



IUCrJ

ISSN 2052-2525

MATERIALS | COMPUTATION

Received 22 January 2018

Accepted 3 May 2018

Edited by A. Fitch, ESRF, France

Keywords: ferroelectrics; piezoelectrics; time-resolved X-ray diffraction; polarization rotation.**Supporting information:** this article has supporting information at www.iucrj.org

Monoclinic distortion, polarization rotation and piezoelectricity in the ferroelectric $\text{Na}_{0.5}\text{Bi}_{0.5}\text{TiO}_3$

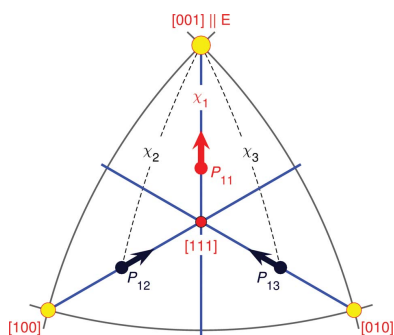
Hyeokmin Choe,^a Johannes Bieker,^b Nan Zhang,^c Anthony Michael Glazer,^{d,e} Pam A. Thomas^e and Semën Gorfman^{f,*}

^aDepartment of Physics, University of Siegen, Walter-Flex Strasse 3, Siegen 57072, Germany, ^bInstitute of Electromechanical Design, Technische Universität Darmstadt, Darmstadt, Germany, ^cElectronic Materials Research Laboratory, Key Laboratory of the Ministry of Education and International Center for Dielectric Research, Xian Jiaotong University, Xian, People's Republic of China, ^dPhysics Department, University of Oxford, Clarendon Laboratory, Parks Road, Oxford OX1 3PU, England, ^eDepartment of Physics, University of Warwick, Gibbet Hill Road, Coventry CV4 7AL, England, and ^fMaterials Science and Engineering, Tel Aviv University, Wolfson Building for Mechanical Engineering, Tel Aviv 6997801, Israel. *Correspondence e-mail: gorfman@tauex.tau.ac.il

The relationship between crystal structure and physical properties in the ferroelectric $\text{Na}_{0.5}\text{Bi}_{0.5}\text{TiO}_3$ (NBT) has been of interest for the last two decades. Originally, the average structure was held to be of rhombohedral ($R3c$) symmetry with a fixed polarization direction. This has undergone a series of revisions, however, based on high-resolution X-ray diffraction, total neutron scattering, and optical and electron microscopy. The recent experimental findings suggest that the true average symmetry is monoclinic (space group Cc), which allows for a rotatable spontaneous polarization. Neither polarization rotation nor its potentially important real role in enhanced piezoelectricity is well understood. The present work describes an *in situ* investigation of the average monoclinic distortion in NBT by time-resolved single-crystal X-ray diffraction under external electric fields. The study presents a high-resolution inspection of the characteristic diffraction features of the monoclinic distortion – splitting of specific Bragg reflections – and their changes under a cyclic electric field. The results favour a model in which there is direct coupling between the shear monoclinic strain and the polarization rotation. This suggests that the angle of polarization rotation under a sub-coercive electric field could be 30° or more.

1. Introduction

$\text{Na}_{0.5}\text{Bi}_{0.5}\text{TiO}_3$ (NBT), a perovskite-based ferroelectric, has been the focus of attention for over two decades (Vakhrushev *et al.*, 1985; Roleder *et al.*, 2002; McQuade & Dolgos, 2016). NBT is of interest because of its potentially important role as an end member of many lead-free substitutes to replace the commercially dominant $\text{PbZr}_{1-x}\text{Ti}_x\text{O}_3$ piezoelectric (Takehana *et al.*, 1991, 2008; ShROUT & Zhang, 2007; Panda, 2009; Rödel *et al.*, 2009). NBT is also an interesting model system in the crystallography of distorted perovskites. Phase transitions in NBT are realized through symmetry-lowering shifts of the *A/B* cations and tilting of the TiO_6 octahedra. This symmetry-lowering results in the formation of domains that are spontaneously polarized, electromechanically active and switchable by an external electric field. Although all these phenomena are ubiquitous in many other perovskite-based materials (Mitchell, 2003), NBT is one of the most complex and a number of unresolved controversies remain. The structure–property relationships of NBT continue to be the subject of debate and prompt continued research work on this unusual material.

OPEN  ACCESS

The commonly accepted crystallographic reference for NBT comes from the neutron powder diffraction work of Jones & Thomas (2002). This work reported a transformation from an average rhombohedral ($R3c$) to an average tetragonal ($P4bm$) phase at ~ 580 K, following the reorientation of the spontaneous polarization from the body-diagonal to the cell-edge direction (the pseudocubic cell setting with the lattice parameter $a \simeq 3.9$ Å is used for lattice directions and reflection indices throughout this paper.). Most surprisingly, the TiO_6 octahedra undergo a change in their tilt system from $a^-a^-a^-$ to $a^0a^0c^+$ (according to Glazer's notation system; Glazer, 1972). The absence of a displacive path for such a phase transition is expressed by the lack of any group-subgroup relationship between the $R3c$ and $P4bm$ space groups. In some respects, this thermally driven $R-T$ phase transformation resembles the compositionally driven one in $\text{PbZr}_{1-x}\text{Ti}_x\text{O}_3$ (PZT) at the morphotropic phase boundary (MPB, $x \simeq 0.48$; Jaffe *et al.*, 1954, 1971). In PZT, the Zr-rich side has been recognized as monoclinic rather than rhombohedral close to the MPB, although recent work has shown that the structure is extremely complex, with mixing of disordered and ordered monoclinic regions (Noheda *et al.*, 1999; Yokota *et al.*, 2009; Zhang *et al.*, 2011, 2014, 2018; Gorfman *et al.*, 2011).

Gorfman & Thomas (2010) have reported a high-resolution X-ray diffraction study of multi-domain NBT 'single' crystals. They demonstrated that the angular separation between the different components of the $\{hkl\}^1$ Bragg reflections (each component is diffracted from a similarly oriented set of ferroelastic domains) violates rhombohedral $R3c$ symmetry and favours monoclinic Cc average symmetry instead. This average monoclinic symmetry of room-temperature NBT has since been confirmed by many other authors (Aksel *et al.*, 2011; Ma *et al.*, 2013; Gorfman *et al.*, 2012; Kitanaka *et al.*, 2014; Levin & Reaney, 2012). The most important implication of monoclinic symmetry is that it allows the polarization to rotate (Vanderbilt & Cohen, 2001) and thus leads to greater susceptibility to an external electric perturbation (Fu & Cohen, 2000). The direct effect of the polarization rotation on the lattice parameters might be the reason for the strong electromechanical coupling in both NBT and PZT. For the case of NBT, the true nature of the monoclinic symmetry remains controversial, especially as the average structure is reported to be different from the local one (Aksel *et al.*, 2013). Neutron scattering studies of pair-distribution function in NBT by Keeble *et al.* (2013) support the local monoclinic symmetry of the Bi positions, where Bi atoms may displace in two different 'monoclinic' directions. At the same time, extended X-ray absorption fine structural studies by Rao *et al.* (2016) show that the local symmetry of the Bi sites is rhombohedral (consistent with $R3c$), while the apparent monoclinic symmetry averages out the combination of different orienta-

tion variants of unit cells of rhombohedral symmetry. Regardless of whether the apparent monoclinic distortion has a true local-scale origin or results from the averaging, attempts to test the susceptibility of the monoclinic phase 'in action' (dynamically under an electric field) are still rare and mainly limited to the use of static electric fields (Ogino *et al.*, 2014; Kitanaka *et al.*, 2014).

The aim of this work is to investigate the apparent monoclinic distortion in NBT under an alternating sub-coercive (<14 kV cm^{-1}) external electric field, test the polarization rotation and clarify if this rotation can give rise to high piezoelectricity. We have implemented a stroboscopic data-acquisition system and high-resolution X-ray diffractometer (beamline P08 at the PETRA III storage ring) to collect reciprocal-space maps (RSMs) around the family of most representative Bragg reflections (Gorfman & Thomas, 2010). We have observed that the monoclinic splitting is indeed strongly sensitive to the external electric field: electric-field-induced shifts of the peaks amount to a piezoelectric effect of as much as 124 pC N^{-1} . The positions of the Bragg peaks in reciprocal space are consistent with the existence of 12 monoclinic domains, in which the polarization vector rotates in one of the 12 monoclinic $\{110\}$ mirror planes. Most importantly, we report that the average shear lattice strain is

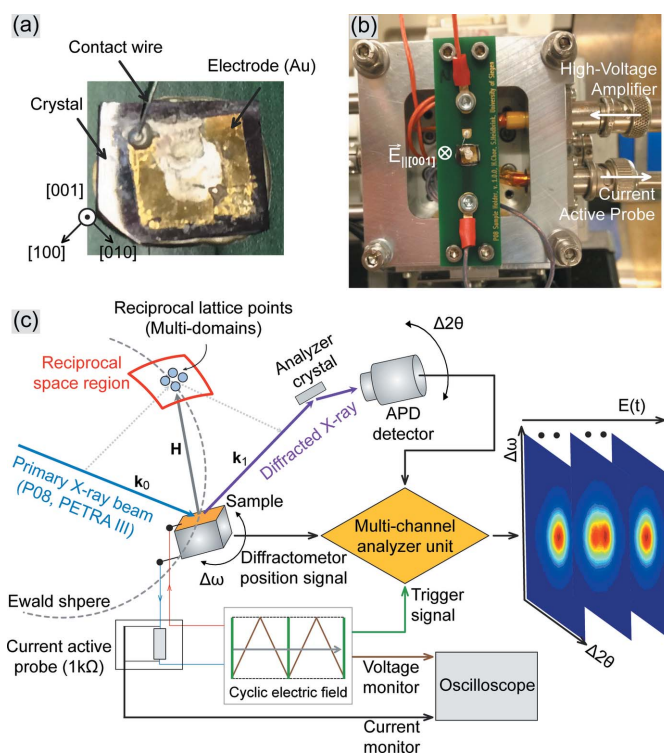


Figure 1 The experimental setup for stroboscopic high-resolution reciprocal space mapping. (a) As-grown and prepared sample, with the crystallographic orientation of the edges shown. (b) A view of the sample holder for application of the electric field. (c) A flow chart of the stroboscopic data-acquisition process. The diffracted signal is collected using a single-photon-counting APD detector placed behind an Si(111) analyser crystal in ω versus 2θ scan mode. A cyclic 100 Hz triangular-shaped electric field is applied to the sample. Detector signals are processed using a custom-built stroboscopic data-acquisition system.

¹ $\{hkl\}_{PC}$ denotes all possible reflections, obtained by applying the cubic $m3m$ point symmetry group to the reflection hkl . For cubic symmetry, the lengths of such reciprocal-space vectors would be the same. However, when the symmetry is distorted the lengths of the vectors vary, giving characteristic splitting of reflections for every crystal system.

nonlinear with electric field and this nonlinearity can be well accounted for by the polarization rotation, with the maximum angle of polarization rotation reaching 35° .

2. Experimental details

The NBT single crystal was grown at the Shanghai Institute of Ceramics by the top-seeded solution-growth method (as described by Ge *et al.*, 2008) and doped with Mn. The crystal was cut to a 0.5 mm thick plate with the surface parallel to (001) and the edges along the [110] and [100] crystallographic directions. Thin (~ 100 nm) gold electrodes were sputtered onto the faces to apply the electric field along [001]. We designed a sample stage, which serially connects the electrodes with a high-voltage supply *via* a 1 k Ω active-probe monitor of the capacitive current. The current and polarization hysteresis loops were monitored continuously during the measurement.

Fig. 1 shows the experimental equipment on the P08 high-resolution four-circle diffractometer at the PETRA III storage ring. The arbitrary function generator (HMF-2550, Hameg) and high-voltage amplifier (AMT-3B20, Matsusada) produce a triangular-shaped AC high-voltage signal/electric field with an amplitude of 14 kV cm^{-1} . This field is significantly smaller than the coercive field of $\sim 45\text{ kV cm}^{-1}$ reported for an Mn-doped NBT single crystal (Ge *et al.*, 2010). We used an avalanche photodiode (APD) single-photon counting detector and Si(111) analyser crystal and measured the scattering intensity as a function of ω (rocking) and 2θ (scattering) angles around the [004]* position of reciprocal space. The output of

the APD detector was introduced directly into a custom-built stroboscopic data-acquisition system (Gorfman *et al.*, 2010, 2013; Gorfman, 2014; Choe *et al.*, 2015, 2017). The system implements the working principle of a multi-channel analyser: it assigns detector counts to one of 10 000 time channels, where each channel has a fixed time delay to the beginning of latest high-voltage cycle. Each point in the RSM was collected for $10\text{ s} = 1000$ electric-field cycles. The frequency of the applied electric field was 100 Hz and the time resolution (channel width) was $1\text{ }\mu\text{s}$. The X-ray energy was set to 15.1 keV ($\lambda = 0.827\text{ \AA}$), just below the 'Bi' L_2 absorption edge, giving an average penetration depth for the measured reflection of $\langle t \rangle = \sin\theta/2\mu = 5.3\text{ }\mu\text{m}$. This is ~ 2.5 times deeper than the penetration depth which was previously used in the experiment of Gorfman & Thomas (2010).

3. Results

Fig. 2(a) reproduces an ω versus 2θ RSM of one of the {002} reflections from our previous studies (Gorfman & Thomas, 2010) (measured using a home-laboratory high-resolution PANalytical MRD diffractometer). This RSM contains two Bragg peaks, separated along the 2θ axis and diffracted from two families of ferroelastic domains. Different 2θ angles mean different lengths of the corresponding reciprocal-lattice vectors [$H = (2\sin\theta)/\lambda$]. Accordingly, such splitting violates the rhombohedral symmetry of the domains, which would constrain the average pseudocubic lattice parameters to be $a = b = c$ and $\alpha = \beta = \gamma$. The work of Gorfman & Thomas (2010)

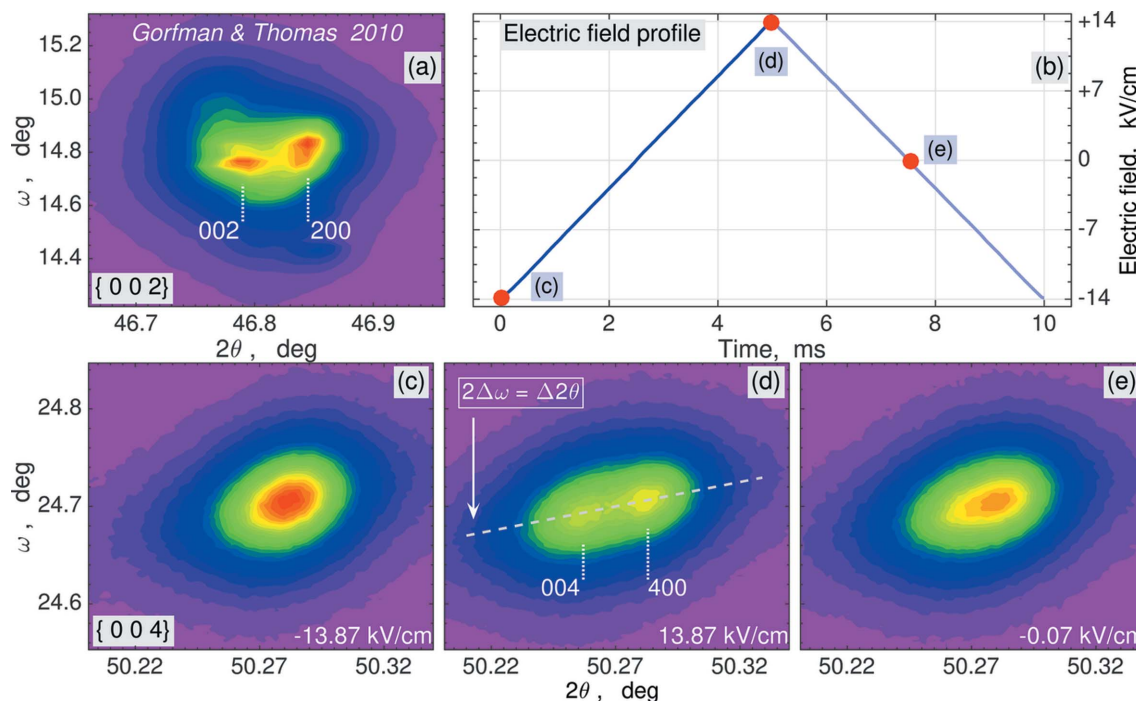


Figure 2

The intensity of the X-ray scattering around the {00*h*} family of reflections. (a) Static ω versus 2θ mesh of the {002} reflections family, regenerated from the earlier data set of Gorfman & Thomas (2010). (b) The time-dependence of the applied external electric field. (c)–(e) Stroboscopically collected ω versus 2θ meshes of the {004} family of reflections, corresponding to different time channels and electric fields. The splitting along the 2θ axis violates the trigonal symmetry of the pseudocubic lattice. This splitting is commonly recognized as a ‘fingerprint’ of the M_A/M_B monoclinic distortion in perovskites (see e.g. Noheda *et al.*, 1999). An animated version of panels (c)–(e) is available in the supporting information.

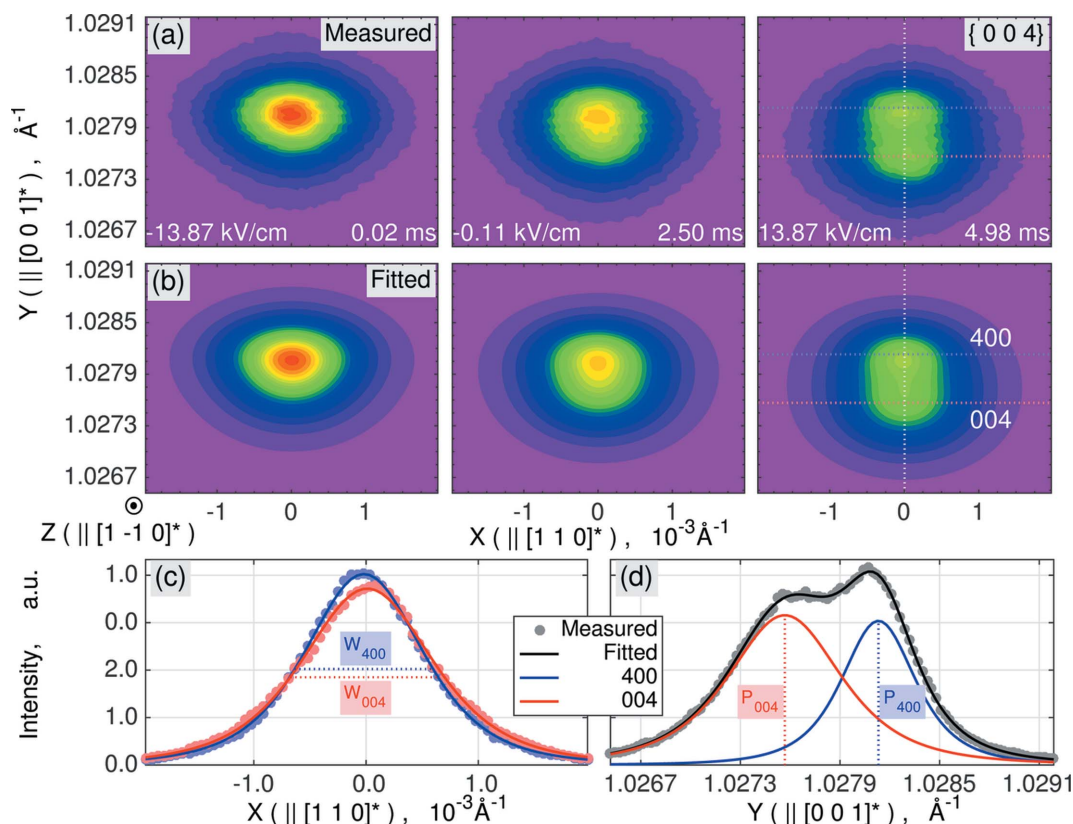


Figure 3 Measured and fitted X-ray scattering intensity distributions around {004} reflections in the reciprocal lattice coordinates $X \parallel [110]^*$ and $Y \parallel [001]^*$. (a) Measured RSMs and (b) their fit by the superposition of a pair of two-dimensional Moffat distribution functions. (c) and (d) X and Y intensity profiles of 400 (higher) and 004 (lower) Bragg-peak components at 13.87 kV cm^{-1} of electric field. An animated version of panels (a) and (b) is available in the supporting information.

reports on the detailed analysis of such splitting in many other families of reflections, which includes 41 different RSMs. The results of this work clearly suggested that the above constraint must be lifted to $a = b \neq c$ and $\alpha = \beta \neq \gamma$, corresponding to an average monoclinic lattice with a mirror plane \parallel to $(1\bar{1}0)^2$. The structure of room-temperature NBT must therefore be described by the monoclinic Cc space group, which is a subgroup of rhombohedral $R3c$.

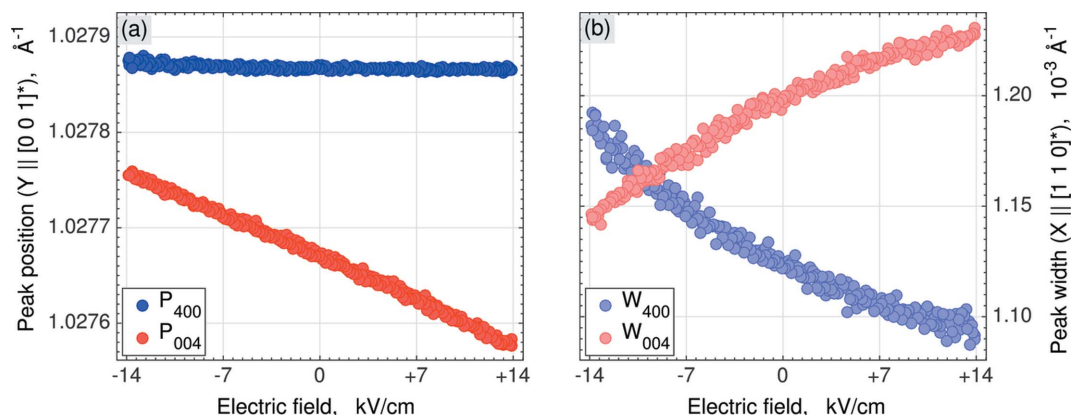
In the present work, we selected the most representative set of Bragg reflections to measure the field dependence of the monoclinic distortion. Figs. 2(c)–2(e) display stroboscopically collected RSMs of the {004} reflection (measured on the P08 beamline at PETRA III), corresponding to three different time channels or electric field states, marked by the circles in Fig. 2(b). Each RSM contained 5928 intensity values on the mesh of 78×76 points along the ω and 2θ directions, respectively. An animated set of 250 RSMs (after binning of every 40 channels to improve the counting statistics) is available in the supporting information. The varying separation of peaks on the 2θ axis suggests that the monoclinic distortion is field dependent. Fig. 3(a) shows three RSMs of an {004}

reflection in Cartesian coordinates, with the horizontal axis $X \parallel [110]^*$ and the vertical axis $Y \parallel [001]^*$, where X and Y are the coordinates of the scattering vector. Both the X and Y axes on these maps lie in the diffraction plane and the Y axis is parallel to the scattering vector. This Y axis corresponds to the $\Delta(2\theta) = 2(\Delta\omega)$ dashed line in Fig. 2(d). The open slit of the detector is perpendicular to the diffraction plane, thus giving rise to automatic intensity integration along the $Z \parallel [1\bar{1}0]^*$ direction.

We have fitted these RSMs using the superposition of two Moffat two-dimensional distribution functions, each of which has adjustable parameters: the positions of the peaks (x_0, y_0), full widths at half maxima (σ_x, σ_y), integrated intensities I and shape parameter β . This means we used 12 model parameters to describe the intensity distribution over 5928 points on each RSM. The details are given in the supporting information. Fig. 3(b) shows three RSMs, calculated using the best-fit values of the parameters. The graphs in Figs. 3(c) and 3(d) then cut reciprocal space along the X and Y directions, clearly showing the Y separation of the peak components. An animated version of this figure (in the supporting information) shows that the split peaks have significantly different time and electric-field dependencies.

Finally, Fig. 4 shows the field dependence of the key model parameters for both contributing Bragg peaks. These key

² These lattice parameters will look conventionally monoclinic ($a_m \neq b_m \neq c_m$, $\alpha_m = \gamma_m = 90^\circ$, $\beta_m \neq 90^\circ$) when the unit-cell vectors are transformed as $\mathbf{a}_m = \mathbf{a} + \mathbf{b}$, $\mathbf{b}_m = \mathbf{a} - \mathbf{b}$ and $\mathbf{c}_m = \mathbf{c}$. In this case the mirror plane becomes parallel to $(010)_m$


Figure 4

Best-fit values of the ‘position’ and ‘width’ parameters of the 004 (red) and 400 (blue) reflections as a function of time and applied electric field. (a) The Y component of the Bragg-peak positions P_{004} and P_{400} . (b) The X component of the Bragg-peak widths W_{004} and W_{400} .

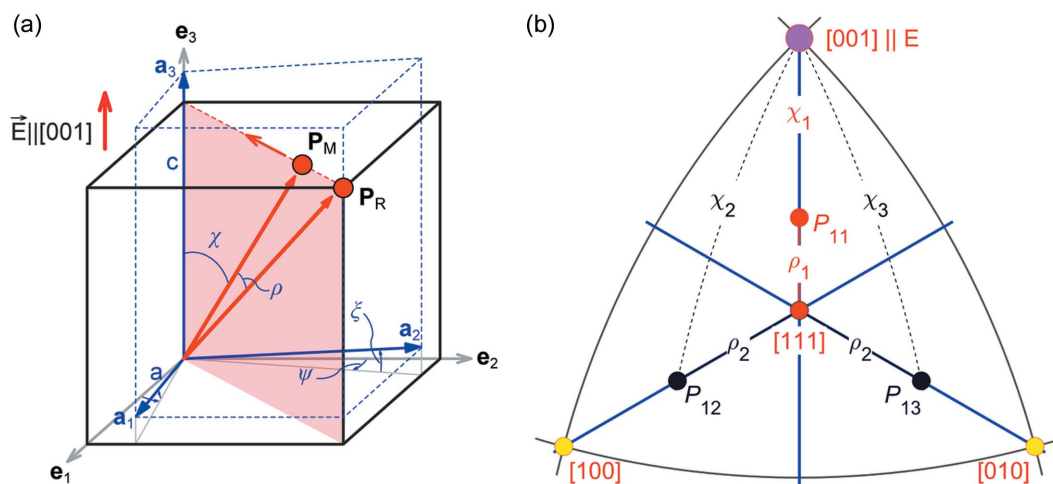
parameters are the peak positions (Figs. 4a and 4b) and peak widths (Figs. 4c and 4d) along Y and X , respectively. In the following we will introduce this model, which will help us to calculate the monoclinic distortion parameters as a function of electric field.

4. Modelling of the field dependence of RSMs

4.1. Monoclinic distortion, polarization rotation and monoclinic domains

We now discuss whether the measured changes in the peak positions and widths may be explained by a model of electric-field-dependent monoclinic distortion and polarization rotation. Fig. 5 shows how the pseudocubic unit cell is distorted after a transition from a cubic, $Pm\bar{3}m$ ($a = b = c$ and $\alpha = \beta = \gamma = 90^\circ$), to a monoclinic, Cc ($a = b \neq c$ and $\alpha = \beta \neq \gamma$), structure. The figure shows the orientation of the pseudocubic basis

vectors \mathbf{a}_1 , \mathbf{a}_2 , \mathbf{a}_3 in a monoclinic domain relative to the Cartesian reference frame \mathbf{e}_1 , \mathbf{e}_2 , \mathbf{e}_3 , aligned with the edges of the cubic unit cell. The monoclinic distortion can be modelled using four free parameters, c , a , ψ and ξ . Here, c and a are the unit-cell lengths in and out of the monoclinic mirror plane, respectively, and ψ and ξ are the shearing angles of the unit cell, as shown in Fig. 5(a). We also assume that all these free parameters can be expressed as a function of the polarization rotation angle ρ , the angle between the monoclinic \mathbf{P}_M and rhombohedral \mathbf{P}_R polarization directions. The positive and negative polarization rotation angles correspond to the monoclinic M_A ($\rho > 0$) and M_B ($\rho < 0$) phases, respectively (Vanderbilt & Cohen, 2001; Zhang *et al.*, 2014). The loss of the three-fold rotational symmetry results in the formation of monoclinic domains, in which the polarization vector rotates in a plane between the unit-cell body-diagonal directions towards one of the three edges (Fig. 5b). Therefore, a maximum of 24 monoclinic domains can be created in which


Figure 5

The distortion of the monoclinic unit cell and the polarization rotation. (a) A schematic showing the monoclinic distortion of the pseudocubic unit cell due to the rotation of the polarization direction. The angle between the polarization vector and the cell-body diagonal is ρ . The unit cell is modelled using four ρ -dependent parameters (c , a , ψ and ξ): c and a are the lengths of the unit-cell edges in and out of the polarization rotation plane, respectively, and ψ and ξ are the unit-cell shear angles. (b) A stereographic projection of polarization rotation planes and directions of polarization P_{11} , P_{12} and P_{13} in the monoclinic domains M_{11} , M_{12} and M_{13} , respectively. An animated version of panel (a) is available in the supporting information.

Table 1

Summary of the twinning matrices for all 12 monoclinic domains and the form of the reciprocal orientation matrices.

Arrows indicate the polarization rotation direction induced by an [001]-oriented electric field.

D	Polarization rotation plane and rotation angle	Twinning matrix, T	Form of the $[\mathbf{U}_B^{(mn)}]$ matrix	{004} reflection position XY
M_{11}	$[111] \rightarrow [001], \rho_1$	$\begin{bmatrix} 1 & 0 & 0 \\ 0 & 1 & 0 \\ 0 & 0 & 1 \end{bmatrix}$	$\begin{bmatrix} A & -B & -C \\ -B & A & -C \\ 0 & 0 & D \end{bmatrix}$	$\begin{bmatrix} -X_1 \\ +Y_1 \end{bmatrix}$
M_{12}	$[111] \leftarrow [100], \rho_2$	$\begin{bmatrix} 0 & 0 & 1 \\ 1 & 0 & 0 \\ 0 & 1 & 0 \end{bmatrix}$	$\begin{bmatrix} D & 0 & 0 \\ -C & A & -B \\ -C & -B & A \end{bmatrix}$	$\begin{bmatrix} -X_2 \\ +Y_2 \end{bmatrix}$
M_{13}	$[111] \leftarrow [010], \rho_2$	$\begin{bmatrix} 0 & 1 & 0 \\ 0 & 0 & 1 \\ 1 & 0 & 0 \end{bmatrix}$	$\begin{bmatrix} A & -C & -B \\ 0 & D & 0 \\ -B & -C & A \end{bmatrix}$	$\begin{bmatrix} -X_2 \\ +Y_2 \end{bmatrix}$
M_{21}	$[\bar{1}\bar{1}1] \rightarrow [001], \rho_1$	$\begin{bmatrix} 0 & \bar{1} & 0 \\ 1 & 0 & 0 \\ 0 & 0 & 1 \end{bmatrix}$	$\begin{bmatrix} A & B & C \\ B & A & -C \\ 0 & 0 & D \end{bmatrix}$	$\begin{bmatrix} 0 \\ Y_1 \end{bmatrix}$
M_{22}	$[\bar{1}\bar{1}1] \leftarrow [\bar{1}00], \rho_2$	$\begin{bmatrix} 0 & 0 & \bar{1} \\ 0 & 1 & 0 \\ 1 & 0 & 0 \end{bmatrix}$	$\begin{bmatrix} D & 0 & 0 \\ C & A & -B \\ C & -B & A \end{bmatrix}$	$\begin{bmatrix} -X_2 \\ +Y_2 \end{bmatrix}$
M_{23}	$[\bar{1}\bar{1}1] \leftarrow [010], \rho_2$	$\begin{bmatrix} \bar{1} & 0 & 0 \\ 0 & 0 & 1 \\ 0 & 1 & 0 \end{bmatrix}$	$\begin{bmatrix} A & C & B \\ 0 & D & 0 \\ B & -C & A \end{bmatrix}$	$\begin{bmatrix} +X_2 \\ +Y_2 \end{bmatrix}$
M_{31}	$[\bar{1}\bar{1}\bar{1}] \rightarrow [001], \rho_1$	$\begin{bmatrix} \bar{1} & 0 & 0 \\ 0 & \bar{1} & 0 \\ 0 & 0 & 1 \end{bmatrix}$	$\begin{bmatrix} A & -B & C \\ -B & A & C \\ 0 & 0 & D \end{bmatrix}$	$\begin{bmatrix} +X_1 \\ +Y_1 \end{bmatrix}$
M_{32}	$[\bar{1}\bar{1}\bar{1}] \leftarrow [\bar{1}00], \rho_2$	$\begin{bmatrix} 0 & 0 & \bar{1} \\ \bar{1} & 0 & 0 \\ 0 & 1 & 0 \end{bmatrix}$	$\begin{bmatrix} D & 0 & 0 \\ C & A & B \\ C & B & A \end{bmatrix}$	$\begin{bmatrix} +X_2 \\ +Y_2 \end{bmatrix}$
M_{33}	$[\bar{1}\bar{1}\bar{1}] \leftarrow [0\bar{1}0], \rho_2$	$\begin{bmatrix} 0 & \bar{1} & 0 \\ 0 & 0 & \bar{1} \\ 1 & 0 & 0 \end{bmatrix}$	$\begin{bmatrix} A & -C & B \\ 0 & D & 0 \\ B & C & A \end{bmatrix}$	$\begin{bmatrix} +X_2 \\ +Y_2 \end{bmatrix}$
M_{41}	$[1\bar{1}\bar{1}] \rightarrow [001], \rho_1$	$\begin{bmatrix} 0 & 1 & 0 \\ \bar{1} & 0 & 0 \\ 0 & 0 & 1 \end{bmatrix}$	$\begin{bmatrix} A & B & -C \\ B & A & C \\ 0 & 0 & D \end{bmatrix}$	$\begin{bmatrix} 0 \\ Y_1 \end{bmatrix}$
M_{42}	$[1\bar{1}\bar{1}] \leftarrow [100], \rho_2$	$\begin{bmatrix} 0 & 0 & 1 \\ 0 & \bar{1} & 0 \\ 1 & 0 & 0 \end{bmatrix}$	$\begin{bmatrix} D & 0 & 0 \\ C & A & B \\ -C & B & A \end{bmatrix}$	$\begin{bmatrix} -X_2 \\ +Y_2 \end{bmatrix}$
M_{43}	$[1\bar{1}\bar{1}] \leftarrow [0\bar{1}0], \rho_2$	$\begin{bmatrix} 1 & 0 & 0 \\ 0 & 0 & \bar{1} \\ 0 & 1 & 0 \end{bmatrix}$	$\begin{bmatrix} A & C & -B \\ 0 & D & 0 \\ -B & C & A \end{bmatrix}$	$\begin{bmatrix} +X_2 \\ +Y_2 \end{bmatrix}$

the polarization rotation angle, ρ , is measured from one of the eight $\langle 111 \rangle$ body-diagonal directions. Because the crystal response is strongly asymmetric with respect to the electric field direction, we must assume that *e.g.* the previous poling history of the sample kept only four rhombohedral domains with the polarization close to the ‘rhombohedral’ $[111]$, $[\bar{1}\bar{1}1]$, $[\bar{1}\bar{1}\bar{1}]$ and $[1\bar{1}\bar{1}]$ directions, resulting in the formation of only 12 monoclinic domains. We mark these domains as M_{nm} , where $m = 1, 2, 3, 4$ correspond to $\mathbf{P}_R \parallel [111]$, $\mathbf{P}_R \parallel [\bar{1}\bar{1}1]$, $\mathbf{P}_R \parallel [\bar{1}\bar{1}\bar{1}]$ and

$\mathbf{P}_R \parallel [1\bar{1}\bar{1}]$, respectively, $n = 1$ corresponds to the domains where the electric field lies in the polarization rotation plane (see Fig. 5*b*) and $n = 2, 3$ correspond to the domains where the electric field is directed out of the polarization rotation plane.

4.2. Modelling of the positions of the diffraction peaks

We use the model and definitions above to simulate the positions in the {004} family of reflections, each diffracted from

one of the 12 monoclinic domains. To do this we introduce the orientation matrix of a monoclinic domain (Fig. 5*a*),

$$[\mathbf{U}_A] = \begin{pmatrix} a \cos \xi \cos \psi & a \cos \xi \sin \psi & 0 \\ a \cos \xi \sin \psi & a \cos \xi \cos \psi & 0 \\ a \sin \xi & a \sin \xi & c \end{pmatrix}. \quad (1)$$

The columns of this matrix are the coordinates of the vectors \mathbf{a}_i in the Cartesian coordinate system \mathbf{e}_i . The reciprocal orientation matrix $[\mathbf{U}_B]$ (the columns of which are the coordinates of the reciprocal basis vectors \mathbf{a}_i^* , such that $\mathbf{a}_i \cdot \mathbf{a}_j^* = \delta_{ij}$) can be obtained as

$$[\mathbf{U}_B] = [\mathbf{U}_A^T]^{-1}, \quad (2)$$

and therefore

$$[\mathbf{U}_B] = \begin{pmatrix} A & -B & -C \\ -B & A & -C \\ 0 & 0 & D \end{pmatrix}, \quad (3)$$

where

$$\begin{aligned} A &= \frac{\cos \psi}{a \cos \xi \cos 2\psi}, & B &= \frac{\sin \psi}{a \cos \xi \cos 2\psi}, \\ C &= \frac{\tan \xi}{2^{1/2} c \sin(\frac{\pi}{4} + \psi)}, & D &= \frac{1}{c}, \end{aligned} \quad (4)$$

are also the functions of the polarization rotation angle, $A(\rho)$, $B(\rho)$, $C(\rho)$ and $D(\rho)$. Transforming the coordinates of the vectors \mathbf{a}_i^* into the laboratory coordinate system X , Y , Z of Fig. 3 (here $X \parallel \mathbf{e}_1 + \mathbf{e}_2$, $Y \parallel \mathbf{e}_3$ and $Z \parallel \mathbf{e}_1 - \mathbf{e}_2$) is done using the matrix equation $[\mathbf{U}_B]_{XYZ} = [XYZ] \cdot [\mathbf{U}_B]$ with

$$[XYZ] = \begin{bmatrix} \frac{1}{2^{1/2}} & \frac{1}{2^{1/2}} & 0 \\ 0 & 0 & 1 \\ \frac{1}{2^{1/2}} & -\frac{1}{2^{1/2}} & 0 \end{bmatrix}, \quad (5)$$

The functional form of the orientation matrices of all other monoclinic domains can be calculated using

$$[\mathbf{U}_B^{(mn)}] = [T^{(mn)}][\mathbf{U}_B][T^{(mn)}]^{-1}, \quad (6)$$

where the columns of the twinning matrices $[T^{(mn)}]$ (see Table 1) are the coordinates of the Cartesian cubic axes $\mathbf{e}_i^{(mn)}$ of the domains M_{mn} in the \mathbf{e}_i coordinate system. Finally, the positions of the Bragg peak diffracted from domain mn are described by the first two components (X and Y) of the third column of $[\mathbf{U}_B^{(mn)}]_{XYZ}$. Table 1 summarizes the twinning matrices for all 12 monoclinic domains and the form of the reciprocal orientation matrices. The arrows indicate the polarization rotation direction induced by an $[001]$ -oriented electric field.

Symmetry dictates that the polarization rotation angles are the same for all monoclinic domains from groups M_{m1} ($\rho = \rho_1$) and M_{m2}/M_{m3} ($\rho = \rho_2$). The following notation is introduced in Table 1:

$$X_1 = 4(2^{1/2})C(\rho_1), \quad Y_1 = 4D(\rho_1), \quad (7)$$

$$X_2 = \frac{4}{2^{1/2}}B(\rho_2), \quad Y_2 = 4A(\rho_2), \quad (8)$$

Fig. 6 represents the right-hand column of Table 1 in the form of a schematic drawing of the positions of the Bragg reflection, diffracted from all 12 monoclinic domains. Note that the separation of the Bragg peaks along the X axis might be significantly smaller than the peak width (arising from *e.g.* crystal mosaicity, later defined as $W_{004}^{(0)}$ and $W_{400}^{(0)}$), and therefore cannot be seen in Fig. 3 directly. Instead, this separation can be extracted from the field-dependent peak broadening, displayed in Fig. 4(*b*). Following equations (7) and (8) and the scheme in Fig. 6, the broadening can be simulated as

$$W_{004} = \frac{4 \tan \xi(\rho_1)}{c(\rho_1) \sin[\frac{\pi}{4} + \psi(\rho_1)]} + W_{004}^{(0)}, \quad (9)$$

$$W_{400} = \frac{4(2^{1/2}) \sin \psi(\rho_2)}{a(\rho_2) \cos \xi(\rho_2) \cos 2\psi(\rho_2)} + W_{400}^{(0)},$$

We further assume that the monoclinic distortion angles ψ and ξ are so small that we can replace all the trigonometric functions above by the corresponding first-order Taylor expansions. This brings us to the following expressions,

$$\Delta P_{004} = -\frac{4}{c_1} \frac{\Delta c_1}{c_1}, \quad \Delta P_{400} = -\frac{4}{a_2} \frac{\Delta a_2}{a_2}, \quad (10)$$

$$\Delta W_{004} = \frac{4(2^{1/2})}{c_1} \Delta \xi_1, \quad \Delta W_{400} = \frac{4(2^{1/2})}{a_2} \Delta \psi_2. \quad (11)$$

Here, the Δ sign stands for the difference between *e.g.* a zero-state value and a non-zero-field value. The equations allow for the direct evaluation of the monoclinic distortion parameters

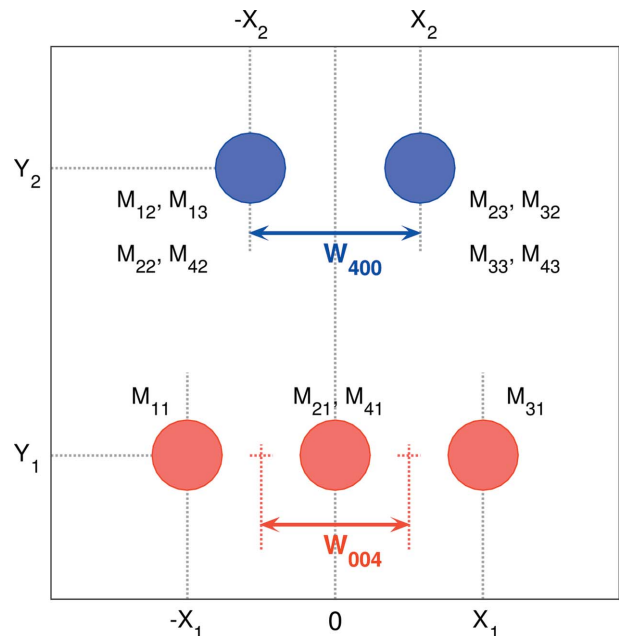


Figure 6 Schematic drawing for the set of $\{004\}$ reflections, corresponding to 12 monoclinic domains. The positions of all peaks are described by four independent parameters, X_1 , Y_1 , X_2 and Y_2 (as used in Table 1), which are directly connected to the monoclinic distortion parameters Δc_1 , $\Delta \xi_1$, Δa_2 and $\Delta \psi_2$ by equations (4) in the text. An animated version is available in the supporting information.

$\Delta c_1 = \Delta c(\Delta \rho_1)$ and $\Delta \xi_1 = \Delta \xi(\Delta \rho_1)$ (domains M_{1n}), and $\Delta a_2 = \Delta a(\Delta \rho_2)$ and $\Delta \psi_2 = \Delta \psi(\Delta \rho_2)$ (domains M_{2n} or M_{3n}).

4.3. Nonlinear electric field response and the model of polarization rotation

According to equations (10) and (11), the monoclinic distortion parameters $\Delta \xi_1$, $\Delta \psi_2$ and Δa_2 have the same field dependencies as the peak-shape parameters W_{004} , W_{400} and P_{400} , respectively. Therefore (Fig. 4), all the derived monoclinic distortions (except for Δc_1) are essentially nonlinear with respect to the magnitude of the electric field.

In the following we address the question of whether this nonlinearity can be accounted for by polarization rotation. More specifically, we will test if the monoclinic strains can be described as linear functions of the polarization rotation angle ρ , rather than of the magnitude of the electric field E , so that $\Delta \xi_1 = F_\xi \Delta \rho_1$, $\Delta \psi_2 = F_\psi \Delta \rho_2$ and $\Delta a_2 = F_a \Delta \rho_2$. To derive the $\rho_1(E)$ and $\rho_2(E)$ dependence, we shall assume that the free energy ΔG (Devonshire, 1954) has a quadratic dependence on ρ with its minimum at $\rho = 0$, so that the total free energy (including the term describing the interaction of electric field and spontaneous polarization) is

$$\Delta G = G_0 \rho^2 - EP \cos \chi, \quad (12)$$

where G_0 is the energy expansion coefficient, P is the length of the polarization vector and χ is the angle between the polarization and electric field directions, as marked in Fig. 5(b):

$$\cos \chi_1 = \cos(\rho - \rho_R), \quad (13)$$

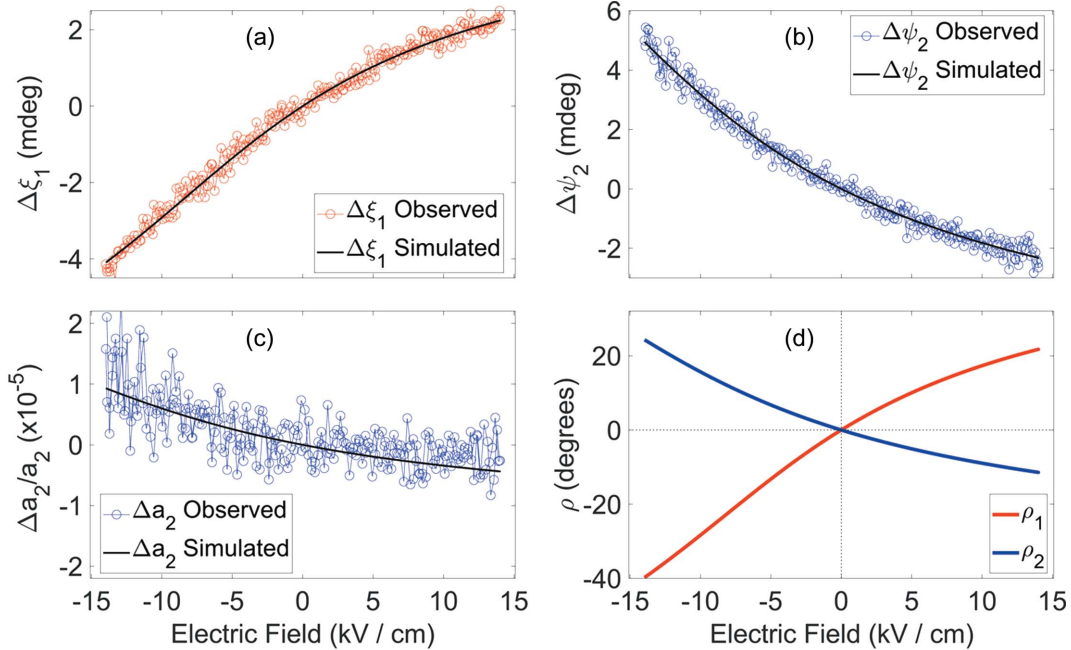


Figure 7

Electric field dependence of the strain parameters and polarization rotation angle. (a)–(c) Comparison between the observed change in the monoclinic strain parameters $\Delta \xi_1$, $\Delta \psi_2$ and Δa_2 (circles) and the calculated change according to equations (15)–(17). (d) The polarization rotation angles ρ_1 and ρ_2 , as defined in Fig. 5(b).

$$\cos \chi_2 = \cos \chi_3 = \frac{1}{2^{1/2}} \sin(\rho_R - \rho), \quad (14)$$

where $\rho_R = \arccos(1/3^{1/2}) \simeq 54.57^\circ$ is the angle between the cube edges and the body diagonal. Substituting equations (13) and (14) into (12) and locating the position of the global minimum by equating $\partial \Delta G / \partial \rho = 0$ gives the polarization rotation angles $\Delta \rho$ in the domains M_{n1} and M_{n2}/M_{n3} :

$$\Delta \rho_1 = -WE \sin(\Delta \rho_1 - \rho_R), \quad (15)$$

$$\Delta \rho_2 = -\frac{WE}{2^{1/2}} \cos(\rho_R - \Delta \rho_2), \quad (16)$$

with $W = G_0/2P$. Using our assumption that the change in the monoclinic distortion parameters is linear with respect to polarization rotation, we rewrite equations (15) and (16) as

$$\Delta \xi_1 = -WE F_\xi \sin\left(\frac{\Delta \xi_1}{F_\xi} - \rho_R\right), \quad (17)$$

$$\Delta \psi_2 = -\frac{WE F_\psi}{2^{1/2}} \cos\left(\rho_R - \frac{\Delta \psi_2}{F_\psi}\right), \quad (18)$$

$$\Delta a_2 = -\frac{WE F_a}{2^{1/2}} \cos\left(\rho_R - \frac{\Delta a_2}{F_a}\right). \quad (19)$$

Equations (17)–(19) can be solved numerically for any given electric field magnitude E , so that the unknown model parameters W , F_ξ , F_ψ and F_a can be found from the best fit to the experimental values. The solutions are shown in Figs. 7(a)–7(c), where both observed [from equations (10) and (11)] and simulated [according to equations (17)–(19)] monoclinic distortion parameters $\Delta \xi_1$, $\Delta \psi_2$ and Δa_2 are displayed. These

figures show that our simplified model above is highly effective in accounting for the nonlinear dependence of all three nonlinear monoclinic distortion parameters. This good match between observed and simulated monoclinic distortion parameters points strongly to a close connection between electric-field-induced polarization rotation and lattice strain, clearly suggesting that the corresponding piezoelectric effects are principally intrinsic rather than extrinsic in origin. Fig. 7(d), however, represents the $\Delta\rho_1 = F_\xi\Delta\xi_1$ and $\Delta\rho_2 = F_\psi\Delta\psi_2$ polarization rotation angles, showing that this nonlinearity must assume rotation of the polarization vector by an angle larger than 35° .

5. Low-field piezoelectric coefficients of a single monoclinic domain

The intrinsic low-field piezoelectric coefficients of a single monoclinic domain can be calculated from the experimental results as $d_{ijk} = \partial x_{jk}/\partial E_i$ ($E = 0$), where x_{jk} and E_i are the components of the strain tensor and electric field vector, respectively, in the domain-related Cartesian coordinate system $\mathbf{e}_i^{(mn)}$. The strain tensor for the monoclinic distortion (Fig. 5a) is:

$$x_{ij} = \begin{pmatrix} \frac{\Delta a}{a} & \Delta\psi & \frac{\Delta\xi}{2} \\ \Delta\psi & \frac{\Delta a}{a} & \frac{\Delta\xi}{2} \\ \frac{\Delta\xi}{2} & \frac{\Delta\xi}{2} & \frac{\Delta c}{c} \end{pmatrix}. \quad (20)$$

The piezoelectric coefficients d_{3jk} describe the strain in response to the electric field applied along $\mathbf{e}_3^{(mn)}$ (parallel to the polarization rotation plane). Such orientations are realized in the M_{m1} domains, so that the field dependence of the monoclinic distortion parameters Δc_1 and $\Delta\xi_1$ can be used to calculate the piezoelectric coefficients d_{333} and d_{323} . Similarly, the first and second rows in the tensor representation of the piezoelectric coefficients $d_{1jk} = d_{2jk}$ describe the strain in response to the electric field being parallel to the $\mathbf{e}_1^{(mn)}$ and $\mathbf{e}_2^{(mn)}$ axes (out of the polarization rotation plane). Such orientations of the electric field are realized in the monoclinic domains M_{m2} and M_{m3} , respectively. Therefore, the field dependence of the monoclinic distortion parameters $\Delta\psi_2$ and Δa_2 gives the piezoelectric coefficients d_{112} and d_{111} , respectively:

$$\begin{aligned} d_{333} &= \frac{1}{c_1} \frac{\partial \Delta c_1}{\partial E}, & d_{323} &= \frac{1}{2} \frac{\partial \Delta \xi_1}{\partial E}, \\ d_{112} &= \frac{\partial \Delta \psi_2}{\partial E}, & d_{111} &= \frac{1}{a_2} \frac{\partial \Delta a_2}{\partial E}. \end{aligned} \quad (21)$$

The numerical values of the corresponding piezoelectric constants are $d_{333} = 124.1 \text{ pC N}^{-1}$, $d_{323} = 20.36 \text{ pC N}^{-1}$, $d_{112} = -43.93 \text{ pC N}^{-1}$ and $d_{111} = 4.29 \text{ pC N}^{-1}$. The field dependence of the strain suggests that the low-field piezoelectric coefficients, d_{323} , d_{112} and d_{111} are associated with the polarization rotation.

6. Discussion

The macroscopic piezoelectric coefficients of NBT materials (ceramics and single crystals) range between 20 and 100 pC N^{-1} (Ge *et al.*, 2010; Foronda *et al.*, 2014; Hiruma *et al.*, 2010, 2009). These have the same order of magnitude as the values in the previous paragraph. Therefore, our results suggest that electromechanical coupling in NBT is predominantly intrinsic. The intrinsic character of the electromechanical coupling is seen in the shifts of the angular positions in the {004} family of twin reflections. We have also argued that some components of the strain can be explained straightforwardly by polarization rotation. This suggestion follows from the nonlinear dependence of the monoclinic lattice distortion parameters ($\Delta\xi_1$, $\Delta\psi_2$ and Δa_2) on the electric field. We must stress, however, that using a polarization rotation argument to explain this dependence produces an unexpectedly large amplitude for the polarization rotation angle: 25° between [111] and [001] (M_A phase) and up to 35° between [111] and [110] (M_B phase). This would mean that the polarization rotation represents such a ‘soft mode of structural changes’ that even a sub-coercive electric loading of 14 kV cm^{-1} can change it greatly. Can a polarization vector really rotate so much in a ferroelectric crystal? Our preliminary analysis favours a positive answer to this question and might lead to the suggestion of different origins for such extreme ‘structural softness’.

Firstly, the large polarization rotation may originate from the intricate structural flexibility of perovskites, where a ‘soft’ displacement of A/B cations from the centres of the corresponding oxygen AO_{12}/BO_6 cages occurs. This explanation assumes that the above displacements are long-range ordered and that the polarization rotates through coherent changes of atomic position in every unit cell of the crystal. Such coherent structural changes would appear as a change in the ‘Bragg intensities’ and could be analysed using the standard structure factor formalism (see *e.g.* Gorfman *et al.*, 2016, 2013, 2006; Tsirelson *et al.*, 2003; Schmidt *et al.*, 2009). Indeed, we observe changes in the integrated intensity for both 400 and 004 Bragg peaks by $\sim 10\%$ (see Fig. S1 in the supporting information) but we have not been able to measure enough Bragg reflection intensities to model the structural changes within the Cc space group.

Secondly, the large polarization rotation may originate from variations in the local structure and short-range order parameters. In this model, by contrast with the first, the displacement of the atoms varies from one unit cell to another, so that the polarization can only be defined on average. The strong structural disorder in NBT has been documented by the observation of diffuse X-ray scattering (Kreisel *et al.*, 2003; Thomas *et al.*, 2010; Gorfman *et al.*, 2015) or total neutron scattering (Keeble *et al.*, 2013). A reverse Monte Carlo simulation of the atomic pair-distribution function in NBT (Keeble *et al.*, 2013) demonstrated that the Bi atoms in the monoclinic {110} planes do indeed differ and involve two co-existing directions of bismuth displacements. From this starting point, it follows that one might suggest that the

application of an external electric field in a particular direction switches a sub-population of atoms, thus changing the distribution of Bi atoms over two metastable states. Such a redistribution would produce a change in the average direction of the polarization vector relatively easily.

Thirdly, the large polarization rotation in NBT is strongly reminiscent of that in the compositionally driven polarization rotation in $\text{PbZr}_{1-x}\text{Ti}_x\text{O}_3$ at the morphotropic phase boundary. The work of Zhang *et al.* (2014) shows that even a minor change in the composition x near the morphotropic phase boundary leads to the rotation of the average direction of the Pb displacement vector by a large angle of well beyond 35° . This large polarization rotation is commonly considered as one of the origins of enhanced piezo-activity at this particular boundary in the phase diagram.

We finally note the ongoing discussion of the true structural origin of the monoclinic phase in NBT. One suggestion is that the monoclinic symmetry can be mimicked by an adaptive phase mechanism (Jin *et al.*, 2003; Wang, 2007), which is a microstructural material state made of periodically arranged nano-domains. Here the effective polarization rotation can be driven by the dynamics in the hierarchical nano-domain pattern, where each domain would have rhombohedral symmetry. The polarization direction of such an assembly is given by the volume average of polarization in the individual sets of nano-domains. Provided that the domains are sufficiently small, the Bragg diffraction pattern of such an adaptive structure would be indistinguishable from the equivalent Bragg diffraction pattern from a truly long-range monoclinic phase.

Finally, the demonstrated time-resolved reciprocal-space mapping approach has the potential to uncover the origins of electromechanical coupling in other ferroelectric perovskites.

Acknowledgements

We acknowledge Dr Genziana Bussone-Grifone and Dr Ute Rutt for their support of the P08 beamline staff at the PETRA III storage ring. We acknowledge Prof Haosu Luo (Shanghai Institute of Ceramics) for providing the single crystal of NBT.

Funding information

The following funding is acknowledged: Bundesministerium für Forschung und Technologie (award No. 05K13PSA to H. Choe and S. Gorfman). We are grateful for financial support provided by the EPSRC (Materials World Network: Nano-scale Structure–Property Relationships in Lead-Free Morphotropic Phase Boundary Piezoelectrics).

References

Aksel, E., Forrester, J. S., Kowalski, B., Jones, J. L. & Thomas, P. A. (2011). *Appl. Phys. Lett.* **99**, 222901.
 Aksel, E., Forrester, J. S., Nino, J. C., Page, K., Shoemaker, D. P. & Jones, J. L. (2013). *Phys. Rev. B*, **87**, 104113.

Choe, H., Gorfman, S., Heidbrink, S., Pietsch, U., Vogt, M., Winter, J. & Ziolkowski, M. (2017). *IEEE Trans. Nucl. Sci.* **64**, 1320–1326.
 Choe, H., Gorfman, S., Hinterstein, M., Ziolkowski, M., Knapp, M., Heidbrink, S., Vogt, M., Bednarcik, J., Berghäuser, A., Ehrenberg, H. & Pietsch, U. (2015). *J. Appl. Cryst.* **48**, 970–974.
 Devonshire, A. F. (1954). *Adv. Phys.* **3**, 85–130.
 Foronda, H., Deluca, M., Aksel, E., Forrester, J. S. & Jones, J. L. (2014). *Mater. Lett.* **115**, 132–135.
 Fu, H. & Cohen, R. E. (2000). *Nature*, **403**, 281–283.
 Ge, W., Li, J., Viehland, D. & Luo, H. (2010). *J. Am. Ceram. Soc.* **93**, 1372–1377.
 Ge, W., Liu, H., Zhao, X., Zhong, W., Pan, X., He, T., Lin, D., Xu, H., Jiang, X. & Luo, H. (2008). *J. Alloys Compd.* **462**, 256–261.
 Glazer, A. M. (1972). *Acta Cryst.* **B28**, 3384–3392.
 Gorfman, S. (2014). *Crystallogr. Rev.* **20**, 210–232.
 Gorfman, S., Glazer, A. M., Noguchi, Y., Miyayama, M., Luo, H. & Thomas, P. A. (2012). *J. Appl. Cryst.* **45**, 444–452.
 Gorfman, S., Keeble, D. S., Bombardi, A. & Thomas, P. A. (2015). *J. Appl. Cryst.* **48**, 1543–1550.
 Gorfman, S., Keeble, D. S., Glazer, A. M., Long, X., Xie, Y., Ye, Z.-G., Collins, S. & Thomas, P. A. (2011). *Phys. Rev. B*, **84**, 020102.
 Gorfman, S., Schmidt, O., Tsirelson, V. G., Ziolkowski, M. & Pietsch, U. (2013). *Z. Anorg. Allg. Chem.* **639**, 1953–1962.
 Gorfman, S., Schmidt, O., Ziolkowski, M., von Kozierowski, M. & Pietsch, U. (2010). *J. Appl. Phys.* **108**, 064911.
 Gorfman, S., Simons, H., Iamsasri, T., Prasertpalichat, S., Cann, D. P., Choe, H., Pietsch, U., Watier, Y. & Jones, J. L. (2016). *Sci. Rep.* **6**, 20829.
 Gorfman, S. & Thomas, P. A. (2010). *J. Appl. Cryst.* **43**, 1409–1414.
 Gorfman, S., Tsirelson, V., Pucher, A., Morgenroth, W. & Pietsch, U. (2006). *Acta Cryst.* **A62**, 1–10.
 Hiruma, Y., Nagata, H., Hidaka, Y., Tsukada, S., Kojima, S. & Takenaka, T. (2010). *Ferroelectrics*, **404**, 162–166.
 Hiruma, Y., Nagata, H. & Takenaka, T. (2009). *J. Appl. Phys.* **105**, 084112.
 Jaffe, B., Cook, W. J. R. & Jaffe, H. (1971). *Piezoelectric Ceramics*. London and New York: Academic Press.
 Jaffe, B., Roth, R. S. & Marzullo, S. (1954). *J. Appl. Phys.* **25**, 809–810.
 Jin, Y. M., Wang, Y. U., Khachatryan, A. G., Li, J. F. & Viehland, D. (2003). *J. Appl. Phys.* **94**, 3629–3640.
 Jones, G. O. & Thomas, P. A. (2002). *Acta Cryst.* **B58**, 168–178.
 Keeble, D. S., Barney, E. R., Keen, D. A., Tucker, M. G., Kreisel, J. & Thomas, P. A. (2013). *Adv. Funct. Mater.* **23**, 185–190.
 Kitanaka, Y., Yanai, K., Noguchi, Y., Miyayama, M., Kagawa, Y., Moriyoshi, C. & Kuroiwa, Y. (2014). *Phys. Rev. B*, **89**, 104104.
 Kreisel, J., Bouvier, P., Dkhil, B., Thomas, P. A., Glazer, A. M., Welberry, T. R., Chaabane, B. & Mezouar, M. (2003). *Phys. Rev. B*, **68**, 014113.
 Levin, I. & Reaney, I. M. (2012). *Adv. Funct. Mater.* **22**, 3445–3452.
 Ma, C., Guo, H. & Tan, X. (2013). *Adv. Funct. Mater.* **23**, 5261–5266.
 McQuade, R. R. & Dolgos, M. R. (2016). *J. Solid State Chem.* **242**, 140–147.
 Mitchell, R. H. (2003). *Perovskites: Modern and Ancient*. Thunder Bay, Canada: Almaz Press.
 Noheda, B., Cox, D. E., Shirane, G., Gonzalo, J. A., Cross, L. E. & Park, S.-E. (1999). *Appl. Phys. Lett.* **74**, 2059–2061.
 Ogino, M., Noguchi, Y., Kitanaka, Y., Miyayama, M., Moriyoshi, C. & Kuroiwa, Y. (2014). *Crystals*, **4**, 273–295.
 Panda, P. K. (2009). *J. Mater. Sci.* **44**, 5049–5062.
 Rao, B. N., Olivi, L., Sathe, V. & Ranjan, R. (2016). *Phys. Rev. B*, **93**, 024106.
 Rödel, J., Jo, W., Seifert, K. T. P., Anton, E. M., Granzow, T. & Damjanovic, D. (2009). *J. Am. Ceram. Soc.* **92**, 1153–1177.
 Roleder, K., Franke, I., Glazer, A. M., Thomas, P. A., Miga, S. & Suchanicz, J. (2002). *J. Phys. Condens. Matter*, **14**, 5399–5406.
 Schmidt, O., Gorfman, S., Bohatý, L., Neumann, E., Engelen, B. & Pietsch, U. (2009). *Acta Cryst.* **A65**, 267–275.
 Shrout, T. R. & Zhang, S. J. (2007). *J. Electroceram.* **19**, 111–124.

- Takenaka, T., Maruyama, K. & Sakata, K. (1991). *Jpn. J. Appl. Phys.* **30**, 2236–2239.
- Takenaka, T., Nagata, H. & Hiruma, Y. (2008). *Jpn. J. Appl. Phys.* **47**, 3787–3801.
- Thomas, P. A., Trujillo, S., Boudard, M., Gorfman, S. & Kreisel, J. (2010). *Solid State Sci.* **12**, 311–317.
- Tsirelson, V. G., Gorfman, S. V. & Pietsch, U. (2003). *Acta Cryst.* **A59**, 221–227.
- Vakhrushev, S. B., Isupov, V. A., Kvyatkovsky, B. E., Okuneva, N. M., Pronin, I. P., Smolensky, G. A. & Syrnikov, P. P. (1985). *Ferroelectrics*, **63**, 153–160.
- Vanderbilt, D. & Cohen, M. H. (2001). *Phys. Rev. B*, **63**, 094108.
- Wang, Y. U. (2007). *Phys. Rev. B*, **76**, 024108.
- Yokota, H., Zhang, N., Taylor, A. E., Thomas, P. A. & Glazer, A. M. (2009). *Phys. Rev. B*, **80**, 104109.
- Zhang, N., Yokota, H., Glazer, A. M., Keen, D. A., Gorfman, S., Thomas, P. A., Ren, W. & Ye, Z.-G. (2018). *IUCrJ*, **5**, 73–81.
- Zhang, N., Yokota, H., Glazer, A. M., Ren, Z., Keen, D. A., Keeble, D. S., Thomas, P. A. & Ye, Z.-G. (2014). *Nat. Commun.* **5**, 5231.
- Zhang, N., Yokota, H., Glazer, A. M. & Thomas, P. A. (2011). *Acta Cryst.* **B67**, 386–398.