

## Research Article

# Facile Solventless Synthesis of a Nylon-6,6/Silver Nanoparticles Composite and Its XPS Study

Raúl A. Morales-Luckie,<sup>1</sup> Víctor Sánchez-Mendieta,<sup>2</sup> Oscar Olea-Mejía,<sup>1</sup>  
Alfredo R. Vilchis-Nestor,<sup>1</sup> Gustavo López-Téllez,<sup>1</sup> Víctor Varela-Guerrero,<sup>1</sup>  
L. Huerta,<sup>3</sup> and Jesús Arenas-Alatorre<sup>4</sup>

<sup>1</sup> Centro Conjunto de Investigación en Química Sustentable UAEM-UNAM (CCIQS), Facultad de Química, Universidad Autónoma del Estado de México, Km. 14.5 de la carretera Toluca-Atlacomulco, 50200 San Cayetano, MEX, Mexico

<sup>2</sup> Facultad de Química, Universidad Autónoma del Estado de México, Paseo Colón y Paseo Toluca, 50120 Toluca, MEX, Mexico

<sup>3</sup> Instituto de Investigaciones en Materiales, Universidad Nacional Autónoma de México, Apartado Postal 70-360, 04510 México, DF, Mexico

<sup>4</sup> Instituto de Física, Universidad Nacional Autónoma de México, Apartado Postal 20-364, 01000 México, DF, Mexico

Correspondence should be addressed to Víctor Sánchez-Mendieta; [vsanchezm@uaemex.mx](mailto:vsanchezm@uaemex.mx)

Received 14 February 2013; Revised 22 April 2013; Accepted 9 May 2013

Academic Editor: Carmina Menchaca-Campos

Copyright © 2013 Raúl A. Morales-Luckie et al. This is an open access article distributed under the Creative Commons Attribution License, which permits unrestricted use, distribution, and reproduction in any medium, provided the original work is properly cited.

Silver nanoparticles were synthesized and supported on thin nylon membranes by means of a simple method of impregnation and chemical reduction of Ag ions at ambient conditions. Particles of less than 10 nm were obtained using this methodology, in which the nylon fibers behave as constrained nanoreactors. Pores on nylon fibres along with oxygen and nitrogen from amide moieties in nylon provide effective sites for *in situ* reduction of silver ions and for the formation and stabilization of Ag nanoparticles. Transmission electron microscopy (TEM) analysis showed that silver nanoparticles are well dispersed throughout the nylon fibers. Furthermore, an interaction between nitrogen of amides moieties of nylon-6,6 and silver nanoparticles has been found by X-ray photoelectron spectroscopy (XPS).

## 1. Introduction

Research on the synthesis of mesoporous materials containing nanoparticles represents a fast-developing area of nanoscience and nanotechnology. This interest is stimulated by several possible application areas of these materials including catalytic [1], magnetic [2], and optoelectronic [3, 4]. Metal nanoparticles dispersed in polymeric matrixes have recently been the subject of intense study aiming to develop nanocomposite films [5–8]. General approaches for the synthesis and support of nanoparticles inside porous materials include impregnation [9] and deposition-precipitation [10]. A drawback of this nanocomposites is the difficulty to disperse nanoparticles in most systems [11]; hence one potential advantage of such metal/polymer systems is that the size and distribution of dispersed metal nanoparticles can be readily controlled based on the properties of the host polymer [12, 13].

Silver particles with a narrow size distribution have been produced upon reversible chemical transformation between metallic and oxide states in a titania matrix [14] and in a mesoporous silica which was grafted with hydrophobic  $-\text{Si}(\text{CH}_3)_3$  groups at the pore surface [15].

Nylon is an electron-rich and polar synthetic polymer (polyamide) usually made from the monomers adipoyl chloride and hexamethylene diamine to form a linear molecular chain (Figure 1). Synthetic nylon membranes have a porous structure [16] and are composed of microfibrils that are interconnected forming a three-dimensional network. Such morphological features provide a unique reaction vessel for synthesizing and supporting metal nanoparticles, allowing enhanced access of guest molecules to catalytic centres, compared with nonporous films.

In this work, a facile synthesis of silver nanoparticles of less than 10 nm in diameter with a narrow size distribution,



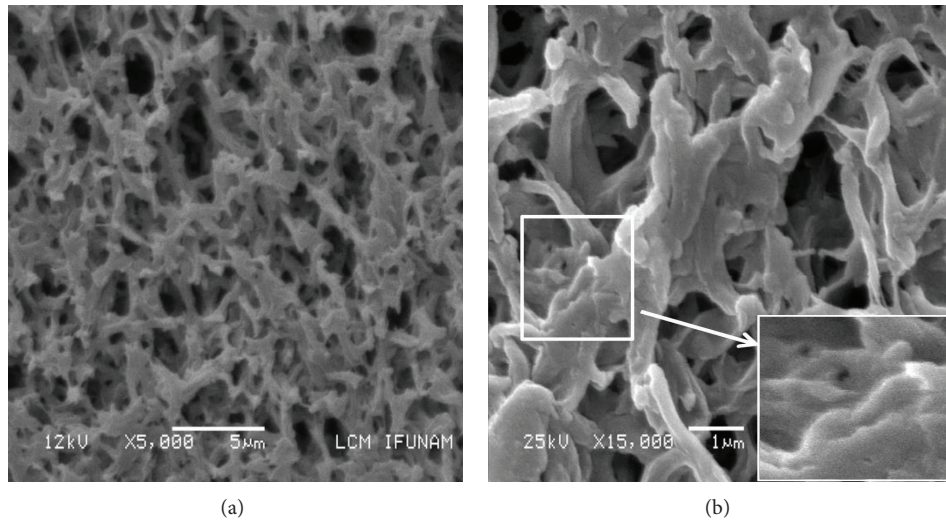


FIGURE 2: SEM image of the nylon-6,6 membrane.

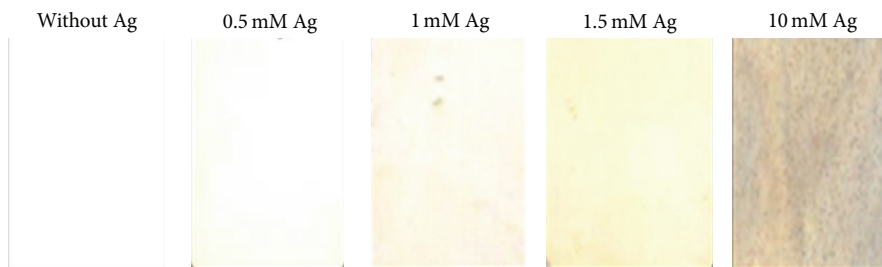


FIGURE 3: Nylon-6,6 without and with Ag nanoparticles formed at different AgNO<sub>3</sub> concentrations.

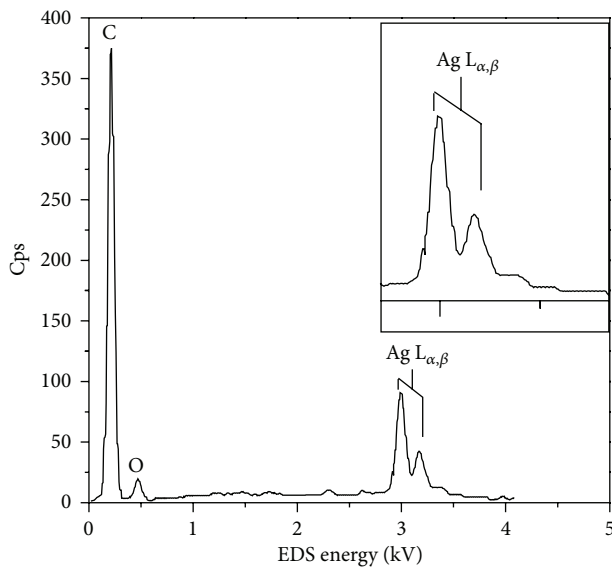


FIGURE 4: EDX spectrum of nylon-6,6 membrane coated with Ag nanoparticles.

located at 390 nm (Figure 5(a)); this is attributable to the surface plasmon resonance of silver nanoparticles in agreement with the observed yellow colour showed in Figure 3 [19]. No absorption was observed at wavelengths longer than 450 nm. These observations imply that Ag nanoparticles were formed. The surface plasmon peak underwent a shift to

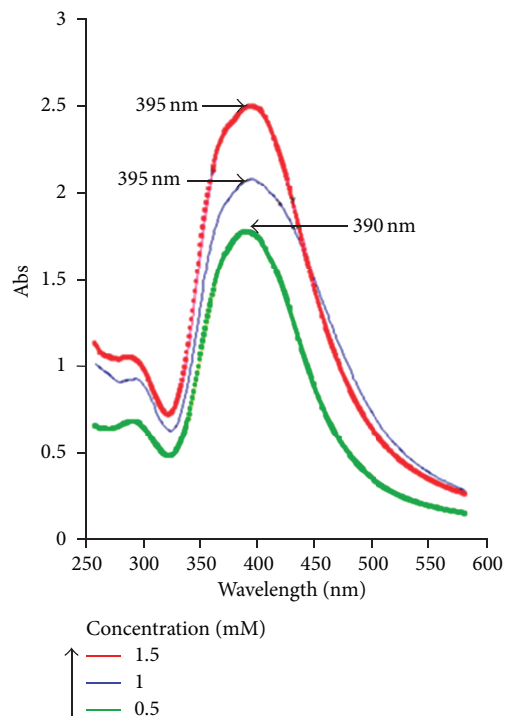


FIGURE 5: UV-Vis spectra of silver nanoparticles in nylon-6,6 membranes. Nanoparticles were prepared using aqueous AgNO<sub>3</sub> at 0.5 mM, 1.0 mM, and 1.5 mM, respectively.

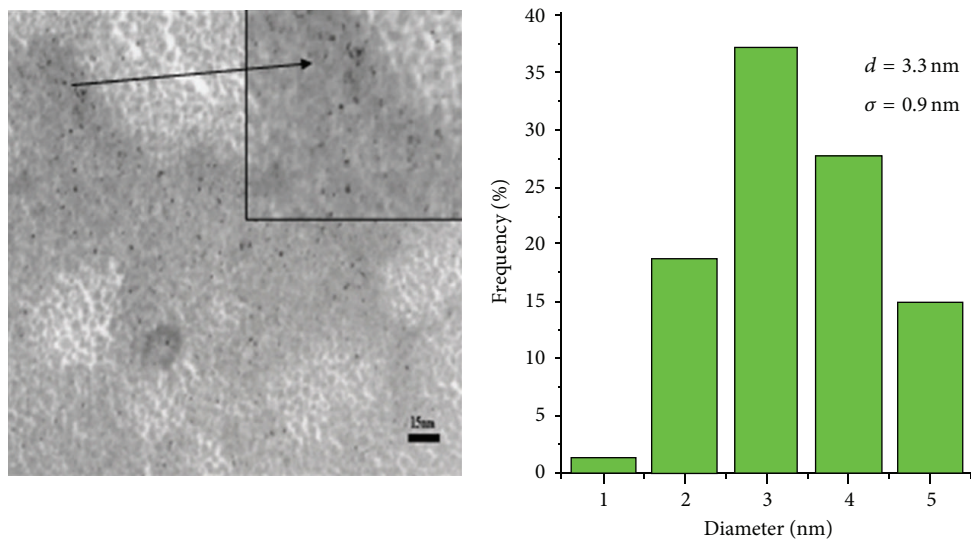
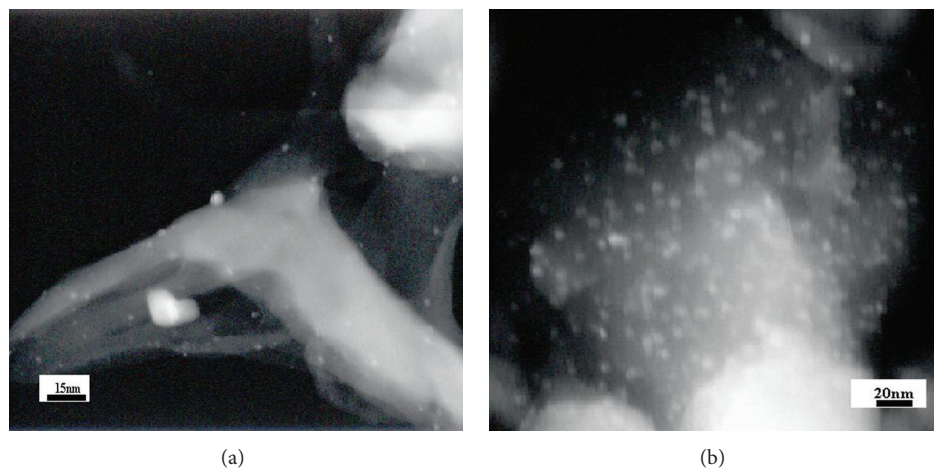
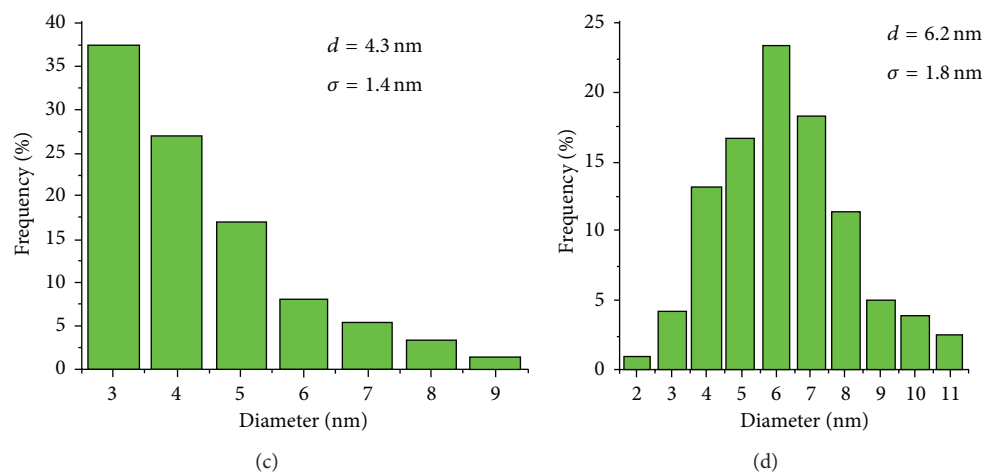


FIGURE 6: TEM micrograph and size distribution histogram of Ag nanoparticles in nylon-6,6 membranes ( $\text{AgNO}_3$  0.05 mM).



(a)

(b)



(c)

(d)

FIGURE 7: HAADF images and size distribution of Ag nanoparticles obtained from (a), (c)  $\text{AgNO}_3$  1.0 mM; (b), (d)  $\text{AgNO}_3$  1.5 mM.



395 nm and was slightly broadened when 1 mM  $\text{AgNO}_3$  was employed (Figure 5(b)). When 1.5 mM  $\text{AgNO}_3$  was used, the surface plasmon absorption becomes broader (Figure 5(c)). It is noted that the absorption intensity of the plasmon band increases as the concentration of aqueous  $\text{AgNO}_3$  augments, also in agreement with the membrane intensity changes in colour, which can be attributed to the increase in the Ag nanoparticles concentration.

These assumptions are confirmed by TEM observations. As shown in Figure 6 monodisperse Ag nanoparticles were obtained at  $\text{AgNO}_3$  0.5 mM. Their mean diameter ( $d$ ) was measured as 3.3 nm with a standard deviation ( $\sigma$ ) of 0.9 nm.

Figure 7 shows Z-contrast images of nylon fibres with Ag particles. Figures 7(a) and 7(c) correspond to a concentration of 1.0 mM of  $\text{AgNO}_3$  resulting in nanoparticles with an average size of 4.3 nm with a standard deviation of 1.4 nm. Using a solution of 1.5 mM  $\text{AgNO}_3$  (Figures 7(b) and 7(d)), the average particle size slightly increases to 6.2 nm with a standard deviation of 1.8 nm. Therefore, it is possible to control the size and size distribution by adjusting the concentration of metal ions in solution.

Because of the porous structure of nylon fibre and the strong interactions between  $\text{Ag}^+$  ions and the carbonyl and amide groups of nylon macromolecule,  $\text{Ag}^+$  ions were uniformly and tightly anchored to the nylon fibres [20]. Such interactions would lower the mobility of  $\text{Ag}^+$  ions, enhance the formation of silver nuclei, limit the formation of several morphologies, and prevent the growth of larger particles [21]. This is particularly true at low  $\text{Ag}^+$  ion concentrations and can explain the formation of monodisperse Ag nanoparticles (after  $\text{NaBH}_4$  reduction) under such conditions. At higher  $\text{AgNO}_3$  concentrations, larger amounts of  $\text{Ag}^+$  ions are embedded on nylon membranes, leading to large and widely distributed particles after reduction [22]. In these kinds of systems, it is expected that the carbonyl and amide groups may also play an important role in stabilization of metal nanoparticles in addition to the porous structure of nylon fibres.

The HRTEM image in Figure 8 shows Ag nanoparticles between 2 and 3 nm in diameter. This image possesses atomic resolution; therefore, several crystalline planes are distinguishable and the interplanar distances can be measured. The interplanar distances shown in the micrograph were measured from the Fourier transforms (FFT) of these nanoparticles (Figure 8, bottom), where the corresponding crystalline planes are specified. The interplanar distances and their corresponding crystalline planes match the ones of metallic Ag (FCC) phase. The measured interplanar distance is 2.36 Å and corresponds to the (111) plane [23].

In order to examine the chemical composition of the nylon-6,6/Ag nanoparticles fibre composite, as well as the possible interaction of silver metal with the nylon moieties, after formation of Ag nanoparticles, X-ray photoelectronic spectroscopy (XPS) was used. Figure 9 shows the XPS survey spectra obtained after  $\text{Ar}^+$  etching for 10 s.

Figure 9 shows the increase in the Ag 3d signal corresponding to the different concentrations of  $\text{AgNO}_3$  added in the solution reaction; a spectrum of nylon-6,6 is used as reference. Clearly an increase of the concentration of

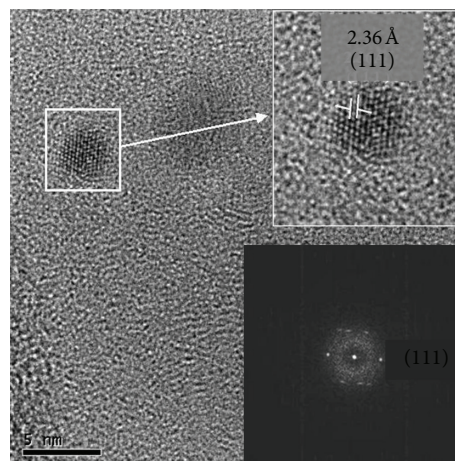


FIGURE 8: HRTEM image of Ag nanoparticles (top) and their corresponding Fourier transforms (FFT) (bottom) pattern.

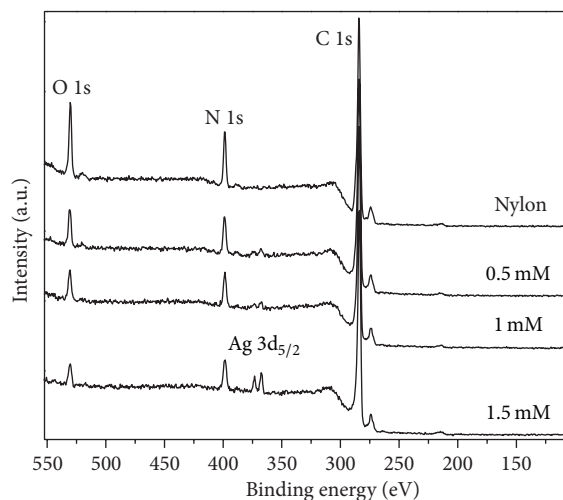


FIGURE 9: XPS spectra of nylon-6,6 and three different nylon-6,6/Ag composites.

the metal precursor salt favours the formation of the nylon/Ag composite.

In order to obtain more information about the chemical state of the Ag nanoparticles present in the composite a curve fit of the various signals was made and is shown in Figure 10.

Table 1 shows the binding energies of the XPS deconvolution of Ag 3d, C 1s, O 1s, and N 1s core levels samples with the Ag metallic, AgO, and nylon-6,6 as references. The peak energy position in the deconvolution and number of peaks were based in data reported by Beamson and Briggs [24] and Hoflund and Hazos [25] and calibrated with metallic Ag foil as reference.

From Table 1, a positive chemical shift in N 1s core level and negative chemical shift in Ag 3d<sub>5/2</sub> of Ag(0) of nanoparticles, with respect to N 1s and Ag 3d<sub>5/2</sub> of nylon and Ag metallic, are observed. From the data in Table 1 the general rule based on the electronegativity can explain both the positive chemical shift of the N 1s core level and the negative chemical shift of the Ag 3d<sub>5/2</sub> core level. This effect can be seen

TABLE I: The binding energy position by XPS of nylon-6,6- and Ag-containing samples core levels Ag 3d, C 1s, O 1s, and N 1s.

Sample	N 1s	O 1s		Ag 3d <sub>5/2</sub>		*C 1s			
		O=C	O-Ag	Ag	AgO (eV)	1	2	3	4
References*	398.69	530.33		368.21					
0.5 mM	398.91	530.33	531.30						
1.0 mM	398.80	530.33	531.30	368.20	367.31	283.84	284.14	285.00	286.81
1.5 mM	398.75	530.33	531.30						

\*References: nylon and metallic Ag.

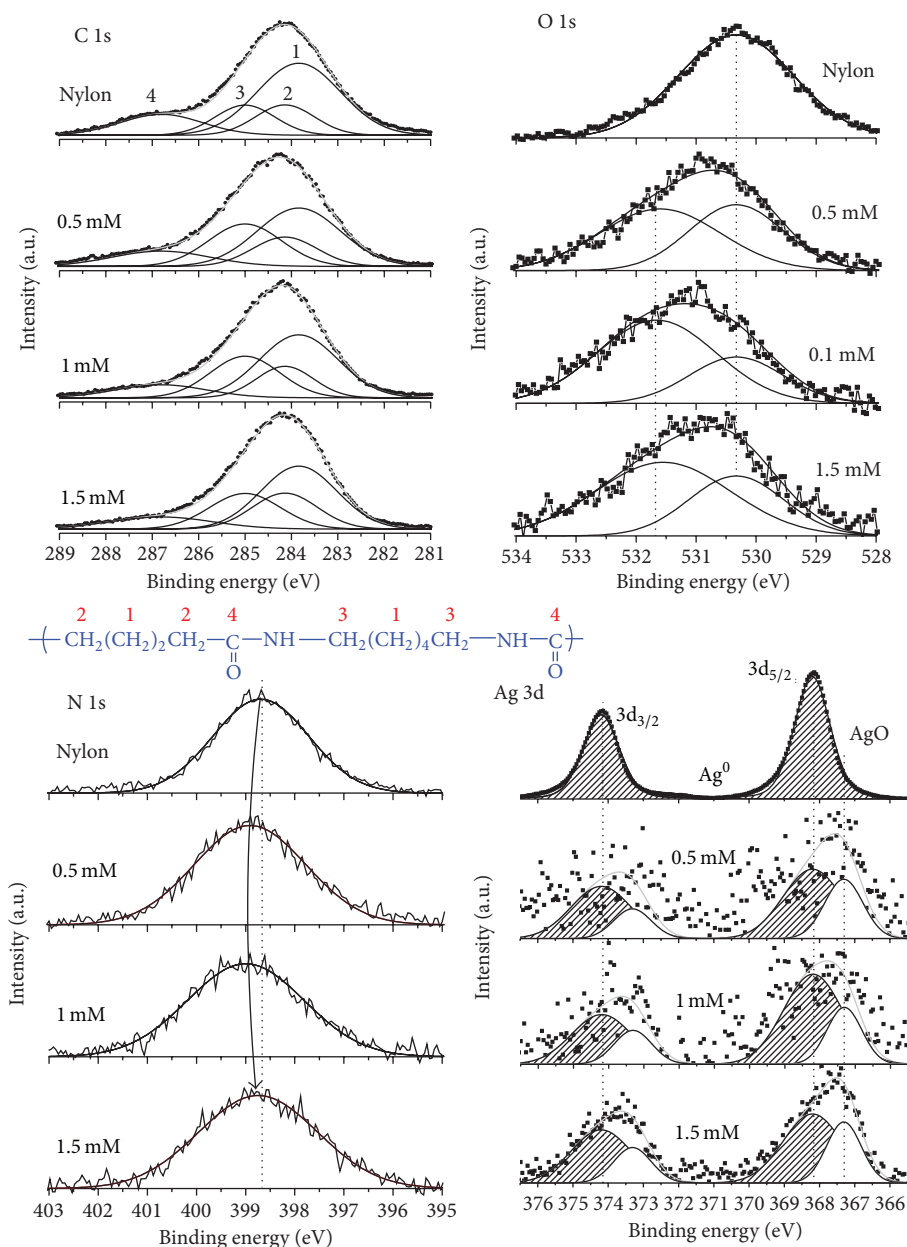


FIGURE 10: The XPS deconvolution for C 1s, O 1s, N 1s, and Ag 3d core levels.

in Figure 10 for N 1s. For the composite samples analysed, all C 1s, O 1s, and Ag 3d<sub>5/2</sub> of AgO core levels are in the same positions as exhibited in Table I and Figure 10. O 1s (530.33 eV) corresponds to O=C and Ag-O at 367.31 eV; C 1s corresponds to (1) C-CH<sub>2</sub> at 283.84 eV, (2) CH<sub>2</sub>-CH<sub>2</sub> at

284.14 eV, (3) CH<sub>2</sub>-N at 285 eV, and (4) C=O at 286.81 eV. It can be also observed that for N 1s the maximum shifting ( $\Delta$ ) occurs for the 0.5 mM simple with  $\Delta = 0.22$  eV; however, for 0.1 mM sample,  $\Delta = 0.11$ , it contains the largest amount of Ag nanoparticles, taking in account the amount

of AgO. For the 1.5 mM sample, Ag 3d peaks can be deconvoluted satisfactorily obtaining thus all oxidation states even in the samples with less Ag concentration. In addition, in Figure 10, the data obtained from the curve fitting show that there is no change in the binding energy of the orbitals of C 1s despite the use of different AgNO<sub>3</sub> concentrations. There is, however, a 0.31 eV shift in the N 1s peak corresponding to the sample with 1.5 mM AgNO<sub>3</sub>. This small shift in binding energy suggests there is an interaction of the nitrogen atoms in nylon during the stabilization of the silver nanoparticles. A similar chemical shift for N 1s binding energy in XPS spectra has been observed for the interactions of PVP with Pt nanoparticles and nitrocellulose with Ru [26, 27]. Hence, pores of nylon fibers, where the nitrogen atoms from the amide groups are found, not only interact with the metal ions presumably through ion-dipole forces, but once the reduction reaction occurs, they also stabilize the Ag nanoparticles. Analysis of the O 1s and Ag 3d<sub>5/2</sub> peaks shows the presence of AgO, which was expected due to the high reactivity of the silver nanoparticles at the sizes obtained. Nevertheless, the fact that there is still a signal corresponding to Ag<sup>0</sup> suggests that the oxidation takes place only on the surface of the nanoparticles, while metallic silver remains at their core; again this is possible due to the stabilizing effect of the amide groups present in the nylon fibers. This nylon-6,6/Ag nanoparticles composite will be probed in dyes removal from aqueous solutions and in applications regarding its antibacterial properties.

#### 4. Conclusions

It was demonstrated that using an aqueous Ag ion impregnation of nylon fibres followed by a reduction with NaBH<sub>4</sub>, a composite of Ag nanoparticles attached to the polymer can be formed. The whole process is carried out at ambient conditions. SEM, HRTEM, and XPS studies confirmed the presence of such Ag nanoparticles in the fibres, with an average size of 3.3 nm. This very simple and versatile synthetic route could be applied to obtain other composites made of metal nanoparticles and natural or synthetic polymer fibres. Moreover, an interaction between nitrogen of amides moieties of nylon-6,6 and silver nanoparticles has been found by X-ray photoelectron spectroscopy.

#### Acknowledgments

Financial support of this work was provided by Universidad Autónoma del Estado de México (UAEM) (Project no. 3246/2012U). The authors are indebted to Fernando Ureña (ININ), Samuel Tehuacanero (IFUNAM) for their assistance in SEM and image digitalisation, respectively.

#### References

- [1] J. Y. Ying, C. P. Mehert, and M. S. Wong, "Synthesis and applications of supramolecular-templated mesoporous materials," *Angewandte Chemie*, vol. 38, no. 1-2, pp. 56-77, 1999.
- [2] M. S. Morey, J. D. Bryan, S. Schartz, and G. D. Stucky, "Pore surface functionalization of MCM-48 mesoporous silica with tungsten and molybdenum metal centers: perspectives on catalytic peroxide activation," *Chemistry of Materials*, vol. 12, no. 11, pp. 3435-3444, 2000.
- [3] J. Liu, X. Feng, G. E. Fryxell, L. Wang, A. Y. Kim, and M. Gong, "Hybrid mesoporous materials with functionalized monolayers," *Advanced Materials*, vol. 10, no. 2, pp. 161-165, 1998.
- [4] J. J. Schneider, N. Czap, J. Hagen et al., "Metallorganic routes to nanoscale iron and titanium oxide particles encapsulated in mesoporous alumina: formation, physical properties, and chemical reactivity," *Chemistry*, vol. 6, no. 23, pp. 4305-4321, 2000.
- [5] J. S. Jung, W. S. Chae, R. A. McIntyre, C. T. Selp, J. B. Wiley, and C. J. O'Connor, "Preparation and characterization of Ni nanoparticles in an MCM mesoporous material," *Materials Research Bulletin*, vol. 34, no. 9, pp. 1353-1360, 1999.
- [6] K. Moller and T. Bein, "Inclusion chemistry in periodic mesoporous hosts," *Chemistry of Materials*, vol. 10, no. 10, pp. 2950-2963, 1998.
- [7] H. Shi, L. Zhang, and W. Cal, "Composition modulation of optical absorption in Ag<sub>x</sub>Au<sub>1-x</sub> alloy nanocrystals *in situ* formed within pores of mesoporous silica," *Journal of Applied Physics*, vol. 87, no. 3, p. 1572, 2000.
- [8] K. W. Powers and L. L. Hench, "The effect of pore size on metal cluster formation in silica sol gel monoliths," *Ceramic Transactions*, vol. 101, p. 253, 2000.
- [9] Y. Plyuto, J. M. Berquer, C. Jaquiod, and C. Ricolleau, "Ag nanoparticles synthesised in template-structured mesoporous silica films on a glass substrate," *Chemical Communications*, no. 17, pp. 1653-1654, 1999.
- [10] W. H. Zhang, J. L. Shi, L. Z. Wang, and D. S. Yan, "Preparation and characterization of ZnO clusters inside mesoporous silica," *Chemistry of Materials*, vol. 12, no. 5, pp. 1408-1413, 2000.
- [11] B. Lebeau, C. E. Fowler, S. Mann, C. Farcet, B. Charleux, and C. Sanchez, "Synthesis of hierarchically ordered dye-functionalised mesoporous silica with macroporous architecture by dual templating," *Journal of Materials Chemistry*, vol. 10, no. 9, pp. 2105-2108, 2000.
- [12] L. Zhang, T. Sun, and J. Y. Ying, "Oxidation catalysis over functionalized metalloporphyrins fixated within ultralarge-pore transition metal-doped silicate supports," *Chemical Communications*, no. 12, pp. 1103-1104, 1999.
- [13] V. L. Colvin, M. C. Schlamp, and A. P. Alivisatos, "Light-emitting diodes made from cadmium selenide nanocrystals and a semiconducting polymer," *Nature*, vol. 370, no. 6488, pp. 354-357, 1994.
- [14] B. O. Dabbousi, M. G. Bawendi, O. Onitsuka, and M. F. Rubner, "Electroluminescence from CdSe quantum-dot/polymer composites," *Applied Physics Letters*, vol. 99, p. 13834, 1995.
- [15] R. Lamber, S. Wetjen, G. Schulz-Ekloff, and A. Baalmann, "Metal clusters in plasma polymer matrices: gold clusters," *Journal of Physical Chemistry*, vol. 99, no. 38, pp. 13834-13838, 1995.
- [16] L. L. Beecroft and C. K. Ober, "Nanocomposite materials for optical applications," *Chemistry of Materials*, vol. 9, no. 6, pp. 1302-1317, 1997.
- [17] K. Akamatsu, N. Tsuboi, Y. Hatakenaka, and S. Deki, "In situ spectroscopic and microscopic study on dispersion of Ag nanoparticles in polymer thin films," *Journal of Physical Chemistry B*, vol. 104, no. 44, pp. 10168-10173, 2000.
- [18] SDP v4. 1 (32 bit) Copyright 2004, XPS International, LLC, Compiled January 17, 2004.

- [19] J. He, I. Ichinose, T. Kunitake, and A. Nakao, "Reversible conversion of nanoparticles of metallic silver and silver oxide in ultrathin  $\text{TiO}_2$  films: a chemical transformation in nano-space," *Chemical Communications*, no. 17, pp. 1910–1911, 2002.
- [20] S. Besson, T. Gacoin, C. Ricolleau, and J. Boilot, "Silver nanoparticle growth in 3D-hexagonal mesoporous silica films," *Chemical Communications*, vol. 9, no. 3, pp. 360–361, 2003.
- [21] T. M. Fahmy, P. M. Fong, A. Goyal, and W. M. Saltzman, "Targeted for drug delivery," *Nanotoday*, vol. 8, no. 8, pp. 18–26, 2005.
- [22] J. He, I. Ichinose, T. Kunitake, A. Nakao, Y. Shiraishi, and N. Toshima, "Facile fabrication of Ag-Pd bimetallic nanoparticles in ultrathin  $\text{TiO}_2$ -gel films: nanoparticle morphology and catalytic activity," *Journal of the American Chemical Society*, vol. 125, no. 36, pp. 11034–11040, 2003.
- [23] W. J. de Ruijter, R. Sharma, M. R. McCartney, and D. J. Smith, "Measurement of lattice-fringe vectors from digital HREM images: experimental precision," *Ultramicroscopy*, vol. 57, no. 4, pp. 409–422, 1995.
- [24] G. Beamson and D. Briggs, *High Resolution XPS of Organic Polymers*, John Wiley & Sons, Chichester, UK, 1992.
- [25] G. B. Hoflund and Z. F. Hazos, "Surface characterization study of Ag, AgO, and  $\text{Ag}_2\text{O}$  using x-ray photoelectron spectroscopy and electron energy-loss spectroscopy," *Physical Review B*, vol. 62, no. 16, pp. 11126–11133, 2000.
- [26] L. Qiu, F. Liu, L. Zhao, W. Yang, and J. Yao, "Evidence of a unique electron donor—acceptor property for platinum nanoparticles as studied by XPS," *Langmuir*, vol. 22, no. 10, pp. 4480–4482, 2006.
- [27] V. K. Kaushik, "XPS core level spectra and Auger parameters for some silver compounds," *Journal of Electron Spectroscopy and Related Phenomena*, vol. 56, no. 3, pp. 273–277, 1991.





**Hindawi**

Submit your manuscripts at  
<http://www.hindawi.com>

