Phase transition in nanomagnetite

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Recently, the application of nanosized magnetite particles became an area of growing interest for their potential practical applications. Nanosized magnetite samples of 36 and 9 nm sizes were synthesized. Special care was taken on the right stoichiometry of the magnetite particles. Mössbauer spectroscopy measurements were made in 4.2–300 K temperature range. The temperature dependence of the intensities of the spectral components indicated size dependent transition taking place in a broad temperature range. For nanosized samples, the hyperfine interaction values and their relative intensities changed above the Verwey transition temperature value of bulk megnetite. The continuous transition indicated the formation of dendritelike granular assemblies formed during the preparation of the samples. © 2008 American Institute of Physics. [DOI: 10.1063/1.2937252]

I. INTRODUCTION

Magnetite, Fe_3O_4 , is a prototype of inverse spinels. At room temperature, it has a cubic spinel structure with a lattice parameter of 8.397 Å. The unit cell consists of a cage of 32 O²⁺ ions arranged in a face centered cubic lattice. Each unit cell has eight Fe³⁺ ions at the tetrahedral site (A sites) with four nearest neighbor O^{2+} ions, and eight Fe^{3+} and eight Fe^{2+} ions at the octahedral sites (B sites) with six nearest neighbor O²⁺ ions. Magnetite is a relatively good conductor at room temperature. On cooling below 120 K, its conductivity sharply drops by two orders of magnitude. Verwey evidenced¹ this sharp anomaly in 1939. The transition was explained² by a change from a dynamic disorder of electrons hopping on B sites above the Verwey transition (T_v) to a long range order of the B site cations in alternate plans of Fe^{3+} and Fe^{2+} ions. On cooling below T_V , the change from cubic (space group $Fd\overline{3}m$) to monoclinic (space group P2/c) structure yields a change in the local symmetry around the cations.³ The Verwey—Mott model assumes a purely electronic mechanism for the transition.⁴ Nevertheless, despite extensive studies, the mechanism responsible for the transition is still not completely understood.⁵

Recently, the application of small particle or nanoparticle magnetite became an area of growing interest due to their potential practical applications. In the case of magnetic nanostructures, data storage is a potential application.⁶ Also, the application of magnetic nanostructures in magnetic resonance imaging and magnetically steered drug delivery^{7–9} is expected.

Mössbauer spectrometry is a very sensitive method that provides information on the Fe ions that are located at different lattice positions, which results in different hyperfine parameters.¹⁰ Below T_V , the spectrum changed significantly.^{11,12} In the case of nanomagnetite, the dependence of the transition on the particle size is interesting but contradicting. A decrease^{13,14} and an increase¹⁵ in T_V were

measured depending on the size of the samples. However, the studies were performed only at some temperature values and not in a broad temperature range, and the preparation of the samples studied were made by using different methods. No transition was observed below 50 nm size.¹⁴ A change in the slope of the zero-field-cooled magnetization graph for 10 and 24 nm sized samples at 100 K was attributed to Verwey transition.¹⁶ Recently, the fractal formation (dendriticlike self-assembly) by magnetite nanoparticles was observed on porous silicon in aqueous media.¹⁷

Because of the different results obtained using special preparation techniques and with the samples measured at some temperature values, we intend to perform measurement on nanomagnetite samples synthesized by using classical methods. Our aim was to prepare bulk and nanomagnetite samples and measure the temperature dependence of Mössbauer spectra to compare the results and conclude on the temperature dependence of the crystal transformation for 9 and 36 nm sized magnetite samples. Up until now, the Mössbauer spectra of nanomagnetite samples have not been measured systematically in a broad temperature range. We measured the samples in a broad temperature range between 4.2 and 300 K.

II. EXPERIMENTAL DETAILS

Bulk magnetite crystals were prepared by reducing melted hematite in CO/CO_2 atmosphere.¹⁸ The reduction and cooling down took place for several hours. The nano-magnetite samples had to be synthesized by different chemical methods depending on the particle size. The sample of 36 nm thickness was prepared by the method described in Ref. 19. The sample of 9 nm size was prepared by the precipitation of magnetite from the basic aqueous solution of FeCl₂ and FeCl₃, as was described earlier.²⁰ After the syntheses, the iron content of the segregated samples was chemically analyzed. The results proved the stoichiometric Fe₃O₄ compositions. The surface of the samples was covered with oleic acid and oleylamine to keep nanodispersion and to

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avoid oxidation on the surface. The structure of the nanomagnetite samples was determined by the x-ray diffraction method. The diffraction results supported the cubic structure. The size distribution was determined by x-ray line profile analysis. The full width at half maximum (FWHM) of the profiles was plotted as a function of the length of the diffraction vector (Williamson-Hall method). These plots showed that the FWHM values are independent of the indices of reflections, which indicates that there are no significant lattice strains in the structure. In this case, the broadening of the diffraction profiles is caused by the smallness of the crystallites (grains) and the crystallite size distribution can be evaluated from a single profile. The crystallite size distribution was determined from the profile of (400) reflection because this peak was relatively strong and did not overlap with neighboring lines. The peak profile was evaluated by the convolutional multiple whole profile fitting method, which is described in detail elsewhere.²¹ In this procedure, the measured line profile was fitted by the convolution of the instrumental peak and the physical profile resulted by the crystallite size distribution. The instrumental profile was measured on a LaB6 peak profile standard material. In the fitting procedure the size distribution density [f(x)] was assumed to be a lognormal function, which is given by

$$f(x) = \frac{1}{\sqrt{2\pi\sigma}} \frac{1}{x} \exp\left[-\frac{\left[\ln(x/m)\right]^2}{2\sigma^2}\right],\tag{1}$$

where x is the crystallite size, and m and σ are the median and variance of the size distribution, respectively. The variance characterizes the width of the size distribution. The line profile fitting gives the values of m and σ directly. From m and σ , the mean crystallite size can be determined as²²

$$\langle x \rangle = m \exp(\sigma^2/2). \tag{2}$$

X-ray line profile analysis gives $m=9\pm 1$ nm, $\sigma = 0.31\pm 0.04$, and $\langle x\rangle = 9\pm 1$ nm for the 9 nm sample, and $m=34\pm 4$ nm, $\sigma=0.35\pm 0.04$, and $\langle x\rangle = 36\pm 4$ nm for the 36 nm sample. It can be seen that the width of the size distribution is the same within the experimental error, while the mean size value is four times higher for the larger sample. It was shown for different nanosized materials that the bar diagram of the crystallite size distribution obtained from transmission electron microscopy micrographs and the size distribution density function, f(x), determined by x-ray analysis give the same results.²²

The Mössbauer absorber samples were prepared in a very pure Ar atmosphere. To keep the dispersion of the nanoparticles, they were mixed with fine BN powder. The measurements at low temperatures were made in a cryostat containing liquid N_2 or He. The application of the Mössbauer spectrometry to determine the phase transition is very effective, because the method is an order-sensitive technique due to the strong dependence of the hyperfine parameters on the Fe-atom surroundings. Therefore, the local positions, the interaction of ions, and their oxidation state can simultaneously be determined. The Mössbauer spectra were measured using the spectrometer in a constant velocity mode. A single line source of 50 mCi ⁵⁷Co in a Rh matrix was used. The spectra



FIG. 1. Mössbauer spectra of bulk magnetite.

were fitted by using a least-squares fitting program. The isomer shift δ values in mm/s are given relative to that of α -iron at room temperature.

III. RESULTS AND DISCUSSION

The Mössbauer spectra of the bulk magnetite sample were measured at different temperatures, and four of them are shown in Fig. 1. The spectra were measured to perform a reliable comparison of the hyperfine parameters of bulk and nanomagnetite samples. The spectra above T_V consist of two components corresponding to Fe³⁺ and the intermediate ions in the lattice denoted by the average valence state Fe^{2.5+}. The intensity ratio of Fe³⁺/Fe^{2.5+} was 1:2 at room temperature, as can be expected for a stoichiometric sample assuming the same Lamb Mössbauer factor of the components. Below T_V , the spectral shape changed considerably, as was already ob-served earlier 12,23 Below T_V , the fit was made earlier by four components.²⁴ However, in most cases, the spectra were fitted by five components,^{12,25} resulting in the best result. Nevertheless, even more, six and nine components were supposed,^{26,27} but the exact reason for the existence of these components was not given. We fitted our spectra with five components below T_V . The parameters obtained are compiled in Table I. Below T_V , component a and component b_1 can be attributed to Fe³⁺ in tetrahedral and octahedral positions, respectively, and b_2 , b_3 , and b_4 to Fe²⁺ in octahedral position. This attribution is in agreement with those published.^{12,25} The appearance of the five components, some with different electric quadrupole splitting values, can be attributed to a complex monoclinic symmetry resulting in an electric field gradient at the cation sites in different sublattices. According to the refined model, the monoclinic cell has an orthorombic Pmca pseudosymmetry, constraining the atomic positions. Therefore, the lattice contains four iron atoms in subcells of the differently distorted octahedrons.²⁸ By fitting the spectra with five components below T_V , the hyperfine interaction parameter values measured by us are close to

TABLE I. Mössbauer parameters of bulk magnetite measured at different temperatures: isomer shift (δ), quadrupole splitting (QS) given in mm/s, internal magnetic field *H* given in T, and relative intensity (RI) values of the spectrum components given as percentages. Ions in (a) tetraheral and (b) octahedral positions.

Temperature	Component	δ	QS	Н	RI
295 K	а	0.27(3)	0.00(3)	49.0(1)	34(1)
	b	0.66(3)	0.00(3)	45.9(1)	66(1)
120 K	а	0.37(3)	0.00(3)	50.3(1)	36(1)
	b	0.77(3)	0.00(3)	47.9(1)	64(2)
115 K	а	0.38(3)	0.01(3)	50.4(1)	37(1)
	b1	0.47(3)	-0.10(3)	51.6(1)	11(1)
	b2	0.77(3)	-0.02(3)	50.5(1)	16(1)
	b3	0.96(3)	-0.3(3)	48.2(1)	27(1)
	b4	0.98(3)	1.88(3)	35.7(1)	9(1)
4.2 K	а	0.39(3)	0.01(3)	50.7(1)	33(1)
	b1	0.51(3)	-0.08(3)	52.3(1)	27(1)
	b2	0.81(3)	-0.04(3)	51.6(1)	9(1)
	b3	0.98(3)	-0.81(3)	48.8(1)	19(2)
	b4	1.02(3)	1.94(3)	35.9(1)	11(1)

those obtained earlier for bulk magnetite. According to Ref. 23, two different Fe^{3+} and four Fe^{2+} lattice positions exist in the lattice. Therefore, six Mössbauer spectral positions are expected to appear. However, only five components can be distinguished. The reason could be that two Fe^{2+} components have very close hyperfine parameters and they could not be resolved.

Four characteristic Mössbauer spectra of the 36 nm thick sample are shown in Fig. 2. The spectrum measured at room temperature is very similar to the spectrum of the bulk phase. The only difference is that the linewidths are slightly broader. The shape of the spectra gradually changes at lower temperatures and the $Fe^{3+}/Fe^{2.5+}$ intensity ratio begins to increase. The spectra from 150 K can be fitted by six compo-



FIG. 2. Mössbauer spectra of 36 nm sized magnetite.

TABLE II. Mössbauer parameters of 36 nm magnetite measured at different temperatures: isomer shift (δ), QS given in mm/s, *H* given in T, and RI values of the spectrum components given as percentages.

Temperature	Component	δ	QS	Н	RI
295 K	а	0.27(3)	0.00(3)	48.7(1)	33(1)
	b	0.65(3)	0.00(3)	45.6(1)	67(1)
140 K	а	0.34(3)	0.01(3)	50.2(1)	30(1)
	b1	0.45(3)	-0.02(3)	51.5(1)	10(1)
	b2	0.50(3)	-0.01(3)	50.7(1)	3(1)
	b3	0.90(3)	-0.58(3)	48.3(1)	6(1)
	b4	0.95(3)	1.85(3)	35.7(1)	0(1)
	b5	0.74(3)	0.10(3)	46.7(1)	51(1)
77 K	а	0.39(3)	0.02(3)	50.5(1)	32(1)
	b1	0.49(3)	-0.05(3)	51.9(1)	22(1)
	b2	0.60(3)	-0.02(3)	51.2(1)	7(1)
	b3	0.94(3)	-0.75(3)	48.6(1)	13(1)
	b4	1.00(3)	1.85(3)	35.8(1)	3(1)
	b5	0.84(3)	0.16(3)	47.1(1)	23(2)
4.2 K	а	0.42(3)	0.01(3)	50.7(1)	30(2)
	b1	0.50(3)	0.08(3)	52.3(1)	27(1)
	b2	0.68(3)	-0.04(3)	51.6(1)	11(1)
	b3	0.97(3)	-0.83(3)	48.8(1)	10(1)
	b4	1.02(3)	1.90(3)	35.9(1)	6(1)
	b5	0.85(3)	0.05(3)	47.4(1)	16(1)

nents. The fit values are included in Table II. The hyperfine parameters of the a and b_1 components are very close to the values of bulk magnetite measured below T_V . The parameter values of b3, b4, and b5 are also close to those obtained for the bulk sample. Hovewer, it was necessary to introduce a sixth sextet component b_2 to get the proper fit. This component was not purely Lorentzian, but it had a Voigt shape and the parameter values are close to those measured for Fe^{3+} . The change in the hyperfine parameters and relative intensities took place surprisingly in a broad temperature range, not as they were observed for the bulk sample. The maximum Fe³⁺/Fe²⁺ values measured at 4.2 K indicate no oxidation effect. The fitted values are deployed in Table II. The $Fe^{3+}/(Fe^{2+}+Fe^{3+})$ ratios are shown in Fig. 4. The Mössbauer spectra of the 9 nm thick sample are shown in Fig. 3. The fitted values are listed in Table III. It is clearly seen that the intensity of the Fe^{2.5+} component decreases already at 210 K. The hyperfine split values taking into account the different temperature values of the measurements are near those measured in the other samples. The $Fe^{3+}/(Fe^{2+}+Fe^{3+})$ intensity ratios are shown in Fig. 4. The phase transition begins at a higher temperature than in the bulk and the 36 nm sized sample. In addition, the transition to the monoclinic phase is complete at lower temperature as in bulk crystals. In the case of the 9 nm sized sample, the transition is more intensive at higher temperature values, as shown in Fig. 4.

It is known that in nonstoichiometric magnetite samples (e.g., containing γ -Fe₂O₃ in various concentration), T_V is reduced from 120 K and a decrease in resistivity with increasing nonstoichiometry was observed.²⁹ For Fe_{3- δ}O₄, the critical value δ =0.0039 was determined.³⁰ Above this value, T_V shifts downward. The measured intensity ratios of the spec-



FIG. 3. Mössbauer spectra of 9 nm sized magnetite.

tral components in our samples exclude nonstoichiometry above this δ value. Recently, a mechanism was proposed for the formation of charge ordered surface states, providing an explanation for the increase in the phase transition temperature at the magnetic surface.³¹ Earlier, Anderson³² calculated much higher Verwey temperatures, taking into account the long and short range charge ordering in magnetite. For our 9 and 36 nm sized samples, the crystal structure is cubic at room temperature and the components of the Mössbauer spectra have the same isomer shift values as the sextet components of the bulk sample. This indicates that the samples

TABLE III. Mössbauer parameters of 9 nm magnetite measured at different temperatures: isomer shift (δ), QS given in mm/s, *H* given in T, and RI values of the spectrum components given as percentages.

Temperature	Component	δ	QS	Н	RI
295 K	а	0.29(3)	0.00(3)	47.1(1)	33(1)
	b	0.61(3)	0.00(3)	44.6(1)	67(1)
210 K	а	0.32(3)	-0.03(3)	48.3(1)	23(1)
	b1	0.41(3)	-0.01(3)	48.2(1)	11(1)
	b2	0.52(3)	-0.08(3)	42.7(1)	6(1)
	b3	0.87(3)	-0.49(3)	48.1(1)	6(1)
	b4	0.92(3)	1.83(3)	35.5(1)	0(1)
	b5	0.67(3)	0.06(3)	46.3(1)	54(2)
77 K	а	0.38(3)	0.01(3)	50.3(2)	33(1)
	b1	0.47(3)	-0.03(3)	51.8(1)	23(1)
	b2	0.62(3)	-0.02(3)	51.1(1)	9(1)
	b3	0.95(3)	-0.70(3)	48.5(1)	12(1)
	b4	1.02(3)	1.87(3)	35.6(1)	2(1)
	b5	0.85(3)	0.13(3)	47.0(1)	21(1)
4.2 K	а	0.41(3)	-0.01(3)	50.5(1)	29(1)
	b1	0.49(3)	0.09(3)	52.1(1)	26(2)
	b2	0.68(3)	0.01(3)	51.5(1)	12(2)
	b3	0.96(3)	-0.90(3)	48.6(2)	10(1)
	b4	1.00(3)	1.95(3)	35.7(1)	5(1)
	b5	0.86(3)	0.01(3)	47.3(1)	18(1)



FIG. 4. The temperature dependence of the relative content ratio of Fe^{3+} in the magnetite samples.

are definitely in cubic phase. By cooling down the nanosized samples, spectral change began showing hyperfine interaction parameters similar to those appearing below T_V in the bulk sample. This change suggests that fractions of the nanosized particles have higher phase transition temperatures than the bulk crystals. It means that the cubic \rightarrow monoclinic phase transition occurs at higher temperature. What is the explanation for this effect? In Ref. 7, it was shown that particle aggregation resulted in magnetic composites. Therefore, the neighboring particles behave in a different way as compared to separated clusters. Indeed, the Mössbauer spectrum of the magnetic dendritelike particle was very similar to our 9 nm sized sample at room temperature, and transition temperatures were considerably different above T_V . Nevertheless, our fit resulted in those components appearing in the spectra of bulk magnetite. These results show that dendritelike self-assemblies can form already during precipitation in aqueous media. The phase transition of the dendrites takes place in a broad temperature range. The magnetic interaction of the particles and the different specific surface may also shift the temperature of the cubic-monoclinic transition and, finally, the phase transition takes place in broad temperature ranges.

IV. CONCLUSIONS

The Mössbauer spectra of 9 and 36 nm sized nanomagnetite and bulk magnetite samples were measured in the temperature range between 4.2 and 300 K. For bulk magnetite, the Verwey transition temperature and the temperature dependence of the hyperfine split parameters were the same as observed earlier. Similar hyperfine interaction parameters were found for 9 and 36 nm sized nanomagnetite samples. We observed first that the temperature dependence of the intensities of spectral components indicated a continuous phase transition taking place in a very broad temperature range. The continuous transition can be explained by the formation of fractal-like granular assemblies formed during the preparation of the samples. The hyperfine interaction values measured in these granules are close to the values measured in bulk crystals. Nevertheless, the interaction of the assemblies influences the cubic-monoclinic transition temperature. Formally, the monoclinic-cubic phase transition temperature increases for smaller particles. The selforganization of magnetic nanostructures is an interesting subject and may find potential applications in modern materials science.

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