

## Mixed Valence Europium Nitridosilicate $\text{Eu}_2\text{SiN}_3$

Martin Zeuner,<sup>†</sup> Sandro Pagano,<sup>†</sup> Philipp Matthes,<sup>†</sup> Daniel Bichler,<sup>†</sup> Dirk Johrendt,<sup>†</sup>  
Thomas Harmening,<sup>‡</sup> Rainer Pöttgen,<sup>‡</sup> and Wolfgang Schnick<sup>\*†</sup>

Ludwig-Maximilians-Universität München, Department Chemie und Biochemie,  
Butenandtstraße 5-13 (D), D-81377 München, Germany, and WWU Münster, Institut für  
Anorganische und Analytische Chemie and NRW Graduate School of Chemistry der  
Westfälischen Wilhelms-Universität, Germany

Received May 25, 2009; E-mail: wolfgang.schnick@uni-muenchen.de

**Abstract:** The mixed valence europium nitridosilicate  $\text{Eu}_2\text{SiN}_3$  has been synthesized at 900 °C in welded tantalum ampules starting from europium and silicon diimide  $\text{Si}(\text{NH})_2$  in a lithium flux. The structure of the black material has been determined by single-crystal X-ray diffraction analysis (*Cmca* (no. 64),  $a = 542.3(11)$  pm,  $b = 1061.0(2)$  pm,  $c = 1162.9(2)$  pm,  $Z = 8$ , 767 independent reflections, 37 parameters,  $R1 = 0.017$ ,  $wR2 = 0.032$ ).  $\text{Eu}_2\text{SiN}_3$  is a chain-type silicate comprising one-dimensional infinite nonbranched *zwei* chains of corner-sharing  $\text{SiN}_4$  tetrahedra running parallel [100] with a maximum stretching factor  $f_s = 1.0$ . The compound is isostructural with  $\text{Ca}_2\text{PN}_3$  and  $\text{Rb}_2\text{TiO}_3$ , and it represents the first example of a nonbranched chain silicate in the class of nitridosilicates. There are two crystallographically distinct europium sites (at two different Wyckoff positions 8f) being occupied with  $\text{Eu}^{2+}$  and  $\text{Eu}^{3+}$ , respectively.  $^{151}\text{Eu}$  Mössbauer spectroscopy of  $\text{Eu}_2\text{SiN}_3$  differentiates unequivocally these two europium atoms and confirms their equiatomic multiplicity, showing static mixed valence with a constant ratio of the  $\text{Eu}^{2+}$  and  $\text{Eu}^{3+}$  signals over the whole temperature range. The  $\text{Eu}^{2+}$  site shows magnetic hyperfine field splitting at 4.2 K. Magnetic susceptibility measurements exhibit Curie–Weiss behavior above 24 K with an effective magnetic moment of  $7.5 \mu_B/\text{f.u.}$  and a small contribution of  $\text{Eu}^{3+}$ , in accordance with  $\text{Eu}^{2+}$  and  $\text{Eu}^{3+}$  in equiatomic ratio. Ferromagnetic ordering at unusually high temperature is detected at  $T_C = 24$  K. DFT calculations of  $\text{Eu}_2\text{SiN}_3$  reveal a band gap of  $\sim 0.2$  eV, which is in agreement with the black color of the compound. Both DFT calculations and lattice energetic calculations (MAPLE) corroborate the assignment of two crystallographically independent Eu sites to  $\text{Eu}^{2+}$  and  $\text{Eu}^{3+}$ .

### Introduction

During the last decade nitridosilicates and oxonitridosilicates emerged from structural ceramics to advanced optical materials. Due to their exceptional chemical and physical stability in conjunction with nonlinear optical properties (e.g., 2-photon excitation, second harmonic generation, SHG) or luminescence (upon doping with  $\text{Ce}^{3+}$  or  $\text{Eu}^{2+}$ ) these materials found industrial application, e.g. as highly efficient phosphors in modern LED technology.<sup>1–7</sup>

From a structural point of view, nitridosilicates represent a significant extension of the broad and varied crystal chemistry of classical oxosilicates, although on a local atomic scale, both oxosilicates and nitridosilicates are made up of analogous and

isosteric building blocks, namely  $\text{SiO}_4$  and  $\text{SiN}_4$  tetrahedra, respectively. The broad diversity of condensed silicate structures is a direct consequence of the variability in connectivity of oxygen in classical silicates and even more of nitrogen in nitridosilicates.<sup>2</sup> Whereas the structural chemistry of oxosilicates is limited to terminal  $\text{O}^{[1]}$  atoms and simply bridging  $\text{O}^{[2]}$ , nitridosilicates emulate the same with  $\text{N}^{[1]}$  atoms and  $\text{N}^{[2]}$  but can additionally form highly crosslinked networks which contain  $\text{N}^{[3]}$  or  $\text{N}^{[4]}$  bridges connecting three or even four neighboring tetrahedral centers (e.g.,  $\text{BaYbSi}_4\text{N}_7$ ).<sup>8,9</sup> Additionally, in nitridosilicates, edge sharing has been observed (e.g.,  $\text{Ba}_5\text{Si}_2\text{N}_6$ ;  $\text{BaSi}_7\text{N}_{10}$ ).<sup>10,11</sup> The crystal chemistry of silicates has become much richer by introducing nitrogen atoms as anionic network formers, yielding new structure types that are naturally not occurring in minerals. Examples for such nitrido- or oxonitridosilicates are the framework silicates  $\text{M}_2\text{Si}_5\text{N}_8$  ( $M = \text{Ca}, \text{Sr}, \text{Ba}$ ) or also the layer silicates  $\text{MSi}_2\text{N}_2\text{O}_2$  ( $M = \text{Ca}, \text{Sr}, \text{Ba}, \text{Eu}$ ).<sup>12–16</sup> Both series of compounds proved to be excellent host

<sup>†</sup> Ludwig-Maximilians-Universität München.

<sup>‡</sup> Westfälische Wilhelms-Universität (WWU Münster).

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structures for rare earth doping and provide highly efficient luminescent materials which are industrially applied in commercial phosphor-converted pc-LEDs.<sup>4,17</sup> Even reduced nitridosilicates  $\text{MSi}_6\text{N}_8$  with  $\text{M} = \text{Sr},^{18} \text{Ba}^{19}$  exhibiting a Si–Si bond have been observed. On the majority highly condensed (oxo)nitridosilicates are obtained, due to the high possible degree of crosslinking, leading predominantly to three-dimensional networks. The synthesis and characterization of nitridosilicates with a low degree of condensation is challenging due to their high sensitivity to oxygen or moisture. The oxonitridosilicate  $\text{Gd}_3[\text{SiN}_3\text{O}]\text{O}^{20}$  represents a nitridic example of an orthosilicate, while  $\text{Ba}_5\text{Si}_2\text{N}_6^{10}$  exhibits a group silicate structure composed of pairs of edge-sharing tetrahedra (cf.  $\text{Ca}_5[\text{Si}_2\text{N}_6]$  and  $\text{Ca}_7[\text{NbSi}_2\text{N}_9]$ ).<sup>21</sup> *Schleid et al.* reported  $\text{Pr}_9\text{Se}_6[\text{Si}_3\text{N}_9]$ , the only known nitrido-cyclo-trisilicate so far.<sup>22</sup> Very recently,  $\text{M}_5\text{Si}_3\text{N}_9$  ( $\text{M} = \text{La}, \text{Ce}$ ) the first nitridosilicate with a chain-like Si–N substructure was reported.<sup>23</sup> The compound is built up of alternating  $\text{Q}^2$ - and  $\text{Q}^3$ -type corner-sharing  $[\text{SiN}_4]$  tetrahedra with additional corner-sharing  $\text{Q}^1$ -units attached to the  $\text{Q}^3$ -tetrahedra, forming branched, zipper-like chains.

The combination of nitridosilicate anionic host structures with mixed valence Eu cations is an unknown phenomenon. In the year 1997 *Huppertz et al.* reported on the first nitridosilicate ( $\text{EuYbSi}_4\text{N}_7$ ), comprising  $\text{Eu}^{2+}$  and  $\text{Yb}^{3+}$  cations.<sup>24</sup> More recently, *Hintzen et al.* mentioned the isostructural compound  $\text{EuYSi}_4\text{N}_7$ .<sup>25</sup> SIALON substitution variants  $\text{EuLn}[\text{Si}_{4-x}\text{Al}_x\text{O}_x\text{N}_{7-x}]$  with  $\text{Ln}^{3+} = \text{Ho}–\text{Yb}$  have been extensively investigated by *Lieb et al.*<sup>26</sup> A different nitridosilicate  $\text{BaEu}(\text{Ba}_{0.5}\text{Eu}_{0.5})\text{YbSi}_6\text{N}_{11}$  was reported by *Huppertz*,<sup>27</sup> containing alkaline earth and rare earth ions in a three-dimensional network structure of vertex-sharing  $[\text{SiN}_4]$  tetrahedra.

In this contribution we describe the synthesis, crystal structure, and magnetic properties of the mixed valence chain-type nitridosilicate  $\text{Eu}_2\text{SiN}_3$  comprising both  $\text{Eu}^{2+}$  as well as  $\text{Eu}^{3+}$ .

## Experimental Section

Unless otherwise stated, all manipulations were performed with rigorous exclusion of oxygen and moisture in flame-dried Schlenk-type glassware on a Schlenk line, interfaced to a vacuum line ( $10^{-3}$  mbar), or in an argon-filled glove box. Argon (Messer-Griesheim, 5.0) was purified by passage over columns of silica gel (Merck),

**Table 1.** Crystal Data of  $\text{Eu}_2\text{SiN}_3$

formula	$\text{Eu}_2\text{SiN}_3$
formula weights [ $\text{g}\cdot\text{mol}^{-1}$ ]	374.04
crystal system	orthorhombic
space group	<i>Cmca</i> (No. 64)
diffractometer type	Nonius Kappa-CCD
radiation $\lambda$ [pm]	70.930 (Mo– $\text{K}\alpha$ )
crystal size [ $\text{mm}^3$ ]	0.05 mm $\times$ 0.05 mm $\times$ 0.04 mm
unit cell dimensions	
<i>a</i> [pm]	542.3(11)
<i>b</i> [pm]	1061.0(2)
<i>c</i> [pm]	1162.9(2)
cell volume [ $10^6 \text{ pm}^3$ ]	669.1(2)
formula units <i>Z</i>	8
$\rho$ [ $\text{g}\cdot\text{cm}^{-3}$ ]	7.43
<i>F</i> (000)	1288
$\mu$ [ $\text{mm}^{-1}$ ]	37.3
temperature [K]	200(2)
$\theta$ range [deg]	3.1–34.4
absorption correction	multi-scan (SADABS) <sup>30</sup>
measured reflections	7709
independent reflections	768
observed reflections	749 ( $F_0^2 \geq 2\sigma(F_0^2)$ )
no. of refined parameter	37
min/max $\Delta\rho_c$ [ $\text{e}\cdot\text{\AA}^{-3}$ ]	–1.22/1.35
goodness of fit	1.182
$R_1/R_1$ (all data)	0.0159/0.0167
$wR_2$ (all data)	0.0320
weighting details	$w^{-1} = \sigma^2 F_0^2 + (0.0P)^2 + 4.760P$ $P = (F_0^2 + 2F_c^2)/3$

molecular sieve (Fluka, 4 Å), KOH (Merck,  $\geq 85\%$ ),  $\text{P}_4\text{O}_{10}$  (Roth,  $\geq 99\%$ , granulate) and a titanium sponge at 700 °C (Johnson Matthey, 99.5%, grain size  $\leq 0.8$  cm).

$\text{Eu}_2\text{SiN}_3$  was synthesized from 130.6 mg (0.86 mmol) europium (Smart-Elements, 99.9%, pieces), 25 mg (0.43 mmol) silicon diimide (synthesized according to the reference 28), 19.8 mg Li (2.86 mmol) and 14.0 mg  $\text{LiN}_3$  (0.29 mmol) in tantalum crucibles (wall thickness 0.5 mm, internal diameter 10 mm, length 300 mm). The latter were arc-welded under a pressure of 1 bar purified argon. The crucible holder was water-cooled in order to avoid a start of the reaction during welding. The crucible was placed in a silica tube (under argon) inside the middle of a tube furnace. The temperature was raised to 900 °C (rate  $120\text{ }^\circ\text{C}\cdot\text{h}^{-1}$ ), maintained for 24 h, subsequently cooled to 500 °C (rate  $5\text{ }^\circ\text{C}\cdot\text{h}^{-1}$ ), and finally quenched to room temperature by switching off the furnace.  $\text{Eu}_2\text{SiN}_3$  crystallizes in black crystals, which hydrolyze rapidly after exposure to air or moisture.

Single-crystal X-ray data were collected on a Nonius Kappa-CCD diffractometer with graphite-monochromated Mo– $\text{K}\alpha$  radiation ( $\lambda = 70.93$  pm) with graded multilayer X-ray optics. The structure was solved by direct methods using SHELXS-97<sup>29</sup> and refined with anisotropic displacement parameters by full-matrix least-squares calculations on  $F^2$  (SHELXL-97)<sup>29</sup> in the orthorhombic space group *Cmca* (no. 64). Details of the X-ray data collection, structural refinements, final atomic coordinates and equivalent displacement parameters are listed in Tables 1, 2, and 3 (the CIF file can be found in the Supporting Information). The data were semi-empirically corrected for absorption by using the software SADABS.<sup>30</sup> Further details of the crystal structure investigations can be obtained from the Fachinformationszentrum Karlsruhe, 76344 Eggenstein-Leopoldshafen, Germany (fax: (+49)7247-808-666; E-mail: crysdata@fiz-karlsruhe.de) on quoting the depository number CSD-420679.

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**Table 2.** Atomic Coordinates of  $\text{Eu}_2\text{SiN}_3$ 

atom	Wyckoff	x	y	z	$U_{\text{eq}}$
Eu1	8f	1/2	0.04545(2)	0.35413(2)	0.00410(7)
Eu2	8f	0	0.14392(2)	0.57650(2)	0.00365(6)
Si1	8f	0	0.23623(13)	0.33629(9)	0.00099(18)
N1	8f	0	0.0750(4)	0.3738(3)	0.0047(6)
N2	8e	1/4	0.2757(4)	0.2500(4)	0.0053(4)
N3	8f	0	0.3326(4)	0.4564(3)	0.0055(6)

The crystal structure was verified by X-ray powder diffraction using Debye–Scherrer geometry on a STOE Stadi P powder diffractometer with Ge(111)-monochromatized Mo– $K_{\alpha 1}$  radiation ( $\lambda = 71.04$  pm) and double checking with patterns calculated from single crystal data (Figure S1, Supporting Information).

Scanning electron microscopy was performed on a JEOL JSM-6500F equipped with a field emission gun at an acceleration voltage of 10 kV. Samples were prepared by placing single crystals on adhesive conductive pads and subsequently coating them with a thin conductive carbon film. Each EDX spectrum (Oxford Instruments) was recorded with the analyzed area limited on one single crystal to avoid the influence of possible contaminating phases. Analysis of three spots per crystallite showed average atomic Eu/Si/O/N compositions (%) of 73.8(6):7.4(2):0:18.8(2). Despite the large standard deviations of the measurements, these results confirm the presence of the elements and are in agreement with the compositions obtained from single crystal X-ray studies (the calculated values Eu/Si/O/N are 81.3:7.5:0:12.2).

Magnetic moments were measured using a SQUID magnetometer (MPMS-XL5, Quantum Design Inc.) between 1.8 and 300 K with magnetic flux densities up to 5 tesla. A sample of about 16 mg was loaded in a gelatin capsule and fixed in a straw as sample holder. Corrections for the sample holder and the core diamagnetism<sup>31</sup> were applied to the data. Effective magnetic moments were calculated by fitting the corrected molar magnetic susceptibility data  $\chi_m$  with the equation  $\chi_m = C/(T - \theta_{\text{CW}})$  where  $C$  is the Curie constant and  $\theta_{\text{CW}}$  is the paramagnetic Curie–Weiss temperature (Weiss constant). The small magnetic contribution of the trivalent europium ions was taken into account by the Van Vleck equation using a spin–orbit coupling parameter  $\zeta_4(\text{Eu}^{3+})$  of  $1326 \text{ cm}^{-1}$ .<sup>31</sup>

The 21.53 keV transition of  $^{151}\text{Eu}$  with an activity of 130 MBq (2% of the total activity of a  $^{151}\text{Sm}:\text{EuF}_3$  source) was used for the  $^{151}\text{Eu}$  Mössbauer spectroscopy. The measurements were performed in the usual transmission geometry in commercial helium bath and flow cryostats. The temperature of the absorber could be varied from 4.2 to 300 K. The source was kept at room temperature in all experiments. The material for the Mössbauer spectroscopy investigation was the same as for the susceptibility measurements. The sample was diluted with silica and placed within a thin-walled glass container at a thickness corresponding to approximately 10 mg Eu/cm<sup>2</sup>.

The electronic structure of  $\text{Eu}_2\text{SiN}_3$  was studied with *ab initio* density functional theory methods<sup>32</sup> using the generalized gradient approximation (GGA) with the Perdew *et al.* Scheme<sup>33</sup> for the exchange–correlation functional. We used the WIEN2k package, which is based on a full-potential augmented-plane-wave method.<sup>34</sup> To correct for neglected correlation effects, we used the GGA+EECE (exact exchange of correlated electrons) method as implemented in the WIEN2k package.<sup>35</sup> The plane wave cutoff was set to

$R_{\text{min}}^{\text{MT}}K_{\text{max}} = 7$ , leading to 8662 plane waves. All calculations were done with 100k points in the Brillouin zone.

## Results and Discussion

**Crystal Structure.**  $\text{Eu}_2\text{SiN}_3$  is a chain-type nitridosilicate comprising one-dimensional infinite nonbranched *zweier* chains<sup>36</sup> of corner-sharing  $\text{SiN}_4$  tetrahedra running parallel [100]. The chains of  $\text{Eu}_2\text{SiN}_3$  exhibit a maximum stretching factor  $f_s = 1$ , comparable to  $\text{Zn}(\text{PO}_3)_2$  ( $f_s = 1.0$ ),  $\text{Ca}_2\text{PN}_3$  ( $f_s = 1.0$ )<sup>37,38</sup> or Johannsenite  $\text{CaMg}[\text{Si}_2\text{O}_6]$  ( $f_s = 0.98$ ).<sup>39</sup>  $\text{Eu}_2\text{SiN}_3$  is isostructural with the nitridophosphate  $\text{Ca}_2\text{PN}_3$ ,<sup>37,38</sup> with  $\text{Rb}_2\text{TiO}_3$ <sup>40</sup> and the alkali oxocobaltates(IV)  $\text{M}_2\text{CoO}_3$  with  $\text{M} = \text{K}, \text{Rb}, \text{Cs}$ .<sup>41</sup> It represents the first example of a nonbranched chain-type silicate in the class of nitridosilicates (Figure 1).

Details of the X-ray data collection, structural refinements, final atomic coordinates, and equivalent displacement parameters are listed in Tables 1, 2, and 3.

The Si–N bond lengths within the chains (173–177 pm) vary only in a small range, and the bridging angle Si–N–Si between the  $[\text{SiN}_4]$  tetrahedra amounts to  $152.11^\circ$  (Figure 2).

There are two crystallographically distinct europium sites. The Eu atoms are coordinated in a distorted pentagonal bipyramidal manner by the nitrogen atoms of the  $[\text{Si}_2\text{N}_6]^{10-}$  chains, whereby each Eu atom is coordinated to three chains. Charge neutrality in  $\text{Eu}_2\text{SiN}_3$  is achieved by two valence states of the europium sites, namely  $\text{Eu}^{2+}$  and  $\text{Eu}^{3+}$  ions occupying the two different Wyckoff sites (8f). The Eu1 cation is coordinated by nitrogen atoms with distances ranging between 255 and 305 pm, while Eu2 exhibits N neighbors between 240 and 275 pm (Figure 3, Table 4). These differences in bond length of the two metal sites can analogously be found in the isostructural compound  $\text{Ca}_2\text{PN}_3$  (Ca1–N: 244–300 pm; Ca2–N: 242–262 pm),<sup>38</sup> probably favoring the crystallization of the mixed valence compound  $\text{Eu}_2\text{SiN}_3$  in this structure type.

These structural data already indicate that there is an ordered distribution of larger  $\text{Eu}^{2+}$  (Eu1) and smaller  $\text{Eu}^{3+}$  (Eu2). Typical distances for  $\text{Eu}^{2+}$ –N in nitridosilicates range between 260–325 pm ( $\text{Eu}_2\text{Si}_5\text{N}_8$ ) and 293–306 pm ( $\text{EuYbSi}_4\text{N}_7$ ).<sup>24</sup> These distances resemble those of the Eu1–N coordination. Since no mixed valence europium nitridosilicate besides  $\text{Eu}_2\text{SiN}_3$  is known so far, only analogous compounds can be used for comparison. The distances for  $\text{Eu}^{3+}$ –N in  $\text{EuN}$  are 250.7 pm,<sup>42</sup> comparable and within the range of the values of Eu2. A rather suitable compound for comparison seems to be  $\text{Eu}_3\text{O}_4$ ,<sup>43</sup> containing mixed valence europium which connects oxygen atoms. The distances  $\text{Eu}^{2+}$ –O (264–296 pm) are significantly longer than those of  $\text{Eu}^{3+}$ –O (230–249 pm). Compared to the Eu–N bonds in  $\text{Eu}_2\text{SiN}_3$  the Eu–O bonds are generally shorter, due to the smaller and more electronegative oxygen. The shortest inter-

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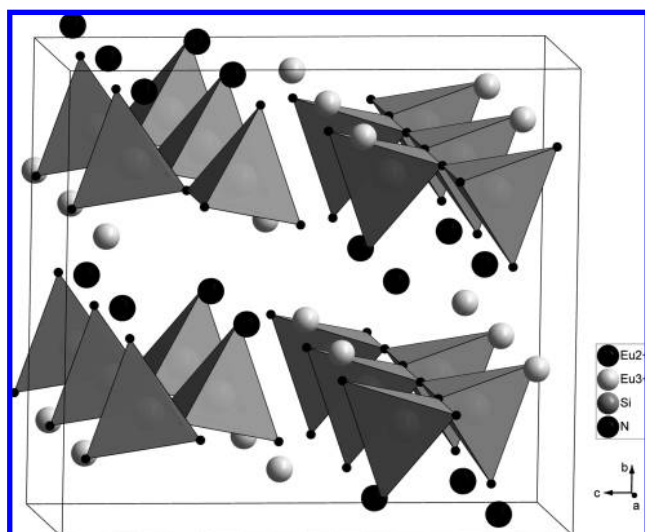
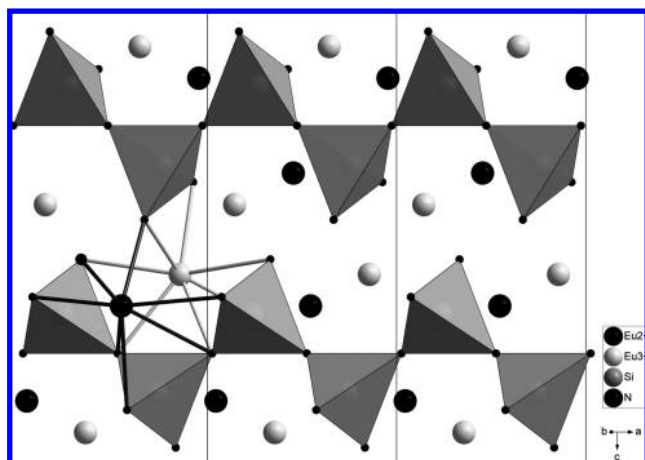
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**Table 3.** Anisotropic Displacement Parameters<sup>a</sup> of  $\text{Eu}_2\text{SiN}_3$ 

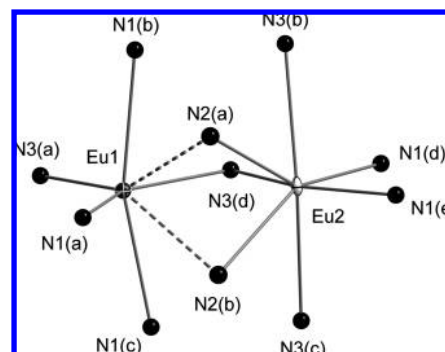
atom	$U_{11}$	$U_{22}$	$U_{33}$	$U_{23}$	$U_{13}$	$U_{12}$
Eu1	0.00461(10)	0.00428(10)	0.00340(10)	0.00056(6)	0.00000	0.00000
Eu2	0.00563(10)	0.00284(10)	0.00247(10)	-0.00005(6)	0.00000	0.00000
Si1	0.0006(4)	0.0016(5)	0.0008(4)	-0.0002(3)	0.00000	0.00000
N1	0.0070(16)	0.0030(14)	0.0040(13)	-0.0013(12)	0.00000	0.00000
N2	0.0033(13)	0.0077(15)	0.0050(13)	0.000	0.0023(11)	0.00000
N3	0.0080(16)	0.0050(15)	0.0034(14)	0.0013(12)	0.00000	0.00000

<sup>a</sup> The anisotropic temperature factor is given as  $\exp[-2\pi^2(U_{11}h^2a^{*2} + \dots + 2U_{13}hla^*c^*)]$ ;  $U_{eq}$  is defined as one-third of the trace of the  $U_{ij}$  tensor.

**Figure 1.** Perspective view of  $\text{Eu}_2\text{SiN}_3$  along [100].**Figure 2.** View along the  $[\text{Si}_2\text{N}_6]_n^{10-}$  chains of  $\text{Eu}_2\text{SiN}_3$ . The two Eu sites connect three chains with one another.

atomic distance between the europium sites is 339 pm which resembles those in  $\text{Eu}_2\text{Si}_5\text{N}_8$  (341.7 pm)<sup>24</sup> and  $\text{EuN}$  (354.8 pm).<sup>42</sup>

**Lattice Energy Calculations According to the Maple Concept.** MAPLE (Madelung Part of Lattice Energy)<sup>44</sup> calculations were performed on  $\text{Eu}_2\text{SiN}_3$ . This method is especially useful for the mixed valence compound since it exclusively considers the electrostatic interactions in ionic crystals, depending on the distance, charge, and coordination of the constituting atoms. The MAPLE sum values of  $31146 \text{ kJ}\cdot\text{mol}^{-1}$  were

**Figure 3.** Nitrogen coordination of Eu1 and Eu2 in  $\text{Eu}_2\text{SiN}_3$  (ellipsoids with a probability of 90 %).**Table 4.** Selected Interatomic Distances [pm] and Angles [deg] in  $\text{Eu}_2\text{SiN}_3$  (standard deviation in parentheses)

Eu1–Eu2	339.3(6) pm	Si–N1	173.3(3) pm
Eu1–N1(a)	267.2(1) pm	Si–N2	176.4(3) pm
Eu1–N1(b)	274.0(1) pm	Si–N3	173.8(2) pm
Eu1–N1(c)	274.0(1) pm		
Eu1–N2(a)	304.5(1) pm	Si–N2–Si	152.1(2) deg
Eu1–N2(b)	304.5(1) pm		
Eu1–N3(a)	255.2(1) pm		
Eu1–N3(d)	255.3(1) pm		
Eu2–Eu1	339.3(6) pm		
Eu2–N1(d)	246.5(1) pm		
Eu2–N1(e)	239.5(1) pm		
Eu2–N2(a)	257.6(1) pm		
Eu2–N2(b)	257.6(1) pm		
Eu2–N3(b)	275.0(1) pm		
Eu2–N3(c)	275.0(1) pm		
Eu2–N3(d)	244.2(1) pm		

calculated starting from the refined structural data with  $\text{Eu}^{2+}$  at the Eu1 site and  $\text{Eu}^{3+}$  at Eu2 (Table 5,  $\text{Eu}_2\text{SiN}_3$  1).

The values are close to the sum of the total MAPLE values of the starting materials (difference 0.4 %). The calculated partial MAPLE values of the crystallographically different atoms are within their typical range. Especially the  $\text{Eu}^{3+}$  values compare well with calculated MAPLE values of  $\text{EuN}^{42}$  or  $\text{Eu}_3\text{O}_4$ ,<sup>43</sup> respectively. The \*MAPLE values (MAPLE/charge) exhibit the same value for  $\text{Eu1} = \text{Eu}^{2+}$  and  $\text{Eu2} = \text{Eu}^{3+}$ , corroborating this assignment. If the occupancy of the europium sites is interchanged ( $\text{Eu1} \rightarrow \text{Eu}^{3+}$  and  $\text{Eu2} \rightarrow \text{Eu}^{2+}$ ,  $\text{Eu}_2\text{SiN}_3$  2), the MAPLE value of the compound differs 1.5 % to the sum of the total MAPLE values of the starting materials. Furthermore, the partial europium values change significantly to anomalous MAPLE values for  $\text{Eu}^{2+}$  and  $\text{Eu}^{3+}$ . Calculations with a charge of +2.5 on both europium sites (not displayed) revealed a favoritism of the Eu1 site for lower charges and a difference of 1.5 % to the sum of the total MAPLE values.

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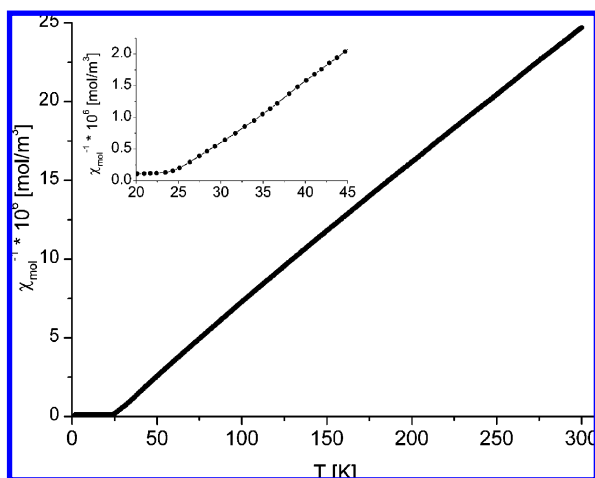
**Table 5.** Results of the MAPLE Calculations for  $\text{Eu}_2\text{SiN}_3$  and Increment Calculations in  $\text{kJ}\cdot\text{mol}^{-1}$ 

$\text{Eu}_2\text{SiN}_3$ (1)	MAPLE	*MAPLE <sup>a</sup>	$\text{Eu}_2\text{SiN}_3$ (2)	MAPLE	*MAPLE <sup>a</sup>
Eu1 = $\text{Eu}^{2+}$	1910	478	Eu1 = $\text{Eu}^{3+}$	3754	417
Eu2 = $\text{Eu}^{3+}$	4254	473	Eu2 = $\text{Eu}^{2+}$	2228	557
Si	9960		Si	10018	
N1	4785		N1	4786	
N2	5459		N2	5396	
N3	4778		N3	4637	

Typical Partial MAPLE <sup>45</sup> Values <sup>b</sup>		
MAPLE ( $\text{EuN} + 1/2 \text{Eu}_2\text{Si}_5\text{N}_8 - 1/2 \text{Si}_3\text{N}_4$ )	MAPLE $\text{Eu}_2\text{SiN}_3$ (1) Eu1 = $\text{Eu}^{2+}$ ; Eu2 = $\text{Eu}^{3+}$	MAPLE $\text{Eu}_2\text{SiN}_3$ (2) Eu1 = $\text{Eu}^{3+}$ ; Eu2 = $\text{Eu}^{2+}$
31270	31146 $\Delta^c = 0.4\%$	30819 $\Delta^c = 1.5\%$

<sup>a</sup> MAPLE/charge. <sup>b</sup>  $\text{Eu}^{2+}$ : 1900–2000;  $\text{Eu}^{3+}$ : 4000–5000;  $\text{Si}^{4+}$ : 9000–10200;  $\text{N}^{3-}$ : 5000–6000. <sup>c</sup>  $\Delta$  = difference.



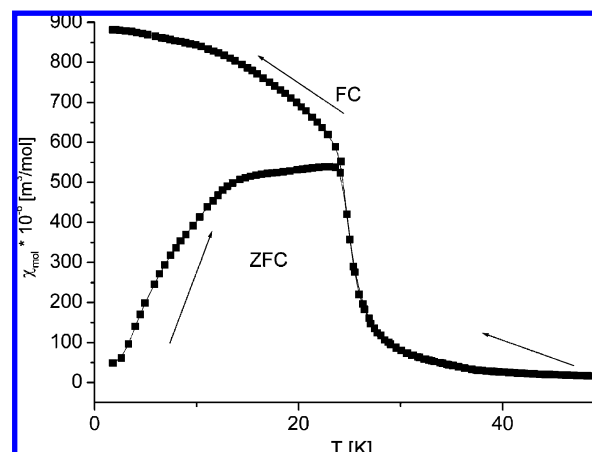
**Figure 4.** Temperature dependence of the inverse magnetic susceptibility of  $\text{Eu}_2\text{SiN}_3$  measured at 0.1 T. The low-temperature behavior is shown in the inset.

**Magnetic Measurements.** The temperature dependence of the inverse magnetic susceptibility measured at magnetic flux density of 0.1 T is shown in Figure 4.

By fitting the  $1/\chi(T)$  data using the Curie–Weiss law only, we obtain an effective magnetic moment  $\mu_{\text{eff}} = 8.12 \mu_{\text{B}}$  per formula unit, which is too high compared with  $7.94 \mu_{\text{B}}$  as expected for  $\text{Eu}^{2+}$ . By considering the Van Vleck contribution of the  $\text{Eu}^{3+}$  ions with a fixed spin-orbit coupling parameter  $\zeta_{4f}$  of  $1326 \text{ cm}^{-1}$  taken from ref 31, the fit resulted in an effective magnetic moment of  $7.5 \mu_{\text{B}}$ . This is in good agreement with  $\text{Eu}^{2+}$  if we bear in mind the crystal field dependency of the  $\zeta_{4f}$  parameter. We were not able to obtain a reliable fit of  $\zeta_{4f}$  from our data; thus, our approach is only an estimation of the small, but noticeable,  $\text{Eu}^{3+}$  contribution. The valence combinations  $\text{Eu}^{2+}\text{Eu}^{2+}\text{SiN}_3$  or  $\text{Eu}^{3+}\text{Eu}^{3+}\text{SiN}_3$  would exhibit magnetic moments of  $11.2 \mu_{\text{B}}$  and  $\sim 0 \mu_{\text{B}}$ , respectively, underlining the presence of mixed valence europium in  $\text{Eu}_2\text{SiN}_3$ .

The positive Weiss constant ( $\theta_{\text{CW}} = 28 \text{ K}$ ) is indicative of ferromagnetic interactions. Subsequent zero-field-cooled (ZFC) and field-cooled (FC) measurements at a magnetic flux density of 3 mT (Figure 5) confirmed the ferromagnetic behavior below  $T_{\text{C}} = 24 \text{ K}$ .

The Curie temperature of  $\text{Eu}_2\text{SiN}_3$  is rather high compared to other ferromagnetic Eu compounds, as collected in Table 6. Europium oxide  $\text{EuO}$  exhibits a very high  $T_{\text{C}}$  value compared



**Figure 5.** Low-temperature susceptibility of  $\text{Eu}_2\text{SiN}_3$  measured at an external field strength of 0.003 T in the zero-field-cooling (ZFC) and field-cooling (FC) modus. Ferromagnetic behavior below  $T_{\text{C}} = 24 \text{ K}$  can be observed.

**Table 6.** Curie Temperatures and Weiss Constants of  $\text{Eu}_2\text{SiN}_3$  and Selected Eu Compounds

group	compound	$T_{\text{C}}$ [K]	$\theta$ [K]
(oxo)nitridosilicates	$\text{Eu}_2\text{SiN}_3$	24	28
	$\text{Eu}_2\text{Si}_5\text{N}_8$ <sup>46</sup>	13	18
	$\text{EuSi}_2\text{O}_2\text{N}_2$ <sup>16</sup>	4.5	2.2
silicates	$\text{Eu}_2\text{SiO}_4$ <sup>47</sup>	10	10
	$\text{Eu}_3\text{SiO}_5$ <sup>47</sup>	19	19
oxides	$\text{EuO}$ <sup>48</sup>	69	76
sulfides	$\text{EuS}$ <sup>48</sup>	16	19
halides	$\text{EuI}_2$ <sup>48</sup>	5	5

to the higher homologue  $\text{EuS}$ . Europium silicates typically show lower values within the range of the nitridosilicate  $\text{Eu}_2\text{Si}_5\text{N}_8$ .<sup>46</sup>

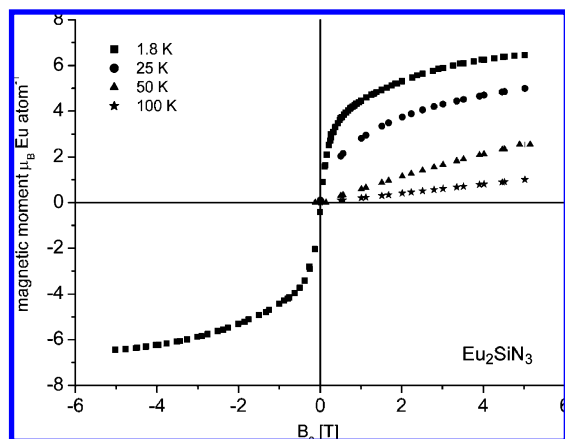
We have finally measured the isothermal magnetization at 1.8, 25, 50, and 100 K (Figure 6). The magnetizations increase linearly at 50 and 100 K, as expected for a paramagnetic compound. Close to the magnetic ordering temperature, the curve is much steeper, and the bend indicates the onset of magnetic ordering.

At 1.8 K, the magnetization is almost saturated close to  $6.5 \mu_{\text{B}}/\text{f.u.}$  at 5 T, in very good agreement with the theoretical value of  $7 \mu_{\text{B}}/\text{f.u.}$  for one  $\text{Eu}^{2+}$  and one  $\text{Eu}^{3+}$  atom according to  $gJ \times J$ .<sup>31</sup> Two  $\text{Eu}^{2+}$  sites would exhibit much higher values around

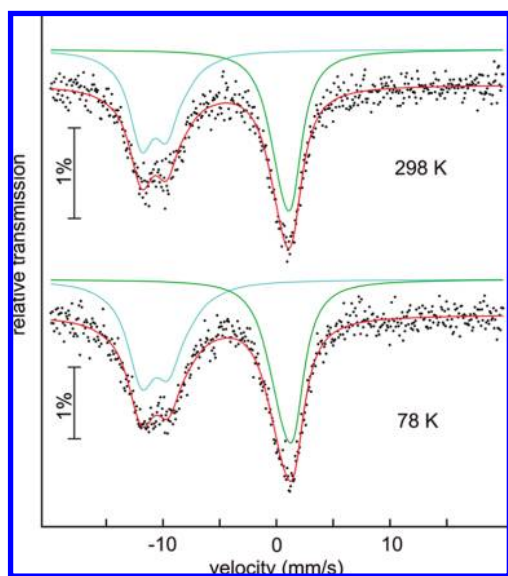
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**Figure 6.** Magnetization versus external field strength of  $\text{Eu}_2\text{SiN}_3$  measured at 1.8, 25, 50, and 100 K.

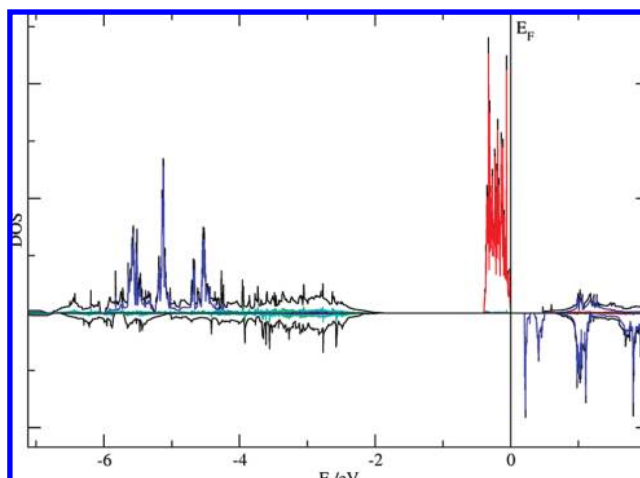


**Figure 7.** Experimental (red line: fitting of measured black points) and simulated (blue:  $\text{Eu}^{2+}$ ; green  $\text{Eu}^{3+}$ )  $^{151}\text{Eu}$  Mössbauer spectra of  $\text{Eu}_2\text{SiN}_3$  at 78 and 298 K.

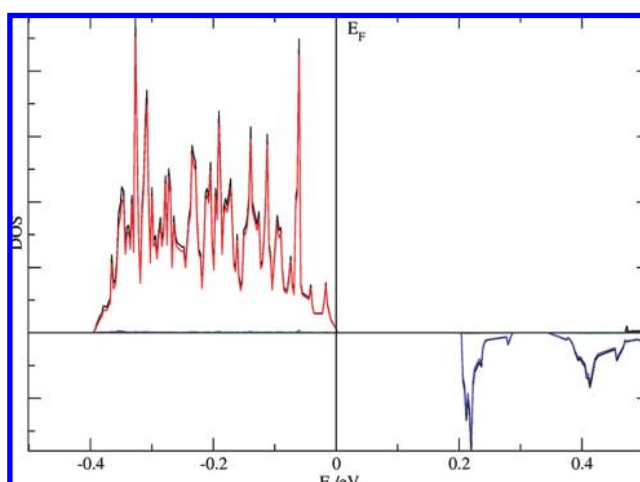
$14 \mu_{\text{B}}$ . The paramagnetic moment and the saturation magnetization at low temperatures are comparable to the values of the nitridosilicates  $\text{Eu}_2\text{Si}_5\text{N}_8$  and  $\text{EuSi}_2\text{O}_2\text{N}_2$ .<sup>16,46</sup> With respect to the very small ferromagnetic hysteresis,  $\text{Eu}_2\text{SiN}_3$  might be classified as a soft ferromagnet.

**Mössbauer Spectroscopy.** The  $^{151}\text{Eu}$  Mössbauer spectra of  $\text{Eu}_2\text{SiN}_3$  at 78 and 298 K are presented in Figure 7 together with transmission integral fits. The corresponding fitting parameters for these and additional measurements at other temperatures are listed in Table 7.

The spectra clearly show two well-separated signals with equal intensity. The signal at an isomer shift of  $-10.57 \text{ mm/s}$



**Figure 8.** Density-of-states (DOS) of  $\text{Eu}_2\text{SiN}_3$ , calculated with the GGA+EECE approach. Eu1 (red); Eu2 (blue), and Si–N (turquoise) bands.



**Figure 9.** Density-of-states (DOS) of  $\text{Eu}_2\text{SiN}_3$ , calculated with the GGA+EECE approach. An energy gap of  $\sim 0.2 \text{ eV}$  discerns the differences between the bonding Eu1 (red) and antibonding Eu2 (blue) bands.

(78 K data) can be attributed to the  $\text{Eu}^{2+}$  ions. It is subject to quadrupole splitting of  $-7.5 \text{ mm/s}$ , and furthermore, an asymmetry parameter of  $\eta = 0.7$  was included in the fit. Compared to the recently reported nitridosilicates  $\text{Eu}_2\text{Si}_5\text{N}_8$ <sup>46</sup> ( $-11.88 \text{ mm/s}$ ) and  $\text{EuSi}_2\text{O}_2\text{N}_2$ <sup>16</sup> ( $-12.3 \text{ mm/s}$ ),  $\text{Eu}_2\text{SiN}_3$  shows by far the smallest isomer shift. This is indicative of a higher degree of covalent bonding in  $\text{Eu}_2\text{SiN}_3$ . Gerth et al.<sup>49</sup> have systemized the course of the isomer shift vs. the ionicity of the respective chemical bonds. A decrease of the isomer shift (also a decrease of the electron density at the nuclei) reflects the participation of the 6s electrons in the covalent bonding. This tendency can be observed by shorter bond lengths of the  $\text{Eu}^{2+}$  site in  $\text{Eu}_2\text{SiN}_3$  compared to  $\text{Eu}_2\text{Si}_5\text{N}_8$  (260–325 pm) and  $\text{EuSi}_2\text{O}_2\text{N}_2$  (280–293

**Table 7.** Fitting Parameters for  $^{151}\text{Eu}$  Mössbauer Spectroscopic Measurements on  $\text{Eu}_2\text{SiN}_3$ <sup>a</sup>

$T$ (K)	$\delta_1$ (mm/s)	$\Gamma_1$ (mm/s)	$\Delta E_{Q1}$ (mm/s)	$\eta$	$\delta_2$ (mm/s)	$\Gamma_2$ (mm/s)	$\Delta E_{Q2}$ (mm/s)	%
35	$-10.67(6)$	$2.7(2)$	$-7.5(3)$	0.7	$0.92(5)$	$2.2(2)$	$4.2(4)$	53(2)/47(2)
78	$-10.57(4)$	$2.8(1)$	$-7.5(2)$	0.7	$0.89(3)$	$2.3(1)$	$3.9(3)$	51(1)/49(1)
150	$-10.76(6)$	$2.9(2)$	$-7.4(3)$	0.7	$0.84(4)$	$2.4(2)$	$3.0(6)$	54(2)/46(2)
298	$-10.66(4)$	$2.5(1)$	$-7.0(2)$	0.7	$0.85(4)$	$2.3(1)$	$3.2(3)$	49(1)/51(1)

<sup>a</sup> Isomer shift ( $\delta$ ), electric quadrupole interaction ( $\Delta E_Q$ ), asymmetry parameter ( $\eta$ ), and experimental line width ( $\Gamma$ ). The indices 1 and 2 refer to the  $\text{Eu}^{2+}$  and  $\text{Eu}^{3+}$  sites. For details see text.

pm). The second signal at 0.89 mm/s (78 K data) is due to  $\text{Eu}^{3+}$ . The latter also shows electric quadrupole splitting, however, with a smaller absolute  $\Delta E_Q$  value as compared to  $\text{Eu}^{2+}$ . Over the whole temperature range investigated (Table 7), the isomer shift values of  $\text{Eu}^{2+}$  and  $\text{Eu}^{3+}$  do not vary more than one standard deviation. This is indicative for stable static mixed valence in  $\text{Eu}_2\text{SiN}_3$ , in good agreement with the magnetic data. In contrast to  $\text{Eu}_{14}\text{Cl}_{33}$ ,<sup>50</sup> no temperature-dependent change in the  $\text{Eu}^{2+}/\text{Eu}^{3+}$  ratio could be observed in the Mössbauer spectra, probably due to the more rigid lattice of the nitridosilicate compound.<sup>51</sup> The valence behavior of  $\text{Eu}_2\text{SiN}_3$  is similar to other static intermediate valence compounds such as  $\text{Eu}_2\text{CuS}_3$ ,<sup>52</sup>  $\text{Eu}_3\text{F}_4\text{S}_2$ ,<sup>53</sup>  $\text{Eu}_5\text{Sn}_3\text{S}_{12}$ ,<sup>54</sup> or  $\text{Eu}_5\text{Zr}_3\text{S}_{12}$ .<sup>55</sup>

The  $\text{Eu}_2\text{SiN}_3$  sample has also been measured at 4.2 K, well below the magnetic ordering temperature of  $T_C = 24$  K. The divalent europium site showed magnetic hyperfine field splitting; however, although the total counting time was 5 days, the resulting spectrum had a poor signal-to-background ratio. A reliable fit of the data was not possible.

**Electronic Structure Calculations.** The electronic structure of  $\text{Eu}_2\text{SiN}_3$  was studied using *ab initio* density functional theory (DFT) methods.<sup>32</sup> Experimental lattice parameters and atomic coordinates were used. The total density of states (DOS) together with the contributions of the Eu *5f* orbitals as well as the silicon and nitrogen atoms is displayed in Figure 8.

The DOS just below the Fermi level is dominated by Eu1 ( $\text{Eu}^{2+}$ ) *4f* states (−0.4 to 0 eV). Above the Fermi level, we find an energy gap  $E_G$  of ~0.2 eV between filled Eu1 and unfilled Eu2 states (Figure 9).

Since the DFT generally underestimates band gaps, the experimental gap is expected to be larger, but certainly below 1 eV due to the black color of the compound. We calculate 6.7 unpaired electrons at the Eu1 ( $\text{Eu}^{2+}$ ) site and 6.0 unpaired electrons at the Eu2 ( $\text{Eu}^{3+}$ ) site in very good agreement with the valence assignments suggested by the MAPLE calculations and the interatomic distances.

## Conclusion

$\text{Eu}_2\text{SiN}_3$  features the structural motif of one-dimensional infinite nonbranched *zweier* chains, hitherto unknown in nitri-

dosilicates, and additionally is the first representative of mixed valence nitridosilicates. Magnetic measurements revealed a magnetic moment of  $7.5 \mu_B/\text{f.u.}$  and ferromagnetic order was detected below 24 K.  $^{151}\text{Eu}$ –Mössbauer spectroscopy confirmed the ratio  $\text{Eu}^{2+}/\text{Eu}^{3+}$  of 1:1. Lattice energy calculations (MAPLE) allowed the crude assignment of the mixed valence europium sites. A small band gap of ~0.2 eV was determined by DFT calculations using GGA+EECE, additionally the assignment of the  $\text{Eu}^{2+}$  and  $\text{Eu}^{3+}$  sites was confirmed. Therefore,  $\text{Eu}_2\text{SiN}_3$  may be classified as a static mixed valent europium compound, regardless of its black color, which implies some polaron activities. Further investigations about the exchange of the  $\text{Eu}^{3+}$  or  $\text{Eu}^{2+}$  ion are currently in progress, which might lead to luminescent materials with interesting magnetic properties.

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**Supporting Information Available:** Complete ref 32 and XRD-pattern ( $\text{Mo-K}_{\alpha 1}$ ) of  $\text{Eu}_2\text{SiN}_3$ . This material is available free of charge via the Internet at <http://pubs.acs.org>.

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