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TlCu_{1.73}Fe_{0.27}Se₂ Studied by means of Mössbauer Spectroscopy and SQUID Magnetometry

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Abstract. TlCu_{2-x}Fe_xSe₂ is a p-type metal for x < 0.5 which crystallizes in a bodycentered tetragonal structure. The metal atoms are situated in ab-planes, ~ 7 Å apart, while the metal - metal distance within the plane is ~ 2.75 Å. Due to the large difference in cation distances, the solid solutions show magnetic properties of mainly two-dimensional character. The SQUID measurements performed for x = 0.27 give the c-axis as the easy axis of magnetisation, but show also clear hysteresis effects at 10 K, indicating a partly ferromagnetic coupling. The magnetic ordering temperature T_c is 55(5) K as found from both SQUID and Mössbauer spectra. At $T \ll T_c$ the magnetic hyperfine fields are distributed with a maximum at about 30 T, which is compared to the measured magnetic moments per iron atom which is 0.97 μ_B /Fe as found from SQUID measurements. The experimental results are compared to results using other methods on isostructural Tl-selenides.

1. Introduction

As has been shown previously, the ternary copper chalcogenides $TlCu_2X_2$ (X= S, Se, Te) are all metallic conductors [1, 2, 3]. The reason, as suggested from various transport measurements in the sulphur and selenium cases [2, 3], is the presence of an electron hole per formula unit at the top of the valence band (VB) due to Se-Se bonding and the fact that copper is monovalent. Consequently, these compounds show only temperature independent Pauli paramagnetism (PM). The electron band structure was investigated indirectly by XPS [4] and theoretical calculations [5, 6], confirming this view.

Substitution on the transition metal site offers the possibility to change the physical properties. By introducing other metals (Me) such as gallium or iron for copper, a semiconductor is obtained at the composition TlCu_{1.5}Me_{0.5}Se₂, suggesting that two extra electrons (from Me^{3+} instead of Cu^+) are simultaneously donated, eventually filling up the valence band [1, 7]. Iron carries at the same time a magnetic moment from its 3d electrons, and interesting magnetic properties are introduced, extensively probed by a battery of experimental methods [7]. Thus, Berger and van Bruggen were able to show how the magnetic properties vary with the degree of substitution in $TlCu_{2-x}Fe_xSe_2$. The paramagnetic iron moments may order below a critical temperature: mainly ferromagnetic (FM) ordering occurs up to $x \sim 0.4$, but near the critical total band-filling value of x = 0.5, the solid solution shows a ferrimagnetic character, since the asymptotic paramagnetic temperature suddenly gets negative. In the metallic range, 0 < x < 0.5, the Curie temperature of these phases showed a maximum near 80 K for $x \sim 0.4$. At the same time, the variation of magnetic moment, as inferred from saturation measurements, had a maximum near 3 μ_B/Fe (i.e. more than in metallic iron) for x~0.3. These effects were interpreted as due to polarisation of the conducting electrons (RKKY type) creating ferromagnetic ordering for low x-values. The increased probability for indirect Fe-Fe coupling for larger x-values makes antiferromagnetic interactions come into play (superexchange) which lower the net moment.

Recently, renewed magnetic measurements on such solutions [8] very nicely confirmed the trend in magnetic moment values, now from single-crystal material at x = 0.16 and x = 0.45 that gave the extra information that the moments are oriented along the c-axis of the tetragonal cell. The parent compound is of the ThCr₂Si₂ structure type (figure 1) where the transition metal atoms are confined to sheets in the ab-plane, with a large sheet separation (roughly 7 Å). Thus, the iron atoms mainly interact within the sheets where they occur with a square sub-lattice arrangement.

In order to probe the iron interactions in this kind of structure, ⁵⁷Fe Mössbauer spectroscopy has been addressed previously mainly for cases with larger iron contents, as TlCuFeSe₂ [7] or TlFe_{1.7}Se₂ [9], the latter with metal vacancies. Within the framework of a renewed interest in two-dimensional magnetism it was decided to try this method anew, now on $x\sim 0.3$, a concentration where a large moment was found for the solid solution TlCu_{2-x}Fe_xSe₂ [7].



Figure 1. The tetragonal crystal structure of $\text{TlCu}_{2-x}\text{Fe}_x\text{Se}_2$, where A and X stand for Tl and Se, respectively, and T for Cu or Fe. Two unit-cells are shown.

2. Experimental

Synthesis and phase characterization

Crystals of nominal composition x = 0.3 (with an ⁵⁷Fe abundance of 33%) were synthesized by mixing stoichiometric amounts of TlSe (presynthesized), Se, Fe and Cu and heating in an evacuated silica tube, followed by controlled slow cooling to allow for nucleation and growth of large crystals. Highly textured crystal material could be obtained in this manner, and the cell parameters were determined from powdered material by x-ray powder diffraction, using a Guinier-Hägg camera equipped with strictly monochromatic CuK α_1 radiation. Germanium was used for calibration. The cell axes were found to be a = 3.9175(3) Å and c = 13.864(2) Å which, according to the Vegard's law calibration curve [7], translates into x = 0.27.

Magnetometry

For the measurements by a SQUID (superconducting quantum interference device) magnetometer (MPMS XL), single crystal material with a mass of 23.5 mg was stacked with a common < 001 > orientation (normal to the flat surface habitus). Thereby it was possible to get data from two principal orientations, either perpendicular to the c-axis or parallel to it, important for the analysis when remembering that the spin-carrying species (iron) is strictly confined to the ab-plane.

The magnetic behaviour as a function of temperature was probed in two manners, recording data on heating, either after applying the field after cooling (zero field cooling, zfc), or applying the field already during the cooling step (field cooling, fc). In both cases, the applied field was 100 Oe. The field dependence of the magnetization was

investigated on changing the field while keeping the sample at a constant temperature.

Mössbauer spectroscopy

The absorber was manufactured by grinding the crystals to a fine powder which was then mixed with BN and finally gently pressed into a ring of copper. The absorber thickness was 0.74 mg Fe/cm^2 . Al-foils were glued on both sides of the ring to keep the absorber intact, being mounted in a He-flow cryostat for low temperature measurements. The transmission spectrometer was of constant acceleration type using 512 channels for storing data. Two spectra, in forward and backward directions of the same source (Co<u>Rh</u> at room temperature) were recorded simultaneously. The spectra recorded in the backward direction were used for calibration with an iron foil at room temperature as absorber. The spectra were folded and analysed using the Mössbauer fitting program "Recoil" [10].

3. Results

3.1. Magnetometry results

The temperature dependence of the magnetisation (expressed as M/H) for a constant field of 100 Oe is given in figure 2. There is a pronounced difference between the two orientations, the M/H value being around 100 times higher with the applied field parallel with the c-axis compared to perpendicular to it. Both orientations indicate, as best determined from the inflexion of the fc branch, an ordering temperature of about 55 K.



Figure 2. The magnetisation versus temperature per applied field and mass in cases that the applied field are in the c-direction (left panel) and in the ab-plane (right panel). Y-scale is very different in the two cases. This indicates that the easy axis is the c-axis.

The magnetisation data are very different for the two orientations (See figure 3). Considering that the crystal material was not exactly single-crystalline, small deviations from complete alignment are expected. However, a complete saturation is not achieved even for large applied fields. Therefore, a small paramagnetic component is likely present in the hysteresis measurements. The data from the ab-plane show correspondingly mainly pure paramagnetic behaviour, only a little marred by a small part of ferromagnetic behaviour. Therefore, the saturation moment was determined in two steps: Firstly, M_{sat} (T) was taken as the intercept on the M-axis for H = 0 from a linear extrapolation using high H-values, realising the symmetry around the origin. Secondly, the M_{sat} (T))-values for T = 10 K and 30 K were extrapolated to T = 0 K. By this procedure, $M_{sat}(0)=3.2$ emu/g was found. From this, and x = 0.27, we obtain an average iron moment of 0.97 μ_B/Fe .



Figure 3. The magnetization as function of applied field, in the c-direction (left panel) and in the ab-plane (right panel). Different scales in y-axis are used.

3.2. Mössbauer spectroscopy and fittings

The Mössbauer spectrum recorded at room temperature (see figure 4) is a symmetrical broad singlet. However, the full width at half maximum, Γ_{exp} is $\approx 0.56(2)$ mm/s, so the Fe-atoms do react to hyperfine interactions. The very symmetrical profile indicates approximately the same centre shifts (CS) for all iron atoms, but a smooth distribution in quadrupole splittings (Δ). CS is defined as the centre of the resonance lines and connected to the isomer shift (δ) and second Doppler shift (SOD) via CS = δ + SOD. Using a two doublet fit, we obtain CS = 0.40(1) mm/s with | Δ |= 0.13(1) mm/s and 0.38(1) mm/s, with Γ_{exp} = 0.28(1) mm/s and 0.35(1) mm/s, respectively. The average value found for | Δ |= 0.27(1) mm/s.

The spectra recorded between 200 K and 50 K show similar bell shaped profiles. The average quadrupole splitting increases with decreasing temperature to 0.38(1) mm/s at 50 K. The intensities for the two doublets do not change significantly when decreasing the temperature and are on the average 59(5)% and 41(5)% for the low $|\Delta|$ and high $|\Delta|$ value components, respectively. The Γ_{exp} for the profile is plotted in figure 5. The extra broadening starts around 58 K and becomes very significant below 50 K. The spectrum at 30 K shows resolved lines but with obviously very broad distribution



Figure 4. The Mössbauer spectra at 300 K down to 40 K. The absorbances were about 3 percent.

in magnetic hyperfine fields. The distribution sharpens up when going to 20 K and is rather narrow at 10 K (figure 6). Assuming a random occupation of the Fe and Cu atoms in the 2-dimensional network results in certain coordination probabilities for the Fe atom in this Cu-Fe plane. The nearest neighbour (nn) distance is 2.77 Å (4 atoms) and the next nearest neighbour (nnn) distance is 3.92 Å (4 atoms), while the nnnn and nnnnn distances are 5.55 Å (4 atoms) and 6.19 Å (8 atoms). Here it may be emphasized that the distance between the Cu-Fe -planes are 6.93 Å in the present case. Restricting to nn and nnn coordination shells give the occupation probabilities presented in table 1 and figure 7.



Figure 5. The Γ_{exp} of the Mössbauer resonance spectra as function of temperature. It shows that the transition temperature is about 55 K as marked by the arrow.

Table 1. Coordination probabilities for an Fe atom in the ab-plane network assuming random occupation of Cu and Fe atoms. Here (x,y) stands for x Fe atoms as nn and y Fe atoms as nnn. The probabilities have been merged into 5 coordinations in row b) and into 3 in row c).

Coordination (x,y)	(0,0)	(0,1)	(0,2)	(1,0)	(1,1)	(1,2)	(2,0)	(2,1)	(3,0)
a) Probabilities	31%	20%	5%	20%	12%	3%	5%	3%	1%
b) Probabilities	31%	25%		20%	15%		9%		
c) Probabilities	31%	25	5%	44%					

The spectra, recorded at 10 K and 20 K were possible to fit with reasonable good result using only three subpatterns with intensities locked to table 1, row c. The centre shifts were restricted to be same for all subpatterns as also the Γ_{exp} . The magnetic hyperfine field B_{hf} and its gaussian distribution σ and the electric quadrupole shift parameter ϵ were however free to vary for each subpattern in the fitting. The spectra and the fittings are shown in figure 6 and the obtained parameter values are given in table 2. The spectrum at 30 K was also fitted by the same model although here the hyperfine field distributions became very large (table 2).



Figure 6. The Mössbauer experimental and fitted spectra for low temperatures, 30 K, 20 K and 10 K. The absorbances were about 1 percent.

Table 2. The magnetic hyperfine fields B_{hf} and their gaussian distributions σ , centroid shifts CS, quadrupole shifts ϵ of the different spectral components at low temperatures. Intensities were constrained to 44%, 25% and 31% for the spectral components 1, 2 and 3 respectively. Estimated errors in B_{hf} and σ are ± 0.5 T, in CS ± 0.01 mm/s and in $2\epsilon\pm0.05$ mm/s. The individual Lorentzian linewidths were the same for all lines and equal to 0.29 mm/s.

Temp.	CS	B_{hf1}	σ_1	$2\epsilon_1$	B_{hf2}	σ_2	$2\epsilon_2$	B_{hf3}	σ_3	$2\epsilon_3$
(K)	(mm/s)	(T)	(T)	(mm/s)	(T)	(T)	(mm/s)	(T)	(T)	(mm/s)
30	0.58	23.2	5.4	-0.12	15.9	4.8	0.02	6.7	4.4	0.05
20	0.58	27.7	3.3	-0.14	22.6	4.2	0.16	15.0	7,4	-0.16
10	0.57	30.1	1.8	-0.14	28.1	2.8	0.25	22.3	6.7	-0.21

4. Discussions

Felton *et al.* [8], using SQUID-techniques, measured a saturation moment of 1.5 μ_B /Fe for x = 0.16 and 0.66 μ_B /Fe for x = 0.45 in TlCu_{1-x}Fe_xSe₂. We measure here, using

the same technique, 0.97 μ_B/Fe for x = 0.27. A linear interpolation, using Felton's values gives 1.18 μ_B/Fe for x = 0.27, thus in reasonable agreement with our result and the VB-model presented in [7]. Felton *et al.* found $T_c = 65$ K for x = 0.16 and two transitions at about 70 K and 130 K (weak) for x = 0.45. We got here $T_c = 55(5)$ K for x = 0.27 and no tendency to transition above 100 K. Our results then agree with Felton's lower transition temperature. The mentioned transition at 130 K may come from the Verwey transition in magnetite, being an impurity too minor to be detected by XRD. Berger and van Bruggen also got minor amounts of magnetite in their samples [7]. Furthermore, the easy axis of magnetisation is found to be parallel with the c-axis for x = 0.27, as earlier also found for x = 0.16 and 0.45 [8]. The magnetic coupling then seems to be of the same type in the whole region x < 0.5, where $TlCu_{2-x}Fe_xSe_2$ is a p-type metal. Also the isostructural non-stoichiometric phase $TlFe_{2-y}Se_2$, with $y \approx 0.3$ - 0.4, does have the easy magnetisation in the c-axis direction [9, 11] while the magnetic moments form a helix in the ab-plane in $TlCo_2Se_2[12]$.



Figure 7. Different Fe coordination possibilities viewed in projection of the Cu-Fe quadratic ab-plane along the crystal [001] - direction. Filled and unfilled circles indicate Fe and Cu atoms, respectively while halffilled circles indicate that the position is occupied either by a Cu or an Fe atom. The coordinations depicted are from left to right: (x,y) = (0, 0), (0, 1-2), (1, 0-2) and (2, 0-1), where (x,y) stands for x Fe atoms as nn and y Fe atoms as nnn. The coordination probabilities are from left to right 31%, 25%, 35% and 9%. The Fe magnetic moment fluctuations are schematically drawn in accordance with the findings from the fitting of the Mössbauer spectra (see below). The fluctuation frequencies decrease for increasing number of Fe as nn and nnn.

The centre shift increases with decreasing temperature (figure 8). However, the high temperature linear slope is $-6*10^{-4}$ mm/(sK), thus somewhat less steep than the second order dopplershift (SOD) high temperature variation, being $-7.3*10^{-4}$ mm/(sK). One cause for the difference could be the volume contraction at lowering temperatures, squeezing more Fe s-electrons into the nuclear volume. However, if it had been mainly Fe 3d-electrons which had been compressed, then the CS should have increased instead at lowering temperatures as a result of shielding effects. In TlFe_{1.7}Se₂ the slope is $-6.7*10^{-4}$ mm/(sK) in the temperature interval 100 K - 458 K [11] and $-7.3*10^{-4}$ mm/(sK) for TlFe_{1.6}Se₂ in the interval 100 K - 438 K as an average value for three Mössbauer patterns [9]. As these isostructural Tl selenides show normal

CS(T)-values, we assume that the deviation from normal SOD-shift behaviour for $TlCu_{1.73}Fe_{0.27}Se_2$ is not due to volume contraction at lowering temperatures. It might be due to a temperature dependent transfer of electrons from Fe to Se instead.



Figure 8. The centre shift (CS) as a function of temperature. For comparison the SOD function is also included.

The centre shift obtained for TlCu_{1.73}Fe_{0.27}Se₂ at room temperature, 0.40 mm/s, is quite low compared to 0.57 mm/s as obtained for TlCuFeSe₂ [13]. The result may be understood using the earlier mentioned VB-model [7]: the Se 4p-band is filled in TlCu_{2-x}Fe_xSe₂ for x = 0.5. For higher x-values there is a back-donation of electrons to iron, resulting in an increased CS (increased ferrous character, probably complete at x = 1). The CS at room temperature in other related structures are: 0.51 mm/s (average value) in TlFe_{1.6}Se₂ [9], 0.55 mm/s in TlFe_{1.7}Se₂ [11] and 0.49 mm/s in TlCo₂Se₂ doped with 2% ⁵⁷Fe [12]. The result for the two former examples is in line with the model: both should have filled Se 4p VB and the ferrous character should be higher in the "Fe_{1.7}" compared to the "Fe_{1.6}"-case, while the situation in the Co-case is unclear. Unfortunately, the full solid solution series TlCu_{2-x}Fe_xSe₂ has not been studied by Mössbauer spectroscopy as a function of x, due to experimental difficulties (high atomic absorption of the 14.4 keV γ -radiation by Tl and Se, making doping by ⁵⁷Fe necessary). A random distribution of iron atoms over the metal sites gives the Fe coordination probabilities presented in table 1. Fe will be surrounded by 4 Cu atoms as nn in 56% of the cases and by 3 Cu + 1 Fe in 35%. Other configurations (2 and more Fe in nearest metal shell) will only cover 9% of the cases. Assuming a more regular Fe surrounding having the lower $|\Delta|$, we can correlate "4 Cu"-surroundings to the smaller $|\Delta|$ -values, while "Fe-neighbours" give higher $|\Delta|$ -value. Other atoms as nnn and nnnn etc. also influence $|\Delta|$, resulting in the smooth $|\Delta|$ -distribution. Such ascribements do not contradict the experimental intensity values of 59(5)% and 41(5)%, respectively. The low values of $|\Delta|$ in this case are reasonable since the CuSe₄-tetrahedron is elongated in the c-direction in TlCu₂Se₂, but close to regular in TlCu_{1.5}Fe_{0.5}Se₂ [7].

The measured room temperature electric quadrupole splitting, $|\Delta| = 0.81$ mm/s in TlCuFeSe₂ [7], is higher than obtained here for TlCu_{1.73}Fe_{0.27}Se₂ (average value 0.27 mm/s). This is reasonable as the ferrous character is higher and the Me-Se₄ tetrahedron is less regular in the former [7], both factors favour a higher $|\Delta|$.

The low temperature Mössbauer results indicate that the sharpening of the spectra when going from 30 K to 10 K is a result of the local influence on the magnetic hyperfine fields. At 10 K the fields for components 1 and 2 (table 1) are rather close being 30.1(5) T and 28.1(5) T. As assigned component 1 is emanating from Fe atoms having 3, 2 or 1 Fe atoms as nn, while component 2 is emanating from Fe atoms with 0 Fe as nn but with 1 or 2 Fe atoms as nnn. Component 3, with a field at 10 K of 22.3(5) T and a broad distribution of 6.7 T is emanating from Fe atoms with no Fe atoms as nn or nnn but with one or more Fe atoms in the third or higher coordination shells. These long Fe-Fe distances result in weak magnetic couplings. Consequently the Fe magnetic moments fluctuate with the result of a lower measured magnetic hyperfine field [14] and a broader distribution. This explanation is further supported by the result for the 20 K and 30 K spectra where the decreased measured fields and even broader gaussian field distributions are due to increased magnetic moments fluctuation frequencies. The fact that the two spectral components 1 and 2 are having narrow spread in the magnetic hyperfine fields at 10 K means that the Fe moment relaxations times for these coordinations are longer than the Mössbauer time window. Since the observed magnetic hyperfine fields are slightly different (7%) for these two components we can conclude that it is possible to correlate not only the hyperfine parameter $|\Delta|$ (shown above from the analysis of room temperature spectrum) but also the hyperfine parameter B_{hf} to the local Fe configurations.

The obtained high-field component at low temperature (≈ 30 T) is somewhat higher than in non-stoichiometric TlFe_{1.7}Se₂ (B_{hf} =27.2 T at 100 K) [11] or TlFe_{1.6}Se₂ (B_{hf} = 27.0 T at 100 K [9]) and considerably higher than in TlCo₂Se₂ (B_{hf} $\leq 12 T$ in a helical magnetic structure [12]).

5. Conclusion

Mössbauer spectroscopy and SQUID measurements have determined TlCu_{1.73}Fe_{0.27}Se₂ to be partly ferromagnetic with the c-axis as the easy axis of magnetisation and a T_c

of 55(5) K. The saturation moment found, 0.97 μ_B , is lower than earlier reported for this compound. Furthermore the local probe technique has been able to correlate the local Fe configurations to specific strength of hyperfine interaction parameters. The maximum saturation hyperfine field is high, 30 T, for this rather Fe diluted compound, higher than the isostructural non-stoichiometric TlFe_{1.7}Se₂. Further Mössbauer studies of other Cu/Fe ratios are needed in order to reveal the full picture of the system.

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