Nanoencapsulation of Luminescent 3-Hydroxypicolinate Lanthanide Complexes

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We have synthesized luminescent nanoparticles comprising a core of lanthanide complexes and shells of amorphous silica using reverse micelles as nanoreactors. 3-Hydroxypicolinate complexes of Eu(III), Tb(III), and the corresponding heteronuclear complexes have been investigated as the photoactive cores. The size of the silica particles is within the nanometer scale, which, together with the ability for surface biofunctionalization, opens up perspectives for their use in bioapplications. Optical studies of the as-prepared nanoparticles reveal that the luminescence properties of the 3-hydroxypicolinate complexes in the matrices are markedly different from their original features.

1. Introduction

The synergy between nanotechnology and the biological sciences is especially important in pushing the frontiers in medicine, ^{1–3} environment, ⁴ and food technology. ⁵ For biomedical sciences, especially, nanotechnology holds a lot of potential in areas such as drug delivery, gene delivery, and bioassays. ^{2,3,6,7} One obvious advantage of employing nanoparticles in bioassays is that nanoparticles are within dimensions of biomolecules and cellular components and, thus, can act as local probes and sensors. ^{3,8} This is especially true for homogeneous assays where, in addition to their small size, nanoparticles also provide large surface areas (by volume of material) for interaction with specific biomolecules.

Water-in-oil microemulsions (W/O's) or reverse micelles have become very important and widely used in the preparation of nanoparticles. Monodispersed particles can be prepared when syntheses are carried out in the nanosized water pools that can be regarded as dynamic nanoreactors.^{6,9–13} They also offer the possibility of size tuning by varying the [water]/[surfactant] ratio *R*; the lower the *R* value, the smaller the average size of the nanoparticles.^{6,14} In this context, W/O's have been used in the preparation of silica,^{6,15} titania,¹¹ perovskites,¹⁶ semiconductor nanoparticles,^{10,17,18} and even organic nanoparticles.¹ Consequently, SiO₂ nanoparticles prepared in W/O's have been reported to encapsulate a wide variety of luminescent compounds, ranging from semiconductor nanocrystals,^{19,20} YF₃ nanoparticles,⁹ organic/inorganic dyes, and lanthanide (Ln) chelates.^{21,22}

In this work, we have investigated this approach to encapsulate lanthanide chelates in silica nanoparticles prepared by the Stober method. 3-Hydroxypicolinate complexes of europium(III) and terbium(III), as well as the mixed-lanthanide (Eu and Tb) complexes, were used for silica encapsulation. The discussion on 3-hydroxypicolinate complexes will be largely skewed in favor of Eu³⁺ due to the high luminescent behavior of both the pure complex and the derived nanocomposites.

Considering that, in general, lanthanide complexes show high luminescence quantum yields, narrow bandwidth and long-lived emission, large Stokes' shifts, and ligand-dependent luminescence sensitization, ^{23,24} these nanodispersed systems can be of interest as new bioprobes.

2. Experimental Section

2.1. Materials. 3-Hydroxypicolinic acid (HpicOH, Aldrich, 98%), sodium bis(2-ethylhexyl) sulfosuccinate (AOT, Aldrich, 96%), europium(III) chloride hexahydrate (Aldrich, 99.9%), terbium(III) chloride hexahydrate (Aldrich, 99.9%), and tetraethyl orthosilicate (TEOS, Aldrich, 98%) were obtained from commercial sources and used without further purification.

2.2. Synthesis of the Lanthanide Complexes. The 3-hydroxypicolinate complexes, $[Ln(H_2O)(picOH)_2(\mu-HpicO)] \cdot nH_2O$ (Ln = Eu (1), Tb (2)), were prepared as reported previously.²⁵ A similar procedure was employed for the synthesis of the mixed-lanthanide complexes, Eu_{0.5}Tb_{0.5} (3), Eu_{0.3}Tb_{0.7} (4), and Eu_{0.1}Tb_{0.9} (5), by adding the stoichiometric amounts of EuCl₃·6H₂O and TbCl₃·6H₂O. The lanthanide hydroxides used as starting compounds were prepared by adding aqueous solutions (8 mL) of KOH (4.5 mmol) to an equal volume of aqueous solutions containing the lanthanide chloride salt (1.5 mmol). The solid formed was stirred for over 90 min, filtered, and washed thoroughly with distilled water. The freshly prepared Ln(OH)₃ (1 mmol) was added to an aqueous solution (25 mL) containing 3-hydroxypicolinic acid (4 mmol). This mixture was stirred during 1 h, then heated for 30 min at 80 °C, and further stirred over 8 h at room temperature. The obtained solid was filtered, washed thoroughly with distilled water, and dried over silica gel.

2.3. Preparation of Silica Nanoparticles. Water-in-oil microemulsions were prepared by dissolving 0.6 g of AOT in 10 mL of heptane, followed by addition of distilled water (50, 100, 150, 200, and 245 μ L, corresponding to R = 2, 4, 6, 8, and 10, respectively) under vigorous stirring. Next, 5 mg of the lanthanide complex or 10 mg of a mixture of the europium complex and the terbium complex [5:5 (6), 3:7 (7), and 1:9 (8)] dissolved in dimethyl sulfoxide (DMSO) was added, followed by addition of 100 μ L of TEOS and, finally, 100 μ L of a NH₄OH aqueous solution (25%). Each step had a 5–7 min

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interval between them. After the mixture was stirred over 24 h, the silica nanoparticles were precipitated out by addition of acetone and then separated by centrifugation. The luminescent silica nanoparticles (1/SiO₂-R) were washed twice with heptane and ethanol, followed by 1-propanol, 1-butanol, and ethanol.

2.4. Instrumentation. FT-IR spectra were obtained as KBr pellets using a Mattson 7000 FT instrument. Raman spectra were recorded using a Bruker RFS100/S FT-Raman spectrometer (Nd: YAG laser, 1064 nm excitation). Elemental analyses for C, H, and N were performed with a CHNS-932 elemental analyzer in the Microanalysis Laboratory of the University of Aveiro. The lanthanide content for the heteronuclear complexes was obtained by ICP (inductively coupled plasma) analysis on an ICP-AES Jobin Yvon model 70+ at the Central Analytical Laboratory of the University of Aveiro.

The scanning electron microscopy (SEM) images were obtained on a Hitachi SU 70 microscope fitted with an energy-dispersive spectroscopy (EDX) accessory (EDX Detector, Bruker AXS, Quantax software). The transmission electron microscopy (TEM) study of the nanoparticles' morphology was carried out with a Hitachi H-9000-NA TEM or using a HR-(EF) TEM JOEL 2200FS, both operating at 200 kV. Samples were prepared by dipping carbon-coated copper grids in dilute ethanol solutions of the samples and then letting the solvent evaporate. Average particle sizes were determined by statistical calculations based on the TEM images only.

The excitation and emission spectra were obtained at 300 K on a Jobin Yvon-Spex spectrometer (HR 460) coupled to a R928 Hamamatsu photomultiplier under continuous excitation of a Xe arc lamp (150 W). The excitation monochromator was a Jobin Yvon (TRIAX 180). The spectra were corrected for the response of the detector. The lifetime measurements and the excitation and emission spectra of 2 were recorded at 300 K on a modular double-grating excitation spectrofluorimeter with a TRIAX 320 emission monochromator (Fluorolog-3, Jobin Yvon-Spex) coupled to a R928 Hamamatsu photomultiplier using the front face acquisition mode. All the photoluminescence spectra were collected on solid samples at room temperature and corrected for optical and detection spectral responses. The roomtemperature lifetime measurements were acquired with the setup described for the luminescence spectra using a pulsed Xe-Hg lamp (6 μ s pulse at half-width and a 20-30 μ s tail). The ⁵D₀(Eu³⁺) lifetimes were monitored within the more intense emission line of the ${}^5D_0 \rightarrow {}^7F_2$ transition, under the excitation that maximizes the Eu³⁺ emission. The emission decay curves (not shown) were well-reproduced by single exponentials.

3. Results and Discussion

3.1. Synthesis and Material Characterization. A series of Eu(III) and Tb(III) complexes with 3-hydroxypicolinic acid were synthesized and structurally characterized by elemental analysis and infrared, Raman, and photoluminescence spectroscopy. Subsequently, the lanthanide complexes, $[Ln(H_2O)(picOH)_2(\mu-HpicO)] \cdot nH_2O (1-5)$, were encapsulated into amorphous silica nanoparticles (complex/SiO₂-R, where R = 2, 4, 6, 8, and 10), as schematically illustrated in Figure 1.

Table 1 shows the infrared and Raman spectroscopic data for 3-hydroxypicolinic acid and its lanthanide complexes together with the elemental analysis for C, H, N, and Ln. Elemental analysis results are in accordance with the molecular formula $[Ln(H_2O)(picOH)_2(\mu-HpicO)] \cdot nH_2O$ for all the lanthanide complexes (1-5), which is consistent with the vibrational data presented below. Additionally, the vibrational data for the mixed-lanthanide complexes (3-5) are in good agree-



Figure 1. Schematic representation of reverse micelles as nanoreactors for the SiO₂ encapsulation of Ln complexes.

ment with those described for the pure lanthanide (Eu and Tb) complexes already reported in the literature.²⁵

To confirm the chemical homogeneity of the heteronuclear bulk samples, the spatial distribution of europium and terbium in the mixed-lanthanide complexes 3–5, and in the derived nanoparticles, was analyzed by EDX. Figure 2 shows the SEM image and corresponding EDX mapping for the Eu and Tb elements of a typical nanosilica sample (5/SiO₂-10). These results show that both lanthanide ions are homogeneously distributed in the final SiO₂ nanoparticles, as expected, in a material containing those ions homogeneously mixed at the molecular level. For comparison, the EDX mapping for the Eu and Tb elements in the starting complexes (3–5) used for the encapsulation is shown in the Supporting Information (Figures S1–S3).

TEM analysis revealed particles with a spherical morphology, as illustrated in Figure 3, for the sample 3/SiO₂-10. Additionally, these TEM results also confirmed the core-shell structure of these hybrid lanthanide/SiO₂ materials that comprise a nanometric core assigned to the encapsulated Ln complex. The R value was used to change the particle size, and spheroidal core/ shell type nanoparticles were observed for the several samples analyzed. Because of the high sensitivity of the morphology of these nanostructures to the synthetic conditions employed, the particle size of the nanoparticles could vary considerably. It was observed that the diameter of the composite particles could range from 30 to 55 nm, with an average size of 45 nm and a particle size dispersion superior to 10%. Nevertheless, the TEM results (Figure 3a) showed core—shell structures with a typical diameter of about 9 nm for the core of the composite particles. For the sake of comparison, similar syntheses made in the absence of the complex result in SiO₂ particles whose TEM images (Figure 3b) did not show the darker cores. On the other hand, samples of the pure complex did not appear as morphologically uniform particles, but still, we cannot exclude the presence of nonencapsulated complexes dispersed in these samples. In fact, the results presented here provide the basis for the SiO₂ nanoencapsulation of lanthanide complexes using reverse micelles, but we hope to continue our investigations to optimize the nanocomposites' morphological characteristics. Because the surfactant molecules can compete for the lanthanide sites, it will be interesting, for example, to see how the morphological homogeneity is favored by varying the chemical nature of the lanthanide complex and/or the surfactant.

3.2. Photoluminescence. Photoluminescence measurements were carried out for the silica nanoparticles with different sizes encapsulating the europium(III) 3-hydroxypicolinate complex $(1/\text{SiO}_2\text{-}R; R = 2, 4, 6, \text{ and } 10)$ and compared with those of the pure europium complex. The excitation and emission spectra of $1/\text{SiO}_2\text{-}R$ together with 1 are shown in Figure 4. The excitation spectra of the materials (Figure 4A) show a large broad band ranging from 250 to 400 nm and a series of sharp

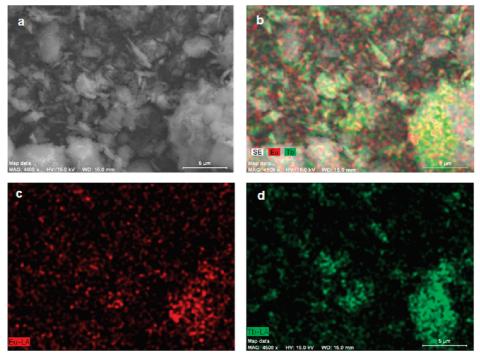


Figure 2. (a) SEM image and corresponding EDX distribution maps of (b) Eu:Tb, (c) Eu, and (d) Tb in 5/SiO₂-10.

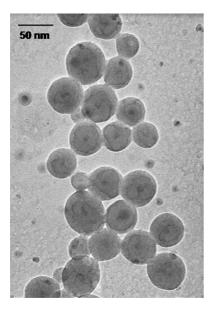
TABLE 1: Analytical and Vibrational Data for the Free Ligand and Its Lanthanide Complexes (1–5)

	analysis ^a (%)				vibrational spectra ^b (cm ⁻¹)			
compound	С	Н	N	Eu; Tb	$\overline{\nu_{\rm as}({ m CO}_2)}$	ν(C-N)	$\nu_{\rm s}({ m CO_2})$	ν (C-O) _h
3-hydroxypicolinic acid					1702 s	1608 s	1320 s	1284 vs
					-	1612(6)	1323(4)	1286(2)
$[Eu(H_2O)(picOH)_2(\mu-HpicO)] \cdot 4H_2O (1)$	32.1	3.1	6.4		1623 vs	1592 vs	1330 sh	1263 s
	(32.9)	(3.4)	(6.4)		1619 vw	1596 vw	1322 vw	1259 w
$[Tb(H2O)(picOH)2(\mu-HpicO)] \cdot 6H2O (2)$	30.1	2.9	6.1		1627 vs	1592 vs	1332 s	1261 s
	(30.9)	(3.7)	(6.0)		1626 υw	1598 w	1322 vw	1259 vw
$[Eu_{0.5}Tb_{0.5}(H_2O)(picOH)_2(\mu-HpicO)] \cdot 5H_2O$ (3)	31.4	2.6	6.0	13.4; 13.9	1625 vs	1592 vs	1332 s	1261s
	(31.9)	(3.6)	(6.2)	(11.5; 12.0)	1625 w	1591 m	1325 m	1260 m
$[Eu_{0.3}Tb_{0.7}(H_2O)(picOH)_2(\mu-HpicO)] \cdot 5H_2O$ (4)	31.0	2.6	6.0	8.1; 18.9	1625 vs	1592 vs	1332 s	1261 s
	(31.8)	(3.6)	(6.2)	(6.9; 16.8)	1628 w	1592 m	1321 m	1261 m
$[Eu_{0.1}Tb_{0.9}(H_2O)(picOH)_2(\mu-HpicO)] \cdot 5H_2O$ (5)	31.4	2.6	6.0	2.8; 23.3	1625 vs	1592 vs	1332 s	1261 s
	(31.8)	(3.6)	(6.2)	(2.3; 21.6)	1624 w	1592 m	1326 m	1261 m

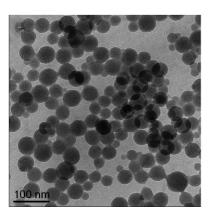
^a Calculated values in parentheses. ^b Infrared and Raman (in italics) data: vs, very strong; s, strong; m, medium; w, weak; sh, shoulder.

lines assigned to intra-4f⁶ transitions. The large broad band can be ascribed to the excited levels of the ligand (L),²⁵ indicating the existence of ligand-to-Eu³⁺ energy transfer. The higher intensity of the large broad band relative to that of the intra-4f6 transitions reveals that the luminescence sensitization via the ligands' excited states is more efficient than f—f direct excitation. The peak position corresponding to the maximum intensity of the broad band ranges from 363 to 380 nm for the 1/SiO₂-R series (375 nm for 1). Excitations within the ligands' absorption band (325–380 nm) for 1/SiO₂ and 1 lead to the typical Eu³⁺ emission spectra (${}^5D_0 \rightarrow {}^7F_{0-4}$), without any broad emission band arising from the organic ligands (Figure 4B). No significant changes are detected on the emission spectra of the 1/SiO₂-R series exciting within the ligands' absorption band or within the direct europium absorption lines (not shown). However, differences are detected in the relative intensity of the ${}^5D_0 \rightarrow$ ⁷F_{2.4} transitions for the 1/SiO₂ series by varying the size of the nanoparticles. The small differences detected in the emission and excitation spectra of the 1/SiO₂-R series indicate that the local environment of the Eu³⁺ is similar even though it is not the same. Further studies will be performed to better understand these differences that lie outside the scope of this work.

A meticulous analysis of the emission spectra of the nanoparticles shows some variations when compared with that of the starting europium complex, namely, of the energy (E) and full width at half-maximum (fwhm) of the ${}^5D_0 \rightarrow {}^7F_0$ transition and maximum splitting of the $^5D_0 \rightarrow {}^7F_1$ transition (ΔE). The corresponding values are listed in Table 2, where 1/SiO₂-10 was selected to represent the $1/SiO_2$ series. The fwhm of the ${}^5D_0 \rightarrow$ $^{7}\text{F}_{0}$ transition for 1 is 20 cm $^{-1}$, whereas for 1/SiO₂-10, it is 30 cm⁻¹, revealing an increased nonhomogeneity for the Eu³⁺ local coordination site.²⁶ This fact brings to the fore the question of whether the process is a mere encapsulation, as opposed to a sort of chemical reaction between the matrix and the europium complex leading to extensive modification of the local environment of Eu³⁺. Changes in the ${}^5D_0 \rightarrow {}^7F_0$ energy are highly dependent on the covalency of the Eu³⁺-L bonds of the first Eu³⁺ coordination sphere. A red shift of the ${}^5D_0 \rightarrow {}^7F_0$ energy with respect to the energy of the gaseous Eu³⁺ (17374 cm⁻¹) means an increase in covalency.^{27–29} Thus, the blue shift found for the ${}^5D_0 \rightarrow {}^7F_0$ transition in $1/SiO_2$ relative to the europium pure complex is an indication of a decreasing covalency of the Eu³⁺-L bonds in the first Eu³⁺ coordination sphere in the silica matrix. A similar shift was found for the silica nanocomposite



(a)



(b)

Figure 3. TEM images of 3/SiO₂-10 (a) and SiO₂ particles (b).

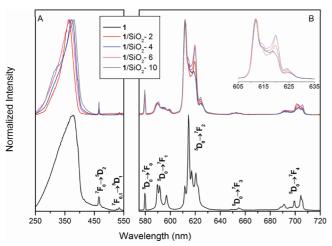


Figure 4. (A) Excitation and (B) emission spectra of **1** and **1**/SiO₂-R monitored at 614.5 (for **1**) and 612 nm (for **1**/SiO₂-R) and excited at 375 (**1**), 363 (R=2), 373 (R=4), 368 (R=6), and 380 nm (R=10). The inset in B shows a detailed view of the $^5D_0 \rightarrow ^7F_2$ transition.

containing the europium complex in our earlier work,²⁵ where it was suggested that the europium complex in the nanocomposite is bound to the silica surface via the silanol groups,

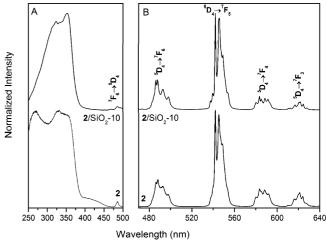


Figure 5. (A) Excitation and (B) emission spectra of **2** and **2**/SiO₂-10 monitored at 544 nm and excited at 320 nm, respectively.

TABLE 2: Energy (*E*) and fwhm of the ${}^5D_0 \rightarrow {}^7F_0$ Transition and Maximum Splitting (ΔE) of the ${}^5D_0 \rightarrow {}^7F_1$ Transition for Selected Materials

	⁵ D ₀	$^{5}D_{0} \rightarrow {}^{7}F_{1}$	
sample	$E \text{ (cm}^{-1})$	fwhm (cm ⁻¹)	$\Delta E \text{ (cm}^{-1}\text{)}$
1	17 249	20	205
1/SiO ₂ -10	17 258	30	260
5	17 258	21	325
5 /SiO ₂ -10	17 247	34	236
8/SiO ₂ -10	17 258	22	347

possibly displacing one bridging carboxylate group from the Eu³⁺ first coordination sphere. Therefore, similar interactions should occur between the europium complex and the silica nanoparticles (1/SiO₂-R). The maximum splitting of the ${}^5D_0 \rightarrow$ ⁷F₁ transition has been reported as being directly proportional to the strength of the ligand (crystal) field interaction. 26,29 This value increased for the 1/SiO₂ series relative to the pure europium complex, indicating a greater crystal field interaction strength than that in 1. The main ${}^5D_0 \rightarrow {}^7F_2$ emission peak shifts from $16\,273~{\rm cm}^{-1}~(\sim\!614.5~{\rm nm})$ for ${\bf 1}$ to about $16\,340~{\rm cm}^{-1}$ (∼612 nm) in the 1/SiO₂ series. Despite all the spectroscopic evidence indicating that the Eu³⁺ local coordination changes from 1 to $1/SiO_2$, the local field splitting of the ${}^5D_0 \rightarrow {}^7F_1$ and $^5D_0 \rightarrow {}^7F_2$ transitions into three and five Stark components, respectively, points out that the Eu³⁺ ions occupy, in both materials, a low site symmetry without an inversion center (high intensity of the ${}^5D_0 \rightarrow {}^7F_2$ transition²⁵).

The experimental lifetime of the excited ⁵D₀(Eu³⁺) energy level, $\tau_{\rm exp}$, for 1/SiO₂-10 is 0.450 \pm 0.003 ms, slightly longer than the corresponding value measured for 1 (0.431 \pm 0.002 ms). The quantum efficiency, q, of the excited 5D_0 state, q = $k_{\rm r}/(k_{\rm r}+k_{\rm nr})$, where $k_{\rm r}$ and $k_{\rm nr}$ are the radiative and nonradiative transition probabilities, respectively, was calculated on the basis of the emission spectrum and 5D_0 lifetime $(\tau_{\rm exp}^{-1} = A_{\rm r} + A_{\rm nr})$. 25,26,29 The $k_{\rm r}$ contribution is calculated from the relative intensities of the ${}^5D_0 \rightarrow {}^7F_{0-4}$ transitions (the ${}^5D_0 \rightarrow {}^7F_{5,6}$ branching ratios are neglected due to their poor relative intensity with respect to that of the remaining ${}^5D_0 \rightarrow {}^7F_{0-4}$ lines). The ${}^5D_0 \rightarrow {}^7F_1$ transition does not depend on the local ligand field and, thus, may be used as a reference for the whole spectrum. An effective refractive index of 1.5 was used, leading to $A_{01} \approx 50 \text{ s}^{-1}$; A_{01} stands for the Einstein's coefficient of spontaneous emission between the ⁵D₀ and the ⁷F₁ Stark levels.

We found an increase in the calculated 5D_0 quantum efficiency for $1/\text{SiO}_2$ -10 (23%) when compared with that of 1 (11%) due

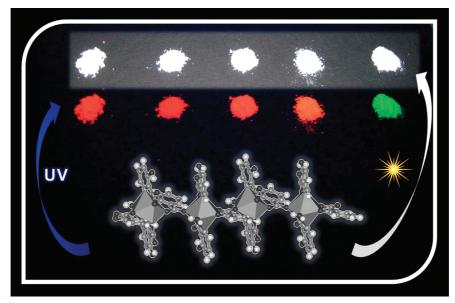


Figure 6. Powdered lanthanide complexes under UV light and natural light, following the sequence 1, 3, 4, 5, and 2, from left to right.

to an increase for the former material of the radiative transition probability, $k_{\rm r}$, and to a decrease of the nonradiative one, $k_{\rm nr}$ $(0.518 \text{ and } 1.70 \text{ ms}^{-1}, \text{ respectively, for } 1/\text{SiO}_2\text{-}10 \text{ and } 0.497$ and 2.07 ms^{-1} for 1).

The increase in q and τ_{exp} for the europium complex when it is embedded in a silica matrix (via a sol-gel approach) over those of a pure europium complex corroborates earlier findings reported by us.25

The terbium(III) complex (2) was also incorporated into the silica matrix (R = 2, 4, 6, 8,and 10). The excitation and emission spectra of 2 and 2/SiO₂-10 are shown in Figure 5. The excitation spectra of the nanoparticles (Figure 5A) show a large broad band ranging from 250 to 380 nm (already ascribed to the excited levels of the ligands) and the ${}^{7}F_{6} \rightarrow {}^{5}D_{4}$ transition with low intensity relative to the large broad band. This large broad band, also observed in 2 and in the 1-based materials described above, shows similar profiles for R = 6, 8, and 10 (Figure S4A, Supporting Information) and is double-crested with the first peak around 320 nm and the second around 350 nm. Indeed, the excitation spectra profile of 1/SiO₂-10 may be considered as approximately a single-peak (maximum at 380 nm) with a shoulder around 320 nm (Figure 4A). Figure 5B shows the emission spectra of 2 and 2/SiO₂-10 excited at 320 nm, which display the typical intra-4f⁸ lines of the Tb³⁺ ion $(^{5}D_{4} \rightarrow {}^{7}F_{6-3})$. Varying the particle size or the excitation wavelength does not produce significant changes in the emission spectra of the terbium-based nanoparticles (Figure S4B, Supporting Information).

The Eu^{3+}/Tb^{3+} mixed complexes **3** ($Eu_{0.5}Tb_{0.5}$), **4** ($Eu_{0.3}Tb_{0.7}$), and 5 (Eu_{0.1}Tb_{0.9}) (prepared and characterized for the first time in this paper) were encapsulated into silica nanoparticles (R =8 and 10). Figure 6 exhibits the digital photographs of the powdered lanthanide complexes 3-5 under UV light and natural light. Figure 7A displays the excitation spectra of 3, 4, and 5 monitoring the ${}^5D_0 \rightarrow {}^7F_2$ (620 nm) transition. The observed large broad band is mainly formed of a peak at 357 nm and a shoulder around 330 nm. The existence of the Tb³⁺-to-Eu³⁺ energy transfer is supported by the detection of the Tb³⁺ $^{7}F_{6} \rightarrow$ ⁵D₄ transition located at 486 nm on the excitation spectra of 3, **4**, and **5** (when the Eu³⁺ emission is being monitored).

The emission spectra of all the mixed-lanthanide complexes excited within the ligands' broad band (357 nm) (Figure 7B)

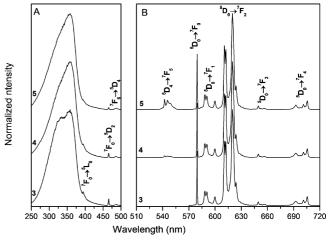


Figure 7. (A) Excitation and (B) emission of 3, 4, and 5 monitored at 620 nm and excited at 357 nm, respectively.

exhibit, besides the Tb³⁺ $^5D_4 \rightarrow ^7F_5$ lines, the typical Eu³⁺ 5D_0 \rightarrow ⁷F₀₋₄ transitions, suggesting the occurrence of ligand-to-Tb³⁺, ligand-to-Eu³⁺, and Tb³⁺-to-Eu³⁺ energy transfer. The most intense emission line, which is the main Stark component of the $^5D_0 \rightarrow ^7F_2$ transition, is located at 16 129 cm⁻¹ (~620 nm), in contrast to 16 273 cm⁻¹ (\sim 614.5 nm) for 1. Additionally, the energy of the ${}^5D_0 \rightarrow {}^7F_0$ transition for the mixed-lanthanide complexes increased above that of 1 (Table 2), although the fwhm was maintained. The ΔE of the ${}^5D_0 \rightarrow {}^7F_1$ transition for **5** (325 cm⁻¹) increased well above the value for **1** (205 cm⁻¹), showing a stronger ligand field interaction in the mixedlanthanide complexes relative to that of 1. Therefore, it can be seen from the foregoing that the presence of the Tb³⁺ ions in the mixed-lanthanide complexes affects the local coordination of the Eu³⁺ ions and, hence, their optical features. It is noteworthy that the relative intensity of the ${}^{7}F_{0} \rightarrow {}^{5}D_{2}$ transition decreases (Figure 7A) and that of the $^5D_4 \, \rightarrow \, ^7\!F_6$ transition increases (Figure 7B) when the Eu3+ ions are gradually substituted by the Tb³⁺ ions.

All the mixed-lanthanide complexes were encapsulated into silica nanoparticles (R = 8 and 10). Their excitation spectra (not shown) roughly approximate those of the mixed-lanthanide complexes whose peaks range from 355 to 370 nm, and their

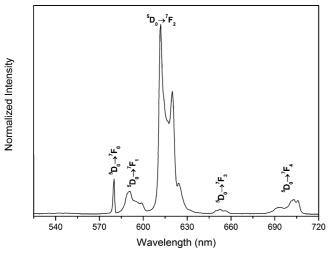


Figure 8. Emission spectra of 5/SiO₂-10 excited at 357 nm.

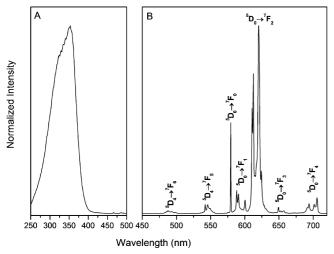


Figure 9. (A) Excitation and (B) emission spectra of 8/SiO₂-10 monitored at 620 nm and excited at 353 nm, respectively.

emission spectra (Figure S5 in the Supporting Information) are very similar. Nevertheless, changes are observed in the emission spectra of 5/SiO₂-10 (Figure 8) relative to the isolated complex 5. When 5 is incorporated into the silica matrix, the fwhm of the ${}^5D_0 \rightarrow {}^7F_0$ transition increased up to 13 cm⁻¹ (Table 2), revealing an increased nonhomogeneity for the Eu³⁺ local coordination site, whereas the energy of the ${}^5D_0 \rightarrow {}^7F_0$ transition decreased up to 11 cm⁻¹, revealing that, on average, the Eu³⁺-L bonds in the nanocomposites experience some increase in covalency. The ΔE of the $^5\mathrm{D}_0 \rightarrow ^7\mathrm{F}_1$ transition for 5/SiO₂-10 decreased significantly to 236 cm⁻¹ compared with that of 5 (325 cm⁻¹); this indicates a lower crystal field interaction strength. The most intense emission line shifts from 16 129 cm⁻¹ $(\sim 620 \text{ nm})$ for the mixed-lanthanide complexes to around 16 340 cm⁻¹ (612 nm) for the mixed-lanthanide silica nanoparticles, and the number of Stark components detected for the $^5D_0 \rightarrow ^7F_{1,2}$ transitions is one less than the number in the mixedlanthanide complexes. It is of considerable importance to emphasize that the Tb³⁺ ${}^5D_4 \rightarrow {}^7F_5$ transition detected in the mixed-lanthanide complexes (Figure 7A) disappeared in the related nanocomposites (Figure S5 in the Supporting Information). For that reason, it can be assumed that the Tb³⁺-to-Eu³⁺ energy transfer in the $3-5/SiO_2$ -R series is more efficient than that in the starting complexes and/or that nonradiative deactivation of the Tb³⁺ excited levels occurs.

The physical mixtures of the europium complex (1) and the terbium complex (2) in the stoichiometric amounts of 5:5, 3:7, and 1:9 were incorporated into the silica matrix, giving the materials named here as 6/SiO₂-R, 7/SiO-R, and $8/SiO_2$ -R (R = 6, 8, and 10), respectively. The excitation and emission spectra of 8/SiO₂-10 (Figure 9) are very similar to those of 6/SiO₂-10 and 7/SiO₂-10 (Figure S6 in the Supporting Information). The excitation spectrum of 8/SiO₂-10 monitoring the ${}^5D_0 \rightarrow {}^7F_2$ (Eu³⁺, 620 nm) transition only shows a large broad band mainly formed of a peak around 350 nm and a shoulder around 320 nm. For 6/SiO₂-10 and 7/SiO₂-10 (Figure S6 in the Supporting Information), the ⁵L₆ and ⁵D₂ Eu³⁺ levels and the ⁵D₄ Tb³⁺ level are detected, the latter being clear evidence of the occurrence of Tb³⁺-to-Eu³⁺ energy transfer. The profile of the large broad band splits into two crests, one centered around 322 nm and the other around 356 nm; this is particularly seen for 6/SiO₂-6, 6/SiO₂-8, and 7/SiO₂-8 (not shown). The similarity between the emission spectra of 8/SiO₂-10 and 5 (Figures 9 and 7 and Table 2) is clear, particularly, for the values of E and the fwhm of the ${}^5D_0 \rightarrow {}^7F_0$ transition. When we consider that the Eu³⁺/Tb³⁺ stoichiometric amounts in 8/SiO₂-10 and 5 are retained, the decrease of the relative intensity of the ${}^5D_4 \rightarrow$ $^{7}F_{5}$ transition (when compared with that of the $^{5}D_{0} \rightarrow ^{7}F_{2}$ transition) of 0.05 in 8/SiO₂-10 (compared with the value of 0.11 for 5) shows that the Tb³⁺-to-Eu³⁺ energy transfer in 8/SiO₂-10 is more efficient than that in 5 and/or that nonradiative deactivation of the Tb³⁺ excited levels occurs. In contrast to that observed for the $3-5/SiO_2$ series, the detection of the Tb³⁺ $^5D_4 \rightarrow {}^7F_{6,5}$ transitions points out the (i) lower efficiency of the Tb³⁺-to-Eu³⁺ energy transfer, (ii) occurrence of radiative deactivation of the Tb3+ excited levels, or (iii) existence of clustering of a given Ln³⁺ in a separate material domain. In conclusion, the Eu³⁺ local environment in this series suffered less alteration than in the remaining nanoparticles, as can be seen from the values of the fwhm of the ${}^5D_0 \rightarrow {}^7F_0$ transition, which are significantly lower than those of the 1/SiO₂ and 5/SiO₂ and closer to 1

4. Conclusions

Reverse microemulsions were used to encapsulate 3-hydroxypicolinate complexes of Eu(III) and/or Tb(III) complexes in silica nanoparticles. Our earlier work to prepare silica nanoparticles by encapsulating 1 using the sol-gel method²⁵ led to particles with an average diameter above 100 nm; here, the preparation of particles with an average diameter of 45 nm and below confirmed that reverse micelles are useful for producing such nanoparticles that are less than 100 nm in diameter. All the photoluminescence results confirmed the presence of the lanthanide complexes in the final nanoparticles and demonstrated the potential of the employed chemical method to yield luminescent SiO₂ nanoparticles. Moreover, the differences in the ⁵D₀ lifetimes and quantum efficiency and profile, energy, and fwhm of the ⁵D₀ \rightarrow ⁷F₀₋₂ transitions between the nanoparticles and their starting complexes clearly indicate modifications in the Eu³⁺ local environment when the complexes were embedded in silica matrices. Clearly, this work showed that heteronuclear lanthanide complexes with adjusted visible photoluminescence could be encapsulated in nanosized SiO₂. Thus, this method provides an alternative strategy to the synthesis of photoluminescent particles containing lanthanide species.^{30,31} In particular, the biofunctionalization of systems similar to

those reported here might have interest for in vitro applications, such as optical biotagging.

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Supporting Information Available: EDX maps of **3**, **4**, and **5** and additional excitation and emission spectra of **2**/SiO₂-*R*, **3–5**/SiO₂-*R*, and **6–8**/SiO₂-10. This material is available free of charge via the Internet at http://pubs.acs.org.

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