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Article

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Monolithic CIGS—Perovskite Tandem Cell for Optimal Light Harvesting without Current Matching

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 - Supporting Information

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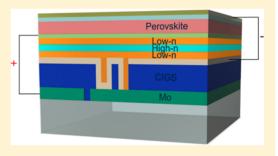
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ABSTRACT: We present a novel monolithic architecture for optimal light harvesting in multijunction thin film solar cells. In the configuration we consider, formed by a perovskite (PVK) cell overlying a CIGS cell, the current extracted from the two different junctions is decoupled by the insertion of a dielectric nonperiodic photonic multilayer structure. This photonic multilayer is designed by an inverse integration approach to confine the incident sunlight above the PVK band gap in the PVK absorber layer, while increasing the transparency for sunlight below the PVK band gap for an efficient coupling into the CIGS bottom cell. To match the maximum power point voltages in a parallel connection of the PVK and CIGS cells, the latter is divided into two subcells by means of a standard



three-laser scribing connection. Using realistic parameters for all the layers in the multijunction architecture we predict power conversion efficiencies of 28%. This represents an improvement of 24% and 26% over the best CIGS and PVK single-junction cells, respectively, while at the same time outperforms the corresponding current-matched standard tandem configuration by more than two percentage points.

KEYWORDS: perovskite solar cells, CIGS solar cells, inverse integration, tandem, serial-parallel configuration

erovskites (PVKs) recently emerged as a solutionprocessed PV technology that in a short period of time 31 reached power conversion efficiencies (PCEs) comparable to 32 the ones from the well-established lower band gap crystalline Si 33 technology. 1,2 One of the remarkable features of PVK cells is 34 their high open-circuit voltage (V_{oc}) , which may reach values as 35 high as 1.24 V (at a band gap of 1.63 eV). Such high voltages, a 36 tunable band gap, 4 and a high charge mobility make them an 37 optimal technology to form a pair with CIGS or crystalline Si-38 based solar cells and lead to a low-cost multijunction 39 technology suitable for terrestrial sun energy harvesting. 5-10 40 In the most standard tandem configuration the two cells are in 41 a serial connection that requires current matching between 42 both cells. 11-14 This constitutes a constraint in the optical 43 design of the optimal tandem architecture that eventually limits 44 the PCE of the final device. To partially circumvent such 45 problems, 3-terminal¹⁵ or 4-terminal^{6,13,16,17} configurations 46 have been proposed. The former one is applicable when the 47 two absorber materials are the same, while in the latter one, 48 there are no material constraints. Mechanically stacked 4-49 terminal devices have shown efficiencies that overcome that of the corresponding single subcells, ^{6,8} but the monolithic 50 character, which may be relevant to achieve a low-cost solar 51 module installation, would be lost to a large extent. A higher 52 cost associated with the connectivity complexity is expected in 53 the installation of such 4-terminal modules. On a theoretical 54 basis it was proposed to connect an indefinite number of cells 55 in each one of the branches of a parallel configuration in series 56 until the voltages of the two branches are matched. ¹⁸ However, 57 no practical implementation of the device configuration was 58 considered. An alternative option proposed piling up three 59 cells, two of them in a series connection and then these two 60 connected in parallel with the remaining one. ¹⁹ The latter 61 combination is an interesting proposal to match the voltage, but 62 the problem of current matching remains.

In this work we propose a new configuration, denoted as S-P 64 (serial-parallel), different from the standard or any other 65 tandem scheme considered until now. We describe a 66 monolithic tandem architecture where current matching 67

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68 between the PVK and CIGS cells is not required. Performing 69 an optical optimization of the structure, we predict an efficiency 70 of 28% using a 1.56 eV band gap PVK and PV parameters that 71 have already been reached for the constituent CIGS or PVK 72 subcells. Furthermore, we discuss the experimental viability of 73 the proposed architecture.

RESULTS AND DISCUSSION

75 As shown in Figure 1a, the configuration is formed with a 76 semitransparent PVK solar cell deposited on top of two CIGS

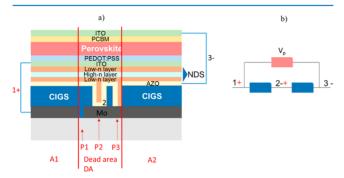


Figure 1. (a) Schematic drawing of the monolithic multijunction device composed of a two-series connected CIGS cell connected in parallel with a PVK cell. (b) Schematic drawing of the connections of the S-P configuration.

77 solar cells laterally connected in series. A schematic drawing of 78 the architecture with connection nodes numbered from 1 to 3 79 is shown in Figure 1b. Voltage matching at nodes 1 and 3 from 80 both branches of the circuit may, in principle, be achieved 81 provided the maximum power point voltage $(V_{
m mpp})$ of the light-82 filtered CIGS cells is close to half that from the PVK cell. The 83 serial connection among the two CIGS cells can be 84 implemented by means of laser scribing, a standard technique 85 for the up-scaling of CIGS cells to modules. As seen in Figure 86 la, a dielectric multilayer referred as a 1-D photonic 87 nonperiodic dielectric structure (NDS) is inserted on top of 88 the first CIGS solar cell serving the double purpose of avoiding 89 short-circuiting of nodes 1 and 2 and maximizing light 90 absorption for the first and second cell. On the other hand, 91 the necessary current matching between the two serial-92 connected CIGS devices can be easily obtained by defining 93 an equal area for both.

In the implementation of the optical optimization described 94 in the Methods we consider a configuration where the CIGS 95 solar cell architecture is the standard one and some changes are 96 introduced to the PVK device, for instance, the use of top and 97 bottom transparent conductive electrodes (TEs). Specifically, 98 the S-P architecture consists of the semitransparent PVK cell on 99 top of a CIGS cell separated by the NDS as follows: glass/Mo/ 100 MoSe/CIGS/CdS/i-ZnO/AZO/low-n film/high-n film/low-n 101 film/TE/PEDOT:PSS/PVK/PCBM/TE/MgF2. As TEs for the 102 semitransparent PVK we considered 90 nm ITO films. The 103 ITO is chosen instead of the common FTO due to its better 104 processing compatibility with the underlying materials and its 105 improved near-infrared transmission. Similar thicknesses of 106 ITO films, prepared at room temperature, have been employed 107 to form a recombination layer and top contact in a monolithic 108 tandem architecture with PVK and silicon, exhibiting high fill 109 factors.10

In the S-P configuration considered here, the optical 111 optimization is performed for a device that includes the NDS 112 between the inner TE/AZO interface as seen in Figure 1a. Such 113 NDS is formed by a combination of three layers of intercalated 114 low and high refractive index materials and has a 2-fold 115 functionality. First, it electrically separates the TE anode of the 116 PVK and the AZO cathode from the CIGS, and second, it 117 confines the incident sunlight above the PVK band gap in the 118 PVK absorber layer, while increasing the transparency for 119 sunlight below the PVK band gap for an efficient coupling into 120 the CIGS bottom cell. To prevent tunnelling from electrical 121 charges between the PVK and CIGS cells, one may use MgF2 as 122 a low-n material, which is a highly insulating transparent 123 material exhibiting a refractive index close to 1.38. For the highn film, a highly transparent dielectric material with a refractive 125 index of 2.2 would be the optimal. Possible materials for the 126 high-index layer may be TiO₂ or Si₃N₄. Considering the three 127 films, the total thickness of the NDS would be above 500 nm, 128 sufficient, in principle, to prevent electrical charge tunnelling. 129 When the inverse integration optical optimization detailed in 130 the Methods section is applied, the EQEs obtained for the 131 NDS-separated PVK and CIGS cells are shown in Figure 2a. 132 f2 Under the assumption of a negligible extinction coefficient for 133 any of the layers within the NDS, we observe an enhancement 134 in the maximum short-circuit current of the PVK and CIGS 135 subcells, especially in the wavelength range between 700 and 136 900 nm. The NDS fine-tuning required to reach maximum 137

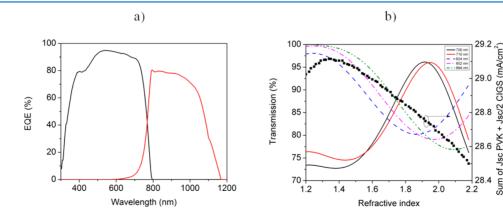


Figure 2. (a) Calculated EQE for the perovskite (black) and the CIGS (red) subcells for the optimal configuration. (b) Sum of PVK and CIGS photocurrent densities (black squares) and transmission of the NDS at 700 nm (black line), 710 nm (red), 824 nm (blue dash), 852 nm (magenta dash dot), and 884 nm (green dash dot dot) as a function of the refractive index of the low-n NDS layer.

Jsc/2 CIGS

Jsc

Sum of

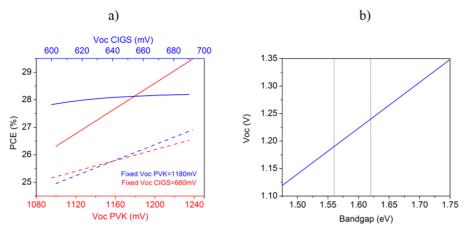


Figure 3. (a) PCEs for the tandem (dashed lines) and S-P (continuous lines) configurations. The red lines correspond to a fixed $V_{\rm oc}$ for the CIGS solar cells at 660 mV, while the $V_{\rm oc}$ of the PVK is allowed to change from 1080 to 1240 mV. The blue lines correspond to a fixed $V_{\rm oc}$ condition of the PVK at 1180 mV, while the $V_{\rm oc}$ of the CIGS is allowed to change from 600 to 690 mV. (b) Estimate of PVK $V_{\rm oc}$ as a function of the PVK band gap. The two reference values are marked as vertical lines.

138 PCE may be extracted from Figure 2b, where the transmission 139 at five different wavelengths just above and below the PVK 140 band gap is shown as a function of the NDS low refractive 141 index material. In this figure one observes that for the cell 142 architecture with the highest short-circuit current sum, when the NDS low-index material is close to 1.38, the inverse integration leads to an NDS transmission that is larger at longer wavelengths to favor absorption at the CIGS and smaller at 146 shorter wavelengths to favor absorption in the PVK. Note that 147 the unbalanced current density in both subcells explains why 148 the PCE of the S-P configuration is considerably larger than the 149 PCE for the current matched tandem, as seen in Figure 3a. The 150 optimal thicknesses for the critical layers inside the S-P 151 structure depend on the selected TE. When using a 90 nm 152 indium tin oxide (ITO) film, the optimal architecture is glass/ $153 \text{ Mo/MoSe/CIGS} (2.5 \mu\text{m})/\text{CdS} (40 \text{ nm})/\text{i-ZnO}(50 \text{ nm})/$ 154 AZO(100 nm)/low-n film (70 nm)/high-n film (220 nm)/low-155 n film (250 nm)/ITO (90 nm)/PEDOT:PSS (20 nm)/PVK 156 (700 nm)/PCBM (20 nm)/ITO (90 nm)/MgF₂ (98 nm), delivering a J_{sc} of 22.06 mA/cm² for the PVK and 14.1 mA/cm²

To confirm that such an increase in photocurrent in the S-P 160 configuration corresponds to an equivalent increase in PCE, we determined the rest of the electrical parameters that character-162 ize the overall IV curves employing the single-diode electrical 163 equivalent circuit model. The single-diode circuit model was 164 applied to each branch of the circuit, and then the currents were added point by point to obtain the resulting IV curve. To 166 reach optimal PCEs, we set the fill factor (FF) between 80% 167 and 81% for all the single-junction cells used in the S-P 168 configuration. A similar procedure was used to determine the 169 optimal configuration for a standard current-matched tandem, 170 which is used as a reference for comparison. Although small variations resulting from the IV curve fitting may be found, the 172 FF of the S-P structure closely approached that of the subcell, 173 with the highest FF for most of the cases considered. On the 174 other hand, the $V_{
m oc}$ of the S-P is roughly limited by the voltage 175 of the branch that has the lowest voltage, as can be seen in 176 Figure S1 of the SI. We studied two cases, one where the $V_{\rm oc}$ of 177 the PVK was fixed to 1180 mV while the $V_{\rm oc}$ of the CIGS was 178 varied in the 600-700 mV range, and a second case where the $V_{\rm oc}$ of the CIGS was fixed to 660 mV and that of the PVK was 180 allowed to change in the 1080–1240 mV range. The ideality

factor, series resistance, shunt resistance, and saturation current 181 were determined to match experimental curves when available 182 (Figure S2 of the SI) or from the computer-generated single- 183 junction IV curves exhibiting the targeted $J_{\rm sC}$ FF, and $V_{\rm oc}$ 184 parameters. Note that the changes in $V_{\rm oc}$ considered here are 185 not associated with band gap modifications, which will be 186 considered below. Indeed the $V_{\rm oc}$ ranges encompass a wide 187 variety of voltages that are often found in experimental CIGS or 188 PVK single-junction devices.

As seen in Figure 3a, when a PVK cell with a $V_{\rm oc}$ of 1180 mV 190 is combined with a CIGS cell in the S-P configuration, the 191 efficiency remains more or less around 28% as the $V_{\rm oc}$ of the 192 CIGS changes. This PCE is clearly larger than the PCE from 193 any of the possible tandem configurations also shown in Figure 194 3a. In the event that the $V_{\rm oc}$ of the PVK is further increased, 195 PCEs above 29% would be reachable, while for the standard 196 tandem PCEs larger than 27% are, in principle, not possible. 197 This confirms that the S-P configuration clearly outperforms 198 the standard tandem by more than two percentage points in 199 almost all cases. Note that, when optimal single-junction cells 200 are considered, both configurations, tandem and S-P, will 201 clearly outperform record PCEs for CIGS²⁰ and PVK²¹ single- 202 junction cells

Provided that in the S-P configuration no current matching is 204 required between the PVK and the CIGS cells, a further 205 increase in the PCE may be achieved by tuning the band gap of 206 either the PVK or CIGS cells. For PVK cells several authors 207 have considered tuning the band gap by changing the relative 208 halide composition and observed an increase in the $V_{\rm oc}$ 209 Although in most cases studied no clear relationship was 210 established between the $V_{\rm oc}$ and the band gap, from the wide 211 band gap tuning study reported by Noh et al.4 one may extract 212 a close to linear correspondence between these two parameters 213 when the band gap is changed from 1.57 to 2.29 eV. The $V_{\rm oc}$ 214 values therein reported are, however, low in comparison to 215 recent reports. For instance, PVK cells with a measured band 216 gap of 1.56 eV have reached a $V_{\rm oc}$ as high as $1.19~{
m V.}^{22}$ Similarly, 217 for a higher band gap PVK of 1.62 eV, a $V_{
m oc}$ of 1.24 eV 3 was 218 recently achieved, which corresponds to a loss in potential 219 (difference band gap and V_{oc}) of 0.39 eV, which is close to the 220 thermodynamic limit. Thus, an increase in $V_{
m oc}$ may be 221 achievable for the other band gaps as well once the charge 222 extraction contacts and the perovskite composition have been 223

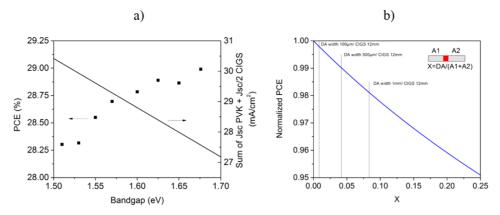


Figure 4. (a) PCE of the CIGS/PVK S-P cells as a function of the PVK band gap (squares) and sum of PVK and CIGS halved short-circuit current densities for the S-P configuration as a function of the PVK band gap (line). (b) PCE as a function of DA width of the serial-connected cells assuming devices with an area of $12 \times 12 \text{ mm}^2$.

224 optimized to overcome limiting effects such as the one 225 described in ref 23. For this work, the above two values are 226 taken as references and used to establish a linear relation 227 between the $V_{\rm oc}$ and the band gap in the range of 1.51 to 1.71 228 eV, shown in Figure 3b. At present, there are material compositions for PVK cells yielding band gaps below 1.56 eV. 13,24,25 However, such compositions have led to suboptimal devices, which were not taken as references for this analysis. A monotonic decrease of the total calculated short-circuit current density as the band gap increases is observed in Figure 4a. Note that we have plotted the sum of the short-circuit current density of the PVK with half of the CIGS to account for the differences 236 in area inherent to the S-P structure. When estimating the PCE 237 using the same single-diode model described above, one observes, as can be seen in Figure 4a, an increase in PCE as the band gap of the PVK increases. Although this may seem 240 counterintuitive provided the band gap of PVK is well above 241 the optimal single-junction solar cell band gap, the explanation 242 is that the band gap increase implies a better voltage matching 243 with the CIGS branch of the circuit.

METHODS

Intercalated Nonperiodic Photonic Structure for Optimal Light Harvesting. To design the specific architecture for the S-P cell, we set the sum of the photocurrents as the target optimization parameter within our numerical calculations.

From Figure 1b, it can be seen that the maximum electrical power *P* derived from the device is given by

$$P = (I_{\text{CIGS}} + I_{\text{PVK}})V_{\text{p}} \times \text{FF}$$
 (1)

253 ,where $I_{\rm CIGS}$ is the short-circuit current generated by the CIGS 254 solar cell and $I_{\rm PVK}$ is the short-circuit current generated by the 255 PVK. $V_{\rm P}$ and FF are the corresponding voltage and fill factor 256 from the parallel connection.

$$I_{PVK} = J_{PVK}(A_T)$$
 (2)

258 with $J_{\rm PVK}$ being the short-circuit current density from the PVK 259 and $A_{\rm T}$ the total illumination area of the device. Neglecting the 260 dead area, $A_{\rm T}$ corresponds to

$$A_{\rm T} = A_1 + A_2 \tag{3}$$

where A_1 and A_2 are the areas for the CIGS subcells (for 262 practical considerations refer to the next subsection). The 263 current flowing through the CIGS cells must fulfill

$$I_{\text{CIGS}} = A_1 J_{\text{CIGS1}} = A_2 J_{\text{CIGS2}} \tag{4}$$

2.66

By simple algebra one may write the PCE as

$$PCE = \frac{[(A_{1}J_{CIGS1}) + (A_{T}J_{PVK})]V_{p} \times FF}{P_{AM1.5}A_{T}}$$
(5) ₂₆₇

We determine the optimal cell architecture by implementing an 268 inverse integration approach where the photocurrent from each 269 one of a large set of possible configuration solutions is 270 numerically computed using a transfer matrix formalism. 271 Configurations are defined by a large set of different input 272 parameters, including thicknesses and/or optical constants for 273 some or all the device layers, which are allowed to vary within a 274 specified range. The target solution is selected from that set of 275 all possible solutions.

When implementing the transfer matrix model, it is assumed 277 that all layers are considered homogeneous and isotropic, such 278 that their optical characteristics can be represented by scalar 279 complex indexes of refraction; interfaces between adjacent 280 layers are parallel and optically flat; incident light is 281 perpendicular to the stack and can be described by plane 282 waves; and the efficiencies in exciton diffusion, charge 283 separation, and carrier transport and collection are wavelength 284 independent. To perform these calculations, one must know 285 the complex index of refraction and thickness for each layer in 286 the architecture. The refractive indexes for each layer were 287 taken from refs 26 and 27 for the CIGS solar cell and from refs 288 28 and 29 for most of the layers in the PVK solar cell.

Using the calculated electric field intensities for the CIGS 290 and PVK, together with their optical properties, the absorption 291 is computed to finally determine the short-circuit current 292 density ($J_{\rm sc}$) for each subcell. $J_{\rm sc}$'s for the PVK and CIGS cells 293 were computed for 10 000 different combinations of layer 294 thicknesses when varying the absorber layers, the entrance 295 transparent electrode, and the intermediate dielectric NDS 296 layers. The thickness ranges were in accordance with the 297 experimental constraints to obtain an optimal FF or $V_{\rm oc}$ for 298 both cells. We assumed an internal quantum efficiency (IQE) 299 of 100% for the PVK cell, as demonstrated in refs 30 and 31 300 and 92% for the CIGS solar cells. The PCE estimates, we 301 obtained the equivalent parallel voltage $V_{\rm p}$ and FF by adding 302

303 the IV curves corresponding to each one of the branches of the 304 circuit.

Experimental Implementation of S-P Structure. The 306 CIGS bottom cells in this design can be deposited in the 307 different ways described in the literature, since there are no 308 special limitations on process temperature or conditions 309 present at this point. In order to reach the desired efficiency, 310 a deposition by co-evaporation or precursor selenization seems 311 to be favorable. The substrate can be rigid (soda lime glass) or 312 flexible (polyimides, steel), especially in view of possible roll-to-313 roll production. The back contact is typically Mo around 500 314 nm. The absorber will be between 2 and 3 μ m thick, followed 315 by a buffer layer (most likely CdS by chemical bath deposition) 316 and a multilayer of intrinsic and doped zinc oxide as TE. The 317 front contact sheet resistance should be below 10 ohm/sq for a monolithic cell interconnection.

Concerning the fabrication of the PVK device on top of the 319 320 CIGS, a number of constraints are introduced to prevent 321 damage of the bottom CIGS layer. For instance, thermal 322 instability has been observed for the n-type buffer layer in CIGS 323 solar cells, where atoms tend to diffuse from the buffer to the 324 absorber layer, significantly reducing the performance of the 325 device by an augmented recombination. Temperatures above 326 300 °C trigger this process and have been shown to be 327 detrimental to all PV parameters of CIGS solar cells.³³ Therefore, annealing processes used to manufacture all the 329 layers within the PVK device are temperature-limited, which 330 excludes the use of frequent configurations such as the 331 mesoscopic PVK on 500 °C sintered TiO2. The PVK layer 332 itself does not require high-temperature annealing and neither does the PCBM. On the other hand, ITO has been shown to provide good-quality transparent electrodes at the bottom or 335 the top of PVK solar cells. 10,34 Connection of PVK cells to 336 modules has been reported in the literature, 35 some with a 337 reduced area loss of 3% of the total 16 cm² module area.³⁶

Similarly to the case of PVK, for the lateral serial connection 339 of the CIGS solar cells three structured patterns denominated 340 P1, P2, and P3 are required, as seen in Figure 1a. The first 341 pattern, P1, and the third pattern, P3, separate the positive and 342 negative contacts of adjacent cells, respectively. The second 343 pattern, P2, allows the negative contact of one cell to be 344 connected with the positive contact of the following cell. The 345 region in between P1 and P3, including both, is the dead area 346 (DA). Typically, laser scribing is employed to define such 347 patterns. If we assume that the dead area corresponds to a 348 fraction X of the total CIGS active cell area, then

$$DA = X(A_1 + A_2) (6)$$

350 Within the optically optimized S-P structure we consider the 351 effect of the DA in the total PCE by normalizing it without 352 neglecting the DA. As can be seen in Figure 4b, under the assumption of a cell larger than 1 cm, the effective loss in PCE 354 is less than 2% even when the separation between P1 and P3, 355 including the width of such laser cuts, is as large as 500 μ m. For 356 separations as small as 100 μ m, which have been recently 357 outperformed in the series connection of CIGS cells, 37 the loss 358 in PCE is negligible.

359 CONCLUSIONS

360 In summary, we proposed and studied a new tandem 361 configuration where the constraint of current density matching 362 is removed. Two CIGS cells laterally connected in series may 363 be connected in parallel with a PVK cell matching the

maximum power point voltages. The configuration studied is 364 monolithic, but the currents from the top PVK and the bottom 365 CIGS cells can be extracted separately. We demonstrated that 366 when the two cells are electrically separated by a dielectric 367 nonperiodic photonic multilayer structure, such photonic 368 structure can be designed to effectively confine the incident 369 sun light above the PVK band gap in the PVK absorber layer. 370 The same photonic structure is used to reach an optimal 371 transparency for sunlight below this band gap and efficiently 372 couple it into the CIGS bottom cell. Using realistic current 373 state-of-the-art parameters for all layers in the multijunction 374 architecture we estimated that PCEs above 28% can be 375 achieved. The PCE of the S-P configuration is for most of the 376 cases studied two percentage points higher in efficiency than 377 that estimated for an optimal series-connected standard tandem 378 configuration or 22.6% and 22.1%, which are the maximum 379 efficiencies reported for CIGS and PVK single-junction cells, 380 respectively. Any improvement in the subcells will further 381 increase the potential efficiency of the S-P cell. For a successful 382 experimental implementation of the S-P configuration, the 383 procedures to fabricate high-quality semitransparent electrodes 384 with the limitations imposed by the layers already deposited 385 must be improved. In addition, the optical optimization of the 386 NDS must be made compatible with its role of preventing 387 short-circuiting between the two subcells.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the 391 ACS Publications website at DOI: 10.1021/acsphoto-392 nics.6b00929.

Curve of equivalent V_{oc} for the S-P configuration when 394 varying the $V_{\rm oc}$ of the PVK subcell or $V_{\rm oc}$ of the CIGS $_{395}$ subcell; fitting data of IV curves; module interconnection 396 schemes for the S-P configuration (PDF)

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