Wang, R. et al. (2013). Nitrogen-doped carbon coated ZrO2 as a support for Pt nanoparticles in the oxygen reduction reaction. International Journal of Hydrogen Energy, 38: 5783 – 5788 <u>http://dx.doi.org/10.1016/j.ijhydene.2013.03.041</u>



# Nitrogen-doped carbon coated ZrO2 as a support for Pt nanoparticles in the oxygen reduction reaction

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#### Abstract

A new nitrogen-doped carbon (CN<sub>x</sub>) support for Pt electrocatalysts was prepared by carbonizing polypyrrole on the surface of  $ZrO_2$  ( $ZrO_2@CN_x$ ) at high temperature. Well-dispersed Pt nanoparticles were easily formed on the  $ZrO_2@CN_x$ . The electrocatalyst was characterized by FT-IR, XRD, TEM, XPS. The electrochemical performances indicate that the presence of  $ZrO_2$  modified the electro-structure of Pt on the catalyst surface and that  $ZrO_2@CN_x$  had superior oxygen reduction activity compared to a nitrogen-doped carbon coated carbon (C@CNx).

#### 1. Introduction

A main objective in fuel cell research is to develop low-cost, high-performance and durable catalyst materials. Current fuel cell systems have high intrinsic cost and fairly poor durability. Pt and Pt-based alloys are the most commonly used electrocatalysts for proton exchange membrane fuel cells (PEMFC) [1,2]. Due to the global scarcity of Pt and its high cost there is an urgent need to reduce its use, in addition to improving the efficiency of PEMFC. Depositing Pt and Pt-based alloys on a conductive, porous support reduces the cost and improves performance. Interaction between the catalyst and the support can improve catalyst efficiency, decrease catalyst loss, assist in charge transfer, reduce catalyst poisoning (e.g. by CO, S, etc.), and in some cases beneficially affect catalyst particle size [3]. Hence, the choice of support material lies central to the behavior, performance, durability and cost effectiveness of the catalyst and thus the overall fuel cell. Conventionally, highly conductive carbon blacks of turbostratic structures with high surface areas, such as Vulcan XC- 72R, Shawinigan and Black Pearl 2000 are used for catalyst support [3]. However, the corrosion of carbon, resulting from electrochemical oxidation during fuel cell operating conditions is a major contributor to poor durability [3]. Recently, improving the performance of carbonaceous supports and exploring novel non-carbonaceous electrocatalyst support materials have become active research areas [4e7].

In recent years, the use of metal oxides as electrocatalyst supports has been explored to reduce the cost and increase the durability of fuel cells. Metal oxides such as  $Ti_4O_7$  [8],  $WO_x$  [9],  $Fe_3O_4$  [10],  $ZrO_2$  [7] and  $SnO_2$  [11] with high surface area, mechanical strength and thermal stability have already proven promising support materials for PEMFC. Among these oxides,  $ZrO_2$ , used as a membrane electrode anode (MEA) supporting Pt, has shown comparable performance to MEAs of commercial Pt/C. However, the MEA performance of Pt/ZrO<sub>2</sub> with Nafion was lower than that of conventional Pt/C due to low electric conductivity of Pt/ZrO<sub>2</sub>. Therefore, enhancement the  $ZrO_2$  support's electric conductivity forms a goal of interest, i.e. to produce a potentially low-cost high-performance catalyst.

Carbon coating provides an effective method to enhance the electric conductivity of oxides. It is also well known that the nitrogen-doped carbon possesses n-type or metallic behavior and show higher electron mobility than conventional carbon [12]. In this study, a method was developed to form a highly conductive nitrogen-doped carbon layer on the surface of  $ZrO_2$  ( $ZrO_2@CN_X$ ), on to which Pt nanoparticles were deposited to form Pt/ $ZrO_2@CN_X$ . The Pt/ $ZrO_2@CN_X$  catalyst was found to exhibit high electrocatalytic activity for the oxygen reduction reaction (ORR).

### 2. Experimental

Nitrogen-doped carbon was formed on the surface of ZrO<sub>2</sub> by polymerization of pyrrole (PPy) in a solution containing (NH<sub>4</sub>)<sub>2</sub>S<sub>2</sub>O<sub>8</sub> as oxidant. The detailed procedure was as follows: 100 mL of deionized water, 2 g of sodium dodecylbenzene-sulphoate and 1 g of ZrO<sub>2</sub> were added to a flask with stirring under nitrogen atmosphere. When a homogeneous suspension was formed, 0.5 mL of pyrrole was slowly added to the above solution. Then, 1.85 g of  $(NH_4)_2S_2O_8$  was added dropwise to the solution which was then stirred at 25 °C for 1 h ZrO<sub>2</sub>@PPy powder was recovered by filtering and dried at 60 °C. To prepare ZrO<sub>2</sub>@CN<sub>X</sub>, ZrO<sub>2</sub>@PPy was carbonized at high temperature in an inert gas oven. A 300 mg of ZrO<sub>2</sub>@PPy was placed in a quartz tube furnace and then heated to 800 ° C with a heating rate of 5 ° C min<sup>-1</sup> under N<sub>2</sub> atmosphere and kept at 800 ° C for 2 h. The furnace was cooled to room temperature and a black powder of ZrO2@CNx was obtained. For comparison, carbon coated with nitrogen-doped carbon (C@CN<sub>x</sub>) was also synthesized using the same procedure. Platinum catalysts (loading ca. 20 wt %) supported on ZrO<sub>2</sub>@CN<sub>x</sub> were prepared by an ethylene glycol (EG) solution method. H<sub>2</sub>PtCl<sub>6</sub> (66.4 mg) was dissolved in 30 mL of EG in a flask. The solution was adjusted to pH 10 by adding 5 wt% of KOH/EG solution. The ZrO<sub>2</sub>@CN<sub>x</sub> black powder was added to the H<sub>2</sub>PtCl<sub>6</sub> solution and then heated at 100 <sup>O</sup>C for 10 h. The resulting catalyst (Pt/ $ZrO_2@CN_x$ ) was filtered, washed with ultrapure water, and

dried in a vacuum oven.  $C@CN_X$  supported Pt (Pt/C@CN<sub>X</sub>) catalyst was synthesized using the same procedure for preparing Pt/ZrO<sub>2</sub>@CN<sub>X</sub>.

X-Ray Diffraction (XRD) and X-Ray Photoelectron Spectroscopy (XPS) spectra were generated by a Shimadzu XD-3A (Japan) goniometer with Cu Ka radiation (40 kV, 30 mA) and a PHI-5702 multifunctional X-ray photoelectron spectrometer (American), respectively. Transmission electron microscopy (TEM) measurements were carried out using a JEM-2010 Electron Microscope (Japan) with an acceleration voltage of 200 kV. The chemical composition of the samples was determined using energy dispersive X-ray analysis (EDX) technique coupled to TEM.

The electrochemical measurements of catalysts were carried out using an Autolab electrochemical work station (PGSTAT128N, Eco Chemie, The Netherlands). A conventional three-electrode electrochemical cell was used comprising a platinum wire as the counter electrode, an Ag/AgCl (KCl 3 M) electrode as the reference electrode, and a glass carbon electrode (5 mm in diameter) as the working electrode. All potentials are quoted with respect to the reversible hydrogen electrode (RHE). The thin film electrode was prepared as follows: 5 mg of catalyst was dispersed ultrasonically in 1 mL of Nafion/ethanol (0.25% Nafion). A 8 mL of the dispersion was transferred onto the glassy carbon disc using a pipette, and then dried in air to form the catalyst layer. Before each measurement, the solution was purged with high-purity N<sub>2</sub> (for oxygen-free solutions) or O<sub>2</sub> gas (for oxygen-saturated solutions) for at least 30 min.

### 3. Results and discussion

Fig. 1 shows the FT-IR spectra of ZrO<sub>2</sub>, C@PPy, C@CN<sub>X</sub>, ZrO<sub>2</sub>@PPy and ZrO<sub>2</sub>@CN<sub>X</sub> recorded from 4000 to 400 cm<sup>-1</sup>. In the spectra of these samples, the band at 1400 cm<sup>-1</sup> is attributed to the C]C stretch vibration, and the bands at 2850 cm<sup>-1</sup> are assigned to the CeH stretching vibration. In the all samples containing PPy and CN<sub>X</sub>, the bands at 1457 cm<sup>-1</sup> and 3400 cm<sup>-1</sup>, which are assigned to the CeN stretching vibration and NeH stretching vibration, respectively, are clearly presented in the FT-IR spectra [13]. After carbonization, the CeH stretching bands at 2850 cm<sup>-1</sup> were absent in the FT-IR spectra of C@CN<sub>X</sub> and ZrO<sub>2</sub>@CN<sub>X</sub>, which indicated that the PPy was completely carbonized after the high temperature treatment. Compared to the FT-IR spectrum of ZrO<sub>2</sub>, the FT-IR spectra of ZrO<sub>2</sub>@PPy and ZrO<sub>2</sub>@CN<sub>X</sub> with strong attenuation of the peaks suggest that ZrO<sub>2</sub> particles were completely coated by PPy and CN<sub>X</sub>.



Fig. 1 – FT-IR spectra of  $ZrO_2$ , C@PPy, C@CN<sub>x</sub>,  $ZrO_2$ @PPy and  $ZrO_2$ @CN<sub>x</sub>.



XRD patterns in Fig. 2 show the crystalline structure of Pt/ C@CN<sub>X</sub> and Pt/ZrO<sub>2</sub>@CN<sub>X</sub>. The characteristic peaks of mono-clinic ZrO<sub>2</sub> (2q at 24.5<sup>o</sup>, 28.2<sup>o</sup>, 31.5<sup>o</sup>, 34.2<sup>o</sup>, 40.8<sup>o</sup>, 49.3<sup>o</sup>, 54.1<sup>o</sup>, 55.4<sup>o</sup>) and tetragonal ZrO<sub>2</sub> (2q at 30.1<sup>o</sup>, 35.3<sup>o</sup>, 50.2<sup>o</sup>, 60.1<sup>o</sup>, 63.0<sup>o</sup>, 65.8<sup>o</sup>, 71.2<sup>o</sup>, 75.2<sup>o</sup>) are clearly shown in the XRD pattern of Pt/ ZrO<sub>2</sub>@CN<sub>X</sub> [14]. Two small peaks pertaining to cubic ZrO<sub>2</sub> are also present. These results indicate that the commercial ZrO<sub>2</sub> particles form a mixture of mainly monoclinic ZrO<sub>2</sub> and tetragonal ZrO<sub>2</sub> with a small amount of cubic ZrO<sub>2</sub>. After CN<sub>X</sub> was formed and Pt nanoparticles were deposited on the surface of the ZrO<sub>2</sub> particles, the characteristic peaks of crystalline Pt at 2q <sup>1</sup>/4 39.7<sup>o</sup>, 46.3<sup>o</sup>, 67.6<sup>o</sup>,

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81.4<sup>O</sup> (JCPDS, No. 04-0802), which are attributed to the (1 1 1), (2 0 0), (2 2 0), (3 1 1) crystal planes of face-centrered cubic (fcc) Pt, respectively, became clearly present in the XRD pattern [15]. The absence of a diffraction peak related to crystalline carbon indicates that the  $CN_X$  formed by carbonization of PPy was amorphous. In the XRD pattern of  $Pt/C@CN_X$ , there is a broad peak at 2q <sup>1</sup>/<sub>4</sub> 25<sup>O</sup> assigned to the amorphous carbon materials, which further indicates that the  $CN_X$  synthesized by carbonizing PPy at high temperature formed amorphous structure. The characteristic peaks of crystalline Pt pertaining to the Pt (111), (200), (220) and (311) planes are also clearly observed in the XRD pattern of Pt/ C@CN<sub>X</sub>.

TEM images in Fig. 3 show that Pt nanoparticles were well dispersed on the surface of C@CN<sub>X</sub> and ZrO<sub>2</sub>@CN<sub>X</sub> supports. The particle size distributions, insert (a) Pt/C@CN<sub>X</sub> and insert (b) Pt/ZrO<sub>2</sub>@CN<sub>X</sub>, show that the average size of Pt nanoparticles on their respective surfaces of was 5.0 and 5.9 nm, respectively. A uniform C@CN<sub>X</sub> support is observed in the magnified TEM image (Fig. 3c). EDX spectrum of Pt/C@CN<sub>X</sub> is presented as the inset in Fig. 3c, where the presence of C, N, O, and Pt elements are identified. As shown in Fig. 3d, ZrO<sub>2</sub> particles were completely covered by CN<sub>X</sub> formed during the carbonization of Ppy, The lattice of Pt and ZrO<sub>2</sub> are clearly shown in Fig. 3d, while the lattice of CN<sub>X</sub> is not observed, which further indicates that the CN<sub>X</sub> formed on the surface of ZrO<sub>2</sub> is amorphous. The presence of C, N, O, Zr and Pt are confirmed by the inset of Fig. 3d.



Fig. 3 – TEM images of (a)  $Pt/C@CN_x$  and (b)  $Pt/ZrO_2@CN_x$ . The particle size distributions of  $Pt/C@CN_x$  and (b)  $Pt/ZrO_2@CN_x$  are inserted in a and b. The EDX spectrum of  $Pt/C@CN_x$  and (b)  $Pt/ZrO_2@CN_x$  are inserted in c and d.



Fig. 4 shows the survey XPS spectra of  $Pt/C@CN_X$  and  $Pt/ZrO_2@CN_X$ , in which elements of Pt, C, Zr, O and N are detected in the  $Pt/ZrO_2@CN_X$  catalyst and Pt, C, O and N are found in the  $Pt/C@CN_X$  catalyst. These results confirm the presence of the N element in both catalysts after carbonization of PPy. XPS also provides an efficient way to study the surface oxidation states. As shown in Fig. 5, two peaks in

the Pt 4f binding energy region of Pt/C@CN<sub>X</sub> and Pt/ZrO<sub>2</sub>@CN<sub>X</sub> are observed at (70.6 eV, 73.9 eV) for Pt/C@CNx and (70.7 eV, 74.2 eV) for Pt/ZrO2@CNx, which are attributed to  $4f_{7/2}$  and  $4f_{5/2}$  of metallic Pt, respectively. Compared with the XPS of Pt/C@CN<sub>x</sub>, a clear shift to higher energy region of the Pt 4f peak in the XPS spectrum of  $Pt/ZrO_2@CN_X$  has been observed and results from the electronic effect on Pt due to the interaction between Pt and support [16]. To evaluate the surface oxidation states of Pt, the Pt 4f spectra were de-convoluted into three doublets which are assigned to the different oxidation states of Pt. The most intense doublet (around 71 eV and 74 eV) is assigned to metallic Pt. The second doublets (around 71.5 eV and 75 eV), can be attributed to the Pt (II) chemical state, i.e. PtO and Pt(OH)<sub>2</sub> [17]. The weakest doublet of Pt at even higher binding energies (around 73.3 eV and 76.3 eV) indicates Pt (IV) in PtO<sub>2</sub> (IV) [18] species existing on the surface of both catalysts. Since the amount of Pt species are related to the relative intensities of these three peaks, the amount of each Pt oxidation states can be calculated from XPS results. The results show that the percentages of each Pt oxidation state for Pt/C@CN<sub>x</sub> are Pt(0) 44%, PtO/Pt(OH)<sub>2</sub>(II) 39%, PtO<sub>2</sub>(IV) 17%, and for Pt/ZrO2@CNx are Pt(0) 39%, PtO/Pt(OH)2(II) 34%, PtO2(IV) 27%.

It has been reported that there are four types of nitrogen species found in carbonbased materials such as coal and char [19]. These comprise pyridinic-N (398.6  $\pm$  0.3 eV), pyrrolic-N (400.5  $\pm$  0.3 eV), quaternary nitrogen (401.3  $\pm$  0.3 eV) and pyridinic-N<sup>b</sup>eO<sup>-</sup> (402e405 eV). As shown in Fig. 6, there are four peaks in the N 1s spectra for both Pt/C@CN<sub>x</sub> and Pt/ ZrO2@CN<sub>x</sub> catalysts. It should be noticed that pyridinic-N, which provides one *p*-electron to the aromatic *p*systems and has a pair of electrons in the plane of the carbon matrix, can increase the electron-donor property of the catalyst. Pyridinic-N can thus weaken the bond of OeO via bonding with nitrogen or the carbon atom adjacent to it, and further facilitate the oxygen reduction. The pyridinic-N is usually located on the edge of carbon plane and carbon vacancy. In Fig. 6a and b the integrated area ratio of pyridinic-N increased when ZrO2@CN<sub>x</sub> was used as support.

The oxidation state of  $ZrO_2$  is shown in the XPS results in Fig. 7. Two Zr 3d peaks were observed at the binding energies of 182.4 eV and 184.5 eV, which are attributed to  $ZrO_2$  [20]. The result indicates that the oxidation state of  $ZrO_2$  did not change after  $CN_X$  was deposited on its surface.

The cyclic voltammetry (CV) results of  $Pt/C@CN_X$  and  $Pt/ZrO_2@CN_X$  in 0.5 M  $H_2SO_4$  electrolyte between 0 and 1.2 V -1 (versus RHE) with a scanning rate of 50 mV s , are shown in Fig. 8. Typical CV curves of Pt are observed for both electrocatalysts, i.e. hydrogen adsorption/desorption peaks in the low potential region, oxide formation/stripping wave/peak in the high potential region, and a flat double layer in between. In the potential range from 0 to 0.2 V, three hydrogen desorption

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peaks are observed at about 0.01, 0.1, 0.18 V, which correspond to the catalytic effects of Pt (111), Pt (110), Pt (100) planes, on the sample of Pt/ZrO<sub>2</sub>@CN<sub>X</sub>. These three desorption peaks on the sample of Pt/C@CN<sub>X</sub> are not as clear as those on Pt/ZrO<sub>2</sub>@CN<sub>X</sub>. The CV curve of Pt/ZrO<sub>2</sub>@CN<sub>X</sub> also shows two hydrogen absorption peaks, which are absent on Pt/C@CN<sub>X</sub>. The results suggest that the crystallinity of Pt nanoparticles on the support of ZrO<sub>2</sub>@CN<sub>X</sub> is higher than that on C@CN<sub>X</sub>, and thus lead to an enhancement of the ORR kinetics [21]. The ORR polarization curves of Pt/ZrO<sub>2</sub>@CN<sub>X</sub> and Pt/C@CN<sub>X</sub> catalysts in oxygen-saturated 0.5 M H<sub>2</sub>SO<sub>4</sub> using a rotating disk electrode at 1600 rpm, Fig. 9.



Fig. 5 – XPS spectra of Pt 4f region for the (a) Pt/C@CN<sub>x</sub> and (b) Pt/ZrO<sub>2</sub>@CN<sub>x</sub>.



Fig. 6 - XPS spectra of N 1s region for the (a) Pt/C@CNx and Pt/ZrO2@CNx.

Here, both the onset and peak potential of  $Pt/ZrO_2@CN_X$  catalyst shift to a more positive position. The half-wave potentials of  $Pt/ZrO_2@CN_X$  and  $Pt/C@CN_X$  catalysts were 0.77 V and 0.73 V, respectively. The half-wave potentials of  $Pt/ZrO_2@CN_X$  is higher than that of  $Pt/C@CN_X$  catalyst, showing that the  $Pt/ZrO_2@CN_X$  is more active towards ORR, apparently due to the  $ZrO_2$  support.



Fig. 7 - XPS spectra of Zr 3d region for the Pt/ZrO2@CNx.



Fig. 8 – Cyclic voltammograms of Pt/C@CN<sub>x</sub> and Pt/ ZrO<sub>2</sub>@CN<sub>x</sub> electrocatalysts in 0.5 M  $\rm H_2SO_4$  at 50 mV  $\rm s^{-1}$  at room temperature.



Fig. 9 – Polarization curves of ORR on Pt/ZrO<sub>2</sub>@CN<sub>x</sub> and Pt/C@CN<sub>x</sub> catalysts in oxygen-saturated 0.5 M H<sub>2</sub>SO<sub>4</sub>; rotation rates 1600 rpm, 30 °C, scan rate: 5 mV s<sup>-1</sup>.



30 °C, scan rate: 5 mV s<sup>-1</sup>, rotation rate: 1600 rpm.

Tafel plots for the kinetic current density normalized to the electrochemical active surface are presented in Fig. 10. The Tafel lines of the Pt/ZrO<sub>2</sub>@CN<sub>X</sub> catalyst clearly moved to a higher potential region, compared to the Pt/C@CN<sub>X</sub> catalyst, suggesting that the Pt/ZrO<sub>2</sub>@CN<sub>X</sub> catalyst had a higher kinetic ORR activity. Both Pt/C@CN<sub>X</sub> and Pt/ZrO<sub>2</sub>@CN<sub>X</sub> show two Tafel regions with slopes of -60 mV dec<sup>-1</sup> at the low current region, and -120 mV dec<sup>-1</sup> at the high current region, which indicate that the ORR on both catalysts follows the same mechanism as in Pt/C catalysts [22]. Therefore the rate determining steps for ORR on ZrO<sub>2</sub>@CN<sub>X</sub> support material appear to be the same as on carbon.

### 4. Conclusions

Forming nitrogen-doped carbon on the surface of metal oxides,  $ZrO_2$ , via carbonization at high temperature provides a useful method for synthesizing support of electrocatalysts. Well-dispersed Pt nanoparticles were successfully synthesized on the surface of  $ZrO_2@CN_X$  supports. The electrochemical data for ORR demonstrate that  $Pt/ZrO_2@CN_X$  is more active than  $Pt/C@CN_X$  due to the  $ZrO_2$  support, and follows the same mechanism as Pt/C catalyst. Further work is underway to determine how  $ZrO_2$  affects the ORR activity and its longterm stability.

#### Acknowledgments

The authors would like to thank the National Natural Science Foundation of China (21163018, 51262028) and the National Science Foundation for Post-doctoral Scientists of China (20110490847, 2012T50554) for financially supporting this work.

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