

Extension of cylindrical samples in the Sentmanat Extensional Rheometer (SER)

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Publication date:
2010

Document Version
Publisher's PDF, also known as Version of record

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Citation (APA):

Yu, K., Marin, J. M. R., Jensen, M. K., Rasmussen, H. K., & Hassager, O. (2010). Extension of cylindrical samples in the Sentmanat Extensional Rheometer (SER). Abstract from 6th Annual European Rheology Conference, Göteborg, Sweden.

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Is there a Relationship between the Elongational Viscosity and the First Normal Stress Difference in Polymer Solutions?

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Elongational viscosity and a non-zero first normal stress difference are typical characteristics of flexible systems. The elongational viscosity of polymer solutions can be several orders of magnitude larger than the solvent viscosity. In contrast, the shear viscosity remains on the order of the solvent viscosity. In many cases, elongational flow affects the processability more severely, e.g., in fiber spinning, spraying, and coating of pesticides, etc. where the flow is elongational, at least to a large extent. However, the definition of the first normal stress $N_1 = \tau_{xx} - \tau_{yy}$ (with τ_{ab} the components of the stress tensor) is not directly comparable to the definition of the elongational viscosity $\eta_e = \frac{\tau_{xx} - \tau_{yy}}{\dot{\epsilon}}$ in elongational flow suggests the existence of a direct relation between these quantities ($\dot{\epsilon}$ is the elongational rate). From an experimental point of view, measurement of the elongational viscosity of dilute polymer solutions is a nontrivial task and only recently a method called Capillary Breakup Extensional Rheometer) became available. We investigate a variety of different situations in shear and elongational flow. The shear flow is created in the cone-plate-geometry of a rheometer. We compare the relaxation time and the elongational viscosity measured in the CaBER with the first normal stress difference and the relaxation time that we measured in our rheometer. All of these quantities depend on different fluid parameters - the viscosity of the polymer solution, the polymer concentration within the solution, and the molecular weight of the polymers - and on the shear rate. We find that the first normal stress coefficient depends quadratically on the CaBER relaxation time, while the relaxation factor depends on the type of polymer only, whilst theoretical considerations predict a linear dependence on the concentration, too. Different scaling relations are presented that can help explaining our

Soft Materials Rheology studied by Diffusing Wave Spectroscopy

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Diffusing wave spectroscopy (DWS) is an optical technique which provides detailed information about the rheological properties of soft materials without interacting mechanically with the sample. DWS measures the time correlation functions of laser light diffusely scattered within the sample. Via the measured intensity correlation function, the squared displacement (MSD) of tracer particles in the medium is obtained. The MSD is then used to determine the storage and loss modulus (G' & G'') based on the general microrheology approach. Since its introduction in 1987/88 tremendous progress has been made in the field of detector technology. Single-photon detectors have response times in the nano second range and quantum efficiencies above 90%. Novel optical schemes such as the "Two-Cell-Echo-Technology" provide orders of magnitude improvement in the dynamic range of the DWS. The combination of both has substantially advanced the range of application and the performance of the DWS. Today frequencies up to 1'000'000 rad/s for moduli in the range 1-100'000 Pa are accessible with DWS. Recently the first commercial instruments based on the DWS approach have become available. We will discuss the most important technological advances in diffusing wave spectroscopy. We will demonstrate the performance of this technique on a number of viscoelastic complex fluids ranging from surfactant micelles to colloidal dispersions (microgel) dispersions and dairy products. We discuss both the opportunities and the challenges of DWS.

Experimental approach to determine high frequency rheological properties of polyethylene

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The determination of the second crossover frequency and the corresponding entanglement relaxation time, τ_e , together with the entanglement modulus, G_e , can be achieved for a number of thermoplastic polymers by applying the TTS method. The knowledge of these parameters is important, because they provide a link to structural parameters like friction factor and others. In contrast, polyethylene, due to its fast crystallization below T_m and weak temperature dependence of viscosity, can be explored rheologically in limited frequency range, only. An experimental determination of the mentioned parameters was only possible for model polyethylenes, and its determination for industrial polyethylenes is missing so far. Here we characterize commercial polydisperse, as well as linear and branched, polyethylenes by expanding the frequency range utilizing new rheometers: piezoelectric- and new crystal resonator technology. We introduced high frequency rheological techniques and show how to implement these new measurements methods, and compare with polymers, for which TTS holds. This allows us to determine G_e and τ_e for PE for the first time. Finally we compare the obtained material parameters with simulation results.

Extension of cylindrical samples in the Sentmanat Extensional Rheometer (SER)

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Polymer extensional flow is the one of the most important deformation in polymer processing. It is the dominant deformation in melt-spinning, bottle-blowing, and roll-coating. Because the molecular structure of the polymeric system strongly influences the extension viscosity, extensional flow measurements are useful for polymer characterization. The Sentmanat extensional Rheometer [1] is a new testing platform for the study of polymers and elastomers in extensional flow. This technique employs a dual wind-up drum technique to perform an uni-axial extensional deformation during experiments. It requires small amount of materials and can be used for polymer melt and soft elastomers characterization over a very wide range of temperatures and kinematic deformations and rates. In order to validate the reliability of this testing platform a finite element technique based on a Lagrangian kinematics description of the 3D time-dependent flow of K-BKZ type fluids [2] is used to simulate extension flow of cylindrical shaped sample in the SER. Here the purpose is to discuss the potential deviations from ideal uni-axial deformation, based on theoretical ideal configurations. Our simulation can setup a theoretical based 'safe' geometry range of cylinder samples for SER experiments. Extension of a cylindrical sample shows the less inhomogeneity compared to a strip shapes sample [3]. Whereas using a sample with an initial diameter of more than 0.5mm will create a too large deviation between expected and obtained Hencky strain. Furthermore, the simulations are able to capture flow instabilities in stress relaxation, which have been experimentally observed [4]. Reference [1]. M.L. Sentmanat, Rheol Acta, 43:657-669, 2004. [2]. J.M.R. Marin, H.K.Rasmussen, J. Non-Newtonian Fluid Mech, 156 (3) , p. 177-188 [3]. K.Y, J.M.R. Marin, H.K.Rasmussen, O.Hassager, J. Non-Newtonian Fluid Mech (accepted) [4]. Y.Wang, P.Boukany, S.Wang, X.Wang, Physical Review Letters, 99, 237801 (2007)