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INFLUENCE OF LAMINATION ORIENTATION AND STACKING ON MAGNETIC CHARACTERISTICS OF GRAIN-ORIENTED SILICON STEEL LAMINATIONS

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ABSTRACT

Analytical and experimental investigations have been carried out upon the behaviour of flux in laminations, where the rolling directions of adjacent sheets are reversed. The paper clarifies the mechanism of the greatly different magnetic characteristics between such laminations and usual ones, where the rolling directions of adjacent sheets are coincident.

1. INTRODUCTION

At the T-joint of a so-called scrap-less-type three-phase transformer core, laminations are alternatively turned over so that the angle between the rolling directions of overlapping sheets may become 90°. The magnetic characteristics of such laminations are greatly different from those of usual laminations, where the rolling directions of adjacent sheets are coincident [1], [2]. Some measurements of flux have been carried out in order to obtain a fuller understanding of the effect of the stacking method of laminations [3]-[5]. If localized flux distributions are obtained, the mechanism of the different magnetic characteristics in these laminations should be explained clearly.

The flux distribution in laminations, where the rolling directions are reversed, is calculated using a new approximation method to solve three-dimensional magnetic fields [6]. Using a single sheet tester [7], the experimental investigations have also been carried out on the effects of stacking, cut angle and the flux density. The paper clarifies the mechanism of the greatly different magnetic characteristics.

2. ANALYSIS

2.1 Model

For simplicity, laminations stacked as shown in Fig. 1 were examined. These laminations were tested in the single sheet tester. The stackings in Figs. 1 (a) and 1 (b) are called as the "//" and "X" stackings respectively. For example, "X" stacking in Fig. 1(b) corresponds to the hatched parts of a so-called scrap-less-type three-phase transformer core shown in Fig. 2. The solid arrows \longleftrightarrow denote the rolling directions of the sheets in the first lamination and the dashed arrows $\dashleftarrow\dashrightarrow$ denote those in the adjacent lamination.

The laminations in Fig. 1 were 0.3(mm) thick highly-oriented silicon steel M-3H (AISI-78). The frequency was 50(Hz).

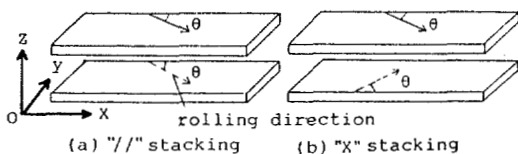


Fig. 1 Stacking methods.

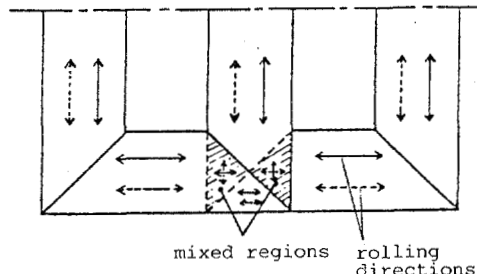


Fig. 2 Three-phase transformer core.

2.2 Method of Analysis

The magnetic characteristics of the "//" stacking shown in Fig. 1(a) were analyzed using the two-dimensional finite element method [8].

As the flux distribution in the lamination shown in Fig. 1(b) is nonuniform in the z-direction, three-dimensional analysis is required. If it is possible to analyze this three-dimensional field approximately by modifying the two-dimensional finite element method, both the computer storage and the computing time can be reduced considerably. The "new approximation method [6]" developed by the authors is used in the analysis of the "X" stacking.

In this approximation method, the flux distribution in each lamination is calculated from vector potentials {A} as well as distribution ratios F_{x1} and F_{y1} , defined later [6]. It is assumed that the z-directional magnetic resistance is negligibly small.

X- and y-directional components of flux densities, i.e. B_{x1} , B_{y1} , B_{x2} and B_{y2} , in the first and the second sheets in Fig. 3 are denoted by

$$B_{x1} = \phi_x F_{x1} / t \tag{1}$$

$$B_{x2} = \phi_x (1 - F_{x1}) / t \tag{2}$$

$$B_{y1} = \phi_y F_{y1} / t \tag{3}$$

$$B_{y2} = \phi_y (1 - F_{y1}) / t \tag{4}$$

where, ϕ_x and ϕ_y are x- and y-directional components of the total flux of the two sheets. "t" is the effective thickness of one sheet in the z-direction. F_{x1} and F_{y1} are the ratios of the x- and y-directional components of the flux in the first sheet to ϕ_x and ϕ_y respectively. The flux distribution in each lamination is calculated from the following equations:

$$\frac{\partial X}{\partial A_i} = 0, \quad \frac{\partial X}{\partial F_{x1}} = \frac{\partial X}{\partial F_{y1}} = 0 \tag{5}$$

Where, A_i is the vector potential at a node i. X is given by

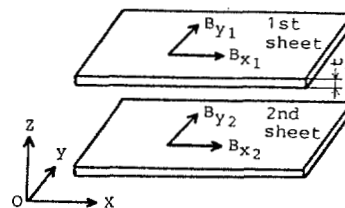


Fig. 3 Flux density in each sheet.

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$$X = X_1 + X_2 \tag{6}$$

$$X_1 = t \iint_S \left(\int_0^{B_{X1}} v_{X1} B_{X1} dB_{X1} + \int_0^{B_{Y1}} v_{Y1} B_{Y1} dB_{Y1} \right) dx dy \tag{7}$$

$$X_2 = t \iint_S \left(\int_0^{B_{X2}} v_{X2} B_{X2} dB_{X2} + \int_0^{B_{Y2}} v_{Y2} B_{Y2} dB_{Y2} \right) dx dy \tag{8}$$

X_1 and X_2 are the energies in the first and the second sheets.

2.3 Method of Experiment

Figure 4 shows the single sheet tester used in the experiment [7]. As this tester has two yokes, the magnetic characteristics of the laminations stacked with two sheets having different cut angles can be investigated. The iron loss was measured using B and H coils [9]. The H coils are set at the upper and lower sides of the two sheets as shown in Fig. 4. The length of the H coil is 2/3 of that of the exciting coil. The B coil is set between the exciting coil and the H coil, and its length is exactly the same as that of the H coil. The interlaminar gap length T_g was adjusted by inserting a sheet made of phenol resin.

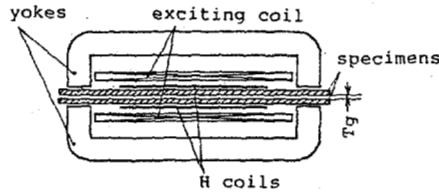


Fig. 4 Single sheet tester.

3. RESULTS AND DISCUSSIONS

Figure 5 shows the effects of the stacking, the cut angle θ shown in Fig. 1, and the maximum flux density B_m , on the iron loss W . These curves were obtained by using the single sheet tester shown in Fig. 4. The iron loss is greatly affected by the stacking method. The mechanism of the different iron loss characteristics is clarified by the localized flux distributions shown in Figs. 6 and 7 which are calculated for the case of $\theta=45^\circ$.

In Fig. 6, one flux line represents 0.1(Wb) when the width and the thickness of the sheet are equal to 1.0(m) respectively. The flux in the "//" stacking flows in a direction deviating from the rolling one.

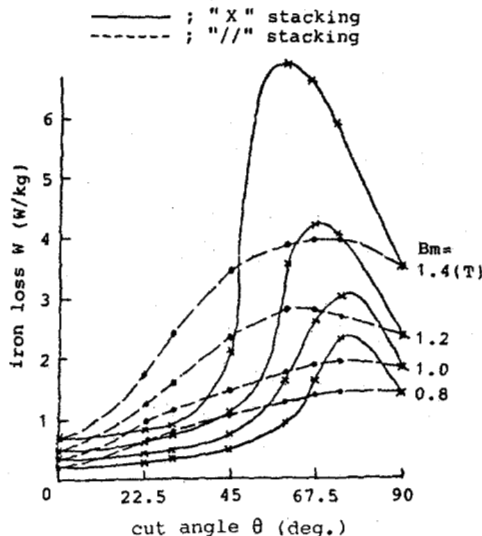


Fig. 5 Angle dependence of iron loss ($T_g=0$ (mm), measured).

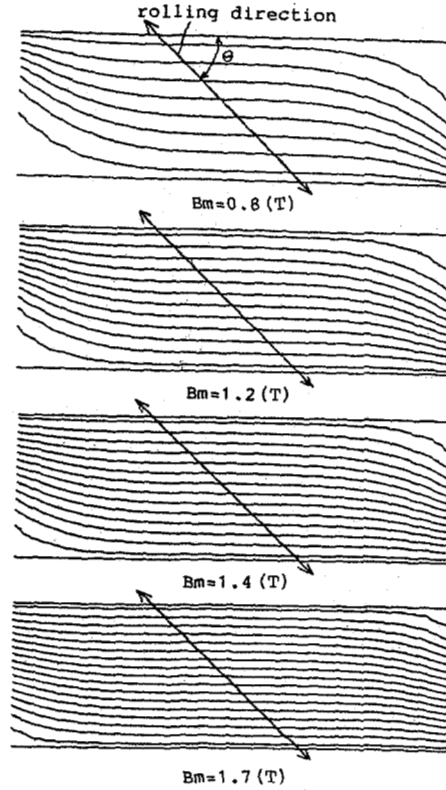
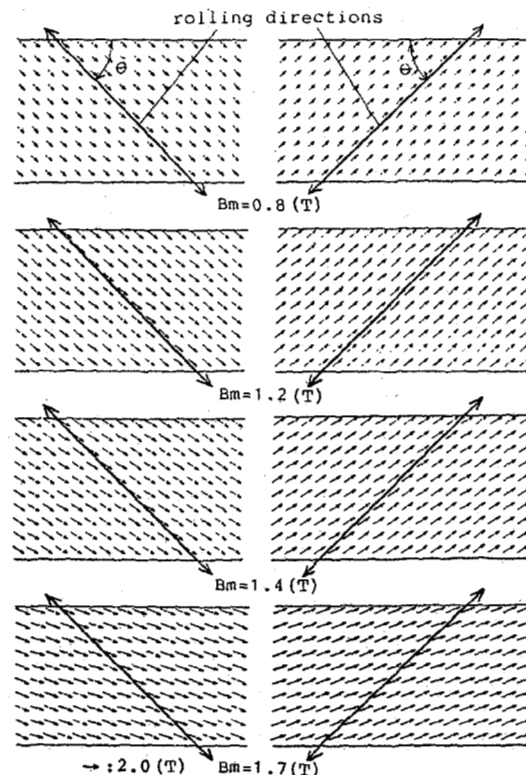


Fig. 6 Flux distributions of "//" stacking ($\theta=45^\circ$).



(a) the first sheet (b) the second sheet

Fig. 7 Flux distributions of "X" stacking ($\theta=45^\circ$).

In the "X" stacking, the flux zigzags transversing to the adjacent sheet as shown in Fig. 7. The flux does not flow continuously in the sheet. It transverses to the adjacent sheet, and changes the direction about 2θ, so that the flux in each sheet may flow in the rolling direction. When the gap length T_g becomes smaller, most of the transversing flux concentrates near the edge of the sheet. From the above-mentioned reason, the flux distribution should be represented by flux density vectors as shown in Fig. 7 instead of the flux lines in Fig. 6.

Figure 8 shows the relationship among the apparent flux density vector B_m and the flux density vectors B₁ and B₂ in the first and the second sheets of the "X" stacking. The apparent flux density B_m is the mean vector of the flux densities B₁ and B₂. The amplitudes B₁ and B₂ of them are denoted by

$$B_1 = B_2 = B_m / \cos \phi \quad (9)$$

where, φ is the angle between vectors B₁ (or B₂) and B_m as shown in Fig. 8. When B_m is small, the angle φ is nearly the same as θ. According to Eq. (9), B₁ or B₂ is nearly equal to √2B_m in the case of Fig. 7. When B_m is increased, φ becomes smaller than θ. Because the magnetic resistance in the φ direction is smaller than that in the θ direction due to the magnetic saturation in this θ direction. In Fig. 7, the flux deviates from the rolling direction when B_m ≥ 1.2(T). Although, Eq. (9) means that B₁ or B₂ reaches infinity when θ approaches 90°, B₁ or B₂ cannot be increased so much because of the magnetic saturation. From this reason, B₁ (or B₂) approaches a saturated value, and finally reaches B_m when θ=90°. The iron loss characteristics in Fig. 5 can be explained from the above-mentioned facts.

Figure 9 shows the estimated iron loss [10] in the case of Fig. 7. When B_m < 1.2(T), the estimated iron loss curve denoted by the solid line agrees well with the measured one denoted by the dashed line. When B_m ≥ 1.2(T), the iron loss estimated is not reliable. Because the flux does not flow in the rolling direction and an accurate estimation method of the iron loss for such a case is not established yet.

The flux distribution in Fig. 7 is checked indirectly by examining the iron loss characteristics of the laminations with the interlaminar gap T_g shown in Fig. 4. When the gap T_g is increased, the iron loss of the "X" stacking approaches that of the "//" stacking as shown in Fig. 10. Because the transverse flux in the interlaminar gap decreases with the increase of T_g.

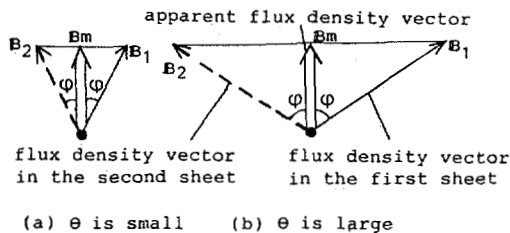


Fig. 8 Flux density vectors.

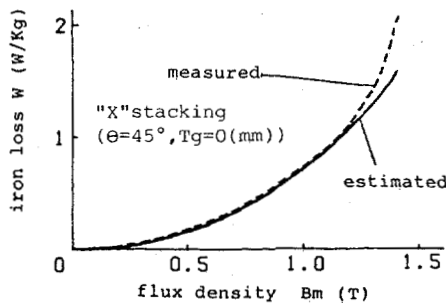


Fig. 9 Measured and estimated iron losses.

Figures 9 and 10 prove indirectly that the calculated flux distributions shown in Figs. 6 and 7 are sufficiently reasonable. The iron loss characteristics in Fig. 5 are explained by such an extraordinary behaviour of the fluxes clarified from Figs. 6 and 7.

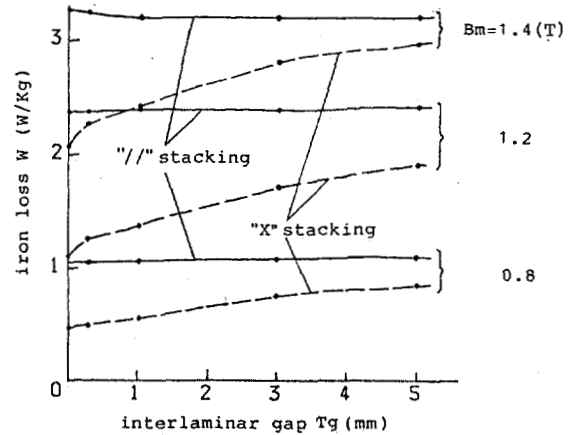


Fig. 10 Effects of the interlaminar gap T_g (measured, θ=45°).

4. CONCLUSIONS

The uniqueness of magnetic characteristics of laminations, where the rolling directions in adjacent sheets are reversed, is explained by investigating the behaviour of flux in such laminations analytically and experimentally. The knowledge of the detailed behaviour of the flux obtained in this paper should give useful suggestions to reduce the iron loss in the T-joints of transformer cores.

The behaviour of flux in laminations with various cut angle will be reported later.

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