role of silver nanoparticle (AgNP) functionalization of chitosan in the electrical characterization is also investigated.

In sum, the focus of this work will be on the chitosan, HPC and chitosan-AgNP composite solution formulation; the deposition of these solutions via spin-coating; and the electrical characterization of CBRAM devices using these thin films as the insulating layer and the comparison between PVD contacts and screen printed contacts.

1.2. Objectives

The main objective of this thesis project is the design, fabrication and characterization of MIM structures as CBRAM devices based on eco-friendly and low-cost biopolymers. With that in mind, two natural occurring biopolymers were chosen as insulating materials: chitosan and hydroxypropyl cellulose (HPC). The focus will be on the biopolymer solution preparation, and thin film deposition via spin-coating, and the electrical characterization associated.

As such, the following themes will be covered:

- Characterization of the chitosan-AgNP's composite solutions and film properties.
- Thickness and area dependence study on the electrical switching mechanics of the devices.
- Effect of the electrolyte material on the electrical functioning of the CBRAM devices.
- Electrical characterization of screen printed devices.

2. GENERAL INTRODUCTION

For a better understanding of the following chapters, a brief introduction on relevant topics will be presented. This chapter will be split into three sections. Firstly, some fundamental concepts about ReRAM, and the different types of ReRAM will be introduced. Then some further insight into state of the art of CBRAM devices architectures and their switching mechanisms will be given. Finally, a brief explanation on the fundamental structures and origins of biopolymer materials and their current role in electronics, with special emphasis on ReRAM devices will be presented.

2.1. Variety of ReRAM devices

ReRAM[21] is a category of memory storage devices that includes every type of NVM (non volatile memory) devices based on the resistivity change of a given material. As such, there are several types of NVM that fall into this category, as illustrated in figure 1.1 a), and they are designated by the physical phenomena that leads to the resistive switching effect. Although all of these memory effects are electrically induced, the actual physical driving force behind the resistance switching mechanism is different for the different ReRAM types. For instance, phase change memory (PCM)[22][23] is characterized by the switching of the insulating material from a crystalline more conductive phase (on state), to an amorphous less conductive phase (off state) by voltage application; thermochemical memory (TCM)[24] effect can be described by the resistivity change of an insulating material due to chemical changes inside the material caused by temperature, which can be electrically induced by joule heating; valence change memory (VCM) and electrochemical memory (ECM) effect rely on redox reactions in the material (anions in the case of VCM and metal cations on the case of ECM) to alter the resistivity between electrodes[8].

Depending on whether it is the insulator or one of the metal electrodes that plays the dominant role towards the resistive switching effect, these devices can be split into two subcategories: electrode dominated, or insulator dominated. Insulator dominated ReRAM can include PCM, VCM and TCM, whereas ECM belong to the electrode dominated category.

Concerning the programming point of view, all ReRAM devices fall into two subcategories: unipolar or bipolar. Unipolar operation is defined by using voltage pulses independent of the polarity for the set and reset functions. In this case, the set voltage is always higher than the reset voltage, and the reset current is always higher than the set current because it is limited by compliance. Bipolar switching is characterized by performing the set operation in one polarity, and the reset operation in the opposite polarity. Typical I-V operation is shown in Figure 2.1 b) for unipolar behavior and c) for bipolar behavior.

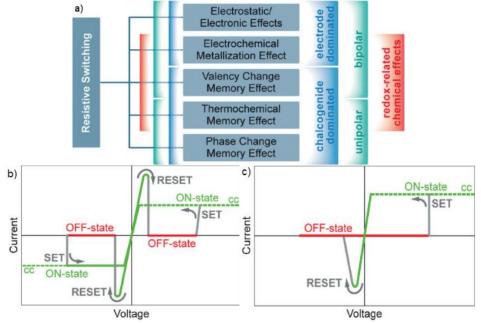


Figure 2.1–Classification of the resistive switching effects that are considered for non-volatile memory applications(a). Typical I-V curve of a unipolar (b) and bipolar (c) resistive memory, (adapted from[25])

2.2. CBRAM fundamentals and applications

CBRAM devices are a type of ReRAM devices characterized by switching its resistivity through the ECM effect. Plainly put, CBRAM are normally constituted by an electrode dominated, asymmetric MIM structure, composed by: an inert electrode (IE), an electrolyte and an electrochemically active electrode (AE) (e.g. copper (Cu) or silver (Ag)). Due to the asymmetry of the device structure, by application of a positive voltage bias on the AE, while grounding the IE, the oxidation of the active metal is provoked and an electric field between the two electrodes is generated along the electrolyte.

$$E = -\frac{V}{d} \tag{2.1}$$

Equation (2.1) describes the electric field (E) as a function of the voltage (V) and distance (d). In many cases, inorganic nanoparticles are dispersed in the electrolytic medium in an effort to charge the dispersed nanoparticles, reducing the effective electric field distance, and therefore reducing the working voltages [34].

The metal cations migrate along the electric field towards the grounded IE, when reaching this electrode, the cations are reduced and crystallized on the surface of the IE. By this process, eventually, a crystallized metallic filament will be formed between the two electrodes, short circuiting them, and changing the resistivity state between the electrodes. Equation (2.2) and (2.3) represent the anodic dissolution of a metal (M) to the ionic form, and the reduction and electrocrystallization of M, respectively:

$$M \longrightarrow M^{z+} + e^{-} \tag{2.2}$$

$$M^{z+} + ze^{-} \longrightarrow M \tag{2.3}$$

Since the behavior described depends on the oxidation of an electrode to achieve a high current state (HCS) and it depends on the reduction of a metallic filament to achieve the low current state (LCS), then the programming of CBRAM devices are characterized by bipolar operation (from hereon in, all current states will be mentioned as LCS and HCS). The setting operation requires a positive voltage bias to be applied on the AE to induce the formation of the filament, and the reset operation is performed at a negative polarity to reduce and break the metallic filament. Figure 2.2 shows a sketch of the operation of a typical CBRAM device comprised of an electrolyte sandwiched between a Ag AE and a platinum (Pt) IE.

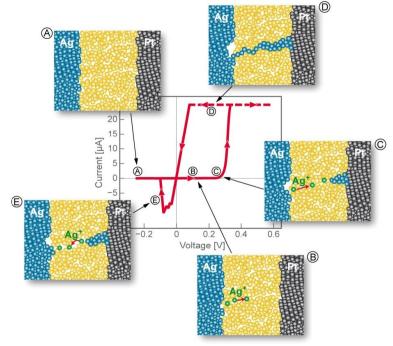


Figure 2.2 – Sketch of the steps ((A) - (D)) of a set operation and a reset operation (E) on a typical CBRAM device [7]

Because the ECM effect is electrode dominated, while the insulator takes a more secondary role towards the switching mechanism, the only condition for the electrolyte material is a high ionic conductivity towards the migrating ions. As such, there is a wide range of materials that fit into this parameter. For the purpose of the current work, we will consider four categories: chalcogenide based, oxide, bilayer and biopolymer electrolytes.

Chalcogenide based materials are regarded as the most used and studied materials for CBRAM devices because of their well-known diffusion mechanisms towards copper (Cu²⁺) and silver (Ag⁺) ions. However, integration with CMOS technology in chalcogenide materials might be a problem. Oxides offer a great integration with CMOS technology, however CBRAM devices based on these materials typically operate at much lower currents and speeds, as well as suffering from high device-to-device and cycle-to-cycle uniformity problems, limiting their real application. Bilayer materials aim to improve oxide based materials uniformity and other performance problems by adding a diffusion layer to the oxide layer in order to better control the diffusion of the AE. Jana et al.[9] provided an extensive overview of the current state of the art on these three types of materials, and Waser et al.[7] extensively reviews the theory behind these switching mechanisms.

Regarding the latter category of materials applied in CBRAMs, an insight into biopolymer electrolytes and their application in electronics will be given in the following chapters.

2.3. Biopolymers: from fundamentals towards electronic applications

Biopolymers are natural occurring polymers originating from living organisms, often with a very well defined structure when compared to their synthetic counterparts. Due to their structure and origin, biopolymers possess very interesting characteristics, such as: biocompatibility, biodegradability, self-sustainability, thermal stability, low cost and solution processability at room temperature[26]. This set of properties made biopolymers to be established over time as an important resource for important economic sectors such as: packaging, food, medicine, paper and textile industries[27]. Most recently, due to the concern for the stagnation of the silicon based electronics industry, biopolymers are starting to gain relevance as an innovative material towards low cost printed and paper electronics[28]

Among various existing biopolymers, two stand out for being the most renewable and abundant on earth: cellulose and chitin. Cellulose mainly originates from plant cell walls, but can also be found on some sea organisms, fungi and bacteria. Its structure consists of a straight chain linkage of D-glucose units through mainly $\beta(1\rightarrow 4)$ glycosidic bonds (some forms may possess $\alpha(1\rightarrow 4)$ glycosidic bonds). Chitin is the second most abundant biopolymer on Earth, after cellulose, and in contrast to cellulose, it is mainly extracted from fungi cell walls and the exoskeletons of crustaceans and insects. It is constituted by N-acetylglucosamine units linked by $\beta(1\rightarrow 4)$ covalent bonds; plainly put, chitins structure can be comparable to that of cellulose, but with one hydroxyl group from each monomer substituted by an acetyl amine group.

For the purpose of solution processability, cellulose cannot be dissolved in common solvents due to the strong hydrogen bonds[29]. Cellulose dissolution has to be conducted, therefore, in other complex solvent systems, such as: lithium chloride/N,N-dimethylacetamide (LiCl/DMAc), N-methylmorpholine-N-oxide monohydrate (NMMO) and ionic liquids[30]. However these dissolution systems suffer from high cost, toxicity and volatility, limiting their applications. Alternatively, by substituting cellulose's hydroxyl groups by other functional groups, cellulose derivatives that are soluble in ordinary solvents can be created. Examples of cellulose derivatives include: ethyl cellulose (EC), hydroxypropyl cellulose (HPC), methyl cellulose (MC) and carboxymethyl cellulose (CMC)[31]. Particularly, HPC results from the hydroxypropylation of cellulose, substituting some of the hydroxyl groups from the repeating glucose units of the cellulose backbone to hydroxypropyl groups, making it soluble in water and other common organic solvents[32].

Chitin, similarly to cellulose, cannot be dissolved in common solvents, as such, chitosan emerges as a derivative of chitin that is soluble in slightly acidic solutions, being the most used solvent system acetic acid aqueous solutions[33]. Chitosan derives from the deacetylation of chitin, therefore, its structure is comprised of N-glucosamine units linked through $\beta(1\rightarrow 4)$ covalent bonds much like chitin. Figure 1.3 illustrates the chemical structure of cellulose, chitin and chitosan.

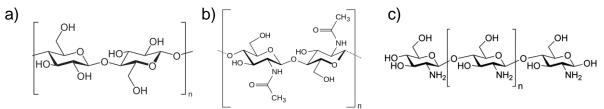


Figure 2.3 – Chemical structures of cellulose (a), chitin (b) and chitosan (c).

Most recently, the strive towards self-sustainable, low cost, green electronics has driven forward the research regarding biopolymer materials in electronics. The most notable role of biopolymer materials in electronics is as a free standing substrate material, and as a dielectric or electrolyte material,

due to its high capacitance, in a variety of systems including: batteries[34], [35], capacitors[36]–[39],thin film transistors (TFT)[40]–[46], and ReRAM devices[12]–[20].

Focusing on the latter application, Hosseini and Lee have reported the usage of AgNP-chitosan composite as a solid polymer electrolyte in CBRAM devices[16]. In their work, resistive switching is achieved in a MIM structure comprised of a Pt IE, AgNP-chitosan as a solid polymer electrolyte (SPE), and a Ag AE, either on a silicon (Si) substrate or a flexible polyethersulfone (PES) plastic substrate. These devices exhibited low voltage operation ($\pm 0.8V$) a high On/Off ratio (I_{on}/I_{off}) (10^5), high data retention (10^4 s), and an endurance of 100 program/erase (P/E) cycles. The same authors went on to develop similar devices while changing both contacts to magnesium (Mg)[13].

Most recently, Hosseini and Lee have developed ReRAM devices using starch and a starchchitosan blend on a PET/ITO substrate that also serves as one electrode, while the other electrode is gold (Au)[18]. The fact that both electrodes are inert, make this an electrolyte dominated device. The authors postulate that the resistive switching comes from the formation of carbon rich filaments (CRF) provenient from the SPE, induced by joule heating.

Another work of interest is that of Nagashima et al[19], where cellulose nanofiber paper(CNP) is functionalized with AgNP's, acting as a SPE for the purpose of building a CBRAM device. In this work, both a Pt/AgNP-CNP/Pt and a Pt/Ag-CNP/Ag architectures are explored, as well as the influence of the device area in the LCS. Both the Pt/AgNP-CNP/Pt and Pt/Ag-CNP/Ag structure show resistive switching, being that the latter, showed much greater switching behavior in terms of lower forming voltages, switching probability and retention (Annex A – C).

Other work regarding the usage of biopolymer in resistive switching memory devices include: a DNA biopolymer nanocomposite[15], silk[14] and AuNP's in silk[20], sericin[17] and albumen[12]. Table 2.1 provides a summarized overview of some existing ReRAM technology using biopolymers.

SPE	BE	TE	WORM	Voltage(V)	I _{On} /I _{Off}	Retention(s)	P/E endurance (cycles)	Ref
AgNP- Chitosan	Pt	Ag	No	±0.8	10 ⁵	10 ⁴	100	[16]
AgNP- Chitosan	Mg	Mg	No	[-3, 2]	10 ²	10 ⁴	60	[13]
Starch- Chitosan	ITO	Au	No	[-4, 2]	10 ³	10 ⁴	—	[18]
AgNP- CNP	Pt	Ag or Pt	No	±0.5	10 ⁶	10 ⁵	100	[19]
DNA composite	ITO	Ag	Yes	±6	10 ²	10 ⁵	—	[15]
Silk	ITO	AI	No	±15	10	10 ³	120	[14]
AuNP-silk	ITO	AI	No	±2	10 ⁴	10 ³	10	[20]
Sericin	Au	Ag	No	[-1.5, 3]	10 ⁶	10 ³	21	[17]
Albumen	ITO	Al	No	[-3, 4]	10 ³	10 ⁴	500	[12]

Table 2.1 - Overview of existing ReRAM technology using biopolymers

The fact that resistive switching phenomena (similar to CBRAM) has already been reported in biopolymer materials utilizing structures were both electrodes are inert, supports the possibility that CBRAM devices based on biopolymers could in fact possess a mixed switching mechanism consisting of thermochemical, valence change and electrochemical metallization effects.

As already demonstrated by Nagashima et al[19], cellulose can in fact be used as a solid polymer electrolyte towards resistive memory applications, although producing very interesting results, the solution processability of this CNP remains challenging.

In this work, based on the work of Hosseini and Lee[16], the usage of chitosan in ReRAM devices comprised of a Pt/chitosan/Ag structure is studied, while adapting the solution deposition method from drop-casting to spin-coating, and studying the influence of the electrolyte thickness and the role of AgNPs embedded in the chitosan matrix in the electrical functioning of the devices. In a similar way, a new type of device using water soluble HPC as the electrolyte layer in ReRAM devices was proposed and achieved, and its effect on the resistive switching properties of the devices was compared to its chitosan counterparts.

3. MATERIALS AND METHODS

This chapter will focus on the steps that were taken in the fabrication and characterization process of the aforementioned devices. First, the AgNP-chitosan composite solution formulation will be presented. Then all the necessary steps towards device fabrication, and finally, how the characterization of the insulating material and electrical characterization of the devices were performed.

3.1. Materials and reagents

Low molecular weight chitosan powder (deacetylation degree: 75%-85%, CAS:9012-76-4), silver nitrate powder (AgNO₃ \geq 99.0%, CAS:7761-88-8) and acetic acid (CH₃CO₂H \geq 99%, CAS:64-19-7) were purchased from Sigma-Aldrich, and hydroxypropyl cellulose powder (M_w ~ 100000, CAS:9004-64-2) was purchased from Acros organics.

3.2. AgNP-chitosan solution formulation

The procedure for the formulation of the AgNP-chitosan composite solution was adapted from [47].

Acetic acid was diluted to obtain a 1% (v/v) aqueous solution, chitosan flakes were then added in a 3% (w/v) and dissolved under vigorous stirring at ambient temperature overnight. 50mL of the prepared chitosan solution were then heated in an oil bath, under reflux, up to 95°C under vigorous stirring (1100rpm), then, 20mL of a previously prepared AgNO₃ aqueous solution with a certain concentration (7mM, 12mM, 26mM, 52mM) were added dropwise (this process results in a diluted 2.1% (w/v) chitosan solution). The mixture was kept under temperature and agitation for a total of 7h, while solution samples were taken hour by hour to evaluate the effect of the reaction time on the concentration of nanoparticles dispersed in the solution.

3.3. Device fabrication

All fabricated devices used the same basic architecture: Glass/Ti/Pt/electrolyte/Ag. Figure 3.1 represents a basic schematic representation of the fabricated CBRAM devices.

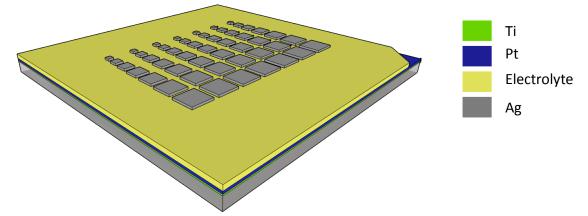


Figure 3.1 – Schematic representation of the fabricated CBRAM devices

For the common bottom contact, a 20nm Ti adhesion layer and a 100nm Pt layer were deposited by e-beam evaporation.

The electrolyte material is comprised of either chitosan, HPC or the AgNP-chitosan composite deposited by spin coating. In order to achieve different thicknesses of the electrolyte layer, different solution concentrations of chitosan and HPC were used (1% (w/v), 2.1% (w/v), 3% (w/v) for chitosan, and 5%wt, 6%wt, 8%wt, and 10%wt for HPC) while maintaining the spin coating parameters for all solutions.

The spin coating process was performed as follows:

• First, the Pt covered glass substrate was subjected to a 30min UV-ozone treatment in order to increase its hydrophilicity.

- Then, the solution is dispensed on the static substrate while being filtered through a 0.45µm syringe filter (the 3% (w/v) chitosan and 10%wt HPC solutions were filtered through a 0.8µm filter instead due to their higher viscosity).
- A two-step spin coating process is performed by spinning the substrate at 1000rpm during 35s and then at 3000 rpm during 15s.
- The substrate is then heated on a hotplate at 60°C during 2min to evaporate any remaining solvents.

The Ag top electrodes were then deposited through e-beam evaporation with a thickness of 100nm and were patterned by shadow mask lithography. The Ag contacts pattern consists of a 7x7 matrix comprised of square contacts, where the area of the square contacts is varied from column to column as illustrated in figure 3.1. The studied areas were: 0.5mm², 0.75 mm², 1mm², 1.25mm², 1.5 mm², 1.75mm² and 2mm².

Furthermore, the usage of Ag electrodes deposited through screen printing was also investigated. To this purpose, a Ag conductive ink (PE-AG-530 Flexible Silver Conductive ink) purchased from conductive compounds was passed through a patterned mesh (using the same pattern described above) followed by a 130°C cure during 90 s on a hotplate. The used mesh has the following characteristics: mesh model 120; mesh count, 305 mesh/inch; aperture, 45µm; thread diameter, 34 µm; opening, 30.5%; screen thickness, 52-57µm. The usage of screen printed Ag electrodes were tested on the intermediate concentrations for both chitosan (2% (w/v)) and HPC (8%wt) electrolytes.

In this work, several devices were tested in order to study the influence of the electrolyte material, electrolyte thickness and the AE material in their electrical performance. Table 3.1 resumes the fabricated devices characteristics and their nomenclature from here on out.

Nomenclature	SPE	SPE Concentration	AE material	
1Q	Chitosan	1% (w/v)	e-beam evaporated Ag	
2Q	Chitosan	2.1% (w/v)	e-beam evaporated Ag	
3Q	Chitosan	3% (w/v)	e-beam evaporated Ag	
5HPC	HPC	5%wt	e-beam evaporated Ag	
6HPC	HPC	6%wt	e-beam evaporated Ag	
8HPC	HPC	8%wt	e-beam evaporated Ag	
10HPC	HPC	10%wt	e-beam evaporated Ag	
2QAgNP	Chitosan-AgNP composite	2.1% (w/v)	e-beam evaporated Ag	
2QSP	Chitosan	2.1% (w/v)	Screen printed Ag	
8HPCSP	HPC	8%wt	Screen printed Ag	

Table 3.1 – Nomenclature and description of all fabricated devices

3.4. Characterization

In order to evaluate the formation of AgNP in the chitosan solutions, UV-Vis spectrophotometry was performed using a PerkinElmer Lambda 950 UV-Vis-NIR spectrophotometer with a 2D detector module in absorption mode, using a wavelength range of 250 nm to 650 nm.

Thickness evaluation of the deposited films was performed by profilometer step measurements using an Ambios XP-Plus 200 Stylus Profilometer with the following parameters: speed = 0.1 mm/s, length = 0.5 mm, range = 100 μ m, stylus force = 0.1 mg.

The morphology and composition of the chitosan-AgNP composite membrane was analyzed by scanning electron microscopy (SEM) and energy dispersive spectroscopy (EDS) using a Carl Zeiss Auriga crossbeam (SEM-FIB) workstation instrument equipped with an Oxford X-Ray energy dispersive spectrometer.

All electrical characterization was performed on a Keithley 4200-SCS semiconductor parameter analyzer connected to a Janis ST-500 probe station. The electrical setup was made so that the IE was always grounded, and the voltage bias on the AE varies. All measurements were carried at normal temperature and pressure (NTP) conditions. All I-V characteristics were performed using a 0.1 step, and a 0.7 s sweep delay. For the mapping of the I-V characteristics of the devices, all measurements are described by a 0 V \rightarrow 6 V \rightarrow 0 V \rightarrow -2 V \rightarrow 0 V cycle, with the exception of the 2QAgNP substrate, were the I-V characteristics are described by a 0 V \rightarrow 3 V \rightarrow 0 V \rightarrow -2 V \rightarrow 0 V. The compliance current (CC) was 1 mA for all I-V characteristics. For the retention and endurance measurements, custom programming was performed for each of the different measurements.

4. Results and discussion

4.1. UV-Visible absorption spectra

In order to evaluate the formation of stable AgNP's in the composite solution, UV-visible absorption spectra was conducted on the solutions containing different silver loadings (7mM, 12mM, 26mM, 52mM), as well as evaluating the effect of the reaction time (1h-7h).

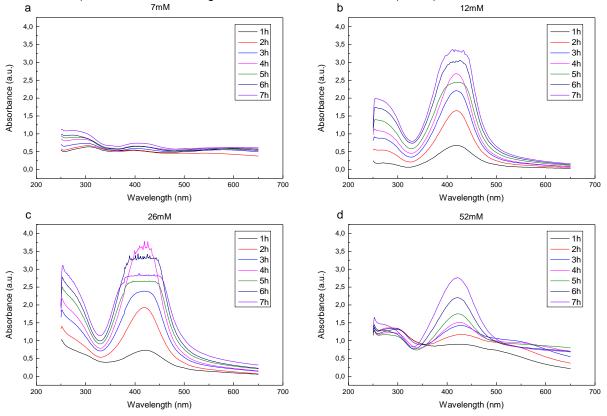


Figure 4.1 – UV-visible absorption spectra of the tested AgNP-chitosan solutions (7mM (a), 12mM (b), 26mM (c), 52mM (d)).

These results can be considered as a comparative study to evaluate the effect of the AgNO₃ concentration and reaction time in the medium. All solutions present a maximum peak on the 400 – 420nm, wavelength range, indicating the presence of metallic AgNP's.

The 7mM solution show a much lower absorption spectra compared to the more concentrated solutions. The 26mM and 52mM solutions show a higher full width at medium height (FWHM) than the 12mM solution, hinting that there is a higher dispersion in nanoparticles sizes in the medium. Therefore the preferred solution is the 12mM solution, for showing high absorption intensity at 420nm and having the lower FWHM.

Regarding the reaction time in the 12mM solution, a peak saturation can be observed for reaction times over 5h, therefore, the 4h reaction time is considered to be the optimum reaction time.

Taking these results into consideration, the 12mM solution at a 4h reaction time was selected to be used as the electrolyte layer in future experiments.

4.2. SEM-EDS analysis

To evaluate the morphological structure of the chitosan-AgNP composite membranes, SEM topographical imaging was carried out, and EDS was performed to analyze the composition of the membrane. Figure 4.2 shows the obtained SEM image of the composite membrane as well as the corresponding EDS analysis.

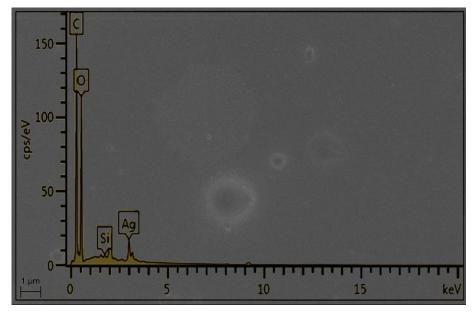


Figure 4.2 – Topographical view of the AgNP membrane obtained by SEM and the corresponding EDS analysis

Even though, there is no distinguishable dispersion of AgNP visible on the obtained SEM image, EDS analysis reveals a silver approximate concentration of 8.8wt%. This result supports the possibility that, even though not being clearly visible on a topographical view, AgNPs can in fact be well dispersed within the chitosan matrix.

4.3. Profilometer analysis

In order to test the effect of different SPE thicknesses in the electrical performance of the devices, various SPE's using the same materials but different concentrations were used, while maintaining the spin coating parameters. This procedure was adopted mainly because, the film thickness dependence over the spin coating parameters such as time and spin speed are not linear (Annex D), therefore it is very challenging to achieve an intermediate film thickness by this method. By varying the solution concentration instead of the spinning parameters, a more linear relation can be obtained.

The profilometer measurements were performed by measuring the thickness of the electrolyte layer on five different sites, the average and the associated error were then calculated and are shown in Table 4.1.

Substrate	Thickness (nm)
1Q	71,4 ± 10
2Q	304 ± 22
3Q	508,4 ± 91
5HPC	362,6 ± 7
6HPC	583 ± 19
8HPC	1135,2 ± 13
10HPC	1815,6 ± 10
2QAgNP	151 ± 11
2QSP	317 ± 8
8HPCSP	1 089 ± 37

Table 4.1 – Measured SPE thicknesses obtained for the different devices

It should be noted that, it was expected for the thickness of the 2Q and 2QAgNP films to be similar, however the thickness of the 2QAgNP film is approximately half of the 2Q film, this might be caused by changes that the nanoparticles may cause on the rheological behavior of the solution. To prove this theory, further rheological studies would need to be conducted.

4.4. Electrical analysis

4.4.1. Typical I-V characteristics

Since the electrical characteristics can vary greatly from device to device in the same substrate, all the electrical characteristics in the different substrates were evaluated. As such, a statistical study was conducted, in order to determine the effect of SPE thickness in the electrical performance.

Figure 4.3 maps the I-V characteristics of all of the devices in the 2Q substrate according to their physical coordinates. The remaining maps can be found from Annex H – M. The rows have been labeled from A – G, and the columns from 1 – 7. Also, the devices electrode area are indicated below each column. From here on in, the device coordinates will be called by their row letter and their column number (e.g. A1).

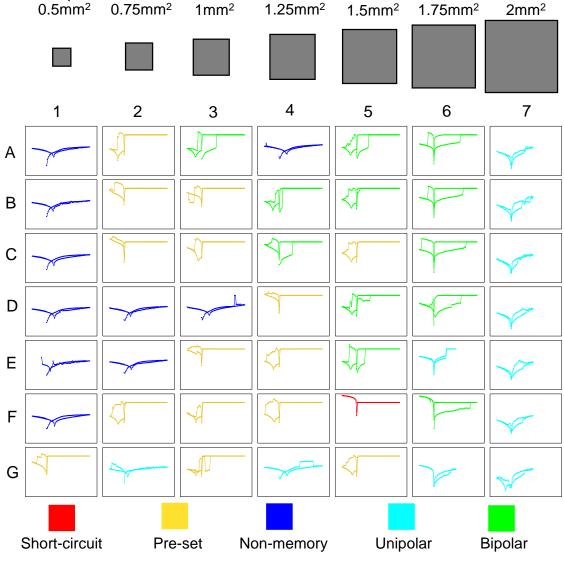


Figure 4.3 – I-V Characteristics of all the devices on the 2Q substrate

By analyzing Figure 4.3, different switching behaviors can be observed along the substrate. For the purpose of this work, these different behaviors can be divided into five categories: non-memory, short circuit, pre-set, unipolar switching and bipolar switching.

Non-memory devices can be designated by simply demonstrating an insulating behavior and no resistive switching (e.g. A1, B1, etc.). These devices might show some small hysteretic behavior because of trapped oxygen vacancies in the electrolyte due to its elevated water content. To note, that even appearing as not working, there is a possibility that they might show resistive switching under different electrical programming. Figure 4.4 shows the behavior of the B1 device on the 2Q substrate, as a standard example of a non-memory device.

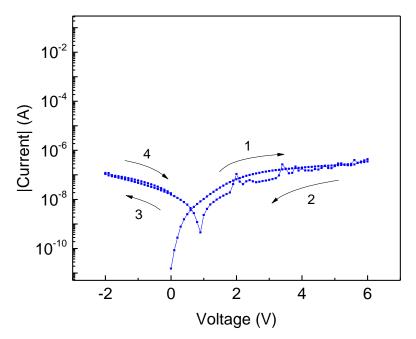


Figure 4.4 – I-V characteristics of the B1 device on the 2Q substrate, a typical example of a non-memory

Short circuited devices appear solely as a short circuit, not being able to ever be reset (e.g. F5). The appearance of these devices are most likely explained by defects of the electrolyte layer, or incomplete film coverage, allowing the deposited AE to be in direct contact with the bottom IE. Figure 4.5 shows the I-V curve of the F5 device, a typical example of a short circuited device.

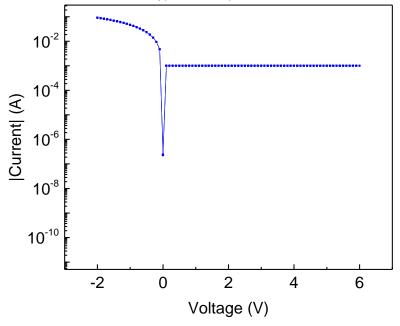


Figure 4.5 – I-V characteristics of the F5 device on the 2Q substrate. A typical example of a short circuited device.

Pre-set devices are characterized by initially appearing in a HCS but being able to be reset by applying a negative voltage bias (e.g.E4, F4). Figure 4.6 illustrates the I-V characteristics of the C5 device on the 2Q substrate, a typical example of a pre-set device.

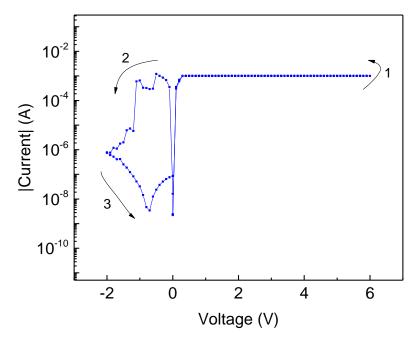


Figure 4.6 – I-V characteristics of the C5 device on the 2Q. A typical example of a pre-set device.

As represented in Figure 4.6, the reset of these devices might show a negative differential resistance (NDR). This means that the reset process is not performed instantaneously, but it rather achieves decreasing current levels until it reaches the final LCS. Under a certain programming, the NDR effect can be utilized towards the programming of multi-level cell's (MLC), meaning that one device might show multiple bit's per cell. Even though this effect is apparent in these devices, the programming of MLC in this case is uncontrollable.

One possible explanation towards the appearance of pre-set devices, is that silver diffuses into the electrolyte matrix when the contacts are deposited through e-beam evaporation[48], then it is possible to rupture the diffused filament through voltage application.

Unipolar switching devices as already described in chapter 2.1, exhibit resistive switching behavior independent of the voltage polarity applied. This behavior is most frequent in electrolyte dominated switching mechanisms such as TCM. Figure 4.7 shows the behavior of the G4 device on the 2Q substrate which is a typical example of unipolar operation.

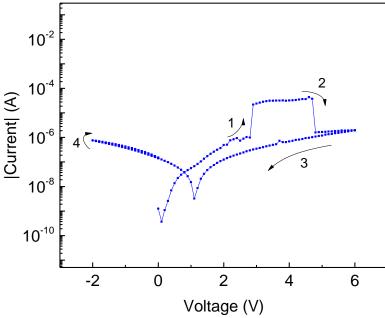


Figure 4.7 – I-V characteristics of the G4 device. A typical example of unipolar operation.

In the figure above, a clear switching to a HCS is visible in (1), the current then drops back to the LCS by applying a higher voltage on the same polarity. This behavior is typical in unipolar operation. It

should be noted that, even though some devices shown in Figure 4.3 do switch to the HCS but do not drop back to the LCS by application of a higher voltage (e.g. D7, E7), we still consider them to exhibit unipolar operation, but the applied voltage simply was not sufficient to switch the device back to the LCS. These considerations are made for two reasons: the HCS does not match the CC, and when the reset operation begins, the device appears in the LCS.

Bipolar switching devices, as described in chapter 2.1 show resistive switching dependent of the voltage polarity. In this case, the devices appear to set at the positive voltage polarity, and reset when a negative voltage bias is applied, as is common for CBRAM devices.

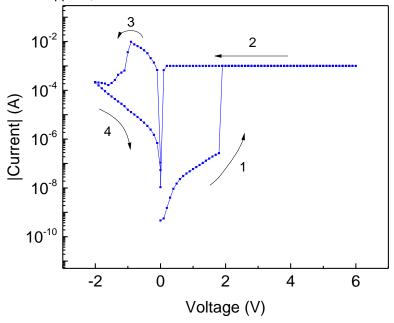


Figure 4.8 – I-V characteristics of the C4 device on the 2Q substrate. A typical example of a bipolar switching device

As illustrated above, a clear switching to the HCS is performed in (1), it should be noted that the HCS must be limited by the CC to prevent the irreversible damaging of the device. The device exhibits a clear window in (2), then at a negative voltage bias, the device switches back to the LCS in (3). Much like the pre-set devices, this devices might exhibit NDR operation in the reset process.

Concerning the device electrode area dependency, by analyzing Figure 4.3, V_{set} seems to lower by decreasing the AE area, hinting that the switching mechanism responsible for the resistive switching might not solely be an ECM effect, since this effect is area independent, but it is possible that the switching mechanism is a mixture of the ECM effect with an electrolyte dependent effect such as TCM or VCM, mostly due to the presence of oxygen vacancies inside the electrolyte layer. Even though V_{set} seems to lower with the decreasing electrode area, since all of the devices are on the same substrate, other effects such as film uniformity issues can cause a great variability in device to device uniformity. We suggest that a new type of substrate should be fabricated were all of the patterned devices have the same size, in order to verify if the V_{set} is consistent in a larger sample group.

4.4.2. Thickness dependency on the electrical performance

Table 4.2 resumes the number of devices that belong to each of the categories mentioned above for the different produced substrates.

Substrate	Non-memory	Short-circuits	Pre-set	Unipolar	Bipolar
1Q	0	47	2	0	0
2Q	10	1	15	11	12
3Q	8	16	14	5	6
2QAgNP	3	17	9	6	14
5HPC	0	35	12	0	2
6HPC	8	19	15	0	7
8HPC	32	5	5	2	5
10HPC	24	2	3	15	5

Table 4.2 – Device count by category in the different substrates

Even though a clear distinction cannot be made in terms of the I-V characteristics of the individual bipolar switching devices in each of the fabricated substrates, by analyzing Table 4.2, it is clear that the thickness of the electrolyte layer plays a very important role in terms of the yields obtained for each of the fabricated substrates.

As expected, it is observable that the number of short-circuits increases dramatically on the substrates with the lower electrolyte thicknesses (1Q and 5HPC). This is explainable by two theories: (1), silver diffusion during the AE deposition process irreversibly short-circuits the devices, or (2), the electric fields at which the devices are subjected in the initial forming process are too high, creating thick filaments that cannot be ruptured. The increase in the number of pre-set devices on the 5HPC substrate, supports (1).

Unipolar and non-memory devices appear only for the intermediate and higher thickness substrates. Concerning the intermediate and higher thickness substrates, no clear distinction can be made utilizing this sample group, suggesting that there is a wide range of SPE thicknesses that can be used. Nevertheless, further statistical analysis would need to be conducted using a larger sample group in order to determine the effect of using higher SPE thicknesses.

4.4.3. Effect of the electrolyte material on the electrical performance.

In order to study how the electrical performance of the devices change with the different used materials, a small statistical study was conducted on the intermediate thickness devices (2Q and 8HPC) as well as the chitosan-AgNP composite device (2QAgNP).

For that purpose, and taking into consideration equation (2.1), the electric field at which the devices switch their current state was calculated for each of the measured bipolar devices (the reset values of the pre-set devices were taken into account as well), taking into account the measured SPE thickness for each of the substrates. Taking those values into account, the probability of encountering one device with a specific forming electric field in the respective substrate is as follows:

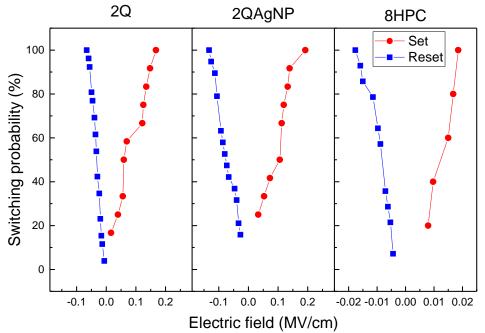


Figure 4.9 – Statistical analysis of the switching probability on the 2Q (left), 2QAgNP (middle) and 8HPC (right) devices

By analyzing the figure above, there is no clear improvement on the usage of the chitosan-AgNP electrolyte instead of a simple chitosan layer. It would be expected that the necessary electric field to set the device would be significantly lower in the AgNP composite, however, the applied electric fields to set and reset the 2Q and 2QAgNP devices are similar. Even though no improvement is clear in this experiment, it is possible that the concentration of AgNP's used was too low to have any effect on the resistive switching of the devices. Further study, using higher concentrations of AgNP's, would be needed to understand the role that AgNP's could have on the resistive switching of these devices.

The 8HPC substrate, shows electrical switching at similar voltage values than the chitosan devices, however, because the SPE films possess a thickness that is about 1 order higher than the chitosan films, the calculated electric field at which the electrical switching occurs is about 1 order lower than the chitosan films. One possibility that could explain the better functioning of HPC at higher thickness values than chitosan is that the defect density in HPC would be much higher than in the chitosan layers, therefore, the thickness of the HPC films need to be higher, in order to not short circuit the device beforehand.

4.4.4. Retention characteristics

In this chapter, results regarding the typical retention characteristics will be presented for devices in the 2Q, 2QSP, 6HPC and 10HPC substrates. Further testing on the other substrates was not possible due to time constraints.

Since it was already shown that device behavior varies greatly from device to device in the same substrate, different programming methods were used in the different measurements in order to obtain the retention results.

The 2Q device was programmed by applying a V_{set} = 1.5 V pulse, during t_{set} = 2.5 s, with a CC = 500 μ A. The HCS was measured by applying a constant V_{read} = 0.2 V bias during 30 h with no CC and measuring the devices current response. The reset was then performed by applying a V_{reset} = -3 V

pulse during $t_{reset} = 2.7$ s with no CC. The LCS was then measured using the same programming as for the HCS. Figure 4.10 shows the results obtained.

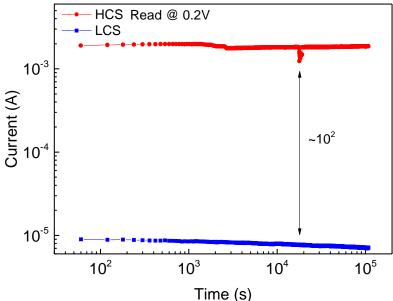


Figure 4.10 - Tipical retention characteristics of the 2Q devices

As illustrated in Figure 4.10, the 2Q device showed remarkable retention characteristics, showing stable data retention times (t_{retention}) of over 10^5 s with a $I_{On/Off} \approx 10^2$.

The 2QSP device was programmed as follows: a 0 V to 6.1 V I-V sweep with a CC = 500 μ A was performed to set the device, the HCS was then measured during 30 h by applying a constant V_{read} = 0.2 V with no CC. The device was then reset by applying a V_{reset} = -3 V during t_{reset} = 2 s, with no CC, then the LCS was measured using the same programming as the HCS.

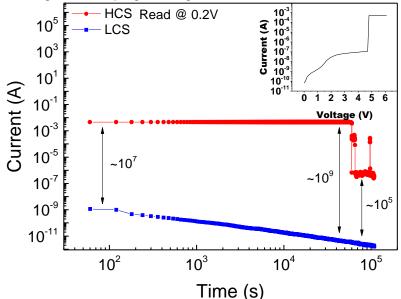


Figure 4.11 – Typical retention characteristics of the 2QSP devices. The inset shows the I-V curve of the initial forming step of the device

The 2QSP device showed retention of one stable HCS with an approximate current value of 4.6 mA during a t_{retention} $\approx 6x10^4$ s, after that, the device seems to switch to a second HCS with an approximate current value of 0.5 µA. The LCS seems to decrease over time, by showing a current variance (ΔI) of approximately 10³. Because of the variance of the LCS, the I_{On/Off} varies from 10⁷ to 10⁹ in the first HCS, and in the second HCS, I_{On/Off} $\approx 10^5$.

The 10HPC device was set by using a Vset = 1.5 V pulse over t_{set} = 2.5 s, with a CC = 5 μ A. The current response was then measured by applying a V_{read} = 0.2 V pulse with a CC of 1 μ A from 10 to 10 min in an effort to avoid stressing the device.

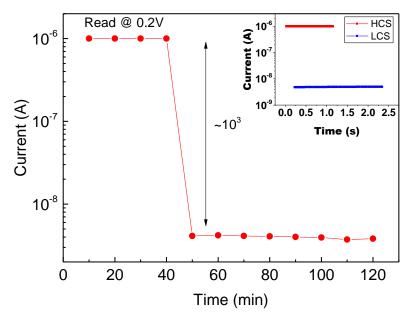


Figure 4.12 – Typical retention characteristics for the 10HPC devices. Inset shows a short retention measurement of both the HCS (red) and LCS (blue)

The tested 10HPC device showed stable HCS = 1 μ A retention during a t_{retention} = 40 min, after which, the current drops to a LCS \approx 0.02 μ A. The inset of Figure 4.12. shows the read current state after a set operation and the current state after a reset operation, demonstrating that the known LCS of the device is comparable to the current state that the device presents for t_{retention} > 40 min. It is probable that this device could show higher retention times if subjected to different programming.

By reducing the thickness of the electrolyte layer, more percolation paths are created between the two electrodes, leading to a better retention of the HCS. Consequently, the lower thickness leads to the stressing of the device on the LCS by application of the read voltage. Figure 4.13 shows the typical retention behavior of a 6HPC device under $V_{read} = 0.2$ V.

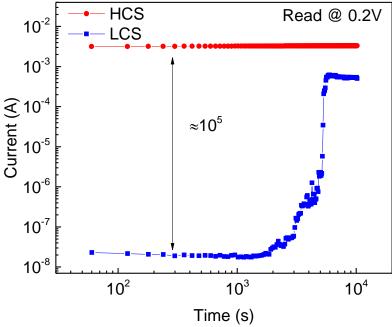


Figure 4.13 – Typical retention characteristics of the 6HPC devices under V_{read} = 0.2 V

The HCS was able to be maintained up to 10^4 s, however, the LCS started to increase its current level at t_{retention} $\approx 2x10^3$ s.

The retention of the LCS can be improved by reducing the used V_{read} . Figure 4.14 – shows the typical retention characteristics of the 6HPC device under $V_{read} = 0.1$ V.

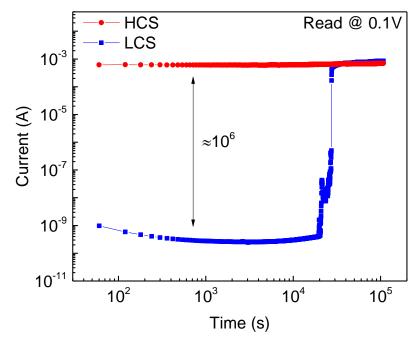


Figure 4.14 – Typical retention characteristics of the 6HPC device under a V_{read} = 0.1 V

At $V_{read} = 0.1$ V, it was possible to maintain the HCS, while improving the retention of the LCS up to $t_{retention} \approx 2x10^4$ s, at which point the LCS turns to the HCS

4.4.5. Endurance characteristics

In this chapter, the typical endurance characteristics for the devices in the following substrates will be presented: 2Q, 10HPC, 2QSP, and 8HPCSP. These tests are made in order to evaluate the devices ability to endure several P/E cycles.

For this purpose, 100 I-V cycles were performed on a 2Q device:

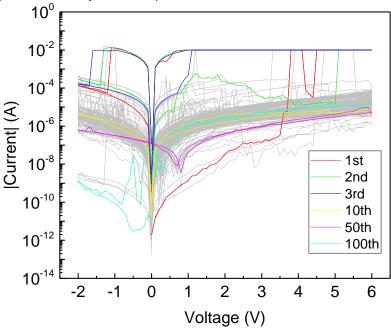


Figure 4.15 – 100 I-V cycles performed on the 2Q device.

It can be observed that the tested device shows rather low endurance, with only around three successful P/E cycles. One explanation for this low endurance is that, by subjecting the device to such high currents for a long period of time, the device switches from a bipolar switching behavior to a unipolar switching behavior. This is further supported by analyzing cycle number 48, which shows a typical unipolar behavior:

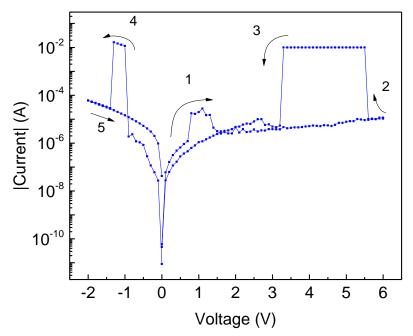


Figure 4.16 – I-V characteristics of the 2Q device at cycle number 48.

This type of behavior could be prevented, by using a different kind of programming. As such, the subsequent testing was performed using timed voltage pulses instead of I-V sweeps in order to reduce the stressing of the devices.

The tests consist on subjecting the selected devices to 100 P/E cycles using timed pulses with certain parameters (V_{set} , t_{set} , V_{reset} , t_{reset} and CC) defined for each of the tested devices, and reading the current response of the device with a $V_{read} = 0.2$ V timed pulse after each set/reset pulse.

The setting parameters for each of the tested devices is presented in Table 4.3:

Substrate	V _{set} (V)	t _{set} (S)	V _{reset} (V)	t _{reset} (S)	CC _{set} (A)	CC _{reset} (A)	HCS CC _{read} (A)	LCS CC _{read} (A)
2Q	1.5	2	-3	2	2x10 ⁻⁶	5x10 ⁻²	5x10 ⁻⁷	5x10 ⁻⁷
10HPC	1.5	2	-3	2	5x10 ⁻⁶	5x10 ⁻²	10 ⁻⁶	10 ⁻⁶
2QSP	3	2	-3	1	5x10 ⁻⁴	10 ⁻¹	10 ⁻¹	5x10 ⁻⁴
8HPCSP	1.5	5	-4	5	10 ⁻³	10 ⁻¹	10 ⁻³	10 ⁻³

Table 4.3 – P/E endurance tests programming parameters for the tested devices

It should be noted that the P/E endurance cycles were performed using Keithley 4200-SCS semiconductor measurement units (SMU) which enable the application of voltage bias for a long period of time (>1s), but offer very low precision in terms of the voltage pulse application times, in contrast to the pulse measurement units (PMU), which offer a very high precision in voltage pulse application times, but can only be used for fast pulses (≤1s). Therefore, the programmed times might differ greatly from the real voltage pulse application times.

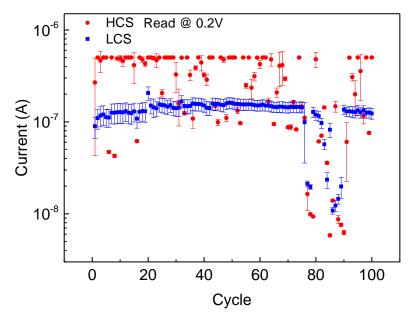


Figure 4.17 - Typical endurance characteristics of the 2Q devices

This device shows two distinct current states with $I_{On/Off} \approx 5$. The presence of intermediate current states between HCS and LCS indicate the presence of an uncontrollable MLC behavior. Furthermore some measurements of the HCS appear on the LCS. One explanation for both of these behaviors is that the device exhibits some V_{set} dispersion (this is a well-known effect for CBRAM devices [9]), and the V_{set} or t_{set} used is not sufficient to set the device at some specific cycles. Another possibility is that after an undetermined number of cycles, the device will start to behave as a unipolar memory, and the set voltage pulse cannot switch the device to the HCS. However, a HCS is still observable at cycle number 100, indicating that the device can exhibit resistive switching for 100 cycles but due to incorrect programming, the device can fail to set at randomly distributed cycles.

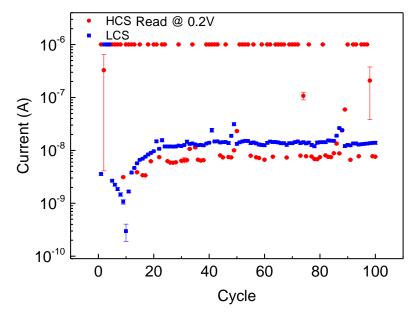


Figure 4.18 – Typical endurance characteristics for the 10HPC devices

The endurance characteristics of the 10HPC device shows similar behavior to that described for the 2Q, however, the LCS shows a current value of 10^{-8} , showing $I_{On/Off} \approx 10^2$. The lower current value of the LCS could be attributed to the higher thickness of the HPC devices when compared to the chitosan devices as shown in Table 4.1. Much like the 2Q device, there are some intermediate current states, indicating an uncontrollable MLC behavior, as well as randomly distributed set failures.

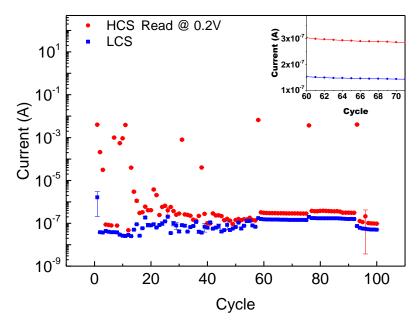


Figure 4.19 – Typical endurance characteristics of the 2QSP device. The inset shows a close up of cycles 60 to 70, demonstranting that a smaller memory window is present

The 2QSP device clearly shows uncontrollable MLC behavior, since there is no well-defined HCS in the initial P/E cycles. There seems, however, to be a tendency for the HCS to decrease, this could be caused by some stressing of the device by the reset voltage pulse. Since the CC does not limit the reset voltage pulse, but limits the set voltage pulse, the filaments within the electrolyte could become iteratively thinner with each cycle, causing the HCS to decrease over time. This issue could possibly be resolved by increasing CC_{set} or t_{set}.

At around cycle number 59, even though the memory window seems to decrease considerably, there seems to be two well defined current states with a $I_{On/Off} \approx 2$. Similar to the previously shown devices, this device might still randomly set, indicating that it might still be working after the endurance test, however, the used programming was not ideal.

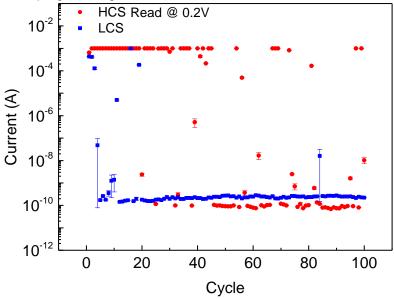


Figure 4.20 - Typical endurance characteristics of the 8HPCSP device

The 8HPCSP device initially shows two stable current states with a $I_{On/Off} \approx 10^7$. Similar to the other presented devices, an uncontrollable MLC behavior is observable by the presence of intermediate current states. At cycle number 46, the device seems to consistently fail the setting procedure. Endurance behavior could be improved by using faster but higher reset pulses.

Endurance characteristics reveal that all tested devices endure multiple P/E cycles. We suggest that most of the failing P/E cycles might be a result of inappropriate programming rather than device

failure. Perhaps, a more appropriate programming would be the usage of faster timed pulses and higher voltage values.

Analyzing the obtained results, no conclusions can be made regarding the degradation of device performance by the usage of screen printed electrodes instead of the evaporated electrodes, suggesting that the fabrication of these devices is adaptable towards printed electronics.

4.4.6. Programming using fast pulses

In an attempt to confirm whether the programming of these types of devices is viable using fast voltage pulses, a simple experiment was performed.

One of the 2QSP devices was selected, and an initial forming was conducted by applying an I-V sweep from 0 V \rightarrow 2 V \rightarrow 0 V, with a CC = 0.5 mA. Figure 4.21 shows the initial switching behavior of the tested device.

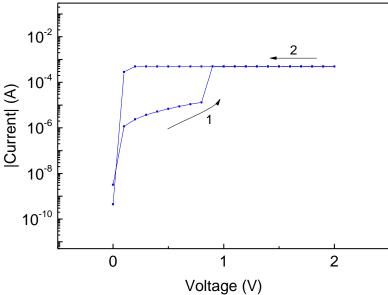


Figure 4.21 - Initial forming step of the tested 2QSP device towards switching using fast pulses

Afterwards, a fast voltage pulse of 0.2 V was applied to read the current value, in order to verify that the device is in fact in the HCS; then a fast -3 V pulse was used to reset the device back to the LCS; and finally, another 0.2 V pulse was used to read the LCS. The programmed pulse parameters are shown in Table 4.4 and the fast I-V pulse measurements are represented in Figure 4.22.

Operation	Voltage amplitude (V)	Period (s)	Pulse width (s)	Rise time (s)	Fall time (s)	Current range (A)
HCS read	0.2	10 ⁻²	10 ⁻³	5x10⁻⁵	5x10⁻⁵	10 ⁻³
Reset	-3	10 ⁻²	10 ⁻³	5x10 ⁻⁴	5x10 ⁻⁴	10 ⁻²
LCS read	0.2	10 ⁻²	10 ⁻³	5x10⁻⁵	5x10⁻⁵	10-4

Table 4.4 – Fast pulses programmed parameters

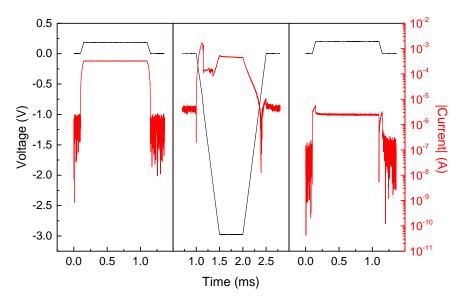


Figure 4.22 – Fast I-V pulses measured on the 2QSP device. HCS read operation on the left, reset operation in the middle and LCS read operation on the right

The depicted current values extracted from the read measurements demonstrates that a reset operation was successful using a -3 V pulse during approximately 2 ms with a $I_{On/Off} \approx 10^2$.

Using a similar programming process, the set operation was attempted, but to no avail, suggesting that, a set process requires a much higher applied voltage or time in order to perform the operation.

Further testing should be attempted in order to correctly determine the voltage/time dependency for the tested devices.

4.4.7. Electrical testing without AE

Throughout all of the electrical testing described above, a possibility was suggested that, even though the main switching mechanism involved would be the ECM effect, other secondary effects could be at play. This would be more evident in the particular cases that exhibit uncontrollable MLC behaviors, as well as some unipolar characteristics.

In an attempt to verify this hypothesis a new type of device with no AE was fabricated. The fabricated structure is comprised of a Pt/10HPC/Pt structure. To measure the I-V characteristics of these devices, 10 cycles consisting of -10 V \rightarrow 10 V \rightarrow -10 V sweeps were performed. The I-V characteristics are shown in Figure 4.23.

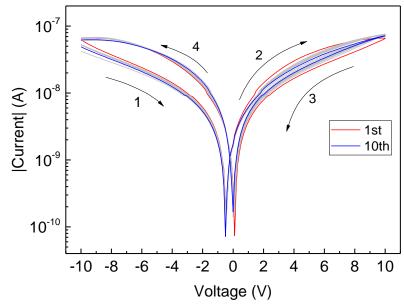


Figure 4.23 - I-V characteristics of the Pt/10HPC/Pt structure

It is clearly demonstrated that no resistive switching occurs in these structures. Nonetheless, some hysteresis can be observed, that could be attributed to the aforementioned switching effects such as VCM or TCM effects, however, without an AE, the primary switching mechanism (ECM) is not present, and thus the device is not able to set to a HCS. This type of device shows very similar behavior to the previously described non-memory devices.

This serves to prove that, the main effect responsible for the electrical switching observed in the previously tested devices is in fact ECM, nevertheless other effects such as VCM and TCM effects cannot be discarded, as it is possible that they could play a secondary role in the switching process.

5. CONCLUSIONS AND FUTURE PERSPECTIVES

In this work, CBRAM devices based on MIM structures involving chitosan and HPC thin films acting as the insulating layer were successfully fabricated. Furthermore, the implementation of AgNP into the chitosan matrix was also studied, as well as the effect of switching the AE from an evaporated Ag electrode to a Ag printed electrode based on an electro conductive ink.

5.1. Final conclusions

Regarding the typical observed I-V characteristics for these devices, five different categories were observed: non-memory, short-circuit, pre-set, unipolar switching and bipolar switching. The number of devices that belong to each of these categories in one substrate is mostly governed by the SPE thickness used. It was shown that the samples with a lower SPE thickness increased the number of short-circuits dramatically, as expected. Unipolar and non-memory devices only appeared on the samples with intermediate and higher thicknesses.

The appearance of pre-set devices seem to suggest the possibility that a filament short-circuiting the device is already present on the SPE matrix, however by application of a negative voltage bias, it is possible to reset the device and initiate subsequent testing. The most probable explanation for the appearance of these devices would be that a thin Ag filament diffuses into the SPE matrix when the Ag AE is deposited through e-beam evaporation.

Devices using chitosan as a SPE showed retention of HCS and LCS of up to 10^5 s (≈ 30 h) with $I_{On/Off} \approx 10^2$. The usage of screen printed electrodes does not seem to degrade the retention times of the devices significantly, as one stable HCS was maintained up to $5x10^4$ s, after that a lower HCS was maintained for the same time. HPC was also able to show high retention times of up to 10^5 s for the HCS and up to $2x10^4$ s for the LCS.

In terms of endurance characteristics, it was shown that a more favorable approach in order not to stress the devices would be to set and reset using timed pulses, instead of a slow I-V sweep. Nevertheless, chitosan and HPC, with evaporated or printed electrodes all showed endurance over several cycles. There were, however, multiple instances in which there was a failure to set the device to HCS. There are several explanations for these failures, in some occasions, the set CC could be too high, making the device adopt a unipolar behavior instead of a bipolar switching behavior; another explanation would be that the devices suffer a shift of V_{set} , making it so that the applied pulse was not enough to set the device.

Switching using fast voltage pulses (≤2 ms) was achieved for the reset operation, but not the set operation, suggesting that the reset process requires much less applied voltage or time to perform the operation. Further study using fast voltage pulses would be needed to correctly determine the voltage/time dependency of the devices.

The functionalization of chitosan with AgNP's showed similar switching probabilities at similar electric fields than the tested chitosan devices with no functionalization, indicating that there is no apparent benefit on the usage of AgNP within the film. However, further study would be needed using higher AgNP concentrations in the film to effectively evaluate how the electrical behavior would change with the usage of nanoparticles embedded into the SPE.

Regarding the electrolyte material, even though chitosan showed retention times of 1 order higher than the tested HPC devices, both materials exhibited resistive switching, and similar endurance characteristics. We suggest that under a different programming, the HPC device could exhibit much more interesting retention properties, the same is true for the endurance testing.

Throughout all of the electrical testing it was shown that the used electrical programming has a significant impact on the electrical functioning of the devices, whether it be, the initial forming characteristics, retention or endurance measurements. Annex N shows a proposed generic process towards the initial programming of the devices.

5.2. Future perspectives

This work showed that the integration of solution processed biopolymers into CBRAM devices is definitely a possibility, and doing so, new doors are opened to further research in the area of solution processed electronics, as well as printed electronics involving memory devices.

In this work, no definite conclusion could be made in regards to the area dependency of the switching effects, mostly due to the limited sample group, as such, a more extensive statistical study would be needed to reach to some conclusion. We also suggest that all of the devices should be fabricated with the same electrode area, only then, if all of the results are systematic for a given electrode size, a conclusion can be made regarding the dependency of the electrode area on the electrical switching phenomena, and if that correlation is made, then it is proven that the electrical switching in these types of biopolymers is not solely due to an ECM effect.

In regards to the thickness dependency on the electrical functioning of the devices, even though it was proven that the thickness plays an important role on the yields presented by each substrate, further statistical analysis could prove useful in order to obtain an optimal thickness for each of the used materials.

Further electrical testing using programming with fast pulses (≤ 1 s) could prove very useful, as this information would not only provide some insight into the switching speeds that these devices could actually achieve, but by using fast pulses, the endurance behavior of the devices could improve significantly as the stressing of the device during the set and reset processes would be reduced significantly.

As a concluding remark, by proving that a cellulose derivative could in fact be used as an insulating layer in CBRAM devices, there is a new array of possibilities that could be tested by combining functionalized paper and cellulosic materials with printed electronics towards the development of memory devices based on resistive switching.

6. BIBLIOGRAPHY

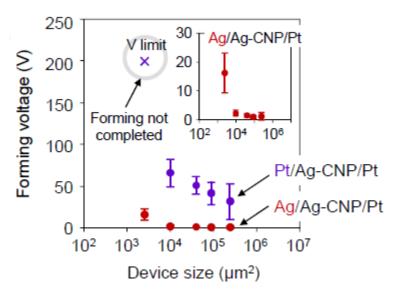
- K. Prall, N. Ramaswamy, and A. Goda, "Charge-Trapping Non-Volatile Memories: Volume 1 --Basic and Advanced Devices," P. Dimitrakis, Ed. Cham: Springer International Publishing, 2015, pp. 37–64.
- [2] K. Aratani, K. Ohba, T. Mizuguchi, S. Yasuda, T. Shiimoto, T. Tsushima, T. Sone, K. Endo, A. Kouchiyama, S. Sasaki, A. Maesaka, N. Yamada, and H. Narisawa, "A novel resistance memory with high scalability and nanosecond switching," *Tech. Dig. Int. Electron Devices Meet. IEDM*, pp. 783–786, 2007.
- [3] S. Dietrich, M. Angerbauer, M. Ivanov, D. Gogl, H. Hoenigschmid, M. Kund, C. Liaw, M. Markert, R. Symanczyk, L. Altimime, S. Bournat, and G. Mueller, "A nonvolatile 2-Mbit CBRAM memory core featuring advanced read and program control," *IEEE J. Solid-State Circuits*, vol. 42, no. 4, pp. 839–845, 2007.
- [4] Y. Hayakawa, A. Himeno, R. Yasuhara, W. Boullart, E. Vecchio, T. Vandeweyer, T. Witters, D. Crotti, M. Jurczak, S. Fujii, S. Ito, Y. Kawashima, Y. Ikeda, A. Kawahara, K. Kawai, Z. Wei, S. Muraoka, K. Shimakawa, T. Mikawa, and S. Yoneda, "Highly reliable TaOx ReRAM with centralized filament for 28-nm embedded application," *Dig. Tech. Pap. Symp. VLSI Technol.*, vol. 2015-Augus, no. 2011, pp. T14–T15, 2015.
- [5] U. Russo, D. Kamalanathan, D. Ielmini, A. L. Lacaita, and M. N. Kozicki, "Study of multilevel programming in Programmable Metallization Cell (PMC) memory," *IEEE Trans. Electron Devices*, vol. 56, no. 5, pp. 1040–1047, 2009.
- [6] Yole Développement, "Émerging Non Volatile Memory (NVM) Technology & Market Trends Report," Lyon, 2015.
- [7] I. Valov, R. Waser, J. R. Jameson, and M. N. Kozicki, "Electrochemical metallization memories--fundamentals, applications, prospects.," *Nanotechnology*, vol. 22, no. 25, p. 254003, 2011.
- [8] R. Waser and M. Aono, "Nanoionics-based resistive switching memories.," *Nat. Mater.*, vol. 6, no. 11, pp. 833–40, 2007.
- D. Jana, S. Roy, R. Panja, M. Dutta, S. Z. Rahaman, R. Mahapatra, and S. Maikap,
 "Conductive-bridging random access memory: challenges and opportunity for 3D architecture.," Nanoscale Res. Lett., vol. 10, no. 1, p. 188, 2015.
- [10] D. Tobjörk and R. Österbacka, "Paper electronics," *Adv. Mater.*, vol. 23, no. 17, pp. 1935–1961, 2011.
- [11] D.-H. Lien, Z.-K. Kao, T.-H. Huang, Y.-C. Liao, S.-C. Lee, and J.-H. He, "All-printed paper memory," *ACS Nano*, vol. 8, no. 8, pp. 7613–7619, 2014.
- [12] Y.-C. Chen, H.-C. Yu, C.-Y. Huang, W.-L. Chung, S.-L. Wu, and Y.-K. Su, "Nonvolatile Bio-Memristor Fabricated with Egg Albumen Film," *Sci. Rep.*, vol. 5, no. MAY 2015, p. 10022, 2015.
- [13] N. R. Hosseini and J.-S. Lee, "Biocompatible and Flexible Chitosan-Based Resistive Switching Memory with Magnesium Electrodes," *Adv. Funct. Mater.*, p. n/a–n/a, 2015.
- [14] M. K. Hota, M. K. Bera, B. Kundu, S. C. Kundu, and C. K. Maiti, "A natural silk fibroin proteinbased transparent bio-memristor," *Adv. Funct. Mater.*, vol. 22, no. 21, pp. 4493–4499, 2012.
- [15] Y. C. Hung, W. T. Hsu, T. Y. Lin, and L. Fruk, "Photoinduced write-once read-many-times memory device based on DNA biopolymer nanocomposite," *Appl. Phys. Lett.*, vol. 99, no. 25, pp. 13–16, 2011.
- [16] N. Raeis Hosseini and J. Lee, "Resistive Switching Memory Based on Bioinspired Natural Solid Polymer Electrolytes," *ACS Nano*, vol. 9, no. 1, pp. 419–426, 2015.
- [17] H. Wang, F. Meng, Y. Cai, L. Zheng, Y. Li, Y. Liu, Y. Jiang, X. Wang, and X. Chen, "Sericin for resistance switching device with multilevel nonvolatile memory," *Adv. Mater.*, vol. 25, no. 38, pp. 5498–5503, 2013.
- [18] N. Raeis-Hosseini and J.-S. Lee, "Controlling the Resistive Switching Behavior in Starch-Based Flexible Biomemristors," *ACS Appl. Mater. Interfaces*, p. acsami.6b01559, 2016.
- [19] K. Nagashima, H. Koga, U. Celano, F. Zhuge, M. Kanai, S. Rahong, G. Meng, Y. He, J. De Boeck, M. Jurczak, W. Vandervorst, T. Kitaoka, M. Nogi, and T. Yanagida, "Cellulose nanofiber paper as an ultra flexible nonvolatile memory.," *Sci. Rep.*, vol. 4, p. 5532, 2014.
- [20] N. Gogurla, S. P. Mondal, A. K. Sinha, A. K. Katiyar, W. Banerjee, S. C. Kundu, and S. K. Ray, "Transparent and flexible resistive switching memory devices with a very high ON/OFF ratio using gold nanoparticles embedded in a silk protein matrix," *Nanotechnology*, vol. 24, no. 34, p. 345202, 2013.

- [21] D. S. Jeong, R. Thomas, R. S. Katiyar, J. F. Scott, H. Kohlstedt, a Petraru, and C. S. Hwang, "Emerging memories: resistive switching mechanisms and current status," *Reports Prog. Phys.*, vol. 75, no. March 2016, p. 076502, 2012.
- [22] H. S. P. Wong, S. Raoux, S. Kim, J. Liang, J. P. Reifenberg, B. Rajendran, M. Asheghi, and K. E. Goodson, "Phase change memory," *Proc. IEEE*, vol. 98, no. 12, pp. 2201–2227, 2010.
- [23] M. Wuttig and N. Yamada, "Phase-change materials for rewriteable data storage," *Nat. Mater.*, vol. 6, no. 11, pp. 824–832, 2007.
- [24] J. J. Yang, I. H. Inoue, T. Mikolajick, and C. S. Hwang, "Metal oxide memories based on thermochemical and valence change mechanisms," *MRS Bull.*, vol. 37, no. 02, pp. 131–137, 2012.
- [25] R. Waser, R. Dittmann, C. Staikov, and K. Szot, "Redox-based resistive switching memories nanoionic mechanisms, prospects, and challenges," *Adv. Mater.*, vol. 21, no. 25–26, pp. 2632–2663, 2009.
- [26] S. Thomas, D. Durand, C. Chassenieux, and P. Jyotishkumar, *Handbook of Biopolymer-Based Materials*. 2013.
- [27] D. Platt, Biodegradable polymers: market report. 2006.
- [28] M. Berggren, D. Nilsson, and N. D. Robinson, "Organic materials for printed electronics.," *Nat. Mater.*, vol. 6, no. 1, pp. 3–5, 2007.
- [29] B. Lindman, G. Karlström, and L. Stigsson, "On the mechanism of dissolution of cellulose," *J. Mol. Liq.*, vol. 156, no. 1, pp. 76–81, 2010.
- [30] T. Huber, J. Müssig, O. Curnow, S. Pang, S. Bickerton, and M. P. Staiger, "A critical review of all-cellulose composites," *J. Mater. Sci.*, vol. 47, no. 3, pp. 1171–1186, 2012.
- [31] M. Granström and P. I. Kilpeläinen, *Cellulose Derivatives: Synthesis, Properties and Applications*, vol. PhD, no. May. 2009.
- [32] E. S. Abdel-Halim and S. S. Al-Deyab, "Utilization of hydroxypropyl cellulose for green and efficient synthesis of silver nanoparticles," *Carbohydr. Polym.*, vol. 86, no. 4, pp. 1615–1622, 2011.
- [33] G. Romanazzi, F. M. Gabler, D. Margosan, B. E. Mackey, and J. L. Smilanick, "Effect of chitosan dissolved in different acids on its ability to control postharvest gray mold of table grape.," *Phytopathology*, vol. 99, no. 9, pp. 1028–1036, 2009.
- [34] A. C. Baptista, J. I. Martins, E. Fortunato, R. Martins, J. P. Borges, and I. Ferreira, "Thin and flexible bio-batteries made of electrospun cellulose-based membranes," *Biosens. Bioelectron.*, vol. 26, no. 5, pp. 2742–2745, 2011.
- [35] R. H. Y. Subban, a. K. Arof, and S. Radhakrishna, "Polymer batteries with chitosan electrolyte mixed with sodium perchlorate," *Mater. Sci. Eng. B*, vol. 38, no. 1–2, pp. 156–160, 1996.
- [36] N. A. Choudhury, P. W. C. Northrop, A. C. Crothers, S. Jain, and V. R. Subramanian, "Chitosan hydrogel-based electrode binder and electrolyte membrane for EDLCs: Experimental studies and model validation," *J. Appl. Electrochem.*, vol. 42, no. 11, pp. 935–943, 2012.
- [37] L.-F. Chen, Z.-H. Huang, H.-W. Liang, W.-T. Yao, Z.-Y. Yu, and S.-H. Yu, "Flexible all-solidstate high-power supercapacitor fabricated with nitrogen-doped carbon nanofiber electrode material derived from bacterial cellulose," *Energy Environ. Sci.*, vol. 6, no. 11, pp. 3331–3338, 2013.
- [38] K. H. Teoh, C.-S. Lim, C.-W. Liew, S. Ramesh, and S. Ramesh, "Electric double-layer capacitors with corn starch-based biopolymer electrolytes incorporating silica as filler," *Ionics (Kiel).*, vol. 21, no. 7, pp. 2061–2068, 2015.
- [39] S. Yamazaki, A. Takegawa, Y. Kaneko, J. ichi Kadokawa, M. Yamagata, and M. Ishikawa, "An acidic cellulose-chitin hybrid gel as novel electrolyte for an electric double layer capacitor," *Electrochem. commun.*, vol. 11, no. 1, pp. 68–70, 2009.
- [40] E. Fortunato, N. Correia, P. Barquinha, L. Pereira, G. Gonçalves, and R. Martins, "High-Performance Flexible Hybrid Field-Effect Transistors Based on Cellulose Fiber Paper," vol. 29, no. 9, pp. 2008–2010, 2008.
- [41] D. Gaspar, S. N. Fernandes, a G. de Oliveira, J. G. Fernandes, P. Grey, R. V Pontes, L. Pereira, R. Martins, M. H. Godinho, and E. Fortunato, "Nanocrystalline cellulose applied simultaneously as the gate dielectric and the substrate in flexible field effect transistors.," *Nanotechnology*, vol. 25, no. 9, p. 094008, 2014.
- [42] R. Martins, P. Barquinha, L. Pereira, N. Correia, G. Gonalves, I. Ferreira, and E. Fortunato, "Write-erase and read paper memory transistor," *Appl. Phys. Lett.*, vol. 93, no. 20, p. 203501, 2008.
- [43] R. Martins, A. Nathan, R. Barros, L. Pereira, P. Barquinha, N. Correia, R. Costa, A. Ahnood, I. Ferreira, and E. Fortunato, "Complementary metal oxide semiconductor technology with and on

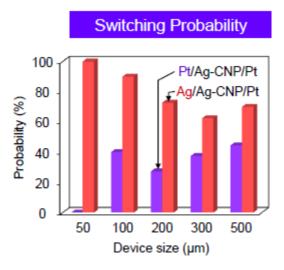
paper," Adv. Mater., vol. 23, no. 39, pp. 4491–4496, 2011.

- [44] L. Pereira, D. Gaspar, D. Guerin, A. Delattre, E. Fortunato, and R. Martins, "The influence of fibril composition and dimension on the performance of paper gated oxide transistors.," *Nanotechnology*, vol. 25, no. 9, pp. 1–11, 2014.
- [45] S.-J. Kim, D.-B. Jeon, J.-H. Park, M.-K. Ryu, J.-H. Yang, C.-S. Hwang, G.-H. Kim, and S.-M. Yoon, "Nonvolatile Memory Thin-Film Transistors Using Biodegradable Chicken Albumen Gate Insulator and Oxide Semiconductor Channel on Eco-Friendly Paper Substrate," ACS Appl. Mater. Interfaces, vol. 7, no. 8, pp. 4869–4874, 2015.
- [46] Z. Bin, S. Jia, H. Xiao, J. Jie, and W. Qing, "Low-Voltage Organic/Inorganic Hybrid Transparent Thin-Film Transistors Gated by Chitosan-Based Proton Conductors," *Electron Device Lett. IEEE*, vol. 32, no. 11, pp. 1549–1551, 2011.
- [47] A. S. Dehnavi, A. Aroujalian, A. Raisi, and S. Fazel, "Preparation and characterization of polyethylene/silver nanocomposite films with antibacterial activity," *J. Appl. Polym. Sci.*, vol. 127, no. 2, pp. 1180–1190, 2013.
- [48] A. C. Dürr, F. Schreiber, M. Kelsch, H. D. Carstanjen, H. Dosch, and O. H. Seeck, "Morphology and interdiffusion behavior of evaporated metal films on crystalline diindenoperylene thin films," *J. Appl. Phys.*, vol. 93, no. 9, pp. 5201–5209, 2003.
- [49] University Colombia, "Spin Coat Theory," *Cost Effective Equipment*, 2009. [Online]. Available: http://nano.columbia.edu/files/cise/spintheory_0.pdf. [Accessed: 23-Mar-2016].

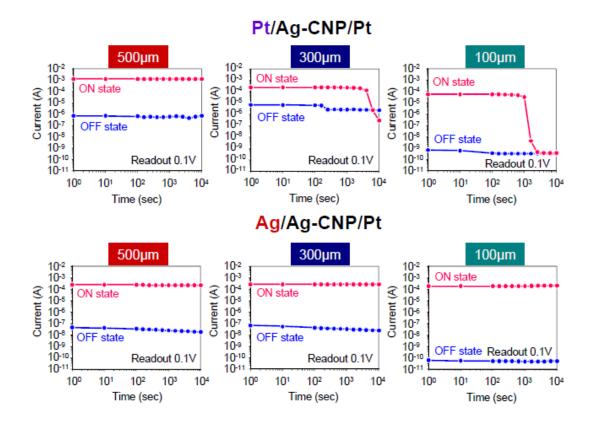
7. ANNEX



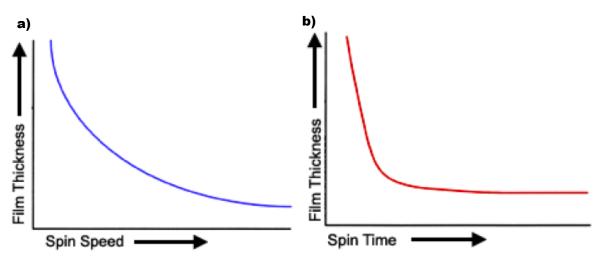
Annex A - Forming voltage dependency on device area and AE material for CNP devices by Nagashima et al[19]



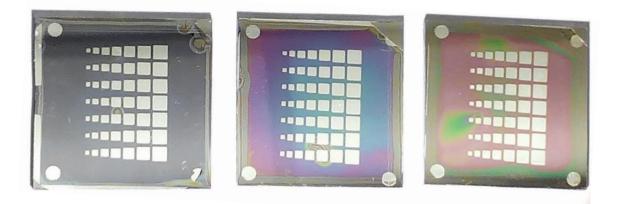
Annex B - Switching probability dependency on device area and AE material on CNP devices by Nagashima et al[19]



Annex C - Retention dependence on area and different AE material on CNP devices by Nagashima et al[19]



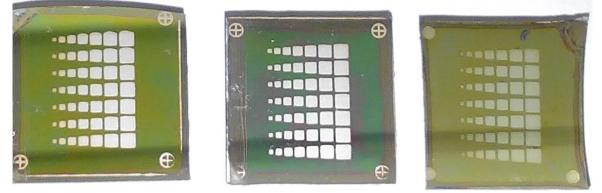
Annex D – General relation between film thickness and spin speed (a) and spin time (b) in spin coating. Adapted from [49]



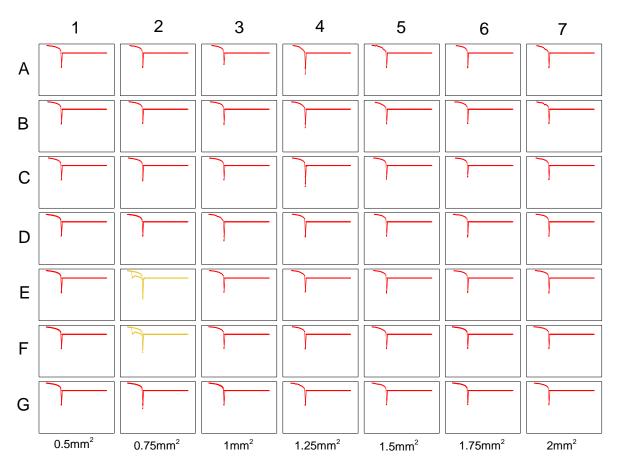
Annex E - Image of the 1Q (left), 2Q (middle) and 3Q (right) substrates

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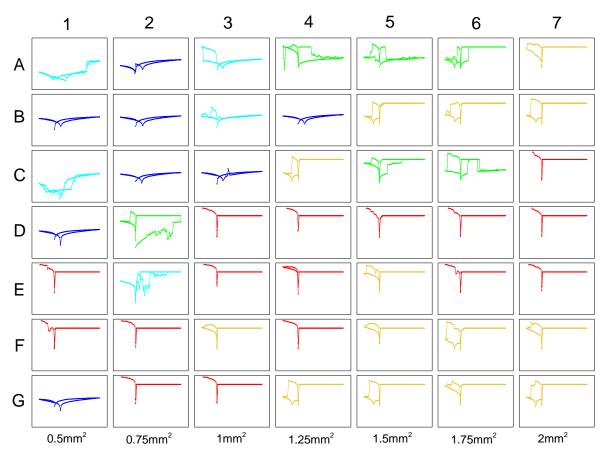
Annex F - Image of the 5HPC (left), 8HPC (middle), and 10 HPC (right) substrates



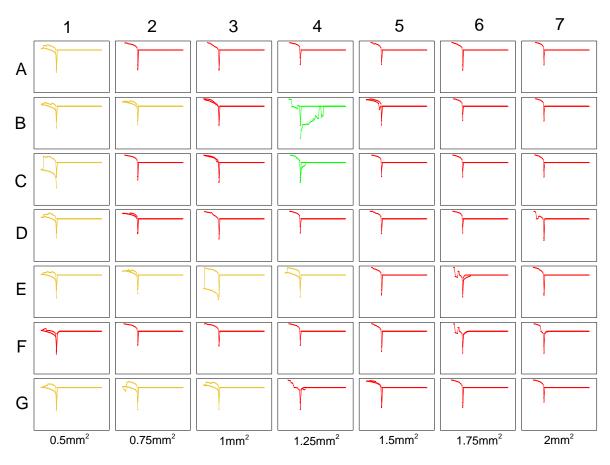
Annex G – Image of the 2QSP (left), 8HPCSP (middle) and 2QAgNP (right) substrates



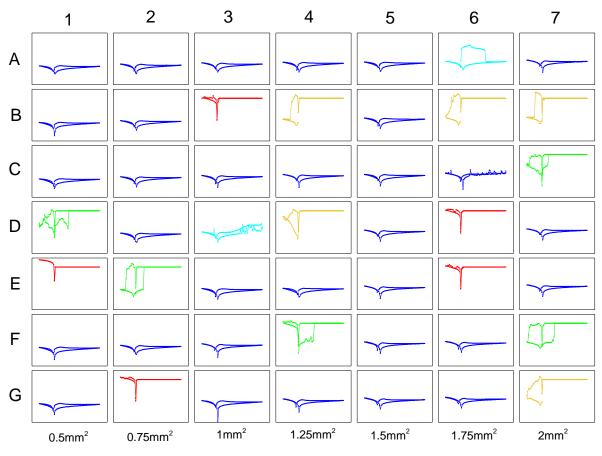
Annex H - I-V characteristics of all of the devices on the 1Q substrate.



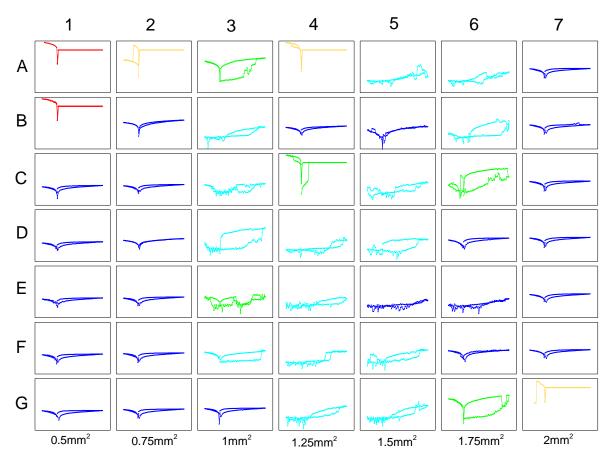
Annex I - I-V characteristics of all of the devices on the 3Q substrate



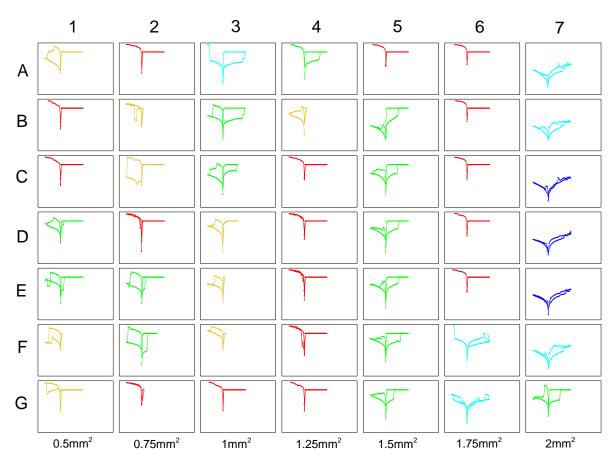
Annex J - I-V characteristics of all of the devices on the 5HPC substrate



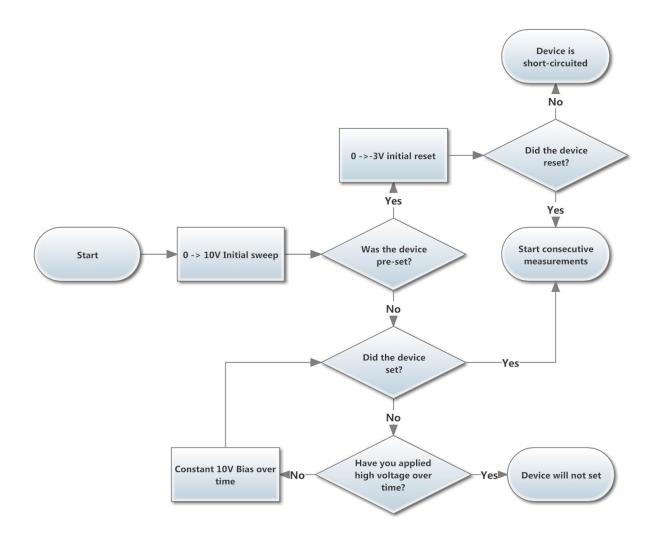
Annex K - I-V characteristics of all of the on the 8HPC substrate



Annex L - I-V characteristics of all of the devices on the 10HPC substrate



Annex M - I-V characteristics of all of the devices on the 2QAgNP substrate



Annex N – Flowchart of the recommended generic initial programming process