PHYSICAL REVIEW B 87, 180407(R) (2013)

Size-induced superantiferromagnetism with reentrant spin-glass behavior in metallic nanoparticles of TbCu₂

C. Echevarria-Bonet, ¹ D. P. Rojas, ² J. I. Espeso, ¹ J. Rodríguez Fernández, ¹ L. Rodríguez Fernández, ³ P. Gorria, ⁴ J. A. Blanco, ⁴ M. L. Fdez-Gubieda, ⁵ E. Bauer, ⁶ G. André, ⁷ and L. Fernández Barquín ¹ Departamento CITIMAC, Universidad de Cantabria, 39005 Santander, Spain ² Departamento de Física e Instalaciones Aplicadas-ETSAM, 28040 Madrid, Spain ³ SERMET, Universidad de Cantabria, 39005 Santander, Spain ⁴ Departamento Física, Universidad de Oviedo, 33007 Oviedo, Spain ⁵ Depto. Electricidad y Electrónica, Universidad del País Vasco (UPV/EHU), 48940 Leioa and BCMaterials, Building No. 500, Technological Park of Biscay, 48160 Derio, Spain ⁶ Institut für Festkörperphysik, Technische Universität Wien, 1040 Wien, Austria ⁷ Laboratoire Léon Brillouin, CEA-SACLAY, 91191 Gif-sur-Yvette Cedex, France (Received 24 January 2013; revised manuscript received 3 May 2013; published 24 May 2013)

An unusual 4f-superantiferromagnetic state characterized by simultaneous antiferromagnetic and spin-glass behaviors induced by particle-size reduction is revealed in metallic nanoparticles (≈ 9 nm) of TbCu₂. The Néel temperature is 46 K and the glassy freezing is below ≈ 9 K and governed by a critical slowing down process. Neutron diffraction at 1.8 K establishes the superantiferromagnetism. The latter is settled by the nanoparticle moments and the freezing mechanism is provided by the surface spins.

DOI: 10.1103/PhysRevB.87.180407 PACS number(s): 75.10.Nr, 75.25.-j, 75.50.Lk, 75.75.-c

The family tree of magnetism is formed by several archetypal arrangements, which are manifestations of the more general physical order/disorder behavior found in various systems such as liquid crystals and superconductors. Indeed, antiferromagnetism (AF) and the glassy spin (spin glass, SG) state have been thoroughly described in numerous materials. ¹⁻³ Other branches of such a tree are based on mictomagnetism (cluster glass) and superparamagnetism, the latter associated with magnetic nanoparticles. ¹⁻⁵ The advent of new production techniques and research in new materials have spawned a variety of behaviors, consequences of competing magnetic interactions. In this sense, ferromagnets with reentrant state, ^{1,2,4,6} superferromagnets, or superspin glasses ⁷⁻⁹ display new spin dispositions derived from the magnetic interactions.

The coexistence of ferromagnetism (FM) and spin glass has been evidenced through the Curie transition and a further SG freezing at lower temperatures. This finding could be observed in either crystalline^{1,4} or amorphous materials.^{6,10} Surprisingly, that coexistence has been sparsely reported for the AF-SG arrangements. 11-15 Reports on Ising Fe_{0.55}Mg_{0.45}Cl₂ and Fe_{0.6}Mn_{0.4}TiO₃¹¹ compounds, YBa₂(Cu_{0.88}Fe_{0.12})₃O_{6.5} semiconductors, 12 insulating materials of Co₂(OH)(PO₄), and Heisenberg perovskites of Pb(Fe_{1/2}Nb_{1/2})O₃^{14,15} have been delivered. The phenomenal research output from magnetic nanoparticles ignited the idea of producing such AF-SG coexistence by a structure comprising a collection of AF nanoparticles. The AF nanoparticles present a weak ferromagnetic component due to uncompensated spins and have been studied in the form of (pseudo)oxides.^{7,16} This gives rise to what Néel coined as superantiferromagnetism of individual particles.¹⁷ In nanometric TbCu₂, one counts with the advantage of a high magnetic Tb signal, a Néel transition (T_N) below room temperature, and the possibility to produce them in large quantities (suitable for neutron diffraction) via high-energy milling. ¹⁸ In bulk state, TbCu₂ crystallizes in the Imma symmetry (orthorhombic, CeCu₂-type crystal structure) and orders at $T_N = 48 \text{ K}^{19}$ with a collinear AF structure. Seizing the particle size (D) as the driving variable to induce spin rearrangements, we report here the existence of a macroscopic AF with a reentrant SG state in nanosized TbCu₂ alloys.

A large (\simeq 5 g) quantity of nanometric TbCu₂ was produced by high-energy mechanical (WC containers) milling in Ar atmosphere (99.99%) during 2 hours.²¹ The master alloy was annealed at 750 °C for 5 days to ensure homogeneity prior to milling. In Fig. 1, the x-ray diffraction pattern performed at RT (Cu-K α $\lambda = 1.5418$ Å) is shown. The pattern has been refined by the Rietveld method (FULLPROF²²) including the resolution function with a LaB₆ calibration. The fit was carried out with a Thompson-Cox-Hastings function, allowing the calculation of the particle size (D) and strain (ϵ) . The result indicates that the *Imma* symmetry of the bulk alloy is retained and that the lattice parameters a = 4.322(1) Å, b = 6.833(2) Å,and c = 7.327(2) Å are similar to those of the bulk alloy.¹⁹ The Bragg error $R_B = 14.6\%$ is low for a nanostructure. The size D = 9(1) nm and $\epsilon = 0.2(1)\%$. Thus, the grinding time to obtain nanoparticles is small compared to other milled RE intermetallics.²¹ The structural characterization was completed via transmission electron microscopy (TEM). The inset of Fig. 1 shows a bright-field image of TbCu₂ nanoparticles with a sharp surface and absence of amorphous environments, and the high-resolution detail in which lattice planes are visible confirms the particle crystallinity. A histogram analysis gives $D_{\text{TEM}} = 7(1) \text{ nm}.$

The static magnetic characterization $M_{\rm DC}(T)$ was recorded in zero-field cooled (ZFC) and field cooled (FC) curves, and the loops $M_{\rm DC}(H)$ (T=5 K), including the initial magnetization at T=5 and 150 K. In Fig. 2 (H=1 kOe), the first manifestation of the PM \leftrightarrow AF \leftrightarrow SG transitions is patent. Thus, there is a peak at $T_N=46.8(5)$ K, followed by an increase of the susceptibility reaching another peak at a lower temperature, marking the freezing transition $T_f=9.2(2)$ K.

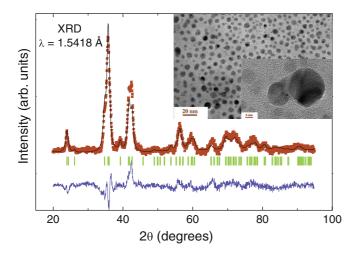


FIG. 1. (Color online) Rietveld refinement of the XRD pattern of 2 h milled alloy ($T=300~\rm K$). The theoretical calculation is depicted as a continuous line and the experimental data as points. The blue line below the spectra is the difference between the calculated and experimental patterns. The size is $D=9(1)~\rm nm$. Inset: Bright-field TEM image, where the particles appear spherical with $D=7(1)~\rm nm$, and HRTEM detail.

Simultaneously, an irreversibility appears for $T \lesssim 40$ K. These two findings disclose the magnetic reentrance. We have also applied different fields ($H \leq 10$ kOe) to this ZFC-FC process and have plotted (see inset of Fig. 2) the δT_f vs $H^{2/3}$ dependence, indicating that the glass temperature follows the de Almeida–Thouless line. This dependence was reported in bulk Pb(Fe_{1/2}Nb_{1/2})O₃, where a coexistence of AF and SG order is put forward. In nanomagnets it has been found in γ -Fe₂O₃ (ferrimagnetic) and ascribed to the canting of the particle surface spins. The linear fit in the $1/\chi$ vs T Curie-Weiss plot provides $\theta_P = 19(1)$ K, and a value of the paramagnetic moment $m_{\rm eff} = 9.83(5)$ μ_B , which compares well with the Tb³⁺ effective moment (9.72 μ_B). Here $\theta_P > 0$, in contrast to the bulk alloy value ($\theta_P = -7$ K), $\theta_P = 1.00$

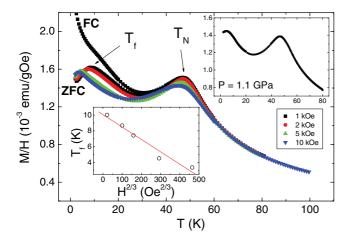


FIG. 2. (Color online) Thermomagnetic DC-susceptibility ZFC curves obtained with magnetic fields, 1 kOe $\leq H \leq$ 10 kOe, where the evolution from AF to SG state can be observed. A FC curve is also plotted at H=1 kOe to evidence the irreversibility. Inset (bottom): Linear T_f variation vs $H^{2/3}$. Inset (top): ZFC DC susceptibility of the alloy at a pressure P=1.1 GPa.

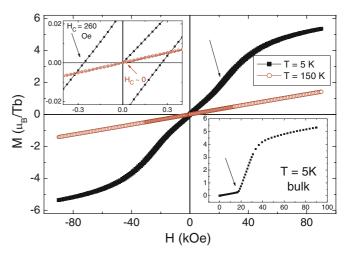


FIG. 3. (Color online) Magnetization curves of TbCu₂ nanoparticles at T=5 K (SG region) and 150 K (paramagnetic), up to H=90 kOe. Inset (left): Low-field coercivity in the SG state. Inset (right): $M_{\rm DC}(H)$ curves showing the metamagnetic transition of bulk alloy. Both bulk and 2 h alloys show the AF metamagnetic transitions, marked with arrows.

that the milling is triggering a weak FM coupling. The latter is necessary for the FM-AF frustration.^{2,4} The ordered magnetic arrangement is robust against hydrostatic pressure (P) variations. In Fig. 2 (top inset) the ZFC $M_{DC}(T)$ curve under P=1.1 GPa is plotted. The Cu-Be pressure cell is inside a SQUID magnetometer and the value of P was extracted from the superconducting transition of Sn. The effect of P is negligible in $\Delta T_N(P)/T_N(P=0)=0.2\%$ but $\Delta T_f(P)/T_f(P=0)=-15\%$ is significant; the SG state is influenced by thermal and field history,^{2,4} and thus by external pressure. These transition temperatures are larger than in Fe_{0.55}Mg_{0.45}Cl₂ ($T_N=7.5$ K, $T_f=3$ K),¹¹ and for the most recent Pb(Fe_{1/2}Nb_{1/2})O₃, $T_N=7.5$ K and $T_f=3$ K.¹⁵ In Co₂(OH)(PO₄), $T_N=70$ K, $T_f=15$ K.¹³

In Fig. 3, the $M_{\rm DC}(H)$ of the TbCu₂ nanoparticles is examined up to H = 90 kOe. At T = 150 K the response is linear of PM character. Below T_f , a large signal is detected reaching $M_{\rm DC}(H=90\,{\rm kOe})=5.35(5)~\mu_B/{\rm Tb}$. The magnetization does not reach saturation, as occurs in ≈15 nm TbAl₂, ²¹ presumably due to the spin canting on the surface. 16,23 Note that bulk TbCu2 is affected by a large anisotropy which complicates the determination of the saturation magnetization.²⁴ The lack of saturation in the bulk alloy is confirmed in the bottom inset of Fig. 3. The extrapolation of the high-field magnetization to H=0 gives $M_{\rm nano}/M_{\rm bulk}\approx 80\%$, hence a sign of the reduced contribution caused by moments at the surface. Another point is the existence of a kink around H = 20 kOe. This is a change of slope which is worth comparing to the $M_{DC}(H)$ response in bulk TbCu₂ (see bottom inset of Fig. 3). There, the kink is more pronounced and corresponds to the metamagnetic transition of AF. The outcome is that the AF is still present within the SG region. Regarding the presence of coercivity ($H_C = 260 \text{ Oe}$), which is also observed in SG systems, 2,4,13 it should be noted that AF compounds should not display coercivity due to the absence of domains. The reentrant Co₂(OH)(PO₄) shows $H_C \lesssim 50 \text{ Oe.}^{13}$

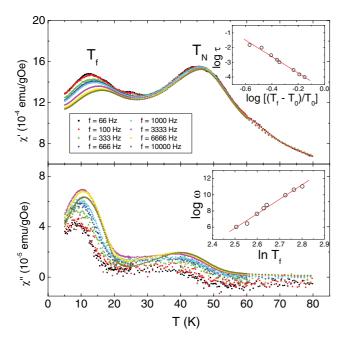


FIG. 4. (Color online) Real $\chi'(T)$ (top) and complex $\chi''(T)$ (bottom) components of the AC susceptibility at several $f = \omega/2\pi$ (h = 1 Oe). In $\chi'(T)$, the Néel transition (T_N) and the low-temperature freezing (T_f) are detected. The latter clearly shifts to higher temperatures with increasing f. In the $\chi''(T)$, the AF transition is barely observed. Inset, bottom: Linear behavior of the δ shift of T_f . Inset, top: Line fit according to the critical slowing down of the relaxation times.

The spin dynamics was studied with an oscillating field h =1 Oe and the frequency was varied between $f = (\omega/2\pi) =$ 10 Hz and 10 kHz. In Fig. 4, the real $[\chi'(T)]$ and complex $[\chi''(T)]$ components of the AC susceptibility are plotted for several frequencies. The shape of the $\chi'(T)$ curve is similar to the ZFC $M_{\rm DC}(T)$. There is a main peak at 46.0(5) K marking the T_N , practically frequency independent, and a smaller contribution, with a broad rounded peak which shifts up in temperature and decreases in magnitude with higher frequency, corresponding to $T_f(\omega)$. In the case of $\chi''(T)$, the Néel transition is nearly absent, marking the onset of antiferromagnetism, while the low T_f peak is observable showing, as for $\chi'(T)$, a ω shift. This shift is commonly quantified as $\delta = \Delta T_f/T_f \log_{10} \omega = 0.057(3)$ (see bottom inset of Fig. 4). This value is larger than that of the individual spins in $Co_2(OH)(PO_4)$ ($\delta = 0.0024$). ¹³ If the magnetic relaxation of the particles were independent, a $\delta = 0.1$ –0.2 is expected.^{4,8} In ideal SGs the relaxation is faster and the values are much lower;² for ferromagnets with reentrant state $\delta = 0.01$ (in crystalline FeAu⁴) and $\delta = 0.013(2)$ with data from amorphous (Fe_{0.20}Ni_{0.80})₇₅P₁₆B₆Al₃. An intermediate metallic case is observed in super-SGs, as that of Fe₂₄Ag₇₆ with single-domain ferromagnetic Fe nanoparticles, where $\delta = 0.010(1)$. Assuming that the spins participating in the freezing follow a critical slowing down, the measuring time τ_m : $\tau_m = \tau_0 (T_f/T_0 - 1)^{-zv}$, where zv is the dynamic exponent. Our analysis reveals a large $\tau_0 = 1.3(5) \times 10^{-5}$ s, as that (10^{-6} s) discussed in Fe-C (4.7 nm) nanoparticles,⁵ $T_0 =$ 9.6(4) K, and zv = 5.7(1.1) (top inset of Fig. 4). The zvvalue is close to reentrant FM and super-SG systems. 8-10

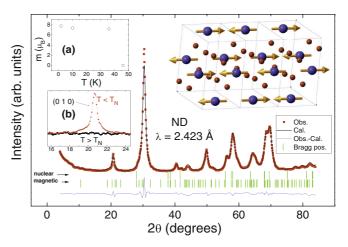


FIG. 5. (Color online) Neutron diffraction pattern at $T=1.8~\rm K$ and its nuclear + magnetic Rietveld refinement. Right inset: The magnetic structure is antiferromagnetic (collinear) with large Tb atoms (blue) holding the magnetic moment (arrow) coinciding with that reported for bulk TbCu₂ (Ref. 20). Cu atoms (red) are of smaller size. Inset (a): Thermal variation of the magnetic moment of Tb atoms. Inset (b): Uprise of first AF peak, below T_N .

So far the existence of *macroscopic* AF and SG states is clear. However a microscopic magnetic structure analysis is required; this has been carried out by neutron diffraction in Laboratoire Léon Brillouin (G4.1 instrument) with $\lambda =$ 2.423 Å. In Fig. 5, the neutron pattern at $T = 1.8 \text{ K} < T_f, T_N$ is shown together to a total Rietveld refinement.²² The lattice parameters are smaller: a = 4.3176(1) Å, b = 6.7842(9) Å,and c = 7.2990(7) Å, than the XRD ones at 300 K. The size calculation yields similar $D_{\text{nuclear}} = 9(1)$ nm and $D_{\text{mag}} =$ 10(1) nm values, establishing that the magnetic structure is a single domain in the particle. A rough estimation assuming the ratio of surface atoms (N_s) with respect to those in the total volume (N_V) through $N_s/N_V = (6a/D) \approx 40\%$, remarking the role of surface spins and supporting the surface moment reduction, although it only includes geometrical effects. A crucial result is that the magnetic structure (right inset of Fig. 5) is a commensurate AF with two propagation vectors $\vec{q}_1 = (0,0,0)$ and $\vec{q}_2 = (1/3,0,0)$, as in bulk. The ordered Tb^{3+} magnetic moment $m=7.76(2)~\mu_B~(R_{\rm mag}=4.2\%~{\rm and}$ $R_B = 1.1\%$). In the temperature variation of the ordered magnetic moment [Fig. 5(a)], there is an expected increase below T_N towards the magnetic ground state. The coherent diffraction across several lattice planes assures the long-range character of the AF interactions, with a mean correlation length $D_{\rm mag} = 10(1)$ nm similar to the nanoparticle size. Hence the AF order survives in the frozen SG region [inset (b) of Fig. 5]. In fact the neutron magnetic intensity reduction (\approx 78%) at T = 1.8 K, considering the (neutron) bulk $m = 8.8 \mu_B$, ²⁰ also points to a disordered environment, affected by the decrease of surface coordination with respect to the atoms in the particles. 16 Another revealing result is observed in Fig. 6. This is the existence of a clear $2\theta < 15^{\circ}$ low-angle scattering. The low-angle scattering increases with decreasing temperature evidencing magnetic correlations with an interparticle length scale. The Q range is limited $(0.1 \text{ Å}^{-1} < Q < 0.7 \text{ Å}^{-1})$ but the variation of the integrated intensity confirms this result (left inset, Fig. 6). This is a sign of the existence of interparticle

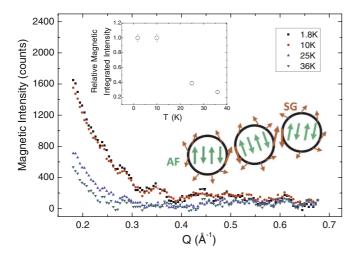


FIG. 6. (Color online) Low-angle magnetic contribution corresponding to the 0.1 Å $^{-1}$ < Q < 0.7 Å $^{-1}$ range. There is a marked increase of curvature pointing to the existence of interparticle magnetic correlations. Left inset: Thermal variation of the integrated magnetic intensity for Q < 0.7 Å $^{-1}$. Right inset: Simplified sketch visualizing the spin nanostructure.

correlations. It is likely that reducing the Q range would allow the precise determination of the coherence lengths involved. ^{28,29} In short, there is an AF order within the percolating magnetic moments inside the particles (see sketch in Fig. 6).

In the few AF systems with a reentrant SG state, $^{11-15}$ the combination of frustration and random bonds gives rise to the freezing. The Sherrington-Kirkpatrick model predicts that long-range ferromagnetism is always present but, for $T \leqslant T_f$, there exists a replica symmetry breaking resulting in the coexistence of FM and SG states. 26 Such an explanation qualitatively supports the phase diagram in Fe_{0.55}Mg_{0.45}Cl₂. 11

For Pb(Fe_{1/2}Nb_{1/2})O₃ both noncollinearity and clustering have been invoked. A very recent model has been proposed involving small clusters with AF order, showing $\chi(T)$ resembling those in Fig. 4, and predicting SG \leftrightarrow PM transitions. However, here the particle size is much larger.

In our rare-earth alloy nanoparticulate system, the major AF coupling is held within the AF nanoparticles in a superantiferromagnetic arrangement and the uncompensated spins of both sublattices provide a source for randomness at the particle surface. As a result, a thin layer of random spins favors the SG behavior. It should be noted that although our experimental evidence is providing a single domain coherence length for each particle, it would not be surprising that such a coherence length could cover a few particles of the randomly placed particles, increasing the range of the AF interaction. The exact definition of such range caused by the RKKY interaction merits further work via small-angle neutron scattering. The spin scenario resembles that of amorphous Fe-rich FeZr, where magnetic clustering is settled in a metallic arrangement, ^{6,29} which presents a reentrant macroscopic signature, and that of the supermagnetism appearing in nanomagnetic particles.^{7–9} Here, a fraction of spins forming the surface acts as a switch, enabling/disabling the coexistence, thus mimicking the variations of concentration in canonical and reentrant SGs.^{2,4,6}

In conclusion, we have defined the Néel transition and the freezing temperature due to the spins at the surface of $TbCu_2$ nanoparticles. The glassy transition does not hamper the AF state. The PM \leftrightarrow AF \leftrightarrow SG phases constitute a very attractive change of magnetic state in a metal. Consequently, the system should be considered as a superantiferromagnet with reentrant behavior assured by the nanoparticle surface.

This work was supported by a CICyT MAT2011-27573-C04 grant. L.F.B. acknowledges pertinent comments by S. N. Kaul (University of Hyderabad, India) and by S. G. Magalhães (Universidade Fluminense, Brazil).

¹C. M. Hurd, Contemp. Phys. **23**, 469 (1982).

²K. Binder and A. P. Young, Rev. Mod. Phys. **58**, 801 (1986).

³E. C. Andrade and M. Vojta, Phys. Rev. Lett. **109**, 147201 (2012).

⁴J. A. Mydosh, *Spin Glasses: An Experimental Introduction* (Taylor & Francis, London, 1993).

⁵C. Djurberg, P. Svedlindh, P. Nordblad, M. F. Hansen, F. Bødker, and S. Mørup, Phys. Rev. Lett. **79**, 5154 (1997).

⁶D. H. Ryan, J. M. D. Coey, E. Batalla, Z. Altounian, and J. O. Ström-Olsen, Phys. Rev. B 35, 8630 (1987); S. N. Kaul, V. Siruguri, and G. Chandra, *ibid.* 45, 12343 (1992).

⁷Steen Mørup, Morten Bo Madsen, Jørgen Franck, Jørgen Villadsen, and Christian J. W. Koch, J. Magn. Magn. Mater. **40**, 163 (1983).

⁸S. Bedanta and W. Kleemann, J. Phys. D **42**, 13001 (2009).

⁹J. Alonso, M. L. Fdez-Gubieda, J. M. Barandiarán, A. Svalov, L. Fernández Barquín, D. Alba Venero, and I. Orue, Phys. Rev. B 82, 054406 (2010).

¹⁰K. Jonason, J. Mattsson, and P. Nordblad, Phys. Rev. B **53**, 6507 (1996).

¹¹Po-zen Wong, S. von Molnar, T. T. M. Palstra, J. A. Mydosh, H. Yoshizawa, S. M. Shapiro, and A. Ito, Phys. Rev. Lett. **55**, 2043 (1985); H. Yoshizawa, S. Mitsuda, H. Aruga, and A. Ito, *ibid.* **59**, 2364 (1987).

¹²I. Mirebeau, C. Bellouard, M. Hennion, G. Jehanno, V. Caignaert, A. J. Dianoux, T. E. Phillips, and K. Moorjani, Physica C 184, 299 (1991).

¹³J. M. Rojo, J. L. Mesa, L. Lezama, J. L. Pizarro, M. I. Arriortua, J. Rodriguez Fernandez, G. E. Barberis, and T. Rojo, Phys. Rev. B 66, 094406 (2002).

¹⁴G. M. Rotaru, B. Roessli, A. Amato, S. N. Gvasaliya, C. Mudry, S. G. Lushnikov, and T. A. Shaplygina, Phys. Rev. B 79, 184430 (2009).

¹⁵W. Kleemann, V. V. Shvartsman, P. Borisov, and A. Kania, Phys. Rev. Lett. **105**, 257202 (2010).

¹⁶S. Mørup, D. E. Madsen, C. Frandsen, C. R. H. Bahl, and M. F. Hansen, J. Phys.: Condens. Matter **19**, 213202 (2007).

¹⁷L. Néel, Comptes Rendus Acad. Sci., Paris **253**, 9 (1961).

- ¹⁸D. P. Rojas, J. I. Espeso, L. Fernández Barquín, J. Rodríguez Fernández, and J. C. Gómez Sal, J. Magn. Magn. Mater. 310, e506 (2007).
- ¹⁹N. H. Luong and J. J. M. Franse, in *Handbook of Magnetic Materials*, edited by K. H. J. Buschow, Vol. 8 (Elsevier, Amsterdam, 1995), p. 415.
- ²⁰V. Ma, Z. Smetana, B. Lebech, and E. Gratz, J. Magn. Magn. Mater. 54–57, 1357 (1986).
- ²¹D. P. Rojas, L. Fernández Barquín, J. Rodríguez Fernández, J. I. Espeso, and J. C. Gómez Sal, J. Phys.: Condens. Matter 19, 186214 (2007).
- ²²J. Rodríguez-Carvajal, Physica B **192**, 55 (1993).
- ²³B. Martinez, X. Obradors, Ll. Balcells, A. Rouanet, and C. Monty, Phys. Rev. Lett. **80**, 181 (1998).

- ²⁴C. A. Poldy and E. Gratz, J. Magn. Magn. Mater. 8, 223 (1978).
- ²⁵Franz Bødker, Mikkel F. Hansen, Christian Bender Koch, Kim Lefmann, and Steen Mørup, Phys. Rev. B **61**, 6826 (2000)
- ²⁶D. Sherrington and S. Kirkpatrick, Phys. Rev. Lett. **35**, 1792 (1975).
- ²⁷C. F. Silva, F. M. Zimmer, S. G. Magalhaes, and C. Lacroix, Phys. Rev. E 86, 051104 (2012).
- ²⁸C. Bellouard, I. Mirebeau, and M. Hennion, Phys. Rev. B **53**, 5570 (1996).
- ²⁹R. García Calderón, L. Fernández Barquín, S. N. Kaul, J. C. Gómez Sal, Pedro Gorria, J. S. Pedersen, and R. K. Heenan, Phys. Rev. B 71, 134413 (2005).