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Cobalt Phosphide Nanorods as an Efficient Electrocatalyst for

Hydrogen Evolution Reaction

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Abstract: Cobalt phosphide (Co₂P) nanorods are found to exhibit efficient catalytic activity in hydrogen evolution reaction (HER), with the overpotential required for the current density of 20 mA/cm² as small as 167 mV in acidic solution and 171 mV in basic solution. In addition, the Co₂P nanorods can work stably in both acidic and basic solution during hydrogen production. This performance can be favorably comparable to typical high efficiently non-precious catalysts, and suggest the promising application potential of the Co₂P nanorods in the field of hydrogen production. The HER process follows a Volmer-Heyrovsky mechanism, and the rates of the discharge step and desorption step appear to be comparable during the HER process. The similarity of charged natures of Co and P in the Co₂P nanorods to those of the hydride-acceptor and proton-acceptor in high efficient Ni₂P catalyst, [NiFe] hydrogenase, and its analogues implies that the HER catalytic activity of Co₂P nanorods might be correlated with the charged natures of Co and P.

Keywords

cobalt phosphide; nanorod; electrocatalyst; hydrogen generation

1. Introduction

The solar-driven splitting of water into molecular hydrogen and oxygen is one of the most promising possibilities for simultaneously solving the global energy crisis and current environmental issues.[1-3] Because of the intrinsically slow hydrogen evolution reaction (HER) kinetics of semiconductors, photocathodes must be decorated with HER catalysts for efficient hydrogen production. Though platinum remains the most effective HER catalyst, having been shown to significantly enhance the hydrogen production capability of photocathodes several decades ago,[4, 5] it is a limited resource and expensive, and so its widespread practical application in the field of solar-driven hydrogen production may be limited. There is therefore a demonstrable need for non-precious HER catalysts.

Recently, a variety of new HER catalysts have been reported, including molybdenum sulfide,[6, 7] molybdenum carbide,[8-10] molybdenum nitride,[10] molybdenum boride,[8] tungsten carbide,[11, 12] tungsten carbonitride,[13] first-row transition-metal dichalcogenides,[14, 15] nickel selenide,[16] nickel phosphide,[17] cobalt phosphide (CoP), [18-20] molybdenum phosphide,[21, 22] and etc. The first-row transition-metal dichalcogenides have similar coordination structure as the active centers in efficient hydrogenase,[15] and the charged natures of metal and P in metal phosphides are similar to those of the hydride-acceptor and proton-acceptor in [NiFe] hydrogenase and its analogues $([Ni(PS3*)(CO)]^{1-}$ and $[Ni(PNP)_2]^{2+}$.[23]

Here the HER performance of cobalt phosphide (Co₂P) nanorods is introduced. Although their structure and composition are different from all heretofore reported HER catalysts, the Co₂P nanorods

exhibit efficient and stable HER catalytic activity in both acidic and basic solutions. The overpotential required for a current density of 20 mA/cm² (η_{20}) is as small as 167 mV in acidic solution and 171 mV in basic solution. The η_{20} of the Co₂P nanorods lies in the top 10 of the reported values of non-precious HER catalysts. It is worth noting that the four reported η_{20} better than that of the Co₂P nanorods were obtained from the composites of catalysts and nanostructured conductive supports, including Mo₁So_v particles loaded on reduced graphene oxide (Mo₁So_y /rGO),[10] CoP nanoparticles loading on carbon nanotube (CoP/CNT),[18] CoP nanowires loaded on carbon cloth (CoP/CC),[20] and MoS₂ loaded on mesoporous graphene foam (MoS₂/MGF).[24] The nanostructured conductive supports have been well known to improve the electron transport among catalysts and therefore the performance of catalyst.[25] The charged natures of Co and P in the Co₂P nanorods are similar to those of the hydride-acceptor and proton-acceptor in Ni₂P catalyst, [NiFe] hydrogenase, and its analogues. Such similarity might afford the efficient catalytic activity of the Co₂P nanorods.

2. Materials and methods

2.1 Synthesis of Co₂P nanorods. Cobalt acetate tetrahydrate (0.50 g, 2 mmol) was mixed with oleylamine (12 g, 45 mmol) in a 100 mL round-bottom flask. The flask was heated via a heating mantle. Dispersion was obtained by stirring the mixture at 70 °C. The dispersion was then heated to 120 °C and triphenylphosphine (5.246 g, 20 mmol) was added to the mixture. The flask was pumped under vacuum at 120 °C for 30 min to remove water, and then refilled with N₂. The temperature of the

heating mantle was increased to 370 °C and maintained at this value for 10 min. The flask was removed from the heating mantle and cooled to room temperature. The black product was isolated and washed by repeated centrifugation/ultrasonication, with hexane as good solvent and ethanol as non-solvent. Finally, the product was dried under vacuum at room temperature.

2.2 Characterization. The morphology of Co₂P nanorods was assessed by transmission electron microscopy (TEM, 200 kV, JEM2100, JEOL) and scanning electron microscopy (SEM, 7001F, JEOL). The energy-dispersive X-ray spectroscopy (EDX) spectrum was recorded using a GENESIS 2000 XM 30T (EDXA) on a JEM2100. For the TEM investigation, the Co₂P nanorods were dispersed in hexane by ultrasonication. The dispersion was dropped onto a carbon-coated copper grid (300-mesh). The copper grid was then dried at 100 °C for 5 min before the TEM characterization. Powder X-ray diffraction (XRD) patterns were collected using a D8 Advance diffractometer with graphitemonochromated Cu K α radiation ($\lambda = 1.54178$ Å). The X-ray photoelectron spectroscopy (XPS) experiments were carried out on an ESCALAB250Xi System (ThermoFisher) equipped with a monochromatic Al Ka (1486.6 eV) source and a concentric hemispherical energy analyzer. The binding energy C 1s peak from surface adventitious carbon (284.8 eV) was adopted as a reference for the binding energy measurements.

2.3 Electrochemical performance. Co₂P nanorods (15 mg) were dispersed in hexane (0.5 mL) with the aid of an ultrasonic horn (2 mm diameter, 130 W, 60 min). The dispersion (17 μ L) was dropped onto a clean Ti foil (0.5 cm²) and dried naturally. The Ti foil was polished by sandpaper (7000 mesh), 4

and then cleaned by acetone, ethanol, de-ionized water (15 min each) prior to the drop-coating of the Co_2P nanorods. The Co_2P nanorods loaded on the Ti foil were annealed in 5% H₂/N₂ at 450 °C for 30 min to remove surface ligands.

All electrochemical measurements were carried out with an electrochemical workstation (CHI 614D, CH Instrument) in a three-electrode configuration, with Co_2P (loaded onto Ti foil) as a working electrode, a graphite rod (6 mm diameter) as a counter electrode, and a mercury/mercurous sulfate electrode (MSE) or mercury/mercury oxide electrode (MMO) as a reference electrode. The samples were assembled into a homemade electrochemical cell, with only a defined area (~0.07 cm²) of the front surface of the sample exposed to solution during the measurements. The counter electrode was separated from working chamber by porous glass frit.

 H_2SO_4 aqueous solution (0.5 M) or KOH aqueous solution (1 M) was used for electrochemical measurements. The MSE is used as the reference electrode in H_2SO_4 solution, and the MMO is used in KOH solution. The solutions were purged with high purity H_2 (99.999%) for 30 min prior to electrochemical measurements. The reversible hydrogen evolution potential (RHE) was determined by the open circuit potential of a clean Pt electrode in the solution of interest bubbled with H_2 (99.999%), being -0.694 V vs. MSE for the 0.5 M H_2SO_4 solution and -0.876 V vs. MMO for the 1 M KOH solution. A potential measured with respect to the MSE electrode was therefore referenced to the RHE by adding a value of 0.694 V for the measurements in the 0.5 M H_2SO_4 solution, and a potential measured with respect to the MMO electrode in the KOH solution was referenced to the RHE by

adding a value of 0.876 V.

Polarization curves of Co₂P samples were measured at a sweep rate of 5 mV/s in rigorously stirred solution (1600 rpm). The uncompensated cell resistance (*R*) was determined by currentinterrupt method, and the experimental potential was corrected by subtracting ohmic drop (*iR*), where *i* is the current corresponding to the experimental potential. The apparent Tafel slope was derived from the *iR*-corrected polarization curve by fitting experimental data to the equation $\eta = a + blogj$, where η is the *iR*-corrected potential, *a* is the Tafel constant, *b* is the Tafel slope, and *j* is the current density. Electrochemical impedance spectroscopy (EIS) measurements were carried out at different potentials in the frequency range of 10⁻² to 10⁶ Hz with 10 mV sinusoidal perturbations and 12 steps per decade in 0.5 M H₂SO₄ solution.

For accelerated degradation investigations, cyclic voltammetric (CV) measurements were carried out with a 50mV/s sweep rate between -0.240 and 0.100 V vs. RHE in the 0.5 M H_2SO_4 solution, and -0.330 and 0.080 V vs. RHE in the 1 M KOH solution, without accounting for uncompensated resistance.

The volume of H_2 during the potentiostatic electrolysis experiment was monitored by volume displacement in a configuration shown in Figure S1 (Electronic Supplementary Information). In this experiment, the backside of the Ti foil was connected to a Cu wire with Ag paste. The Cu wire was threaded to a glass tube (6 mm diameter), and the backside and front side of sample electrode were then sealed with epoxy resin with the exception of an exposed area (~0.5 cm²). A Freescale

MPXV7002DP differential pressure transducer was employed to monitor pressure variation in the gas gathering tube, and then the volume of generated H₂ was computed from pressure variation in the gas gathering tube (see details in Electronic Supplementary Information). The current and charge passing the Co₂P nanorods were measured with the electrochemical workstation and the voltage change of the differential pressure transducer was monitored with a digital multimeter (4 1/2 digits). Prior to experiment, the relationship between volume of gathered gas and the variation of output voltage of the differential pressure transducer (i.e., pressure variation in the gas gathering tube) was calibrated by injecting known amounts of air into the gas gathering tube and recording the variation of output voltage of the differential pressure transducer.

3. Results and discussion

Co₂P nanorods were synthesized using low cost precursors at 370 °C. Cobalt acetate and triphenylphosphine were adopted as cobalt source and phosphorus source, respectively, and oleylamine was used as heating media and surface capping agent. The overall structural feature of the resultant Co₂P nanorods was revealed by XRD experiments (Figure 1). The observed well-defined peaks are consistent with the good crystallinity of the products, the peaks matching well those from orthorhombic phase Co₂P (JCPDS No. 32-306, a = 5.6465 Å, b = 6.6099 Å, c = 3.5130 Å).



Figure 1. Experimental XRD pattern of the Co₂P nanorods.

The morphology and microstructure of the Co₂P nanorods were assessed via TEM. A typical TEM image can be found in Figure 2a. The product has a rod-like morphology, with 9.8 ± 1.3 nm diameter and 110.0 ± 11.8 nm length (Figure S2, Electronic Supplementary Information). The rod-like morphology of the Co₂P was further confirmed by a SEM image (Figure S3, Electronic Supplementary Information). A high resolution TEM (HRTEM) image of the Co₂P nanorods can be found in Figure 2b, in which lattice fringes can be clearly seen. These fringes suggest the good crystallinity of Co₂P, in accordance with that revealed by the XRD experiment. The fast Fourier transform (FFT) pattern of the lattice fringes shows the corresponding diffraction pattern of the Co₂P nanorods (inset of Figure 2b). The pattern can be indexed to the [-210] zone-axis patterns of orthorhombic phase Co₂P ($1/d_{OA} = d_{(121)} = 2.2 \text{ Å}$, $1/d_{OB} = d_{(002)} = 1.7 \text{ Å}$, $1/d_{OC} = d_{(-1-21)} = 2.2 \text{ Å}$, $\angle AOB = \angle BOC = 51^\circ$). The FFT pattern reveals that the Co₂P nanorods grew along the [002] direction. HRTEM images from other nanorods reveal the 8

same growth direction. EDX spectra recorded from nanorods confirm the presence of Co and P in the nanorods (Figure S4, Electronic Supplementary Information).



Figure 2. (a) TEM and (b) HRTEM images of the Co₂P nanorods.

The chemical states of Co and P on the surface of the Co₂P nanorods were revealed by XPS experiments. Figure 3 shows the XPS spectra of the Co 2p window and the P 2p window recorded from the Co₂P nanorods subjected to annealing (450 °C, 5% H₂/N₂, 30 min). Three distinct doublets can be clearly seen in the XPS spectrum of the Co 2p window (Figure 3a), with Co 2p_{3/2} binding energies at 778.2, 781.7 and 785.6 eV. The peak at 778.2 eV suggests that there are reduced Co species in Co₂P.^[36] These reduced Co species are partially charged (Co⁸⁺, 0 < δ < 2), and δ must have a small value, because the corresponding Co 2p_{3/2} binding energy (778.2 eV) is very close to that of metallic Co (777.9 eV).[26] The P 2p window of the XPS spectrum shows two doublets with peak binding

energies at 129.4 and 133.5 eV. The doublet at 129.4 eV suggests the presence of partially negatively charged P (P^{δ} , $-1 < \delta < 0$), and the value of δ is likely to be small, because the corresponding energy (129.4 eV) is close to that of elemental P (130.0 eV).[27] The Co 2p peak observed at 781.7 eV and its satellite at 785.6 eV can be assigned to Co oxidation state, which could be associated with cobalt phosphate formed on the surface of the Co₂P nanorods.[28] The presence of cobalt phosphate on the surface of Co₂P nanorods is also suggested by P 2p peak centered at 133.5 eV and O 1s peak centered at 531.6 eV (Figure S5, Electronic Supplementary Information), which resemble well those of P and O species in cobalt phosphate.[28]



Figure 3. XPS spectra of (a) the Co $2p_{3/2}$ and (b) the P 2p windows.

The HER performance of the Co₂P nanorods was evaluated by polarization curve measurements (*j*-V plots). Prior to measurements, The Co₂P nanorods were applied to Ti foils (loading amount ca. 1 mg/cm²) by drop coating, and annealed at 450 °C in a 5% H₂/N₂ atmosphere for 30 min to remove surface ligand.[17] The measurements were carried out in a 0.5 M H₂SO₄ solution with a three-

electrode configuration (see details in Materials and methods). Figure 4a shows the j-V plots of the Co₂P nanorods, bare Ti foil, commercial Pt/C (Johnson Matthey, Hispec 3000, 20 wt.%) loaded on a glassy carbon electrode (GCE), and a bare GCE. It was verified that cobalt phosphate can be dissolved in the 0.5 M H₂SO₄ solution (Figure S6, Electronic Supplementary Information), therefore the cobalt phosphate presented on the surface of the Co₂P nanorods would be dissolved and should not affect the catalytic activity of Co₂P nanorods during the electrochemistry measurements. It can be seen that the bare Ti foil exhibits negligible current flow in the potential range of -250 to 0 mV vs. RHE, suggesting that the current flow of the Co₂P nanorods supported on the Ti foil sample in this potential range is induced by the Co₂P nanorods, but not the Ti foil. The onset of current is found at ca. -70 mV vs. RHE for the Co₂P nanorods. The η_{20} is 167 mV for the Co₂P nanorods, and the overpotential required for a current density of 10 mA/cm² (η_{10}) is 134 mV. The η_{20} and η_{10} values of different catalysts are usually compared in order to evaluate their efficiencies, because in photoelectrochemical applications a photocathode produces a current flux of 10-20 mA/cm² under 1 sun of AM 1.5 G illumination.[3] The performance of representative HER catalysts is summarized in Table 1. It can be seen that the η_{20} and η_{10} values of the Co₂P nanorods in our experiments are only larger than those of CoP nanoparticles, [19] Ni₂P nanoparticles, [17] Ni-Mo nanopowders, [29] MoP nanoparticles, [21, 22] CoP/CC, [20] CoP/CNT,[18] Mo₁Soy/ rGO [10] and MoS₂/ MGF),[24] and smaller than other listed catalysts. Ni-Mo nanopowders degrade rapidly in acidic condition, rendering their exploitation problematic.[29] The conductive CC, CNT, rGO, MGF in CoP/CC, CoP/CNT, Mo₁Soy/rGO or MoS₂/MGF enhances the 11

HER performance of these composites, in contrast to the Co₂P nanorods in our experiments which are loaded on the Ti foil. This comparison demonstrates that the HER performance of the Co₂P nanorods in our experiments is favorably comparable to most values of recently reported high efficiency nonprecious electrocatalysts.



Figure 4. (a) Polarization curves of Pt/C on GCE (loading amount: ~0.285 g/cm²), Co₂P on Ti foil, bare GCE, and bare Ti foil in 0.5 M H₂SO₄ solution (*iR* corrected). The bare Ti foil was subjected to the same annealing treatment as Co₂P on Ti foil sample prior to the measurement. (b) Polarization curves of Co₂P on Ti foil in 0.5 M H₂SO₄ at different CV sweeps (without *iR* correction).

Catalyst	Substrate	Mass density	η_{onset}^{a}			Tafel slope	J_0^b	Electrolyte
		(mg/cm)	(mv)	(mV)	(mV)	(mv/dec)	(mA/cm)	
Ni ₂ P[17]	Ti foil	1		117	130	$46_{\eta=25-125}$ $81_{\eta=150-200}$	3.3×10^{-3}	0.50M H ₂ SO ₄
CoP[30]	Ti foil	2			85	50	0.14	0.50 M H ₂ SO ₄
CoP[20]	carbon cloth	0.92	38	67	100	51	0.288	0.50 M H ₂ SO ₄
CoP/CNT[18]	GCE	0.285	40	122	160	54	0.13	0.50 M H ₂ SO ₄
MoP[22]	GCE	0.36	40	125	151	54	0.086	0.50 M H ₂ SO ₄
MoP[21]	GCE	0.86	50	135	167	54	0.034	0.50 M H ₂ SO ₄
Ni-Mo nanopowder[29]	Ti foil	1			70			2 M NaOH
Ni-Mo nanopowder[29]	Ti foil	3			80			$0.5 \text{ M H}_2 \text{SO}_4$
Ni-Mo nanopowder[29]	Ti foil	1		79	107			1 M NaOH
Bulk Mo ₂ C[8]	carbon paste electrode	1.4	≧100	208	224	$56_{\eta=100-220}$	1.3×10^{-3}	0.50 M H ₂ SO ₄
Bulk MoB[8]	carbon paste electrode	2.5	≧100	212	227	55 _{η=140-210}	1.4×10^{-3}	0.50 M H ₂ SO ₄
Mo ₂ C/CNT[9]	carbon paper	2	63	149		55.2	1.4×10^{-2}	0.1 M HClO ₄
Fe-WCN[13]	RRDE	0.4	100	220		47.1		$H_2SO_4 (pH 1) + Na_2SO_4 (0.5 M)$
Mo ₁ Soy[10]	carbon paper	1.4		177		66.4	1.3×10^{-2}	0.1 M HClO ₄
Mo ₁ Soy-RGO[10]	carbon paper	0.47		109		62.7	3.7×10^{-2}	0.1 M HClO ₄
Mo ₂ C/C[10]	carbon paper	2		311		87.6	8.1×10^{-3}	0.1 M HClO ₄
Co _{0.6} Mo _{1.4} N ₂ [31]	GCE	0.243		202	267		2.3×10^{-4}	0.1M HClO ₄
MoS ₃ (33%)/MWCNT- NC[32]	silver electrode	0.255	130	206	226	$40_{\eta=135\text{-}174}$	1.35×10^{-4}	$1 \text{ M H}_2\text{SO}_4$
Core-shell MoO ₃ - MoS ₂ nanowires[33]	FTO		150-200	254	272	$50-60_{\eta=200}$		$0.5 \ M \ H_2 SO_4$
Defect-rich MoS ₂ nanosheets[34]	GCE	0.285	120	190	214	$50_{\eta=120\text{-}180}$	$8.91\times 10^{\text{-3}}$	$0.5 \ M \ H_2 SO_4$
MoS ₂ @Au[35]	Au electrode	0.00103	90	226		69	9.3×10^{-3}	$0.5 \text{ M H}_2\text{SO}_4$
amorphous MoS ₃ - CV[7]	GCE			211	229	$40_{\eta=170-200}$	1.3×10^{-4}	$1 \text{ M H}_2 \text{SO}_4$
MoS ₂ /RGO hybrid[25]	GCE	0.285	100	154	176	41		$0.5M H_2SO_4$
MoS ₂ /MGF[24]	GCE	0.21	100	146	159	$42_{\eta=90-120}$		$0.5 \text{ M} \text{H}_2 \text{SO}_4$
MoS ₂ /CNTs[36]	glass carbon disk	0.136	90	184	230	44.6		$0.5 \ M \ H_2 SO_4$
$Cu_2MoS_4[37]$	GCE	0.0425	135	321		95		pH 0 H ₂ SO ₄
WS ₂ /RGO[38]	GCE	0.4	150-200	265	292	58		$0.5M H_2SO_4$
WS ₂ nanosheets[39]	GCE	0.0001-0.0002 or ca. one continuous laver	80-100	233	275	55		0.5 M H ₂ SO ₄
WS ₂ nanosheets[40]	GCE	0.285	60	151	177	72	2.5×10^{-3}	1 M H ₂ SO ₄
Cobalt-sulfide	FTO		43	165	227	93		1.0 M pH 7 PBS
NiWS _x [42]	FTO		165	373	430	96-120 150	10 ^{-2.66}	pH 7 PBS
$\frac{1}{CoWS_{*}[42]}$	FTO		95	271	311	78_{m} 120 150	10-2.25	pH 7 PBS
$\frac{CoMoS_{x}[42]}{CoMoS_{x}[42]}$	FTO		75	241	2.82	$85_{m-120,150}$	10-2.89	pH 7 PBS
FeS ₂ [15]	GCE		10	192.6	202	62.5	7×10^{-4}	0.5 M H ₂ SO ₄
FeSe ₂ [15]	GCE					65.3	3.5×10^{-4}	0.5 M H ₂ SO ₄
$Fe_{0.43}Co_{0.57}S_{2}[15]$	GCE			264		55.9	1.3×10^{-3}	0.5 M H ₂ SO ₄
CoS ₂ [15]	GCE			232		44.6	6.5×10^{-5}	0.5 M H ₂ SO ₄
CoSe ₂ [15]	GCE	0.037	45	231		42.4	6.5×10^{-5}	0.5 M H ₂ SO ₄
Co _{0.56} Ni _{0.44} Se ₂ [15]	GCE			250		49.7	6.3×10^{-5}	0.5 M H ₂ SO ₄
$Co_{0.32}Ni_{0.68}S_2[15]$	GCE					66.8	3.0×10^{-4}	0.5 M H ₂ SO ₄
NiS ₂ [15]	GCE					41.6	1.4×10^{-4}	$0.5 \text{ M H}_2 \text{SO}_4$

Table 1. Summary of the HER performance of representative catalysts.

NiSe ₂ [15]	GCE	250	56.9	5.7×10^{-4}	0.5 M H ₂ SO ₄
^a nonset: overp	otential for the onset of	reductive current			

^b J₀: exchange current density

High durability is of importance for a good electrocatalyst. The stability of the Co₂P nanorods was evaluated by cyclic voltammetry (CV) sweeps between -0.240 and 0.100 V vs. RHE in the 0.5 M H₂SO₄ solution. The corresponding *j*-V curves of the CV sweeps (without *iR* correction) are shown in Figure 4b. After 1000 CV sweeps the η_{20} increases from 181 mV to 194 mV, the increase of the η_{20} being smaller than 15 mV. The current density at 0 V vs. RHE in CV sweeps (2 mA/cm², Figure 4b) is larger than that in Figure 4a (~ 0 mA/cm²). It is apparent that the larger current density is correlated with the faster scan rate (50 mV/s) in CV sweeps, implying that the current density at 0 V vs. RHE in CV sweeps could be associated with the capacitive process on the surface of the catalyst. In addition, current density exhibits only ca. 10% decrease after 12 h potentiostatic electrolysis (Figure S7, Electronic Supplementary Information). The CV sweeps and potentiostatic electrolysis experiment suggest the HER stability of the Co₂P nanorods in acidic solution.

Thus far, only rare excellent catalysts (e.g. Pt and Mo₂C)[8] have been shown to perform reliably in both acidic and basic solution. For this reason, the HER performance of the Co₂P nanorods in basic solution (KOH, 1 M) was also evaluated. The corresponding results are shown in Figure S8 (Electronic Supplementary Information), with a η_{20} value of 171 mV (*iR* corrected). After 1000 CV sweeps between -0.330 and 0.080 V vs. RHE, the η_{20} increases from 217 mV to 229 mV (without *iR* correction). The relatively small increase of the η_{20} (12 mV) and a potentiostatic electrolysis

experiment (Figure S8c, Electronic Supplementary Information) shows the stability of the Co₂P nanorods in the basic solution. These experiments demonstrate that the Co₂P nanorods can function efficiently and stably under the basic condition as well. It has been reported that the η_{20} of Ni₂P nanoparticles in basic solution (KOH, 1 M) is 205 mV, and the HER performance declined rapidly on cycling.[17] The reason for the different stabilities of the Co₂P nanorods and the Ni₂P nanoparticles in basic solution remains unknown, but the different crystal structures (orthorhombic Co₂P and hexagonal Ni₂P) are likely to be an important factor.

The faradaic efficiency of the Co₂P nanorods during H₂ evolution was probed by comparing the volume of generated gas and quantity of charges passing the Co₂P nanorods while a potentiostatic electrolysis measurement was carried out. The volume of generated gas was monitored by the water displacement method. Figure 5 shows the comparison of the theoretical volume of hydrogen and the experimentally measured volume of hydrogen. 10 C of charge pass through the cathode (~0.5 cm²) in 1310 s, 1.25 mL of H₂ should be produced according to the theoretical computation. In the experiment, the volume of generated H₂ was measured to be 1.18 mL, very close to the theoretical value. In addition, the faradaic yield of H₂ production is nearly 100% over the time scale of the measurement (Figure 5).



Figure 5. Current efficiency for HER under potentiostatic electrolysis (applied potential: -150 mV vs. RHE).

To obtain insight into the HER process and mechanism with the Co_2P nanorods, electrochemical impedance spectroscopy (EIS) investigations were carried out. The EIS experiments were carried out at different applied potentials, the results being shown in Nyquist plots in Figure 6. The EIS spectra display two semicircles. The semicircles at high frequencies (shown more clearly in Figure 6b) can be related to the contact between the catalyst (Co_2P) and the catalyst support (Ti foil), while those at low frequencies correspond to the kinetics of the HER process on the surface of the catalyst.

The diameters of the semicircles at low frequencies are potential-dependent, decreasing with increasing applied potential. This is qualitatively in accordance with faster HER kinetics occurring at higher overpotential. The kinetics of electrochemical reaction at an electrode's surface is usually assessed by charge transfer resistance (R_{ct}), with a smaller R_{ct} value corresponding to faster kinetics. R_{ct} can be deduced from EIS spectra by data fitting, in the present case using the equivalent circuit shown in Figure S9 (Electronic Supplementary Information). The results are listed in Table S1 16

(Electronic Supplementary Information). The high frequency constant phase element (CPE_1) and resistance element (R_1) are nearly potential-independent (Figures S10a and S10b, Electronic Supplementary Information). In contrast, R_{ct} is potential-dependent (Figure S10d, Electronic Supplementary Information) whereas the low frequency constant phase element (CPE_{dl}) is nearly potential-independent (Figure S10c, Electronic Supplementary Information). The invariant CPE_{dl} value suggests that the active surface area of the catalyst is maintained at different applied potentials, consistent with a high durability of the catalyst.



Figure 6. Nyquist plots of EIS responses of the Co₂P nanorods at different applied potentials in 0.5 M H₂SO₄ solution: (a) the overall spectra, and (b) data near the high frequency region. The scatters show the experimental data, and lines show the results of fitting to the equivalent circuit in Figure S9.

In general, classic two-electron-reaction models suggest that the HER process proceeds in two steps: a discharge step (Volmer reaction: $H_3O^+ + e^- \rightarrow H_{ads} + H_2O$) followed by a desorption step (Heyrovsky reaction: $H_{ads} + H_3O^+ + e^- \rightarrow H_2 + H_2O$), or a discharge step followed by a recombination step (Tafel reaction: $H_{ads} + H_{ads} \rightarrow H_2$), where H_{ads} represents an H atom absorbed at the active site of the catalyst. The rate-limiting step can be identified by the Tafel analysis of the catalyst, with a Tafel 17

slope of 116, 38, or 29 mV/decade assigning the rate-determining step in the HER process to Volmer, Heyrovsky, or Tafel reaction. The polarization curve (Figure 4a) is presented in a Tafel plot in Figure 7a, and the apparent Tafel slope obtained by fitting the linear portion of the Tafel plot to the Tafel equation. The apparent Tafel slope of Co_2P fitted from the polarization curve is 71.0 mV/decade. The apparent Tafel slope extracted from the polarization curve will be higher than the true value, if electron transport in the catalyst is sufficiently slow. For example, it has been shown that different loading amounts of MoS_x on GCE result in different Tafel slopes,[43] and the introduction of reduced graphene oxide to MoS₂ can markedly reduce the Tafel slope.[25] Hu and Vrubel et al. demonstrated that the contribution of the electron transport process in the catalyst to the Tafel slope can be eliminated if the Tafel slope is derived from EIS data.[43] The log(1/R_{et})- η plot of the Co₂P nanorods is shown in Figure 7b, showing a Tafel slope of 51.7 mV/decade.

The Tafel slope of 51.7 mV/decade lies between 38 and 116 mV/decade, suggesting that a Volmer-Heyrovsky mechanism might be responsible for the HER process,[10] and that the rates of the discharge step and the desorption step might be comparable during the HER process.[16] The Tafel slope fitted from the polarization curve is larger than that obtained from R_{ct}, suggesting that electron transport resistance in the catalyst may be well comparable to the charge transfer resistance at the catalyst/electrolyte interface. The apparent Tafel slope would thereby be reduced and, accordingly, the HER performance could be further improved by reducing the electron transport resistance in the catalyst, for example via the introduction of conductive graphene.[25] On the other hand, the turnover

frequencies (TOFs) of the HER were estimated according to the number active sites determined by CV sweep. [7] In this approach the total number of active sites is the upper limit value. Figure S11 (Electronic Supplementary Information) shows the plot of TOFs versus the applied potentials in the 0.5 M H₂SO₄ solution. The Co₂P nanorods show a TOF of 0.725 S⁻¹ at -143 mV vs. RHE. This value is larger than that of CoP/CC (\sim -75 mV vs. RHE),[20] and smaller than that of MoS₃ deposited on GCE (\sim -220 mV vs. RHE). [7]



Figure 7. (a) Apparent Tafel slope fitted from polarization data measured in 0.5 M H₂SO₄ solution, and (b) Tafel slope fitted from EIS data measured in 0.5 M H₂SO₄ solution. Scatters show experimental data and the solid lines show the corresponding fitting.

The HER catalytic activity of the Co_2P nanorods might be correlated with the unique charged natures of Co and P species. The kinetics of HER in [NiFe] hydrogenase and its analogues $([Ni(PS3^*)(CO)]^{1-}$ and $[Ni(PNP)_2]^{2+})$ have been assessed via density functional theory simulation.[23] It was revealed that the efficient hydrogenase and its analogues have both proton acceptor and hydride

acceptor sites. The proton acceptor site is the nonmetal site having a small negative charge to trap protons (e.g., O of Glu23 in hydrogenase, -0.44 e; S in Ni(PS3*)(CO)]¹⁻, -0.4 e; N in $[Ni(PNP)_2]^{2+}$, -0.34 e). The hydride acceptor site (Ni in hydrogenase, Ni(PS3*)(CO)]¹⁻, and $[Ni(PNP)_2]^{2+}$) is the isolated metal atom providing moderate bonding to hydrogen, and the hydride acceptor site has a slightly positive charge, suggested by the calculated total electron density of hydrogenase, $Ni(PS3^*)(CO)$ ¹⁻, and $[Ni(PNP)_2]^{2+}$. The proton acceptor and hydride acceptor sites work cooperatively, resulting in high HER catalytic activities. Ni₂P(001) also has proton acceptor sites (P) and hydride acceptor sites (Ni), and it has been predicted [23] and then proved [17] to be highly active HER catalyst. The hydride acceptor (Ni) and proton acceptor (P) in Ni₂P has small positive and negative charges, respectively, suggested by the fact that the peak binding energies of Ni (853.5 eV) and P (129.5 eV) in Ni₂P are very close to those of the corresponding zero valence species (852.8 eV for Ni and 130.0 eV for P).[44] It has been revealed that Co and P in Co₂P also have a small positive and negative charge (Figure 3 and corresponding discussion). The charged natures of Co and P in Co₂P are therefore similar to those of hydride acceptor and proton acceptor in Ni₂P, [NiFe] hydrogenase, $[Ni(PS3^*)(CO)]^{1-}$, and $[Ni(PNP)_2]^{2+}$, respectively. The similarity in charged natures implies that similar catalytic mechanism might work in the Co₂P nanorods as Ni₂P, [NiFe] hydrogenase, [Ni(PS3*)(CO)]¹⁻, and [Ni(PNP)₂]²⁺), resulting to high HER catalytic activity reported here.

4. Conclusions

In conclusion, Co₂P nanorods were demonstrated as a new efficient HER electrocatalyst. The HER performance of the Co₂P nanorods is comparable to those of high efficiently non-precious electrocatalysts. Accelerated degradation and potentiostatic electrolysis experiments have demonstrated the stability of the Co₂P nanorods in acidic and basic solution during the HER process. The faradaic yield of H₂ production is nearly 100%. The HER process was found to follow a Volmer-Heyrovsky mechanism, and the rates of the discharge and desorption step appear to be comparable during the HER process. The HER performance could be further improved by the introduction of conductive species. Studies directed toward this goal are currently underway.

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Accepted manuscript

Research highlight

A convenient method is used to fabricate cobalt phosphide (Co₂P) nanorods.

Co₂P nanorods exhibit efficient electrocatalytic activity in hydrogen evolution reaction.

Co₂P nanorods work stably in both acidic and basic solutions.

Accepted manuscript

Graphical abstract

