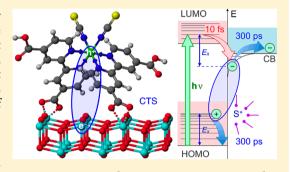
Dynamics of Interfacial Charge Transfer States and Carriers Separation in Dye-Sensitized Solar Cells: A Time-Resolved Terahertz Spectroscopy Study

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Supporting Information

ABSTRACT: Electron injection from a photoexcited molecular sensitizer into a wide-bandgap semiconductor is the primary step toward charge separation in dye-sensitized solar cells (DSSCs). According to the current understanding of DSSCs functioning mechanism, charges are separated directly during this primary electron transfer process, yielding hot conduction band electrons in the semiconductor and positive holes localized on oxidized dye molecules at the surface. Comparing results of ultrafast transient absorption and time-resolved terahertz measurements, we show here that intermediate interfacial charge transfer states (CTSs) are rather formed upon ultrafast injection from photoexcited Ru(II)bipyridyl dye-sensitizer molecules into mesoporous TiO₂ films. Formation



and dissociation of these CTSs were found to strongly depend on their ionic environment and excess excitation energy. This finding establishes a new mechanism for charge separation in DSSCs. It also offers a rationale for the effect of electrolyte composition in liquid-based devices and of ion doping in solid-state solar cells under working conditions.

■ INTRODUCTION

Dye-sensitized solar cells (DSSCs) present a promising way in the quest for cost-effective photovoltaics, owing in particular to their ease of fabrication, high efficiency under low-density illumination, and customizable aesthetics. DSSCs belong to the family of hybrid donor-acceptor bulk heterojunction photovoltaic converter systems. They are based on a mesoporous TiO2 film, constituting the photoanode, onto which a monolayer of a molecular sensitizer is adsorbed. The pores of the dyed nanocrystalline film are filled with either a liquid redox-active electrolyte or a solid-state organic hole transport material.1-

In the absence of a built-in electric field at the interface able to separate photogenerated electron-hole pairs, light-induced charge separation in bulk heterojunction cells relies on the kinetic competition between several forward electron transfer, charge transport, and recombination processes. 4-6 Hence, a detailed investigation of the dynamics of interfacial charge transfer is essential for the understanding of the device functioning mechanism. The kinetics of electron injection from photoexcited heteroleptic Ru(II) bipyridyl dyes into titanium dioxide (TiO₂) semiconductor nanoparticles have been extensively studied. Most of the reports describe the kinetics of the primary light-induced, interfacial charge transfer process as being multiexponential and different rationales for the observed kinetic heterogeneity have been proposed.^{7–12} Although a majority of studies report ultrafast electron injection taking place within a sub-100 fs time scale, recent measurements carried out by using transient IR spectroscopy in the

presence of an electrolyte suggest that charge separation in Ru(II) complex dye-sensitized solar cells might be much slower. 13,14 Results obtained here and their interpretation reconcile these apparently contradicting observations. While initial femtosecond photoinduced electron injection is confirmed, actual charge separation to yield free mobile carriers is shown to be hindered by electron trapping in charge transfer states, whose dissociation in the presence of an electrolyte or in contact with a hole transport material can extend over hundreds of picoseconds.

EXPERIMENTAL SECTION

Ultrafast Time-Resolved Terahertz Spectroscopy **Setup.** The initial pulsed laser beam ($\lambda = 800 \text{ nm}$ wavelength, 45 fs pulse duration, 4.5 mJ/pulse energy, 1 kHz repetition rate) provided by an amplified Ti:sapphire laser (Libra, Coherent) was split into different paths. Approximately 1.4 mJ/pulse was used to pump a white light-seeded optical parametric amplifier (OPA) (OPerA Solo, Coherent), whose wavelength-tunable output constituted the optical pump to photogenerate carriers in the sample. At 590 nm pump wavelength, a typical energy of 8 μ J/pulse was used, corresponding to a maximum energy fluence at the sample of 200 μ J cm⁻²/pulse (5.9 × 10¹⁴ photons cm⁻²/pulse). A second

Received: July 17, 2015 Revised: November 3, 2015 Published: November 4, 2015 part of the beam (0.9 mJ/pulse) produced single cycle terahertz (THz) pulses by optical rectification in a 1 mm thick ZnTe crystal. The THz beam was focused by parabolic gold mirrors on the sample and used as a probe. The transmitted THz pulses were detected in the time-domain through free space electro-optic sampling. A third part of the initial laser beam was used as gating beam and was guided over a delay line onto a 0.5 mm thick ZnTe detector crystal. Birefringence induced in the ZnTe crystal by the electric field associated with the transmitted THz radiation was converted into an intensity modulation using a quarter-wave plate and a Wollaston prism polarizer and finally detected by a pair of balanced photodiodes (Nirvana 2007, New Focus).

Femtosecond Transient Absorption Spectroscopy (TAS) Setup.¹⁶ An amplified Ti:sapphire femtosecond laser (CPA 2001, Clark-MXR) was used as a light source. The laser had a 1 kHz repetition rate, with typical pulse energy of 950 μ J at a fundamental wavelength of 775 nm and pulse duration of 140 fs. Half the source intensity was used for pumping two noncollinear phase-matched OPAs (NOPA Plus, Clark MXR), allowing for the generation of monochromatic pump and probe beams. The probe two-stage NOPA allowed the generation of a near-IR beam, with λ_{probe} = 840 nm. The spectral profile of the pulses was measured using a dual-channel CCD spectrometer (S2000, Ocean Optics). The output pulses of the NOPAs were compressed by SF10 prisms pairs down to sub-50 fs duration. The temporal profile of the pulses was determined using a portable autocorrelator (PulseCheck, A·P·E). The relative path length between the two beams was changed with a delay line, alternatively on the pump or on the probe pulse. The motorized translation stage (M531 PD, Physik Instrumente) allowed time steps of 10 fs and generated delays up to 2 ns. The pump beam was chopped at half the repetition frequency of the pulsed laser source. The change in transmittance of the sample, i.e., change in intensity of the probe beam, was measured by a photodiode placed after the sample and protected by cutoff filters to avoid any scattered light from the pump. The lowamplitude signal was extracted by a lock-in amplifier (SR-830, Stanford Research) referenced to the chopper. To ensure isotropic excitation of the sample, the pump polarization was set at the magic angle (54.7°) relative to the probe pulse with a $\lambda/2$ waveplate. A lens allowed tuning the size of the pump beam before the sample to both decrease the fluence and allow for complete overlap of the probed spot area. The energy fluence on the sample was limited to a maximum of 230 μ J cm⁻²/pulse (6.8 \times 10¹⁴ photons cm⁻²/pulse). Temporal overlap between the pump and probe pulses at the sample position was measured with a Kerr gating technique and gave a typical instrument response function (IRF) of 120 fs.

Sample Preparation. Liquid Electrolyte Samples. $10.2~\mu m$ thick TiO_2 films were prepared by screen-printing a TiO_2 paste on a quartz substrate and sintering at $500~^{\circ}C$, according to a previously described procedure. To increase adhesion of the film onto the substrate, a compact TiO_2 underlayer was applied by spray pyrolysis. The paste consisted of TiO_2 particles with an average diameter of $\sim 20~nm$ dispersed in aqueous ethyl cellulose. The procedure yielded mesoporous transparent films with an average pore diameter of $\sim 26~nm$ and a porosity of 0.68. $(Bu_4N)_2[cis-Ru^{II}(Hdcbpy) (NCS)_2] (N719)^{18}$ was used as dye sensitizer. Derivatization of mesoporous TiO_2 surface was achieved by immersing films overnight in a 0.4~mM N719 solution in ethanol. The solvent or the electrolyte was applied onto the sample just before measurement by placing a quartz

cover slide on the sample and by sucking the solution by capillarity between the dye-sensitized ${\rm TiO_2}$ film and the cover slide.

Solid-State Samples. Samples consisted of 2.3 μ m thick TiO₂ films made by screen-printing a paste, whose particle size and ethyl cellulose content were optimized for solid-state DSSCs.¹⁹ The obtained transparent films contained 22 nm diameter particles and were characterized by an average pore diameter of 28 nm and a porosity of 0.671. Z907 dye (cisbis(isothiocyanato)(2,2'-bipyridyl-4,4'-dicarboxylato)(4,4'-dinonyl-2'-bipyridyl)RuII) was used as a sensitizer and adsorbed onto the film as for the liquid electrolyte samples. 18 The holeconducting material 2,2',7,7'-tetrakis(N,N-di-4-methoxyphenylamino)-9,9'-spirobifluorene (spiro-MeOTAD) was applied onto the films by spin-coating: 40 µL of spiro-MeOTAD solution was deposited onto the sensitized substrates and allowed to infiltrate for 30 s in order to maximize the penetration of the HTM prior to spin-coating for 30 s at 2000 rpm. Three different solutions were prepared: Solution 1 contained only spiro-MeOTAD, solution 2 spiro-MeOTAD and Li+, and solution 3 spiro-MeOTAD and 4-tert-butylpyridine (tBP). 72 mg of spiro-MeOTAD was dissolved in 400 μ L of chlorobenzene by heating the solution at 60 $^{\circ}\text{C}$ for 20 min. In solution 2, 15 μ L of a Li⁺ solution, obtained by dissolving 17 mg of LiTFSI in 100 µL of MeCN, was added to the HTM solution. Solution 3 was obtained by adding 7 μ L of solid tBP to solution 1. Preparation of all solutions and spin-coating were carried out in a dry glovebox in the absence of oxygen. After preparation, samples were stored in the glovebox in the dark until they were measured.

RESULTS

A combination of ultrafast vis/NIR transient absorption and time-resolved THz spectroscopies was applied to scrutinize the details of interfacial photoinduced charge separation in liquid electrolyte- and solid-state hole transport material-based DSSCs. An efficient molecular redox dye-sensitizer (S), (Bu₄N)₂[cis-Ru^{II}(Hdcbpy) (NCS)₂] (N719), was adsorbed onto the surface of nanocrystalline TiO2 transparent films deposited on fused silica plates. The films had their pores filled either with pure acetonitrile (CH3CN) solvent or with an electrolyte used in efficient DSSCs and containing 0.6 mol L⁻¹ iodide (I^-), 0.1 mol L^{-1} triiodide (I_3^-), 0.6 mol L^{-1} 1,3dimethylimidazolium (DMI+), 0.1 mol L-1 lithium cations (Li⁺), and 0.6 mol L⁻¹ 4-tert-butylpyridine (tBP) dissolved in acetonitrile.²⁰ Upon femtosecond laser pulse excitation of the samples, optical absorption of the oxidized state of the dye (S⁺) was promptly observed in the NIR spectral domain with a rise time <50 fs. The optical transient absorbance at λ = 840 nm signal allowed monitoring directly the kinetics of photoinduced electron injection from the MLCT electronic excited state of the dye (S*) into the conduction band of titanium dioxide: Sl $\text{TiO}_2 + h\nu \rightarrow \text{S*}|\text{TiO}_2 \rightarrow \text{S}^+|\text{TiO}_2 + \text{e}^-_{\text{cb}}(\text{TiO}_2)$ (Figure 1A). At a probe wavelength of 840 nm, the decadic molar extinction coefficient of S*, $e^-_{cb}(TiO_2)$, and S⁺ species has values of 2.2 × 10^3 , 1.3×10^3 , and 5.4×10^3 mol⁻¹ L cm⁻¹, respectively. 16,21 Provided that the charge injection quantum yield is larger than 25%, the extinction coefficient values imply that the transient absorption signal is dominated by the dynamics of the products of the charge injection: $\Delta A(S^+ + e^-_{ch}) > \Delta A(S^*)$. A spectrokinetic model was established that allowed to isolate and safely extract the kinetics of appearance of S⁺ species formed upon charge injection from signals measured at 840

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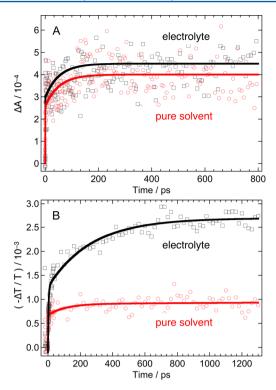


Figure 1. (A) Transient absorbance measured at $\lambda_{\text{probe}} = 840$ nm upon fs laser pulse excitation at $\lambda_{\text{pump}} = 590$ nm of N719 dye adsorbed on transparent mesoporous TiO_2 films. The sensitized films were immersed in pure acetonitrile solvent (red circles) and in a $(\text{I}_3^-/\text{I}^-)$ redox couple-based liquid electrolyte (black squares). Data were fitted with an analytical convolution function of a Gaussian instrument response and two exponential rises (solid line). (B) Relative amplitude change of the transient THz signal transmitted through identical samples upon pulsed excitation at $\lambda_{\text{pump}} = 590$ nm in similar conditions.

nm. ^{16,21} In accordance with previously reported data, sub-50 fs charge injection is believed to take place from nonthermalized singlet ¹MLCT excited states of the dye. ^{7,11,12}

An additional kinetic component with a time constant of 20–50 ps was consistently observed, which accounted for ~20% of the total amplitude of the signal. This significantly slower process was rationalized by charge injection from the relaxed triplet ³MLCT excited state of dye molecules, which underwent intersystem crossing in competition with direct injection from the singlet state. ^{7–9,12} Figure 1A shows that the efficiency and dynamics of both ultrafast components of the injection process appear to be essentially independent of the medium (pure solvent or electrolyte) surrounding the dye, thus confirming previous measurements carried out in our laboratory with a probe wavelength of 860 nm. ⁹ In the presence of iodide-based electrolytes, incident photon-to-current conversion efficiencies measured with solar cells built from similar systems typically exceed 80%, demonstrating that the electron injection quantum yield must be close to unity. ²

Charge injection process kinetics was monitored in identical samples by use of time-resolved THz spectroscopy. Dyesensitized films were excited by optical ultrashort pulses and probed by single-cycle THz pulses. The intensity of the THz transmittance change after optical photoexcitation is proportional to the change of the sample's electric permittivity, which arises from the photogeneration of mobile charge carriers. As the mobility of S⁺ species in the electrolyte and along the

surface is negligible, 24 the relative amplitude change $-\Delta T/T$ of the transmitted THz radiation is directly related to the effective photoconductivity of the TiO₂ film, which is proportional to the product of the density and the mobility of conduction band electrons.

Data reported in Figure 1B show that in pure acetonitrile solvent the time evolution of the THz signal closely matched the dynamics observed in the NIR: A steep absorbance within the duration of the probe pulse was followed by a second kinetic component developing within the first 200 ps and accounting for ~20% of the total signal amplitude. When the film was immersed in the electrolyte solution, the initial amplitude change of the THz signal increased by a factor of 2. A larger photoconductivity of nanoparticles is indeed expected when the real part of the dielectric function of the surrounding medium increases.²⁵ From linear time-domain THz spectroscopy measurements, the dielectric constants of pure acetonitrile solvent and of the electrolyte were evaluated as being $\varepsilon_s = 2.8$ and $\varepsilon_{\rm e}$ = 3.6, respectively, at a frequency of 1 THz, where the difference $\varepsilon_{\rm s} - \varepsilon_{\rm e}$ is maximum. Using the Drude–Smith model for intraparticle conductivity and by applying Maxwell–Garnett effective medium approximation 26,27 to the TiO₂ spherical nanoparticles/liquid composite with a 50:50 volume fraction, an increase by a factor of 2 of the effective photoconductivity of the samples was estimated, in good agreement with the 2-fold enhancement of the observed initial THz transmittance change.²⁸

Figure 1B shows moreover that a further slow growth of the THz absorption was recorded with a half-reaction time of $t_{1/2} \approx 300$ ps. This effect of the electrolyte was not observed in the transient absorption of the oxidized state (S⁺) of the sensitizer (Figure 1A), where the yield and dynamics of the primary light-induced electron transfer process appear invariant. Since the concentration of e^-_{cb}/S^+ pairs produced upon pulsed excitation is constant, the slow rise of the observed THz absorption must then reflect an increase of the charge carriers mobility after they have been injected, rather than the density of injected electrons.

Signals with qualitatively analogous dynamics were observed in the mid-IR (\overline{v} = 1960 cm⁻¹) upon excitation at λ = 540 nm. 13,14 In both reported studies, an ultrafast rise taking place in <200 fs was monitored, which was followed by a slow upsurge of the transient absorbance change on the subnanosecond time scale, typically accounting for more than 80% of the total amplitude. The probed spectral domain was characterized by a broad, featureless absorption spectrum assigned to free conduction band electrons in TiO₂. When two-dimensional infrared (2D IR) spectroscopy is applied to similar dye-sensitized systems, the broad background absorption of electrons disappears to reveal sharp absorption features belonging to the oxidized state of the dye.²⁹ The electron injection kinetics monitored at $\overline{v} = 2000 \text{ cm}^{-1}$ in such conditions for a Re sensitizer is strikingly identical to the one we observe at 840 nm and with THz probe in the absence of electrolyte (Figure 1A). This suggests that the arrival of mobile carriers in the conduction band might be delayed considerably compared to the femtosecond formation of S+ species generated by the injection process and, therefore, that the observation of the appearance of free electrons in titanium dioxide does not necessarily provide direct information on the dynamics of interfacial electron transfer. The delayed emergence of free charge carriers can be rationalized by direct injection of electrons into trap states at the TiO₂/dye interface, which might have their origins in defects or in electrostatic

interactions as in charge transfer states. The mobility of trapped electrons is indeed close to zero and increases to that of free conduction band carriers in TiO₂ nanoparticles upon detrapping.

Figure 2 compares the influence of Li⁺ cations and 4-tert-butylpyridine (tBP), commonly added to I_3^-/I^- redox couple-

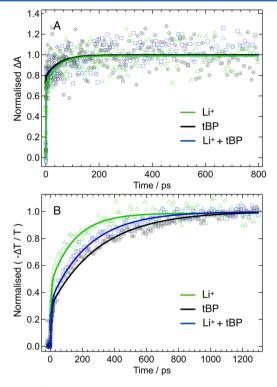


Figure 2. (A) Time course of the transient absorbance measured at $\lambda_{\text{probe}} = 840$ nm following fs laser pulse excitation at $\lambda_{\text{pump}} = 590$ nm of N719 dye-sensitized mesoporous TiO₂ films immersed in solutions of 1,3-dimethylimidazolium iodide (DMII, 0.6 mol L⁻¹) and iodine (0.1 mol L⁻¹) in CH₃CN with lithium iodide (LiI, 0.1 mol L⁻¹) (green triangles), 4-tert-butylpyridine (tBP, 0.5 mol L⁻¹) (black circles), or a mixture of both LiI (0.1 mol L⁻¹) and tBP (0.5 mol L⁻¹) (blue squares) added. (B) Relative amplitude change of the transient THz signal transmitted through identical samples upon pulsed excitation at $\lambda_{\text{pump}} = 590$ nm in similar conditions. Traces were normalized at the plateau value extrapolated from the fitting of experimental points at time delays \geq 1.3 ns.

based electrolytes in DSSCs. Figure 2A shows that the yield and dynamics of the ultrafast electron photoinjection, monitored by the appearance of the S⁺ species characteristic optical absorbance, are not sensitive to the additives. In contrast, the dynamics of THz transmittance changes (Figure 2B) consist in a fast rise, essentially taking place during the probe pulse, and a second slower component extending on several hundreds of picoseconds. The latter kinetic component is strongly affected by the various additives. The slow component observed in the presence of tBP represents 70% of the total transmittance change and is characterized by a half-reaction time $t_{1/2} = 350$ ps. Addition of Li⁺ cations does not affect significantly the amplitude of the slow component, but it clearly shortens the reaction time.

Similar influence of tBP and Li⁺ species contained in the electrolyte upon THz signal dynamics was previously reported.²⁷ This was assigned to a shift of the flat-band potential of the oxide induced by the adsorption of potential-

controlling ions on its surface, which was expected to affect the kinetics of charge injection. The latter interpretation, however, is in contradiction with our observation in the NIR of the appearance of the dye-oxidized state S⁺, whose rate is unaffected by the electrolyte composition (Figure 2A). Adsorption of Li⁺ cations onto the surface of TiO₂ produces a downward shift of the conduction band energy of the semiconductor material. An opposite effect is expected for tBP, whose role in the DSSCs is precisely to neutralize protons released at the surface by the dye upon its adsorption and raise the oxide conduction band edge energy.³⁰ Though both additives mixed with a large concentration of iodide in the electrolyte yielded a similar effect on the THz signal compared to the pure solvent, which was to add a slow component in the hundreds of picoseconds time domain and to markedly increase the transient conductivity.

In a quest of a new rationale for the observed effect of the electrolyte, we performed time-resolved THz photoconductivity measurements in the conditions of Figure 1B with various pump wavelengths ranging from 590 nm down to 420 nm. Figure 3 displays the obtained results restricted to a selection of

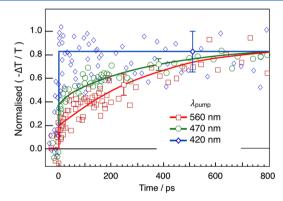


Figure 3. Relative amplitude change of the transient THz signal transmitted through N719 dye-sensitized TiO₂ samples in a (I₃ $^-$ /I $^-$) redox couple-based liquid electrolyte upon pulsed excitation at $\lambda_{\rm pump} = 560$ nm (squares), 470 nm (circles), and 420 nm (diamonds). Traces were normalized to the plateau value extrapolated from fitting curves at a delay time of 800 ps.

three different pump wavelengths. For $\lambda_{\rm pump}=560$ nm, a fast initial part amounting to ca. 25% of the total signal was followed by a slower rise of the photoconductivity, whose time constant was $t_{1/2}=350$ ps. At shorter pump wavelengths, the fast initial part was observed to increase, reaching 50% of the total signal at 470 nm and practically 100% for $\lambda_{\rm pump} \leq 450$ nm. At large excitation wavelengths, a majority of injected electrons seem to be primarily localized and acquire their mobility only within hundreds of picoseconds in the presence of an electrolyte. Conversely, larger excess excitation energies appear to produce an increasing initial proportion of mobile carriers, eventually reaching a situation where no slow rise of the mobility in the subnanosecond time scale takes place anymore. Blank experiments carried out with bare TiO₂ films pumped at $\lambda \geq 420$ nm in identical conditions yielded no THz absorption signal.

In functional DSSCs containing I₃⁻/I⁻ redox couple-based liquid electrolytes, holes are transferred to iodide anions in the microsecond time scale and do not interfere then with subnanosecond carriers dynamics. This picture is complicated in solid-state dye-sensitized solar cells (ss-DSSCs), where the

liquid electrolyte is replaced by spiro-MeOTAD, an amorphous solid organic hole transport material (HTM) filling the pores of the dye-sensitized nanostructured TiO₂ film.³¹ At the hybrid organic-inorganic heterojunction, hole injection from the dye oxidized state into the HTM, $S^+|TiO_2| + HTM \rightarrow S|TiO_2| +$ h⁺(HTM), takes place in the picoseconds time scale, typically 4-5 orders of magnitude faster than in liquid electrolytes.³² Solid-state samples were prepared by adsorbing the dye cisbis(isothiocyanato)(2,2'-bipyridyl-4,4'-dicarboxylate)(4,4'-dinonyl-2'-bipyridyl)Ru^{II} (Z907) on mesoporous TiO₂ films.³³ Z907 is spectroscopically and thermodynamically similar to the N719 carboxylated derivative used in liquid electrolyte-based systems. The hydrophobic dinonyl aliphatic moiety of Z907 allows for a better wetting of the dye-sensitized surface by the organic HTM and the formation of a more conformal heterojunction.³⁴ Hence, its use is preferred to that of N719 dye in the fabrication of efficient solid-state DSSCs. Upon pulsed laser excitation, solid-state samples displayed a steep rise of transient absorption observed at short times, corresponding to the appearance of the oxidized state of the dye produced by ultrafast electron injection (Figure 4A). In the subnanosecond time domain, a decay of the signal was observed in the presence of HTM, which is assigned to the regeneration of the dye ground state by injection of a hole into the organic material. At the probe wavelength of 840 nm, the dye excited state S*, the

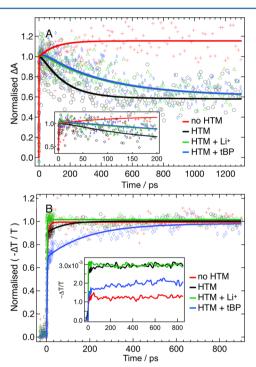


Figure 4. (A) Hole injection dynamics probed at $\lambda_{\text{probe}} = 840$ nm of solid-state samples following femtosecond laser pulse excitation at $\lambda_{\text{pump}} = 590$ nm of the amphiphilic ruthenium complex sensitizer *cis*-RuLL'(SCN)₂ (L = 4,4'-dicarboxylic acid-2,2'-bipyridine, L' = 4,4'-dinonyl-2,2'-bipyridine) (Z907). The pores of the dye-sensitized nanocrystalline TiO₂ films were left empty (red crosses), filled by spin-coating with pristine *spiro*-MeOTAD (black circles) or filled with the HTM doped with lithium bis(trifluoromethanesulfonyl)imide (LiTF-SI, 15 mol %) (green triangles) or tBP (81 mol %) (blue diamonds). (B) Normalized transient THz transmittance changes of the same samples upon pulsed excitation at $\lambda_{\text{pump}} = 590$ nm in similar conditions. The inset shows the relative amplitude changes for the investigated systems before normalization.

oxidized state S⁺, conduction band electrons e⁻_{cb}, and the oxidized state of the hole transport medium h⁺(HTM) all contribute to the transient absorbance signal. Here again, the respective contributions were unravelled using a model including the kinetic heterogeneity of electron injection. Fitting of experimental data then yielded a fast regeneration time constant $\tau = 153$ ps ($t_{1/2} \approx 100$ ps) for pristine *spiro-MeOTAD*, $\tau = 370$ ps for Li⁺-doped HTM, and $\tau = 384$ ps ($t_{1/2} \approx 270$ ps) for tBP-containing samples.

Similar to the effect of the electrolyte, filling the pores of the nanostructured TiO2 film with the organic hole-conducting material produced a marked increase by a factor 2.3 of the transient THz conductivity signal measured at the end of the probe pulse (Figure 4B). An additional slow component accounting for approximately 10% of the signal was also observed with a half reaction time $t_{1/2} \approx 100$ ps, concomitant with the hole injection into the HTM. Blank experiments demonstrated that the transient photoconductivity signal is not due to holes in the HTM, whose mobility by hopping mechanism is practically null at THz frequency, but only to conduction band electrons in TiO2. Figure 4A shows that Li+ and tBP species inserted in the organic hole transport material yielded identical hole injection rates. The same two additives, however, had a profoundly different effect upon the THz signal dynamics (Figure 4B). In the presence of Li⁺ cations, the photoconductance reached immediately its maximum value, and no slow component was observed. On the contrary, addition of tBP to the spiro-MeOTAD limited the initial THz photoconductivity to a level comparable to the system deprived of HTM. A slow rise component was observed in this case, which corresponded to an increase by 50% of the average mobility of photoinjected electrons with a time constant $t_{1/2} \approx$ 200 ps.

DISCUSSION

Ultrafast optical pump-probe transient absorption measurements monitoring the dye sensitizer's oxidized state yield an evaluation of the population of electrons photoinjected in TiO₂ upon laser pulsed excitation. Besides, time-resolved THz spectroscopy experiments provide a direct estimate of the photoconductivity of the nanocrystalline oxide material, which stands for the product of the injected conduction band electrons' density and average mobility. Combining both techniques by applying them on the same samples allows then extracting the time evolution of the carrier's mobility. In liquid junction systems, a marked increase of the carrier mobility by a factor of ~2 is observed upon addition of electrolyte species in the solvent. This indicates that about half of the photoinjected electrons get trapped within the first picosecond following ultrafast interfacial electron transfer from the dye's excited state into the conduction band of TiO2. The upsurge of the mobility of conduction band electrons in the subnanosecond time scale, long after they have been injected into the oxide semiconductor, the ability of an electrolyte and in particular that of small cations such as Li⁺ to detrap electrons, and the effect of the excitation wavelength point toward the formation and dissociation of charge transfer states (CTSs) at the interface. According to this interpretation, Coulomb interaction between injected electrons trapped at the surface of the solid and positive charges represented by adsorbed oxidized dye species S+ prevent part of the injected charge carriers from moving during the first picoseconds and from contributing to the THz photoconductivity. In the

presence of a concentrated electrolyte, bound electron—hole pairs (e^-_{cb} ····S⁺) formed across the interface dissociate upon polarization reorganization of the ionic environment in the subnanosecond time scale, screening the Coulomb interaction and diminishing the CTSs binding energy. Freed conduction band electrons can then contribute to the photoconductivity of TiO_2 nanoparticles, whose THz absorption increases markedly. Although acetonitrile is a polar solvent, it is apparently unable to dissociate charge transfer states, which appear in contact with the pure solvent to be remarkably stable for more than a nanosecond. A large excess excitation energy provided by optical pumping at short wavelength appears to prevent the formation of CTSs or speed up considerably their dissociation, presumably because it exceeds the binding energy of the electron—hole pair in the charge transfer state.

This rationale based on the formation and dissociation of charge transfer states is able to explain the effect of Li+ and tBP additives in the liquid electrolyte as well as in the solid organic hole transport material. In liquid junction systems, CTSs formed upon ultrafast photoinduced charge injection dissociate in the subnanosecond time scale only in the presence of an electrolyte. This dissociation is faster for small Li⁺ cations able to quickly rearrange on the surface and effectively screen trapped electrons. The charge-screening kinetics, whose estimated rate constant is here $k = 3 \times 10^9$ s⁻¹, appears to be 10⁵ times faster compared to the value obtained from spectroelectrochemical and Stark spectroscopy measurements applied to a comparable system.³⁵ This shows that local reorganization of ions adsorbed to the surface, which allow for the liberation of electrons trapped in CTSs is much faster that the macroscopic reorganization of the electrolyte and solvent molecules around dye-sensitized TiO2 nanoparticles. Addition of tBP does not provide supplementary ions in the electrolyte. Conversely, this additives' role is to capture protons that could be present at the surface.³⁰ As protons are expected to have the same effect as lithium cations and promote the dissociation of CTSs, the effect of tBP opposes to that of Li⁺ by slowing down the charge separation at the interface.

In solid-state samples containing pristine spiro-MeOTAD hole transport material, H⁺ cations freed during the adsorption of the protonated Z907 dye sensitizer are the only ionic species that could be present at the surface of TiO₂. Contrary to N719 dye, the anchoring carboxylic groups of Z907 complex prior to adsorption are protonated. A larger concentration of H⁺ associated with the surface is therefore expected in ss-DSSCs compared to the liquid electrolyte-based systems. In the presence of HTM, localization of photoinjected electrons is practically hindered with a large initial photoconductivity signal and only a small fraction of trapped electrons being released in the picosecond time scale. The latter slow contribution is totally suppressed upon addition of Li⁺ cations. tBP-doping of spiro-MeOTAD reduces ions presence at the interface by neutralizing adsorbed protons. As a consequence, about half of the photogenerated electrons are initially trapped at the interface and freed later, concomitantly with hole injection into the HTM, with a time constant $t_{1/2} = 180$ ps. This suggests that hole injection dynamics is the cause of the observed slow increase in the THz mobility of injected electrons rather than its consequence.

Charge transfer states and charge transfer excitons are well-established concepts in organic photovoltaics. A lot of efforts are currently devoted to fully understand the formation and dissociation of CTSs. ^{9,36–40} At organic heterojunctions, the low

dielectric constant of materials favors the formation of CTSs, and electron-hole binding energies of the order of 250 meV $(\sim 10kT)$ are typical.⁴¹ At hybrid interfaces, involving a low dielectric constant inorganic metal oxide such as ZnO ($\varepsilon_{r,GHz}$ = 3.7),⁴² evidence has been found that electron injection proceeds over an intermediate, electrostatically bound state. 43-45 Our present work provides evidence that CTSs can as well exist in DSSCs at the surface of TiO2, a material characterized by a strong relative permittivity ($\varepsilon_{r GHz} = 17$).⁴⁶ This finding is consistent with a previously suggested recombination mechanism occurring via surface Coulomb traps. 47,48 It shows that the bulk dielectric function of the solid is not appropriate to predict the screening of electrostatic interactions between electrons trapped in surface states and holes localized on neighboring oxidized adsorbed dye molecules.

A new picture of a charge injection mechanism involving the formation and dissociation of charge transfer states is sketched in Figure 5. Following absorption of light, vibrationally

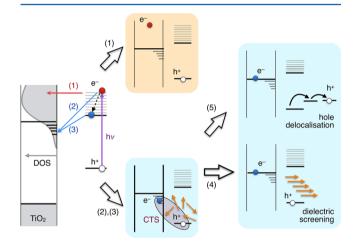


Figure 5. Scheme of energy levels and processes involved in charge transfer states formation and dissociation in DSSCs. An explanation is provided in the text.

unrelaxed excited dye sensitizer molecules optimally coupled to the TiO2 surface undergo electron injection in the conduction band of the solid. This process takes place on a < 50 fs time scale and competes kinetically with vibrational relaxation. 11,12 Injection from hot molecular excited states can proceed isoenergetically, yielding hot electrons in the conduction band, whose kinetic energy carry away the excess optical excitation energy (Figure 5, process 1). Alternatively, electrons can be injected directly at the bottom of the conduction band or into underlying trap states (Figure 5, process 2). In the first case, a hot electron would thermalize to the conduction band bottom edge within a few hundreds of femtoseconds, but the probability of being trapped into a surface state at the immediate proximity of the oxidized dye molecule is obviously low. In the second case, the excess excitation energy is converted into vibrational excitation of the dye cation, and hence, the localized injected electron might be trapped in an interfacial charge transfer state. The same result would be obtained if electron injection occurs from vibrationally relaxed singlet ¹MLCT state of the sensitizer or from a triplet ³MLCT state yielded by ultrafast intersystem crossing (Figure 5, process 3). ^{49,50} Since the excess excitation energy in this case is reduced, the formation of a CTSs between

correlated electrons and dye cations should be more probable. The binding energy of excitons is expected to be sensitive to changes in the surrounding charge distribution. Here we propose that CTSs are split by a local redistribution of ionic charges and dipoles (Figure 5, process 4) present at the surface or by fast hole injection of the dye cation into an HTM (Figure 5, process 5).

The geometry of the dye-sensitizer molecule adsorbed on the surface is expected to influence the distance at which injected electrons and holes localized in the oxidized dye molecule interact and, hence, modulate the strength of the Coulomb attraction between them. For N719 dye anchored on ${\rm TiO_2}$ anatase (101) facet, the e⁻—h⁺ minimum separation distance is ca. 7 Å (Figure 6). Assuming an average dielectric constant of the intra-CTS medium $\varepsilon_{\rm r}=3$, an upper limit for the binding energy $E_{\rm b}=0.7$ eV is estimated from the Coulomb law.

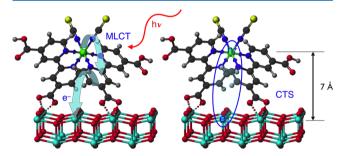


Figure 6. Ball-and-stick molecular model of N719 dye-sensitizer adsorbed on TiO₂ anatase (101) surface. Ultrafast electron injection into the solid oxide takes place upon photoexcitation of the Ru(II) complex by a metal-to-ligand charge transfer (MLCT) transition. Provided the excess excitation energy is not too large ($\Delta E_{\rm ex}$ < 1 eV, $\lambda_{\rm pump} \geq$ 470 nm), a charge transfer state (CTS) is formed through Coulombic interaction between the positive charge (h⁺) located within the HOMO of the oxidized dye cation and the injected electron (e[−]) trapped at a neighboring, uncoordinated surface Ti(IV) site, with an e[−]−h⁺ separation distance ≥7 Å.

Hot CTS are produced naturally by encompassing the excess excitation energy $\Delta E_{\rm ex}$ of the dye defined as $\Delta E_{\rm ex} = h\nu - \Delta E_{0.0}$ $-\Delta G_{\rm ini}^0$, where $h\nu$ is the absorbed photon energy, $\Delta E_{0.0}$ the excitation energy of the dye at its absorption threshold, and ΔG^0_{ini} the standard free energy for charge injection from the lowest excited state of the sensitizer to the bottom edge of the conduction band of the acceptor material. The absorption threshold of adsorbed N719 dye is at a wavelength of 750 nm corresponding to $\Delta E_{0.0} = 1.65$ eV. For N719 photoexcited at λ_{ex} = 470 nm, in conditions where approximately half of the photoinjected electrons appear to be trapped in CTS (Figure 3), and assuming $\Delta G^0_{\rm inj} = 0.1$ eV, a value $\Delta E_{\rm ex} = \sim 1.1$ eV is calculated. This excess excitation energy largely exceeds the Coulombic binding energy of charge transfer states. However, it closely matches the CTS dissociation energy E_d calculated as the sum $E_{\rm d} = E_{\rm b} + E_{\rm t} = \sim 1.1$ eV, where $E_{\rm t} = \sim 0.4$ eV is the energy depth of electron trap states at the surface of TiO2 nanoparticles.51

CONCLUSION

Ultrafast optical transient absorption and time-resolved terahertz spectroscopies were applied to the photoanode of dye-sensitized solar cells based on Ru(II) complexes adsorbed on mesoporous titania films. The combination of both techniques allowed monitoring the time evolution of the

mobility of photoinjected electrons in TiO2. Provided the excess excitation energy of the dye sensitizer was not too large $(\lambda_{\rm exc} \ge 450 \text{ nm})$, a significant part of the injected carriers was found to be trapped within a subpicoseconds time scale. Detrapping of electrons with a time constant of several hundreds of picoseconds was observed to take place in the presence of ionic species or upon positive charge transfer to a solid-state hole transport material. These observations are rationalized by the formation and dissociation of charge transfer states, which are arising from the Coulomb interaction between electrons trapped at the surface of titania and positive charges located on oxidized dye cations. Irradiation of sensitizer molecules in the blue part of their absorption spectrum ($\lambda_{\rm exc}$ < 450 nm) results in a large excitation excess energy, which prevents the trapping of photoinjected electrons or causes the prompt dissociation of charge transfer states on the subpicoseconds time scale.

Results obtained here and conclusions drawn from them reconcile what appeared so far as contradictory reports regarding charge injection kinetics in dye-sensitized TiO2 and provide a new, consistent interpretation of previous observations. They highlight the importance of dielectric screening by ions in carrier separation, which is of particular importance in solid-state dye-sensitized solar cells, where the hole transport medium does not intrinsically contain charges. They shed new insight into the possible mechanism of action of dopants, such as in particular Li⁺ and H⁺ cations. They show that the existence of charge transfer states is a common property of liquid electrolyte-based and solid-state dye-sensitized solar cells. Finally, they suggest a unified picture for free carriers generation and separation in OPV and hybrid organicinorganic photovoltaic systems using wide-bandgap semiconductors, where the involvement of interfacial charge transfer excitons and charge transfer states represents a paradigm shift.

ASSOCIATED CONTENT

S Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.jpcc.5b06911.

Details on femtosecond transient absorption data treatment, confidence intervals of the fitting of transient absorbance experimental points, and blank measurements to assess the contribution to the THz absorption signal of holes in *spiro-MeOTAD HTM (PDF)*

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Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

Funding of this work has been provided by the Swiss National Science Foundation (grant no. 200021_157215) and the National Center of Competence in Research program "Molecular Ultrafast Science and Technology" (NCCR-MUST), a research instrument of the SNSF. The authors

thank Amalie Dualeh for help in the preparation of the solidstate samples, Joël Teuscher for fruitful discussions, and Pascal Comte and Shaik M. Zakeeruddin for providing TiO₂ pastes, dyes, and electrolytes used in this study.

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Supporting Information

Dynamics of Interfacial Charge Transfer States and Carriers Separation in Dye-Sensitized Solar Cells A Time-Resolved THz Spectroscopy Study

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1. Femtosecond transient absorption data treatment

The extinction coefficients of all the species contributing to transient absorbance signals observed at the probe wavelength were determined and were used in a spectro-kinetic model to isolate the dynamics of the appearance of dye oxidized state S^+ and, hence, the kinetics of the charge injection from the excited state of the dye S^* to the conduction band of TiO_2 (Eq. 1)

$$S \mid TIO_2 + h\nu \rightarrow S^* \mid TiO_2 \rightarrow S^+ \mid TiO_2 + e^-_{cb}(TiO_2)$$
 (1)

The probe wavelength λ_{probe} = 840 nm was chosen in order to observe the oxidized state S⁺ of the dye with minimal contribution from the excited state S* (Figure S1).

$$S^{+} \mid TiO_{2} + HTM \rightarrow S \mid TiO_{2} + h^{+}(HTM)$$
 (2)

In solid-state dye-sensitized solar cells, charge transfer to the hole transporting material (Eq. 2) is orders of magnitude faster than the dye regeneration by common redox mediator used in liquid electrolytes. The deconvolution of the dynamics of processes (1) and (2), therefore, is more critical. The procedure for the fitting of the femtosecond transient absorption signals is described in refs. 16 and 21.

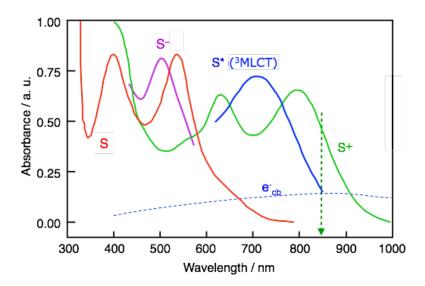


Figure S1. Relative absorbance spectra of the various species involved in the charge injection process according to the values of extinction coefficients determined in ref. 21. S represents here the amphiphilic Ru^{II} complex dye Z907. Spectra of the parent dye complex N719 in comparable conditions is practically identical. The probe wavelength of 840 nm was selected, as it ensures the largest contribution to the absorbance signal of the S $^+$ species compared to S * and e $^-$ cb.

2. Confidence intervals of the fitting of transient absorbance data

95% confidence interval bands drawn here below in Figure S2 show that the fitting of noisy experimental points does not suffer much uncertainty and provides a clear picture of the time evolution of the transient absorbance monitored at 840 nm.

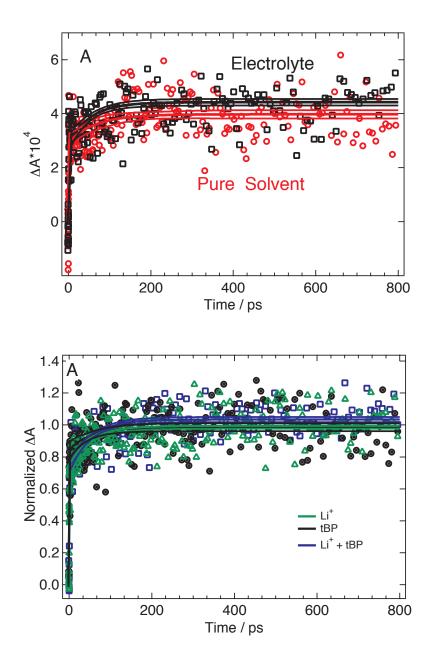


Figure S2. Experimental points reported in Figure 1A (top) and 2A (bottom). The limits of 95% confidence intervals bands are added here on both sides of the fitting curves.

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3. Contribution of holes in spiro-MeOTAD HTM to the THz absorption signal

Blank measurements were performed on samples composed of mesoporous $Al_2O_3 \mid 2907 \mid spiro$ -MeOTAD (Figure S3). Fast hole transfer from Z907 excited state to the HTM along a reductive quenching pathway is expected to be quite effective in this case, where no electron injection into the oxide can take place. As we fail to observe any THz signal, we conclude that holes injected in the oxidized HTM, whose transport takes place through a hopping mechanism, are characterized by a negligible conductivity. This demonstrates that the transient photoconductivity signal recorded in solid-state dye-sensitized systems is not due to holes in the HTM but only to conduction band electrons in TiO_2 .

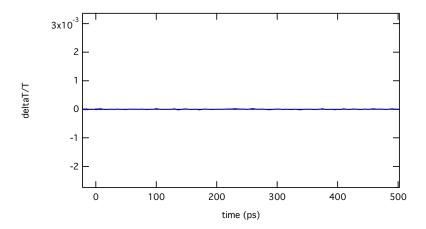


Figure S3. Transient THz transmittance change of a solid-state sample following fs laser pulse excitation at $\lambda_{pump} = 590$ nm of the amphiphilic ruthenium complex sensitizer *cis*-RuLL'(SCN)₂ (L = 4,4'-dicarboxylic acid-2,2'-bipyridine, L' = 4,4'- dinonyl-2,2'-bipyridine) adsorbed on a mesoporous Al₂O₃ film. The pores of the dyed nanocrystalline alumina scaffold were filled with pristine *spiro*-MeOTAD HTM by spin-coating. Other conditions are identical to those used to produce data of Figure 4B.

S4