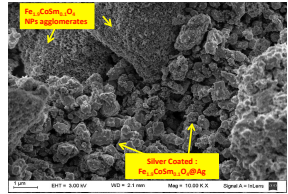


Piezo and Magneto Resistance of Magnetically Oriented Dispersions of Magnetic Particles Covered with Silver in Bulk and Screen-Printed Elastomer Polymers

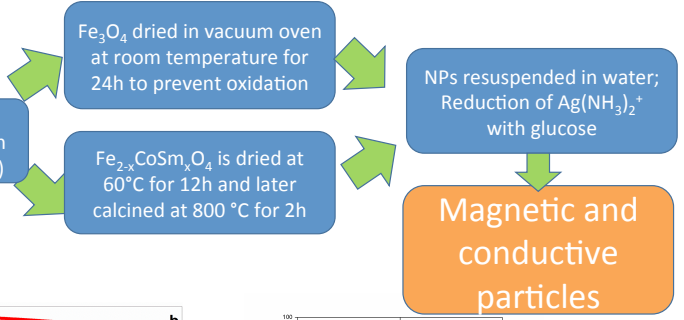
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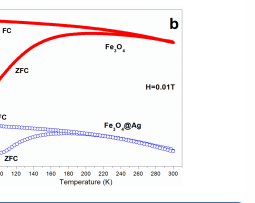
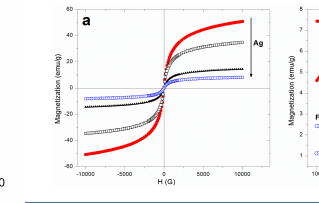
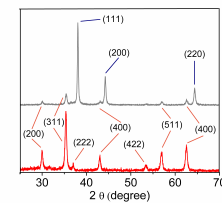
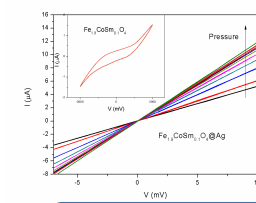
Step 1 Synthesis and characterization of particles, simultaneously magnetic and conductive



Inorganic precursors Fe(III), Fe(II), Co(II) Sm(III) in acidic solution → coprecipitation of mixed oxides in alkaline solution (Fe_3O_4 and $Fe_{2-x}CoSm_xO_4$)



Silver Coating. Left: SEM imaging of $Fe_{2-x}CoSm_xO_4$ and silver coated particles. The uncoated ferromagnetic nanoparticles tend to aggregate due to magnetic dipole interactions. The formed silver coated particles are micrometrical size aggregates

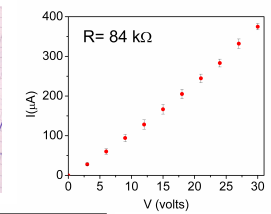
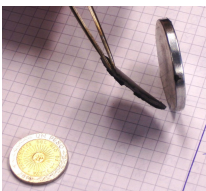
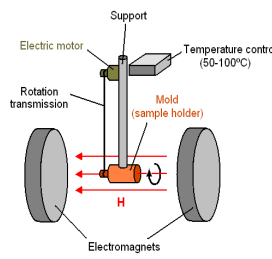
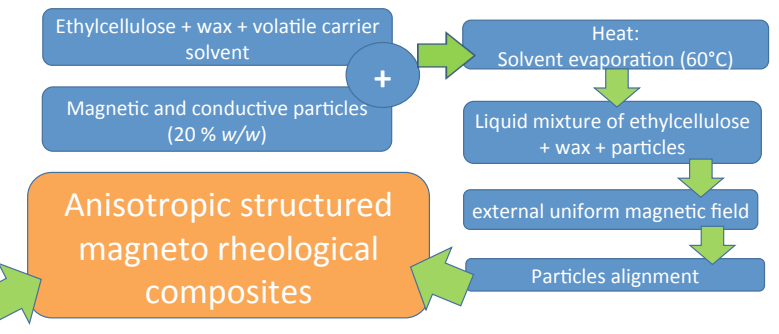
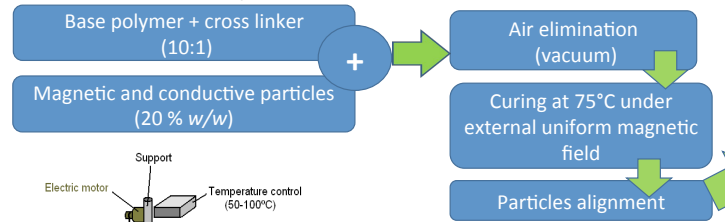


Left: Cyclic voltammograms for ferromagnetic $Fe_{3-x}CoSm_xO_4$ and for $Fe_{2-x}CoSm_xO_4@Ag$. The conductivity of the particles increases when pressure is applied.
Right: XRD patterns for Fe_2CoO_4 (—) and $Fe_2CoO_4@Ag$ (—)

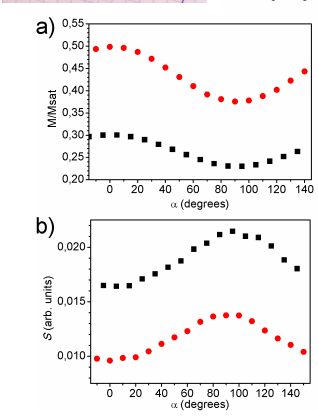
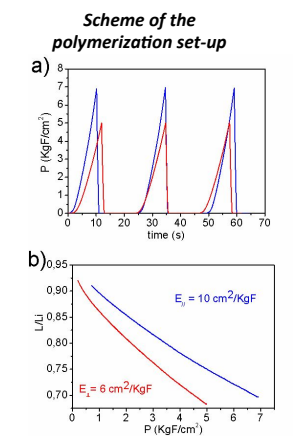
VSM and SQUID (a) Magnetic hysteresis curves at 25 °C of Fe_3O_4 nanoparticles (—●—) and $Fe_{2-x}CoSm_xO_4@Ag$ microparticles synthesized in different conditions: $AgNO_3$ at 50°C for 20 hours (—□—), $[Ag(NH_3)_2]^+$ at room temperature for one hour (—▲—) and $[Ag(NH_3)_2]^+$ at 50°C for one hour (—○—) (b) ZFC and FC magnetization vs. Temperature curves ($H=0.01T$) for the Fe_3O_4 nanoparticles (—●—) and $Fe_2O_4@Ag$ particles (—○—)

Effect of Sm^{3+} doping and heat treatment: VSM loops, the values of M_{sat} , M_{rem} and H_c are affected by Sm^{3+} proportion and heat treatment (800 °C for 2 hours).

Step 2 Elaboration of composites: dispersion of particles in organic matrix (PDMS or ethylcellulose + waxes)



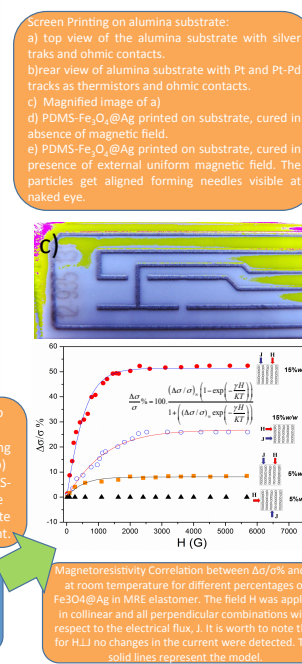
I vs. V plot of $Fe_2CoO_4@Ag$ -PDMS (20% w/w) film (shown at the left) for a pressure of 0.5 KgF/cm²



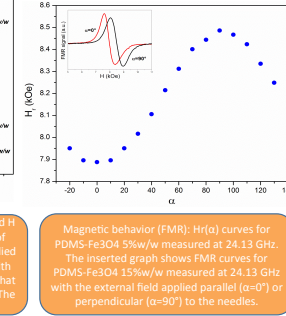
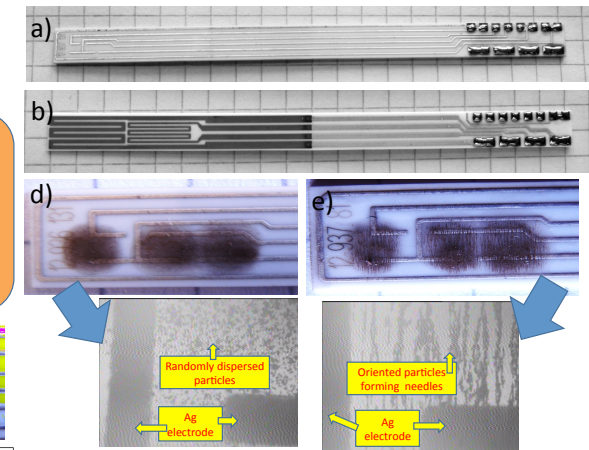
Texture analysis of PDMS-NPs composites: a) slow compression, fast decompression (strain 30%) on PDMS-NPs (10% w/w, $x=0.2$, not calcined) composite. b) Relative thickness (L/L_0) vs. applied pressure (P) for PDMS-NPs (10% w/w, $x=0.2$, not calcined) composite. (—) pressure applied parallel to the needles. (---) pressure applied perpendicular to the needles.

a) Remanence magnetization (normalized to $M_{s,0}$) vs angle respect H (α) for PDMS- $Fe_{1.9}CoSm_{0.104}$ composites (10% w/w) using particles with and without heat treatment. b) Magneto viscosity as a function of α for PDMS- $Fe_{1.9}CoSm_{0.104}$ composites. (*) Composite prepared with calcined particles. (•) Composite prepared with particles before heat treatment.

The total resistance of the composite is considered as two resistors in series: $R_t = R_i + R_a$. The first resistance R_i is an intrinsic inter-grain resistance which is an activated process present in any polycrystalline material (dependent on temperature but not on H). R_a is an activated process (with an associated conductivity $R_a^{-1} = 1/\sigma_a$) that accounts the effect of matching partially the spin polarizability of both grains by spin-alignment through the presence of H .



Step 3 Screen printing on alumina substrate with the fluid mixture



$Fe_3O_4@Ag$ -PDMS composites present magneto resistant properties. Magneto resistant studies of $Fe_2CoO_4@Ag$ -PDMS composites are currently in progresss.

Magneto-resistivity Correlation between $\Delta\sigma/\sigma$ and H at room temperature for different percentages of $Fe_3O_4@Ag$ in MR-E elastomer. The field H was applied in collinear and all perpendicular combinations with respect to the electrical flux J . It is worth to note that for $H \perp J$ no changes in the current were detected. The solid lines represent the model.