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6 Luminescent Oil Soluble Carbon Dots Towards 7 8 9 10 11 White Light Emission: A Spectroscopic Study 12 13 14 15

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36 ABSTRACT Carbon dots (C-dots) are emerging as new emitting nanomaterials for
37 optoelectronics, bioimaging and biosensing, thanks to their high quantum yield (QY),
38 biocompatibility, low toxicity and cost effective sources. Although the origin of their
39 photoluminescence (PL) mechanism (i.e. their strong blue-green emission and excitation
40 dependent fluorescence) is still controversial, it has been demonstrated to depend on the
41 synthetic protocols and experimental conditions, able to modify the surface properties. Here oil-
42 dispersible C-dots, synthesized by carbonization of citric acid in the presence of hexadecylamine
43 in high boiling organic solvent, are thoroughly investigated by systematically controlling the
44 synthetic reaction parameters. Similarly to what found for water soluble C-dots, citric acid in the
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3 presence of amine-containing passivating agents improves the PL emission of C-dots, *via* the
4 formation of molecular fluorescent derivatives alongside the carbonization process. We
5 demonstrate that, at growth temperature of 200°C, such C-dots exhibit an interesting and intense
6 white emission, when excited in the blue region, thus resulting in a biocompatible colloidal white
7 emitting single nano-objects. The incorporation of the nanoparticles in a poly(methyl
8 methacrylate) (PMMA) host matrix, to obtain freestanding nanocomposite films, is demonstrated
9 not to affect the colour point, that still falls in the white colour region of the 1931 CIE diagram.
10 Remarkably, the emission properties are retained even after several months of films exposure to
11 air and sunlight, thus confirming the colour stability of the nanoparticles against ageing.
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25 1. INTRODUCTION

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28 Recently, carbon nanodots (C-dots) have emerged as an effective alternative to the classical
29 inorganic semiconductor quantum dots (QDs) for a wide range of applications, comprising
30 sensors, photovoltaics, catalysis, bioimaging, drug delivery and optoelectronics. Compared to
31 QDs, C-dots combine an intense and tunable photoluminescence (PL) with good
32 biocompatibility, low toxicity, high resistance to photo-bleaching and inexpensive cost.¹
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40 Since their discovery, extensive efforts have been devoted to developing novel and original
41 synthetic approaches and understanding the electronic and physical-chemical properties of the C-
42 dots. In general, both top-down and bottom-up strategies have been exploited to prepare C-dots,
43 offering the latter a wider choice of organic precursors and/or ligand molecules and diverse
44 carbonization conditions.¹⁻³ Currently, several methods have been proposed to obtain water-
45 dispersible C-dots, being generally based on hydrothermal reactions between an organic
46 precursor bringing carboxylic moieties as carbon source, typically citric acid (CA), and a variety
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3 of amines (e.g. ethanol-amine, ethylenediamine) as nitrogen source and surface passivating
4 agents. The formation of the C-dots has been generally described as a sequence of steps denoted
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6 as poly-condensation and thermal pyrolysis or carbonization. The first step has been reported to
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8 produce amide derivatives generated from reaction between the carboxylic and amine groups that
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10 upon carbonization induce the formation of water-dispersible and highly emitting C-dots.
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12 Although quite simple, the water phase strategies suffer from the undesired formation of larger
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14 carbon structures induced by the hydrophilic surface functional groups as aggregation sites, and
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16 the resulting very broad size distribution requires post-synthetic purification (by means of
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18 filtration, dialysis, centrifugation, column chromatography and gel electrophoresis) to narrow the
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20 size distribution of C-dots and thus improving their properties.⁴⁻⁵ Meanwhile, the hydrophilicity
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22 of such C-dots hinders their use in some specific applications, such as the fabrication of LEDs or
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24 optoelectronic devices and sensors, vulnerable to water exposure. Conversely, attracting
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26 perspectives can be offered by the synthesis of oil-soluble C-dots,⁶⁻⁸ able to provide a precise
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28 control on size and emission, with a high QY. However, since one of the first synthesis proposed
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30 by Liu and co-authors,⁹ only few studies have reported the preparation of C-dots in non-
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32 coordinating high boiling point organic solvents.⁶⁻⁸ The mechanism proposed by Liu and co-
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34 authors⁹ relies on a hot-injection approach, similar to that used for colloidal inorganic QDs.
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36 Highly luminescent oil-soluble C-dots have been prepared by quickly injecting anhydrous CA as
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38 carbon source into a heated mixture of octadecene (ODE) and 1-hexadecylamine (HDA), acting
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40 as non-coordinating solvent and surface passivating agent, respectively. This simple synthetic
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42 approach results in C-dots with multicolored PL emission, high luminescence efficiency optical
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44 properties dependent on reaction time. The reproducibility of this synthetic strategy facilitates
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46 fundamental studies and practical application of these oil soluble C-dots.
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3 The still debated origin and nature of C-dot optical properties and PL emission, in particular,
4 have inspired a number of studies¹⁰⁻¹⁴ since the deep and comprehensive understanding of the
5 fluorescence mechanisms is decisive to fully exploit their potential for technological
6 applications. A general consensus from various reports highlights two main contributions to the
7 radiative emission of C-dots whose intensity and peak position depend on the excitation
8 wavelength: i) a contribution from the intrinsic carbogenic core, consisting of sp^2 domains and ii)
9 a contribution deriving from the surface states, i.e. emission traps, surface defects, functional
10 groups at C-dot surface.^{13, 15-17}

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12 Also, in hydrothermal synthesis it has been demonstrated that the use of CA, or similar
13 compounds, as carbon source and the concomitant addition of amines induces an enhancement of
14 the luminescence of the prepared nanoparticles (NPs), irrespectively from the synthetic
15 parameters. As an example, an increased PL QY has been detected for water-soluble C-dots
16 obtained by hydrothermal treatment of CA in the presence of amines, which has been reasonably
17 ascribed to the formation of amide-containing molecular fluorophores during the precursor poly-
18 condensation through heating.¹⁵ Molecular fluorescent species can be formed in parallel
19 alongside carbonization, and can notably influence the optical properties of the reaction products.
20 Yang and co-workers¹⁸ isolated, by separation and purification from the reaction mixture, an
21 organic fluorescent derivative, the imidazo[1,2- α]pyridine-7-carboxylic acid, 1,2,3,5-tetrahydro-
22 5-oxo-, IPCA, formed during the hydrothermal synthesis of C-dots from CA with
23 ethylenediamine at 140°C for 10 h. Interestingly, the optical properties of such a fluorophore
24 well match the absorption and emission properties of core and edge states of C-dots, with its QY
25 (nearly 86%) considerably higher than that of the carbon states and its PL characteristic almost
26 independent from the excitation wavelength. Similarly, in a recent paper, Rogach et al.¹⁰ studied

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3 the synthesis of different samples of C-dots obtained by the hydrothermal reaction at 200°C for
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5 5h of CA with three amine ligands. The optical properties of the collected products have been
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7 compared with those of citrazinic acid, firstly recognized in 1884 as the most basic blue
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9 fluorescent molecule, presumably forming alongside the carbonization reaction of CA and amines,
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11 thus confirming the contribution of such a molecular unit on the emission properties of the as
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13 synthesized C-dots. These studies emphasize that when CA is used as carbon source in the
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15 presence of primary bifunctional or easily decomposing amines such as ethylenediamine or
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17 hexamethylene-tetramine, fluorescent molecules are formed, such as IPCA or citrazinic acid,
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19 respectively, mainly connected to the carbon core, and strongly contributing to the optical
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21 properties of the C-dots. Conversely, tertiary amines such as ethanolamine cannot induce the
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23 formation of such amide based fluorophore.
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28 Here the synthesis of oil soluble C-dots prepared by carbonization of CA precursor in hot
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30 coordinating solvent⁹ in the presence of hexadecylamine (HDA) has been investigated. We chose
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32 to take advantage of the tunability and reproducibility of this synthetic procedure, which is less
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34 extensively investigated compared to the more popular water phase strategies, to get insights in
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36 the composition, temporal and temperature dependent- chemical structures and optical properties
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38 of the as-synthesized C-dots. A comprehensive and systematic physical chemical study has been
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40 performed to evaluate the effect of the synthetic parameters on the optical characteristics of the
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42 resulting nanostructures. The influence of HDA, reaction time and temperature on the UV-Vis
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44 absorption and excitation-dependent PL emission and QY have indicated that, at lower reaction
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46 temperature, carbonization extent is low, while highly emissive molecular structures or
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48 aggregates arising from CA and HDA condensation appear to have a dominant effect on the
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50 absorption and emission spectra and QYs. In this case, two emission bands contribute to the
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3 fluorescence for excitation in the range between 390 to 410 nm, that, remarkably, results in an
4 overall intense emission in the white region, according to CIE 1931. This interesting outcome
5 meets the technological requirement and interest devoted to the design of white emitting
6 materials based on NPs and nanocrystals. It is worth to note that the white emission in
7 nanostructured materials is often obtained by mixing semiconductor nanocrystals or fluorophores
8 with complementary emission, which increases the complexity of the fabrication processes,¹⁹⁻²¹
9 or by using single “magic-size” CdSe QDs,²² limited in their practical application due to the
10 toxicity Cd-heavy metal²³⁻²⁵. Here we demonstrated that white emission is obtained from single
11 C-dots purposely synthesized by playing with the growth parameters. Such NPs have great
12 potential as emitting components in the fabrication of safe white emitting diodes (white LEDs),
13 offering a sustainable alternative to inorganic nanocrystals.^{6, 26-29}

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28 Subsequently, the oil-soluble C-dots have been easily incorporated in an organic
29 poly(methylmethacrylate) (PMMA) host matrix resulting in a nanocomposite fabricated as
30 freestanding solid film. The intense white emission is retained in the polymer matrix, ruling out
31 any detrimental effect on the original C-dot fluorescence due to their interaction with the organic
32 host.

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40 Overall, the obtained results confirm the relevance of the proposed synthetic strategy not only
41 for fundamental study on PL mechanisms of oil-soluble C-dots, but also the high versatility and
42 plain processability of as-synthesized samples and their great technological potential in the
43 realization of optoelectronic and photonic devices.

44 45 46 47 48 49 2. EXPERIMENTAL SECTION

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53 **2.1 Chemicals.** All chemicals were used as received, without any further purification or
54 distillation. Anhydrous citric acid, CA ($\geq 97\%$), octadecene, ODE (90%), were purchased by

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3 Aldrich. 1-Hexadecylamine, HDA (98%) was purchased by Fluka. All solvent used were of
4 analytical grade and purchased from Aldrich.
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7 Poly(methyl methacrylate), PMMA, with high MW, was purchased by BDH Chemicals LTD
8 29774.
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11 **2.2 Synthesis of C-dots.** C-dots were synthesized by the thermal carbonization of CA as
12 organic precursor, in non-coordinating solvent (ODE), in the presence of HDA as coordinating
13 agent.⁹ All the preparative steps were performed under nitrogen atmosphere using a Schlenk line,
14 and starting from dried and degassed reactants.
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21 In a typical synthesis, a variable amount (0.8 or 1.5 g) of HDA was dissolved in 35 mL of
22 ODE under vacuum at 110°C for 30 min. Then, the mixture was heated at $T > 153^\circ\text{C}$ under
23 nitrogen flux, and 1 g of CA was quickly injected into the reaction flask under vigorous stirring,
24 at temperature ranging from 200 and 300°C. The heating was kept at that temperature for the
25 desired time. At the end of the reaction, the resulting solution was cooled at room temperature,
26 and C-dots were purified by washing with acetone several times. The resulting C-dots were
27 dispersed in common weakly polar solvents, such as chloroform or toluene.
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33 **2.3 C-dot/PMMA nanocomposite preparation.** C-dot/PMMA nanocomposites were
34 prepared by adding a proper amount (0.175 g) of C-dots to a solution of PMMA in chloroform
35 (0.1 g/mL). The mixture was let under mechanical stirring until a homogeneous dispersion of C-
36 dots in PMMA matrix was achieved. Freestanding films of C-dots in PMMA were then obtained
37 by drop casting 50 μL of the chloroform nanocomposite dispersion onto a properly cleaned glass
38 substrate, and removing the resulting film after the complete evaporation of the solvent.
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51 **2.4 Spectroscopic investigation.** Attenuated Total Reflection Fourier Transform Infrared
52 (ATR-FTIR) spectra were recorded by means of a PerkinElmer Spectrum One Fourier Transform
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3 Infrared spectrometer equipped with both a deuterated triglycine sulfate detector and a three-
4 bounce, 4 mm diameter diamond microprism as internal reflection element. The resolution was 4
5 cm^{-1} . The measurements were performed by dropping aliquots (3–5 μl) of the chloroform
6 solutions of the C-dots, directly onto the upper face of the diamond crystal and the spectra were
7 acquired upon solvent evaporation.
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10 UV-Vis absorption spectra were recorded with a Cary 5000 (Varian) UV/Vis/NIR
11 spectrophotometer. PL spectra were recorded by using a Fluorolog 3 spectrofluorimeter
12 (HORIBA Jobin-Yvon), equipped with double grating excitation and emission monochromators.
13 All optical measurements were performed at room temperature on samples obtained directly
14 from synthesis without any size sorting treatment.
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17 TR-PL measurements were performed by Time-Correlated Single Photon Counting (TCSPC)
18 technique, with a FluoroHub (HORIBA Jobin-Yvon). The samples were excited at 375 nm by a
19 picosecond laser diode (NanoLED 375L) emitting $\tau \approx 80$ ps pulses at a 1 MHz repetition rate. The
20 PL signals were dispersed by a double grating monochromator and detected by a picosecond
21 photon counter (TBX ps Photon Detection Module, HORIBA Jobin-Yvon). The temporal
22 resolution of the experimental set up was ~ 200 ps.
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25 The emission quantum yields and 1931 CIE color point measurements of both solution and
26 solid samples were obtained by means of a “Quanta-phi” integrating sphere coated with
27 Spectralons and mounted in the optical path of the spectrofluorometer, using as an excitation
28 source a 450 W xenon lamp coupled with a double-grating monochromator.
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31 **2.5 Morphological investigation.** Transmission electron microscopy (TEM) imaging was
32 carried out using a JEOL JEM1011 microscope, operating at an accelerating voltage of 100 kV
33 and equipped with a W electron source. The images were acquired using a Gatan Orius
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3 SC1000CCD camera. C-dots were deposited by drop casting onto TEM carbon-coated copper
4 grids. The statistical analysis of the samples was performed by use of a freeware image analysis
5 program. In particular, the average NP size and the percentage relative standard deviation (s%)
6 were calculated for each sample to get information on the NPs size distribution.
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13 3. RESULTS AND DISCUSSION 14 15

16 Luminescent C-dots have been synthesized by a one-step procedure similar to the one reported
17 previously by Liu et al.⁹ consisting in the carbonization of anhydrous citric acid (CA) as
18 precursor, carried out in a mixture of octadecene (ODE) and hexadecylamine (HDA) as non-
19 coordinating solvent and surface ligands. CA, which is abundant in nature, has been chosen as
20 carbon source, since it represents an important starting material for the bottom-up synthesis of C-
21 dots, being involved in fluorescent derivatives that strongly contribute to improve the QY. It may
22 form chemical structures that can evolve from small fluorescent molecules, such as citrazinic
23 acid derivatives, generated from the intramolecular condensation of CA with amines, to polymer
24 clusters and carbon cores alongside carbonization.
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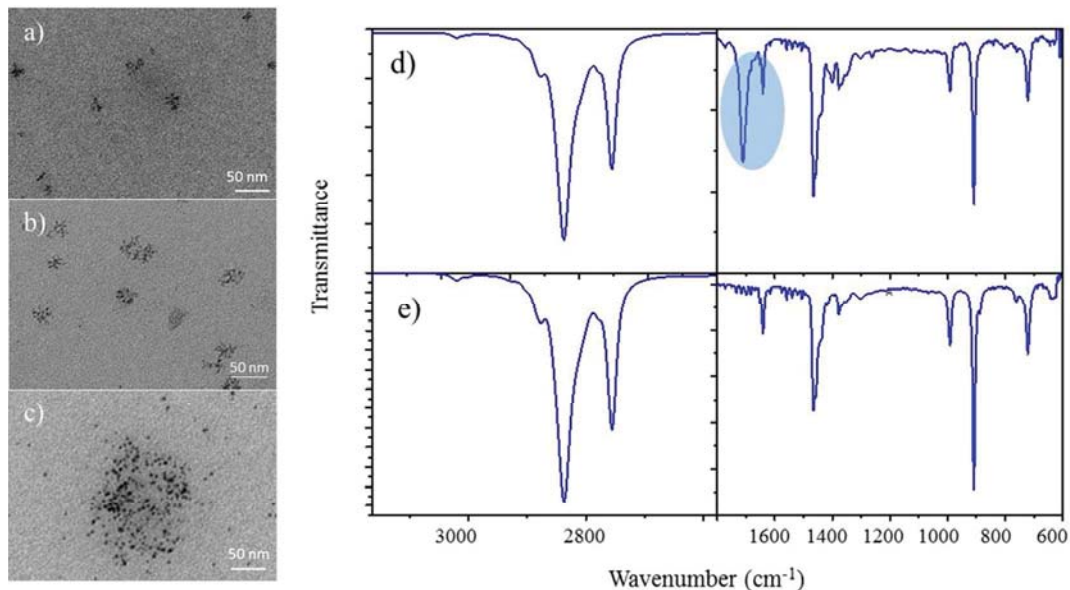
37 Here, the synthetic conditions have been systematically varied to assess the role of the
38 preparative parameters, namely (i) capping agent concentration, (ii) reaction time and (iii)
39 temperature on the optical and morphological properties of the resulting NPs. UV-Vis absorption
40 and excitation dependent emission and QY values have been investigated to get insights in the
41 optical properties and chemical structures of the as-synthesized C-dots.
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50 **3.1 The role of the capping agents.** 51

52 Three C-dot batches have been synthesized by injecting 1 g of CA at 250°C in the reaction
53 mixture without HDA ("bare C-dot") and with 0.8 and 1.5 g of HDA ("HDA-added C-dots"),
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3 setting the reaction growth temperature at 300°C and reaction time at 30 minutes. The TEM
4 investigation, performed after reaction mixture purification, reported in Figure 1 (a-c), shows
5 small dark spots less than 5 nm ascribed to C-dots with a spherical shape, whose size is not
6 significantly affected by the amount of added HDA (Figure 1 a-b)
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12 FT-IR measurements (Figure 1d-e) have been performed, which generally provide information
13 on the functional groups of the nanostructures,³⁰ and here is useful to predict their chemical
14 structures of the as-synthesized type of C-dots. FT-IR spectrum (Figure 1d) of C-dots
15 synthesized in the presence of HDA (irrespectively of the HDA amount added) shows a sharp
16 peak at 1690 cm⁻¹, ascribed to the amide carbonyl stretching vibration, probing that carbonyl
17 groups of the CA were converted into amide group during the synthesis.^{20, 31-32}
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Figure 1. TEM micrographs (a, b, c) and FTIR spectra (d, e) of C-dots samples prepared with
HDA 1.5 g (a, d) and 0.8 g (b) and bare C-dots (c, e).

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Being completely absent in the bare C-dots sample, we can undoubtedly confirm that under the
synthetic conditions, amide structures were formed thanks to conjugation between primary amine
HDA and carboxylic groups of CA.

UV-vis absorbance spectra of all the three batches of C-dots show an intense absorption in the UV region, below 300 nm (Figure 2a), with an evident, broad peak at around 350 nm for HDA-added C-dots. Such a band becomes more intense and symmetric as the HDA amount added to the reaction increases from 0.8 g to 1.5 g, with a concomitant appearance of an absorption tail at lower energies, at wavelength longer than 400 nm.

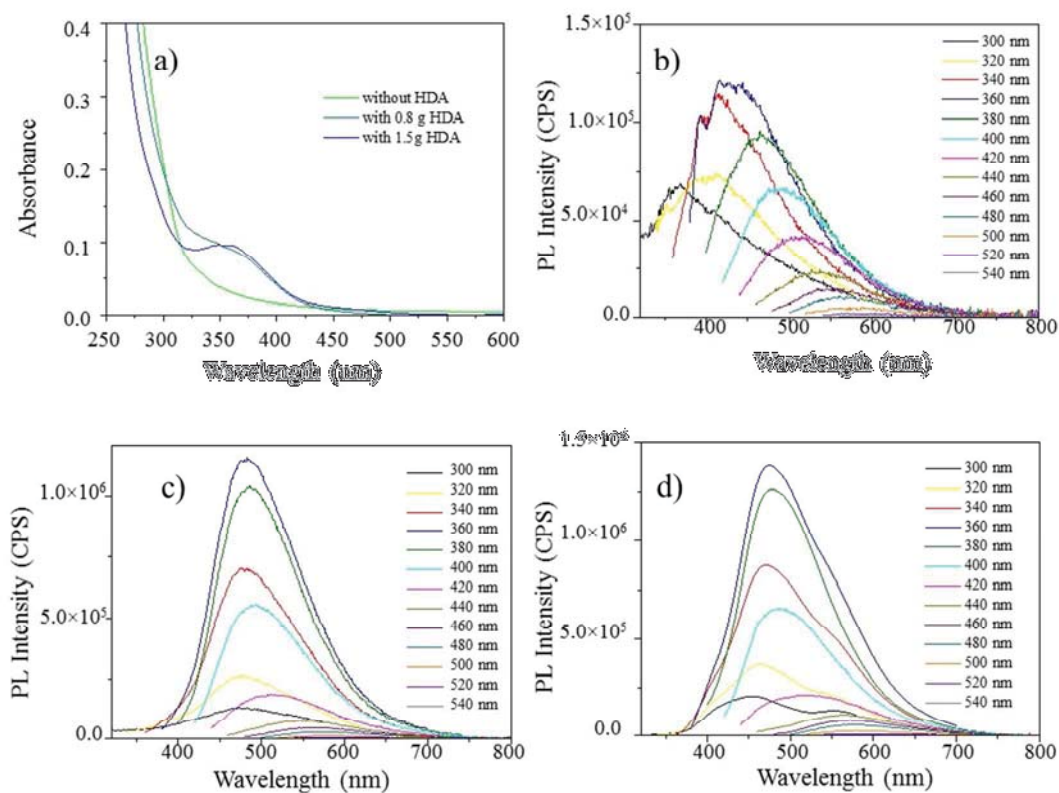


Figure 2. (a) UV-vis absorbance spectra of chloroform dispersion of C-dots synthesized without (green line) or with (blue and cyan line) hexadecylamine (HDA) capping agent. PL spectra of chloroform dispersion of C-dots synthesized without HDA (b), and in the presence of 0.8 g (c) and 1.5 g (d) of HDA, respectively, recorded at increasing excitation wavelength.

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3 As widely reported in literature, the intense absorption below 300 nm can be safely ascribed to
4 the π - π^* transition of the sp^2 domain originating by the conjugated C=C double bonds of the
5 carbon core,³¹⁻³³ while the signal at 350 nm can be attributed to the n - π^* transitions of the C=O
6 bond of the carbonyl and of the amide functional groups at the C-dots surface, being only
7 detected in HDA-added samples and completely absent in the absorption spectrum of bare C-
8 dots. The extended absorption tail, above 400 nm, has been assigned to the surface state sub-
9 bands absorbing at lower energy with respect to the n - π^* transitions.¹⁰ It is worth to note that, in
10 spite of different reaction conditions, the absorption features of HDA added C-dots strongly
11 resemble to those found by Rogach et al.¹⁰ for hydrothermally synthesized water soluble C-dots,
12 prepared by condensation of CA and ethylenediamine. In that case the high absorption below 300
13 nm, generally ascribed to the carbogenic core, was attributed also to π - π^* transitions of the
14 molecular fluorophores formed during carbonization process as citrazinic acid or its derivatives,
15 as IPCA. In addition, the broad peak around 340 nm was ascribed to n - π^* transitions of organic
16 fluorophores. Moreover, a recent investigation reveals that such an absorption band can be due
17 to the twofold contribution of the n - π^* transitions and π - π^* charge transfer involving aggregation
18 of surface or molecular states within or among the C-dots.¹¹

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41 Wide and asymmetric emission bands characterize the PL spectra of the three as synthesized
42 C-dots samples reported in Figure 2b-d. Emission has been recorded at different excitation
43 wavelengths in the range between 300-500 nm, in order to depict the contribution of the
44 carbogenic core, surface related states and/or the organic fluorophores potentially forming during
45 the reaction.^{10, 18} It is worth to note that both carbon core and edge state emission result
46 excitation dependent, while the emission of the fluorescent molecular derivatives is expected to
47 be independent from the excitation wavelength. The trend of PL peak wavelength versus
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excitation wavelength in Figure 3a clearly indicates a tight dependence of the emission peak from the excitation for both the bare C-dots (green line in Figure 3a), and the HDA added C-dots (blue and cyan line in Figure 3a), irrespectively from the amount of HDA used during the reaction. However, while for the bare C dots, the PL peak centered at 367 nm for λ_{ex} of 300 nm, continuously red shifts by increasing excitation wavelength, the HDA added C-dots show a PL peak, whose position remains almost unchanged at nearly 470 nm for $\lambda_{\text{ex}} < 400$ nm, and slightly moves to higher value by increasing excitation wavelength up to 582 nm for $\lambda_{\text{ex}} = 500$ nm. Moreover, the diagram reporting PL intensity versus excitation wavelength reported in Figure 3b shows a strong excitation dependent PL intensity for syntheses carried out in presence of HDA, with the highest emission value obtained for excitation at 360 nm.

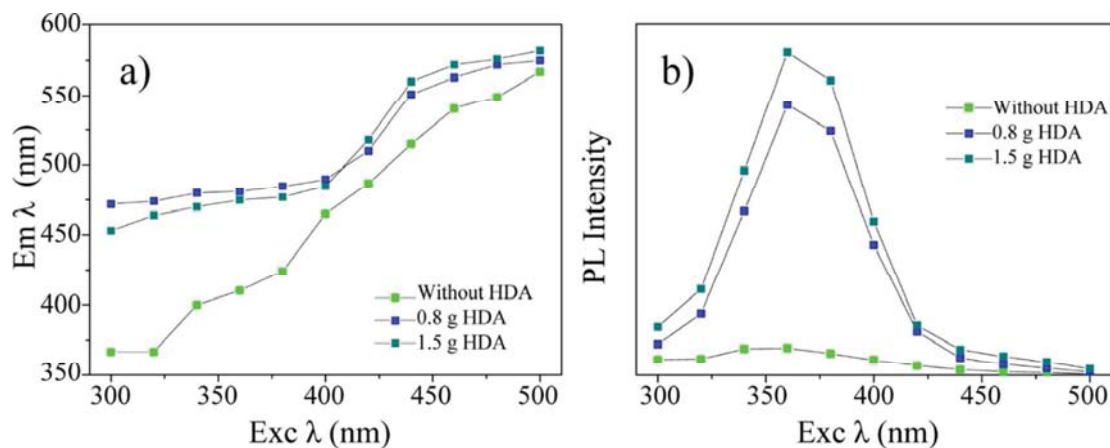


Figure 3. PL peak position (a) and PL intensity (b) as a function of the excitation wavelength for the bare C-dots (green symbols), and for the C-dots synthesized in the presence of 0.8 and 1.5 g of HDA (blue and light blue symbols, respectively).

This trend is in agreement with the absolute PL QY values reported in Table S1: PL QY of bare C-dots has a constant and low (nearly 1) value, irrespectively from the excitation wavelength, while HDA added C-dots show PL QY between 10 and 20% for excitation wavelength below 400 nm. This value drops down to 3% for $\lambda_{\text{ex}} = 420$ nm. Comparing the

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3 emission properties of the samples synthesized without and with HDA suggests the contribution
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5 of different emission process (i.e emission states or emitting species) to the PL behavior.
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8 The observed emission features of oil-soluble C-dots are very similar to what reported in the
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10 hydrothermal synthesis of water soluble C-dots carried out in the presence of amines.¹⁰ In that
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12 case, the detected fluorescence was attributed to the formation of derivatives of citrazinic acid
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14 and to surface states, in the high (<400 nm) and low energy range, respectively.¹⁰ As a molecular
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16 fluorophore, the citrazinic acid has a PL peak that does not change in position (PL peak at nearly
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18 440 nm) with excitation wavelength and has higher PL intensity below 400 nm. In our case,
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20 although the fluorescence cannot be univocally assigned to the occurrence of a specific
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22 fluorophore, it can be safely inferred that most HDA or amino terminated molecules in the
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24 synthesis may induce, through the poly-condensation with carboxylic precursors, the formation
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26 of highly emissive fluorophores that can be accounted for the excitation-independent (for λ_{ex}
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28 below 400 nm) behavior of C-dots prepared in the presence of HDA and to the high PL QY
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30 value.³⁴⁻³⁵ Blank experiment carried out by heating at 300°C for 30 min (and even for prolonged
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32 time) the mixture formed of 1.5 g HDA in ODE without CA has been also carried out. The
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34 spectroscopic characterization (See Supporting Information Figure S1) reveals absorption and
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36 emission features matching well those of the bare C-dots, with a very low emission intensity.
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38 Therefore, the PL of the oil-soluble, HDA added, C-dots can be reasonably assumed to be
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40 originated from molecular fluorophores (i.e. IPCA and the citrazinic acid derivatives^{10, 18}),
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42 arising from the condensation of CA carboxylic moieties and amines groups of HAD, when both
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44 CA and HDA are present in the reaction mixture. Moreover, such blank experiments allow for
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46 highlighting the heterogeneity of the contribution from both carbon core nanoparticles and
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48 fluorophores, as reported by Kumbhakar *et al.*³⁶⁻³⁷.
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To gain information on the recombination dynamics, a time resolved (TR) emission study has been carried out by using a time-correlated single-photon counting (TCSPC) system. The PL decay of bare and HDA added C-dots has been probed at 435 nm, corresponding to the PL maximum, and are reported in Figure 4.

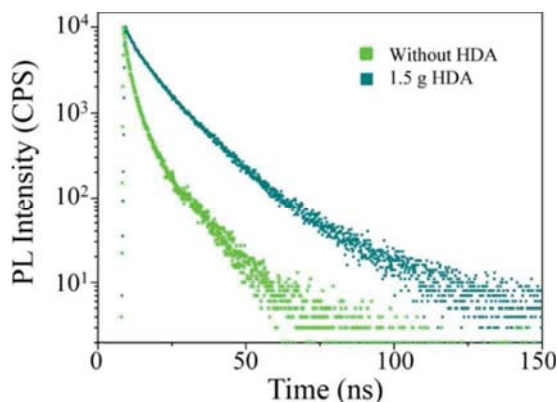


Figure 4. Comparison of time-resolved PL decays of CHCl_3 dispersions of bare C-dots and HDA-capped C-dots (1.5 g), exciting the solutions with a 80 picosecond pulsed light at 375 nm and detecting at 435 nm.

All the PL decay experimental data are best fitted by a bi-exponential function, although fitting ranging from mono-exponential to 4-exponential equations have been reported, depending on the synthetic procedure and on the surface properties of C-dots.^{12, 17, 32, 34} The fitting parameters are reported in Table S2, where A1 and A2 are the pre-exponential factors, τ_1 and τ_2 the decay times. The average lifetime, τ_{Avg} has been calculated according to our previous study³⁸ and reported in Table S2. The recombination dynamics of both the samples show a slow and a fast component, respectively. In bare C-dots, where the carbogenic core mostly contributes to the PL emission, the faster recombination is the dominant component of the decay. Conversely, in HDA-capped C-dots, the slower component prevails. Moreover, the average decay lifetime

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3 increases passing from 4.6 to 12.2 ns for the bare C-dots and HDA-capped C-dots, respectively.
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5 The TR-PL measurements, together with the QY data, indicate that the presence of the
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7 coordinating agent promotes the formation of fluorophores that remarkably contribute to the
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9 radiative PL decay.
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13 **3.2 Effect of the reaction time**

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15 The reaction time has been varied while keeping unchanged the others synthetic conditions. In
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17 particular, 1.5 g of HDA have been added to the high boiling solvent (ODE) in the reaction flask
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19 and after the injection at 250°C of 1 g of the CA precursor, the reaction mixture has been heated
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21 at 300°C for 8, 30 and 180 minutes, respectively. The temporal evolution of absorption and PL
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23 characteristic versus reaction allows to consider the role of chemical composition and particle
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25 size on the optical properties of the formed C-dots.
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29 Figure 5a reports the UV-Vis spectra of chloroform dispersions of C-dots obtained at different
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31 reaction times. Increasing the reaction time from 8 to 30 and 180 minutes, the intensity of the
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33 absorption band at about 350 nm decreases (Figure 5a red, blue and black lines, respectively).
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35 After heating at 300°C for 180 min, only a slight shoulder remains in the same absorption region.
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38 (Figure 5a, black line).
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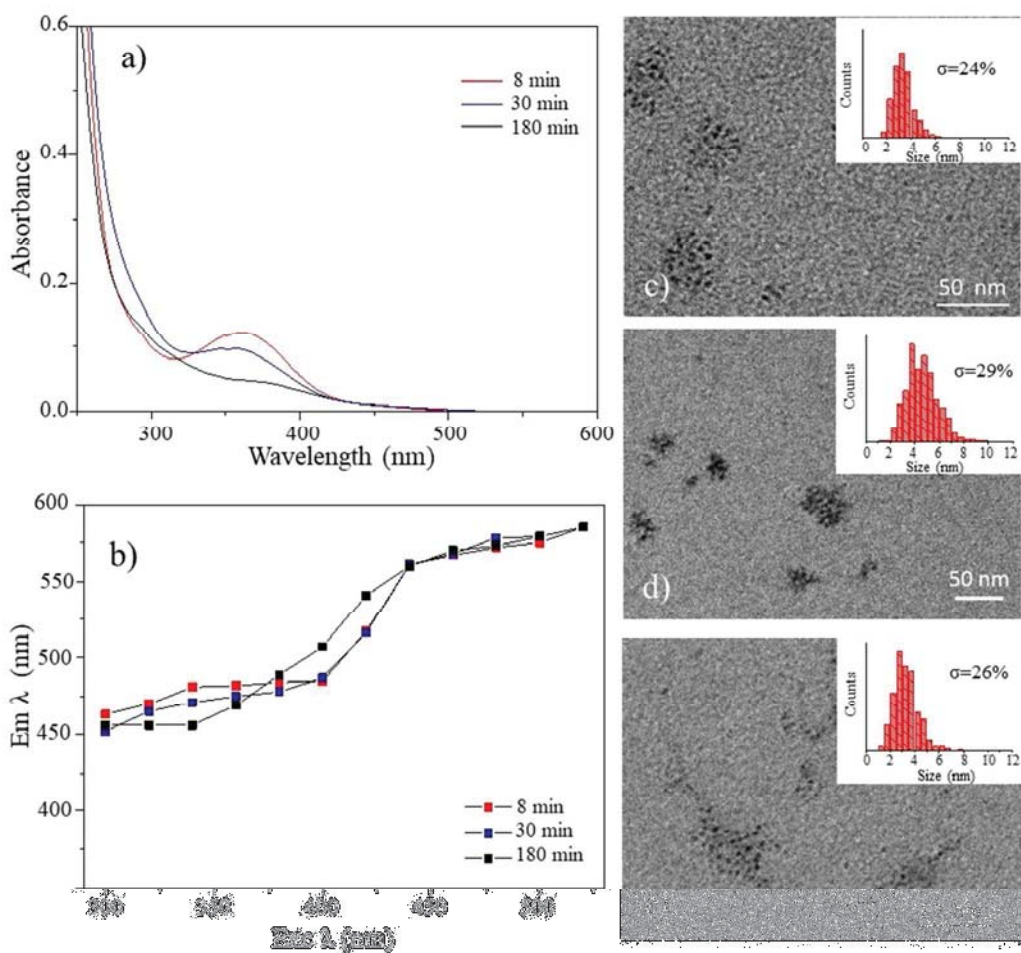


Figure 5. a) UV-vis absorption spectra and b) PL peak position as a function of the excitation wavelength of chloroform dispersion of C-dots carbonized at 300°C for 8, 30 and 180 min; TEM images of C-dots carbonized at 300°C for c) 8, d) 30 and e) 180 min.

TEM micrographs in Figure 5c-e, reveal that, with the reaction proceeding, only a slight modification in the average size (from 3 to 5 nm) and broadening of the size distribution of the spherical NPs can be observed up to 30 minutes of reaction, similarly to what reported by Liu et al.⁹ However, at 180 min a slight focusing of the size distribution is measured (Figure 5 d-e). At different reaction time PL spectra show the typical wide asymmetric fluorescence band that bathochromically shifts with the excitation wavelength (Figure S2 A-C) over distinct regions.¹⁰

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3 The emission wavelength versus excitation wavelength scattered plots reported in Figure 5b for
4 the three samples show a quite similar trend for the samples prepared at the early stage of
5 carbonization (Figure 5b red and blue symbols for 8 and 30 minutes reaction time respectively).
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7 Conversely, at longer reaction time (Figure 5b black symbols for 180 min reaction time), a more
8 pronounced dependence of the PL peak position from the excitation is observed. Also PL
9 intensity sensibly decreases, becoming half with respect to those measured for sample growth
10 time of 30 minutes. These results suggest that, at the early stage of the reaction, molecular
11 fluorophores (free or attached at the C-dots surface) will form alongside carbonization,
12 contributing to the PL properties. However, as the carbonization reaction proceeds these
13 fluorescent derivatives most likely decompose or degrade, becoming less influent in the C-dots
14 PL properties.
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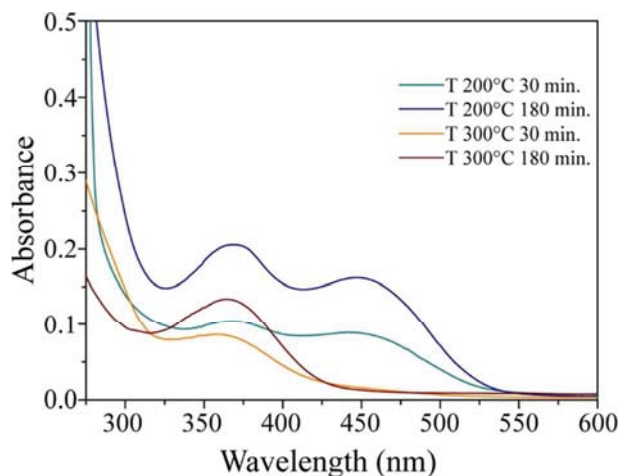
28 Also, absolute PL QYs, measured (excitation wavelength at 360 nm) for all the three samples,
29 provide a clear support for the PL data (Table S3), confirming the hypothesis of molecular
30 fluorophores consumption or degradation during the ongoing of the carbonization reaction. The
31 longer the reaction time, the lower the quantum efficiency, since upon disappearance of the
32 molecular fluorophores the main contribution to the PL remains the less efficient carbon states
33 (core and edge states).
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42 The temporal evolution of the optical and morphological properties versus reaction time,
43 indicates that two concomitant competitive processes can be considered: (i) progress in the
44 carbonization may lead to the formation of carbogenic cores as dominant emissive species: (ii)
45 during the pyrolytic process the molecular and surface states may be modified. Indeed, when the
46 reaction is stopped at the early stage, right after the precursor injection, a large fraction of
47 molecular fluorophores can form, leading to high QY values.
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3.3 Effect of the reaction temperature

Finally, to further investigate the role of the carbonization process in the synthesis of the amino added C-dots, the effect of the reaction temperature has been also studied. In particular, the C-dot synthesis has been performed by keeping constant the CA and HDA amount (1 g and 1.5 g, respectively), and adding the precursor to the non-coordinating solvent heated at 200°C. The carbonization reaction has been carried out at different temperatures (200°C or 300°C), for 30 or 180 min.

UV-vis absorption spectra of samples are displayed in Figure 6. As stated before, at 300°C, regardless the reaction time, only a broad band around 350 nm and a strong absorption in the UV region are detected. Decreasing the growth temperature at 200°C, an additional absorption band at about 450 nm is observed, which becomes more intense at increasing reaction time. According to a recent study,¹¹ the occurrence of such characteristic absorption band can be explained taking into account the possible occurrence of aggregates starting from the molecular fluorophores, resulting in absorption at lower energies. In this regard, FT-IR analysis do not show any significant variation in vibrational signals of C-dots grown at 200°C with respect to the other HDA-capped C-dots heated at higher temperature (data not reported).



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3 **Figure 6.** UV-vis absorption spectra of chloroform dispersion of C-dots as a function of the
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5 reaction temperature at two different reaction times.
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9 The emission spectra of C-dots synthesized at 300°C, displaying a single broad emission band,
10 are reported in Figure S3 A-B, while Figure 7 shows the emission spectra of C-dots grown at
11 200°C, for reaction time of 30 min (Figure 7a) and 180 min (Figure 7b), respectively.
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13 Interestingly, in this case the PL broad band results from the overlapping of two distinct
14 contributions in spectral ranges centered approximately at 435 and 575 nm, respectively (Fig. 7
15 c-d) and their relative intensities modify with the excitation wavelength. At short λ_{Exc} the overall
16 emission principally arises from the band centered at 435 nm.
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24 At longer excitation ($\lambda_{\text{Exc}} > 400$ nm), the band centered at 575 nm becomes progressively
25 stronger, while the band centered at 435nm decreases in intensity. For $\lambda_{\text{Exc}} > 420$ nm (Figure 7a-b,
26 purple line) the low energy band becomes the sole emissive contribution. A clear interrelation
27 between this new emission state at lower energy and the appearance of an absorption feature at
28 450 nm for these types of samples, as reported in Figure 6, is out of debate. Figure 7 also
29 compares the PL intensity (c) and the peak position (d) as a function of the excitation
30 wavelength. The PL intensity shows the same trend of C-dots characterized by emission
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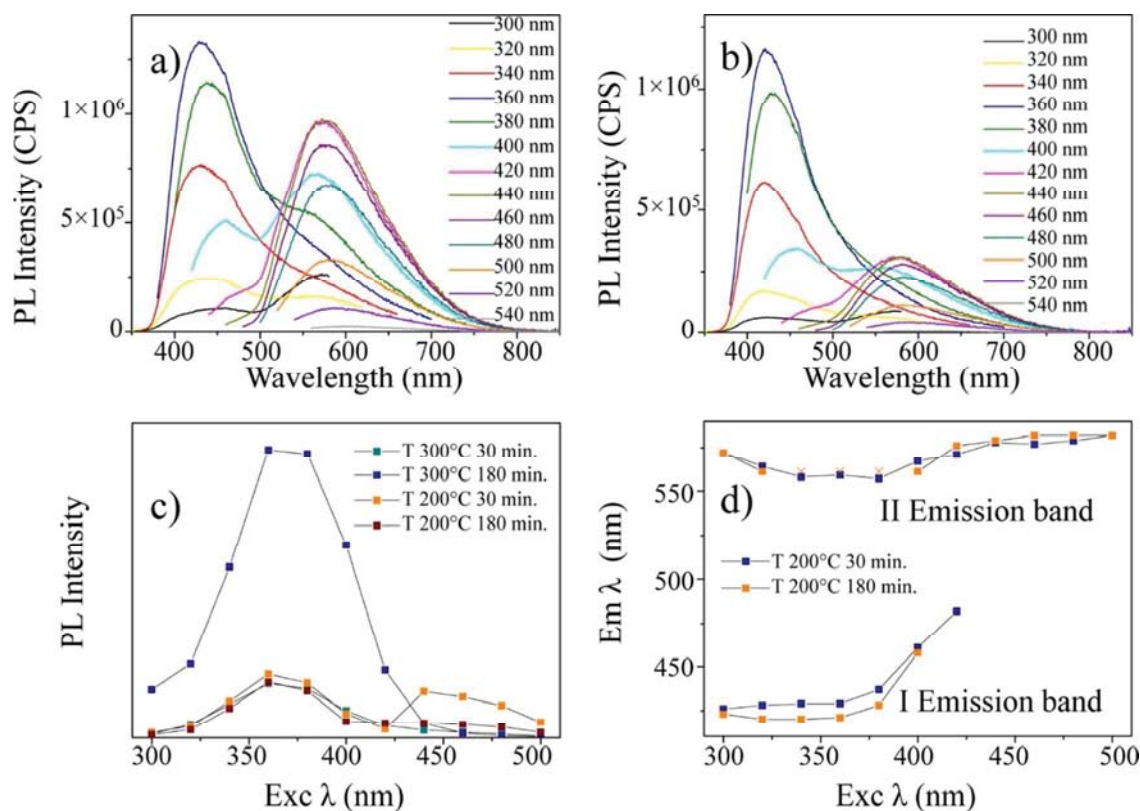


Figure 7. PL spectra of chloroform dispersion of C-dots grown at 200°C for 30 (a) and 180 (b) min, as a function of the excitation wavelength; (c) variation of PL intensity as a function of the excitation wavelength of C-dots grown at different temperatures and reaction times; PL peak position of the first and second emission band of C-dots synthesized at 200°C for 30 and 180 min.

molecular species and surface-related functional groups, as discussed before.¹⁰ In addition, a pronounced emission tail is observed at excitation wavelength longer than 420 nm for samples grown at 200°C. Figure 7 b-c indicate, for the first emission band, a dependence from the λ_{Exc} in the range 360-420 nm for both the samples grown at 30 and 180 min. On the contrary, the peak of the second emission band only slightly shifts from its position, mainly when excitation wavelength is longer than 400 nm. Such second emission band can be considered due to the fluorescence of molecular species, being independent from excitation.¹⁰ In particular, the

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3 progressive evolution towards red emissive states is consistent with the possible occurrence of
4 aggregation among the molecular fluorophores, whose absorption and emission signals are
5 shifted to lower energies.¹¹ These overall evidences suggest a molecular origin of the emission at
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8 low energy.
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12 The PLE spectra of sample grown at 300°C and performed in correspondence of the maximum
13 PL emission at 435 and 575 nm show an intense excitation band, corresponding to the absorption
14 transitions centered at 350 nm, a broad contribution at higher wavelength and a weak shoulder at
15 270 nm (Fig. 8-a). Conversely, for samples grown at 200°C, an additional excitation band is
16 observed in correspondence of the absorption shoulder at 450 nm (Fig. 8 b-c). It is worth to note
17 that a peak at high energy is present also for longer reaction time. Then, when C-dots are
18 synthesized at lower temperature, a second band arises in the absorption, excitation and emission
19 spectra. The presence of this intense band significantly contributes to the overall quantum
20 efficiency. Indeed, the QYs of samples grown at 200°C are much higher than those of C-dots
21 grown at 300°C (Table S3), characterized by a single PL band, reaching the value of 42% for
22 excitation at 360 nm for reaction time of 30 min. The result is in agreement with the work of
23 Kriesmann et al¹⁵, demonstrating that at reaction temperature lower than 230°C, the PL
24 mechanism is principally due to the highly emissive fluorophores formed by the condensation
25 reaction between the carboxylic moieties of the CA and the amino functionalities of the
26 passivating agent. At higher temperature, the pyrolysis starts to form carbonaceous core, and in
27 such experimental conditions, the PL properties are expected to arise from both intrinsic core
28 domains and fluorophores, that can be present as either intermediates or interacting species at the
29 surface of the C-dots. Surface states can contribute to the emission properties of C-dots
30 according to the preparative conditions with a variety of complex excited states. Then, the
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reaction time and the heating temperature play a decisive role in tuning the C-dot optical properties. Indeed, as the reaction time proceeds and the temperature is increased, the carbonization process advances and the carbogenic core is formed at

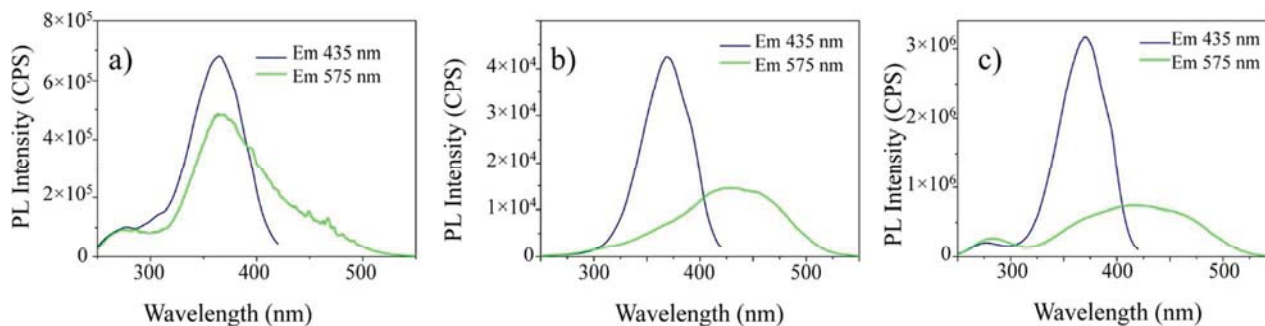


Figure 8. PL excitation spectra of chloroform dispersion of C-dots grown at 300°C for 30 min (a) or 200°C for 30 (b) or 180 min (c).

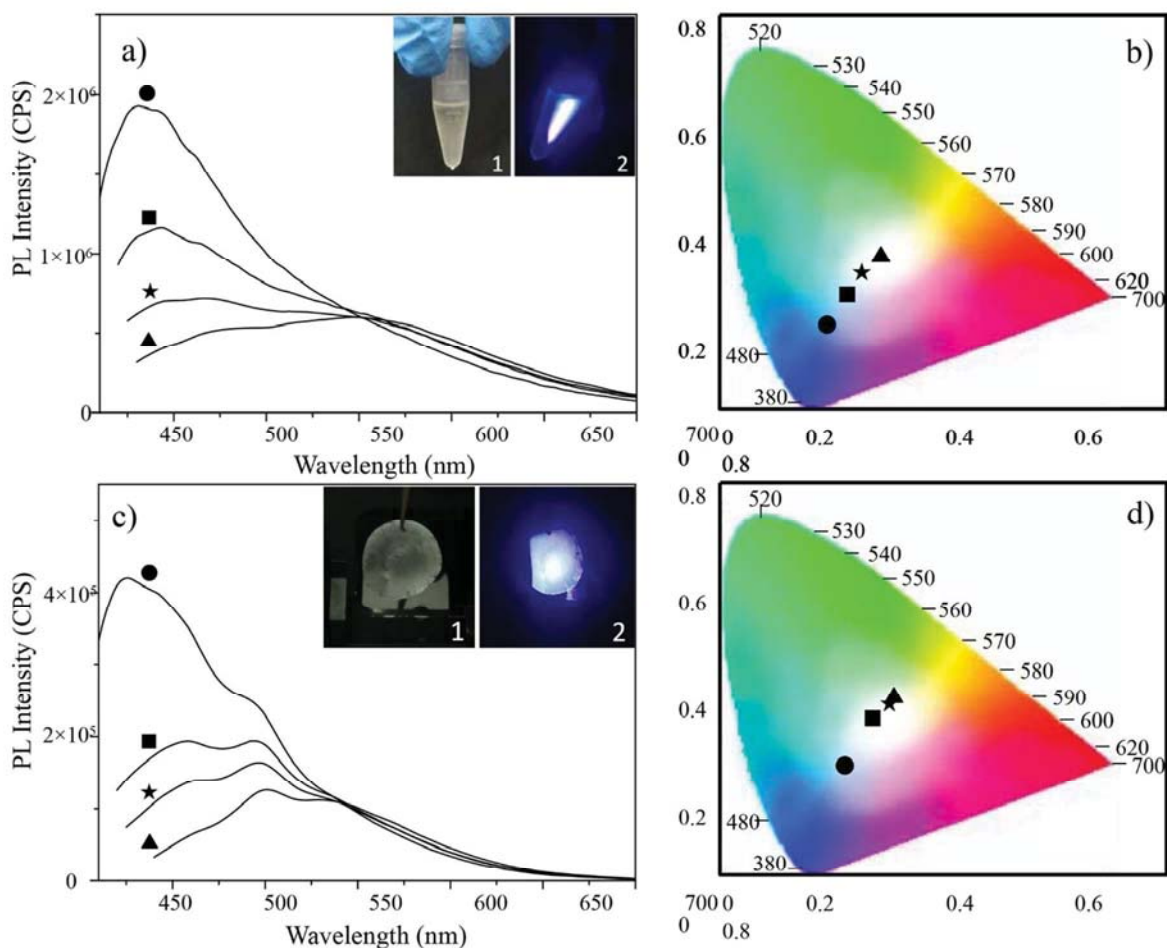
expenses of molecular species and surface connected functional groups, thus causing a decrease in QY, as already highlighted.

The presence of an additional band in absorption, excitation and emission spectra in the red region is indicative for the presence of multiple excited states emitting at low energies. Similarly to the findings of Kumbhakar et al.¹¹, although the different synthetic conditions, we can suppose the formation of complex aggregates of molecular fluorophores emitting at low energies that contribute to the overall PL emission of C-dots prepared in such specific synthetic conditions.

3.4 White PL emission

The synergistic effect of the carbogenic core and the molecular species and/or surface functional groups on C-dot optical properties accounts for intriguing implications arising from the multicolored emission of NPs showing the double emission band. Indeed, illuminating the chloroform dispersion of C-dots grown at 200°C by light of suitable wavelength, a white emission is observed. Figure 9-a shows the PL spectra of the C-dots grown at 200°C for 180 min

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3 by exciting at increasing wavelengths. For excitation from 390 to 410 nm, the PL emission
4 broaden and cover almost the entire visible range of the spectrum, thanks to the enhanced
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6 broaden and cover almost the entire visible range of the spectrum, thanks to the enhanced
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8 contribution of the red component. In such conditions, when the C- dots are excited at 405 or 410
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10 nm, they show an intense white PL emission (inset of Figure 9-a). A similar behavior has been
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12 observed for C-dots grown at 200°C for 30 min (data not shown). The calculated CIE 1931
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14 chromaticity coordinates in Fig. 9-b are (0.267, 0.315), (0.300, 0.361) corresponding to the
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16 excitation wavelength of 405 and 410 nm. Conventionally, “pure” white light is positioned as
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18 single point at (0.333, 0.333) coordinates in the CIE 1931 diagram. However, the white emission
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3 **Figure 9.** (a) and (c) PL emission spectra of C-dots in solution and film, incorporated in PMMA,
4 obtained exciting at 390, 400, 405 or 410 nm (filled dot symbol, square symbol, star symbol or
5 triangle symbol, respectively). In the insets, a picture of the vial and of the film under
6 illumination with a laser diode at 405nm. (b) and (d) Position in the CIE 1931 diagram of C-dot
7 dispersion and PMMA based film at the corresponding wavelength.
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15 complementary colors that properly combine, giving rise to a more extended white emission
16 gamut.³⁹ Therefore, under illumination at 405 and 410 nm, the C-dot chloroform dispersion
17 displays high PL emission in the white gamut (Inset of Fig. 9-a, and Fig.9-b), with QY of 24 and
18 22%, respectively.
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25 Then, suitably playing with the synthetic parameters it is possible to prepare single carbon NPs
26 characterized by an intense and stable white emission, being at the same time biocompatible,
27 harmless and reasonably priced. The possibility to have such nano-objects disperse in organic
28 solution and their high quantum efficiencies open the way to their successful exploitation in
29 white light emitting devices.
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37 In order to test the effective viability of the white emitting C-dots in practical applications, the
38 as synthesized NPs have been simply incorporated into a suitable polymeric matrix, namely
39 poly(methylmethacrylate), PMMA, to obtain nanocomposite solid films. Indeed, within the
40 polymeric matrix, the C-dots can preserve or further enhance their optical properties taking
41 advantage of the improved processability and mechanical properties of PMMA.⁴⁰
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48 PL emission spectra of the prepared nanocomposite films show the typical excitation
49 dependent wide emission completely covering the visible spectrum (Figure 9-c). The PL
50 emission consists of two main broad bands, contributing to the overall emission with intensities
51 that depend on the excitation wavelength: the ratio between the underlying areas of the two
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3 contributing bands significantly changes with the increasing of λ_{Exc} . Upon incorporation in the
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5 polymeric matrix, the C-dots confirm the white light emission when excited at 400, 405 or 410
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7 nm (Inset of fig. 9-c and Fig. 9-d). The coordinates of the CIE 1931 diagram of nanocomposite
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9 films are (0.291, 0.369), (0.319, 0.403), (0.335, 0.435) for excitation at 400, 405 and 410 nm,
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11 respectively (Fig. 9-d), displaying high PL emission in the white gamut (Inset of Fig. 9-c, and
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13 Fig 9-d), with QY of 23, 20 and 18%, respectively.
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17 In addition, the color stability in time of the white emitting C-dot/PMMA nanocomposite film
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19 has been tested by measuring the color point and the QY of the sample stored in air at room
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21 temperature, after several months of aging. The results (see Figure S4) show that the color point
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23 of the nanocomposite still falls in the white color region of the CIE 1931 diagram without
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25 substantial decrease of quantum efficiency (QY of 19% and 18% for excitation at 405 and 410
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27 nm, respectively). The spectroscopic results indicate that upon incorporation of the C-dots in the
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29 organic matrix the emission features of the fabricated films are preserved, thus suggesting that
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31 the surface interactions with the host polymer do not affect the emissive states.
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36 4. CONCLUSIONS

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40 In summary, oil-soluble C-dots have been synthesized by a hot-injection method and
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42 systematically investigated by extensive physical and chemical characterization. Synthetic
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44 parameters as capping agent concentration, reaction time and temperature have been modified in
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46 order to evaluate their effect on the C-dot optical features. We have found that the presence of
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48 HDA, is fundamental in forming by poly-condensation both the amide bonds at C-dot surface
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50 and the intermediate molecular fluorophores that mostly contribute to the overall emission.
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52 Increasing reaction time and temperature advances the carbonization thus promoting the
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54 formation of carbogenic cores and the consumption of the molecular fluorophores, ultimately
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3 resulting in QY decrease. Interestingly, at lower reaction temperature, an additional absorption
4 and emission band is observed. Irradiating such C-dots by visible light in the range 400-410 nm
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7 results in a bright fluorescence in the white gamut. Here, the proper choice of synthetic
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10 parameters allows for preparing functional materials with white emission based on
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12 biocompatible, harmless and low- cost single carbon NPs. Their incorporation in a PMMA host
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14 matrix demonstrates the effective viability of such white emitting C-dots in practical
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16 applications. In particular, the polymeric matrix preserves the C-dot unique optical features and
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18 the nanocomposite solid films exhibit an intense white emission even after their storage in air at
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20 room temperature for several months.
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24 25 26 ASSOCIATED CONTENT

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29 **Supporting Information.** Partial characterization data, including PL spectra of C-dots at
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31 increasing reaction time, tables with QY values at different reaction conditions, table reporting
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33 the fitting parameters of the PL decays and the 1931 CIE diagram of the C-dot /PMMA film after
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48 49 **Author Contributions**

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51 The manuscript was written through contributions of all authors. All authors have given
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53 approval to the final version of the manuscript.
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